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Conformational Changes and Flexibility of DNA Devices Observed by Small-Angle X-Ray Scattering

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ABSTRACT

Self-assembled DNA origami nanostructures enable the creation of precisely defined shapes at the molecular scale. Dynamic DNA devices that are capable of switching between defined conformations could afford completely novel functionalities for diagnostic, therapeutic, or engineering applications. Developing such objects benefits strongly from experimental feedback about conformational changes and 3D structures, ideally in solution, free of potential biases from surface attachment or labeling. Here we demonstrate that small-angle X-ray scattering (SAXS) can quantitatively resolve the conformational changes of a DNA origami two-state switch device as a function of the ionic strength of the solution. In addition, we show how SAXS data allow for refinement of the predicted idealized 3D structure of the DNA object using a normal mode approach based on an elastic network model. The results reveal deviations from the idealized design geometries that are otherwise difficult to resolve. Our results establish SAXS as a powerful tool to investigate conformational changes and solution structures of DNA origami and we anticipate our methodology to be broadly applicable to increasingly complex DNA and RNA devices.

KEYWORDS

DNA origami, small angle X-ray scattering, SAXS, normal modes, conformational changes

A fundamental aim of nanotechnology is to design synthetic objects that can adopt specific conformational states and carry out functions at the molecular scale, e.g. in transport, signal transduction, or molecular circuitry. Molecular self-assembly of DNA is a particularly successful approach towards creating versatile structures at the nanometer scale.¹⁻⁴ When using the DNA origami technique, a several kilobase long circular single-stranded scaffold strand is folded into custom target shapes with the assistance of hundreds of short single-stranded staple strands. By exploiting the specificity of DNA base pairing, precisely controlled shapes reaching over 100 nm in size and molecular weights of several MDa can be created.^{2,5-8}

While an important initial focus in the design of self-assembled DNA structures was to create static objects of well-defined shapes,^{1-3,6} more complex functions require dynamic 3D nanostructures that can undergo controlled conformational changes. Examples of dynamic

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3 DNA origami structures include a DNA box with a closable lid,⁹ a DNA nanorobot,¹⁰ a
4 reconfigurable plasmonic nanostructure,¹¹ or a DNA tweezers.¹² Such dynamic DNA
5 structures are promising candidates for applications ranging from nano-engineering^{13,14} to
6 medical diagnostics and therapeutics.^{15,16} An important challenge in this context is the precise
7 control over the 3D shape and mechanical flexibility of the target design in solution to
8 achieve desired functionality.
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12 So far, structural characterization of DNA origami structures has predominantly relied on
13 atomic force microscopy (AFM) imaging^{14,17,18} and negative-stain transmission electron
14 microscopy (TEM).^{6,19-21} While these techniques are well suited to image static structures,
15 they both rely on immobilizing samples on a surface and involve steps such as drying or
16 staining the samples, which renders the solution conditions far from physiological. Cryo-
17 electron microscopy provides less harsh conditions and has recently been successfully applied
18 to DNA origami structures^{9,22} but still requires immobilized samples embedded in vitrified
19 ice, potentially biasing the conformation of the sample and making it difficult to detect
20 conformational changes upon variation in solution conditions.
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23 In contrast, small-angle X-ray scattering (SAXS) can probe molecular conformations and
24 transitions and provides low-resolution structural information on molecules and molecular
25 assemblies in solution.^{23,24} As SAXS can operate under virtually arbitrary solution
26 conditions,^{25,26} the technique is ideally suited to detect conformational changes triggered by
27 changes in solution environment, such as ionic strength, denaturant, temperature, or ligand
28 binding. SAXS has proven very powerful to detect the large structural changes associated
29 with the folding of proteins²⁷⁻²⁹ and nucleic acids,³⁰⁻³² but can also readily detect more subtle
30 conformational changes, e.g. triggered by the binding of small-molecule ligands.³³⁻³⁶
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33 Recently, Gerling *et al.*³⁷ established a framework based on shape-complementary
34 recognition for the programmable and reversible assembly and disassembly of complex 3D
35 shapes built from DNA. One important example involves a dynamic “switch” device, where
36 multiple weak base stacking interactions were exploited to change conformations between a
37 closed and an open state as a function of temperature or ionic strength of the solution. Here
38 we use small-angle X-ray scattering (SAXS) to probe the structure and conformational
39 changes of the switch device in solution. In particular, we detect and evaluate the
40 conformational changes upon variations in solution conditions and observe quantitative
41 agreement with solution Förster resonance energy transfer (FRET) measurements. In addition,
42 we refine the 3D structure of the switch objects against the scattering data using a normal
43 mode based flexible fitting procedure and find evidence for swelling and structural
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3 rearrangements away from idealized DNA helix geometries. Together, our results establish
4 SAXS as a powerful technique to probe the structures and conformations of DNA devices.
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8 ***DNA origami switch samples based on base stacking interactions.*** We performed SAXS
9 measurements on three different variants of a DNA origami switch object that is based on
10 shape-complementarity and base stacking interactions.³⁷ A dynamic variant of the switch
11 (switch D) can undergo conformational changes between an x-shaped open and a rectangular-
12 shaped closed state (Figure 1; see Supporting Information and Supplementary Figures S1-S3
13 for details). This variant consists of two rigid bundles of DNA double helices arranged in a
14 honeycomb lattice that form the two arms of the structure. The arms are connected in the
15 middle of the structure by a single Holliday junction that acts as a pivot point for the
16 rotational degree of freedom (Figure 1). The structure of the closed state is prescribed by
17 shape-complementary patterns of double helical protrusions (red domains, Figure 1) and
18 recessions (blue domains, Figure 1) that can precisely dock into each other when the two arms
19 of the switch object come close together. The closed state is stabilized by up to 16 short-range
20 stacking interactions of the terminal bases of shape-complementary surface topographies. The
21 conformational equilibrium sensitively depends on ambient conditions such as the salt
22 concentration or the temperature of the solution.³⁷ TEM images of the
23 switch D variant show that at low salt concentrations the great majority of structures assume
24 the open state, while at high salt concentrations switch D particles predominantly populate the
25 closed state.³⁷ As reference structures we employed two static variants of the switch object
26 that are permanently locked in the open and closed states, respectively. In the static closed
27 variant (switch C) stacking interactions are replaced by stronger hybridization interactions of
28 3-bases-long single-stranded overhangs of corresponding staple strands holding the two arms
29 of the switch object in the closed conformational state (Figure 1, right). In the static open
30 variant (switch O) all stacking interactions are deactivated and the two arms are connected by
31 additional crossovers holding them at an opening angle of $\sim 90^\circ$ (Figure 1, left).
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49 ***DNA origami structures give rise to high signal-to-noise SAXS profiles at 25 - 100 nM***
50 ***concentrations.*** To estimate the minimum concentrations required for synchrotron-based
51 SAXS measurements on our large (~ 16000 nucleotides (nt) or ~ 5 MDa) DNA origami
52 structures, we used prior SAXS data of smaller nucleic acids in combination with
53 extrapolation based on a scaling relationship (see Supporting Information). We analyzed the
54 concentrations used for these SAXS measurements that resulted in a sufficient signal-to-noise
55 ratio for structural analyses (which we loosely define as analyses that go beyond Guinier
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3 fitting of the lowest q values) for a range of nucleic acid samples (Figure 2a, blue symbols).
4 The dataset ranges from an 8 nt DNA³⁸ to a large (~400 nt) ribozyme³¹ and includes both
5 RNA^{35,36,39,40} and DNA samples,^{41,42} as well as data for a ~14 knt DNA origami structure⁹
6 recorded at an in-house X-ray source. The data are well described by a scaling relationship of
7 the form $c \sim MW^\nu$, where c is the required concentration, MW the molecular weight, and the
8 scaling exponent ν was fitted to be $\nu = 1.30$ (Figure 2a, dashed line, and Supplementary
9 Information). The scaling relation predicts that concentrations of ~10-50 nM are sufficient to
10 obtain a good scattering signal for a ~16 knt DNA structure. Experimentally, we indeed
11 obtained good signal-to-noise scattering profiles for concentrations as low as 25 nM of the
12 DNA origami structures (Figure 2b and Supplementary Figure S4), in excellent agreement
13 with the predicted scaling relationship (Figure 2a, red star). Additional measurements at 50
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16 100 nM concentration display even higher signal-to-noise ratios (especially in the higher q -
17 range) and are superimposable after scaling by concentration, indicating the absence of
18 aggregation, radiation damage or interparticle interference (Figure 2b and Supplementary
19 Figure S4). Interparticle interference effects occur if the particles in solutions are, on average,
20 sufficiently close to interact, e.g. via excluded volume or electrostatic effects. We note that
21 interparticle interference effects are expected to be (even) weaker for larger macromolecular
22 assemblies as the typical intermolecular distances increase for higher molecular weights due
23 to the lower required concentration (Figure 2a, inset). For instance, the average intermolecular
24 distance of a 24 bp DNA sample (Radius of gyration $R_g \sim 2.4$ nm) measured at a
25 concentration of 0.2 mM is around 20 nm, whereas for our DNA origami objects ($R_g \sim 28$ nm,
26 see below) measured at 25 nM it is around 400 nm. Remarkably, due to their large size, the
27 DNA origami objects give rise to scattering profiles with a dynamic range of ≥ 4 orders of
28 magnitude in intensity, with features identifiable up to $q \approx 3$ nm⁻¹. We note that while the
29 scaling argument and extrapolation shown here only provide a rough estimate of the required
30 sample concentrations, we anticipate that it can provide a useful guideline to other SAXS
31 experiments on nucleic acid assemblies as well.

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51 ***SAXS reveals structural features of DNA origami objects.*** For a first structural
52 characterization, we analyzed the scattering data from the static switch samples, which serve
53 as reference samples for the dynamic switch variant. When comparing the scattering profiles
54 of the switch O and switch C samples, we observe significant differences in the q -range below
55 $q < 0.5$ nm⁻¹ (Figure 3a), in line with global structural differences in the open and closed
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3 states. For higher q , corresponding to smaller length scales, the scattering curves largely
4 coincide, exhibiting two distinct peaks.
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7 We performed a Guinier analysis of the scattering profiles in the low q range to determine
8 the overall radii of gyration (R_g) (see Supporting Information and Supplementary Figure S5).
9 We obtained an R_g of (27.9 ± 0.1) nm for the switch C (Table 1), which is in excellent
10 agreement with a theoretical R_g of 28 nm, derived from approximating the closed switch as a
11 rectangular beam, with $R_g^{theo} = 1/3[(W/2)^2 + (H/2)^2 + (L/2)^2]^{1/2}$, where W , H and L are the width,
12 height and length of the object, respectively (Figure 1). For the switch O sample we found an
13 average R_g of (29.0 ± 0.2) nm (Table 1), overall similar to switch C, which is expected as the
14 approximate distances from the center of mass are conserved upon the transition from the
15 closed to the open state.
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21 For elongated rod-like particles, where the axial dimension is much larger than the radial
22 dimension (as is the case for the DNA origami structures investigated in this work) the
23 scattering intensity can be factorized in an axial and radial scattering component.⁴³ Analysis
24 of the intermediate q -range then permits the calculation of the radius of gyration for the radial
25 cross-section (R_c) (see Supporting Information). We obtained an average R_c value for the
26 switch C of (6.7 ± 0.1) nm corresponding to a radius of the cross-section $R \sim 9.4$ nm, which is
27 in good agreement with the cross-sectional dimensions of the design model (Figure 1). The
28 switch O can be thought of as being assembled from two rods where the cross-section is half
29 of the size as for the switch C sample. Here, a smaller average R_c value of 4.8 nm,
30 corresponding to a radius $R \sim 6.8$ nm, is fully consistent with the expected reduction of the
31 cross-sectional area when the switch changes from the closed to an open conformation.
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40 A Kratky representation ($q^2 \cdot I(q)$ vs. q) of the scattering data of switch C and switch O
41 reveals a number of peaks that can be related to structural features (Figure 3b). The peak and
42 shoulder at lowest q -values (“1”, Figure 3b) for the switch O and switch C samples,
43 respectively, at $q \sim 0.06$ nm⁻¹ are related to the overall dimensions of the objects ($d \sim 100$ nm)
44 and to their R_g via $q \approx \sqrt{3}/R_g \approx 0.06$ nm⁻¹. The major peaks (“2”, Figure 3b) at $q \approx 0.14$ nm⁻¹
45 and at $q \approx 0.19$ nm⁻¹ for the switch C and the switch O sample, respectively, are related to the
46 maximum of the cross-sectional intensity expected at $q_{max} = 1/R_c$. The fitted R_c values of 6.7
47 nm for the switch C and 4.8 nm for the switch O sample (Table I) suggest $q_{max} \sim 0.15$ nm⁻¹
48 and $q_{max} \sim 0.2$ nm⁻¹, in very good agreement with the observed peak positions in the Kratky
49 plot. In the higher q -range, both scattering profiles display a small and broad peak (“3”,
50 Figure 3b) at $q \sim 1.0$ nm⁻¹ ($d \sim 6.3$ nm, Figure 1 red arrow *a*) and a more pronounced peak
51 (“4”, Figure 3b) at $q \sim 1.6$ nm⁻¹ ($d \sim 3.9$ nm, Figure 1 red arrow *b*), which corresponds to the
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3 distances between and within the honeycomb lattice, respectively (Figure 1). These values are
4 in approximate agreement with the theoretical values and the relative number of these
5 distances is approximately the same for both conformations, consistent with the similarity of
6 the scattering curves in the higher q -regime. We note that features relating to the structure of
7 single DNA helices (such as their diameter, the minor groove/major groove periodicity, and
8 the spacing between base pairs) occur on even shorter length scales and thus correspond to q -
9 values $\geq 3 \text{ nm}^{-1}$, which have been probed in wide-angle X-ray scattering measurements,⁴⁴ but
10 are not the focus of the present work.

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12 To more directly visualize the contribution of features on various length scales, we calculated
13 the pair distance distribution function $P(r)$ (see Supporting Information and Supplementary
14 Figure S6), which describes a histogram of all pairwise distances r within the sample (Figure
15 3c). For both static open and closed structures, we find a maximum pairwise distance D_{max} of
16 95 nm, in good agreement with the expected maximum distance from the designed structures.
17 The shape of the $P(r)$ function obtained for the switch C variant is peaked at low r with a long
18 tail out to higher r , characteristic of an elongated object. In contrast, the $P(r)$ of the switch O
19 exhibits an overall more Gaussian shape, characteristic of a more globular object. In the
20 switch C $P(r)$ function, we observe a well-defined peak at an intraparticle distance of 16 nm,
21 which corresponds to the maximum transverse distance of the closed state (Figure 1). This
22 peak is not apparent in the $P(r)$ function of the switch O sample, as expected, since the
23 opening of the switch reduces the transverse distance to 8 nm. For the open conformation we
24 find a smaller feature at 12 nm that is related to the height of the switch object and also
25 contains contributions from the maximum transverse distance of ~ 8 nm (Figure 1), which
26 become more exposed in the open state. The dominant $P(r)$ peak for the switch O, however,
27 occurs around 40 nm, the distance associated with the length of each of the two opened arms.

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29 ***Conformational populations of the dynamic switch variant.*** Having demonstrated that
30 SAXS clearly reveals the large-scale conformational changes between the open and closed
31 versions of the static switch object, we next analyzed the conformational states of the
32 dynamic version of the switch (switch D) at high (30 mM) and low (5 mM) magnesium
33 chloride (MgCl_2) concentrations (Figure 4a,b and Supplementary Figure S8). In general, the
34 scattering profile from an ensemble is given by the sum of the scattering profiles for the
35 individual components, weighted by their relative occupancy. In the case of a two-state
36 system, the scattering profile can be described by a linear superposition of the two states:

$$I(q) = f_1 \cdot I_1(q) + f_2 \cdot I_2(q) \quad (1)$$

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3 $I_1(q)$ and $I_2(q)$ are the scattering profiles and the coefficients f_1 and f_2 are fractional
4 occupancies of states 1 and 2, respectively. Using the scattering profiles of the switch O and
5 switch C objects for the open and closed states, we fitted the scattering profiles of the
6 dynamic variant at 30 mM MgCl₂ (switch D30) and 5 mM MgCl₂ (switch D05) as a linear
7 superposition of the two states (Supplementary Figure S8). Under both conditions, the two-
8 state fits provide an overall excellent description of the experimental data, suggesting that the
9 conformations of the dynamic switch variant can be well approximated by a two-state model
10 featuring the open and closed states. For both samples slight deviations of the fit become
11 apparent at higher q -values, which might imply that there exist structural differences within
12 the internal honeycomb lattice between the dynamic and static versions. This might be
13 attributed to the different concentrations of MgCl₂ in the sample solutions, which have an
14 impact on structural integrity and flexibility owing to its efficacy in screening interhelical
15 repulsion and stabilizing DNA Holliday junctions (see also below).^{45,46} In addition, previous
16 TEM studies on the switch D05 sample revealed a slightly reduced opening angle compared
17 to the fixed opening angle of 90° for the switch O sample,³⁷ which might cause some
18 additional differences in the scattering profiles.

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Complementary to analyzing $I(q)$, we applied a two-state model analogous to Equation 1
to the $P(r)$ functions (Figure 4c). We find that the $P(r)$ function of the switch D30 sample can
be described very accurately by the two-state model. For the $P(r)$ function of the switch D05
sample again slight deviations between the two-state model and the data are observable, but
overall the two-state description is still accurate.

The fitted parameters f_1 and f_2 in Equation 1 provide a direct measure of the relative
populations of the two states. Figure 4d shows the relative populations of the closed
conformation determined from the scattering intensity and $P(r)$ fits (the corresponding
populations of the open conformation are the complement to 100%). From the $I(q)$ fits, we
find a population of $(77 \pm 1)\%$ in the closed state for the switch D30 sample, in agreement
with the expectation that screening of electrostatic repulsion at high salt concentration should
lead to a predominant population of the closed conformation. In contrast, the occupancies
derived for the switch D05 sample are $(3 \pm 2)\%$ for the closed state, in line with the prediction
that electrostatic repulsion at lower ionic strength favors the open configuration. The $P(r)$ fits
gave identical results, within experimental error (Figure 4d). These findings are further
supported by the fact that the fitted cross-sectional radii of gyrations of the switch D object in
5 and 30 mM MgCl₂ are close to values determined for the switch O and switch C
conformations, respectively (Table 1).

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3 The results of the SAXS analyses can be compared to data obtained from ensemble FRET
4 measurements and TEM imaging on switch D particles at varying MgCl_2 concentrations³⁷
5 (see Supporting Information and Supplementary Figure S7). Data from solution-based
6 ensemble FRET measurements are in good agreement, within experimental errors, with the
7 SAXS results (Figure 4d). From TEM imaging data, higher fractional occupancies for the
8 closed state were obtained compared to the solution-based methods: $(93 \pm 1)\%$ of the objects
9 were identified to be in the closed state at a MgCl_2 concentration of 25 mM and $(13 \pm 2)\%$ of
10 closed particles were found at a MgCl_2 concentration of 5 mM. The deviations of the TEM-
11 determined fractions to the solution-based values are modest, but statistically significant for
12 the SAXS derived values (Figure 4d) and might be related to several factors: First, for the
13 TEM analysis switch D particles were picked from TEM images for each salt condition and
14 manually assigned to be either open or closed; errors were determined from binomial
15 counting statistics. This process might introduce a slight bias, as overlapping objects could
16 not be classified and as partially closed switch objects were considered as closed. Second,
17 TEM imaging requires immobilization of samples on a surface potentially affecting their
18 conformation. Furthermore, the staining process for TEM imaging can alter the global shape
19 of the particles.⁴⁷ In addition, single-molecule FRET experiments, which likewise require
20 surface immobilization of the switch D particles, gave similar results as the TEM data.³⁷
21 Taken together, the data suggest that surface immobilization and/or staining might create a
22 modest bias towards the closed conformation and can give rise to a small population of
23 partially closed conformations, possibly due to direct surface interactions or excluded volume
24 effects. In order to test whether the transition from the open to the closed state of the dynamic
25 switch variant upon the addition of MgCl_2 ions can be described as a two-state process, we
26 performed SAXS experiments on switch D samples for varying MgCl_2 concentrations ranging
27 from 3 mM to 30 mM (Figure 4e). The scattering profiles at different MgCl_2 concentrations
28 exhibit an iso-scattering point around $q \sim 0.3 \text{ nm}^{-1}$ tentatively suggesting that the
29 conformational transition can be described as a two-state process. For a more quantitative
30 analysis, we performed a two-state fit of the scattering profiles at each MgCl_2 concentration
31 according to Equation 1 and fitted the resulting populations by a thermodynamic model (using
32 Equations S8 and S9 in Supporting Information) assuming a linear dependence of the free
33 energy ΔG and the ion concentration c . From a least squares fit we obtained for $\Delta G_0 = 1.2$
34 kcal/mol at the reference ion concentration of 5 mM and the slope $m_c = -0.3 \text{ kcal}/(\text{mol mM})$
35 in good agreement with values based on ensemble FRET measurements (Figure 4f and
36 Supplementary Figure S7). In addition, the two state-fits yield a good fit of the full scattering
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3 profiles over the entire range of MgCl_2 concentrations (Supplementary Figure S8). These
4 findings show that the switch D transition from the open to the closed state can be described
5 adequately, at least at the current level of signal-to-noise, by a two-state model employing a
6 single open and closed conformation, without the need to introduce intermediates states or
7 conformations.
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11 Overall, we find quantitative agreement between SAXS and solution FRET derived
12 population estimates and approximate agreement with the TEM derived values, confirming
13 the switching mechanism in the dynamic switch variant. Our findings highlight the
14 importance of solution-based techniques when performing structural characterization of
15 complex DNA structures.
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21 ***Comparison of experimental SAXS data to idealized models and model refinement.*** In
22 addition to detecting conformational transitions and providing global measures of size and
23 shape (such as R_g , R_c , and D_{max}) SAXS can provide information about the full 3D solution
24 structure of macromolecules and their assemblies.^{24,48} Even though the resolution of SAXS
25 experiments is typically insufficient to compute a unique structure, it is possible to test and
26 refine structural models against experimental SAXS data.⁴⁹⁻⁵² First, we compared our
27 experimental data to scattering profiles of the switch O and switch C samples predicted from
28 idealized atomistic models generated by CanDo⁵³ (see Supporting Information). The
29 computed profiles from the CanDo models reproduce the overall shape of the experimental
30 curves and reveal similar characteristic peaks (Figure 5a,b; Supplementary Figure S9). In
31 addition, we obtain R_g and R_c values from the theoretical scattering curves, which are in good
32 agreement with the experimentally determined values (Table 1, Supplementary Table S1).
33 However, small, but systematic deviations between the experimental and theoretical profiles
34 are apparent. There is an additional peak in the theoretical scattering patterns for the open
35 state at $q \sim 0.26 \text{ nm}^{-1}$ and the peaks that are visible in both experimental and theoretical
36 curves are shifted, mostly to higher q in the theoretical curves compared to experiment.
37 Furthermore, the ratios of the peak intensity values at low and high q differ between the
38 experimental and theoretical curves. In addition, we determined the $P(r)$ functions from the
39 theoretical data and calculated a histogram of distances directly from the atomistic model
40 coordinates (Supplementary Figure S9). In comparison to the experimental data, the peaks
41 are more pronounced and deviations from the experimental peak positions are observable.
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57 We note that even though the different methods to compute scattering profiles from the
58 structures exhibit some differences (see Supporting Information and Supplementary Figure
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3 S9), they do give overall very similar results and show comparable deviations from the
4 experimental data, suggesting that the details of the scattering computations are relatively
5 unimportant and can not explain the observed differences to the experimental data. In
6 principle, both the hydration layer of partially ordered water molecules around a
7 macromolecule in solution^{49,54} and the ion atmosphere around charged nucleic acids^{55,56}
8 contribute to the scattering profile. For simple DNA duplexes, the effect of the ion
9 atmosphere has been studied in detail and while the it is observable, the ion cloud's
10 contribution to the scattering is pattern is relatively minor, typically increasing e.g. the radius
11 of gyration by a few angstroms.^{55,57} We have performed electrostatic calculations using
12 linearized Poisson-Boltzmann theory (Supplementary Information and Supplementary Figures
13 S10 and S11) to compare the electrostatic potential in the vicinity of a DNA origami structure
14 with a simple DNA duplex. Our results suggest that the electrostatic potential and,
15 consequently, the ion density around our DNA origami structures is only slightly elevated and
16 overall similar in magnitude and spatial extent compared to a single double-stranded DNA
17 helix (Supplementary Figures S10 and S11), consistent with previous reports in the
18 literature.^{58,59} Taken together, these observations suggest that for the very large DNA
19 structures considered in this work contributions from the ion atmosphere to the scattering
20 profile are small or negligible. In addition, we tested whether altering the density of the
21 solvent or the contrast of the hydration layer in the range of physically plausible values would
22 explain the observed differences between the CanDo derived models and our experimental
23 data, but again found that while changing the hydration shell gives rise to small changes in the
24 scattering profiles, these changes are insufficient to account for the observed differences
25 (Supplementary Figure S12).

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41 Combined, the differences between experimental and predicted scattering profiles indicate
42 that the switch objects adopt conformations in solutions that differ from the idealized models
43 generated by CanDo. Such deviations have been suggested previously: Pan *et al.*⁵³ found an
44 average root mean square deviation (RMSD) of 3.2 Å between the CanDo derived model and
45 the crystal structure of a DNA tensegrity motif. In general, electrostatic repulsion between
46 adjacent helix bundles or at crossovers resulting in the bowing out of double helical
47 domains^{2,4} can lead to local displacements of nucleobase positions. Theoretical calculations
48 and experimental evidence based on TEM data suggest an important role of flexibility for
49 several DNA origami structures,⁶⁰⁻⁶² indicating maximum root-mean-square fluctuation
50 amplitudes of a few nanometers.⁶⁰ In addition, a cryo-EM structure of a DNA origami object
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3 observed deviations between the idealized structure and the experimentally determined
4 density map.²²
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6 There is currently no established method to refine DNA origami structures quantitatively
7 against experimental data. A considerable challenge in this regard is the large size of our
8 switch objects that renders refinement e.g. based on all-atom molecular dynamics^{61,63}
9 challenging. As a computationally tractable approach, we turned to normal mode refinement
10 of the CanDo derived model against the experimental SAXS data using an elastic network
11 model. Normal mode analysis⁶⁴⁻⁶⁸ based on coarse-grained elastic network models has proven
12 to describe large-scale conformational changes surprisingly well as compared to considerably
13 more complex approaches⁶⁶ and has been applied to deform macromolecular structures to fit
14 and refine experimental data from cryo-EM,⁶⁹ X-ray crystallography,^{67,70} and SAXS data.⁷¹⁻⁷³
15 We iteratively refined the switch C and switch O structures against the experimental SAXS
16 data by normal mode based deformations (see Supporting Information). The resulting
17 structures yield significantly better fits to the data (Figure 5a,b): the goodness-of-fit statistic
18 X^2 (defined in Equation S4 in the Supporting Information) is reduced from 0.5% to 0.06%
19 and from 6.7% to 1.9% for the switch C and switch O structures, respectively. We find that
20 for the refined structures the highly symmetric lattice structure is significantly deformed
21 (Figure 5c,d). In comparison to the initial models, some parts in the refined closed and opened
22 switch objects swell and bulge out. This effect is especially pronounced in double helices
23 around the center of the structure, where the two arms are connected to each other
24 (Supplementary Figure S13). In addition, the refined structures show the helices at the ends
25 and sides of the arms slightly bend outwards (Supplementary Figure S13). Interestingly, these
26 effects are more pronounced in the switch C compared to switch O structure. The RMSD for
27 the refined switch C structure compared to the initial model is 22.3 Å; for the switch O, the
28 refined structure has an RMSD of 8.4 Å relative to the starting model. The larger
29 deformations in the switch C object compared to switch O might be due to the more compact
30 structure and, therefore, higher charge density, that would make electrostatic repulsion more
31 relevant for this object. Taken together, these data suggest an important role of flexibility and
32 local deformations in DNA origami objects, which has to be considered when designing
33 complex origami structures.
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54 In summary, we have demonstrated the ability for SAXS to sensitively monitor
55 conformational changes of self-assembled DNA origami objects in solution. SAXS provides a
56 number of advantages: First, being a solution-based technique, SAXS is free of potential
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3 biases and perturbations from the proximity of a surface. Second, SAXS is a label free
4 method, without the need to chemically modify the structure of interest. Third, SAXS reads
5 out the global conformation of molecules or molecular assemblies in solution, as defined by
6 their electron density, thus avoiding concerns whether e.g. variations in fluorescence might
7 stem from local conformational changes or photophysical effects upon changes in solution
8 condition. Taken together, these advantages render SAXS a very promising novel approach
9 for detecting conformational states of dynamic DNA origami objects and we anticipate that
10 many of the techniques' capabilities that were previously demonstrated in other contexts can
11 be extended towards monitoring conformational changes in DNA nanostructures, including
12 temperature controlled⁷⁴ and/or time-resolved SAXS^{31,32,75} measurements and the detection
13 and characterization of structural intermediates and molecular ensembles.^{36,39,76}

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16 Quantitative comparison of the experimental SAXS data to theoretical profiles derived
17 from 3D models of the DNA objects reveal considerable flexibility and deformations away
18 from the idealized "design" structure. Such deformations will have to be taken into account
19 for high-resolution designs in the future. In addition, this work highlights the ability of SAXS
20 to critically test structural models against solution-based data, even for very large DNA
21 objects, which constitutes a promising approach that is complementary to the more routinely
22 used methods.

23 24 25 26 27 28 29 30 31 32 33 34 **ASSOCIATED CONTENT**

35
36 Supplementary Methods, Table S1, and Supplementary Figures S1-S13. This material is
37 available free of charge via the Internet at <http://pubs.acs.org>.

38 39 40 41 **AUTHOR INFORMATION**

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Author Contributions

L.K.B., T.G., H.D., and J.L. designed the study; T.G. assembled and purified samples; L.K.B., S.M.S., and P.W. performed SAXS measurements; W.Z. performed structure refinement. All authors analyzed data, contributed to writing the paper, and have given approval to the final version of the manuscript.

Notes

The authors declare no competing financial interest.

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ABBREVIATIONS

SAXS, small-angle X-ray scattering; FRET, (Förster) fluorescence resonance energy transfer; AFM, atomic force microscopy; Switch C, closed switch; switch O, open switch; switch D, dynamic switch.

TABLES

Table 1. Comparison of the radius of gyration (R_g) and the cross-sectional R_g (R_c) for the static and dynamic versions of the switch object derived from experimental and theoretical scattering profiles. Experimental data correspond to averaged results from concentration scaled scattering profiles for sample concentrations of 25, 50 and 100 nM. *Values were determined from Guinier fits of the predicted scattering profiles in the fitting range $q_{max} \cdot R_g < 1.3$.

Sample	R_g (nm)	R_c (nm)
Switch C	27.9 (\pm 0.1)	6.7 (\pm 0.1)
Switch O	29 (\pm 0.2)	4.8 (\pm 0.0)
Switch D30	28.1 (\pm 0.1)	6.0 (\pm 0.1)
Switch D05	27.5 (\pm 0.2)	4.8 (\pm 0.0)
<i>CRYSOL (closed)</i>	28.2*	6.4
<i>CRYSOL (open)</i>	29.5*	4.4

FIGURES

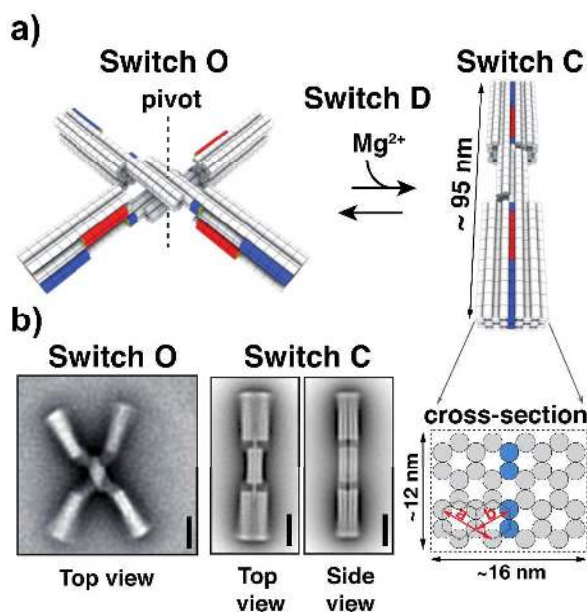


Figure 1. Illustrations of static and dynamic switch devices. (a) Schematics of the switch devices used in this study. The dynamic switch object (switch D) changes from an x-shaped open to a rectangular-shaped closed conformation upon addition of magnesium ions. Shape-complementary protrusions and recessions are indicated by the red and blue DNA double helical domains, respectively. Static switch variants are locked in the open (switch O, left) and closed (switch C, right) state. The schematic of the cross-sectional area of switch C indicates the horizontal and vertical dimensions including inter-helical distances of $a = 6$ nm and $b = 4$ nm, which give rise to a peak in the scattering profiles of switch C and switch O. (b) Corresponding average negative-stain TEM micrographs of switch O in the presence of 5 mM MgCl_2 and of switch C at a MgCl_2 concentration of 25 mM. Scale bars, 20 nm.

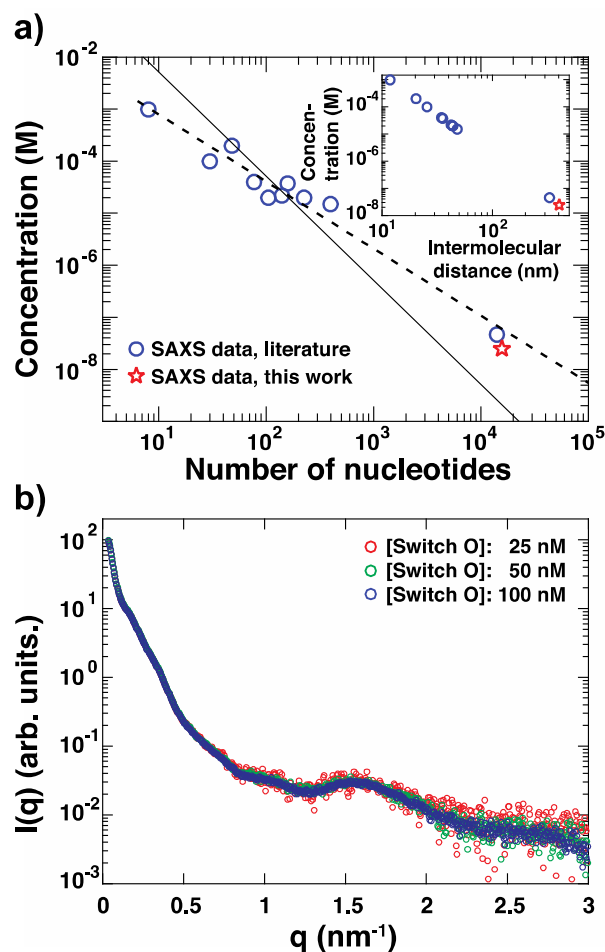


Figure 2. Concentration requirements and SAXS signals for large DNA origami structures. (a) Concentrations required to obtain a suitable SAXS signal, as a function of molecule size (in number of nucleotides), for a range of nucleic acid samples investigated previously (blue circles). The solid line is a fit of the relationship a/nt^2 , where nt is the number of nucleotides and a a fitting constant. The dashed line is a fit of the relationship b/nt^v where b and v are fitting constants. From the best fit we find $v \sim 1.30$. The red star corresponds to measurements of the DNA origami switch samples in this study that were guided by the scaling behavior. Inset: Inter-molecular distances calculated for the required SAXS concentrations of the different nucleic acid samples. (b) Averaged scattering profiles of the switch O measured at three different concentrations: 25 nM (red circles), 50 nM (green circles), 100 nM (blue circles). Data are scaled by concentration.

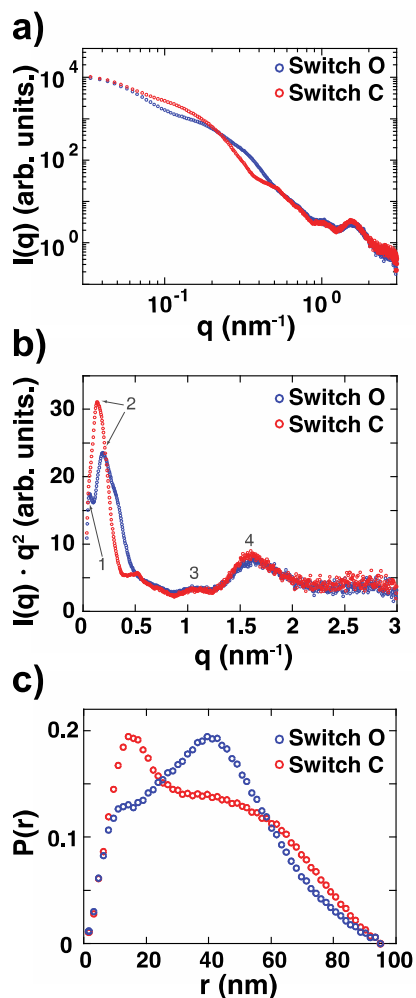


Figure 3. SAXS data reveal conformational features for open and closed switch objects. (a) Double-logarithmic representation of scattering intensity profiles obtained from the switch O and switch C sample. (b) Kratky representation of the data from (a) scaled by a constant factor. Numbers indicate peaks, which are described in the main text. (c) Pair distance distribution function $P(r)$ calculated from data shown in (a) assuming a maximum particle dimension D_{max} of 95 nm. $P(r)$ functions are normalized to equal areas.

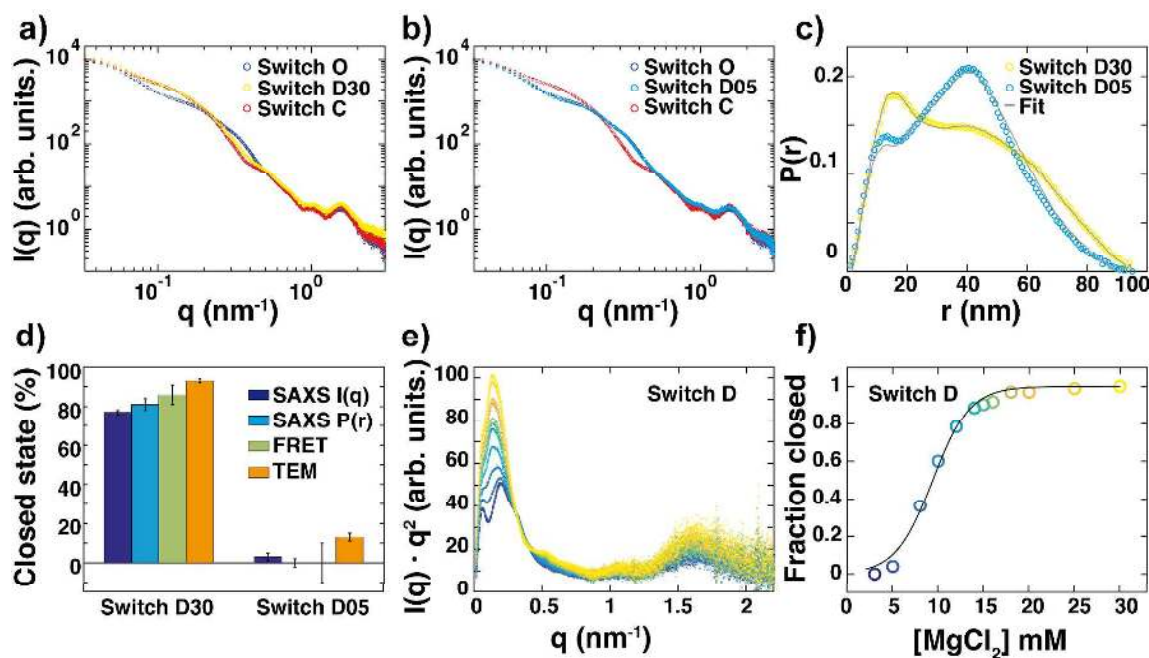


Figure 4. Characterization of conformational states of dynamic switch structures. (a) Comparison of the scattering profile from the switch D30 (yellow) to the scattering profiles of the static switch samples switch O (blue) and switch C (red). (b) Scattering profile from the switch D05 sample (cyan) in comparison to scattering curves from the static structures (same color code as in (a)). (c) $P(r)$ functions of the dynamic switch variants (cyan, yellow circles) and the resulting two-state model fits (grey lines). (d) Comparison of the relative fractions of the closed states determined from the scattering profiles (blue bars), the $P(r)$ functions (cyan bars), ensemble FRET (green bars), and TEM imaging (orange bars) for the switch D30 and switch D05 samples, corresponding to MgCl_2 concentrations of 30 mM and 5 mM, respectively. For TEM imaging the highest MgCl_2 concentration was 25 mM. (e) Kratky representation of the scattering profiles of switch D samples for varying MgCl_2 concentrations: 3 mM (dark blue, bottom), 5 mM, 8 mM, 10 mM, 12 mM, 14 mM, 15 mM, 16 mM, 18 mM, 20 mM, 25 mM and 30 mM (light yellow, top). Data are normalized to the intensity at zero scattering angle and scaled by a constant factor. (f) Fraction of closed switch particles for MgCl_2 titration experiments shown in (e), determined from a two-state model. Solid lines represent a two-state model with a free energy term that depends linearly on the MgCl_2 concentration.

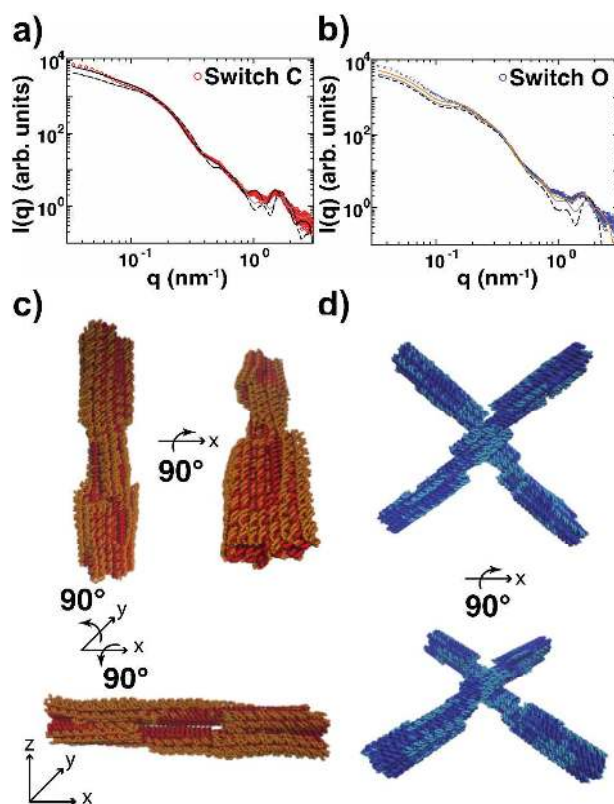


Figure 5. Normal mode based refinement of DNA origami structures against SAXS data. (a) shows data for the switch C construct and (b) the corresponding results for the switch O sample. Experimental scattering profiles are shown as red or blue circles. Scattering profiles predicted from the initial, CanDo derived models using all atoms and the software *CRY SOL* are shown as grey lines and using a one-bead-per-base representation as dashed black lines. Scattering profiles for the final models (computed using the one-bead-per-base representation) after normal mode based refinement are shown as black (for switch C) and orange (for switch O) lines. (c) and (d) show the initial models for the switch C and switch O objects as red and blue tubes and the final models after normal mode refinement as orange and cyan spheres, respectively.

REFERENCES

- (1) Seeman, N. C. *Nature* **2003**, *421* (6921), 427–431.
- (2) Rothmund, P. W. K. *Nature* **2006**, *440* (7082), 297–302.
- (3) Winfree, E.; Liu, F.; Wenzler, L. A.; Seeman, N. C. *Nature* **1998**, *394* (6693), 539–544.
- (4) Castro, C. E.; Kilchherr, F.; Kim, D.-N.; Shiao, E. L.; Wauer, T.; Wortmann, P.; Bathe, M.; Dietz, H. *Nature Methods* **2011**, *8* (3), 221–229.
- (5) Douglas, S. M.; Dietz, H.; Liedl, T.; Högberg, B.; Graf, F.; Shih, W. M. *Nature* **2009**, *459* (7245), 414–418.
- (6) Dietz, H.; Douglas, S. M.; Shih, W. M. *Science* **2009**, *325* (5941), 725–730.
- (7) Benson, E.; Mohammed, A.; Gardell, J.; Masich, S.; Czeizler, E.; Orponen, P.; Högberg, B. *Nature* **2015**, *523* (7561), 441–U139.
- (8) Ke, Y.; Ong, L. L.; Shih, W. M.; Yin, P. *Science* **2012**, *338* (6111), 1177–1183.
- (9) Andersen, E. S.; Dong, M.; Nielsen, M. M.; Jahn, K.; Subramani, R.; Mamdouh, W.; Golas, M. M.; Sander, B.; Stark, H.; Cristiano L. P. Oliveira; Pedersen, J. S.; Birkedal, V.; Besenbacher, F.; Gothelf, K. V.; Kjems, J. *Nature* **2009**, *459* (7243), 73–76.
- (10) Douglas, S. M.; Bachelet, I.; Church, G. M. *Science* **2012**, *335* (6070), 831–834.
- (11) Kuzyk, A.; Schreiber, R.; Zhang, H.; Govorov, A. O.; Liedl, T.; Liu, N. *Nature Materials* **2014**, *13* (9), 862–866.
- (12) Liu, M.; Fu, J.; Hejesen, C.; Yang, Y.; Woodbury, N. W.; Gothelf, K.; Liu, Y.; Yan, H. *Nature Communications* **2013**, *4*.
- (13) Zhang, D. Y.; Seelig, G. *Nat Chem* **2011**, *3* (2), 103–113.
- (14) Marras, A. E.; Zhou, L.; Su, H.-J.; Castro, C. E. *PNAS* **2015**, *112* (3), 713–718.
- (15) Kuzuya, A.; Ohya, Y. *Acc. Chem. Res.* **2014**, *47* (6), 1742–1749.
- (16) Zhao, Y.-X.; Shaw, A.; Zeng, X.; Benson, E.; Nyström, A. M.; Högberg, B. *ACS Nano* **2012**, *6* (10), 8684–8691.
- (17) Zhang, F.; Jiang, S.; Wu, S.; Li, Y.; Mao, C.; Liu, Y.; Yan, H. *Nature Nanotechnology* **2015**, *10* (9), 779–784.
- (18) Aghebat Rafat, A.; Pirzer, T.; Scheible, M. B.; Kostina, A.; Simmel, F. C. *Angew. Chem. Int. Ed. Engl.* **2014**, *53* (29), 7665–7668.
- (19) Langecker, M.; Arnaut, V.; Martin, T. G.; List, J.; Renner, S.; Mayer, M.; Dietz, H.; Simmel, F. C. *Science* **2012**, *338* (6109), 932–936.
- (20) Han, D.; Pal, S.; Nangreave, J.; Deng, Z.; Liu, Y.; Yan, H. *Science* **2011**, *332* (6027), 342–346.
- (21) Schreiber, R.; Luong, N.; Fan, Z.; Kuzyk, A.; Nickels, P. C.; Zhang, T.; Smith, D. M.; Yurke, B.; Kuang, W.; Govorov, A. O.; Liedl, T. *Nature Communications* **2013**, *4*, 2948.
- (22) Bai, X.-C.; Martin, T. G.; Scheres, S. H. W.; Dietz, H. *PNAS* **2012**, *109* (49), 20012–20017.
- (23) Koch, M. H. J.; Vachette, P.; Svergun, D. I. *Quarterly Reviews of Biophysics* **2003**, *36* (02), 147–227.
- (24) Lipfert, J.; Doniach, S. *Annu Rev Biophys Biomol Struct* **2007**, *36* (1), 307–327.
- (25) Rambo, R. P.; Tainer, J. A. *Annu Rev Biophys* **2013**, *42* (1), 415–441.
- (26) Putnam, C. D.; Hammel, M.; Hura, G. L.; Tainer, J. A. *Quarterly Reviews of Biophysics* **2007**, *40* (3), 191–285.
- (27) Eliezer, D.; Jennings, P. A.; Wright, P. E.; Doniach, S.; Hodgson, K. O.; Tsuruta, H. *Science* **1995**, *270* (5235), 487–488.
- (28) Akiyama, S.; Takahashi, S.; Kimura, T.; Ishimori, K.; Morishima, I.; Nishikawa, Y.; Fujisawa, T. *PNAS* **2002**, *99* (3), 1329–1334.

- 1
2
3 (29) Doniach, S. *Chem. Rev.* **2001**, *101* (6), 1763–1778.
- 4 (30) Fang, X.; Littrell, K.; Yang, X.-J.; Henderson, S. J.; Siefert, S.; Thiyagarajan, P.; Pan,
5 T.; Sosnick, T. R. *Biochemistry* **2000**, *39* (36), 11107–11113.
- 6 (31) Russell, R.; Millett, I. S.; Tate, M. W.; Kwok, L. W.; Nakatani, B.; Gruner, S. M.;
7 Mochrie, S. G. J.; Pande, V.; Doniach, S.; Herschlag, D.; Pollack, L. *Proc. Natl.*
8 *Acad. Sci. U.S.A.* **2002**, *99* (7), 4266–4271.
- 9 (32) Herschlag, D.; Russell, R.; Millett, I. S.; Doniach, S. *Nat. Struct Biol.* **2000**, *7* (5),
10 367–370.
- 11 (33) Tuukkanen, A. T.; Svergun, D. I. *FEBS J.* **2014**, *281* (8), 1974–1987.
- 12 (34) Bessman, N. J.; Bagchi, A.; Ferguson, K. M.; Lemmon, M. A. *Cell Rep* **2014**, *9* (4),
13 1306–1317.
- 14 (35) Rambo, R. P.; Tainer, J. A. *RNA* **2010**, *16* (3), 638–646.
- 15 (36) Lipfert, J.; Das, R.; Chu, V. B.; Kudaravalli, M.; Boyd, N.; Herschlag, D.; Doniach,
16 S. *J. Mol. Biol.* **2007**, *365* (5), 1393–1406.
- 17 (37) Gerling, T.; Wagenbauer, K. F.; Neuner, A. M.; Dietz, H. *Science* **2015**, *347* (6229),
18 1446–1452.
- 19 (38) Sim, A. Y. L.; Lipfert, J.; Herschlag, D.; Doniach, S. *Phys. Rev. E* **2012**, *86* (2),
20 021901.
- 21 (39) Ali, M.; Lipfert, J.; Seifert, S.; Herschlag, D.; Doniach, S. *J. Mol. Biol.* **2010**, *396* (1),
22 153–165.
- 23 (40) Lipfert, J.; Ouellet, J.; Norman, D. G.; Doniach, S.; Lilley, D. M. J. *Structure* **2008**,
24 *16* (9), 1357–1367.
- 25 (41) Bai, Y.; Das, R.; Millett, I. S.; Herschlag, D.; Doniach, S. *Proc. Natl. Acad. Sci.*
26 *U.S.A.* **2005**, *102* (4), 1035–1040.
- 27 (42) Meisburger, S. P.; Sutton, J. L.; Chen, H.; Pabit, S. A.; Kirmizialtin, S.; Elber, R.;
28 Pollack, L. *Biopolymers* **2013**, *99* (12), 1032–1045.
- 29 (43) Kratky, O.; Glatter, O. *Academic Press, London* **1982**, 1–267.
- 30 (44) Zuo, X.; Cui, G.; Merz, K. M.; Zhang, L.; Lewis, F. D.; Tiede, D. M. *Proc. Natl.*
31 *Acad. Sci. U.S.A.* **2006**, *103* (10), 3534–3539.
- 32 (45) Chiu, T. K.; Dickerson, R. E. *J. Mol. Biol.* **2000**, *301* (4), 915–945.
- 33 (46) Martin, T. G.; Dietz, H. *Nature Communications* **2012**, *3*, 1103.
- 34 (47) Frank, J. *Annu Rev Biophys Biomol Struct* **2002**, *31* (1), 303–319.
- 35 (48) Hura, G. L.; Menon, A. L.; Hammel, M.; Rambo, R. P.; Poole, F. L., II; Tsutakawa,
36 S. E.; Jenney, F. E., Jr; Classen, S.; Frankel, K. A.; Hopkins, R. C.; Yang, S.-J.; Scott,
37 J. W.; Dillard, B. D.; Adams, M. W. W.; Tainer, J. A. *Nature Methods* **2009**, *6* (8),
38 606–612.
- 39 (49) Svergun, D.; Barberato, C.; Koch, M. *J Appl Crystallogr* **1995**, *28* (6), 768–773.
- 40 (50) Schneidman-Duhovny, D.; Hammel, M.; Sali, A. *Nucl. Acids Res.* **2010**, *38* (Web
41 Server issue), W540–W544.
- 42 (51) Poitevin, F.; Orland, H.; Doniach, S.; Koehl, P.; Delarue, M. *Nucl. Acids Res.* **2011**,
43 *39* (Web Server issue), W184–W189.
- 44 (52) Ravikumar, K. M.; Huang, W.; Yang, S. *The Journal of Chemical Physics* **2013**, *138*
45 (2), 024112–024118.
- 46 (53) Pan, K.; Kim, D.-N.; Zhang, F.; Adendorff, M. R.; Yan, H.; Bathe, M. *Nature*
47 *Communications* **2014**, *5*, 5578–7.
- 48 (54) Svergun, D. I.; Richard, S.; Koch, M. H.; Sayers, Z.; Kuprin, S.; Zaccari, G. *Proc.*
49 *Natl. Acad. Sci. U.S.A.* **1998**, *95* (5), 2267–2272.
- 50 (55) Pollack, L. *Annu Rev Biophys* **2011**, *40* (1), 225–242.
- 51 (56) Lipfert, J.; Doniach, S.; Das, R.; Herschlag, D. *Annu. Rev. Biochem.* **2014**, *83* (1),
52 813–841.
- 53 (57) Meisburger, S. P.; Pabit, S. A.; Pollack, L. *Biophysj* **2015**, *108* (12), 2886–2895.
- 54
55
56
57
58
59
60

- 1
2
3 (58) Gavryushov, S.; Zielenkiewicz, P. *Biophysj* **1998**, *75* (6), 2732–2742.
4 (59) Yoo, J.; Aksimentiev, A. *Biophysj* **2014**, *106* (2), 695a.
5 (60) Kim, D. N.; Kilchherr, F.; Dietz, H.; Bathe, M. *Nucl. Acids Res.* **2012**, *40* (7), 2862–
6 2868.
7 (61) Yoo, J.; Aksimentiev, A. *PNAS* **2013**, *110* (50), 20099–20104.
8 (62) Sedeh, R. S.; Pan, K.; Adendorff, M. R.; Hallatschek, O.; Bathe, K.-J.; Bathe, M. *J*
9 *Chem Theory Comput* **2016**, *12* (1), 261–273.
10 (63) Maffeo, C.; Yoo, J.; Aksimentiev, A. *Nucl. Acids Res.* **2016**, gkw155.
11 (64) Miyashita, O.; Gorba, C.; Tama, F. *Journal of Structural Biology* **2011**, *173* (3), 451–
12 460.
13 (65) López-Blanco, J. R.; Miyashita, O.; Tama, F.; Chacón, P. *Normal Mode Analysis*
14 *Techniques in Structural Biology*; John Wiley & Sons, Ltd: Chichester, UK, 2001; pp
15 1–12.
16 (66) Tama, F.; Sanejouand, Y. H. *Protein Eng.* **2001**, *14* (1), 1–6.
17 (67) Delarue, M. *Acta Crystallogr Sect D Biol Crystallogr* **2008**, *64* (Pt 1), 40–48.
18 (68) Hinsén, K. *Proteins* **1998**, *33* (3), 417–429.
19 (69) Tama, F.; Valle, M.; Frank, J.; Brooks, C. L. *Proc. Natl. Acad. Sci. U.S.A.* **2003**, *100*
20 (16), 9319–9323.
21 (70) Schröder, G. F.; Levitt, M.; Brunger, A. T. *Nature* **2010**, *464* (7292), 1218–1222.
22 (71) Gorba, C.; Tama, F. *Bioinform Biol Insights* **2010**, *4*, 43–54.
23 (72) Zheng, W.; Tekpinar, M. *Biophys. J.* **2011**, *101* (12), 2981–2991.
24 (73) Panjkovich, A.; Svergun, D. I. *Physical Chemistry Chemical Physics* **2016**, *18* (8),
25 5707–5719.
26 (74) Pérez, J.; Vachette, P.; Russo, D.; Desmadril, M.; Durand, D. *J. Mol. Biol.* **2001**, *308*
27 (4), 721–743.
28 (75) Pollack, L. *Biopolymers* **2011**, *95* (8), 543–549.
29 (76) Baird, N. J.; Westhof, E.; Qin, H.; Pan, T.; Sosnick, T. R. *J. Mol. Biol.* **2005**, *352* (3),
30 712–722.
31
32
33
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