

# Antibacterial coating of Ti-6Al-4V surfaces using silver nano-Powder Mixed Electrical Discharge Machining

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## Abstract

Previous studies have revealed the potential of powder mixed electrical discharge machining (PMEDM) with regards to concurrently machining part geometry and coating an antibacterial layer on medical devices. This study is aimed at further demonstrating this potential. In order to do so, the PMEDM process was varied by adding different concentrations of silver nano-particles into the dielectric fluid and used to machine Ti-6Al-4V. Afterwards, the resulting machined and coated surfaces were characterized with regards to surface integrity, the coating layer's thickness, microhardness and chemical elements as well as antibacterial property. Material removal rate, tool wear and pulse signals were also analysed in order to give an insight on process feasibility. From both qualitative and quantitative results, it could be established that the surfaces machined and coated by PMEDM method have demonstrated a significant reduction of not only the amount of *S. aureus* bacteria, but also the number of bacterial clusters on the coating layer's surface. Moreover, the coating layer's silver content, which depends on the powder concentration suspended in the dielectric fluid, plays a vital role in the antibacterial property. As compared to surfaces without silver, surfaces containing approximately 3.78% silver content showed a significant decrease in both bacterial numbers and clusters, whereas a further increase in silver content did not result in a considerable bacterial number and cluster reduction. Regarding the machining performance, as compared to EDM without powder, machining time is remarkably decreased by using the PMEDM method.

*Keywords:* Antibacterial coating; PMEDM; titanium implants; coating layer; antibacterial property.

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## 1. Introduction

1 The demand for medical implants has been rapidly increasing. For example, it is predicted that the  
2 medical device market will record a minimum compound annual growth rate of 4.5% between 2017 and  
3 2024 [1, 2]. However, implant related infections still pose a major challenge for the medical device industry.  
4 Although antibiotics have been successfully used in clinical treatments [3], antibacterial coatings of metal  
5 implants are considered as potential long term solutions to this problem [4].

### 6 1.1. Antibacterial coating processes

7 In order to coat an antibacterial layer on metal surfaces, several methods have been studied and their  
8 efficiency demonstrated. For example, titanium oxide layers deposited on titanium substrates by anodic  
9 oxidation showed a significant reduction in the number of viable *E. coli* on the coating surface [5]. These  
10 layers were combined with silver using plasma electrolytic oxidation [6], anodic spark deposition [7] or plasma  
11 spraying [8], in order to enhance the antibacterial properties. In other studies, coating layers with a thickness  
12 of approximately 2  $\mu\text{m}$  were deposited on titanium surfaces by physical vapor deposition (PVD), with a  
13 significant antibacterial potency and the absence of cytotoxic effects being reported [9]. Moreover, chemical  
14 vapour deposition (CVD) has been used for silver-silica surface coatings and has shown a significant reduction  
15 of a variety of bacteria [10]. Furthermore, PVD and CVD methods have also been studied appertaining to  
16 coating thin films for medical application [11, 12]. On the other hand, thermal spraying has been used in  
17 coating silver-containing hydroxyapatite layers on titanium implant surfaces. The results of both *in vitro*  
18 and *in vivo* studies showed a significant reduction in the number of *S. aureus* bacteria on these layers as  
19 compared to hydroxyapatite layers [13, 14]. Another method, namely ion implantation, has been studied  
20 and has shown excellent antibacterial effects [15] and no cytotoxicity of the coating surface [16]. Other  
21 results have shown that silver ion implantation is more suitable than copper ion implantation for coating  
22 antibacterial surfaces of medically applied metals [17].

### 23 1.2. PMEDM for surface modification

24 PMEDM has been studied for surface modification whereby emphasis has been on characterizing how  
25 the process mechanism allows for material deposition in and onto the machined surface [18]. During the  
26 machining process, materials from particles that are mixed into the dielectric fluid are transferred to the  
27 workpiece surface, therefore modifying its characteristics. Research has shown that the microhardness of  $\beta$ -  
28 titanium implant surfaces has been increased 2-fold when silicon particles are suspended in the dielectric fluid  
29 [19]. The PMEDM process also enhances fatigue endurance and biocompatibility of  $\beta$ -titanium implants  
30 for orthopedic applications [20]. An approximate 76% surface hardness increase as well as an elimination of  
31 microcracks on the machined surface have been reported after adding titanium powder into hydrocarbon-  
32 based dielectric fluid [21]. In another study, silicon and manganese powders were used to improve recast layer

33 hardness by approximately 40% as compared to EDM process without powder [22]. Both microhardness  
34 and wear resistance were significantly increased by mixing tungsten powder into the dielectric fluid [23].  
35 PMEDM using titanium carbide particles has also been combined with ultrasonic vibration to not only  
36 improve machining performance, but also to enhance the hardness and wear resistance of the workpiece  
37 surface [24].

38 Similar to conventional EDM, the modified layer while using PMEDM is also affected by machining  
39 parameters such as current and pulse-on-time. Besides, powder concentration is an important parameter  
40 which influences the properties of the resulting machined surface [25]. Powder material has shown its impact  
41 during PMEDM. For example, using  $\text{Al}_2\text{O}_3$  powder provided a higher modified layer thickness whereas TiC  
42 powder resulted in a higher machined surface hardness [26]. The thickness of the modified layer has also  
43 been increased by using smaller powder particle sizes [27]. However, the modified layer thickness was not  
44 affected by varying machining time [28].

### 45 *1.3. Summary of literature review*

46 From literature, it can be established that many methods have been used for coating antibacterial  
47 layers on medical materials. However, these methods, until now, are only used for coating. PMEDM has  
48 demonstrated its ability in, not only machining medical materials, but also in modifying their surfaces.  
49 Therefore, a method that has the ability to concurrently machine and coat an antibacterial layer on medical  
50 device surfaces is proposed and investigated in this study. Moreover, so as to clearly understand how the  
51 process influences the coating layer, the PMEDM machining and coating mechanism is explained.

## 52 **2. PMEDM's machining and coating mechanism**

53 The PMEDM machining and coating mechanism is, as shown in Fig. 1, best explained by categorizing  
54 it into four discharge phases as follows:

- 55 (i) Preparation phase: After applying a potential difference between the tool and workpiece electrodes,  
56 powder particles in the electric field arrange themselves in chains (Fig. 1(a)) [29].
- 57 (ii) Ignition phase: Where voltage is highest and exceeds the dielectric resistivity, a single ignition occurs  
58 after which multiple ignitions take place (Fig. 1(b)) [30].
- 59 (iii) Discharge phase: A plasma channel, composed of ion and electron flows which collide and implant  
60 into the workpiece and tool electrode surfaces respectively, is formed (Fig. 1(c)). Thermal energy  
61 causes the melting of materials from workpiece, tool electrode as well as powder particles. The plasma  
62 channel grows until reaching a balance point between the internal pressure of the formed bubble and  
63 the dielectric fluid's hydrostatic pressure (Fig. 1(d)). Thereafter, it bursts (Fig. 1(e)) and blows  
64 off most of the molten mixture of materials from the tool electrode, powder particles and workpiece.

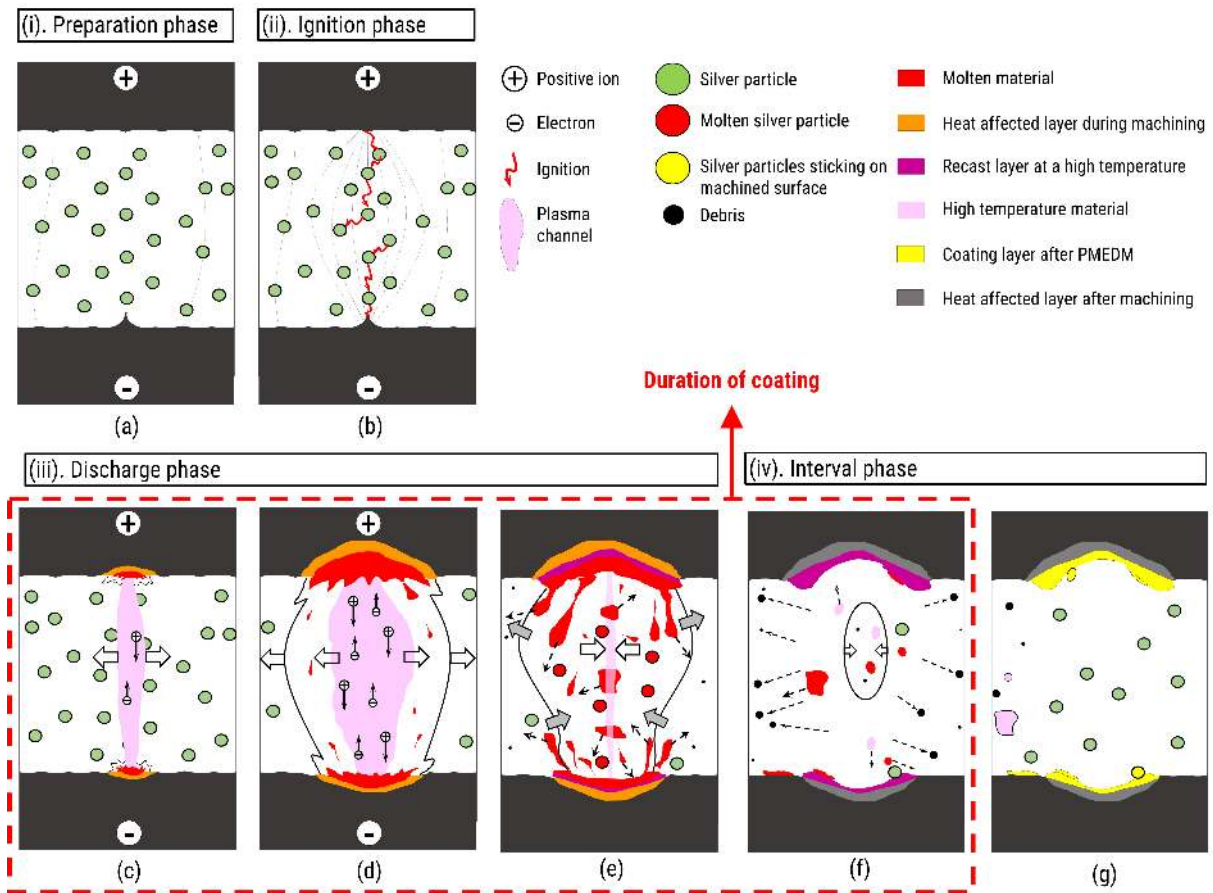


Figure 1: PMEDM's machining and coating mechanism of a single spark. Stages (c-f) represent the duration of the concurrent machining and coating process.

65 However, some of the molten material remains on the workpiece surface thus generating the main part  
 66 of the coating layer.

67 (iv) Interval phase: During this phase, the dielectric fluid cools the workpiece and removes the machined  
 68 debris. Nevertheless, some of the molten materials, after bursting, are sputtered and re-solidified onto  
 69 the machined surface. In addition, some particles, which remain partially or fully molten during the  
 70 recast layer's formation process, stick to sections of the machined surface and forms the rest of the  
 71 resulting coating layer (Fig. 1(f)). Afterwards, fresh dielectric fluid, containing new powder particles,  
 72 continues flowing and cooling the workpiece in anticipation of a new discharge cycle (Fig. 1(g)).

73 **3. Methodology**

74 *3.1. Material selection*

75 Ti-6Al-4V is one of the key materials used to manufacture medical devices owing to its desirable properties  
76 which include excellent biocompatibility, corrosion resistance and high fatigue strength among others. In this  
77 study, (30 x 20 x 1) mm<sup>3</sup> Ti-6Al-4V sheets were used as workpieces. To analyse the chemical composition  
78 of the samples before machining, Energy dispersive X-Ray spectroscopy (EDS) was used. The results were  
79 as presented in Fig. 2.

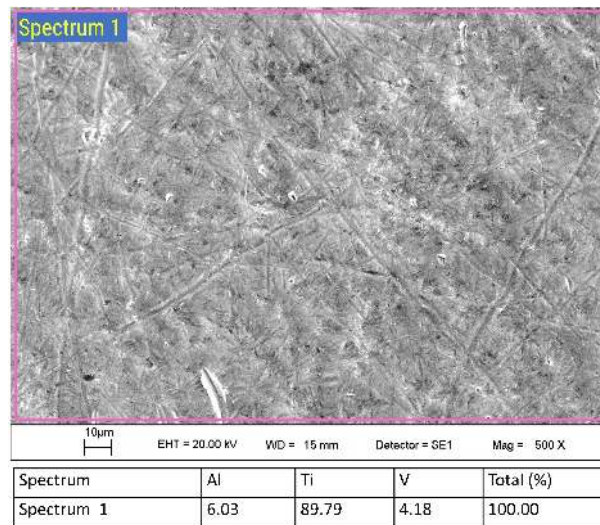


Figure 2: Chemical composition of the sample before machining and coating.

80 From literature, it is clear that silver has a stronger antibacterial effect and is more biocompatible than  
81 other materials such as copper, zinc or titanium dioxide. In addition, various coating methods have shown  
82 no cytotoxicity when depositing silver on medical device surfaces. For these reasons, silver nano-particles  
83 (99.9 % purity) with diameters of  $\approx$  50-60 nm were utilised. The choice of particle size is significant since it  
84 not only influences the coating layer, but is also significant to the stability of the PMEDM process because  
85 large particles could clog the narrow machining gap that is necessary for the sparking process.

86 *3.2. Sample preparation*

87 In this study, the antibacterial property of the coating layer, which is dependent on silver content  
88 deposited in the coating surface during the PMEDM process, is evaluated. In order to do so, the surface  
89 roughness and structure of the samples have to be relatively similar, since they influence bacterial adhesion  
90 and growth. Moreover, surfaces with  $R_a > 0.2 \mu\text{m}$  not only increase plaque accumulation [31], but also  
91 facilitate biofilm formation [32]. Therefore, this  $R_a$  value was referenced for designing surface roughness of  
92 the investigated samples.

93 For the antibacterial evaluation, 11 points were concurrently machined and coated on the workpiece  
 94 surface as shown in Fig. 3. The point numbered “0” was machined without silver powder whereas points  
 95 “1” to “10” were machined with varying silver powder concentrations suspended in the dielectric fluid from  
 96 low to high respectively.

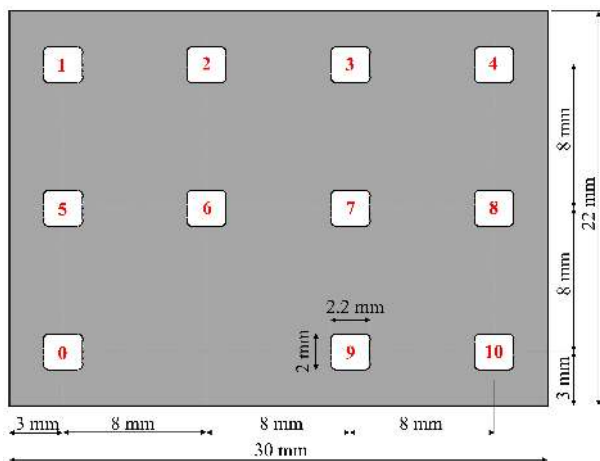


Figure 3: Arrangement of the machined points (white color) on the workpiece.

### 97 3.3. Experimental setup

98 The results from preliminary researches [33, 34] showed that silver powder concentration plays a vital  
 99 role in the deposited silver content. However, it only caused a negligible change on surface roughness.  
 100 Furthermore, other factors such as discharge energy, dielectric material and electrode size have significant  
 101 influences on both roughness and chemical composition of the machined surface. Based on these findings,  
 102 experimental conditions for this study were established as can be seen in Table 1.

Table 1: Experimental conditions.

Machine	Sarix 100 $\mu$ -EDM machine
Tool electrode	- Material: WC-6 wt.% Co - Outer diameter: 0.6 mm - Inner diameter: 0.17 mm
Dielectric fluid	HEDMA111 hydrocarbon
Powder concentrations	0; 2.5; 5; 7.5; 10; 12.5; 15; 17.5; 20; 22.5 and 25 (g/l)
Discharge energy	$\approx 9.98$ ( $\mu$ J)
Polarity	Positive at tool electrode

103 In addition, since discharge energy plays an essential role in not only the surface roughness, but also the

104 thickness of the resulting coating layer, it is necessary to establish the discharge energy boundary which is  
 105 suitable for realising the aforementioned  $R_a = 0.2 \mu\text{m}$  in PMEDM. It enables a discharge energy increment  
 106 which results in a thicker coating layer and consequently allows for a better analysis. For this reason, extra  
 107 experiments were carried out using various discharge energies ( $\approx 9.98 \mu\text{J}$ ,  $\approx 17.5 \mu\text{J}$  and  $\approx 24.46 \mu\text{J}$ ) using  
 108 a 25 g/l powder concentration in the dielectric fluid.

109 In order to carry out the experiments, due to a high-density powder settling at the bottom of the tank, a  
 110 suitable circulation system as shown in Fig. 4 was angularly designed for tank 1. It ensures that all dielectric  
 111 fluid flows out of this tank and into tank 2 where a stirrer is used to continuously mix silver particles with  
 112 the dielectric fluid.

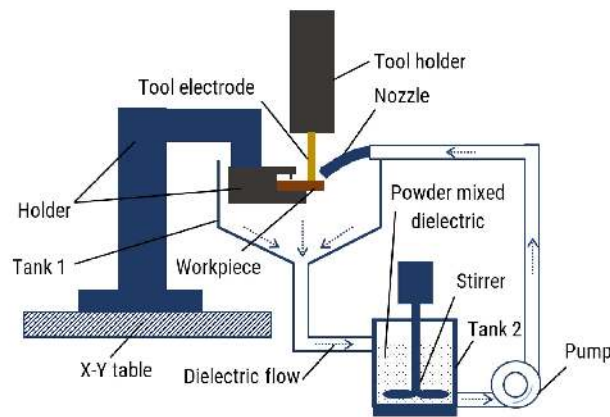


Figure 4: Schematic showing the designed PMEDM dielectric fluid circulation system.

113 As regards the geometry of the machined and coated surface, an area was machined and coated as shown  
 114 in Fig. 5. Five overlapping machining paths of the tool electrode's trajectory (shown in blue color) with a  
 115 20  $\mu\text{m}$  cutting depth are repeated three times to achieve a 60  $\mu\text{m}$  target depth.

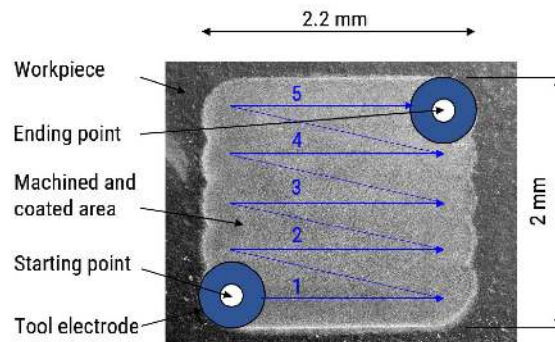


Figure 5: Schematic showing the machined and coated area and tool electrode's trajectory.

116 *3.4. Coating characterization*

117 For effective analysis of the coating layer, the samples have to be carefully analysed. To do so, after  
118 machining, the samples were cleaned for 10 minutes in an ultrasonic bath filled with ethanol at room  
119 temperature, and then dried. Afterwards, their surface topography and composition was analysed using a  
120 scanning electron microscope (SEM) and energy dispersive X-Ray spectroscopy. A Keyence VK9700 3D  
121 Laser scanning confocal-microscope was used to scan the machined surfaces after which MountainsMap 7.4  
122 scanning topography software was used to analyse surface roughness, surface structure and the volume of  
123 removed material. For microhardness testing, the measurement according to “Quasi Continuous Stiffness  
124 Measurement” (DIN EN ISO 14577) was performed by an UNAT nanoindenter for not only the Ti-6Al-4V  
125 substrate, but also the coating surfaces containing different silver contents.

126 In order to analyse the coating layer’s cross section, the samples were carefully grinded, polished by a  
127 wet abrasive paper and then chemically etched using a Kroll solution (1.5% hydrofluoric acid and 4% nitric  
128 acid in water). Afterwards, SEM and EDS were used to observe and analyse the thickness as well as the  
129 elemental composition of the coating layer.

130 *3.5. Material removal rate and tool wear rate characterization*

131 Apart from demonstrating the PMEDM’s performance on machining, it is also important to investigate  
132 the material removal rate (MRR) and tool wear rate (TWR). In order to do so, wear length, after specified  
133 machining duration, was measured by an in-process measuring program. Furthermore, the machining time  
134 of each experiment was recorded, thus allowing MRR or TWR to be calculated by a quotient of “material  
135 removal volume” or “tool wear volume” per machining time, respectively.

136 *3.6. Antibacterial tests*

137 As regards to the antibacterial property of the layer coated by PMEDM method, *S. aureus* SH1000  
138 pSB2035 [35] constitutively expressing green fluorescent protein (GFP) was cultured overnight. The coated  
139 Ti-6Al-4V test plate was incubated with 25 ml lysogeny broth (LB) and inoculated with a 1:1000 dilution  
140 of the *S. aureus* overnight culture with an optical density of 1 at  $\lambda$  600 nm ( $OD_{600}$ ). After 24 hours of  
141 incubation, the coated Ti-6Al-4V test plate was removed, washed once with phosphate buffered saline and  
142 subsequently fixed with 4% paraformaldehyde for 30 minutes at room temperature. The amount of bacteria  
143 and clusters was analysed in 630 x magnification microscopic pictures (Zeiss fluorescence microscope Axio  
144 Observer.Z1) of the respective areas on the coated Ti-6Al-4V test plate. GFP fluorescence was excited at  
145 488 nm and read out at 500-550 nm. Four areas on each numbered coating surface (from “0” to “10”) were  
146 evaluated and the mean values are given with SEM.



147 **4. Results and discussion**

148 To evaluate the PMEDM's efficiency in concurrent machining part geometry and coating an antibacte-  
149 rial surface, different analysis criteria such as discharge energy, the coating layer's antibacterial property  
150 and microhardness, the deposited silver distribution, as well as machining performance are analysed and  
151 discussed.

152 *4.1. Influence of discharge energy on surface roughness*

153 Regarding the influence of the pulse energy on surface roughness, as shown in Fig. 6, the roughness  
154 values of the samples are rapidly increased on increasing the discharge energy. It can also be realised that  
155 a  $0.2 \mu\text{m}$   $R_a$  value of the PMEDMed surface could be achieved as long as the discharge energy was  $\leq 17.5 \mu\text{J}$ .

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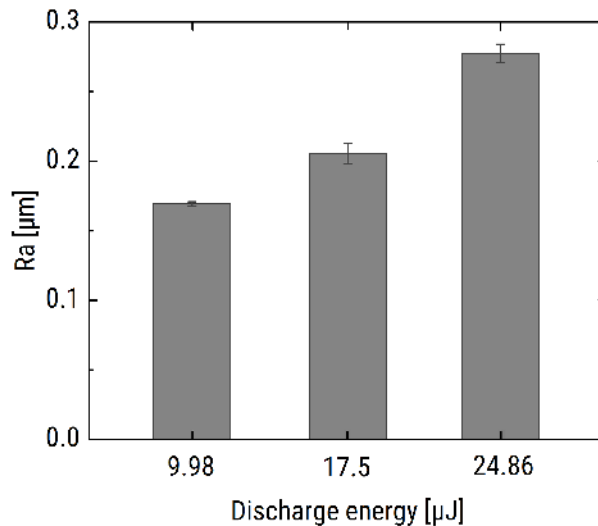


Figure 6:  $R_a$  surface roughness values of samples machined at various discharge energies using 25 g/l silver powder concentration.

157 *4.2. Silver content and in vitro antibacterial evaluation*

158 As can be seen in Fig. 7, powder concentration plays a vital role in the silver content of the coating  
159 layer, whereby a higher powder concentration results in an increase of the deposited silver. Consequently, 11  
160 samples machined and coated without silver and with silver contents from 1.58% to 9.61% were performed  
161 by applying 11 powder concentrations, thus allowing for the antibacterial evaluation based on the coating  
162 layer's silver content. The silver content's results from EDS analysis are semi-quantitative and include a  
163 possible  $\pm 0.2$  wt.% error. Nevertheless, these results are sufficient for the purpose of this study.

164 Fig. 8 shows microscopy evaluation of *S. aureus* bacteria expressing green fluorescent protein, in which  
165 each viable bacterium is visible as a green dot on the dark background at different silver contents of the

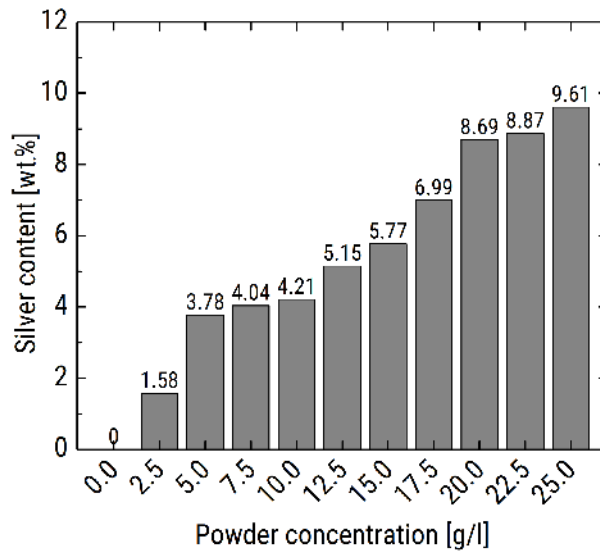


Figure 7: Silver content of the samples depending on powder concentration.

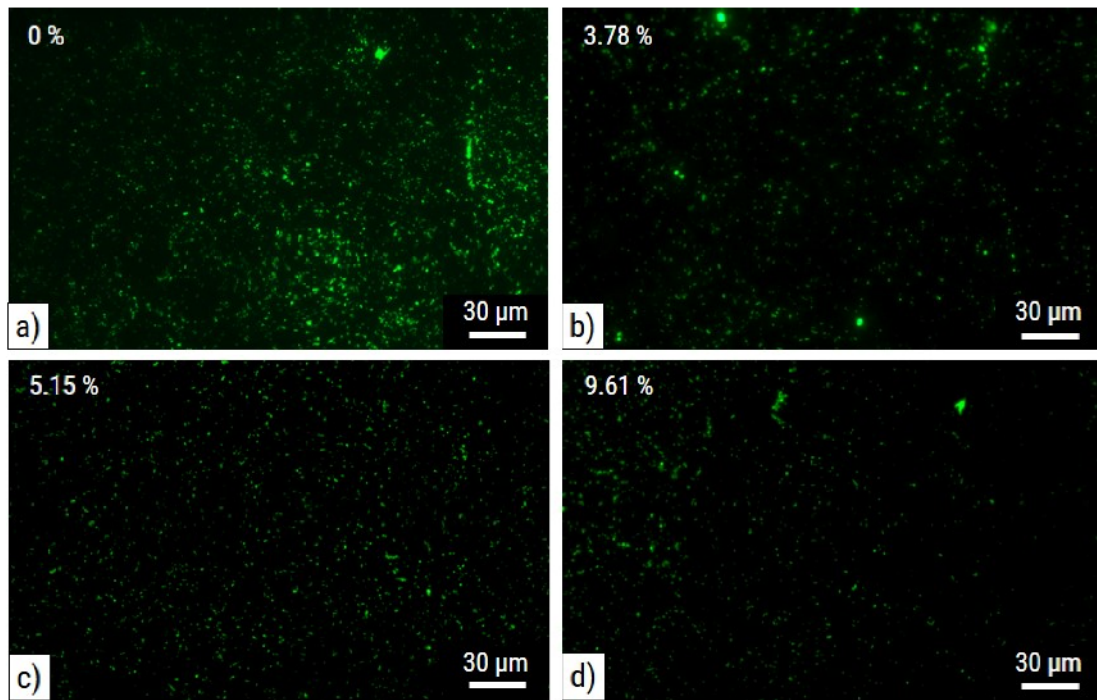


Figure 8: *S. aureus* bacterial mapping on the surfaces coated by silver contents of a) 0%, b) 3.78%, c) 5.15% and d) 9.61%

166 coated Ti-6Al-4V test plate. The bacteria were cultivated for 24 hours before evaluating the amount of  
 167 plate-adherent individual *S. aureus* bacteria and *S. aureus* clusters using epifluorescence microscopy. The  
 168 quantification of adherent bacteria in Fig. 9 and bacterial clusters containing a larger number of *S. aureus*  
 169 in Fig. 10 is shown in the graphs below. It was observed that the surfaces containing silver exhibit an

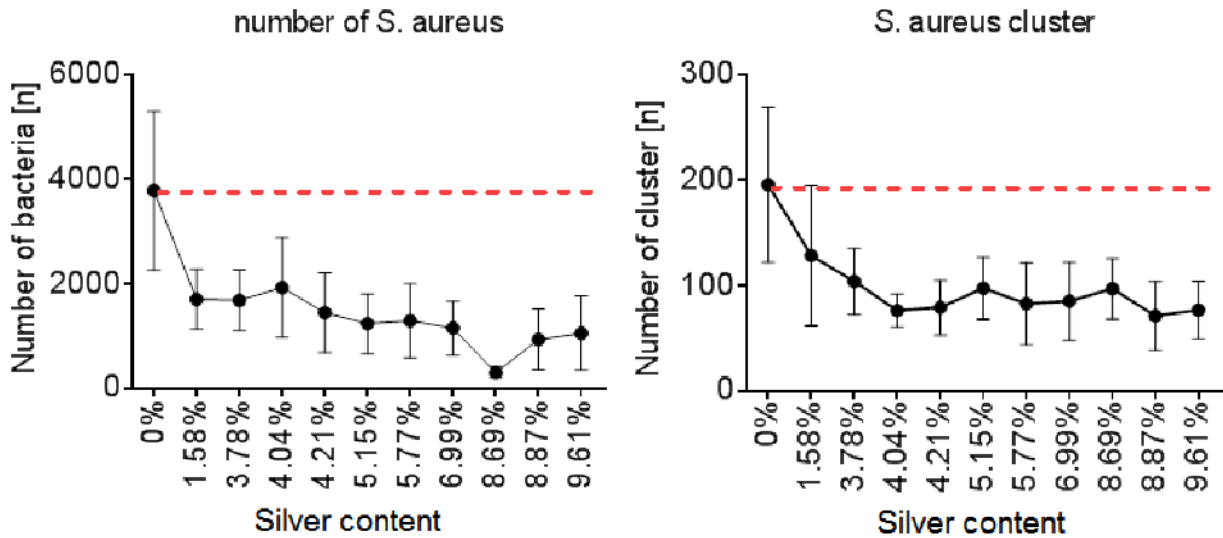


Figure 9: Influence of silver content on the number of bacteria. Figure 10: The number of clusters on the sample's surfaces.

170 antibacterial effect with a significant reduction of both individual bacteria as well as bacterial clusters. A  
 171 significant reduction in bacterial numbers and clusters was observed already at 3.78% of silver with no  
 172 significant further decrease in bacterial and cluster numbers at higher silver contents. Of note, some *S.*  
 173 *aureus* bacteria, as well as clusters, remained even at very high silver concentrations 9.61%.

174 It is known that bacteria can acquire a resistance towards silver mediated by the silver-resistance genes  
 175 (*silE*, *silS* and *silP*) in bacteria [36], which can also occur in *S. aureus* strains [37]. A recent study, however,  
 176 showed that the prevalence in silver resistance in *S. aureus* was only about 6% and restricted to the presence  
 177 of the *silE* gene. Interestingly the silver resistant bacteria remained sensitive against silver in the tested  
 178 wound dressing anyway [38]. Another study using a silver-hydroxyapatite coating on implants has also shown  
 179 that about 3% silver embedded in a hydroxyapatite matrix reduced, but did not fully prevent, Methicillin-  
 180 Resistant Staphylococcus Aureus (MRSA) biofilm formation [39], which is in line with the results from  
 181 this study. Silver has been shown to enhance the effectiveness of antibiotics [40], therefore, the incomplete  
 182 eradication of *S. aureus* in our experiments should not be a problem for the in vivo application, as normally  
 183 antibiotics are given after endoprosthetic joint replacement.

#### 184 4.3. Analysis of silver distribution on and in the coating layer

185 It can be seen from Fig. 11 that some silver was stuck on the machined surface. Since the samples were  
 186 cleaned in an ultrasonic bath for 10 minutes with the machined surface facing downwards, it can therefore  
 187 be concluded that this sticking phenomenon was not caused by settling of silver powder during the cleaning  
 188 process. Based on the PMEDM mechanism, it is possible to explain that some of the observed silver powder  
 189 was as a result of the spattering process, whereas the rest are from a fresh reflow of dielectric fluid which

190 occurred when the coating layer was still molten after discharge. Silver sticking on the machined surface is  
191 one of the major challenges facing PMEDM for medical applications.

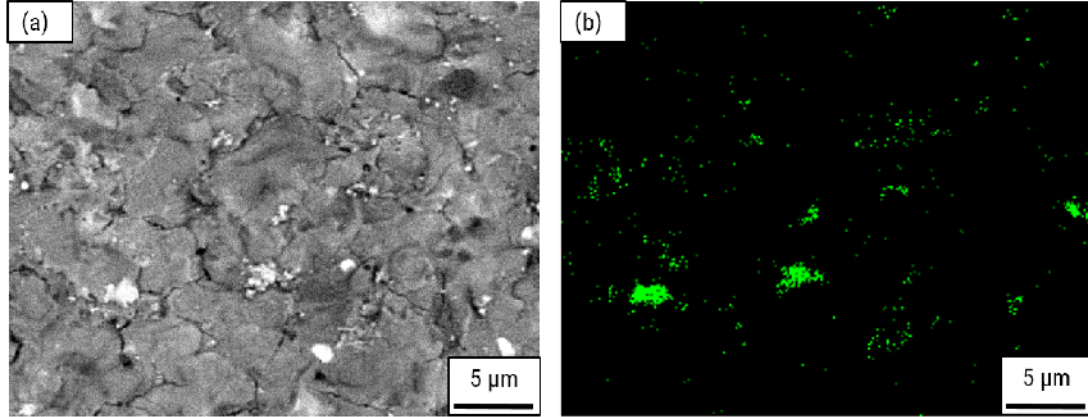


Figure 11: Distribution of silver on the coating layer for samples machined using dielectric fluid with 15 g/l powder concentration: (a) SEM image and (b) silver mapping (identical area).

192 As shown in Fig. 12, the coating layer's cross section was analysed. It can be established that it is  
193 possible to coat a continuous layer on the Ti-6Al-4V surface. EDS analysis shows that materials from the  
194 workpiece, tool electrode as well as the dielectric fluid are contained in the coating layer. Furthermore, Figs.  
195 11 and 12 show that silver is not only deposited on the machined surface, but also implanted in the recast  
196 layer. During machining, owing to a very high temperature in the plasma channel (approximately 10.000 °K)  
197 as well as a thermal evaporation process, silver is molten and mixed with other molten material from tool  
198 electrode and workpiece, as well as mixed with carbon from the hydrocarbon-based dielectric fluid. After  
199 discharge, the coating layer is formed through a re-solidification process. Silver contained in the coating layer  
200 can be categorized into three morphologies: alloying of molten silver with the molten tool and workpiece  
201 materials; embedded silver particles on and into the coating layer; and spattered silver re-solidified on the  
202 coating surface. Therefore, the silver distribution of the coating layer is complex. However, silver content  
203 decreases within the coating layer towards the substrate surface.

204 In order to analyse the coating layer thickness, a Keyence VK9700 3D Laser scanning confocal-microscope  
205 was used to scan the surface after which a Keyence VK analyser software was used to analyse the coating  
206 layer's cross sectional thickness as shown in Fig. 13. Fifty areas with different coating thicknesses were  
207 analysed thus allowing for a calculation of an average thickness as well as standard deviation. From the  
208 results, it can be established that a  $2.49^{-0.5}$  μm thick coating layer could be coated by using a 17.5 μJ  
209 discharge energy.



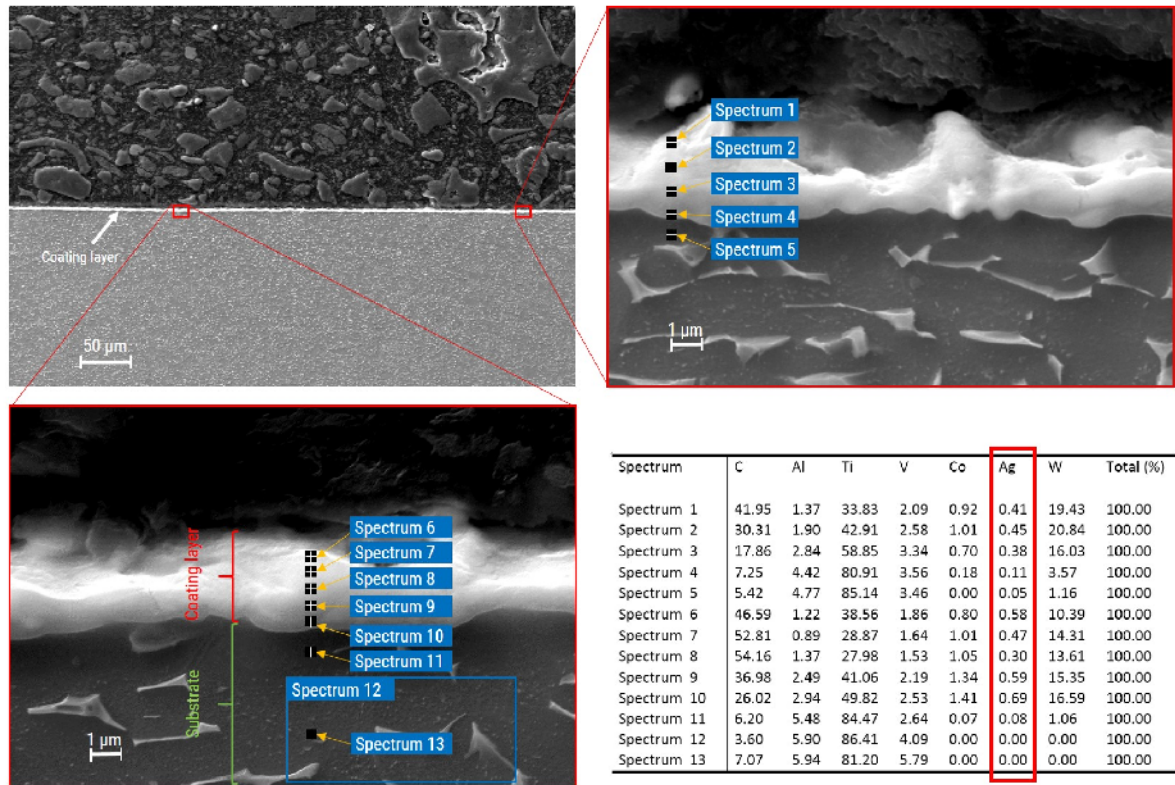


Figure 12: SEM and EDS analysis of the coating layer showing how silver content varies with coat depth.

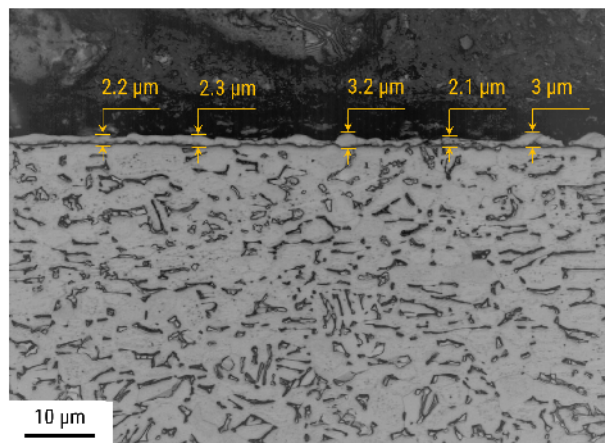


Figure 13: Coating layer thicknesses for a sample machined and coated using a 17.5  $\mu\text{J}$  discharge energy.

#### 220 4.4. Analysis of the coating layer's microhardness

221 A nanoindenter was used to measure the microhardness of both Ti-6Al-4V substrate as well as the  
 222 coating surfaces varying silver contents. The tested nanoindentation positions on the sample are shown  
 223 in Fig. 14. The indentation point was also scanned by a Keyence VK9700 3D Laser scanning confocal-

214 microscope. Afterwards, MountainsMap 7.4 scanning topography software was used to analyse and check  
 215 the indentation depth. From Fig. 14, a  $\approx 0.7 \mu\text{m}$  depth of the indentation could be realised.

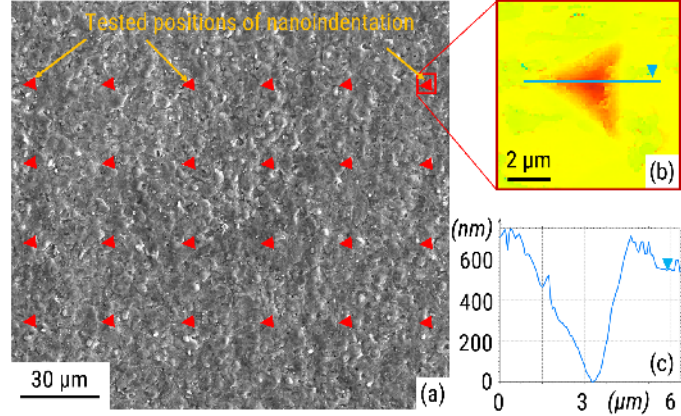


Figure 14: (a) Sample surface showing nanoindentation positions (in red color) used to test for microhardness, (b) the nanoindentation's graph and (c) depth profile.

216 As shown in Tab. 2, the microhardness of Ti-6Al-4V is 379.18 (HV). After EDM without powder, a  
 217 coating layer with a 528.39 (HV) microhardness was coated on the substrate surface. By using nano-silver  
 218 particles mixed into the dielectric fluid in PMEDM, a reduction of the coating layer's microhardness, due  
 219 to the deposition of silver, could be established, whereby an increase of silver content reduces the surface  
 220 hardness.

Table 2: Microhardness values for both Ti-6Al-4V substrate as well as the coating layers containing various silver contents

	Ti-6Al-4V substrate	The coating layers containing different silver contents		
		0 (wt.%) Ag	3.78 (wt.%) Ag	9.61 (wt.%) Ag
Microhardness (HV)	379.18	528.39	527.19	521.62

#### 221 4.5. Analysis of the process efficiency

222 It is clear that EDM without powder has been widely used for machining medical devices. However, long  
 223 machining time is still a challenge for this process. The addition of silver nano-particles into the dielectric  
 224 fluid, in this study, is used not only for machining Ti-6Al-4V, but also for reducing the machining time. The  
 225 results showed that MRR, compared to the EDM without powder, is significantly enhanced by suspending  
 226 powder particles in a hydrocarbon-based dielectric fluid. For example, as can be seen in Fig. 15, the material  
 227 removal rate is improved from  $\approx 0.007$  to  $\approx 0.02$  ( $\text{mm}^3/\text{min}$ ) by using 10 g/l silver powder concentration.

228 In explanation, this increase can be attributed to fact that the addition of silver powder reduces dielectric  
 229 resistivity, resulting in an enlargement of the machining gap, which enhances flushing conditions. Simulta-

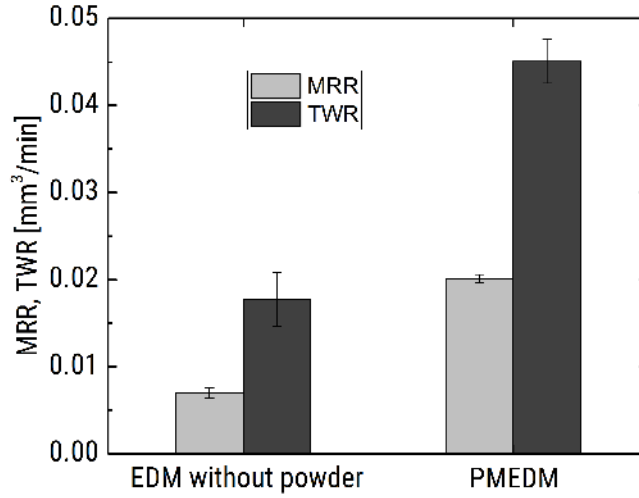


Figure 15: Comparison of material removal rate (MRR) and tool wear rate (TWR) between EDM without powder and PMEDM using 10 g/l silver powder concentration and 9.98  $\mu\text{J}$  discharge energy.

neously, the presence of silver particles in the discharge gap facilitates better spark ignition. Therefore, the number of pulses, which is a primary reason for the improvement in MRR, is increased.

In order to prove this explanation, during the experiments, pulse signals were recorded for 20 ms using a Tektronix DPO4101 digital oscilloscope with a TCP312 current and P6139A voltage probe. Afterwards, MATLAB R2017 software was utilised to analyse these signals. The results show that the pulse number per time of PMEDM is higher than that of EDM without powder (Fig. 16). The number of pulses for 20 ms, as compared to EDM without powder, increases from 1758 to 5090 by using 10 g/l powder concentration.

This pulse increase affects both MRR and TWR whereby, owing to the relative erosive wear between electrodes, as compared to EDM without powder, TWR using PMEDM is also significantly increased (Fig. 15). However, wear ratio, which is calculated by the quotient of TWR per MRR, is reduced from  $\approx 2.54$  to  $\approx 2.32$ .

As part of the PMEDM efficiency on machining, surface roughness and structure, on the other hand, due to the influence on the bacterial adhesion and growth, are important factors that needed to be analysed.

As can be seen in Tab. 3, the surface roughness of the samples machined by EDM without powder and PMEDM using varying powder concentrations and 9.98  $\mu\text{J}$  pulse energy are represented. It can be realized that all samples have surface roughness values not larger than 0.2  $\mu\text{m}$  in  $R_a$  which fits with the aforementioned designed surface roughness upper threshold. Other surface roughness parameters such as  $R_z$  and  $R_{pk}$  are also vital as they have a relationship with the adhesion of bacteria [41].

Furthermore, surface roughness values of all samples are relatively similar. This similarity of the surface roughness values is significant for further antimicrobial evaluation since it allows to attribute any differences on the observed antibacterial properties to the silver content. If the surface roughness values were significantly varied, then their contribution to bacterial adhesion and accommodation would have to be separately

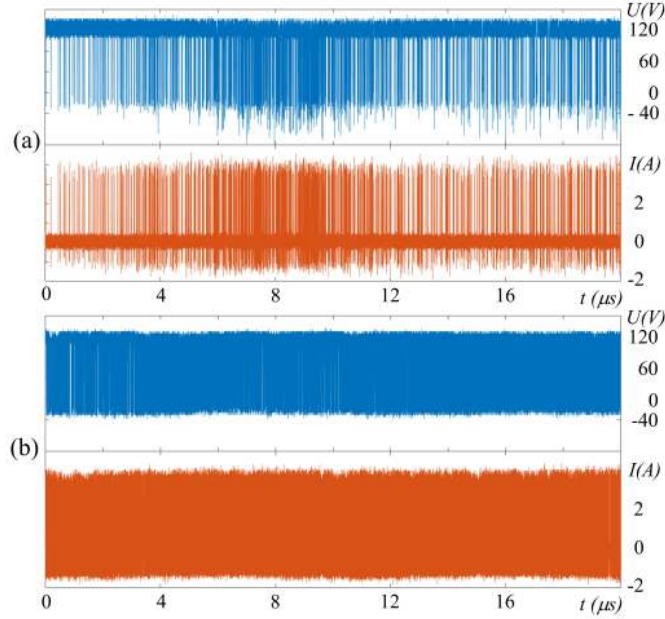


Figure 16: Current and voltage waveforms: (a) EDM without powder and (b) PMEDM with 10 g/l powder concentration.

Table 3: Surface roughness of the samples machined by EDM without powder and PMEDM with different powder concentrations, using 9.98  $\mu\text{J}$  discharge energy.

Powder concentration (g/l)	0	2.5	5	7.5	10	12.5	15	17.5	20	22.5	25
$R_a$ ( $\mu\text{m}$ )	0.2	0.169	0.173	0.166	0.155	0.167	0.181	0.162	0.169	0.159	0.171
$R_z$ ( $\mu\text{m}$ )	1.45	1.25	1.14	1.26	1.21	1.33	1.23	1.27	1.14	1.23	1.15
$R_{vk}$ ( $\mu\text{m}$ )	0.283	0.227	0.218	0.22	0.214	0.215	0.223	0.206	0.231	0.203	0.221

251 factored into the analysis.

252 Although having relatively similar surface roughness values by varying powder concentration suspended in  
 253 the dielectric fluid, surface structures of the samples should be analysed. EDMed surface structures are very  
 254 complex because they are formed by a combination of overlapping craters and layers of spattered material  
 255 from the discharges. Consequently, in order to analyse surface structure of the samples, MountainsMap 7.4  
 256 scanning topography software was used. Fig. 17 shows 3D images of the machined surfaces with various  
 257 silver concentrations mixed into the dielectric fluid (Fig. 17b, c and d) and without powder (Fig. 17a). It  
 258 can be seen that the surface machined using EDM without powder has slightly bigger and higher peaks than  
 259 PMEDMed surfaces. However, there is no obvious difference in the surface structure of the samples.

260 Based on the results regarding material removal rate, tool wear, the surface roughness and structure, as  
 261 well as the deposited silver content, it can be established that the machining time of PMEDM is significantly



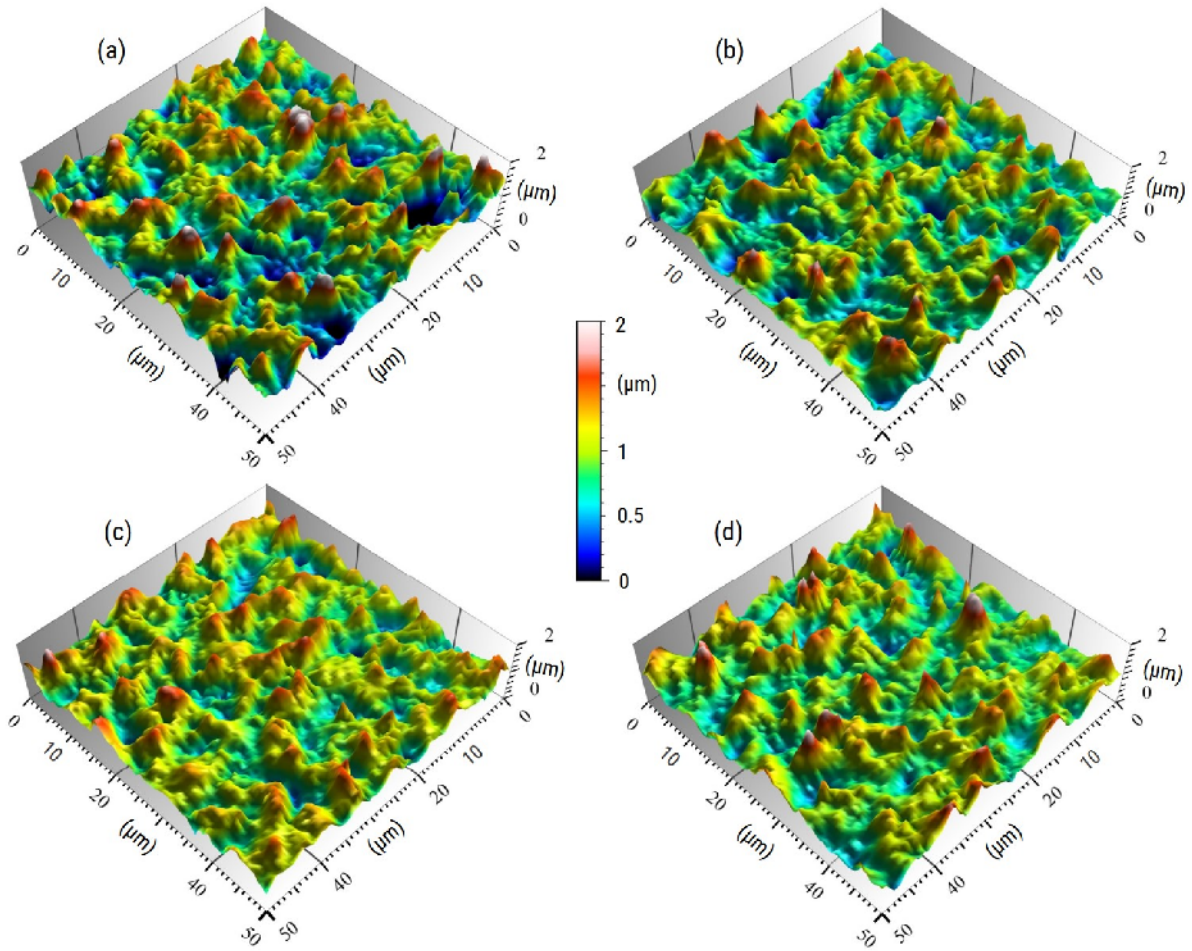


Figure 17: 3D images: (a) EDMed surface machined without powder; and PMEDMed surfaces machined using (b) 5 g/l, (c) 15 g/l and (d) 25 g/l silver powder concentrations.

262 reduced as compared to EDM without powder, whereas the surface quality is similar. Therefore, PMEDM  
 263 has demonstrated its ability in not only improving the machining efficiency, but also in coating layers with  
 264 controllable silver contents on the workpiece surface, while at the same time resulting in negligible changes in  
 265 surface roughness and structure. All of these abilities of PMEDM have an effect on antibacterial properties.  
 266 Therefore, it is possible to concurrently machine and coat surfaces fulfilling specific requirements regarding  
 267 both silver content and surface quality using the PMEDM method.

## 268 5. Conclusions

269 In this study, the capability of PMEDM using silver nano-particles in machining Ti-6Al-4V medical  
 270 material and coating an antibacterial layer on the machined surface has been investigated. From the results,  
 271 the following conclusions can be drawn:

- 272 • It is possible to realise a continuous coating layer incorporating constituent elements from tool elec-  
273 trode, workpiece, dielectric fluid as well as powder particles which were deposited during the machining  
274 and coating process.
- 275 • Silver content in the coating layer tends to decrease with increasing depth of coat.
- 276 • Both qualitative and quantitative antibacterial results have demonstrated an excellent antibacterial  
277 property of the surfaces which are machined and coated by the PMEDM method.
- 278 • Silver content in the coating layer plays a vital role in reducing the amount of not only *S. aureus*, but  
279 also bacterial cluster on the coating surfaces.
- 280 • Machining efficiency is significantly enhanced by suspending silver particles into dielectric fluid. It is  
281 possible to increase material removal rate of EDM process up to  $\approx 286\%$  whereas wear ratio is reduced  
282 by  $\approx 8.7\%$ .
- 283 • PMEDM has demonstrated a considerable potential to concurrently machine geometry's part and coat  
284 antibacterial layers on medical devices.

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