



## SPECIAL TOPIC: Graphene Oxides towards Practical Applications

# Coordination polymer nanowires/reduced graphene oxide paper as flexible anode for sodium-ion batteries

Zehang Sun, Ke Tan, Linrui Hou, Yang Liu\* and Changzhou Yuan\*

The growing demand for lithium-ion batteries (LIBs) will lead to the shortage and high cost of lithium resource in the foreseeable future. Thus, it is essential to develop alternative battery technologies which are reliable and cost-effective [1]. Sodium-ion batteries (SIBs) have been regarded as one of the most promising alternatives for LIBs due to the abundance of sodium [2]. However, the large radius of  $\text{Na}^+$  ion always brings about low kinetics and instability of batteries. Therefore, the major challenge in advancing SIB technology lies in finding suitable host materials for  $\text{Na}^+$  ions, especially high performance anode materials. Until now, the sodium storage capabilities of some inorganic (e.g., metal oxides and sulfides, alloying type metals) and organic (e.g., carboxylates, imides) materials have been tested [3–7]. Although metal-based inorganic electrodes exhibit high capacity in the storage of  $\text{Na}^+$  ions, large volume expansion and severe capacity fading arise during the conversion or alloying reactions [3–5]. Organic electrodes in SIBs mainly based on C=O, C=N bonds and doping reactions provide the possibility for stable accommodation of large  $\text{Na}^+$  ions [6]. Besides, organic electrode materials possess other advantages such as abundance, environmental friendliness, structural diversity and flexibility. However, the inherent poor conductivity and high solubility in non-aqueous electrolytes become the bottlenecks limiting their further developments [7].

Polymerization has been confirmed to be an efficient way to mitigate the solubility of organics in electrolyte. The large extended backbone of repeating units could largely improve the stability of organics. Thus, different kinds of polymers such as conductive polymers (e.g., polypyrrole, polyaniline); carbonyl polymers (e.g., polyimides, polyquinones), radical polymers (e.g., nitroxide-based polymer) and coordination polymers (CPs) (e.g.,

Prussian blue) have been synthesized and tested as cathode or anode materials for SIBs [6–8]. Among them, CPs, especially their porous type namely metal organic frameworks (MOFs), have attracted lots of attention due to their facile synthesis, structural versatility, large surface area and enormous channels [9]. The CPs can be treated as one-, two- or three dimensional (1D, 2D or 3D) networks resulting from the coordination of metal ions and organic ligands [10]. As one important kind of CPs, MOFs always have well defined crystalline and porous structures while CPs may not. The CPs-based materials have been used as electrode materials for SIBs. For example, Prussian blue and its analogues with a general chemical formula of  $A_xM[\text{Fe}(\text{CN})_6]_y \cdot z\text{H}_2\text{O}$  (A represents alkali metal ion,  $M_a$  and  $M_b$  are transition metal ions) have been treated as one of the most promising SIB cathodes as  $\text{Na}^+$  ions can frequently insert into and desert from their lattices in a relatively high voltage window [11]. The CPs can also store  $\text{Na}^+$  ions in low voltage potentials. For instance, Co- and Cr-based MOFs were synthesized by Dong *et al.* [12] through the coordination of 5-amino-isophthalic acid with  $\text{Co}^{2+}$  and  $\text{Cr}^{2+}$  ions, respectively, and tested as anode materials for SIBs. It has been confirmed that the redox reactions were carried out between  $\text{Na}^+$  ions and carboxyl/amine groups; nevertheless, the valences of  $\text{Co}^{2+}$  and  $\text{Cr}^{2+}$  were not changed during the charge and discharge processes. Recently, Co-hexaaminobenzene (Co-HAB) was synthesized by Park *et al.* [13], which could provide a three-electron redox reaction in a voltage window of 0.5–3.0 V, presenting a new promising anode material for SIBs.

Although, some good results have been achieved, there are still obstacles in designing and constructing CPs-based electrodes. One of the challenges is to improve their electrical conductivity and structural stability during

School of Materials Science & Engineering, University of Jinan, Jinan 250022, China

\* Corresponding authors (emails: [mse\\_liuy@ujn.edu.cn](mailto:mse_liuy@ujn.edu.cn) (Liu Y); [ayuan@163.com](mailto:ayuan@163.com) or [mse\\_yuancz@ujn.edu.cn](mailto:mse_yuancz@ujn.edu.cn) (Yuan C))

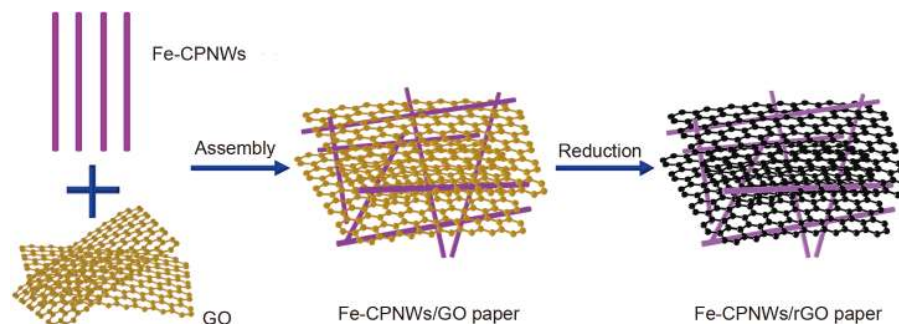
electrochemical reactions. Slurry coating method is commonly used in constructing CP electrodes by using the current collector, conductive agent and polymeric binder, resulting in a complicated technology and low energy densities [14–16]. The fabrication of CPs/carbon hybrids is an efficient way to solve these problems because carbonaceous materials (e.g., graphene) can not only act as fast electron transport networks but also provide stable structural protections [17–19]. Moreover, the excellent mechanical properties of graphene make it suitable substrate to construct flexible electrodes. Herein, flexible, interconnected CP nanowires/reduced graphene oxide (CPNWs/rGO) paper was constructed through a facile assembly and reduction process with Fe-based CP nanowires (Fe-CPNWs) and GO as precursor. The Fe-CPNWs constructed by  $\text{Fe}^{2+}$  (metal ions) and nitrilotriacetic acid (organic ligands) have a high aspect ratio and a large amount of functional groups (e.g., carboxyl), which can not only act as building blocks for free-standing electrodes but also provide enormous active sites for  $\text{Na}^+$  ion storage. The hybrid paper was used as a free-standing electrode for SIBs, showing a relatively high specific capacity, superior rate capability, and stable cycling performance.

As schematically illustrated in Fig. 1, the synthesis of Fe-CPNWs/rGO paper is facile with two steps. Firstly, the 1D Fe-CPNWs with  $\text{Fe}^{2+}$  ions on the surface are electrostatically attached to the 2D GO nanosheets containing negatively charged oxygen-containing functional groups, then assembled into paper structure *via* filtration. Secondly, GO is thermally reduced at a relatively low temperature of  $300^\circ\text{C}$ , which can improve the electrochemical conductivity of GO nanosheets without damaging the crystal structure of Fe-CPNWs.

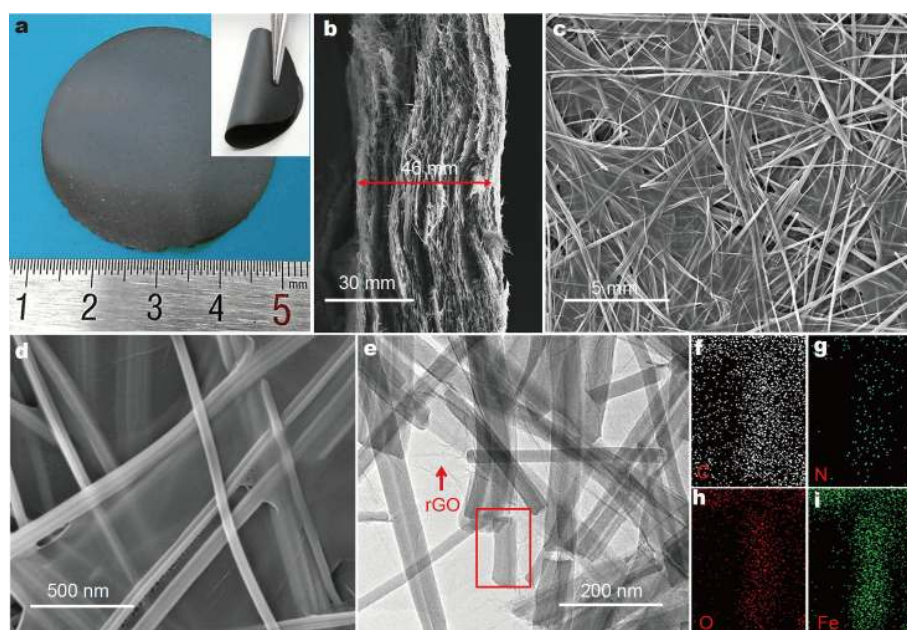
As shown in Fig. 2a, the free-standing and flexible paper with a diameter of  $\sim 4$  cm was obtained after the synthetic procedures. The paper structure exhibits good mechanical deformations such as bending, twisting and

folding (Fig. 2a and Fig. S1). Field-emission scanning electron microscopy (FESEM) and transmission electron microscopy (TEM) were used to investigate the morphology and microstructure of the Fe-CPNWs/rGO paper. The cross-section FESEM image (Fig. 2b) exhibits the thickness of paper structure is  $\sim 46$   $\mu\text{m}$ . Meanwhile, the top-view image (Fig. 2c) depicts the hybrid paper is constructed by 1D Fe-CPNWs (with a diameter of  $\sim 150$  nm) and 2D rGO nanosheets with enormous macro-pores. The magnified image (Fig. 2d) exhibits the Fe-CPNWs are closely attached to the surface of rGO nanosheets, consisting with the TEM image (Fig. 2e). Furthermore, the Fe-CPNWs consist of C, N, O and Fe elements confirmed by the elemental mapping images (Fig. 2f–i). Noteworthy, there is little morphology change of Fe-CPNWs before and after the thermal treatment (Fig. S2). As reported, the 1D/2D hybrid structure always endows materials with good mechanical and electrochemical properties due to their synergistic effect [20,21]. Moreover, the porous networks can provide enough channels for the soakage of electrolyte and diffusion of  $\text{Na}^+$  ions [22,23]. Thus, outstanding electrochemical performance may be expected.

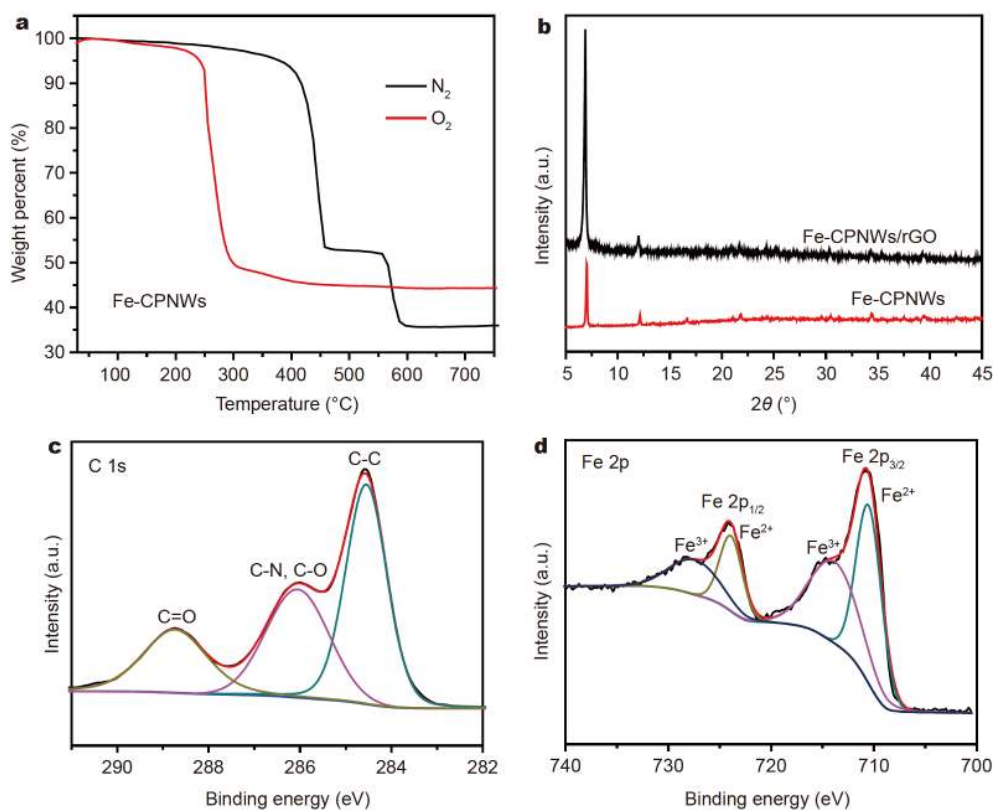
A series of measurements including thermal gravimetric analysis (TGA), X-ray diffraction (XRD), and X-ray photoelectron spectroscopic (XPS) have been taken to investigate the crystal structure and composition of Fe-CPNWs and their hybrids. As shown in Fig. 3a, the Fe-CPNWs have a high thermal stability with nearly no weight loss before  $300^\circ\text{C}$  under  $\text{N}_2$ . Afterwards, there are two drastic weight decrements ranged in  $400$ – $458^\circ\text{C}$  and  $556$ – $600^\circ\text{C}$ , which can be ascribed to the decomposition of organic ligands and the following carbonization process. The Fe-CPNWs show less thermal stability in  $\text{O}_2$  atmosphere as the sample begins to be oxidized at a lower temperature of  $\sim 200^\circ\text{C}$ . The final weight loss is 55.6 wt% due to the oxidation of Fe-CPNWs to iron oxide ( $\text{Fe}_2\text{O}_3$ ). It can be inferred that the chemical formula of Fe-



**Figure 1** Schematic illustration for the construction of Fe-CPNWs/rGO paper.



**Figure 2** (a) Digital images, (b) cross-section and (c, d) top-view FESEM images, (e) TEM image and (f–i) corresponding elemental mapping images (C, N, O, Fe) of the Fe-CPNWs/rGO paper.



**Figure 3** (a) TGA of Fe-CPNWs conducted in  $N_2$  and  $O_2$  atmospheres, respectively. (b) XRD patterns of pristine Fe-CPNWs and Fe-CPNWs/rGO paper. (c, d) XPS spectra (C 1s and Fe 2p) of the Fe-CPNWs/rGO paper.

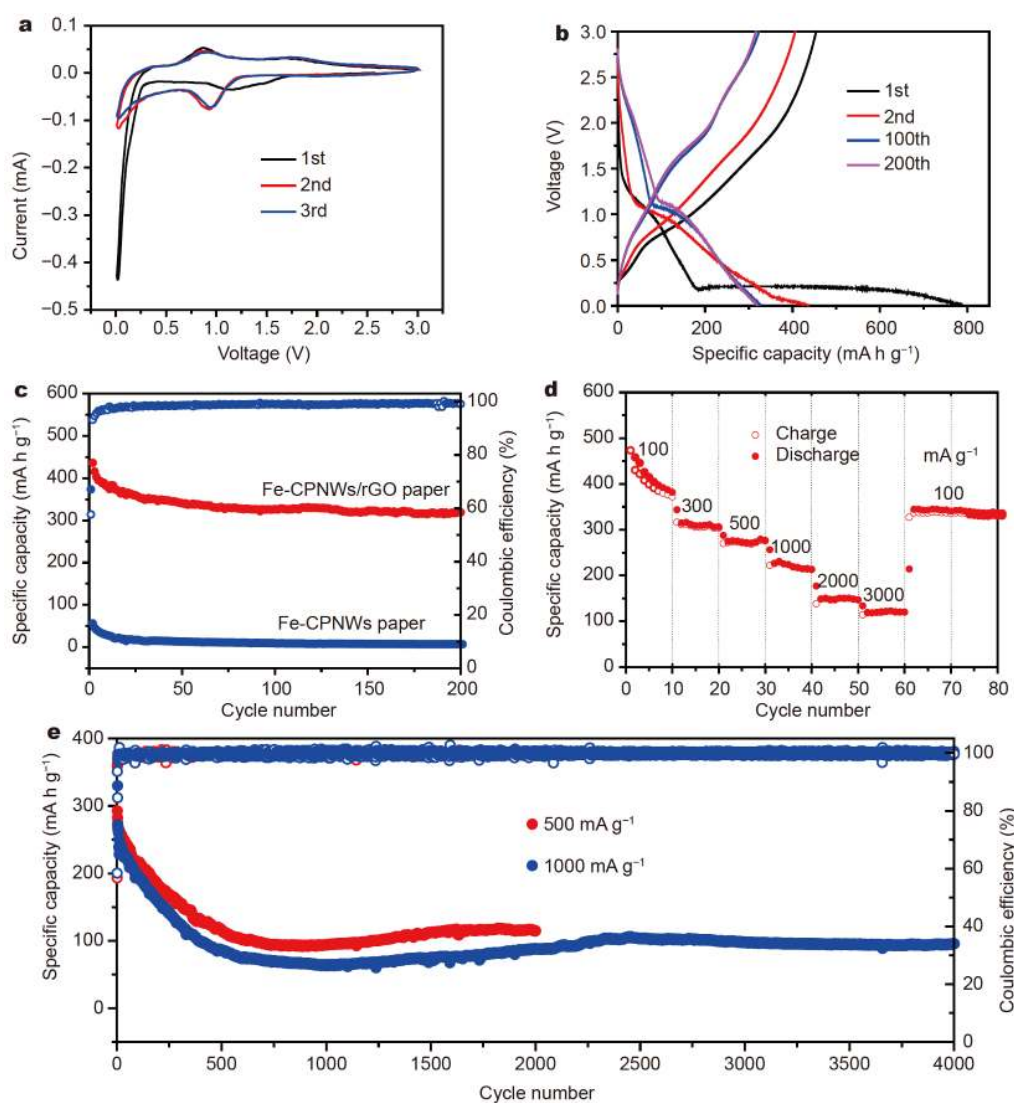
CPNWs is  $\text{Fe}_3(\text{NA})_2$  (NA represents nitrilotriacetic acid) based on the weight loss. According to the results of TGA, a suitable temperature of 300°C has been chosen to modestly reduce the functional groups attached on GO, and keep the intact structure of Fe-CPNWs. Fig. 3b shows that all the diffraction peaks of Fe-CPNWs in the hybrid paper are in accordance with the pristine samples, confirming there is no crystal structure change. However, the intensity of peaks becomes stronger, which can be attributed to the crystallization process during the thermal treatment.

XPS were analyzed to find out the surface chemical compositions of Fe-CPNWs/rGO. As shown in all survey spectrum (Fig. S3), there are C, O, N, Fe elements coexisting in the hybrid, which is in accordance with the elemental mapping. The C 1s spectrum can be deconvoluted into three peaks. The C–C peak located at 284.6 eV is assigned to the carbon framework of graphene. The other two peaks at 286.1 and 288.8 eV belong to C–N/C–O (amine/hydroxyl) and C=O (carboxyl and carbonyl) groups, deriving from the organic ligands (NA) and partial reduced graphene oxide nanosheets [24]. The Fe 2p spectrum (Fig. 3d) displays two couple peaks located at 710.6/724.0 eV and 714.4/727.9 eV, respectively. The former peaks can be attributed to  $\text{Fe}^{2+}$  ions coordinated with carboxyl (OCO–Fe) and amine (N–Fe) groups [24,25], while the latter  $\text{Fe}^{3+}$  ions come from the oxidation of surface  $\text{Fe}^{2+}$  ions due to their less stability in air.

Benefiting from the flexible mechanical property, the Fe-CPNWs/rGO paper is directly used as free-standing anode for SIBs. Fig. 4a shows the cyclic voltammetry (CV) curves of the paper electrode tested in a voltage window of 0.01–3.0 V at a scan rate of 0.1  $\text{mV s}^{-1}$ . During the first cathodic scan, the broad reduction peak from 1.6 to 0.9 V can be ascribed to the organic ligand transformation (i.e., carboxylate to enolate conversion), which becomes stronger in the following cycles due to the activation process [26]. The sharp peak sloping down to 0.01 V is associated with the formation of solid electrolyte interface (SEI) film and the side reactions between the surface functional groups attached on polymers and  $\text{Na}^+$  ions [12,27–29]. Two oxidation peaks centered at 1.79 V can be ascribed to the  $\text{Na}^+$  desertion process. In the subsequent cycles, a new reduction peak at 0.93 V appears, due to the structure reconstruction in the first cycle. The CV curves are basically overlapped thereafter, indicating a good reversibility. To further investigate the electrochemical mechanism of electrodes, *ex-situ* XPS analyses of the electrode materials collected at the dis-

charge state after 5 cycles were done. As shown in the Fe 2p spectrum (Fig. S4a), there are only two peaks at 711.0 and 724.6 eV, which can be ascribed to the  $\text{Fe}^{2+}$  ions [25]. The absence of Fe(0) indicates the valence of centered metal atoms remains unchanged during the electrochemical reactions. However, the peak intensities and areas of C=O and C–N/C–O groups decrease significantly at the discharge state (C 1s spectrum, Fig. S4b). The above results point out that the organic moieties especially carboxylic groups play an essential role in the reversible redox reactions.

The initial discharge and charge capacities of the Fe-CPNWs/rGO paper are 787 and 454  $\text{mA h g}^{-1}$  with a coulombic efficiency (CE) of 57.7% (Fig. 4b). The irreversible part can be attributed to the formation of SEI film and the side reactions [12,27–29]. The CE reaches over 96% for the 5th cycle and can be kept from then on. Fig. 4c shows the cycling performance of the Fe-CPNWs/rGO paper at a current density of 100  $\text{mA g}^{-1}$ . The paper electrode demonstrates a very stable cycling performance as a discharge capacity of 319  $\text{mA h g}^{-1}$  can still be kept after 200 cycles, which is superior to the Fe-CPNWs paper (7  $\text{mA h g}^{-1}$ ) without rGO addition (Fig. S5a). Evidently, the rGO substrate can not only improve the electrochemical conductivity of the hybrid electrodes (as confirmed by electrochemical impedance spectra, Fig. S6), but also help keep their structures intact (Figs S7 and S8). In addition to good cycling stability, the Fe-CPNWs/rGO paper also exhibits excellent rate capability, yielding average reversible capacities of 404, 308, 272, 220, 148 and 120  $\text{mA h g}^{-1}$  at the current densities of 100, 300, 500, 1000, 2000 and 3000  $\text{mA g}^{-1}$ , respectively. Meanwhile, the capacity can restore to 340  $\text{mA h g}^{-1}$  after deep cycling at a high rate. In contrast, the Fe-CPNWs paper without rGO shows inferior rate performance as only 3  $\text{mA h g}^{-1}$  can be obtained at 500  $\text{mA g}^{-1}$  (Fig. S5b). To further confirm the rate capability and cyclic stability, long cycling performances of the Fe-CPNWs/rGO paper electrode under high rates of 500 and 1000  $\text{mA g}^{-1}$  were tested (Fig. 4e). The capacity loss at the initial 500 cycles can be ascribed to the formation of unstable SEI layers at a high current density, which slow down the Na-ion transport behavior and degrade the electrochemical performance [16,22]. However, stable SEI layer is formed during the long cycling performance and capacities of 115 and 100  $\text{mA h g}^{-1}$  can be achieved after 2000 and 4000 cycles with very low capacity decay rates of 0.03% (at 500  $\text{mA g}^{-1}$ ) and 0.02% (at 1000  $\text{mA g}^{-1}$ ) per cycle. The electrochemical performance of the Fe-CPNWs/rGO paper is superior to many organic anodes previously re-



**Figure 4** Electrochemical performance of the Fe-CPNWs/rGO paper for SIBs. (a) CV curves ( $0.1 \text{ mV s}^{-1}$ ) between 0.01 and 3.0 V. (b) Galvanostatic charge and discharge voltage profiles ( $100 \text{ mA g}^{-1}$ ). (c) Comparative cycling performance ( $100 \text{ mA g}^{-1}$ ) of Fe-CPNWs/rGO paper and Fe-CPNWs paper without rGO. (d) Rate capability at various current rates from 100 to  $3000 \text{ mA g}^{-1}$ . (e) Long-term cycling performance at 500 and  $1000 \text{ mA g}^{-1}$ .

ported (Table S1) [12,18,30–34]. In addition, a full sodium ion cell was assembled with  $\text{Na}_3\text{V}_2(\text{PO}_4)_3/\text{rGO}$  (NVP/rGO) as the cathode and Fe-CPNWs/rGO as the anode. A voltage window of 0.5–3.9 V has been chosen according to the voltage profiles and corresponding  $dQ/dV$  curves of NVP/rGO and Fe-CPNWs/rGO (Fig. S9a, b). The full cell delivers a relatively stable cycling performance as a charge capacity of  $171 \text{ mA h g}^{-1}$  can be kept after 100 cycles at  $100 \text{ mA g}^{-1}$  (Fig. S9c, d). Moreover, a flexible pouch-type SIB was also fabricated and the light emitting diode (LED) can be easily lightened under the normal and bending states (Fig. S10).

In conclusion, CP-based flexible SIB anodes consisting

of 1D Fe-CPNWs and 2D rGO are constructed *via* the assembly and controlled reduction process. The multi-dimensional hybrid electrodes exhibit excellent mechanical and electrochemical properties. When tested as free-standing paper anodes for SIBs, the hybrids exhibit high rate capability ( $404\text{--}120 \text{ mA h g}^{-1}$  at a current density of  $100\text{--}3000 \text{ mA g}^{-1}$ ) and long cyclic life ( $100 \text{ mA h g}^{-1}$  can be kept after 4000 cycles at  $1000 \text{ mA g}^{-1}$ ). The organic moieties (carboxyl and amine groups) are the redox centers in CPs. This work provides a new flexible anode candidate for SIBs.

Received 9 December 2019; accepted 26 December 2019;  
published online 17 January 2020

- 1 Liu Y, Sun Z, Tan K, *et al.* Recent progress in flexible non-lithium based rechargeable batteries. *J Mater Chem A*, 2019, 7: 4353–4382
- 2 Deng J, Luo WB, Chou SL, *et al.* Sodium-ion batteries: From academic research to practical commercialization. *Adv Energy Mater*, 2018, 8: 1701428
- 3 Jin T, Han Q, Wang Y, *et al.* 1D nanomaterials: Design, synthesis, and applications in sodium-ion batteries. *Small*, 2018, 14: 1703086
- 4 Huang ZD, Zhang TT, Lu H, *et al.* Bimetal-organic-framework derived CoTiO<sub>3</sub> mesoporous micro-prisms anode for superior stable power sodium ion batteries. *Sci China Mater*, 2018, 61: 1057–1066
- 5 Li D, Zhao X, Yu R, *et al.* Formation of multi-shelled nickel-based sulfide hollow spheres for rechargeable alkaline batteries. *Inorg Chem Front*, 2018, 5: 535–540
- 6 Zhao Q, Lu Y, Chen J. Advanced organic electrode materials for rechargeable sodium-ion batteries. *Adv Energy Mater*, 2017, 7: 1601792
- 7 Lee S, Kwon G, Ku K, *et al.* Recent progress in organic electrodes for Li and Na rechargeable batteries. *Adv Mater*, 2018, 30: 1704682
- 8 Kim J, Kim JH, Ariga K. Redox-active polymers for energy storage nanoarchitectonics. *Joule*, 2017, 1: 739–768
- 9 Zhou W, Lv S, Liu X, *et al.* A directly grown pristine Cu-CAT metal–organic framework as an anode material for high-energy sodium-ion capacitors. *Chem Commun*, 2019, 55: 11207–11210
- 10 Batten SR, Champness NR, Chen XM, *et al.* Coordination polymers, metal–organic frameworks and the need for terminology guidelines. *CrystEngComm*, 2012, 14: 3001–3004
- 11 Lee HW, Wang RY, Pasta M, *et al.* Manganese hexacyanomanganate open framework as a high-capacity positive electrode material for sodium-ion batteries. *Nat Commun*, 2014, 5: 5280
- 12 Dong C, Xu L. Cobalt- and cadmium-based metal–organic frameworks as high-performance anodes for sodium ion batteries and lithium ion batteries. *ACS Appl Mater Interfaces*, 2017, 9: 7160–7168
- 13 Park J, Lee M, Feng D, *et al.* Stabilization of hexaaminobenzene in a 2D conductive metal–organic framework for high power sodium storage. *J Am Chem Soc*, 2018, 140: 10315–10323
- 14 You Y, Wu XL, Yin YX, *et al.* High-quality Prussian blue crystals as superior cathode materials for room-temperature sodium-ion batteries. *Energy Environ Sci*, 2014, 7: 1643–1647
- 15 Ren W, Qin M, Zhu Z, *et al.* Activation of sodium storage sites in Prussian blue analogues *via* surface etching. *Nano Lett*, 2017, 17: 4713–4718
- 16 Wu S, Wang W, Li M, *et al.* Highly durable organic electrode for sodium-ion batteries *via* a stabilized  $\alpha$ -C radical intermediate. *Nat Commun*, 2016, 7: 13318
- 17 Kim JK, Kim Y, Park S, *et al.* Encapsulation of organic active materials in carbon nanotubes for application to high-electrochemical-performance sodium batteries. *Energy Environ Sci*, 2016, 9: 1264–1269
- 18 Wang H, Hu P, Yang J, *et al.* Renewable-juglone-based high-performance sodium-ion batteries. *Adv Mater*, 2015, 27: 2348–2354
- 19 Ba D, Li Y, Sun Y, *et al.* Directly grown nanostructured electrodes for high-power and high-stability alkaline nickel/bismuth batteries. *Sci China Mater*, 2018, 62: 487–496
- 20 Jin T, Han Q, Jiao L. Binder-free electrodes for advanced sodium-ion batteries. *Adv Mater*, 2019, 46: 1806304
- 21 Tan S, Jiang Y, Wei Q, *et al.* Multidimensional synergistic nanoarchitecture exhibiting highly stable and ultrafast sodium-ion storage. *Adv Mater*, 2018, 30: 1707122
- 22 Liu Y, Yang Y, Wang X, *et al.* Flexible paper-like free-standing electrodes by anchoring ultrafine SnS<sub>2</sub> nanocrystals on graphene nanoribbons for high-performance sodium ion batteries. *ACS Appl Mater Interfaces*, 2017, 9: 15484–15491
- 23 Gui Q, Ba D, Zhao Z, *et al.* Synergistic coupling of ether electrolyte and 3D electrode enables titanates with extraordinary coulombic efficiency and rate performance for sodium-ion capacitors. *Small Methods*, 2019, 3: 1800371
- 24 Li C, Yin X, Chen L, *et al.* Synthesis of cobalt ion-based coordination polymer nanowires and their conversion into porous Co<sub>3</sub>O<sub>4</sub> nanowires with good lithium storage properties. *Chem Eur J*, 2010, 16: 5215–5221
- 25 Gu X, Yan C, Yan L, *et al.* Carbonates (bicarbonates)/reduced graphene oxide as anode materials for sodium-ion batteries. *J Mater Chem A*, 2017, 5: 24645–24650
- 26 Kim SH, Lee HH, Kim JM, *et al.* Heteromat-framed metal-organic coordination polymer anodes for high-performance lithium-ion batteries. *Energy Storage Mater*, 2019, 19: 130–136
- 27 Sharma N, Szunerits S, Boukherroub R, *et al.* Dual-ligand Fe-metal organic framework based robust high capacity Li ion battery anode and its use in a flexible battery format for electro-thermal heating. *ACS Appl Energy Mater*, 2019, 2: 4450–4457
- 28 Li C, Lou X, Shen M, *et al.* High anodic performance of Co 1,3,5-benzenetricarboxylate coordination polymers for Li-ion battery. *ACS Appl Mater Interfaces*, 2016, 8: 15352–15360
- 29 He H, Sun D, Tang Y, *et al.* Understanding and improving the initial coulombic efficiency of high-capacity anode materials for practical sodium ion batteries. *Energy Storage Mater*, 2019, 23: 233–251
- 30 Wang S, Wang L, Zhu Z, *et al.* All organic sodium-ion batteries with Na<sub>4</sub>C<sub>8</sub>H<sub>2</sub>O<sub>6</sub>. *Angew Chem*, 2014, 126: 6002–6006
- 31 Deng W, Qian J, Cao Y, *et al.* Graphene-wrapped Na<sub>2</sub>C<sub>12</sub>H<sub>6</sub>O<sub>4</sub> nanoflowers as high performance anodes for sodium-ion batteries. *Small*, 2016, 12: 583–587
- 32 Wu X, Ma J, Ma Q, *et al.* A spray drying approach for the synthesis of a Na<sub>2</sub>C<sub>6</sub>H<sub>2</sub>O<sub>4</sub>/CNT nanocomposite anode for sodium-ion batteries. *J Mater Chem A*, 2015, 3: 13193–13197
- 33 Zhu H, Yin J, Zhao X, *et al.* Humic acid as promising organic anodes for lithium/sodium ion batteries. *Chem Commun*, 2015, 51: 14708–14711
- 34 Zhao Q, Gaddam RR, Yang D, *et al.* Pyromellitic dianhydride-based polyimide anodes for sodium-ion batteries. *Electrochim Acta*, 2018, 265: 702–708

**Acknowledgements** This work was supported by the National Natural Science Foundation of China (51772127 and 51772131), Taishan Scholars (ts201712050), Major Program of Shandong Province Natural Science Foundation (ZR2018ZB0317), the Natural Science Doctoral Foundation of Shandong Province (ZR2019BEM038) and the Natural Science Doctoral Foundation of the University of Jinan (XBS1830).

**Author contributions** Sun Z, Tan K and Hou L performed the experiments; Liu Y and Yuan C designed and engineered the samples, and wrote the paper. All authors contributed to the general discussion.

**Conflict of interest** The authors declare no conflict of interest.

**Supplementary information** Experimental details and supporting data are available in the online version of the paper.



**Zehang Sun** received a BE degree from Liaocheng University in 2017. Currently, he is pursuing his MSc degree at the University of Jinan. His research interests focus on the structural design and electrochemical analyses of electrode materials for alkali-ion batteries.



**Changzhou Yuan** received a PhD from Nanjing University of Aeronautics and Astronautics in 2009. He is a distinguished professor of Taishan Scholar in the School of Materials Science and Engineering, University of Jinan. He is a Highly Cited Researcher (in Materials Science/Cross-field) by Clarivate Analysis from 2016 to 2019, and a Most Cited Chinese Researchers (in materials science) by Elsevier in 2016–2018. His current research focuses on the design and synthesis of micro/nano-materials for electrochemical energy-related applications.



**Yang Liu** obtained his PhD degree in chemical technology from Dalian University of Technology in 2017. He then joined the University of Jinan in September 2017. His research focuses on lithium ion batteries and the potential alternative energy storage devices.

### 配位聚合物纳米线/还原氧化石墨烯复合柔性薄膜电极的构筑及储钠性能研究

孙泽航, 谭可, 侯林瑞, 刘洋\*, 原长洲\*

**摘要** 构建基于有机材料的高性能柔性储钠电极面临诸多挑战. 本工作通过可控组装及还原的方式, 实现了铁基配位聚合物纳米线/还原氧化石墨烯柔性薄膜的构筑. 多维复合薄膜可直接用作钠离子电池自支撑负极, 且具有较高的储钠容量( $200 \text{ mA g}^{-1}$ 电流密度下可逆容量为 $319 \text{ mA h g}^{-1}$ )和优异的倍率性能( $3000 \text{ mA g}^{-1}$ 大电流密度下比容量可保持在 $\sim 120 \text{ mA h g}^{-1}$ ). 研究表明有机配体(氨三乙酸)中的羧基及氨基为储钠活性位点, 而配位金属离子( $\text{Fe}^{2+}$ )不参与电化学反应.