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OPEN Conduction Threshold in Accumulation-Mode InGaZnO Thin **Film Transistors**

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The onset of inversion in the metal-oxide-semiconductor field-effect transistor (MOSFET) takes place when the surface potential is approximately twice the bulk potential. In contrast, the conduction threshold in accumulation mode transistors, such as the oxide thin film transistor (TFT), has remained ambiguous in view of the complex density of states distribution in the mobility gap. This paper quantitatively describes the conduction threshold of accumulation-mode InGaZnOTFTs as the transition of the Fermi level from deep to tail states, which can be defined as the juxtaposition of linear and exponential dependencies of the accumulated carrier density on energy. Indeed, this permits direct extraction and visualization of the threshold voltage in terms of the second derivative of the drain current with respect to gate voltage.

Thin film transistors (TFTs) with disordered channel layers comprising say amorphous oxide or organic semiconductors, are becoming ubiquitous especially where low temperature processability and flexibility are essential¹⁻³. To facilitate TFT systems design, a complete understanding on device physics and related device parameter extraction constitutes the first and inevitable step. In this vein, the work presented here on conduction threshold is a very fundamental device parameter. The work is applicable not only to thin film field effect transistor (FET) but also the pervasive silicon metal-oxide-semiconductor FET (MOSFET). Given its significance as a crucial indicator of carrier transport and instability in FETs⁴⁻¹⁰, the conduction threshold has to be determined physically rather than by empirical means¹¹.

Silicon MOSFETs conduct in inversion mode and the physical reason underlying the conduction threshold is well understood^{4,5}. It corresponds to the critical state in which the induced or (inverted) carrier density becomes the same as the substrate's majority carrier density⁵. For example, in an n-channel MOSFET, this transition point is when the electron density goes from linear to exponential dependence on energy in the solution of the Poisson-Boltzmann equation 5. This transition can be approximated empirically as the intercept made from a linear extrapolation on the drain current (I_{DS}) vs. gate voltage (V_{GS}) characteristic at very low drain voltage (V_{DS})⁵.

In contrast to the silicon MOSFET, the TFT works in carrier accumulation mode regardless of the channel material⁶⁻⁹. This means that there is no polarity inversion of the induced carriers, so it is difficult to physically define the accumulation threshold. Moreover, because of the structural disorder of the semiconductor channel layer, there are localized deep and tail states distributed within the so-called band-gap, and depending on the position of the Fermi-level, free carriers or trapped carriers can be prevalent^{12,13}. As a result, the conduction threshold is difficult to extract even empirically^{6,14–16}. The density of carriers trapped at these localized states can be greater than free carriers, and there is no physical criteria for which all the traps are filled. In this family of devices, it is the localized trap states that determine the behaviour of field-effect mobility as a function of gate bias.

In this letter, we present a systematic analysis of the conduction threshold leading to a physical definition of the threshold voltage in accumulation-mode InGaZnO TFTs. By solving the Poisson-Boltzmann equation coupled with measurements of the current-voltage characteristics, the total carrier density is captured as a function of the surface potential. Similar to the MOSFET, it turns out that there is a transition point in which the accumulated carrier concentration switches from a linear to an exponential dependence on energy, i.e. the free carrier density increases rapidly from this threshold level by virtue of trap-limited conduction. We find that this threshold can be directly extracted as the gate voltage in which the second derivative of I_{DS} with respect to V_{GS} peaks. This is visualized with the proposed grayscale image spectroscopy, allowing an ease of analysis on threshold voltage and related properties. The studies presented here show that the conduction threshold is independent of the band

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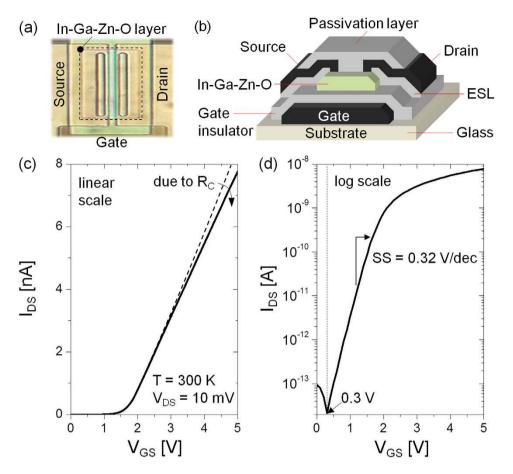


Figure 1. (a) Photomicrograph and (b) schematic of the fabricated InGaZnO TFTs examined in this work. Measured drain current (I_{DS}) vs. gate voltage (V_{GS}) in (c) linear scale and (d) log scale, respectively. Here, we indicate an effective flat-band voltage ~0.3 V.

mobility and is largely determined by the sub-threshold characteristics associated with the deep states including interface states.

Results and Discussion

Microscopic conduction threshold. For the investigations reported here, we used a semiconducting oxide TFT test structure with a 50 nm thick InGaZnO channel layer, as seen in Fig. 1(a,b). The device is a typical bottom gate structure where Mo is used for electrodes. Here, the gate insulator is formed as a bilayer of SiO_x and SiN_x, and the total gate-insulator capacitance of this structure (C_{ox}) is ~10 nF/cm². As a final step, a single layer of SiO_x is used for passivation layer and etch-stop layer (ESL). The TFT has a 50 µm channel width (W) and 10 µm channel length (L). Its I_{DS} vs. V_{GS} measured for V_{DS} fixed at 10 mV is shown in Fig. 1(c,d). Here, we retrieve an effective flat-band voltage (V_{FB}) of 0.3 V and a sub-threshold slope (SS) of 0.32 V/dec. Using the measured I_{DS} – V_{GS} characteristics, the microscopic picture in terms of free and trapped carrier densities can be captured as a function of V_{GS}. The free charge density (Q_{free}) relates to the drain current as^{12,13},

$$I_{DS} = \mu_b Q_{free}(W/L) V_{DS}.$$
 (1)

where μ_b is the band mobility. Equation (1) is valid for both sub- and above-threshold regimes as long as $V_{DS} < kT/q$, where kT thermal energy and q the elementary charge (see Supplementary Information S1). In addition, we replace V_{DS} with $V_{DS} - R_C I_{DS}$, where R_C is a contact resistance. The value of R_C retrieved for the test structure considered here is $50 \, k\Omega$. The main-unknown in Equation (1) is μ_b , which typically ranges from 10 to $30 \, cm^2/V \cdot s^{7,9}$. From Equation(1) and the measured $I_{DS} - V_{GS}$, Q_{free} can be stated as a function of V_{GS} , and related with the free carrier density (n_{free}) through $Q_{free} = q \, n_{free} \lambda_{free}$. Here, λ_{free} is an effective thickness of the induced free carrier sheet, and is expressed as an harmonic average between the free carrier Debye length (λ_D) and thickness of In-Ga-Zn-O channel layer (t_S), viz. $\lambda_{free} = (\lambda_D^{-1} + t_S^{-1})^{-1}$, assuming that the penetration depth of the vertical electric field is limited within $t_S^{17,18}$. The λ_D can be obtained using $\epsilon_S kT/(qQ_{free})^{5,16}$, where ϵ_S is the permittivity of the In-Ga-Zn-O channel layer. The value of λ_{free} is shown in Fig. 2(a). From this, we express n_{free} as $Q_{free}/(q\lambda_{free})$, which is still a function of V_{GS} , i.e. $n_{free}(V_{GS})$. To express it as a function of surface potential (ϕ_S), we relate this to Boltzmann's equation, i.e. $n_{free} = N_C \exp[((E_{F0} + q\phi_S - E_C)/kT]^{5,12}$. Here, N_C is the effective density

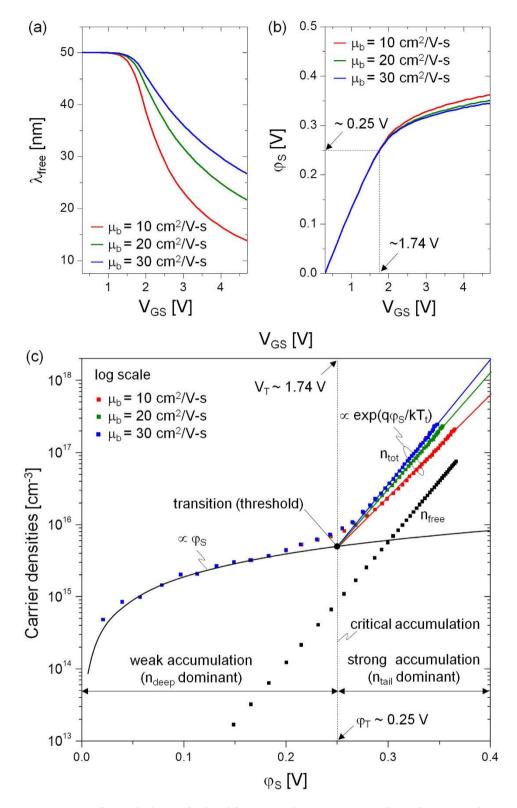


Figure 2. (a) Effective thickness of induced free carrier sheet (λ_{free}) vs. V_{GS} . (b) Surface potential (φ_S) as a function of V_{GS} and (c) total induced carrier density (n_{tot}) and n_{free} as a function of φ_S for different μ_b values of 10, 20, and 30 cm²/V-s, respectively.

of free carriers, E_{F0} an equilibrium Fermi level, and E_C the energy level of the conduction band minima. For the device considered here, N_C is $\sim\!2\times10^{18}\,cm^{-3}$. We now have a correspondence between V_{GS} and ϕ_{S} ,

$$\phi_{S} = \frac{kT}{q} \ln\left(\frac{n_{free}(V_{GS})}{N_{C}}\right) - \frac{(E_{F0} - E_{C})}{q} \equiv f(V_{GS}).$$
(2)

Using Equation (2), ϕS is computed for three different values of μ_b , which is contained in $n_{free}(V_{GS})$. This is shown in Fig. 2(b).

Now, we solve the integral form of Poisson's equation, which yields the relationship between the surface electric field (E_s) and integral of total accumulated carrier density n_{tot} for φ_s as follows¹²,

$$E_{S} \equiv g(V_{GS}) = \sqrt{\frac{2q}{\varepsilon_{S}}} \int_{0}^{\varphi_{S}} n_{tot} d\varphi.$$
(3)

For more detailed derivation, see Supplementary Information S2. In addition, we have expressed the left hand side of the Equation (3) as a function of V_{GS} , i.e. $E_S = C_{ox}(V_{GS} - V_{FB})/\epsilon_S \equiv g(V_{GS})$ by virtue of Gauss' Law (see Supplementary Information S3). Based on Equations (2) and (3), n_{tot} can be written as,

$$n_{tot}(\varphi_S) = \frac{\varepsilon_S}{2q} \frac{\partial (g(f^{-1}(\varphi_S)))^2}{\partial \varphi_S},\tag{4}$$

in which n_{tot} is $n_{free} + n_{deep} + n_{tail}$. Here n_{deep} and n_{tail} are the densities of carriers trapped in deep and tail states, respectively. In the term, n_{deep} the effect of carriers trapped at the interface states is embedded since n_{deep} consists of defect state carriers at the bulk (i.e. n_{bulk}) as well as interface (i.e. n_{int}), thus $n_{deep} = n_{bulk} + n_{int}^{13,17}$. Equation (4) can be considered as a differential form of the Poisson-Boltzmann equation, allowing examination of the behaviour of carrier densities n_{tot} with φ_s , as depicted in Fig. 2(c) in a semi-log plot. A transition is observed at $\varphi_s = 0.25$ V, for a corresponding gate voltage of 1.74 V, whereby the carrier density n_{tot} goes from linear to an exponential dependency on energy. Based on this observation, we can define the threshold voltage (V_T) of the device as 1.74 V, which corresponds to the transition surface potential (φ_T) of 0.25 V. The physical meaning underlying this transition is further discussed in the following.

From Fig. 2(c), we find that the trapped carriers are dominant for all energy levels, i.e. $n_{free} < n_{tail} + n_{deep}$, thus trap-limited conduction (TLC) always prevails. Depending on the value of φ_s , the one or the other can prevail. For example, below φ_T , n_{deep} prevails and the linear dependency of n_{tot} can be explained as,

$$n_{deep}(\varphi_S) = q\zeta_d \varphi_S,\tag{5}$$

where ζ_d is a deep state density related to either n_{bulk} or n_{int} . From Fig. 2(c), the value of ζ_d in the examined InGaZnO TFT is estimated as ~2.1 × 10¹⁶ cm⁻³ eV⁻¹. Besides, ζ_d can also be retrieved from the relation $C_{deep}/(q^2t_S)$, where C_{deep} is a deep state capacitance, which can be deduced independently from the sub-threshold slope (SS) through the relationship, $C_{deep} = C_{ox} [SS/(kT/q\cdot ln10) - 1]^{17}$. For the channel thickness considered here $(t_s = 50 \text{ nm})$, $C_{deep} \sim 43.7 \text{ nF/cm}^2$, yielding $\zeta_d \sim 3 \times 10^{16} \text{ cm}^{-3} \text{ eV}^{-1}$. It is consistent with the value empirically estimated using Fig. 2(c). Note that, along with the C_{deep} and φ_T , an analytical expression of threshold voltage (V_T) can be derived from the charge neutrality equation, $C_{ox}(V_T - V_{FB} - \varphi_T) = C_{deep}\varphi_T$ yielding as,

$$V_T \approx \left(1 + \frac{C_{deep}}{C_{ox}}\right)\varphi_T + V_{FB}.$$
(6)

With Equation (6), V_T is estimated as 1.64 V using $C_{deep} \sim 43.7 \text{ nF/cm}^2$, $C_{ox} \sim 10 \text{ nF/cm}^2$, $\varphi_T = 0.25 \text{ V}$ and $V_{FB} = 0.3 \text{ V}$. Although it may be inaccurate due to its approximation to be analytical, Equation (6) can allow a quick estimation of V_T . Also, it is useful to see its proportionality with deep states (i.e. bulk and interface states) through the term, C_{deep} , for example.

After the transition ($\phi_T = 0.25 \text{ V}$), n_{tail} prevails. Indeed, the carrier density now increases exponentially, seen as linear on the semi-log plot Fig. 2(c). And it can be described by the following relation for $kT_t > kT^{12,16,19}$,

$$n_{tail}(\varphi_S) \approx N_{tc}kT_t \exp\left(\frac{E_{F0} + q\varphi_S - E_C}{kT_t}\right),$$
(7)

where N_{tc} is the tail state density at E_C and kT_t is the characteristic energy of tail states associated with the exponential profile of the energy of the band tail states^{6,12,14,16,19}. As seen in Fig. 2(c), the transition is always maintained as an intersection point regardless of the assumed value of μ_b .

To check the relationship between the transition point and electronic states for electrons, we deduce the density of states using the values of n_{tot} and n_{free} (see Fig. 2(c)), through¹²,

$$N_{deep,tail}(E) \approx \frac{\partial \left(n_{tot}(\varphi_S) - n_{free}(\varphi_S) \right)}{\partial \varphi_S} \bigg|_{E_F \to E}.$$
(8)

Here, the Fermi level E_F replaces ϕ_S by virtue of the relation, $E_F - E_{F0} = q\phi_S$. Figure 3 shows the profile of the density of states as a function of energy while assuming $\mu_b = 20 \text{ cm}^2/\text{V-s}$. We identify a transition energy E_T that demarcates the deep and tail states, confirming that all the deep states (N_{deep}) are filled when E_F arrives at E_T (see also the inset of Fig. 3). This demarcation defines the threshold voltage. Based on this observation of E_D it is

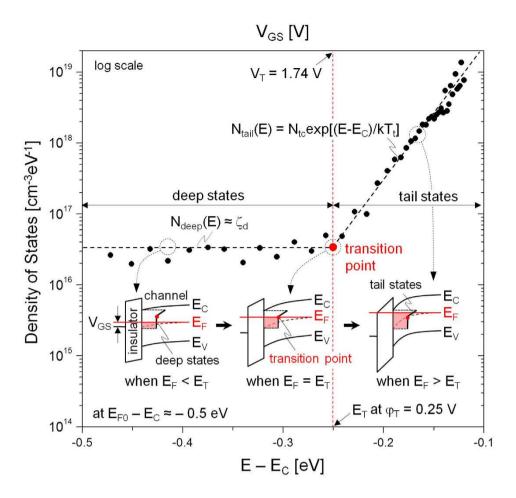


Figure 3. Density of states as a function of energy from E_{c} **.** Inset: band diagrams for three different cases of transition energy (E_{T}): below E_{T} , at E_{T} , and above E_{T} . Here, E_{V} denotes the valence band maxima, and $N_{deep}(E)$ and $N_{tail}(E)$ the respective density of deep and tail states as a function of energy.

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expected that E_T can move toward a higher energy when N_{deep} is increased while maintaining the profile of tail states, resulting in a higher V_T (see Supplementary Information S6). As seen in Fig. 3, at energies higher than E_T , the tail state distribution is found to follow an exponential dependence, i.e. $N_{tail}(E) = N_{tc}exp[(E - E_C)/kT_t]^{6,12,16}$. Using the exponential dependence of the tail states, the retrieved values of N_{tc} and kT_t are $\sim 10^{20}$ cm⁻³ eV⁻¹ and 27 meV, respectively, as seen in Fig. 3. Thus $kT_t > kT$ at T = 300 K. This confirms that the TLC is dominant at T = 300 K. Besides, the exponent (α) in the power-law for the $I_{DS} - V_{GS}$ curve above V_T is defined as $2kT_t/kT - 1$ for $kT_t > kT^{6,12,14,20}$. The value of α at T = 300 K is about 1.07 for $kT_t = 27$ meV, which consistent with the observed linear dependence of I_{DS} on V_{GS} as seen in Fig. 1(c). As implied here, the role of N_{tc} and kT_t is important in the TFT conduction, relating to the material composition. For example, with decreasing the ratio of In/Ga, field effect mobility of In-Ga-Zn-O TFTs can be reduced, suggesting a higher density of tail states (N_{tc})^{21,22}. In other words, if N_{tc} is higher, E_F can be pinned at a lower energy from the conduction band minima (see Supplementary Information S6). This leads to a lower mobility reminiscent of the a-Si TFT whereas Fermi level in oxide TFT can even go into the conduction band due to its lower tail state density²³.

Based on the characteristic behaviour of the carrier densities shown in Fig. 2(c), the significance of the free carrier density (n_{free}) relative to the total carrier density ($n_{tot} = n_{free} + n_{deep} + n_{tail}$) can be defined as a ratio (χ_{TLC}) in accordance to the TLC theory^{6,14,16},

$$\chi_{TLC} \equiv \frac{n_{free}}{n_{free} + n_{deep} + n_{tail}}.$$
(9)

Using Equation (8) together with the values shown in Fig. 2(c), the behaviour of χ_{TLC} and its first derivative with respect to V_{GS} (i.e. $\partial \chi_{TLC}/\partial V_{GS}$) as a function of V_{GS} is shown in Fig. 4(a,b), respectively. We see that $\partial \chi_{TLC}/\partial V_{GS}$ peaks at $V_{GS} \sim 1.74$ V regardless of the value of μ_b . This is consistent with our earlier observation of the common intercept seen in Fig. 2(c). Indeed, χ_{TLC} shows a similarity with the first derivative of the current-voltage characteristics, $\partial I_{DS}/\partial V_{GS}$, as seen in Fig. 4(a,c). As it turns out, $\partial I_{DS}/\partial V_{GS}$ is the trans- conductance of the transistor, which in turn is proportional to the field effect mobility (μ_{FE}). Thus, μ_{FE} is proportional to χ_{TLC} . Indeed in accordance with the TLC theory, μ_{FE} can be defined as $\mu_b n_{free}/n_{tot} = \mu_b \chi_{TLC}^{-620.24}$. At high V_{GS} ,

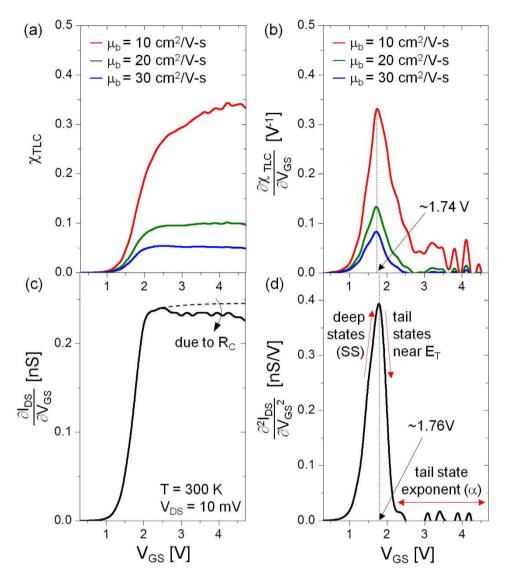


Figure 4. (a) Trap-limited conduction (TLC) ratio (χ_{TLC}) as a function of V_{GS} . (b) The first derivative of χ_{TLC} with respect to V_{GS} . (c) The first derivative (i.e. $\partial I_{DS} / \partial V_{GS}$) and (d) second derivative (i.e. $\partial^2 I_{DS} / \partial V_{GS}^2$) of I_{DS} vs. V_{GS} .

the behaviour of $\partial I_{DS}/\partial V_{GS}$ without effect of R_C looks similar to χ_{TLC} for $\mu_b = 20 \text{ cm}^2/\text{V-s}$ (see Supplementary Information S4). This suggests that μ_b of the examined device is ~20 cm²/V-s. The similarity between $\partial \chi_{TLC}/\partial V_{GS}$ and the second derivative $\partial^2 I_{DS}/\partial V_{GS}^2$ becomes even more striking in which the peak is at 1.74 V regardless of the band mobility (μ_b) values assumed for this analysis (see Fig. 4(b)). More importantly, the experimentally observed peak in $\partial^2 I_{DS}/\partial V_{GS}^2$ vs. V_{GS} appears at ~1.76 V which corroborates with theoretical prediction with a discrepancy less than 2%, as seen in Fig. 4(d). From the nature of the $\partial^2 I_{DS}/\partial V_{GS}^2$ curve, we clearly identify the roles of deep and tail states, respectively, on either side of the peak where the threshold voltage lies. The deep states and corresponding sub-threshold slope determines the slope of the rising edge while the tail states and its exponent α are associated with that of the falling edge (see Fig. 4(d)).

Macroscopic observation of the threshold. The above analysis can be extended to measurements of device behaviour at different temperatures. Figure 5(a) shows $I_{DS} - V_{GS}$ characteristics at T = 300 K, 200 K, 100 K, and the $\partial^2 I_{DS}/\partial V_{GS}^2$ in each case is shown in Fig. 5(b). As the ambient temperature decreases, we observe that V_T shifts to more positive V_{GS} values. This is because of reduced thermal activation¹². As temperature decreases, the current-voltage characteristic approaches a more power-law behaviour, as indicated in Fig. 5(a). Consequently, its second derivative converges to a non-zero value for $V_{GS} > V_T$ and which increases with decreasing temperature, as seen in Fig. 5(b). This is mainly associated with the tail state exponent (i.e. $\alpha = 2 k T_t / k T - 1$), which increases with decreasing temperatures, as indicated in Fig. 5(b). This is mainly due to the smaller sub-threshold slope (SS) at lower temperatures, which is typical of field effect transistors and can be intuitively explained with the relation, $SS = \ln(10) \cdot k T/q(1 + C_{deep}/C_{ox})^{17}$, assuming C_{deep} is independent of temperature.

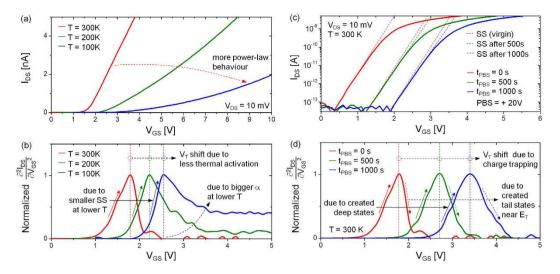


Figure 5. (a) Measured I_{DS} vs. V_{GS} for three different temperatures (T = 300 K, 200 K, and 100 K). (b) Normalized $\partial^2 I_{DS} / \partial V_{GS}^2$ for different temperatures. (c) Three curves of I_{DS} vs. V_{GS} measured before and after +20 V positive bias stress (PBS) for 500s and 1000s, respectively. (d) Normalized $\partial^2 I_{DS} / \partial V_{GS}^2$ for different bias stress times. Note that each $\partial^2 I_{DS} / \partial V_{GS}^2$ has been normalized to its peak value.

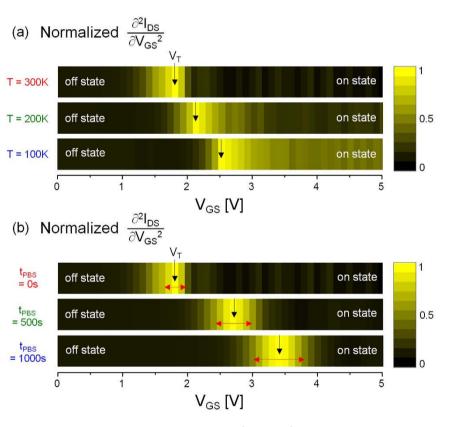


Figure 6. Grayscale images constructed from $\partial^2 I_{DS} / \partial V_{GS}{}^2$ (a) for different temperatures, and (b) for different stress durations, respectively.

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The observations of the current-voltage characteristics and its second derivative presented hitherto considerably simplify the interpretation of the shift in threshold voltage in bias stress experiments. More importantly, the shift in V_T can be retrieved with minimizing an interruption to the stress measurement conditions. Figure 5(c) shows the measured $I_{DS} - V_{GS}$ characteristics following positive bias stress (i.e. PBS at $V_{GS} = +20$ V while maintaining $V_{DS} = 0$ V) for stress durations (t_{PBS}) of 500 sec and 1000 sec. As seen in Fig. 5(d), there is a positive V_T -shift, which increases with stress duration. Since the value of $\partial^2 I_{DS} / \partial V_{GS}^2$ in a well-above V_T regime

remains zero for all cases, as seen in Fig. 5(d), it suggests that there is no significant degradation in band tail states (e.g. kT_t), thus unaffecting the tail state exponent α . This suggests electron trapping into the gate insulator as the significant and likely source of V_T -shift. However, the $\partial^2 I_{DS} / \partial V_{GS}^2$ curve appears to spread out more for longer t_{PBS}. Indeed, the slope of the rising edge (at $V_{GS} < V_T$) becomes less steep, suggesting creation of deep defects^{25–27}. Since the slope of falling edge decreases, as seen in Fig. 5(d), it can be argued that tail states are newly created near E_T (see Supplementary Information S5). So, it is obvious that the analysis presented here is especially useful under bias stress, as it is indicative of the nature of defects created in the sub-threshold and above-threshold regimes.

As another outcome of the presented analysis, we propose an image spectroscopy constructed from the second derivatives, as seen in Fig. 5. This method allows a visualization of V_T position as well as a property change near V_T . As an example, Fig. 5(a) shows $I_{DS} - V_{GS}$ characteristics measured at different ambient temperatures (300 K, 200 K, 100 K), and the V_T for each case is visualized in a grayscale image, as indicated in Fig. 6(a). Here, it is found that V_T is shifted to more positive in V_{GS} as decreasing temperature. Besides this, it is also observed that the grayscale image above V_T is getting brighter as decreasing temperature. This confirms that $I_{DS} - V_{GS}$ curvature in on state (i.e. above-threshold regime) becomes more nonlinear, as can be seen in Fig. 5(b). As another example, it can be applied to analyze the V_T instability with respect to bias stresses. In Fig. 5(c), we show the measured $I_{DS} - V_{GS}$ curves after applying a positive bias stress ($V_{GS} = +20$ V while $V_{DS} = 0$ V) for 500 sec and 1000 sec, respectively. As clearly seen in the constructed grayscale image (see Fig. 6(b)), there is a positive V_T shift and its increase towards more positive direction in V_{GS} with a longer stress time (t_{PBS}). Since the on state image brightness for each case looks dark and similar each other, as seen in Fig. 6(b), we can say that there isn't a significant degradation in band tail states which represents above-threshold regime (i.e. on state) curvature. So, this can be explained with electron trapping into gate insulator. However, the bright parts near V_T appear to spread out more after applying the PBS with a longer t_{PBS} . This suggests that the sub-threshold slope has been slightly bigger, and tail states are newly created near E_T . This is consistent with the results in Fig. 5(d).

Conclusions

The channel layer in amorphous thin film transistors in accumulation mode generally suffer from structural disorder resulting in localized deep and tail states in the energy gap. This makes the classical empirically-based extraction of the conduction threshold difficult, if not ambiguous, since depending on the Fermi level, conduction of free or trapped carriers can be prevalent determined by the shape of the localized tail states. This paper resolves the ambiguity in interpretation of the conduction threshold in accumulation-mode InGaZnO thin film transistors and provides a quantitative means of uniquely extracting the threshold voltage. The conduction threshold has been identified to take place at the transition of the Fermi level from deep to tail states, following which the carrier density goes from a linear to exponential dependence on energy. While the analysis presented here can be applied to study the relative changes in deep defects and tail states at pre- and post-transition caused by effects of ambient temperature and bias stress, a more precise analytical description is needed to quantitatively assess and subsequently predict the nature of the density of states when subject to environmental factors. Work along these lines is in progress. Nevertheless the results demonstrated here give physical and quantitative insight into the V_T and related properties in accumulation-mode thin film transistors.

References

- 1. Nathan, A. et al. Flexible electronics: the next ubiquitous platform. Proc. the IEEE 100, 1468 (2012).
- 2. Wager, J. F. et al. Transparent electronics. Science 300, 1245 (2003).
- 3. Lee, S., Jeon, S., Chaji, R. & Nathan, A. Transparent semiconducting oxide technology for touch free interactive flexible displays. *Proc. the IEEE* **103**, 644 (2015).
- 4. Garrett, C. G. B. & Brattain, W. H. Physical theory of semiconductor surfaces. Phys. Rev. 99, 376 (1955).
- 5. Sze, S. M. Physics of semiconductor devices 2nd edn (Wiley, 1981).
- 6. Street, R. A. Hydrogenated amorphous silicon 1st edn (Cambridge University Press, 1991).
- 7. Fortunato, E. et al. Oxide semiconductor thin-film transistors: a review of recent advances. Adv. Mater. 24, 2945 (2012).
- 8. Jahinuzzaman, S. M. *et al.* Threshold voltage instability of amorphous silicon thin-film transistors under constant current stress. *Appl. Phys. Lett.* **87**, 023502 (2005).
- 9. Lezama, I. G. & Morpurgo, A. F. Threshold voltage and space charge in organic transistors. Phys. Rev. Lett. 103, 066803 (2009).
- 10. Nomura, K. *et al.* Origins of threshold voltage shifts in room-temperature deposited and annealed a-In–Ga–Zn–O thin-film transistors. *Appl. Phys. Lett.* **95**, 013502 (2009).
- 11. Ortiz-Conde, A. et al. A review of recent MOSFET threshold voltage extraction methods. Microelectronics Reliab. 42, 583 (2002).
- 12. Lee, S. & Nathan, A. Localized tail state distribution in amorphous oxide transistors deduced from low temperature measurements. *Appl. Phys. Lett.* **101**, 113502 (2012).
- 13. Street, R. A. Thompson, M. J. & Johnson, N. M. Localized states in doped amorphous silicon. Phil. Mag. B 51, 1(1985).
- 14. Lee, S. et al. Trap-limited and percolation conduction mechanisms in amorphous oxide semiconductor thin film transistors. App. Phys. Lett. 98, 203508 (2011).
- 15. Powell, M. J. The physics of amorphous-silicon thin-film transistors. IEEE Trans. Elec. Dev. 36, 2753 (1989).
- 16. Shur, M. & Hack, M. Physics of amorphous silicon based alloy field-effect transistors. J. Appl. Phys. 55, 3831 (1984).
- 17. Lee, S. & Nathan, A. et al. Modeling sub-threshold current-voltage characteristics in thin film transistors. IEEE J. Disp. Tech. 9, 883 (2013).
- 18. Ghatak, K. P. & Bhattacharya, S. Debye screening length: effects of nanostructured materials 1st edn (Springer, 2014).
- Tiedje, T., Cebulka, J. M., Morel, D. L. & Abeles, B. Evidence for exponential band tails in amorphous silicon hydride. *Phys. Rev. Lett.* 46, 1425 (1981).
- 20. Lee, S. et al. Localized tail states and electron mobility in amorphous ZnON thin film transistors. Sci. Rep. 5, 13467 (2015).
- Olziersky, A. et al. Role of Ga₂O₃-In₂O₃-ZnO channel composition on the electrical performance of thin-film transistors. Mat. Chem. and Phy. 131, 512 (2011).
- Kamiya, T. *et al.* Present status of amorphous In–Ga–Zn–O thin-film transistors. *Sci. and Tech. of Adv. Mat.* 11, 044305 (2010).
 Martins, R. *et al.* Role of order and disorder on the electronic performances of oxide semiconductor thin film transistors. *J. Appl. Phys.* 101, 044505 (2007).
- 24. Shehu, A. et al. Layered distribution of charge carriers in organic thin film transistors. Phys. Rev. Lett. 104, 246602 (2010).
- 25. Powell, M. J. & Deane, S. C. Improved defect-pool model for charged defects in amorphous silicon. Phys. Rev. B 48, 10815 (1993).
- 26. Lee, S. et al. Oxygen defect-induced metastability in oxide semiconductors probed by gate pulse spectroscopy. Sci. Rep. 5, 14902 (2015).
- 27. Mandoc, M. M. et al. Trap-limited electron transport in disordered semiconducting polymers. Phys. Rev. B 75, 193202 (2007).

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Author Contributions

S.L. and A.N. designed the experiments, and performed the analysis. S.L. carried out the experiments. All the authors contributed to discuss about the paper. S.L. and A.N. wrote the paper.

Additional Information

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