

Application of Dimensional Analysis in Calibration of a Discrete Element Model for Rock Deformation and Fracture

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Summary

A discrete element approach was used in the simulation of rock fracture. The numerical synthetic material was made of rigid circular particles or cylinders that have interaction through normal and shear springs. The cylinders were bonded to each other at the contact points to withstand the applied loads. To characterize the microscopic properties of this synthetic material, a dimensional analysis approach was presented. It was shown that the dimensionless parameters and graphs obtained were useful tools for fast and efficient calibration of a synthetic material. This calibration method was employed for finding a numerical model for Pennsylvania Blue Sandstone. The numerical model could mimic many deformational and failure characteristics of the sandstone in both conventional and some non-conventional stress paths.

Keywords: Rock fracture, discrete element method, dimensional analysis, calibration.

List of Symbols

D	specimen width and specimen diameter
E	Young's modulus
E'	apparent Young's modulus
k_n	normal stiffness of a contact
k_s	shear stiffness of a contact
n_b	normal bond of a contact
p	maximum applied load in a Brazilian test and average of maximum and minimum principal stresses
q	half of difference between maximum and minimum principal stresses

q_u	uniaxial compressive strength
R	cylinder radius
R_{ave}	average radius of cylinders
R_{max}	maximum radius of cylinders
R_{min}	minimum radius of cylinders
s_b	shear bond of a contact
t	specimen length in a Brazilian test
$\Delta\varepsilon_{axial}$	axial strain increment
$\Delta\varepsilon_{lateral}$	lateral strain increment
$\Delta\sigma_{axial}$	axial stress increment
ϕ	friction angle
μ	friction coefficient of a contact
σ_0	genesis pressure
σ_1	axial stress and maximum principal stress
σ_3	confining pressure and minimum principal stress
σ_t	Brazilian tensile strength
ν	Poisson's ratio
ν'	apparent Poisson's ratio

1. Introduction

The issue of rock deformation and fracture has been studied intensively by researchers in the past. Two general paths are usually followed to investigate rock fracture, a continuum model and a discrete model. In the continuum model, different techniques are implemented; the most common ones are softening plasticity and damage mechanics models. In the softening approach, rock friction angle or cohesion can be considered to evolve by an invariant of plastic strain or plastic work (Vermeer and De Borst, 1984), while in the damage model, rock stiffness is reduced with deformation (De Borst, 2002). In the continuum framework, the geomaterial failure and localization of deformation can be studied as a bifurcation phenomenon as well (Rudnicki and Rice, 1975; Vardoulakis and Sulem, 1995).

Crack propagation in rock features a complex behavior that usually needs to be studied numerically. The difficulty with the numerical rock softening and damage analysis is the dependency of the failure load on the mesh resolution. For example, it is well known that in the finite element analysis of rock softening, by reducing the mesh size, the dissipated energy in the fracture zone or shear band reduces which is not physically meaningful (Bazant and Planas, 1998). To remedy this difficulty, non-local continuum (Bazant and Pijaudier-Cabot, 1988), Cosserat continuum (De Borst and Muhlhaus, 1991) and gradient plasticity techniques (De Borst and Muhlhaus, 1992) have been proposed in the literature.

Numerical continuum methods based on micromechanics and flaw distribution have been proposed as well. In a multi-laminate model (Pande and Xiong, 1982), discrete failure planes are considered in otherwise a continuum system. The original multi-laminate model later on was renamed a microplane model in that a geometric damage tensor was introduced (Carol et al., 1991). In a microplane model, a non-local averaging technique needs to be implemented to avoid the sensitivity of the results to mesh resolution. A displacement discontinuity technique was used by Van de Steen et al. (2003) to model rock fracture. This technique was successfully implemented in simulating rock spalling around an underground excavation, although no comment

was presented regarding calibration of the model. Fractures can be studied as mathematical cracks as well. In this case, linear fracture mechanics or cohesive crack models (Bazant and Planas, 1998) can be used, although extension of these models to biaxial compressive loading is not straightforward.

In the discrete element approach of rock fracture, the interaction of particles is considered explicitly in the model. This provides an embedded internal length in the formulation of the problem that prevents the numerical difficulties observed in the classical local continuum softening models. The application of this technique in geomechanics was initially proposed by Cundall and Strack (1979) and later on extended and implemented in the PFC computer programs (Itasca, 1999). With this technique, crack propagation, rock softening and dilation, anisotropic rock behavior in damaged zone, cyclic energy dissipation, spontaneous emission of acoustic energy and Coulomb failure envelope can be reproduced. Nevertheless, this powerful technique has a few shortcomings. Firstly, it needs calibration, i.e. some micro-mechanical properties must be specified to result in a synthetic material with desired macro-mechanical properties such as Young's modulus, Poisson's ratio, tensile strength, and failure envelope. For a simple contact bond model implemented in this paper, the micro-mechanical properties for interaction of two circular particles or cylinders are normal and shear spring stiffnesses (k_n , k_s), normal and shear bonds (n_b , s_b) to glue the particles in contact and Coulomb friction coefficient (μ) which is activated when the bond between any two particles is broken. Therefore, these five parameters must be obtained during the calibration step. The calibration can be performed by using a trial and error approach in that extensive effort is needed to acquire appropriate micro-parameters to result in macroscopic properties similar to those of a specific rock. A more efficient method was employed by Huang (1999) for calibration of a circular particle interaction (CPI) model in that a dimensional approach was established and some dimensionless parameters were introduced.

The second problem with a CPI model is the difficulty of calibrating it for the failure envelope. In general, a CPI model shows a friction angle which is smaller than that for a hard rock (Potyondy and Cundall, 2001). To increase the friction angle of a particle assemblage, non-circular particles can be used (Ting et al., 1993), but this requires additional computational time to detect contact of adjacent particles; contact detection of circular particles is much faster compared to that of other particle shapes.

The third difficulty of a CPI model is ratio of unconfined compressive strength to tensile strength which is usually lower than that of a rock. Fakhimi (2004) proposed a slightly overlapped circular particle interaction (SOCPI) model to resolve the above second and third problems. The calibration issue of a SOCPI material that is based on the dimensional analysis is addressed in this paper. The calibration graphs introduced were used to develop a synthetic model for Pennsylvania Blue Sandstone. Several physical and numerical tests are reported to verify the validity of the numerical model.

All the numerical analyses in this paper were conducted using CA2 computer program which was developed by the first author (Fakhimi, 1998). CA2 is a hybrid finite difference-discrete element code for large deformation static or dynamic deformation of solid materials. The first step in discrete element modeling of rock, using CA2, is sample preparation. This step involves with generation of walls that define the domain of analysis. The walls in CA2 are not rigid but are discretized to smaller elements and are used to confine the cylinders. Next, the cylinders with uniformly

distributed radii are generated and inflated by increasing the cylinders radii. An explicit algorithm is used to obtain the equilibrium configuration of this particle assembly. At this stage of analysis, a contact bond model with no friction and bond at the contact points is assumed for the interaction of cylinders. After equilibrium, the induced stress in the surrounding walls is examined. By changing the number of cylinders, the induced stress can be modified. This stress is called genesis pressure (σ_0) and its role is to create a small overlap between cylinders in contact. This small overlap is responsible for obtaining more realistic friction angle and ratio of compressive to tensile strength of the synthetic material (Fakhimi, 2004). The last step in sample preparation is to initialize the normal and shear contact stresses to zero and to introduce normal and shear bonds and friction between the cylinders in contact. More details about sample preparation are given in Fakhimi (2004).

2. Calibration Charts

Calibration of a synthetic model made of rigid circular cylinders is a tedious task if it is undertaken by using a trial and error approach. A much more convenient and efficient method is to introduce some calibration curves using a dimensional analysis. In this paper, a synthetic material was intended to be calibrated for Young's modulus, Poisson's ratio, failure envelope and tensile strength. A SOCPI model is calibrated by assigning appropriate parameters including normal and shear spring constants (k_n , k_s), normal and shear bonds (n_b , s_b), friction coefficient (μ), genesis pressure (σ_0), average cylinders radius (R) and specimen width (D).

The parameters that dictate the elastic properties of a SOCPI material are k_n , k_s , σ_0 , R and D . These five parameters have two independent dimensions, i.e. force (F) and length (L). Hence, considering the p theorem in dimensional analysis (Sedov, 1993), $5 - 2 = 3$ dimensionless parameters are involved in describing the elastic deformation of the synthetic material, i.e.

$$E' = k_n f(R/D, k_s/k_n, \sigma_0/k_n), \quad (1a)$$

$$\nu' = g(R/D, k_s/k_n, \sigma_0/k_n), \quad (1b)$$

where f and g are dimensionless functions and E' and ν' are apparent Young's modulus and Poisson's ratio of the synthetic material, defined in a uniaxial test as:

$$E' = \Delta\sigma_{\text{axial}} / \Delta\varepsilon_{\text{axial}} \quad (2a)$$

$$\nu' = -\Delta\varepsilon_{\text{lateral}} / \Delta\varepsilon_{\text{axial}}. \quad (2b)$$

If a numerical experiment is considered as a biaxial plane strain test, true Young's modulus and Poisson's ratio of synthetic material can be obtained from the following equations:

$$\nu = \nu' / (1 + \nu'), \quad (3a)$$

$$E = E' (1 - \nu^2). \quad (3b)$$

Several numerical uniaxial tests were conducted on the synthetic material with $R/D = 0.013$ and different k_s/k_n and σ_0/k_n values, using CA2 computer program.

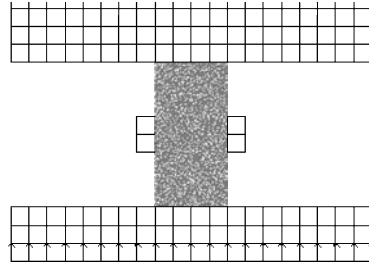


Fig. 1. Numerical unconfined compressive test set-up

The numerical uniaxial test set up together with the synthetic specimen is shown in Fig. 1. The upper and lower platens or walls are made of finite difference grid and are frictionless. The upper wall is fixed vertically while the lower one was moved with a constant velocity to compress the specimen. Two small and very flexible walls were glued to the lateral sides of the specimen. These walls were like measurement devices to ac-

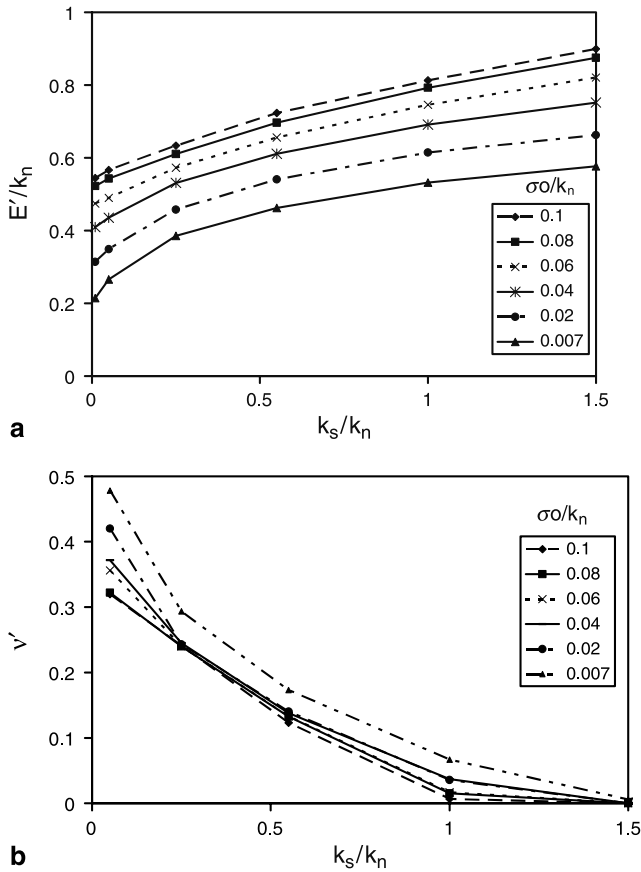


Fig. 2. Elastic constants of synthetic material versus k_s/k_n for different σ_0/k_n , **a** normalized apparent Young's modulus, **b** apparent Poisson's ratio

quire average lateral deformation of the middle of the specimen and consequently the Poisson's ratio. The results of the numerical analyses are shown in Fig. 2a and b. Parametric studies and also the work by Potyondy and Cundall (2001) show that E' and ν' are almost independent of R/D values provided that this ratio is kept small enough. Therefore, by introducing the macroscopic parameters E' and ν' and assuming a value for σ_0/k_n , normal and shear spring constants can be estimated from Fig. 2a and b.

The uniaxial strength of the synthetic material (q_u) was obtained by conducting several numerical tests. This parameter is a function of σ_0 , k_n , k_s , R , D , n_b , s_b , and μ . These eight parameters have two dimensions. Hence, dimensionless unconfined compressive strength can be expressed as:

$$q_u R/n_b = f_1(\sigma_0/k_n, k_s/k_n, k_n R/n_b, n_b/s_b, R/D, \mu), \quad (4)$$

where f_1 is a dimensionless function. Numerical analyses demonstrate that q_u is a weak function of μ and $k_n R/n_b$ (Villegas, 2004). In addition, it has been shown that the biaxial compressive strength of a synthetic material made of circular cylinders is not a function of R/D value, provided that R/D is kept a small number (Potyondy and

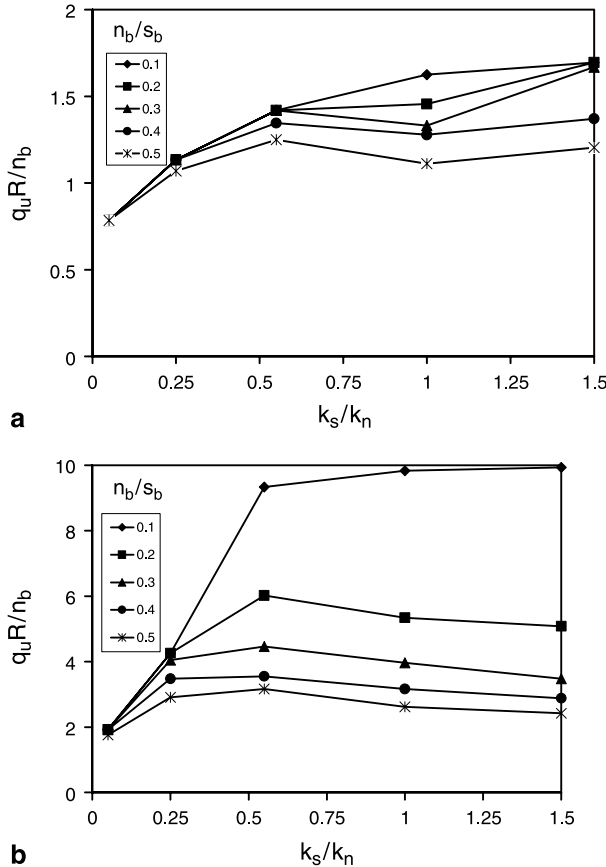


Fig. 3. Dimensionless uniaxial strength versus k_s/k_n for **a** $\sigma_0/k_n = 0.007$ and **b** $\sigma_0/k_n = 0.08$

Cundall, 2001). Thus, only three dimensionless parameters namely, σ_0/k_n , k_s/k_n and n_b/s_b were needed to consider. By conducting uniaxial tests on the synthetic materials with $R/D=0.013$ ($R_{\min}=0.44$ mm and $R_{\max}=0.58$ mm with a uniform random distribution of cylinders radii), $\mu=0.5$, $k_n=50$ GPa, and $n_b=20,000$ N/m, two charts, each for a specific value of σ_0/k_n (0.007 and 0.08) were obtained. Figure 3a and b show these charts where the dimensionless uniaxial strength is plotted versus k_s/k_n for different n_b/s_b values. Similar charts for $\sigma_0/k_n=0.04$ and 0.1 are reported in Villegas (2004). Comparison of these figures verifies that the dimensionless unconfined compressive strength of synthetic material increases as higher values of genesis pressure is used. This should be expected, as with a higher genesis pressure, the number of contact points per unit area of the specimen increases which results in a stronger specimen. It is interesting to observe the effect of k_s/k_n parameter on the specimen strength. Figure 3 indicate that by reducing the k_s/k_n parameter to a value less than 0.5, the dimensionless strength decreases. The crack pattern, in the synthetic material is also affected by the k_s/k_n ratio. This is demonstrated by the crack pattern for the synthetic specimens with different k_s/k_n value (Fig. 4). A crack in CA2 is a line perpendicular to the line connecting centers of cylinders with broken bond. It is evident from Fig. 4 that for a small k_s/k_n value, a diffuse crack pattern is obtained, in contrast to a shear banding when a greater value of k_s/k_n is used. The reason for this observation is that by increasing k_s value, greater microscopic shear stresses and hence shear cracks are developed in the specimen with the consequence of shear banding. In addition, it can be shown that a synthetic specimen with a small k_s/k_n value, features a substantial amount of hardening in its stress-strain diagram, while a greater value of k_s/k_n , results in a more brittle behavior (Villegas, 2004).

Several numerical biaxial tests were conducted to evaluate failure envelope of synthetic material. Strength of synthetic material (σ_1) is a function of σ_0 , R , D , μ , k_s ,

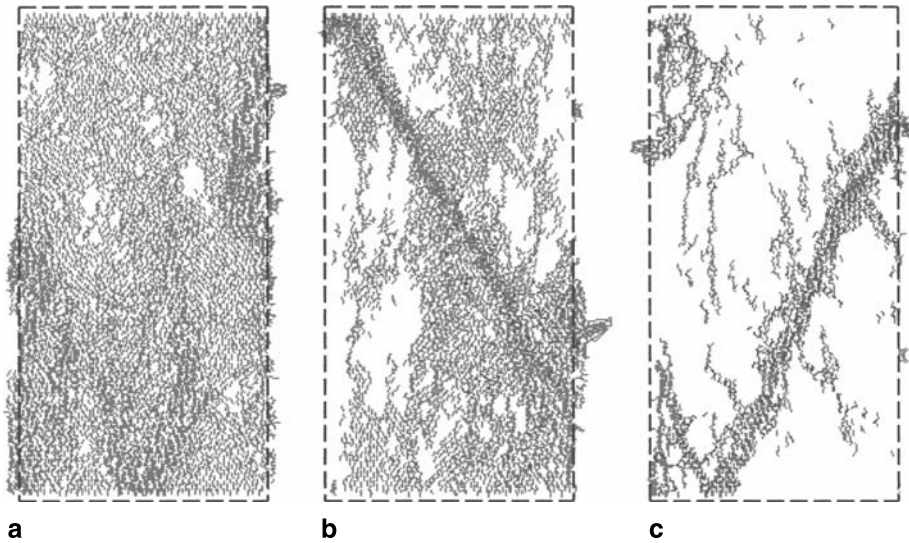


Fig. 4. Crack pattern in the damaged specimen with **a** $k_s/k_n=0.05$, **b** $k_s/k_n=0.55$, and **c** $k_s/k_n=1.5$

k_n , n_b , s_b , and the confining pressure (σ_3). Similar to uniaxial strength, σ_1 is a weak function of μ and $k_n R/n_b$. Therefore, the biaxial dimensionless strength can be written as:

$$\sigma_1 R/n_b = g_1(\sigma_3 R/n_b, k_s/k_n, \sigma_0/k_n, n_b/s_b), \tag{5}$$

where g_1 is a dimensionless function. The normalized failure envelopes for different values of σ_0/k_n are shown in Fig. 5a–c. In this series of numerical anal-

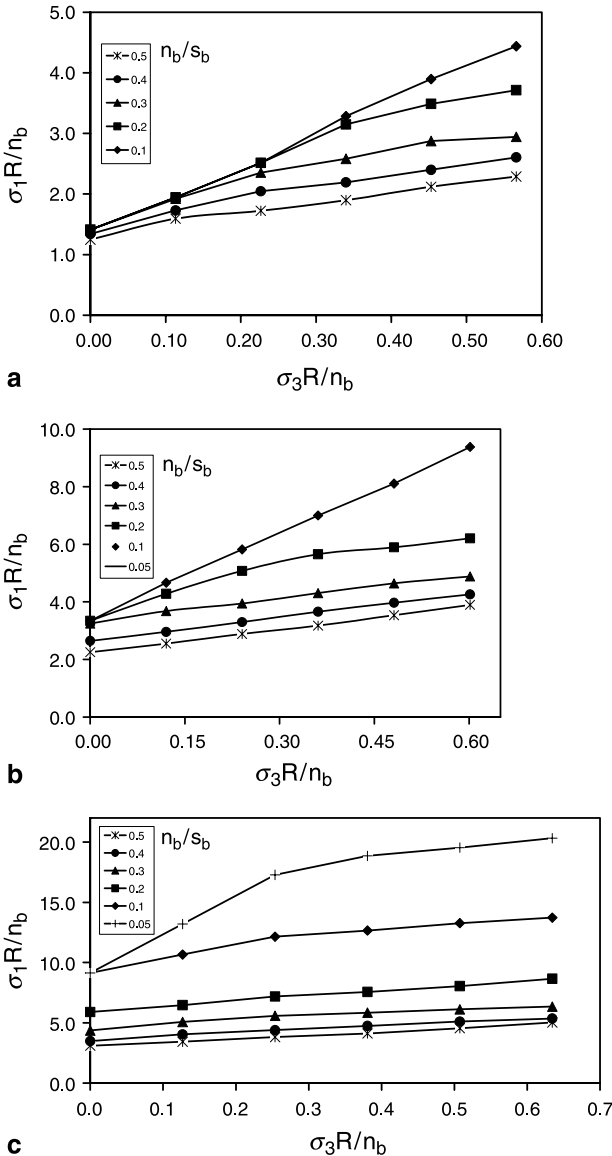


Fig. 5. Normalized failure envelopes for **a** $\sigma_0/k_n = 0.007$, **b** $\sigma_0/k_n = 0.04$, and **c** $\sigma_0/k_n = 0.08$

yses, the following parameters were fixed: $R/D = 0.013$ (with $R_{\min} = 0.44$ mm and $R_{\max} = 0.58$ mm), $\mu = 0.5$, $k_n = 50$ GPa, $k_s = 27.5$ GPa and $n_b = 20,000$ N/m. Different values of n_b/s_b (0.05, 0.1, 0.2, 0.3, 0.4, 0.5) were used in the analyses. Figure 5 indicate that by increasing the genesis pressure, specimens with higher strength are obtained. In addition, it is noticed that with larger σ_0/k_n values together with smaller values of n_b/s_b parameter, curvature of failure envelope similar to that of a rock can be simulated. The slope of failure envelope in a σ_1, σ_3 space is equal to $\tan^2(45 + \phi/2)$ in that ϕ is the friction angle. Inspection of the curves in Fig. 5 for higher σ_0/k_n and smaller n_b/s_b values show that friction angles of 60° or more can be simulated. This has become feasible due to the application of non-zero genesis pressure.

To investigate the role of microscopic friction coefficient (μ) on the failure envelope, two sets of biaxial tests with $\sigma_0/k_n = 0.08$, $k_s/k_n = 0.55$, $n_b/s_b = 0.2$, $k_n R/n_b = 1269$, and $R/D = 0.013$ were conducted. For the first set of biaxial tests, a friction coefficient of 0.5 was used while for the second set, μ was 0.8. Figure 6 shows the peak and residual failure envelopes for these two synthetic materials. This figure demonstrates that while μ has a small effect on the peak failure envelope, its role on the residual strength is negligible. Consequently, a great percentage of the induced friction, in the numerical model, is due to the shear displacement along the developed irregular failure surface and to a much less extent the microscopic friction angle (μ) plays a role.

Some numerical Brazilian tests were performed to evaluate tensile strength of synthetic material. The numerical test set up for the Brazilian test is shown in

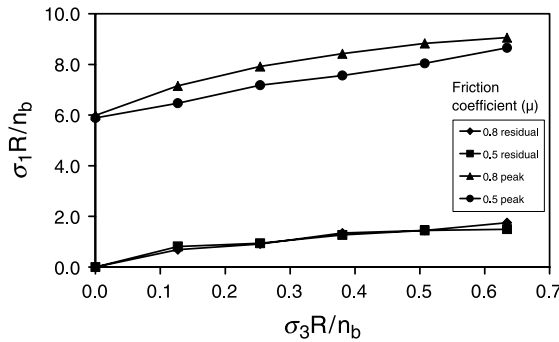


Fig. 6. Normalized peak and residual failure envelopes for two different friction coefficients

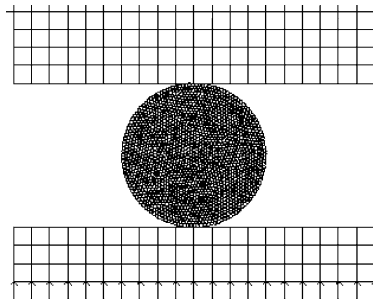


Fig. 7. Numerical Brazilian test set-up

Fig. 7. The upper platen is fixed in the vertical direction while the lower platen was moved with a constant velocity of 0.2×10^{-8} m/step in the upward direction. Both upper and lower platens were discretized to the smaller finite elements. Two series of numerical Brazilian tests were conducted. In the first series, for specific values of σ_0/k_n , the tensile strength was obtained by changing k_s/k_n and n_b/s_b parameters. The tensile strength was obtained from the following equation (Goodman, 1989):

$$\sigma_t = 2p/\pi Dt, \quad (6)$$

where p is the maximum applied load, D is the specimen diameter and t is its length that is equal to unity in the numerical tests. Figure 8a and b show the dimensionless tensile strength ($\sigma_t R/n_b$) versus k_s/k_n parameter for $\sigma_0/k_n = 0.1$ and 0.04 , respectively. Similar to the observation with the unconfined compressive strength, it is evident that increase in genesis pressure results in higher tensile strength. The reason for the observed maximum points of dimensionless tensile strength as shown in Fig. 8a is not clear to the authors. One possible explanation is that by increasing n_b/s_b parameter with higher values of σ_0/k_n , the intensity of micro shear cracks are increased in the model that results in change of the trend of the failure load.

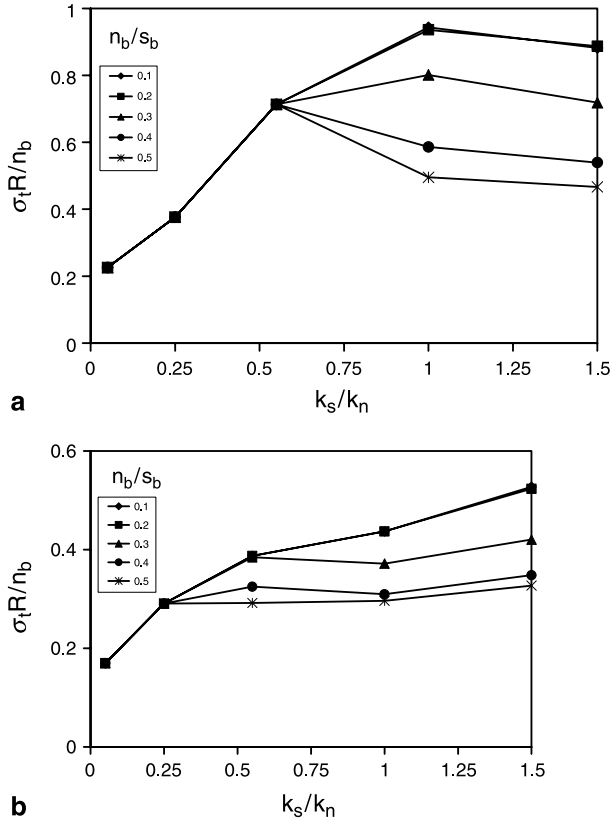


Fig. 8. Dimensionless Brazilian tensile strength versus k_s/k_n for **a** $\sigma_0/k_n = 0.1$ and **b** $\sigma_0/k_n = 0.04$

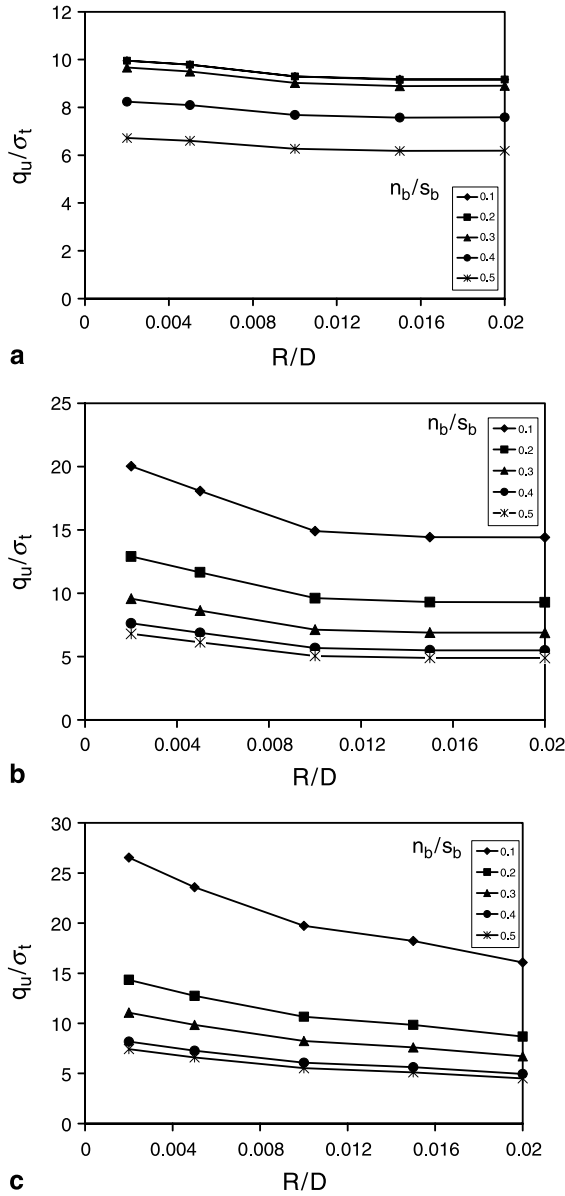


Fig. 9. Ratio of unconfined compressive strength to Brazilian tensile strength versus specimen resolution for **a** $\sigma_0/k_n = 0.04$, **b** $\sigma_0/k_n = 0.08$, and **c** $\sigma_0/k_n = 0.1$

In the second series of Brazilian tests, the size effect was investigated. Figure 9 show the ratio of unconfined compressive strength to tensile strength versus R/D for σ_0/k_n of 0.04, 0.08 and 0.1. Since unconfined compressive strength appears to be independent of R/D , it is realized that σ_t reduces by decreasing R/D value, for a fixed n_b/R . This means that for a numerical model with fixed average cylinder radius (R), by

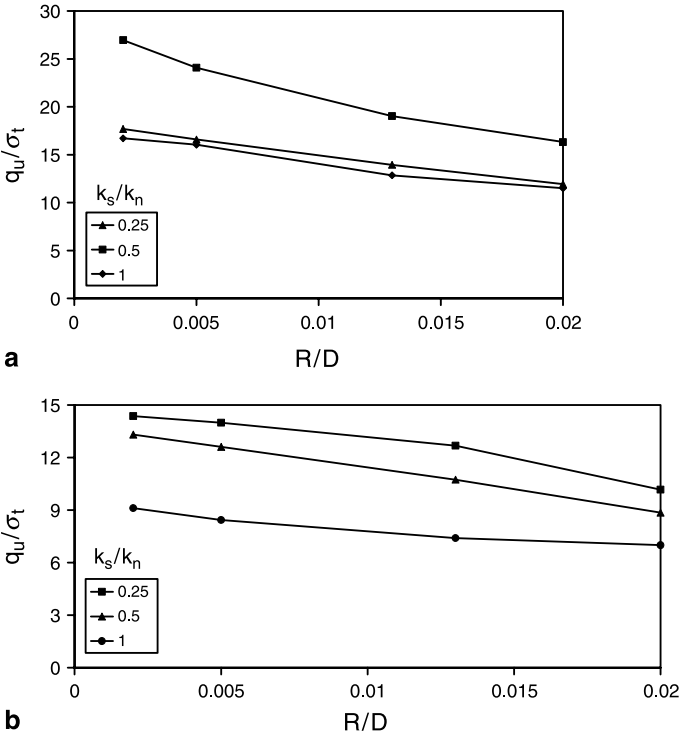


Fig. 10. Ratio of unconfined compressive strength to Brazilian tensile strength versus specimen resolution for different k_s/k_n values. **a** $\sigma_0/k_n = 0.1$, $n_b/s_b = 0.1$, **b** $\sigma_0/k_n = 0.08$, $n_b/s_b = 0.2$

increasing the specimen size (D), the tensile strength decreases. This size effect has already been reported by Potyondy and Cundall (2001) and Huang (1999) for a CPI model and is confirmed here for a SOCPI material. Notice that specimens with higher genesis pressure and lower n_b/s_b values demonstrate more intensive size effect. The reason for the observed size effect in a numerical Brazilian test goes back to the dependence of fracture toughness of the synthetic material to the specimen resolution; by increasing the cylinders radii for a fixed n_b/R , the fracture toughness increases (Potyondy and Cundall, 2001). Figure 10a and b demonstrate q_u/σ_t parameter versus R/D for different k_s/k_n values. These two graphs are important for the calibration procedure as is shown in this paper. An important observation here is that by introducing a non-zero genesis pressure, a q_u/σ_t value of more than 25 can be obtained. This ratio is usually around 5 for a CPI model (Huang, 1999), which is very low compare to those for rocks. Therefore, a SOCPI model is capable of simulating realistic compressive to tensile strength ratio.

3. Calibration Procedures

To examine the usefulness of the calibration charts, an effort was made to develop a synthetic material similar to Pennsylvania Blue Sandstone. The mechanical properties

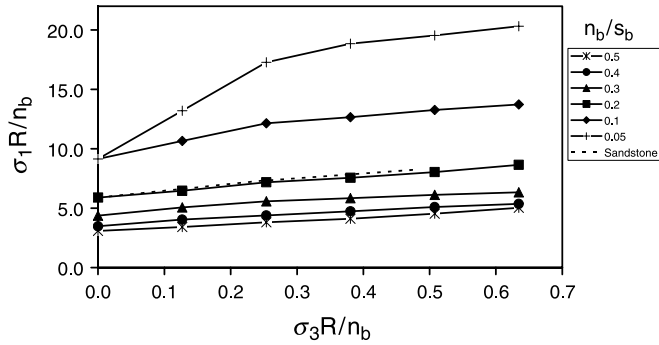


Fig. 11. Normalized failure envelopes for $\sigma_0/k_n=0.08$ together with the normalized failure envelope of Pennsylvania Blue Sandstone

of this rock were obtained by performing laboratory uniaxial, triaxial and Brazilian tests. This sandstone has a Young's modulus (E) of 27 GPa, a Poisson's ratio (ν) of 0.15, a tensile strength (σ_t) of 9.9 MPa, and an unconfined compressive strength (q_u) of 122 MPa. Four steps were employed to calibrate a synthetic material similar to the sandstone as explained in the following:

- The first step was to search among dimensionless or normalized failure envelopes that are shown in Fig. 5. In Fig. 11, the normalized failure envelope of the sandstone together with those of synthetic materials for $\sigma_0/k_n=0.08$ is shown. This figure confirmed that a curve with $n_b/s_b=0.2$ had a good match with the failure envelope of sandstone. Hence, $\sigma_0/k_n=0.08$ and $n_b/s_b=0.2$ were selected for the synthetic material.
- Equations 3a and 3b together with $E=27$ GPa and $\nu=0.15$ were used to obtain the apparent Young's modulus and Poisson's ratio of the sandstone as: $E'=28$ GPa and $\nu'=0.18$. With $\nu'=0.18$ and $\sigma_0/k_n=0.08$, $k_s/k_n=0.42$ was found from Fig. 2b. Next, $E'/k_n=0.66$ was acquired from Fig. 2a, with $k_s/k_n=0.42$ and $\sigma_0/k_n=0.08$. Therefore, the values of $k_n=42.4$ GPa and $k_s=17.8$ GPa were obtained. Finally, by considering $\sigma_0/k_n=0.08$ and $k_n=42.4$ GPa, a genesis pressure of 3.4 GPa was calculated.
- Laboratory experiments showed that the ratio of unconfined compressive strength to tensile strength of sandstone was 12. This value together with $n_b/s_b=0.2$, $\sigma_0/k_n=0.08$, and $k_s/k_n=0.42$ were used in Fig. 10b to obtain an R/D equal to 0.013. With a specimen width of $D=40$ mm and $R/D=0.013$, the average cylinders radius of 0.52 mm was calculated.
- The last step was to acquire the normal and shear bonds of the synthetic material. An approximate value of R/n_b was obtained through step (a), when the failure envelope was selected. In fact, R/n_b was the scaling factor in plotting the sandstone failure envelope in Fig. 11. A more accurate estimate of R/n_b value that incorporates the effect of k_s/k_n parameter is obtained by using Fig. 3b. With $\sigma_0/k_n=0.08$, $n_b/s_b=0.2$ and $k_s/k_n=0.42$, a $q_u R/n_b$ equal to 5 was acquired from Fig. 3b. Inserting $R=0.52$ mm and $q_u=122$ MPa in the $q_u R/n_b=5$, a normal bond of $n_b=12,664$ N/m was obtained which together with $n_b/s_b=0.2$, resulted in a shear

bond of $s_b = 63,320 \text{ N/m}$. Finally, a reasonable friction coefficient of $\mu = 0.5$ was assumed for the synthetic material.

In conclusion, the parameters obtained through the above calibration steps were: $\sigma_0 = 3.4 \text{ GPa}$, $k_n = 42.4 \text{ GPa}$, $k_s = 17.8 \text{ GPa}$, $n_b = 12,664 \text{ N/m}$, $s_b = 63,320 \text{ N/m}$, $R_{ave} = 0.5 \text{ mm}$ and $\mu = 0.5$.

4. Numerical Experiments on the Synthetic Material

The synthetic material with the microscopic parameters obtained in the previous section should have mechanical properties similar to Pennsylvania Blue Sandstone. To verify that, rectangular synthetic specimens $40 \text{ mm} \times 80 \text{ mm}$ in dimension for biaxial tests and circular specimens 40 mm in diameter were created using CA2 computer program. The uniaxial test result on the synthetic specimen is demonstrated in Fig. 12 that shows the axial stress-axial strain and axial stress-lateral strain curves. This figure compares well with that of laboratory uniaxial test on the sandstone (Fig. 13). It is interesting to realize that by development of microcracks at an axial

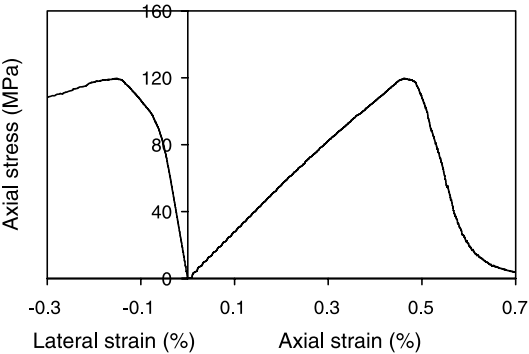


Fig. 12. Axial stress-axial strain and axial stress-lateral strain curves of the synthetic material in a numerical uniaxial test

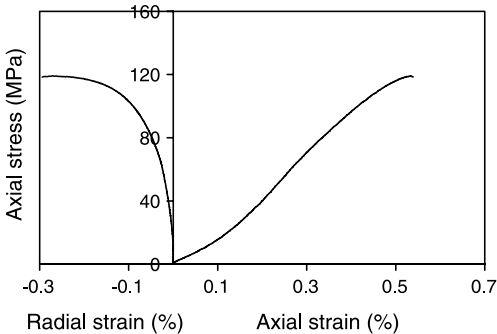


Fig. 13. Axial stress-axial strain and axial stress-lateral strain curves of the sandstone in a physical uniaxial test

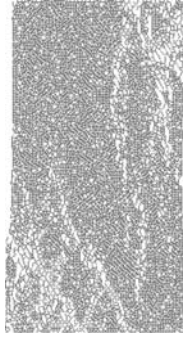


Fig. 14. Damaged specimen in a numerical uniaxial test

stress of about 60% of unconfined compressive strength, dilatation emerges as is clearly visible in axial stress-lateral strain curves of both physical and numerical specimens. From the initial slopes of the curves in Fig. 12, Young's modulus and Poisson's ratio of the synthetic material was calculated to be $E=27.2$ GPa and $\nu=0.145$ that are in good agreement with those for the sandstone. The damaged numerical specimen at the end of the uniaxial test is shown in Fig. 14. In this figure, damaged contacts of the cylinders are shown with lighter color; the synthetic specimen shows both shear banding and axial splitting at the end of numerical test.

Some numerical biaxial tests with confining pressure of $\sigma_3=5, 10, 15, 20$ MPa were conducted as well. Figure 15 shows the stress-strain curves for these biaxial tests from which the peak and residual failure envelopes can be deduced. These failure envelopes together with those for the sandstone are shown in Fig. 16 with great agreement. This figure confirms that the curvature of failure envelope has been simulated accurately.

For the numerical Brazilian test, a set up demonstrated in Fig. 7 was employed, and a tensile strength of $\sigma_t=9.9$ MPa was inferred that is in excellent agreement with that of the physical experiment. The damaged synthetic Brazilian specimen (Fig. 17)

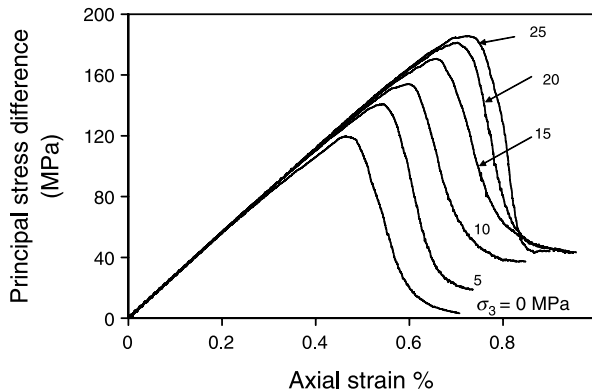


Fig. 15. Principal stress difference ($\sigma_1 - \sigma_3$) versus axial strain in numerical biaxial tests with different confining pressures

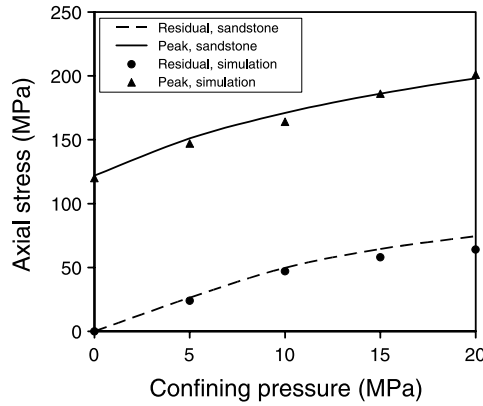


Fig. 16. Peak and residual failure envelopes of the sandstone and the synthetic material

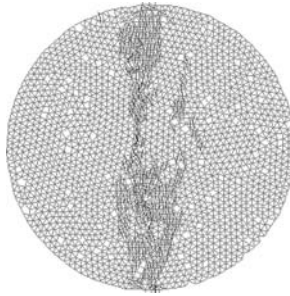


Fig. 17. Developed cracks in the numerical Brazilian test

indicates the development of a macroscopic tensile crack along the vertical diameter that is consistent with physical observation.

In addition to the triaxial and Brazilian laboratory tests, two non-conventional tri-axial tests were conducted on the Pennsylvania Blue Sandstone to investigate the effect of two different stress paths on the rock strength. For S2 sample of sandstone, a triaxial test was conducted. The initial confining stress was 25 MPa and subsequently, an axial stress of 175 MPa was applied and kept constant. The confining stress was then reduced gradually until the sample failure. The sample failed at a confining stress of 10 MPa. The final state of stress is shown in Fig. 18 together with the failure envelope. In this figure, p and q are defined as:

$$p = (\sigma_1 + \sigma_3)/2, \quad (7a)$$

$$q = (\sigma_1 - \sigma_3)/2. \quad (7b)$$

For the second physical experiment, sample S5, a confining stress of 10 MPa was applied initially and followed by an axial stress of 150 MPa. The confining stress was then increased to 20 MPa and kept constant while the axial stress was increased until the sample failure. The sample failed at an axial stress of 218 MPa. From Fig. 18, it is

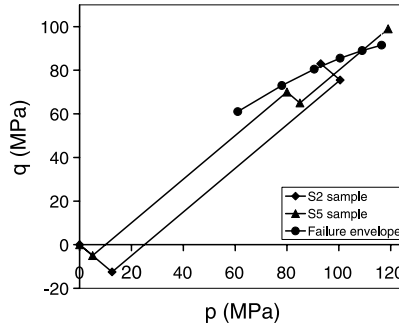


Fig. 18. Stress paths followed for S2 and S5 samples of the sandstone

clear that samples S2 and S5 that were subjected to stress paths different from that in a triaxial compression test, show strengths that are relatively consistent with the prediction obtained by the conventional failure criterion.

To approximately simulate these two stress paths, two numerical biaxial tests were conducted. In a biaxial test, the stress path is not the same as that in a triaxial test, but the effect of intermediate principal stress is usually small and it is ignored here. For the synthetic sample S2b, an initial lateral pressure of 25 MPa was followed by an axial stress of 175 MPa. Thereafter, the axial stress was kept constant while the lateral pressure was gradually reduced until sample failure. For the synthetic sample S5b, a confining pressure of 10 MPa was initially applied and followed by an axial stress of 150 MPa. The confining pressure was then increased to 20 MPa and kept constant while the axial stress gradually increased until the sample failure. The stress paths and failure points are shown in Fig. 19 together with the failure envelope of the synthetic material. It is clear that the failure points are in close agreement with the conventional failure envelope. Besides, comparison of Figs. 18 and 19 indicates that the calibrated synthetic material has been capable of simulating the failure of sandstone reasonably well, even when some non-conventional stress paths were followed.

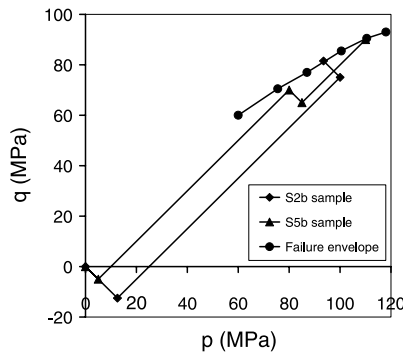


Fig. 19. Stress paths followed for S2b and S5b samples of the synthetic material

5. Conclusion

Dimensional analysis was employed to establish a fast and efficient tool for calibration of synthetic materials made of bonded circular particles. Several dimensionless parameters and graphs were presented. Careful examination of these graphs shows that the applied genesis pressure during sample preparation is an important factor in dictating the deformation and strength properties of a synthetic material. By using high genesis pressure that induces small overlap of circular cylinders, more realistic friction angle, failure envelope and tensile strength can be simulated that was not possible previously by using circular cylinders with zero genesis pressure.

A synthetic material similar to Pennsylvania Blue Sandstone was developed using the proposed systematic approach for sample calibration. Numerical experiments showed that this model could mimic some of the deformation and failure characteristics of the sandstone closely. For example, the peak and residual failure envelopes and their curvature were simulated accurately. In addition, two non-conventional stress paths were applied to both physical and numerical specimens that showed results in close agreement. In summary, it was shown that the numerical model together with the calibration charts offer a robust and useful tool in simulating many aspects of rock mechanical behavior.

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