

A Highly Elastic, Capacitive Strain Gauge Based on Percolating 2 Nanotube Networks

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S Supporting Information 6

ABSTRACT: We present a highly elastic strain gauge based on capacitive sensing 7 8 of parallel, carbon nanotube-based percolation electrodes separated by a dielectric 9 elastomer. The fabrication, relying on vacuum filtration of single-walled carbon nanotubes and hydrophobic patterning of silicone, is both rapid and inexpensive. 10 We demonstrate reliable, linear performance over thousands of cycles at up to 11100% strain with less than 3% variability and the highest reported gauge factor for a 12 device of this class (0.99). We further demonstrate use of this sensor in a robotics 13 context to transduce joint angles. 14

KEYWORDS: Nanotubes, percolation, sensor, strain, elastomer 15

onventional, metal-foil strain gauges are limited to 16 applications involving both relatively stiff substrates and 17 18 strains under 5% (above which the gauge mechanically fails).^{1,2} 19 As such, they are particularly ill-suited for use in the low-20 stiffness, high-strain environments that characterize many 3-821 medical applications and, increasingly, soft robotics systems.³ Recently, many efforts to address this problem have focused on 22 23 carbon nanotube-based percolation networks that are variously ²⁴ coupled to elastomeric matrices.^{9–14} Percolating networks 25 consist of a collection of conductive particles in two- or 26 three-dimensions that overlap with each other to create a 27 conductive network. Such a network is able to maintain 28 conductivity even when deformed due to continual contact 29 among the conductive particles. The critical concentration, or 30 percolation threshold, required to achieve stable percolation is a 31 direct function of the geometry of these particles and scales 32 with aspect ratio.¹⁵ Carbon nanotubes are particularly well 33 suited for use in percolation networks as they have both a high 34 intrinsic conductivity and aspect ratios as high as 1:1000.

The most common approach to percolation sensors relies on 35 ₃₆ piezoresistance (strain induces a change in the resistance of the 37 network). A variety of techniques have been developed for 38 producing these networks using carbon nanotubes. Some of the 39 earliest approaches relied on carefully dispersing nanotubes in 40 unpolymerized polymer before curing the ensemble to produce 41 a conductive composite.^{7,16} While quite effective for flexible 42 electronics, this approach results in significant reinforcement of 43 the elastic modulus, thereby making the material to stiff for 44 stretchable applications.¹⁶ More recent techniques involve 45 infiltration of an elastomer into a vertically aligned array of 46 nanotubes grown on a wafer, direct contact transfer of wafer-47 grown or vacuum filtered nanotubes microcontact stamping, 48 airbrushing of dispersed nanotubes onto a substrate, and 49 manual assembly of sheets of aligned nanotubes onto the 50 surface of an elastomeric support.^{2,17-20} While effective,

piezoresistive designs can be difficult to tune as they inherently 51 rely on the mechanical stability of the percolation network and 52 are susceptible to hysteresis and variable gauge factors (the 53 normalized change in resistance divided by the applied strain) 54 as the network adjusts over time. 55

An alternative approach relies on the piezocapacitance of 56 dielectric elastomers sandwiched by percolation electrodes. 57 Here, any deformation that brings the electrodes closer 58 together results in an increase in capacitance, and vice versa. 59 While these devices are most often used as low-strain pressure 60 sensors that transduce forces normal to the sensor surface, they 61 can also be used to transduce planar, tensile strains (Figure 62 fl 1a).⁸ To date, such planar strain piezocapacitors have been used 63 f1 to transduce strains up to 30% by using either nanotubes or 64 conductive polymers.²⁰⁻²² While we also opted to use 65 nanotube percolation electrodes, our design was optimized 66 for cyclic, planar strains up to 100% (Figure 1b-c). 67

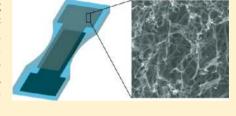
The general mechanism by which such a device couples 68 planar strain to a change in capacitance relies on Poisson 69 contraction. Uniaxial, planar strain results in a Poisson's ratio-70 mediated contraction of the orthogonal axes that brings the two 71 electrodes closer together resulting in a corresponding increase 72 in capacitance. The simplified, linear mechanics of this process 73 (Figure 1a) are described by eqs 1-374

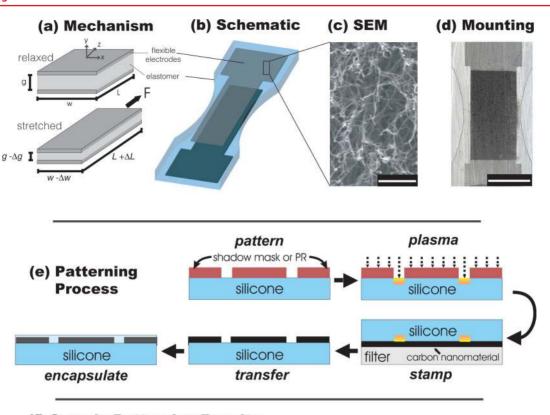
$$\frac{\Delta L}{L} = \varepsilon_z = \frac{\sigma_z}{E_{\text{silicone}}} \tag{1}$$

$$\varepsilon_x = \varepsilon_y = \nu \varepsilon_z$$
 (2)

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(f) Sample Patterning Results

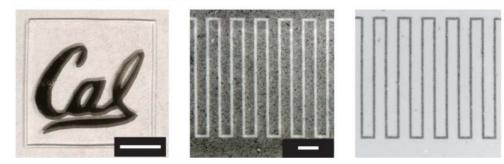


Figure 1. Mechanism, design, and fabrication of a Poisson Capacitor. (a) Poisson contraction converts planar strain to a decrease in the gap distance between the percolating electrodes and an increase in capacitance. (b) Schematic of our device geometry; all sensing is carried out in the middle region of the sensor. (c) SEM data demonstrating percolation of the nanotubes within the electrode; scale bar is 500 nm. (d) Close-up image of the sensing region of the device. Darker region is due to overlap of electrodes, and striations are due to texture of the background; scale bar is 0.75 cm. (e) Process flow for preparing a sensor. (f) Results from patterning. (left) Produced using atmospheric plasma and paper shadow-mask, fringing due to shadows; scale bar is 5 mm. (center) RIE oxygen plasma and photoresist was used to produce 300 μ m wide digits; scale bar is 750 μ m. (right) Nanotubes left on the filter after previous transfer, line width is 100 μ m; same scale as at left.

$$L_{\text{stretch}} = L + \Delta L = L + \varepsilon_z L$$

$$w_{\text{stretch}} = w - \Delta w = w - \nu \varepsilon_z w$$

$$g_{\text{stretch}} = g - \Delta g = g - \nu \varepsilon_z g$$
(3)

75 Here, ν is Poisson's ratio (~0.5 for silicone elastomers); ε is 76 strain; σ is stress; *E* is the elastic modulus of silicones; and *L*, *w*, 77 and *g* are the initial dimensions of the capacitor. Equation 1 is 78 Hooke's law relating modulus, stress, and strain. While models 79 taken from nonlinear elasticity theory are generally used for 80 large deformations, the simple linear model described here has 81 an equivalent goodness-of-fit ($R^2 = 0.997$) to nonlinear models 82 when fitted to mechanical testing data taken from our sample 83 devices (Figure S1 and Supporting Information).²³

84 Equations 2 and 3 describe how Poisson's ratio couples strain 85 applied in one axis to opposite strains induced in the other two axes. If we relate this to the simplest model for parallel plate ⁸⁶ capacitance, we arrive at eq 4 ⁸⁷

$$\Delta C_{\text{simple}} = e_0 e_{\text{silicone}} \left(\frac{w_{\text{stretch}} L_{\text{stretch}}}{g_{\text{stretch}}} \right)$$
$$= e_0 e_{\text{silicone}} \left(\frac{w(1 - \upsilon \varepsilon_z) L(1 + \upsilon \varepsilon_z)}{g(1 - \upsilon \varepsilon_z)} \right)$$
$$= e_0 e_{\text{silicone}} (1 + \varepsilon_z) \frac{wL}{g}$$
(4)

Here, e_0 is the permittivity of free space, and e_{silicone} is the 88 relative permittivity of silicone. This simple electromechanical 89 equation directly relates the change in capacitance of the 90 parallel plate geometry to the applied strain and the initial 91

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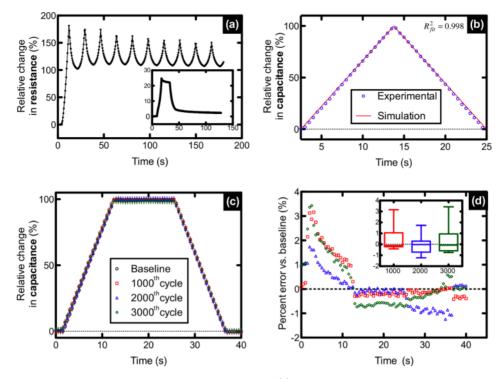


Figure 2. Performance characterization of Poisson capacitor strain gauge. (a) Resistive performance of a single percolation electrode when undergoing 100% cyclic strain. Main plot indicates significant hysteresis, while inset shows both relaxation during a 100% step strain and subsequent hysteresis during recovery. (b) Single, 100% strain cycle of the capacitive sensor (blue) overlaid (red) by the fit from parallel plate model with Palmer correction (red). (c) Demonstration of repeatability over 3000 cycles of 100% strain. After every 1000 cycles, the sample was subjected to a 100% step strain before being relaxed back to basal strain. (d) Stability plot showing how much sensor performance deviated from baseline performance over 3000 cycles. Legend is the same as for (c).

92 geometry of the sensor. It is interesting to note that the final 93 equation, despite resulting from Poisson contraction, does not 94 depend on the value of Poisson's ratio as long as the material is 95 assumed to be isotropic. The equation further predicts a linear 96 output, assuming that the overlap geometry remains stable. This implies that, regardless of resistive hysteresis (i.e., 97 piezoresistive effects, changes in the percolation network 98 99 nanostructure), the sensor performance remains stable as 100 long as the effective overlap area between percolation electrodes remains stable. While this model describes the 101 102 general mechanics and agrees with that used by Loh et al. to describe a strain gauge based on a stiff polymer layer with 103 104 nanotube/gold particle electrodes strained at 1%, it does not take into account the fringing fields normally obtained with 105 finite-size parallel plate capacitors. To take these into account, 106 we used the Palmer correction factor, which is a multiplier that 107 acts on all terms in the simple model, for all of simulations (eq 108 109 S1 and Supporting Information).²⁴

Our design, shown in mounted form in Figure 1d, relies on 110 111 producing carbon nanotube percolation electrodes that are stable over large deformations and can be precisely patterned 112 nto the elastomeric substrate such their shape and orientation 113 can be specified independently of those of the elastomer. We 114 115 chose silicone as the substrate both due its excellent elastomeric properties and its ability to facilitate patterning on both sides of 116 sheet via the hydrophobic to hydrophilic transition that 117 a results from atmospheric (or oxygen) plasma treatment.²⁵ As 118 119 raw single-walled carbon nanotubes (SWNT) are highly 120 hydrophobic, they will preferentially adhere to the hydrophobic 121 (untreated) regions of the patterned silicone (Figure 1f).

Our fabrication process is summarized in Figure 1e. First, 122 275 μ m thick silicone substrates were laser-cut into the test 123 structures (Figure 1b). The pattern resolution necessary for the 124 sensor electrodes was achievable using laser-cut sticker-paper 125 masks that were applied to both sides of the silicone substrates. 126 The masked substrate was positioned on its side in an 127 atmospheric plasma cleaner to allow the plasma access to both 128 sides of the silicone to render the unmasked regions 129 hydrophilic. Upon removal from the chamber, the masks 130 were removed and the substrate was ready to receive 131 nanotubes. The percolation networks were produced using 132 SWNT that were dispersed in a surfactant solution via 133 sonication and subsequently vacuum filtered and collected on 134 a 20 nm pore-size filtration membrane.^{26,27} The result of this 135 process is a percolation network resting on top of a filtration 136 membrane. By bringing both sides of the plasma-treated 137 silicone into contact with these percolation networks, we 138 achieved direct nanotube transfer to the untreated regions of 139 the silicone and produced the overlapping electrodes necessary 140 for a parallel plate capacitor. Should submillimeter features be 141 required, the shadow mask can be replaced with a photoresist 142 layer that can be removed via acetone immersion after plasma 143 treatment. Between these two masking techniques, we have 144 successfully demonstrated patterns ranging from centimeters in 145 size to 100 μ m wide serpentines (Figure 1f). 146

As a final step to protect and stabilize the newly transferred 147 nanotube layers, we used an airbrush to coat the substrates with 148 thin layers of silicone while avoiding spraying the ends of the 149 device (the contact pads).²⁸ All of our tested sensors were 150 designed with a 1.7 cm \times 0.75 cm electrode overlap area and an 151 initial silicone spacer thickness of 275 μ m. During testing, the 152 162

153 contact pads were clamped against copper foil leads connected 154 to an electrical impedance spectroscopy system (EIS). While 155 we used EIS to precisely characterize the device performance, a 156 capacitive Wheatstone bridge can be used as a lightweight, 157 portable alternative. Once connected to the EIS, the sensor was 158 placed in a customized mechanical stretcher capable of 159 cyclically stretching the sensor to 100% strain at 2 mm/s. 160 The fabrication process with special attention to the patterning 161 method is discussed in greater detail in Supporting Information. Sensors made in this fashion were cycled between 0 and 163 100% strain (Figure 2a). For comparison, we first tested the 164 piezoresistive performance of our electrodes. As anticipated, 165 there was significant resistive hysteresis that resulted in a 166 permanent increase in baseline resistance of slightly over 100% 167 relative to the initial resistance (~15 k Ω by two-point 168 measurement). In addition to hysteresis, there was pronounced 169 viscoelastic behavior (inset, Figure 2a), where the sample was 170 strained to 100% and held at that strain for 10 s before being 171 relaxed. During this process, it is clear that the percolation 172 network undergoes stress relaxation. Lastly, there was a 173 continual decrease in the piezoresistive gauge factor with cycling (~ 0.5 over the first 10 cycles to ~ 0.25). 174

By contrast, when used as capacitive strain sensors the 175 176 performance was markedly stable and reliable. The Poisson 177 capacitors averaged a baseline capacitance of 16 pF (n = 6; 178 measured by EIS). All measurements used an AC excitation frequency of 10 kHz (selected by examining the frequency 179 180 response curve, Supporting Information Figure S2). Devices 181 subjected to 100% cyclic strain (Figure 2b) exhibited a relative 182 change in capacitance (blue circles) that closely matched (R^2 = 183 0.998) that predicted by the linear elastic model with the 184 Palmer correction factor (see Supporting Information). This 185 stable performance occurs despite the wide variability in the 186 resistive properties of the two percolation electrodes. Moreover, 187 the performance under increasing tension versus that under 188 decreasing tension is essentially symmetric. When comparing 189 strain versus normalized change in capacitance for these two 190 cases, the average slopes differed by <2%.

The actual stability of the device was assessed over the course 191 192 of 3000 cycles at 100% cyclic strain (Figure 2c). During testing, 193 a new sensor was mounted and immediately subjected to a 194 "step-and-hold" test where it was stretched at 2 mm/s up to 195 100% strain and held for ~12.5 s before reversing the direction 196 of the stretcher. This data set was taken to be the baseline 197 response curve. Following this, the sensor was cycled from 0 to 198 100% strain with a sawtooth wave profile. Every 1000 cycles 199 (up to 3000 total cycles), the sensor was subjected to an 200 additional step-and-hold test to assess its stability. When used 201 as a Poisson capacitor, the device exhibited stable performance 202 with little hysteresis (Figure 2d). The offset error did not 203 exceed 3% of the baseline, and the trend was not monotonic, 204 implying a relatively stable network absent of significant 205 hysteresis. To our knowledge, variability of within 3% at over 206 3000 cycles represents some of the lowest values reported for 207 stretchable strain gauges (a typical value for a recent nanotube resistive strain gauge was over 10% hysteresis after the first 2.08 cycle).¹⁵ Moreover, if the data are normalized with respect to 209 210 the data taken at 1000 cycles as opposed to the first cycle this 211 error drops to less than 1.5% (data not shown), implying that 212 there is slight settling that occurs over the first 1000 cycles.

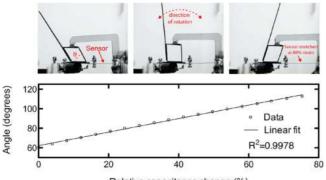
The sensitivity of the Poisson capacitor lies very near to the 213 214 theoretical limit for an elastomeric parallel plate capacitor, 215 which predicts a gauge factor of 1. Here, the gauge factor is

defined as $(\Delta C/C_0)/\varepsilon$. Calculating the mean gauge factor from 216 the data in Figure 2c, we obtain a gauge factor of 0.99 that is 217 uniform throughout the entire 100% strain range. A similar 218 piezocapacitor recently reported was tested to 30% strain with a 219 gauge factor of 0.4.20 The only sensor to our knowledge that 220 explored similar strain (100+%) and cyclic testing relies on a 221 nanotube piezoresistor and has a gauge factor of 0.86 for strains 222 less than 40%, and 0.06 for strains greater than 60%.² 223 Additionally, strain gauge sensitivity can vary with temperature, 224 and we controlled for thermally induced expansion and 225 dielectric constant variation by conducting all experiments at 226 25 °C. However, we also directly calculated the sensor's 227 sensitivity to temperature by using published values for silicone 228 material properties and adding both thermal expansion and 229 thermally induced dielectric constant variation into the Palmer 230 model.²⁹ These calculations suggest that the sensor's capacitive 231 output will vary on the order of 0.01 pF/°C. Given that the 232 sensor operates on the order of 10 pF, this represents a 233 sensitivity of 0.1%/°C, meaning that it should be fairly resistant 234 to temperature variations. 235

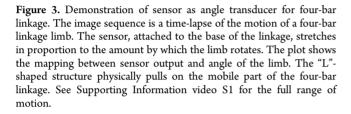
As a demonstration of an alternative to traditional rigid 236 transducers and encoders for robotics, we built a proof-of- 237 concept for robotics applications where size, weight, and power 238 strongly constrain design options. Two such examples where 239 these constraints are crucial are the MEDIC and RoACH 240 (platforms of centimeter-scale walking and running robots). 241 Here, the exoskeleton of the robot consists entirely of origami- 242 style composite laminates that have been cut and folded to 243 produce from four to six "legs" in the form of four-bar 244 linkages.^{30–32} This process results in a strong and light robot 245 (RoACH is only 2.4 g) but also places a premium on sensing 246 and actuation components. At present, the legs are controlled 247 via open-loop, contractile shape memory alloy actuators and 248 passive return springs. Using a Poisson capacitor instead of a 249 return spring would additionally allow feedback control, in turn 250 allowing much more consistent locomotion and more complex 251 behaviors.

Using the same smart composite manufacturing process as 253 used with MEDIC and DASH, we built a scaled up version off 254 of leg linkage and attached the Poisson capacitor in the place of 255 the return spring. Supporting Information video S1 presents the 256 full range of motion of the linkage when coupled to the sensor 257 and stretching apparatus. Figure 3 presents stills of this motion 258 f3 sequence above a plot mapping "limb angle" to relative change 259 in capacitance. To do this, the system was cycled through 80% 260 strain of the sensor (limited by the geometry of the linkage), 261 during which time the 4-bar linkage rotates the "limb" through 262 a wide arc while the sensor stretches. In essence, this allows us 263 to transduce changes in joint angles without relying on an angle 264 encoder or any other traditional, rigid sensing component such 265 as an linear variable differential transformer. While a simple 266 demonstration, this type of sensing can clearly be scaled to a 267 variety of different joints and linkages in a diverse array of 268 robotics platforms. 269

We have demonstrated a high-strain elastomeric, parallel- 270 plate capacitive strain gauge that relies on the Poisson effect to 271 translate uniaxial strain into a scaled deformation that brings 272 the two percolating electrodes closer together. By operating in 273 this fashion, we bypass a number of the typical problems 274 underlying percolating nanotube electrodes and piezoresistive 275 designs such as hysteresis and a variable gauge factor. The 276 device was fabricated using a novel hydrophobicity patterning 277 technique that can be performed rapidly (<20 min for full 278



Relative capacitance change (%)



279 assembly) with minimal equipment and at low cost (<\$1/ 280 sensor) making it an attractive option for anyone needing to 281 rapidly prototype this type of sensor (or similar nanotube 282 devices). Despite the anticipated, unstable resistive properties 283 of our stretchable electrodes, the capacitive output of the sensor 284 remains stable to within 3% over 3000 cycles. Further, the 285 sensor exhibits the highest reported gauge factor (GF = 0.99) 286 of any strain gauge capable of reversibly undergoing 100% 287 strains or higher. Coupling this to the linear performance 288 throughout the entire 100% strain cycle means that is readily 289 calibrated and adaptable to practical sensing applications, as we 290 have demonstrated by incorporating it into a typical robotic 291 linkage as an alternative to an angle encoder.

ASSOCIATED CONTENT 292

Supporting Information 293

294 Characterization of the silicone mechanical properties, detailed 295 description of the fabrication processes, discussion of the 296 patterning technique, a discussion of the Palmer model for 297 parallel plate capacitors, capacitance versus frequency response 298 of the sensor, and a video of the four-bar linkage leg going 299 through its full range of motion. This material is available free 300 of charge via the Internet at http://pubs.acs.org.

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304 Notes

305 The authors declare no competing financial interest.

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REFERENCES 313

314 (1) Ajovalasit, a; Zuccarello, B. J. Strain Anal. Eng. Des. 2005, 40, 315 643-653.

341

- (2) Yamada, T.; Hayamizu, Y.; Yamamoto, Y.; Yomogida, Y.; Izadi- 316 najafabadi, A.; Futaba, D. N.; Hata, K. Nat. Nanotechnol. 2011, 1-6. 317 (3) Cutkosky, M. R.; Kim, S. Philos. Trans. R. Soc., Ser. A 2009, 367, 318 1799-813. 319
- (4) Rogers, J. a; Someya, T.; Huang, Y. Science (New York, N.Y.) 320 2010, 327, 1603-7. 321
- (5) Lacour, S. P.; Benmerah, S.; Tarte, E.; Fitzgerald, J.; Serra, J.; 322 McMahon, S.; Fawcett, J.; Graudejus, O.; Yu, Z.; Morrison, B. Med. 323 Biol. Eng. Comput. 2010, 945-954. 324
- (6) Carta, R.; Jourand, P.; Hermans, B.; Thoné, J.; Brosteaux, D.; 325 Vervust, T.; Bossuyt, F.; Axisa, F.; Vanfleteren, J.; Puers, R. Sens. 326 Actuators A 2009, 156, 79-87. 327
- (7) Sekitani, T.; Nakajima, H.; Maeda, H.; Fukushima, T.; Aida, T.; 328 Hata, K.; Someya, T. Nat. Mater. 2009, 8, 494-9. 329
- (8) Takei, K.; Takahashi, T.; Ho, J. C.; Ko, H.; Gillies, A. G.; Leu, P. 330 W.; Fearing, R. S.; Javey, A. Nat. Mater. 2010, 9, 1-6. 331
- (9) Sekitani, T.; Noguchi, Y.; Hata, K.; Fukushima, T.; Aida, T.; 332 Someya, T. Science (New York, N.Y.) 2008, 321, 1468-72. 333
- (10) Hu, L.; Yuan, W.; Brochu, P.; Gruner, G.; Pei, Q. Appl. Phys. 334 Lett. 2009, 94, 161108. 335
- (11) Huang, Y. Y.; Terentjev, E. M. Adv. Funct. Mater. 2010, 20, 336 4062 - 4068.337
- (12) Jung, Y. J.; Kar, S.; Talapatra, S.; Soldano, C.; Viswanathan, G.; 338 Li, X.; Yao, Z.; Ou, F. S.; Avadhanula, A.; Vajtai, R.; Curran, S.; 339 Nalamasu, O.; Ajayan, P. M. Nano Lett. 2006, 6, 413-8. 340
- (13) Bokobza, L. Polymer 2007, 48, 4907-4920.
- (14) Gui, X.; Cao, A.; Wei, J.; Li, H.; Jia, Y.; Li, Z.; Fan, L.; Wang, K.; 342 Zhu, H.; Wu, D. ACS Nano 2010, 4, 2320-2326. 343
- (15) Kyrylyuk, A. V.; Hermant, M. C.; Schilling, T.; Klumperman, B.; 344 Koning, C. E.; van der Schoot, P. Nat. Nanotechnol. 2011, 345 DOI: 10.1038/NNANO.2011.40. 346

(16) Sekitani, T.; Noguchi, Y.; Hata, K.; Fukushima, T.; Aida, T.; 347 Someya, T. Science (New York, N.Y.) 2008, 321, 1468-72. 348

- (17) Shin, M. K.; Oh, J.; Lima, M.; Kozlov, M. E.; Kim, S. J.; 349 Baughman, R. H. Adv. Mater. 2010, 22, 2663-2667. 350
- (18) Zhou, Y.; Hu, L.; Grüner, G. Appl. Phys. Lett. 2006, 22-24. 351 (19) Lee, K.; Lee, S. S.; Lee, J. a; Lee, K.-C.; Ji, S. Appl. Phys. Lett. 352 2010, 96, 013511. 353
- (20) Lipomi, D. J.; Vosgueritchian, M.; Tee, B. C.-K.; Hellstrom, S. 354 L.; Lee, J. a; Fox, C. H.; Bao, Z. Nat. Nanotechnol. 2011, 1-6. 355
- (21) Loh, K. J.; Kim, J.; Lynch, J. P.; Kam, N. W. S.; Kotov, N. a 356
- Smart Mater. Struct. 2007, 16, 429-438. 357 (22) Kollosche, M.; Stoyanov, H.; Laflamme, S.; Kofod, G. J. Mater. 358
- Chem. 2011, 21, 8292. 359
- (23) Shergold, O. A.; Fleck, N. A.; Radford, D. Int. J. Impact Eng. 360 2006, 32, 1384-1402. 361
- (24) Palmer, H. B. Trans. Am. Inst. Electr. Eng. 1937, 56, 363-366. 362 (25) Tsougeni, K.; Tserepi, A.; Boulousis, G.; Constantoudis, V.; 363
- Gogolides, E. Plasma Processes Polym. 2007, 4, 398-405. 364
- (26) Hu, L.; Hecht, D. S.; Gruner, G. Nano Lett. 2004, 4, 2513- 365 2517. 366
- (27) Zhang, M.; Fang, S.; Zakhidov, A. a; Lee, S. B.; Aliev, A. E.; 367 Williams, C. D.; Atkinson, K. R.; Baughman, R. H. Science (New York, 368

N.Y.) 2005, 309, 1215-9. 369

- (28) Choonee, K.; Syms, R. R. a.; Ahmad, M. M.; Zou, H. Sens. 370 Actuators A 2009, 155, 253-262. 371 372
- (29) Baker, E.; Barry, A. Ind. Eng. 1946, 5-8.
- (30) Kohut, N. J.; Hoover, A. M.; Ma, K. Y.; Baek, S. S.; Fearing, R. S. 373 IEEE Int. Conf. Rob. Autom. 2011, DOI: 10.1109/ICRA.2011.5980360. 374
- (31) Hoover, A. M.; Fearing, R. S. 2008 IEEE Int. Conf. Rob. Autom. 375 2008, 886-892. 376

(32) Hoover, A. M.; Steltz, E.; Fearing, R. S. Intelligent Robots and 377 Systems 2008, 26-33, DOI: 10.1109/IROS.2008.4651149. 378