Comparison of the irreversible thermomagnetic behaviour of some ferro- and ferrimagnetic systems

P S ANIL KUMAR^{\dagger}, **P** A JOY^{*} and S K DATE

Physical and Materials Chemistry Division, National Chemical Laboratory, Pune 411 008, India [†]Present Address: Information Storage Technology Group, MESA Research Institute, University of Twente, 7500 AE Enschede, The Netherlands

MS received 10 March 1999

Abstract. The magnetic behaviour of two ferromagnetic oxides and two ferrimagnetic oxides (ferrites) are compared to study the effect of magnetocrystalline anisotropy on thermal-history-dependence of magnetization of these ordered magnetic systems. All four compounds show thermomagnetic irreversibility ($M_{\rm FC} > M_{\rm ZFC}$) below a certain temperature, $T_{\rm irr}$. The highly anisotropic ferromagnetic oxide, SrRuO₃ and the hard ferrite, SrFe₁₂O₁₉ show sharp peaks below T_c in their $M_{\rm ZFC}(T)$ curves, whereas for the soft ferrite Ni_{0.8}Zn_{0.2}Fe₂O₄ and the low anisotropic ferromagnetic oxide La_{0.7}Ca_{0.3}MnO₃ only a broad maximum is observed in $M_{\rm ZFC}$, when measured in small magnetic fields. The shapes of the $M_{\rm ZFC}(T)$ curves are inversely-related to the magnitude of the coercivities (H_c) of the compounds in relation to the applied field, and the temperature dependence of H_c . $M_{\rm FC}$ and $M_{\rm ZFC}$ are related through the coercivity for all four magnetic systems.

Keywords. Ferromagnet; ferrimagnet; thermomagnetic irreversibility; magnetocrystalline anisotropy; coercivity.

1. Introduction

Spin glasses are frustrated magnetic systems characterized by their peculiar time and thermal-history-dependent magnetic behaviour. The spins are frozen in random directions due to lack of long-range magnetic interactions, when the system is cooled through its freezing temperature, $T_{\rm f}$, to a temperature $T < T_{\rm f}$. Thermal irreversibility between field-cooled (FC) and zero-field-cooled (ZFC) susceptibilities below $T_{\rm f}$, with a cusp in the low-field $c_{\rm ZFC}$ as well as in the ac susceptibility, long time relaxation of magnetization, etc are the characteristic properties of a spin glass (Mydosh 1993). Nishigori et al (1998) have recently reported that for the ferrimagnetic compounds, TbPdIn and DyPdIn, the field-cooled (FC) and zero-fieldcooled (ZFC) magnetizations show thermal irreversibility below the ferrimagnetic ordering temperature concluding that the irreversibility is due to the appearance of spinglass state. The origin of this spin-glass state is attributed to spin frustrations, due to the rare earth ions forming a deformed Kagome lattice. Similarly, while Maignan et al (1997) have reported the spin-glass-like behaviour of Pr_{0.7}Ca_{0.3}MnO₃ from low-field susceptibility studies, neutron-diffraction data of the compound evidence a ferromagnetic component. Itoh et al (1994) have concluded that there is no true long-range ferromagnetic order in Srrich compositions in the La_{1-x}Sr_xCoO₃ system, and that they are cluster glasses. Long-time relaxation of M_{ZFC} , large difference between M_{FC} and M_{ZFC} below a characteristic temperature, and the absence of magnetic saturation are indicated as evidences for the cluster glass behaviour of compositions with x > 0.18. Similar spin-glass-like properties have been reported recently for other magnetic systems also (Greedan *et al* 1996; Pecher *et al* 1999) for which neutron diffraction studies indicate long-range magnetic ordering. The origin of the spin-glass-like nature of these magnetic systems is generally attributed to the competing ferromagnetic and antiferromagnetic exchange interactions, structural disorder, etc.

Our results from a comparison of the low field magnetic behaviours of the ferromagnetic oxides La_{0.7}Ca_{0.3}MnO₃ and Pr_{0.7}Ca_{0.3}MnO₃ have indicated that the properties similar to that of a spin glass, observed for Pr_{0.7}Ca_{0.3}MnO₃, are due to its relatively larger magnetocrystalline anisotropy compared to that of La_{0.7}Ca_{0.3}MnO₃ (Anil Kumar et al 1998a). The magnetic behaviour of La_{0.5}Sr_{0.5}CoO₃ similar to that of a cluster glass as reported (Itoh et al 1994), could also be explained based on the high anisotropy of the compound (Anil Kumar et al 1998b). Roy et al (1994, 1997) reported the thermomagnetic irreversibility of different ordered magnetic systems, and concluded that the features of these ordered systems which are similar to those of spin glasses are due to a combination of intrinsic disorder along with inherent anisotropy. In this paper we have compared the low-field FC

^{*}Author for correspondence

and ZFC magnetic behaviours of the ferromagnetic oxides, $SrRuO_3$ and $La_{0.7}Ca_{0.3}MnO_3$, and the ferrimagnetic oxides, $Ni_{0.8}Zn_{0.2}Fe_2O_4$ and $SrFe_{12}O_{19}$, having widely differing magnetocrystalline anisotropies. $SrRuO_3$ is a well-known ferromagnetic oxide with very high magnetocrystalline anisotropy (Kabayasi 1976) and there is no magnetic saturation even at very high magnetic fields and low temperatures, whereas $La_{0.7}Ca_{0.3}MnO_3$ easily saturates in low magnetic fields and its calculated ferromagnetic moment is comparable to the expected value (Hwang *et al* 1995). Similarly, the magnetocrystalline anisotropy of the hexagonal ferrite $SrFe_{12}O_{19}$ (hard ferrite) is very high (Smit and Wijn 1959; Goldman 1990) compared to that of the cubic ferrite $Ni_{0.8}Zn_{0.2}Fe_2O_4$ (soft ferrite).

2. Experimental

The methods of synthesis and characterization of polycrystalline samples of SrRuO₃ (SRH), La_{0.7}Ca_{0.3}MnO₃ (LCM), Ni_{0.8}Zn_{0.2}Fe₂O₄ (NZF) and SrFe₁₂O₁₉ (SRF) have been reported previously (Kulkarni *et al* 1989; Anil Kumar *et al* 1996, 1998c; Joy *et al* 1997). Magnetic measurements (field-cooled (FC) and zero-field-cooled (ZFC) magnetizations and magnetic hysteresis) were performed using a EG&G PAR model 4500 vibrating sample magnetometer. Magnetic hysteresis measurements (maximum field = 15 kOe) were performed after cooling the sample through its T_c to the required temperature in zero magnetic field.

3. Results and discussion

Temperature variation of the zero-field-cooled magnetizations (M_{ZFC}) of SrRuO₃ (SRH), La_{0.7}Ca_{0.3}MnO₃ (LCM), Ni_{0.8}Zn_{0.2}Fe₂O₄ (NZF) and SrFe₁₂O₁₉ (SRF), measured at low magnetic fields, are shown in figures 1 and 2. For LCM and NZF, M_{ZFC} increases slowly with increasing temperature and shows a broad maximum below the Curie temperature. M_{ZFC} of SRF decreases slowly as the temperature is increased, a broad minimum is observed between 523 and 623 K (see inset of figure 2) and again slowly increases up to ~ 730 K above which a drastic increase in the magnetization is observed. A sharp peak is obtained at 737 K followed by the transition to paramagnetic state at $T_c = 740$ K. Similar sharp peak is observed for SRH also at 160 K close to its $T_c = 161$ K.

The magnetic hysteresis behaviour of the ferro- and ferrimagnetic compounds are shown in figure 3. The coercivities of the low anisotropy compounds, LCM and NZF, are very small compared to the coercivities of SRH and SRF with very large magnetocrystalline anisotropies $(H_c \approx 2K_1/M_s, \text{ where } K_1 \text{ is the anisotropy constant and } M_s \text{ is saturation magnetization (Brown 1962))}$. Temperature dependence of coercivities (H_c) of the ferromagnetic and ferrimagnetic compounds are compared in figures 4 and 5, respectively. The coercivity of LCM decreases linearly from a value of 35 Oe at 82 K ($T_c = 245$ K), as the temperature is increased, whereas for SRH, the coercivity is very high (~ 3500 Oe) at low temperatures. Similarly, for NZF the coercivity decreases from 21 Oe at 323 K ($T_c = 750$ K) compared to the high coercivity of SRF at

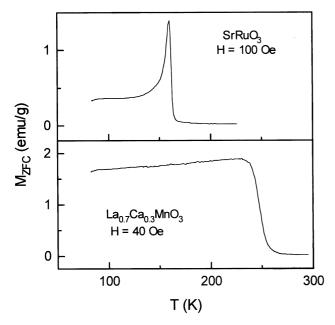


Figure 1. Temperature dependence of M_{ZFC} of SrRuO₃ and La_{0.7}Ca_{0.3}MnO₃ measured at the low magnetic fields indicated.

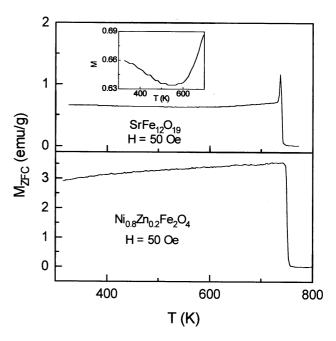


Figure 2. Temperature dependence of M_{ZFC} of $SrFe_{12}O_{19}$ and $Ni_{0.8}Zn_{0.2}Fe_2O_4$ measured at the low magnetic fields indicated. Inset: expanded M_{ZFC} (*T*) curve of $SrFe_{12}O_{19}$.

room temperature (Jahn and Muller 1969). H_c of SRF increases from ~ 4000 Oe at 323 K to ~ 5500 Oe at 473 K and decreases further to ~ 4000 Oe at 673 K, as the temperature is increased.

A comparison of figures 1 and 4 and 2 and 5 shows that for all the four compounds, the temperature dependence of the low-field $M_{\rm ZFC}$ is inversely related to the magnitude

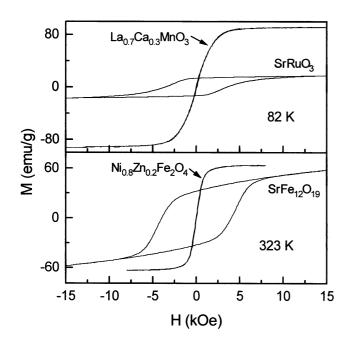


Figure 3. Magnetic hysteresis curves of $SrRuO_3$, $La_{0.7}Ca_{0.3}MnO_3$, $SrFe_{12}O_{19}$ and $Ni_{0.8}Zn_{0.2}Fe_2O_4$ recorded at the temperatures indicated.

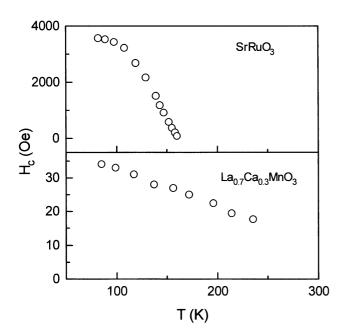


Figure 4. Temperature dependence of H_c of SrRuO₃ and La_{0.7}Ca_{0.3}MnO₃.

of the coercivities of the compounds in relation to the applied field, and the temperature dependence of the coercivity. For NZF and LCM the coercivity decreases very slowly with increasing temperature and $M_{\rm ZFC}$ increases slowly as the temperature is increased. For SRF, $M_{\rm ZFC}$ decreases up to about 473 K and again increases up to 730 K as the temperature is increased. A reverse trend is observed for H_c in this temperature region, showing maximum H_c at ~ 473K. This relation between the temperature variations of H_c and M_{ZFC} of SRF may be clearly seen in figure 2 where the expanded $M_{\rm ZFC}$ curve measured at 50 Oe is shown as inset. Above 723 K, the coercivity decreases very fast up to the T_c and the M_{ZFC} curve shows a drastic increase up to 737 K. A sharp peak is observed in $M_{\rm ZFC}$ at this temperature. For SRH also, a similar trend is observed between $M_{\rm ZFC}$ and $H_{\rm c}$. As the applied magnetic field is increased, the peak in the $M_{\rm ZFC}$ curves becomes broader and the peak temperature is shifted to lower temperatures, as shown for SRF and SRH in figure 6. The difference between T_c and temperature at which a maximum is observed in $M_{ZFC}(T)$ is related to the magnitude of the applied field and the coercivity at that temperature. The peak in $M_{\rm ZFC}$ is observed at a temperature where the coercivity is larger than the applied field and a broad maximum is observed when H_c is comparable or smaller than the applied field. This observation shows the influence of the coercivity on the shape of the $M_{\rm ZFC}$ curves.

FC and ZFC magnetization curves of LCM at $H_{app} = 100$ Oe and SRH at $H_{app} = 2000$ Oe are shown in figure 7 and that of NZF at $H_{app} = 50$ Oe and SRF at $H_{app} = 2000$ Oe are shown in figure 8. All the compounds

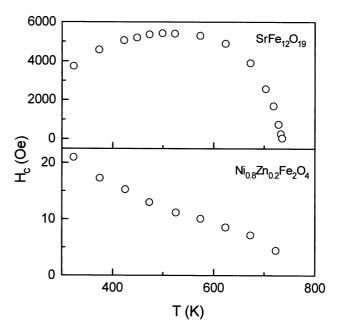


Figure 5. Temperature dependence of H_c of $SrFe_{12}O_{19}$ and $Ni_{0.8}Zn_{0.2}Fe_2O_4$.

show thermomagnetic irreversibility ($M_{\rm FC} > M_{\rm ZFC}$) below a certain temperature, $T_{\rm irr}$. Furthermore, with increasing applied field, there is a corresponding decrease in the value of $T_{\rm irr}$, as observed for La_{0.5}Sr_{0.5}CoO₃ (Anil Kumar *et al* 1998b). $M_{\rm FC}$ and $M_{\rm ZFC}$ of different magnetic systems were found to be related through the expression (Joy *et al* 1998)

$$M_{\rm ZFC} \approx M_{\rm FC} \frac{H_{\rm app}}{H_{\rm app} + H_{\rm c}},$$

where H_{app} is the applied field and H_c the coercivity. M_{ZFC} values calculated using the above expression from the measured $M_{\rm FC}$ at different temperatures are shown in figures 7 and 8. It may be seen that the magnitudes of the calculated $M_{\rm ZFC}$ at different temperatures are comparable to that of the measured M_{ZFC} for LCM and NZF. For SRH and SRF also, the shapes of the calculated $M_{\rm ZFC}$ curves are identical to those of the measured $M_{\rm ZFC}$, though there are slight differences in their magnitudes. This difference between the calculated and measured $M_{\rm ZFC}$ s may be attributed to the presence of a small residual net magnetic field during the process of zero field cooling (Joy et al 1998). A small net magnetic field present during the zero field cooling will greatly affect the $M_{\rm ZFC}$ of the highly anisotropic compounds than that of the low anisotropy compounds because the magnitude of the measured M_{ZFC} at low temperatures is determined by the magnetic anisotropy.

The results show that $M_{\rm ZFC}$ is related to $M_{\rm FC}$ through the coercivity, and the difference between $M_{\rm FC}$ and $M_{\rm ZFC}$ (i.e.

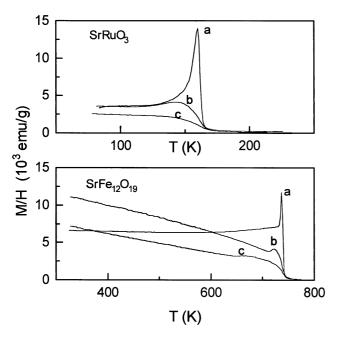


Figure 6. Temperature dependence of M_{ZFC}/H of SrFe₁₂O₁₉ and SrRuO₃, measured at different applied fields; 50 Oe (**a**); 2000 Oe (**b**); 5000 Oe (**c**).

thermomagnetic irreversibility) is due to the magnetocrystalline anisotropy since coercivity is a measure of the anisotropy. Because of the large magnetocrystalline anisotropy of SrRuO₃ and SrFe₁₂O₁₉, their thermomagnetic irreversibility is much larger and sharp peaks are observed in the M_{ZFC} curves, whereas the low anisotropy

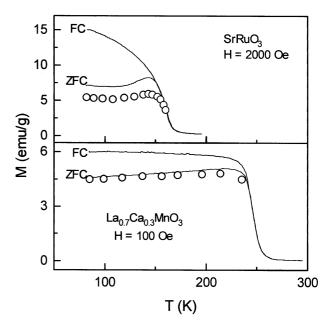


Figure 7. Temperature dependence of M_{FC} and M_{ZFC} of SrRuO₃ and La_{0.7}Ca_{0.3}MnO₃ measured at the magnetic fields indicated. The circles represent the calculated M_{ZFC} from the measured M_{FC} using the expression given in the text.

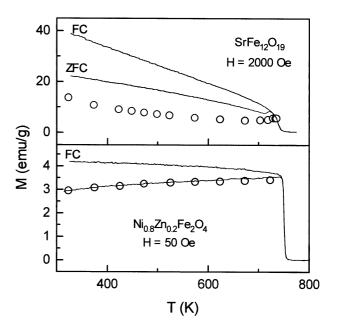


Figure 8. Temperature dependence of $M_{\rm FC}$ and $M_{\rm ZFC}$ of ${\rm SrFe_{12}O_{19}}$ and ${\rm Ni}_{0.8}{\rm Zn}_{0.2}{\rm Fe}_2{\rm O}_4$ measured at the magnetic fields indicated. The circles represent the calculated $M_{\rm ZFC}$ from the measured $M_{\rm FC}$ using the expression given in the text.

compounds $La_{0.7}Ca_{0.3}MnO_3$ and $Ni_{0.8}Zn_{0.2}Fe_2O_4$ show broad maximum in their M_{ZFC} curves and relatively small irreversibility. The shape of the M_{ZFC} curve is determined by the applied magnetic field strength in relation to the coercive field at a given temperature.

The thermomagnetic irreversibility arises from the magnetic anisotropy due to the difference in the way in which the FC and ZFC measurements are performed on the polycrystalline samples, the sample is either cooled in a magnetic field (FC) or in zero magnetic field (ZFC) through the magnetic ordering temperature. The response of the spins to the external magnetic field depends on the competition between magnetocrystalline anisotropy energy and the applied magnetic field strength. At low measuring fields, compared to the anisotropy field, all the spins will not be oriented in the direction of the applied magnetic field. The magnitude of $M_{\rm ZFC}$ at low temperatures below the $T_{\rm c}$ of the compounds will depend on the anisotropy, and therefore the difference between $M_{\rm FC}$ and $M_{\rm ZFC}$ will be larger for the highly anisotropic compounds. Hence, the irreversibility between $M_{\rm FC}$ and $M_{\rm ZFC}$ and a maximum (cusp or peak) in the ZFC magnetizations of the ferro- and ferrimagnetic systems are due to their magnetic anisotropy and not due to any spin-glass-like behaviour as reported for many ordered magnetic systems.

4. Conclusions

The field-cooled and the zero-field-cooled magnetizations of two ferromagnetic compounds, $La_{0.7}Ca_{0.3}MnO_3$ and SrRuO₃, and two ferrimagnetic compounds, $Ni_{0.8}Zn_{0.2}Fe_2O_4$ and SrFe₁₂O₁₉, having widely differing magnetocrystalline anisotropies, are compared. All four compounds show thermal irreversibility between M_{FC} and M_{ZFC} . M_{FC} and M_{ZFC} are related to each other through the coercivity and applied magnetic field. The extent of irreversibility and the shape of $M_{ZFC}(T)$ curve at a given applied field are directly related to the coercivity in relation to the magnitude of the applied magnetic field. The magnetic behaviour of $La_{0.7}Ca_{0.3}MnO_3$ is similar to that of $Ni_{0.8}Zn_{0.2}Fe_2O_4$ which are low anisotropy materials and the properties of the highly anisotropic compounds, $SrFe_{12}O_{19}$ and $SrRuO_3$, are comparable to each other.

References

- Anil Kumar P S, Shrotri J J, Kulkarni S D, Deshpande C E and Date S K 1996 *Mater. Lett.* **27** 293
- Anil Kumar P S, Joy P A and Date S K 1998a Solid State Commun. 108 67
- Anil Kumar P S, Joy P A and Date S K 1998b J. Phys.: Condens. Matter 10 L487
- Anil Kumar P S, Joy P A and Date S K 1998c J. Mater. Chem. 8 1219
- Brown Jr W F 1962 *Magnetostatic principles in ferromagnetism* (Amsterdam: North-Holland)
- Goldman A 1990 *Modern ferrite technology* (New York: van Nostrand)
- Greedan J E, Raju N P, Maignan A, Simon Ch, Pederson J S, Niraimathi A M, Gmelin E and Subramanian M A 1996 *Phys. Rev.* **B54** 7189
- Hwang H Y, Cheong S-W, Radaelli P G, Marezio M and Batlogg B 1995 *Phys. Rev. Lett.* **75** 914
- Itoh M, Natori I, Kubota S and Motoya K 1994 J. Phys. Soc. Jpn. 63 1486
- Jahn L and Muller H G 1969 Phys. Status Solidi 35 723
- Joy P A, Date S K and Anil Kumar P S 1997 *Phys. Rev.* B56 2324
 Joy P A, Anil Kumar P S and Date S K 1998 *J. Phys.: Condens. Matter* 10 11049
- Kabayasi A 1976 J. Phys. Soc. Jpn. 41 1876
- Kulkarni S D, Shrotri J J, Deshpande C E and Date S K 1989 J. Mater. Sci. 24 3739
- Maignan A, Varadaraju U V, Millange F and Raveau B 1997 J. Magn. Magn. Mater. 168 237
- Mydosh J A 1993 Spin glasses—an experimental introduction (London: Taylor & Francis)
- Nishigori S, Hirooka Y and Ito T 1998 J. Magn. Magn. Mater. 177-181 137
- Pecher S, Chevalier B, Laffargne D, Darriet B, Roisnel T and Etourneau J 1999 J. Magn. Magn. Mater. **191** 282
- Roy S B, Pradhan A K and Chaddah P 1994 J. Phys.: Condens. Matter 6 5155
- Roy S B, Pradhan A K, Chaddah P and Sampathkumaran E V 1997 J. Phys.: Condens. Matter **9** 2465
- Smit J and Wijn H P J 1959 *Ferrites* (Eindhoven: Philips Technical Library)