# NANO EXPRESS

**Open Access** 

# Analysis of the Negative-SET Behaviors in Cu/ZrO<sub>2</sub>/Pt Devices



Sen Liu<sup>1,2</sup>, Xiaolong Zhao<sup>2</sup>, Qingjiang Li<sup>1</sup>, Nan Li<sup>1</sup>, Wei Wang<sup>1,2</sup>, Qi Liu<sup>2\*</sup> and Hui Xu<sup>1\*</sup>

## Abstract

Metal oxide-based electrochemical metallization memory (ECM) shows promising performance for next generation non-volatile memory. The negative-SET behavior has been observed in various oxide-based ECM devices. But the underlying mechanism of this behavior remains unaddressed and the role of the metal cation and oxygen vacancy in this behavior is unclear. In this work, we have observed two kinds of negative-SET (labeled as N-SET1 and N-SET2) behaviors in our Cu/ZrO<sub>2</sub>/Pt devices. Both the two behaviors can result in hard breakdown due to the high compliance current in reset process. The *I-V* characteristic shows that the two negative-SET behaviors have an obvious difference in operation voltage. Using four-probe resistance measurement method, the resistance-temperature characteristics of the ON-state after various negative-SET behaviors have been studied. The temperature dependence results demonstrate that the N-SET1 behavior is dominated by Cu conductive filament (CF) reformation caused by the Cu CF overgrowth phenomenon while the N-SET2 is related to the formation of oxygen vacancy CF. This work may provide a comprehensive understanding of the switching mechanism in oxide-based ECM devices.

**Keywords:** Electrochemical metallization memory, Resistive switching, Conductive filament, Temperature dependence, Oxygen vacancy

### Background

The resistive switching (RS) phenomenon, occurred in resistive random access memory (RRAM) also named memristive device, has been widely studied for various applications [1–7]. The binary metal oxide RRAM, which consists of an insulating layer sandwiched between two metal electrodes, has been considered one of the most promising candidate for the next-generation non-volatile memory due to the simple structure and high performance [8-10]. According to the electrochemical activeness of the electrode and switching mechanism, the metal oxide RRAM is divided into electrochemical metallization memory (ECM), valence change memory (VCM), and thermochemical memory (TCM) [11, 12]. The ECM device, also called conductive bridge random access memory (CBRAM), consists of an electrochemical active electrode (e.g., Cu, Ag, or Ni) and an inert electrode (e.g., Pt, W, or Au). The resistance

switching phenomenon of ECM device is dominated by the formation and dissolution of conductive filament (CF) which is relative to the redox process of the active metal atoms [13–15]. The VCM device consists of two metal electrodes which are not easily oxidized or the oxidized form are not easily reduced back to the metal atom. The resistive switching of VCM device is triggered by a drift of oxygen ion-related defects, typically oxygen vacancies [11]. While TCM effect is used to explain the unipolar phenomenon in the metal oxide RRAM, in which a current-induced temperature increase leads to a fuse-antifuse process.

According to the ECM theory, a CF consisting of active electrode atoms can be formed in the ECM device when a positive voltage is applied on the active electrode in forming (P-Forming) or set (P-SET) process. Vice versa, the CF will be dissolved when a negative voltage is applied on the active electrode in reset (N-RESET) process. As a result, the ECM device will be operated in bipolar switching mode. In addition, when a negative voltage is applied to a virgin oxide-based ECM device, which is called negative forming (N-Forming) process, an oxygen vacancy CF can also be formed to bridge the



© The Author(s). 2016 **Open Access** This article is distributed under the terms of the Creative Commons Attribution 4.0 International License (http://creativecommons.org/licenses/by/4.0/), which permits unrestricted use, distribution, and reproduction in any medium, provided you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made.

<sup>\*</sup> Correspondence: liuqi@ime.ac.cn; xuhui@nudt.edu.cn

<sup>&</sup>lt;sup>2</sup>Key Laboratory of Microelectronic Devices and Integrated Technology, Institute of Microelectronics, Chinese Academy of Sciences, Beijing 100029,

China <sup>1</sup>College of Electronic Science and Engineering, National University of Defense Technology, Changsha 410073, China

electrodes [16, 17]. The ECM effect and VCM effect are coexistent in the oxide-based ECM device due to the inevitable oxygen vacancies in the oxide layer, especially when a negative voltage is applied on the device [18, 19]. Recently an unexpected negative-SET (N-SET) behavior has been observed in several ECM structure devices [20–24], which means that a CF is formed in the negative voltage. The N-SET behavior will cause an overset issue under DC sweeping mode due to the high compliance current in N-RESET process, resulting in hard breakdown. However, the underlying mechanism of the N-SET behavior remains unaddressed and the role of the metal cation and oxygen vacancy in the CF formation process at the negative voltage is unclear.

In our previous work, we have demonstrated that the metallic CF overgrowth in ECM devices can cause N-SET behavior [24]. While in this work, we have further observed two kinds of N-SET behaviors (labeled as N-SET1 and N-SET2) in our Cu/ZrO<sub>2</sub>/Pt devices. The operation voltages of the N-SET1 and N-SET2 behaviors are  $\sim$ -2.5 and  $\sim$ -4 V, respectively. Both the N-SET1 and N-SET2 behaviors can result in hard breakdown. The statistical results of operation voltages extracted from 80 devices show that the voltage distribution of N-RESET and N-SET1 have an obvious overlap and the voltage distribution of N-SET2 is close to that of N-Forming. In order to reveal the nature of the CFs, the temperature dependences of the CFs after N-SET1 and N-SET2 behaviors were measured to compare with that of the P-Forming and N-Forming devices. The results show that the CF of N-SET1 behavior has a similar temperature coefficient with that of P-Forming behavior, while the CF of N-SET2 behavior has a close temperature coefficient with that of N-Forming behavior. Based on these results, we demonstrated that the N-SET1 behavior is dominated by Cu CF reformation caused by the Cu CF overgrowth phenomenon while the N-SET2 is related to the formation of oxygen vacancy CF. Finally, a mechanism based on the coexistence of the ECM and VCM effects is proposed to explain the two kinds of N-SET behaviors.

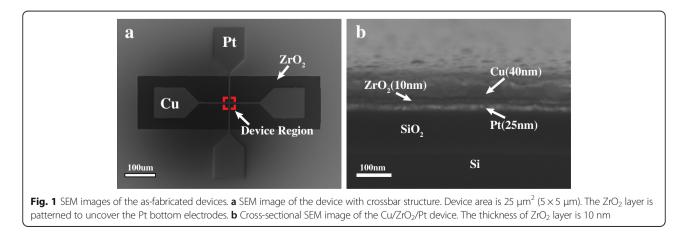
#### Methods

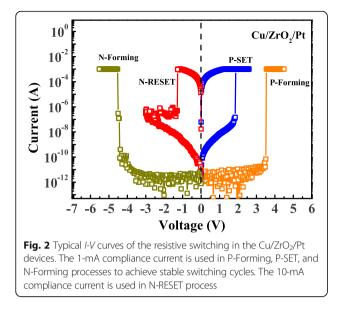
The ECM devices with the structure Cu/ZrO<sub>2</sub>/Pt were fabricated as follows. First, the vertical lines of Pt/Ti (25/ 5 nm) were deposited on the SiO<sub>2</sub>/Si substrate by ebeam evaporation after the first lithography process. A thin Ti layer (5 nm) was used to enhance the adhesion of Pt and SiO<sub>2</sub> layers. Then, the insulting oxide layer of  $ZrO_2$  (10 nm) was grown by magnetron sputtering after the second lithography process. The ZrO<sub>2</sub> layer was patterned to uncover the Pt pad, which facilitates to minimize the sheet resistance of the probe and Pt pad. After the third lithography process, horizontal lines of the Pt/Cu (10/40 nm) were deposited by magnetron sputtering. The Pt layer (10 nm) was covered on Cu layer to prevent the oxidation of Cu. The cell areas of the devices range from 4  $\mu$ m<sup>2</sup> (2 × 2  $\mu$ m) to 25  $\mu$ m<sup>2</sup> (5 × 5 µm). The scanning electron microscope (SEM) image of the as-fabricated device and the cross-section SEM image of the device region are shown in Fig. 1.

The *I-V* characteristics of the devices were measured with Agilent B1500 semiconductor parameter analyzer. During the electrical measurement, the voltage was applied to the Cu electrode while the Pt electrode was tied to ground. The 1- and 10-mA compliance currents were applied in set and reset process to achieve reproducible switching cycles respectively. The four-probe resistance measurement was applied to obtain the accurate ON-state resistance of the device by eliminating the effect of the electrode line resistance and sheet resistance of the probe and pad. The temperature dependence of the ON-state of the device was performed in the temperature of 300 to 410 K.

#### **Results and Discussion**

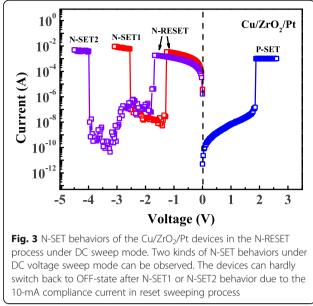
The typical *I-V* curves of the Cu/ZrO<sub>2</sub>/Pt devices are shown in Fig. 2. The initial resistance of the virgin device is ~ $10^{11} \Omega$  at 0.1-V read voltage. Similar to other ECM devices, a forming process is required to achieve reproducible switching cycles [25, 26]. When a positive



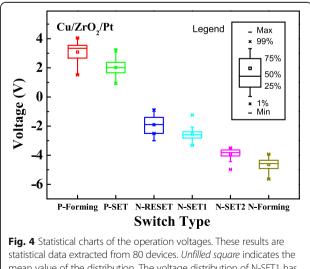


voltage sweeping  $(0 \rightarrow 4.5 \text{ V})$  was applied on a virgin device, a current abrupt can be observed at about 3.3 V and then the device switched to ON-state. After P-Forming process, the Cu/ZrO<sub>2</sub>/Pt device shows bipolar resistive switching behavior and the device can switch between ON-state and OFF-state repeatedly by N-RESET  $(0 \rightarrow -3 \text{ V})$  and P-SET  $(0 \rightarrow 2.5 \text{ V})$  process, as shown by the blue and red lines in Fig. 2. The typical resistances of the ON-state and OFF-state are on the order of  $10^2$  and  $10^{10} \Omega$ , respectively. It is worth noting that the device can also switch to ON-state at ~-4.5 V when a negative voltage sweep  $(0 \rightarrow -5.5 \text{ V})$  was applied on a virgin device in the N-Forming process, as shown by the dark yellow line in Fig. 2.

During the switching cycles, two kinds of N-SET (N-SET1 and N-SET2) behaviors were occasionally observed after N-RESET behavior, as shown in Fig. 3. The N-SET1 behavior occurred at ~-2.5 V, while the N-SET2 occurred at ~-4 V after N-RESET behavior. The device can hardly switch back to OFF-state after N-SET1 or N-SET2 behavior due to the high compliance current in the reset sweeping process. Then the statistical characteristics of the various operation voltages extracted from 80 Cu/ZrO<sub>2</sub>/Pt devices are shown in Fig. 4. Specifically, the device is triggered by the P-Forming  $(0 \rightarrow 4.5 \text{ V})$ process. Then, 15 switching cycles with N-RESET ( $0 \rightarrow$ -3 V) and P-SET (0  $\rightarrow$  2.5 V) processes are implemented on each device until the N-SET1 behavior happens. If the N-SET1 behavior has not been observed after the 15 switching cycles, another N-RESET process  $(0 \rightarrow$ -5.5 V) is applied on the device following with P-SET process  $(0 \rightarrow 2.5 \text{ V})$ . The operation voltages of P-Forming, P-SET, N-RESET, N-SET1, and N-SET2 are extracted from the 65 Cu/ZrO<sub>2</sub>/Pt devices. In addition, the operation voltage of N-Forming is extracted from the



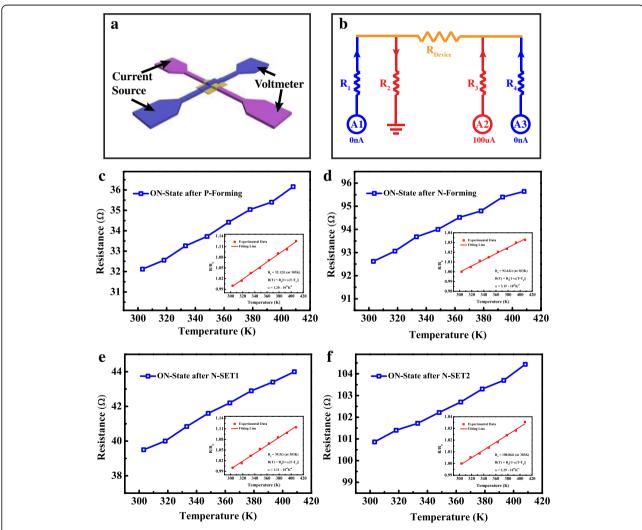
other 15 Cu/ZrO<sub>2</sub>/Pt devices with N-Forming process  $(0 \rightarrow -5.5 \text{ V})$ . As a result, the N-SET1 behaviors have been observed in 24 of the 65 devices after 15 switching cycles. But it should be noted that the occurrence probability of N-SET1 increases with the switching cycles. The statistical results demonstrate that the voltage distributions of N-RESET and N-SET1 have an obvious overlap, indicating that the N-SET1 behavior cannot be avoided by controlling the sweeping voltage in N-RESET process. In addition, the results reveal that the voltage distributions of N-SET1 and N-SET2 have no overlap and the mean value of the N-SET1 and N-SET2 are  $\sim -2.6$  and  $\sim -4 \text{ V}$ , respectively. The obvious voltage

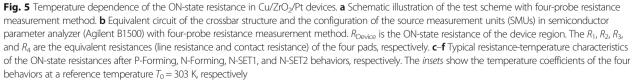


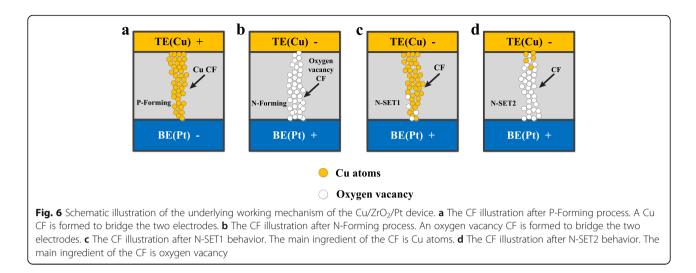
statistical data extracted from 80 devices. *Unfilled square* indicates the mean value of the distribution. The voltage distribution of N-SET1 has an obvious overlap with that of N-RESET and the distribution of N-SET2 is close to N-Forming

difference of the N-SET1 and N-SET2 behaviors indicate that the underlying mechanisms of the two behaviors may be different.

To further clarify the underlying mechanism, the resistance-temperature characteristics of the ON-state after various operation behaviors have been studied. The fourprobe resistance measurement is used to eliminate the effect of the line resistance and contact resistance [27], and the schematic illustration is shown in Fig. 5a. The equivalent circuit of the crossbar structure is shown in Fig. 5b, in which  $R_{\text{Device}}$  is the device resistance and  $R_1$ ,  $R_2$ ,  $R_3$ , and  $R_4$  are the total resistances of line resistances and contact resistances of the four electrode pads, respectively. Four source measurement units (SMUs) are used to implement with four-probe resistance measurement and the configurations of the SMUs are shown in Fig. 5b. The use of four-probe resistance measurement method makes it possible to measure the accurate resistance of the device along with the change of temperature. The ON-state resistances after P-Forming, N-Forming, N-SET1, and N-SET2 behaviors have been measured as a function of temperature ranging from 300 to 410 K, as shown in Fig. 5c–f. All the ON-state resistances show a linear increase with temperature, which is typical for electronic transport in metals. The metallic resistance as a function of temperature change can be written as  $R(T) = R_0[1 + \alpha(T - T_0)]$  [28, 29], in which  $R_0$  is the resistance at temperature  $T_0$  and  $\alpha$  is the resistance temperature coefficient. Thus, the typical temperature







coefficients of the ON-state resistances after P-Forming, N-Forming, N-SET1, and N-SET2 behaviors can be calculated to  $1.2 \times 10^{-3} \text{ K}^{-1}$ ,  $3.15 \times 10^{-4} \text{ K}^{-1}$ ,  $1.11 \times 10^{-3} \text{ K}^{-1}$ , and  $3.29 \times 10^{-4} \text{ K}^{-1}$  at a reference temperature  $T_0 = 303$  K, respectively. As a result, the ON-state devices after P-Forming and N-SET1 behaviors have similar temperature coefficients which are consistent with the characteristics of Cu CF in the previous literature [30, 31]. While the temperature coefficients of the ON-state devices after N-Forming and N-SET2 behaviors, which are obviously smaller than that of P-Forming and N-SET1, are consistent with the characteristics of oxygen vacancy CF in the previous works [32, 33]. The results indicate that the main ingredients of the CFs after N-SET1 and N-SET2 behaviors are Cu atoms and oxygen vacancy, respectively.

Here, we propose a microscopic model to describe the underlying switching mechanism in our Cu/ZrO<sub>2</sub>/Pt devices. The virgin Cu/ZrO<sub>2</sub>/Pt device can be formed in both positive and negative voltage polarities. More specifically, when a positive voltage is applied on a virgin device in the P-Forming process, the Cu atoms are oxidized into Cu<sup>2+</sup> and reduced to Cu atoms inside the ZrO<sub>2</sub> layer [13, 24]. Then, a Cu CF is formed via electrochemical reaction and electromigration inside the ZrO<sub>2</sub> layer, as shown in Fig. 6a. Vice versa, when a negative voltage is applied on a virgin device in the N-Forming process, an oxygen vacancy CF will be formed due to the oxygen ion diffusion and migration [16, 17], as shown in Fig 6b. Generally, when the devices are triggered by P-Forming, the ECM and VCM effects are coexistent in the switching cycles. During the N-RESET process, the Cu CF is dissolved due to the thermalassisted electrochemical reaction by applying a negative voltage on the device [13]. But the N-SET1 behavior may occasionally occur when the Cu atoms precipitated in the inert electrode after several switching cycles provide sufficient Cu<sup>2+</sup> source to reform the Cu CF in the N-RESET process [24], as shown in Fig. 6c. Otherwise, the N-SET2 behavior can be observed at a higher negative voltage than that of N-SET1. The VCM effects dominate the formation of the oxygen vacancy CFs in the devices, as shown in Fig. 6d. Due to the high compliance current, the device can hardly turn to OFF-state when the two N-SET behaviors occurred.

#### Conclusions

In summary, two kinds of N-SET behaviors have been observed in our Cu/ZrO<sub>2</sub>/Pt devices. The detailed I-V characteristics reveal that the two N-SET behaviors have an obvious different voltage and the N-SET behaviors may cause hard breakdown because of the high compliance current in the N-RESET process. Subsequently, the statistical results demonstrate that the N-SET1 voltage distribution has an overlap with that of N-RESET, indicating that the N-SET1 behavior cannot be avoided by controlling the operation voltage. The following resistance-temperature characteristics of the four ON-state devices reveal that the main ingredients of the CFs after P-Forming and N-SET1 behaviors are Cu atoms, while the main ingredient of the CFs after N-Forming and N-SET2 behaviors is oxygen vacancy. The results further confirm that the ECM effect and VCM effect are coexistent in our devices. Although the work is implemented with Cu/ZrO<sub>2</sub>/Pt device structure, the experimental method can be easily extended to other ECM devices.

#### Abbreviations

CBRAM: Conductive bridge random access memory; CF: Conductive filament; ECM: Electrochemical metallization memory; N-Forming: Negative forming; N-RESET: Negative reset; N-SET: Negative-SET; P-Forming: Positive forming; P-SET: Positive set; RRAM: Resistive random access memory; RS: Resistive switching; SEM: Scanning electron microscope; SMU: Source measurement unit; VCM: Valence change memory

#### Acknowledgements

This work was supported by the National Natural Science Foundation of China under Grant Nos. 61521064, 61422407, 61474136, 61334007, 61574166,

6127409, 61322408, 61522408, 61574169, 61471377, 61604177, and 61306117, the National High Technology Research Development Program under Grant No. 2014AA032901, Beijing Training Project for the Leading Talents in S&T under Grant No. Ijrc201508, NUDT Research Funding Program JC-15-04-02, the Opening Project of Key Laboratory of Microelectronics Devices and Integrated Technology, Institute of Microelectronics, the Chinese Academy of Sciences, Youth Innovation Promotion Association CAS under Grant No. 2015096, and the CAEP Microsystem and THz Science and Technology Foundation under the Grant No. CAEPMT201504.

#### Authors' Contributions

QL designed and conducted all experiments. SL and XZ fabricated samples. SL carried out the electrical measurement. QL and SL wrote the manuscript. All authors critically read and contributed to the manuscript preparation. All authors read and approved the final manuscript.

#### **Competing Interests**

The authors declare that they have no competing interests.

#### Received: 12 September 2016 Accepted: 29 November 2016 Published online: 07 December 2016

#### References

- Waser R, Dittmann R, Staikov G, Szot K (2009) Redox-based resistive switching memories-nanoionic mechanisms, prospects, and challenges. Adv Mater 21:2632
- Yang Y, Gao P, Gaba S, Chang T, Pan X, Lu W (2012) Observation of conducting filament growth in nanoscale resistive memories. Nat Commun 3:732
- Zhang M, Long S, Wang G, Liu R, Xu X, Li Y, Xu D, Liu Q, Lv H, Miranda E, Suñ J, Liu M (2014) Statistical characteristics of reset switching in Cu/HfO<sub>2</sub>/Pt resistive switching memory. Nanoscale Res Lett 9:694
- Wong HSP, Lee HY, Yu S, Chen YS, Wu Y, Chen PS, Lee B, Chen FT, Tsai MJ (2012) Metal-oxide RRAM. Proc IEEE 100:1951
- Maikap S, Jana D, Dutta M, Prakash A (2014) Self-compliance RRAM characteristics using a novel W/TaO<sub>x</sub>/TiN structure. Nanoscale Res Lett 9:292
- Chang T, Jo SH, Lu W (2011) Short-term memory to long-term memory transition in a nanoscale memristor. ACS Nano 5:7669
- Pi S, Ghadiri-Sadrabadi M, Bardin JC, Xia Q (2015) Nanoscale memristive radiofrequency switches. Nat Commun 6:7519
- Akinaga H, Shima H (2010) Resistive random access memory (ReRAM) based on metal oxides. Proc IEEE 98:2237
- Pan F, Gao S, Chen C, Song C, Zeng F (2014) Recent progress in resistive random access memories: materials, switching mechanisms, and performance. Mater Sci Eng R 83:1
- Chang KC, Chang TC, Tsai TM, Zhang R, Hung YC, Syu YE, Chang YF, Chen MC, Chu TJ, Chen HL, Pan CH, Shih CC, Zheng JC, Sze SM (2015) Physical and chemical mechanisms in oxide-based resistance random access memory. Nanoscale Res Lett 10:120
- 11. Ilia V, Rainer W, John RJ, Michael NK (2011) Electrochemical metallization memories—fundamentals, applications, prospects. Nanotechnol 22:254003
- 12. Yang JJ, Strukov DB, Stewart DR (2013) Memristive devices for computing. Nat Nanotechnol 8:13
- Liu Q, Sun J, Lv H, Long S, Yin K, Wan N, Li Y, Sun L, Liu M (2012) Real-time observation on dynamic growth/dissolution of conductive filaments in oxide-electrolyte-based ReRAM. Adv Mater 24:1844
- Celano U, Goux L, Belmonte A, Opsomer K, Franquet A, Schulze A, Detavernier C, Richard O, Bender H, Jurczak M, Vandervorst W (2014) Threedimensional observation of the conductive filament in nanoscaled resistive memory devices. Nano Lett 14:2401
- Sun J, Liu Q, Xie H, Wu X, Xu F, Xu T, Long S, Lv H, Li Y, Sun L, Liu M (2013) In situ observation of nickel as an oxidizable electrode material for the solid-electrolyte-based resistive random access memory. Appl Phys Lett 102: 053502
- Tsuruoka T, Terabe K, Hasegawa T, Aono M (2010) Forming and switching mechanisms of a cation-migration-based oxide resistive memory. Nanotechnol 21:425205
- Yang L, Meng Y, Song Y, Liu Y, Dong Q, Lin Y, Huang R, Zou Q, Wu J (2012) Dual resistive switching characteristics in Cu<sub>x</sub>Si<sub>y</sub>O resistive memory. Appl Phys Exp 5:111102

- Liu T, Verma M, Kang Y, Orlowski MK (2012) Coexistence of bipolar and unipolar switching of Cu and oxygen vacancy nanofilaments in Cu/TaO<sub>x</sub>/Pt resistive devices. ECS Solid State Lett 1:Q11
- Chung YL, Cheng WH, Jeng JS, Chen WC, Jhan SA, Chen JS (2014) Joint contributions of Ag ions and oxygen vacancies to conducting filament evolution of Ag/TaO<sub>x</sub>/Pt memory device. J Appl Phys 116:164502
- Guan W, Long S, Liu Q, Liu M, Wang W (2008) Nonpolar nonvolatile resistive switching in Cu doped ZrO<sub>2</sub>. IEEE Electron Device Lett 29:434
- Zhong L, Jiang L, Huang R, Groot CH (2014) Nonpolar resistive switching in Cu/SiC/Au non-volatile resistive memory devices. Appl Phys Lett 104:093507
- Chen C, Gao S, Tang G, Song C, Song C, Zeng F, Pan F (2012) Cu-Embedded AIN-based nonpolar nonvolatile resistive switching memory. IEEE Electron Device Lett 33:1711
- Huang JS, Chen LM, Lin TY, Lee CY, Chin TS (2013) Nonpolar electrical switching behavior in Cu-Si(Cu)O<sub>x</sub>-Pt stacks. Thin Solid Films 544:134
- Liu S, Lu N, Zhao X, Xu H, Banerjee W, Lv H, Long S, Li Q, Liu Q, Liu M (2016) Eliminating negative-SET behavior by suppressing nanofilament overgrowth in cation-based memory. Adv Mater. doi:10.1002/adma. 201603293
- Sun H, Liu Q, Li C, Long S, Lv H, Bi C, Huo Z, Li L, Liu M (2014) Direct observation of conversion between threshold switching and memory switching induced by conductive filament morphology. Adv Funct Mater 24:5679
- Liu S, Wang W, Li Q, Zhao X, Li N, Xu H, Liu Q, Liu M (2016) Highly improved resistive switching performances of the self-doped Pt/HfO<sub>2</sub>:Cu/Cu devices by atomic layer deposition. Sci China-Phys Mech Astron 59:127311
- Pickett MD, Strukov DB, Borghetti JL, Yang JJ, Snider GS, Stewart DR, Williams RS (2009) Switching dynamics in titanium dioxide memristive devices. J Appl Phys 106:074508
- Guan W, Liu M, Long S, Liu Q, Wang W (2008) On the resistive switching mechanisms of Cu/ZrO<sub>2</sub>:Cu/Pt. Appl Phys Lett 93:223506
- 29. Bid A, Bora A, Raychaudhuri AK (2006) Temperature dependence of the resistance of metallic nanowires of diameter ≥15 nm: applicability of bloch-Grüneisen theorem. Phys Rev B 74:035426
- Yazdanparast S, Koza JA, Switzer JA (2015) Copper nanofilament formation during unipolar resistance switching of electrodeposited cuprous oxide. Chem Mater 27:5974
- 31. Huang Q, Lilley CM, Bode M, Divan R (2008) Surface and size effects on the electrical properties of Cu nanowires. J Appl Phys 104:023709
- Wang ZQ, Xu HY, Zhang L, Li XH, Ma JG, Zhang XT, Liu YC (2013) Performance improvement of resistive switching memory achieved by enhancing local-electric-field near electromigrated Ag-nanoclusters. Nanoscale 5:4490
- Wang ZQ, Xu HY, Li XH, Zhang XT, Liu YX, Liu YC (2011) Flexible resistive switching memory device based on amorphous InGaZnO film with excellent mechanical endurance. IEEE Electron Device Lett 32:1442

# Submit your manuscript to a SpringerOpen<sup>®</sup> journal and benefit from:

- Convenient online submission
- Rigorous peer review
- Immediate publication on acceptance
- Open access: articles freely available online
- ► High visibility within the field
- Retaining the copyright to your article

#### Submit your next manuscript at > springeropen.com