worked MP35N alloy is not due to the formation of a hexagonal close packed ϵ-phase as indicated by previous investigators. The bands in the structure which were previously identified as the ϵ-phase are, in fact, deformation twins. However, strengthening during aging of MP35N alloys is at least partially due to the formation of fine ϵ-precipitates during aging. Changes in microstructure cannot directly account for the observed improvement in hydrogen embrittlement resistance after aging at 1060 K.

We would like to acknowledge T. E. Mitchell and V. Lou of Case-Western Reserve Institute for their help in the interpretation of preliminary microstructural data prior to this study. We would also like to acknowledge the technical assistance of Carl Klein.

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Correction to *Met. Trans. A*, 1979, vol. 10A

A Study of the Early Stages of Tempering in an Fe-1.2 Pct Alloy, by D. L. Williamson, K. Nakazawa, and G. Krauss

Page 1351, title should read:

A Study of the Early Stages of Tempering in an Fe-1.2 Pct C Alloy

Page 1362, the following note should appear at the end of the article:

The reader is referred to two recent papers dealing with the MES parameters of ϵ-carbide: H. Ino and T. Ito: *J. Phys. Colloq. C2*, 1979, vol. 40, pp. C2-644; J. Foct, J. P. Senateur, J. M. Dubois, and G. LeCaer: *J. Phys. Colloq. C2*, 1979, vol. 40, pp. C2-647. The *"e"* and *"f"* components observed by Ino and Ito after a 413 K × 1 h temper are similar to the M2 and M4 components described here. The work of Foct *et al* suggests that the value of H we obtain for M4 would be consistent with 3 carbon nearest neighbors at Fe sites in η-Fe_2C.