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# "Breaking the O=O Bond": Deciphering the Role of Each Element in Highly Durable CoPd<sub>2</sub>Se<sub>2</sub> toward Oxygen Reduction Reaction

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**Supporting Information** 

**ABSTRACT:** In-depth insight into oxygen reduction reaction (ORR) electrocatalyst with high figures of merit (activity, stability, and selectivity) is highly crucial to rationally design electrocatalyst with a potential to replace state-of-the-art Pt/C. This work reports the synthesis of CoPd<sub>2</sub>Se<sub>2</sub> nanoparticles that show remarkably high stability of 50000 electrochemical cycles toward ORR. Morphology of the particles is characterized using SEM and TEM microscopy techniques and simulated using Bravais–Friedel–Donay–Harker (BFDH) morphology calculation method. A deconvoluted approach was used to understand the role of each element in the compound. DFT calculation was performed to have an in-depth analysis of the active site. Co and Pd provided an active site for the O<sub>2</sub> adsorption and Pd dissociates the O=O bond. The back-donation of substantial electron density to the  $\pi^*$  antibonding orbital of the molecule expedites the 4e- reduction of O<sub>2</sub> throughout the entire potential range. During the electrochemical stability test, Se forms a protective layer and prevents the active Co and Pd sites from OH poisoning.



**KEYWORDS**: oxygen reduction reaction, DFT calculation, d-band center, durability, BFDH morphology

### 1. INTRODUCTION

The 21st century has witnessed a huge boom in the worldwide population, and as a consequence, the demand for energy increases exponentially. With decreasing fossil-fuel reserves, the demand can only be fulfilled with the contribution from renewable sources of energy. This has triggered the research on alternative energy sources such as fuel cells that can convert the energy stored in a chemical bond to electrical energy. However, the major constraint posed to the fuel cell emerges from the sluggish kinetics of the ORR catalyst that drops the cell potential owing to high overpotential of the ORR process. Achieving 4e<sup>-</sup> reduction of O<sub>2</sub> to H<sub>2</sub>O is highly desirable to obtain high energy efficiency. Besides, the catalyst should be able to sustain the activity for a prolonged period as well. Thus, activity, selectivity, and stability of electrocatalyst are the three figures of merit to classify the efficiency of an electrocatalyst. State-of-the-art ORR catalysts Pt/C and Pd/C possess high activity and selectivity; however, they degrade due to -OH poisoning within few thousand working cycles. The slowness of O/OH hydrogenation, during ORR, also impedes the adsorption of  $O_2$  by occupying the adsorption sites with O/ OH.<sup>1</sup> Low abundance of Pt further adds to the problem.<sup>2–6</sup> It is highly challenging to develop a catalyst that can mimic all of the desired properties of Pt/C and possess better stability.

Stability of an electrocatalyst is challenged by strong adsorption of intermediate species formed during the ORR process. Therefore, protection of active site by lowering the adsorption energy of the binding adsorbate is essential. This can be attained either by altering the pH of the electrolyte or

through modifying the catalyst surface.<sup>7-9</sup> Since adsorption of charged species is highly potential-dependent, the use of alkaline medium is preferred in industry and space programs.<sup>10-13</sup> However, this does not entirely resolve the stability issue. Consequently, a secondary element is required that can protect the active site from degradation.  $^{\rm 14-17}$  Pd has a high surface energy of around  $1.43 \text{ J/m}^2$ . This results in the agglomeration of Pd nanoparticles which then decreases the activity of the catalysts.<sup>18</sup> Selenium, because of its low surface energy of 0.10  $I/m^2$ , forms a thin layer on the surface and justifies the role of the secondary element by preventing oxidation of the transition metal up to potentials of about 0.8-0.9 V vs RHE in the alkaline medium.<sup>19,20</sup> This has been explained in our previous work, wherein we proved that during the electrochemical ORR measurements on Pd<sub>17</sub>Se<sub>15</sub> catalyst in 0.1 M KOH (pH  $\sim$  13), Se forms a protective oxide layer and prevents the metal active sites from oxidative leaching. Thus, the active sites remain protected for a prolonged time.<sup>20</sup> Thus, chalcogenides represent a potential alternative to the Pt-based catalyst in terms of stability. A group of chalcogenide-based catalysts such as  $PtTe_{2}^{21} NiS_{2}^{22} CoS_{2}^{23} Co_{1-x}S_{2}^{24} Co_{3}S_{4}^{25} Co_{5}S_{8}^{26} CoSe_{2}^{27-29} Co_{7}Se_{8}^{30} ReS_{2}^{31} MoS_{2}^{32-34}$  and  $WS_{2}^{35,36}$  are reported for promising applications toward electrochemical reactions. Another major issue with Pd is the

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formation of significant amounts of  $H_2O_2$ , which is known to corrode the catalyst surface.<sup>37</sup>

Having understood the issue of durability in our previous work,<sup>20</sup> we tried to understand other figures of merit such as selectivity and activity in this work. ORR process can yield either H<sub>2</sub>O or H<sub>2</sub>O<sub>2</sub> formed via 4e<sup>-</sup> or 2e<sup>-</sup> transfer, respectively. Product formed during the reaction is mainly determined by the adsorption energy with which an oxygen molecule is bonded to the catalyst surface and the ability of the catalyst to dissociate the O=O bond. Pt/TiC was observed to exhibit 4 times better activity than Pt/C due to its ability to easily dissociate the O=O bond.<sup>38</sup> There have been several reports suggesting that the adsorption energy of O<sub>2</sub> on the surface should neither be too strong that the surface gets poisoned nor be too weak that the reaction does not progress further.<sup>39,40</sup> There are two types of adsorption modes reported in the literature: end-on mode (Pauling model), where O<sub>2</sub> adsorbs through a single bond on the catalyst surface, and the side-on mode (also known as Yeager model), where an O2 molecule forms a bridge-like adsorption with two bonds interacting with two sites.<sup>41</sup> Choi et al. proposed that side-on adsorption of  $O_2$  forms  $H_2O$  and end-on adsorption of  $O_2$  can form either of the products. Dissociative adsorption of  $O_2$  requires adjacent metals for the dissociation to take place.<sup>42,43</sup>  $H_2O$  can also form due to chemical decomposition of  $H_2O_{24}$ which also requires secondary metal. However, the preferred product is H<sub>2</sub>O to maximize the energy conversion efficiency of the process.

In this work,  $CoPd_2Se_2$  chalcogenide nanoparticles have been synthesized that show high activity and stability for 50000 electrochemical cycles and has exclusive selectivity toward  $H_2O$  over a wide-potential range. The catalyst retains its selectivity toward  $H_2O$ , better than Pt, over a prolonged time. The compound also exhibits high methanol-tolerance activity. We performed a comprehensive analysis to understand the role of Co, Pd, and Se toward these remarkable figures of merit. Cobalt and palladium in this work provide an active site for the dissociative process. A small amount of selenium, because of its low surface energy, forms an overlayer on the surface of the catalyst and protects Co and Pd active sites from degradation, thus maintaining its durability toward the reaction for 50000 cycles.

#### 2. EXPERIMENTAL PROCEDURES

**2.1. Chemicals.** Palladium acetylacetonate ( $Pd(C_5H_7O_2)_2$ , 99%), cobalt acetylacetonate ( $Co(C_5H_7O_2)_2$ , 99%), selenous acid ( $H_2SeO_3$ , 98%), trioctylphosphine ( $P(C_8H_{17})_3$ , 90%), and oleylamine ( $C_{18}H_{35}NH_2$ , 70%) were purchased from Sigma-Aldrich. All of the reagents were used without further purification. Milli-Q water (18.2 M $\Omega$  cm) was used throughout the synthesis and electrochemical measurements.

**2.2.** Synthesis of CoPd<sub>2</sub>Se<sub>2</sub>. CoPd<sub>2</sub>Se<sub>2</sub> nanoparticles were synthesized by colloidal synthesis method. A 0.1 mmol amount of Pd(acac)<sub>2</sub>, 0.1 mmol of H<sub>2</sub>SeO<sub>3</sub>, and 0.07 mmol of Co(acac)<sub>2</sub> were mixed together in 25 mL of oleylamine in a 50 mL two-necked round bottom (RB) flask. A 1000  $\mu$ L aliquot of trioctylphosphine was then added to the solution. The RB flask was then fitted to a condenser, vacuumized, and purged with Ar gas. It was then heated at 220 °C for 3 h. The product obtained was repeatedly washed with a hexane–ethanol mixture for several times and then dried in a vacuum oven at 60 °C for 6 h.

**2.3. Characterization.** The PXRD measurements at room temperature were carried out on a Rigaku miniflex X-ray diffractometer with Cu K $\alpha$  as the X-ray source ( $\lambda = 1.5406$  Å). The instrument is equipped with a position sensitive detector in the

angular range  $20^{\circ} \le 2\theta \le 90^{\circ}$  with the step size  $0.02^{\circ}$  and scan rate of 1 s/step calibrated against corundum standard. The experimental patterns were compared to the pattern simulated from the database.

In order to extract structural information from the diffraction data, we used the Rietveld method as implemented in the FullProf suite program. Quantitative microanalyses on all of the samples were performed with a FEI NOVA NANOSEM 600 instrument equipped with an EDAX instrument. Data were acquired with an accelerating voltage of 20 kV and a 100 s accumulation time. The EDAX analysis was performed using P/B-ZAF standardless method (where Z =atomic no. correction factor, A = absorption correction factor, F =fluorescence factor, and P/B = peak to background model) on selected spots and points. TEM images and selected area electron diffraction patterns were collected using a JEOL JEM-2010 TEM instrument, and color mapping was done in TECHNAI. The samples for these measurements were prepared by sonicating the nanocrystalline powders in ethanol and drop-casting a small volume onto a carbon-coated copper grid. XPS measurement was performed on an Omicron Nanotechnology spectrometer using a Mg K $\alpha$  ( $\lambda$  = 1253.6 eV) X-ray source with a relative composition detection better than 0.1%.

Pd K-edge EXAFS measurements of the catalysts were carried out in transmission mode at the PETRA P65 beamline of DESY.<sup>44</sup> Pellets for the measurements were made by homogeneously mixing the sample with an inert cellulose matrix to have an X-ray absorption edge jump close to one. Standard data analysis procedure was used to extract the EXAFS signal from the measured absorption spectra. Background subtraction, normalization, and alignment of the EXAFS data were performed by using the ATHENA software.<sup>45</sup> Theoretical EXAFS models were constructed and fitted to the experimental data in ARTEMIS. EXAFS data of the catalyst materials were fitted simultaneously with k-weights of 1, 2, and 3. The  $S_0^2$  value was fixed to 1 prior to EXAFS fitting. The data were Fourier transformed in the range of 4-13.8 Å<sup>-1</sup>. They were then fitted in R space between 1 and 3 Å for all samples. The fitting parameters were bond-length change between atoms ( $\Delta R$ ), change in energy scale between data and theory  $(\Delta E_0)$ , and mean-square displacement of the bond length ( $\sigma^2$ ), while the coordination numbers (CNs) were fixed from crystallography data of CoPd<sub>2</sub>Se<sub>2</sub>.

2.4. Electrochemical Studies. All of the electrochemical measurements were carried out using a CHI 760E electrochemical workstation with three electrode channels at room temperature. A conventional three electrode setup consisting of a glassy carbon (GC) (having diameter of 3 mm) as working electrode, platinum wire as counter electrode, and Hg/HgO (MMO) as reference electrode was used. Before all of the measurements, the electrolyte was deaerated with continuous purging of nitrogen gas for 30 min. The catalyst ink was prepared by dispersing 1 mg of catalyst in 200  $\mu$ L of mixed solvent solution (IPA:H<sub>2</sub>O = 1:3 (v/v)). Nation solution (5 wt %, Sigma-Aldrich) is diluted with isopropyl alcohol (IPA) to 0.05 wt %. From the prepared catalyst ink, 5  $\mu$ L of the slurry was drop-cast on GC electrode. Upon drying, a 5  $\mu$ L aliquot of 0.05 wt % Nafion solution has been added to the drop-cast electrode and dried overnight in air. The GC electrode was polished with 0.05  $\mu$ m alumina slurry and washed several times with distilled water prior to the deposition of catalyst slurry. Commercial Pd/C and Pt/C (20 wt %, Sigma-Aldrich) (with the same Pd/Pt loading on the electrode) were used for comparison of activity with the same loading as that of the synthesized catalysts. The commercial catalysts were tested by keeping all of the experimental conditions the same as in CoPd<sub>2</sub>Se<sub>2</sub>. Chronoamperometric (CA) measurements were performed in 0.1 M KOH electrolyte solution. Linear-sweep voltammetry (LSV) was recorded with a sweep rate of 5 mV/s in 0.1 M KOH electrolyte solution under steady state conditions. Tafel plots (TPs) were derived from LSV measurement. Accelerated degradation test (ADT) was performed with graphite rod (purity > 99.9%, Sigma-Aldrich) to avoid Pt dissolution. Stability tests were carried out at a scan rate of 100 mV  $s^{-1}$  and within a potential window of 0.4–0.9 V vs RHE. The potential window for the stability test was selected on the basis of previous literature.<sup>46–48</sup> After every 2000 potential cycles, LSV graphs



**Figure 1.** (a) Rietveld refinement of the powder XRD pattern of as synsthesized  $CoPd_2Se_2$  nanoparticles. (b) Se *k*-edge XANES spectra, (c) elemental mapping, and (d) SEM image of  $CoPd_2Se_2$  nanoparticles. (e, f) Morphology of the  $CoPd_2Se_2$  nanoparticles compared with BFDH generated morphology of  $CoPd_2Se_2$ . Simulated morphology confirms that particles have the same morphology, oriented along different directions as denoted by *x-*, *y-*, and *z*-axes. Particles are marked as A, B, C, and D, and their corresponding simulated morphologies are presented alongside.

were recorded to check the stability of the catalyst. Selectivity toward  $H_2O$ , before and after 1000 cycles, was checked using RRDE electrode with 4 mm diameter and keeping all other experimental conditions similar. All of the electrochemical data, presented in this work, are referenced to RHE scale.

**2.5. Theoretical Calculation.** DFT calculations were performed using PWSCF v.5.1 code embedded in the Quantum Espresso suite for quantum simulation of materials.<sup>49</sup> The PWSCF calculations were performed using generalized gradient approximation (GGA) with PBE<sup>50</sup> exchange–correlation functional and ultrasoft pseudopotential. Kinetic energy ( $E_k$ ) cutoff for the wave functions of 35.0 Ry was used to truncate the plane-wave basis. Furthermore, Gaussian smearing was used with a degauss value of 0.005 Ry. Systems were relaxed until the Hellmann–Feynman forces on each atom are in the order of  $10^{-2}$ – $10^{-3}$  eV/Å. The calculated Fermi energy ( $E_F$ ) was set to zero for all of the DOS plots. A vacuum of 10 Å was used to avoid any interaction among the slabs.

Systematic density functional theory was carried out to determine the interaction between the adsorbate and adsorbant. We applied the d-band model proposed by Hammer and Nørskov.<sup>51</sup> The d-band center ( $\varepsilon_{av}$ ) is given by the following relation in this model:

$$\varepsilon_{\rm av} = \frac{\int_{-\infty}^{E_{\rm f}} E\rho(E) \, \mathrm{d}E}{\int_{-\infty}^{E_{\rm f}} \rho(E) \, \mathrm{d}E}$$

where  $\rho$  is the projected density of d-states of surface atoms, E is the energy, and  $E_{\rm F}$  is the Fermi level.

Change in the average band center was calculated using the following formula:

$$\Delta = \varepsilon_{\rm av,initial} - \varepsilon_{\rm av,final}$$

where  $\varepsilon_{\rm av,initial}$  and  $\varepsilon_{\rm av,final}$  represent the average band center before and after the adsorption of O\_2 molecule.

# 3. RESULTS AND DISCUSSION

**3.1. Crystal Structure.** The crystal structure of  $CoPd_2Se_2$  is isotypic to  $K_2ZnO_2^{52}$  crystallizing in the orthorhombic



**Figure 2.** (a) Comparison of RRDE polarization curve of Pd/C (20 wt %), Pt/C (20 wt %), and CoPd<sub>2</sub>Se<sub>2</sub> recorded at 1600 rpm in O<sub>2</sub>-saturated 0.1 M KOH. Disk current represents current due to reduction of O<sub>2</sub> to H<sub>2</sub>O and H<sub>2</sub>O<sub>2</sub>, and ring current signifies current due to oxidation of H<sub>2</sub>O<sub>2</sub>. (b) Polarization curve of the catalyst before and after ADT test showing negligible shift in the onset potential. (c) Comparison of the Tafel slope before and after the ADT test shows a similar Tafel slope. Negligible change in the (d) polarization curve and (e) Tafel slope of the catalyst in the presence of methanol, showing high methanol tolerance activity.

system with *Ibam* space group. Parallel chains of edge-sharing  $CoSe_{4/2}$  tetrahedra run along [0, 0, 1] and [1/2, 1/2, 1] as shown in Figure S1.<sup>53</sup> Co occupies tetrahedral voids formed by Se, and Pd atom is surrounded by five Se atoms in a distorted quadratic—prismatic arrangement. Since both of the active sites are encapsulated within the Se cage, thus, the active sites are expected to have protection from poisoning intermediates.

Oleylamine used during the synthesis not only can act as a solvent and a surfactant but can also act as an electron donor at elevated temperature.<sup>54</sup> Trioctylphosphine (TOP) acts as a stabilizing agent and prevents agglomeration of nano-particles.<sup>55</sup> Adjusting the ratio between them results in well-separated nanoparticles with intermediate particle size.<sup>56</sup>

Rietveld refinement of the experimental PXRD pattern in Figure 1a and Fourier transformed Pd and Se *k*-edge XAFS data (Figure S2) confirm the purity of the phase. The PXRD pattern could be indexed as orthorhombic with *Ibam* space group and lattice parameters a = 5.993 Å, b = 10.493 Å, and c = 5.003 Å. Comparison of the Se *k*-edge XANES spectra (Figure 1b) with reference Se foil (zero oxidation state) and CoSe and NiSe (-2 oxidation state) indicates that the Se in the compound has a partial negative oxidation state. XPS spectra of individual elements are presented in Figure S3. STEM and elemental mapping in Figure 1c confirm the homogeneous distribution of elements throughout. The FESEM images in Figure 1d show a well-defined morphology of particles. The EDAX composition is presented in Figure S4.

The TEM image shows that the nanoparticles are less than 100 nm in size (Figure 1e). At first sight, these particles seem to have different morphologies. However, BFDH morphology simulation of the nanocrystals, followed by spatial orientation along a different direction, shows morphology similar to that obtained in the TEM image (Figure 1f). Only four sets of planes (002), (121), (110), and (020) are found to contribute to this morphology (Table S1).

**3.2.** Activity. The glassy carbon modified electrode was first activated prior to RDE measurements in N<sub>2</sub>-saturated 0.1 M KOH solution (Figure S5a). To obtain a stable current density, 100 CV activation cycles were performed. To evaluate the ORR catalytic activity of CoPd<sub>2</sub>Se<sub>2</sub>, the linear sweep voltammetry curves were then recorded in O<sub>2</sub>-saturated 0.1 M KOH solution, at a scan rate of 5 mV s<sup>-1</sup>. The polarization curves for ORR at different rotation rates are shown in Figure

S5b in which current densities are normalized with respect to the geometrical surface area (0.0706 cm<sup>2</sup>). The catalyst exhibits current density of 1 mA cm<sup>-2</sup> at a potential of 0.94 V vs RHE compared to Pd/C (0.90 V) and Pt/C (0.93 V). However, after an initial cycling of 1000 cycles using RRDE,  $CoPd_2Se_2$  showed negligible change in the potential, suggesting extraordinary stability similar to the state-of-theart catalysts Pt/C and Pd/C (Figure 2a). The nature of Pt/C and Pd/C polarization curve corroborates well with the previous literature results.<sup>57–59</sup>

The activity of the catalyst was also checked for OER in 0.1 M KOH. The catalyst was found to show good activity compared to state-of-the-art RuO<sub>2</sub>/C (Figure S6). The potential gap between the OER and ORR process, taken at the current densities of 10 and -3 mA cm<sup>-2</sup>, respectively, was found to be  $\Delta E = E_{\text{OER}@10} - E_{\text{ORR}@-3} = 829$  mV. Further study on OER was not performed as the present study completely focuses on the ORR aspect of the catalyst.

3.3. Durability Test and Kinetics Study for CoPd<sub>2</sub>Se<sub>2</sub>. A durability test was performed to assess their ability to sustain activity as shown in Figure 2b. Cyclic potential sweeps were performed between 0.4 and 0.9 V vs RHE, at a scan rate of 0.1  ${V/s}$  and a rotation speed of 800 rpm in O<sub>2</sub>-saturated 0.1 M KOH. An accelerated degradation test (ADT) was performed using RDE working electrodes with 3 mm diameter and graphite rod (purity > 99.9%, Sigma-Aldrich) as the counter electrode to avoid Pt dissolution. After 50000 cycles, the halfwave potential of CoPd<sub>2</sub>Se<sub>2</sub> remains high with a slight negative shift in the mixed kinetic-diffusion limited region. Tafel plots were plotted to understand the kinetics of the catalyst. The Tafel slope obtained toward the ORR before cycling was 56.2 mV dec<sup>-1</sup> at lower overpotential, which decreased to 46.5 mV  $dec^{-1}$  (Figure 2c). This Tafel slope value indicates that, during the initial reaction, the oxygen reduction, catalyzed by the CoPd<sub>2</sub>Se<sub>2</sub> chalcogenide, is controlled by the first chargetransfer step, similar to that of the state-of-the-art Pt catalyst. During the course of the reaction, a value close to  $40 \text{ mV dec}^{-1}$ indicates the dissociation of O=O as the rate-determining step.<sup>60</sup>

Methanol tolerance test was performed to check the stability of the catalyst in the case of fuel crossover in a fuel cell. Linearsweep voltammetry was run in the presence of 1 M methanol, keeping all of the experimental parameters the same (Figure

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2d). However, no significant change in the nature of the curve was observed even after the addition of methanol. The Tafel slope value also remains almost unaffected by the addition of methanol (Figure 2e). Chronoamperometric measurements were carried out for 50000 s to further test the durability of the catalyst (Figure S7).

As seen from Figure S8, the size of the particle grows with the number of cycles. This may be due to electrochemical sintering of the nanoparticles. However, the homogeneity of the particles remained intact even after 30 K cycles (Figure S9). It was tough to do elemental mapping on the particles after 50 K as the surface was enriched with Se and the electron beam damaged the surface (Figure S10). So, FESEM-EDAX images were collected as shown in Figure S11.

**3.4. Selectivity toward H<sub>2</sub>O Pathway.** The plot of 1/j vs  $\omega^{-1/2}$  for the potential range of 0.1–0.6 V (vs RHE) yields a series of essentially parallel straight lines, having a slope value of *B*. The linearity and parallelism of all lines in Figure S5c indicate that the electron number transferred per oxygen molecule and the active surface area of the catalyst do not obviously change in the potential range measured. A value of  $n \sim 4$  is obtained for the catalyst CoPd<sub>2</sub>Se<sub>2</sub>, unlike Co–Se and CoSe<sub>2</sub>/C which have values of n = 3.4 and 3.5, respectively.<sup>61</sup> To further confirm the selectivity toward H<sub>2</sub>O, RRDE measurement was carried out following the same experimental procedure. The percentage of H<sub>2</sub>O<sub>2</sub> produced and the number of electrons consumed during the reaction were recorded before and after 1000 cycles (Figure 2) and tabulated in Table S2.

To demonstrate the efficiency of our catalyst, state-of-the-art catalysts Pd/C (20 wt %) and Pt/C (20 wt %) were used as the references. The percentage of  $H_2O_2$  produced in the case of CoPd<sub>2</sub>Se<sub>2</sub> is less than 3.5% during the first cycle, which is comparable to that of Pt/C (~2.5%) and much better compared to Pd/C (~13%). However, after 1000 cycles, the percentage of  $H_2O_2$  produced on Pt/C, across the entire potential range, is higher than that of CoPd<sub>2</sub>Se<sub>2</sub>. Thus, CoPd<sub>2</sub>Se<sub>2</sub> exhibits very high selectivity toward  $H_2O$  over a wide potential range.

For achieving high energy efficiency,  $4e^-$  reduction to  $H_2O$  is important. This is possible only when  $O_2$  adsorbs side-on in a bridge site of a strongly adsorbing atom. However, strong O adsorption can pose a challenge for the stability of the catalyst. This is clearly seen in state-of-the-art catalyst Pt/C which exhibits superior activity but undergoes degradation due to active site poisoning by strongly adsorbing -OH intermediates. Thus, designing a catalyst with both good activity and good stability remains a challenge. Given the oxophilic character of Co and Pd active sites, Se is necessary to act as a sacrificially protective agent. In a similar system, we have recently found that the Se–Se coordination number decreases during electrochemical ORR, which is due to Se occupying edges and surfaces due to its lower surface energy  $(0.10 \text{ J/m}^2)$  compared to Co  $(2.34 \text{ J/m}^2)$  and Pd  $(1.43 \text{ J/m}^2)$ .<sup>20</sup>

Using density functional theory (DFT), we calculated the adsorption energy of ORR intermediates on the (100) plane (taken as a representative plane) (Figure S12). The binding energies for the systems under consideration were obtained from

$$\Delta E_{\rm b} = E({\rm ads/slab}) - E({\rm slab}) - E({\rm ads})$$

where E(ads/slab) is the energy of the final optimized adsorption configuration on the slab, E(ads) is the energy of

the adsorbed molecule, and E(slab) is the energy of the catalyst surface. With this definition, a negative  $\Delta E_{\text{b}}$  value corresponds to a stable adsorption on the surface.

Sites A–G in Figure 3a represent various adsorption sites of  $O_2$  on the CoPd<sub>2</sub>Se<sub>2</sub> surface. Since  $4e^-$  dissociation of  $O_2$  is



**Figure 3.** (a) Adsorption of  $O_2$  molecule on probable adsorption sites, (b) end-on and side-on adsorption configuration of  $O_2$  molecule and (c) DFT structural input and output configuration of  $O_2$  adsorbing on site **D**, **E** and **F**. Green, yellow, blue and red atom represents Co, Se, Pd and O atom, respectively.

observed using an electrochemical K-L plot. Thus, it is expected that the molecule adsorbs side-on instead of end-on (Figure 3b). Henceforth, A-C adsorption sites can be ignored due to end-on adsorption on these sites.

Adsorption energy of the  $O_2$  adsorbate is listed in Table 1. We observe that  $O_2$  molecule has higher negative adsorption

Table 1. O<sub>2</sub> Adsorption Energies and Optimized Bond Lengths of O–O and M–O Bonds

adsorption site	adsorption energy (eV)	$d_{o-o}$ (Å)	$d_{\text{co-O}}$ (Å)	$d_{\rm PD-O}$ (Å)
site F	-3.032	1.338	1.724	2.146
site G	-2.845	3.612	1.649	2.069

energy when adsorbed on site F (-3.032 eV) compared to that on site G (-2.845 eV). However, complete dissociation of O<sub>2</sub> molecule takes place on site G ( $d_{\text{O-O}} = 3.612 \text{ Å}$ ) unlike on site F ( $d_{\text{O-O}} = 1.338 \text{ Å}$ ) as shown in Figure 4a. Since the adsorption energy is calculated wrt the O<sub>2</sub> molecule, we ignored the energy of dissociation which is consumed during the process. An energy of 8.75 eV is wasted to dissociate an O<sub>2</sub> molecule to its corresponding atom. Adding this energy gives a more stable adsorption (-11.59 eV) on site G compared to site F. Free energy of H<sub>2</sub>O on the catalyst surface was calculated to be 0.04 eV, indicating easy desorption of the H<sub>2</sub>O formed.

Thus, we use the bridge site G, having a Co–Pd bond distance of 4.496 Å, as a descriptor for a stable adsorption site for  $O_2$  dissociation. As seen in Figure 4c, all of the planes obtained by BFDH morphology calculation have a high density of active site G represented by red arrows. Thus, complete dissociation of  $O_2$  molecule to O and complete 4e<sup>-</sup> reduction of  $O_2$  are expected.



Figure 4. (a) Schematic presentation of site F and site G and change in the average band center when adsorbate withdraws or donates electron density, (b) plot of average band center for all of the orbitals of each adsorption site, and (c) high density of site G in the planes obtained from BFDH morphology calculation.



**Figure 5.** Change in average band center for O<sub>2</sub> adsorption site in (a) (100) and (b) (020) planes of CoPd<sub>2</sub>Se<sub>2</sub>. The trend is compared with that of the (100) plane of (c) Pt<sub>3</sub>CO and (d) Pt. Green, yellow, blue, white, and red atoms represent Co, Se, Pd, Pt, and O atoms, respectively. Orbitals 1–5 represent  $d_z^2$ ,  $d_{xz}$ ,  $d_{yz}$ ,  $d_{yz}$ ,  $d_{xz}$ , d

Due to shortened M–O bond length, it is expected that  $\pi$  orbital on oxygen molecule interacts strongly with the d orbital of the transition metal and increased O=O bond length confirms that there is substantial back-donation from filled d-orbitals to the empty  $\pi^*$  antibonding orbital of the molecule. Thus, a detailed analysis on the PDOS of corresponding metal sites was requisite to understand the role of each metal site (Co and Pd). Since site G was chosen as the descriptor for the other planes, so a detailed PDOS analysis was done on Co and Pd active sites as shown in Figure 4b.

A change in the average band center was recorded for each of the metal orbitals of the adsorbing atom (Table S3). A positive change signifies an electron acceptance, and a negative change is indicative of the electron donation of the orbitals. Figure 5 clearly demonstrates that all of the Co d orbitals played the role of accepting electron density from the  $\pi$  orbital of the oxygen molecule, thus forming a strong Co–O bond. However, few of the d orbitals of Pd  $(d_{z^2}, d_{xz}, d_{yz})$  acted as an

electron acceptor, and some of these orbitals  $d_{x^2-y^2}$  and  $d_{xy}$ ) back-donated to the empty  $\pi^*$  antibonding orbital of the molecule. Thus, Pd formed a strong Pd-O bond and also played a vital role in dissociating the O<sub>2</sub> molecule. Site F was unable to dissociate the O2 molecule due to insufficient backdonation from only  $d_{x^2-y^2}$  orbital. Since O<sub>2</sub> adsorbs side-on and better fits into the hollow site G, so it is expected that it interacts strongly with the back-donating  $d_{x^2-y^2}$  and  $d_{xy}$  orbitals, which is not possible in the case of site F. Mismatch in the spatial overlap of the orbitals in adsorbate and adsorbant rationalizes the elongation of O2 on site F and complete dissociation on site G. The operating mechanism of CoPd<sub>2</sub>Se<sub>2</sub> is found to be different than those of Pt and Pt<sub>3</sub>Co (taken as references). The positive change in the average band center in these reference compounds indicates weakening of the O=O bond through strong electron acceptance from O<sub>2</sub>. Thus, Co weakens the O=O bond through strongly interacting with one of the O atoms and Pd weakens through strong interaction and

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back-donation of electron density. Se provides necessary protection to the active sites from poisoning intermediates. The role of each element is thus deconvoluted and can be used to design and improve the selectivity of  $O_2$  catalyst toward  $H_2O$  or  $H_2O_2$  products.

# 4. CONCLUSION

In conclusion, a ternary metal chalcogenide catalyst for oxygen reduction was synthesized by colloidal synthesis method for the first time, which demonstrated significantly high ORR activity, stability, and selectivity. Detailed insight into the roles of Co, Pd, and Se gained both by experiment and from DFT calculations may possibly inspire logical designing of catalyst with a better figure of merit. Our study also highlights the important aspect of alloying and shape-tailoring to control the exposed crystallographic facet of the catalysts to enhance the ORR performance and hence to develop more advanced non-Pt-based electrocatalysts. The strategy demonstrated here can be extended to design and develop the other Pd-based highly active and robust electrocatalysts in the field of electrocatalysis.

#### ASSOCIATED CONTENT

#### **S** Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsaem.9b01400.

Experimental details, tables, crystal structure, XANES, XPS, EDAX, LSV polarization curve, chronoamperometeric curve, TEM, and adsorption configuration of various ORR intermediates (PDF)

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#### Notes

The authors declare no competing financial interest.

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