

Advances in Mechanical Engineering 2016, Vol. 8(8) 1-8 © The Author(s) 2016 DOI: 10.1177/1687814016663964 aime.sagepub.com

(S)SAGE

Effects of heterogeneity and load amplitude on fatigue rate prediction of a welded joint

Chunguo Zhang¹, Jizhuang Hui¹, Pengmin Lu¹, Xiaozhi Hu² and Jia Liang¹

Abstract

It is a contradiction to homogeneous material fatigue behavior characterized by widely used linear Paris law, welded-joint fatigue issues need to be reassessed because fatigue crack growth behavior going through heterogeneous region will be different. For a welded joint, log(da/dN) is no longer linearly related to $log(\Delta K)$ in heterogeneous region because of the change in fatigue properties resulting from the welding process. Theoretical model of the fatigue crack growth rate without artificial adjustable parameters was proposed by considering the effects of heterogeneity in a welded joint and load-amplitude variation on fatigue crack growth curve. In this fatigue heterogeneous region, the relationship between $\log(da/dN)$ and $\log(\Delta K)$ is similar to a concave-down parabola. Predicted results from the proposed model agreed better with the experimental data obtained from fatigue tests conducted in this study and open published literatures for welded joints in comparison to the widely used Paris model.

Keywords

Cyclic loading, fatigue, fracture mechanics, life-cycle assessment, structural engineering

Date received: 23 November 2015; accepted: 19 July 2016

Academic Editor: Chow-Shing Shin

Introduction

Fatigue crack growth behaviors have been studied for various types of widely used engineering structural materials in the past several decades, and a number of fatigue rate models have been establish with the help of the stress and strain field ahead of the crack tip together with a suitable failure criterion.¹⁻⁷ However, most of these fatigue rate models intrinsically contain adjustable material parameters that need to be determined numerically or experimentally. So far, Paris model is the most commonly used model which characterizes materials fatigue behavior, and works well for the second stage of fatigue in which fatigue crack growth rate (da/dN) is nearly linearly correlated with stress intensity factor range (ΔK) in a log-log coordinate space.³ However, the linear relationship is only for homogeneous materials, thus is not competent enough to characterize

heterogeneous materials because of the change in fatigue properties.⁸

It is a contradiction to homogeneous material fatigue behavior characterized by widely used linear Paris law, welded-joint fatigue issues need to be reassessed because fatigue crack growth behavior going through heterogeneous region (HR) will be different. For a welded joint,

¹Key Laboratory of Road Construction Technology and Equipment, MOE, Chang'an University, Xi'an, P.R. China ²School of Mechanical and Chemical Engineering, The University of Western Australia, Perth, WA, Australia

Corresponding author:

Chunguo Zhang, Key Laboratory of Road Construction Technology and Equipment, MOE, Chang'an University, Nan Er Huan Zhong Duan, Xi'an 710064, P.R. China. Email: zcguo2008@163.com

 \odot Creative Commons CC-BY: This article is distributed under the terms of the Creative Commons Attribution 3.0 License (http://www.creativecommons.org/licenses/by/3.0/) which permits any use, reproduction and distribution of the work without further permission provided the original work is attributed as specified on the SAGE and Open Access pages (https://us.sagepub.com/en-us/nam/ open-access-at-sage).

 $\log(da/dN)$ is no longer linearly related to $\log(\Delta K)$ in HR because of the change in fatigue properties resulting from the welding process.

Because of its simplicity and wide general applications, Paris law remains to be the most popular fatigue model to quantify fatigue crack growth,⁹ and it offers a sound base for modifications and improvements. To account for the effects of various factors on da/dN prediction, a number of modifications to the Paris model have been proposed, for instance, as follows: (1) the fracture toughness of material and the stress ratio during fatigue test were suggested to incorporate in the Paris model as reported in Forman et al.;¹⁰ (2) based on theoretical research and experimental validation, Barter et al. proposed the following form: $da/dN = C_1 a^{(1 - m^*/2)}$ $(\Delta K_{eff})^{m^*}$ (C₁, a*, m* are constants);^{11–13} and (3) an effective ΔK (ΔK_{eff}) was used instead of ΔK to account for the effect of crack closure.¹⁴ Due to its simplicity and wide general applications, fatigue Paris model offers a sound base for further modifications and improvements, for example, if "structural fatigue" issues need to be reassessed when fatigue crack growth behavior goes through dissimilar materials (e.g. welded joints).

Different to those aforementioned fatigue models for homogeneous materials, the primary objective of this study is to develop a fatigue rate model to predict the da/dN of the HR in a welded joint under constant amplitude loading with stress ratio (R) of 0, which accounts for both heterogeneity and load-amplitude effects. The da/dN predictions by the commonly used Paris model and the proposed model were compared and validated using open published experimental data.

Fatigue crack growth models

Effect of heterogeneity

Paris model is widely used to characterize material fatigue behavior, and works well for the second stage of fatigue, in which log(da/dN) is nearly linearly correlated with $log(\Delta K)$. According to the Paris formulation. $da/dN = c (\Delta K)^m$, da/dN gradually increases with increase in ΔK during fatigue crack propagation, which is suitable for homogeneous materials. However, fatigue issues of welded joints need to be reassessed because fatigue crack growth behavior going through HR, for example, heat-affected zone (HAZ), melted parent metal, or welded interfaces, will be different. It is highly likely for HR in a welded joint, log(da/dN) is no longer linearly related to $log(\Delta K)$ because of the change in fatigue properties resulting from the welding process; thus, their relationship needs to be reformulated. By taking weld-repaired infrastructure as an example, a transition region exists between weld metal (WM) and infrastructure. Potential problems resulted from welding



Figure 1. Welded-joint fatigue issues: (a) a weld-repaired infrastructure and (b) the potential da/dN versus ΔK curve.

process, for example, welding defects, cracking, residual stresses, and embrittlement, decrease the fatigue resistance in the transition region. The transition region that may consists of HAZ or part of it, melted parent metal, WM or part of it, and welded interfaces is the HR because of the change in fatigue properties (Figure 1(a)). In this fatigue HR, $\log(da/dN)$ is no longer linearly related to $\log(\Delta K)$ and the potential relationship needs to be reassessed, as shown in Figure 1(b).

A multitude of open published experimental data shows that $\log(da/dN)$ in fatigue HR of welded joints first increases to a local maximum at a point termed the weakest point of fatigue performance in this study, then decreases gradually to a local minimum along with the increase in $\log(\Delta K)$,^{15–18} and the relationship between $\log(da/dN)$ and $\log(\Delta K)$ is similar to a concave-down parabola. Assuming a parabolic fit for $\log(da/dN)$ – $\log(\Delta K)$ concave-down, da/dN for the HR of a welded joint can be defined as

$$\frac{\mathrm{da}}{\mathrm{dN}} = 10^{\mathrm{c}} (\Delta \mathrm{K})^{\mathrm{b} + \mathrm{a} \, \mathrm{lg}(\Delta \mathrm{K})} \tag{1}$$

where "a,""b," and "c" are parameters related to the configuration and dimensions of a welded joint, and the weakest point of fatigue performance in the welded joint.

The potential HR includes WM or part of it, melted parent metal, HAZ or part of it, and weld interface(s). According to equation (1), the $\log(da/dN)-\log(\Delta K)$ curve from the HR of a welded joint is an ideal concave-down parabola. It should be mentioned that the fatigue curve of the HR may be part of a parabola and will depend on the configuration, dimensions, and weakest-point site of a welded joint.

Effect of load-amplitude variation

Because equation (1) is an empirical model, fatigue data at a given load amplitude (e.g. P_0) are needed to determine the values of parameters " a_0 ,"" b_0 ," and " c_0 " in equation (1) by curve fitting. The values of these parameters are subsequently used to calculate new values " a_1 ,"" b_1 ," and " c_1 " to predict da/dN at a new load amplitude (e.g. P_1). The procedural steps for the model formulation are as follows.

According to linear elastic fracture mechanics, the stress components at a point near the crack tip during fatigue tensile tests with load amplitude P can be written as follows¹⁹

$$\sigma_{\rm x} = \sigma_{\rm y} = \frac{({\rm K}_1)_{\rm p}}{\sqrt{2\pi\gamma}} \tag{2}$$

where K_I is the stress intensity factor and γ is the polar radius of the point near the crack tip.

For a given crack length, the above equations indicate that the stress components near the crack tip are functions of K_I only because γ is a constant. K_I is a function of the applied load, crack length, and sample configuration and dimensions. For instance, as for the widely used compact tension (CT) sample, the following equation is used to determine stress intensity factor K_I

$$K_{I} = \frac{P}{B\sqrt{W}} \times f(\alpha) = \frac{P}{B\sqrt{W}} \times f\left(\frac{a}{W}\right)$$
 (3)

According to equations (2) and (3), the stress value at the crack tip, which corresponds to da/dN, increases with applied load amplitude P; that is, for $P_1 > P_0$, $((K_1)_{p_1}/\sqrt{2\pi\gamma}) > ((K_1)_{p_0}/\sqrt{2\pi\gamma})$, and $(da/dN)_{P_1} > (da/dN)_{P_0}$.

Because the fatigue crack growth curve from the HR of a welded joint is a concave-down parabola in the log-log coordinate space, log(da/dN) increases nonlinearly with the increase in $log(\Delta K)$ as the applied load amplitude increases. As a result, the crack growth curves from fatigue testing with different load amplitudes cannot coincide for identical welded joints as shown in Figure 2. For a specific welded joint, the fatigue resistance of the material ahead of the crack is governed hypothetically by the local stress range perpendicular to the crack growth direction for tensile (Mode I) loading.³ Taking as an example a point in the fatigue crack growth path marked "A" in Figure 2, the ΔK and da/dN are $(\Delta K_A)_{P_0}$ and $((da/dN)_A)_{P_0}$, respectively. If the load amplitude increases to P_1 , the corresponding ΔK and da/dN become $(\Delta K_A)_{P_1}$ and $((da/dN)_A)_{P_1}$, respectively. Thus, the "A" in the fatigue curve of P₀ was changed to "A'" in the new curve for P₁; likewise, the "O-point" was moved to "O'-point" in fatigue crack growth curve for the respective load amplitudes. Thus, for the HR of a welded joint, the



Figure 2. The effect of load-amplitude variation on the fatigue curve translation in log–log coordinate space.

 $\log(da/dN)$ - $\log(\Delta K)$ curve translates to upper right when the amplitude of cyclic loading increases. Vice versa, the $\log(da/dN)$ - $\log(\Delta K)$ curve translates to lower left when the amplitude of cyclic loading decreases, for example, from P₀ to P₂.

It should be mentioned that even after the translation of the log(da/dN)–log(ΔK) curve, the shape of the curve is preserved. Additionally, the symmetry axis and the vertex of the new fatigue curve for P₁ are different from those of the original fatigue curve for P₀. In other words, the proposed model given by equation (1) is suitable for the HR of a welded joint, but the parameters "a,""b," and "c" are different for different load amplitudes even for the exact same sample.

On a concave-down parabola, the dependent variable reaches its maximum on the axis of symmetry from the view of mathematical knowledge. For example, for a load amplitude of P_0 , log(da/dN) reaches its local maximum "O," at which point the following relationship, is satisfied

$$\lg (\Delta K_{P_0}) = -\frac{b_0}{2a_0}$$
 (4)

Likewise, for a load amplitude of P_1 , $\log(da/dN)$ reaches its local maximum "O'," at which point the following relationship, is satisfied

$$\lg(\Delta K_{P_1}) = -\frac{b_1}{2a_1} \tag{5}$$

For an identical sample, the weakest point of fatigue performance is fixed; thus, the fatigue crack length at which the local $(\log(da/dN))_{max}$ occurs is a constant at

different load amplitude which corresponds to a fixed "a" in equation (3).

For a specific welded-joint sample, the parameters "B,""W," and "d" are constants and independent of the load amplitude; moreover, the site of the weakest point, and thus the local $(\log(da/dN))_{max}$ is fixed. Combining equations (3)–(5), the following can be obtained

$$-\frac{b_1}{2a_1} = lgP_1 - lgP_0 - \frac{b_0}{2a_0}$$
(6)

Considering only a narrow band near the weakest point in a welded joint, it is assumed that the material properties within the narrow band are homogeneous, which is suitable for the Paris model. As for the narrow band, the following formulation can be obtained according to da/dN under fatigue test at load amplitudes of P_0 and P_1

$$\left(\frac{\mathrm{d}a}{\mathrm{d}N}\right)_{\mathbf{P}_{1}} = \left(\frac{\mathrm{d}a}{\mathrm{d}N}\right)_{\mathbf{P}_{0}} \times \left(\frac{\Delta K_{\mathbf{P}_{1}}}{\Delta K_{\mathbf{P}_{0}}}\right)^{\mathrm{m}} \tag{7}$$

According to proposed model, the local $(\log(da/dN))_{max}$ at load amplitudes P_0 and P_1 are, respectively

$$lg\left(\frac{da}{dN}\right)_{A} = \frac{4a_{0}c_{0} - b_{0}^{2}}{4a_{0}}$$
(8)

$$\lg \left(\frac{da}{dN}\right)_{A'} = \frac{4a_1c_1 - b_1^2}{4a_1}$$
(9)

Incorporating equations (8) and (9) into equation (7), the following formulation can be obtained

$$\frac{4a_1c_1 - b_1^2}{4a_1} = \frac{4a_0c_0 - b_0^2}{4a_0} + m \times \lg\left(\frac{P_1}{P_0}\right)$$
(10)

As the amplitude of cyclic loading increased from P_0 to P_1 , the parabolic fatigue curve is translated to the upper right of the coordinate space. The transformation of the fatigue curve due to the change in load amplitude is a result of the increased ΔK at corresponding points, which leads to higher da/dN. In addition, the parabola is stretched by a coefficient of more than one. Based on a number of experimental data obtained from welded joints, the coefficient of transformation of the fatigue curve can be defined by the arithmetic mean of the two load amplitudes; thus, the following formulation can be obtained

$$\mathbf{a}_1 = \lg\left(\frac{\mathbf{P}_0 + \mathbf{P}_1}{2}\right) \times \mathbf{a}_0 \tag{11}$$

In summary, the proposed model accounting for load-amplitude effects, based on the form of da/dN at a given load amplitude P_0 , can be described by equations (6), (10), and (11).

Validation of the theoretical models

Fatigue model considering heterogeneity

To emphasize influence of the fatigue rate model on curve fitting, the fittings of $\log(da/dN)$ – $\log(\Delta K)$ to the proposed model for the HR of three welded joints and to the Paris model have been compared with the experimental data of fatigue tests from literatures^{16,17,20} at a given load amplitude (Figures 3–5).

Obviously, the commonly used Paris law fails to predict the da/dN in the HR of the welded joints indicating the linear relationship is not competent enough to predict fatigue rate from heterogeneous materials (e.g.



Figure 3. Comparison of fatigue rate model fitting and experimental data for API 5L X65 weldment after hydrogen-charging in Tsay et al.¹⁶ (a) Comparison of Paris model fitting and experimental data and (b) comparison of proposed model fitting and experimental data.



Figure 4. Comparison of fatigue rate model fitting and experimental data for EH36 TMCP steel weldment in Tsay et al.¹⁶ (a) Comparison of Paris model fitting and experimental data and (b) comparison of proposed model fitting and experimental data.



Figure 5. Comparison of fatigue rate model fitting and experimental data for weld-repaired Bisplate80 with a soft buffer layer in Zhang et al.²⁰ (a) Comparison of Paris model fitting and experimental data (PM: parent metal) and (b) comparison of proposed model fitting and experimental data.

welded joints). That is because welded-joint fatigue issues belong to structural fatigue; thus, the fatigue crack growth behavior going through HR will be different because of the change in fatigue resistance resulting from the welding process. Comparison of Figures 3–5 illustrates how each model affects the fitting results of da/dN in the HR of welded joints. For each of the three welded joints, the fitting fatigue curve from the proposed model, which accounts for heterogeneity of fatigue properties, are better consonant with the open published experimental data from fatigue tests in comparison to the Paris model. As the log(da/dN)–log(Δ K) curve is a typical concave-down parabola, da/dN increases to a local $(\log(da/dN))_{max}$ at a point termed the weakest point of fatigue performance, then decreases gradually to a local $(\log(da/dN))_{min}$, and finally returns to its steady-state rate as the crack advances through the HR.

Fatigue model considering heterogeneity and amplitude effects

To emphasize the effects of both heterogeneity and load-amplitude variation on da/dN prediction, fatigue tests were carried out for identically samples under the same fatigue loading conditions but at different load

Figure 6. Schematic representation showing welded-joint specimen: (a) welding process, (b) weld-block after removing weld reinforcement, (c) E-CT sample of welded HSLA, and (d) schematic of E-CT specimen.

amplitude, 25 and 30 kN in this section. In present section, fatigue data at a given load amplitude ($P = P_0 = 25 \text{ kN}$) are used to determine the values of parameters "a₀,""b₀," and "c₀" in equation (1) by curve fitting. The values of these parameters are subsequently used to calculate new values "a₁,""b₁," and "c₁" to predict da/dN at a new load amplitude ($P = P_1 = 30 \text{ kN}$).

Specimen and fatigue test. The parent metal (PM) employed in this study was Bisplate80. Flux cored arc welding was used as the joining process for these components while CO_2 was used as the shielding gas. Extended compact tension (E-CT) specimens with through-the-thickness notches for fatigue were machined according to the specifications of ASTM E647 (23) (10 mm thick). The welded blocks were then sliced and machined into the required dimensions as shown in Figure 6. The WM region was centered in the gauge length of the tensile specimens, and the crack length was measured from the loading line.

The fatigue crack growth tests were performed at room temperature using hydraulic fatigue testing machine Instron 8501 with a load capacity of 100 kN.



Figure 7. Comparison of Paris model prediction and experimental da/dN of weld-repaired Bisplate80 at a load amplitude of P = 30 kN.

Constant amplitude tensile loads with a haversine waveform at a frequency of 5 Hz were used with the R of 0 throughout the tests. The Paris fatigue curves for weldrepaired Bisplate80 were measured twice with two identical specimens under the same loading condition but at different load amplitude, 25 and 30 kN, respectively.

Fatigue model prediction and experimental data. The predicted log(da/dN)–log(ΔK) curve obtained from proposed model prediction (accounting for both heterogeneity and fatigue-curve translation effects) for the HR of weld-repaired Bisplate80 (Figure 7) and that obtained from the Paris model prediction (Figure 8) have been compared with the experimental data of fatigue test at a load amplitude of P = 30 kN (named as P₁). Comparison between Figures 7 and 8 illustrates how fatigue model affects the predicting results of da/dN in the HR of a welded joint as the predicted results from the proposed model agree better with the experimental data in comparison to the Paris model.

The da/dN was calculated for the weld-repaired Bisplate80 under constant amplitude fatigue at a load amplitude $P_1 = 30 \text{ kN}$ according to the Paris formula, and is

$$\frac{da}{dN} = 2.231 \times 10^{-12} \times (\Delta K)^{3.188}$$
(12)

Based on the da/dN formulations for the weldrepaired Bisplate80 under constant amplitude fatigue at a given load amplitude $P_0 = 25 \text{ kN}$, the values of the



Figure 8. Comparison of proposed model prediction and experimental da/dN of weld-repaired Bisplate80 at a load amplitude of P = 30 kN.

corresponding new parameters " a_1 ,"" b_1 ," and " c_1 " for the proposed model were calculated from the expressions in equations (6), (10), and (11). Thus, under constant amplitude fatigue at P = 30 kN, da/dN formulation for the HR of the weld-repaired Bisplate80 was obtained and is

$$\frac{\mathrm{da}}{\mathrm{dN}} = 10^{-92.5755} \times (\Delta K)^{88.519 - 22.4787 \, \mathrm{lg}(\Delta K)}$$
(13)

For the weld-repaired Bisplate80, the correlation coefficient ($\gamma_{Pro} = 0.9316$) from the proposed model prediction, which account for both heterogeneity and amplitude effects, were higher than the reference values ($\gamma_{Paris} = 0.7470$) obtained from the Paris model prediction, and the corresponding mean squared errors were lower ($\sigma_{Pro} = 2.8964 \times 10^{-13} < \sigma_{Paris} = 1.0295 \times 10^{-12}$); these differences emphasize the influence of the fatigue rate model on da/dN prediction.

Discussion

Experimental results reported in a number of references indicate that the $\log(da/dN)$ - $\log(\Delta K)$ curve could be modeled as a concave-down parabola for the HR in a welded joint. Due to the heterogeneity resulted from welding process, the value of $\log(da/dN)$ changes nonlinearly with $\log(\Delta K)$ in welded joints, which is strongly dependent on the configuration of the joint. For a given welded joint, the fatigue crack growth curve translation is closely related to the load amplitude owing to the nonlinear relation between $\log(da/dN)$ and $\log(\Delta K)$, and fatigue test results at a known load amplitude are needed to calculate the parameters in equation (1) for any other load amplitude. Basing on fatigue data at known load amplitude, this study was to propose a fatigue rate model to predict the da/dN of the HR in a welded joint at any new load amplitude. Compared to the Paris fatigue model, the proposed model is superior in three respects: (1) it is capable of predicting the relationship of the da/dN to the ΔK for the H-region of welded joints, (2) it accounts for both joint heterogeneity and load-amplitude effects, and (3) it did not contain any material parameters that required prior experimental or numerical determination.

For the weld-repaired Bisplate80 in this study, the value of the mean square errors σ associated with the proposed model is one order of magnitude smaller than the corresponding value for the Paris model. The results also show that the correlation coefficient γ , indicating the agreement between experimental and predicted results, for the proposed model is above 0.93, which is much higher than the corresponding value of approximately 0.74 for the Paris model. Hence, it can be concluded that the accuracy of the proposed model is better than the widely used Paris model.

Conclusion

To emphasize the influence of the fatigue rate model on da/dN prediction accuracy, a novel fatigue crack growth model for HR of welded joints was proposed in this study and validated in details using experimental data of constant amplitude fatigue tests (open published and carried out in the study). From this study, the following conclusions can be obtained:

- 1. Predicted results from the proposed model agreed better with the experimental data in comparison to the widely used Paris model.
- 2. Fatigue data at a given load amplitude are needed to determine the values of parameters related to the configuration and dimensions of a welded joint and the weakest point of fatigue performance in the welded joint by curve fitting and then to calculate new values of parameters to predict da/dN at any new load amplitude.
- 3. A fairly good agreement existed between the predicted $da/dN-\Delta K$ relationship and the corresponding experimental data. Satisfactory agreement was obtained for HR of a welded joint.

It should also be mentioned that the influences of crack closure, residual stresses, and the load history have not been considered in this study. As the abovementioned factors are structure/specimen geometry dependent, this complicated issue is beyond the scope of this article and can only be dealt with in the future in a separate paper.

Author's Note

Authors Chunguo Zhang and Jizhuang Hui contributed equally to this work.

Acknowledgement

The authors would like to thank the University of Western Australia for the weld sample preparation and part of fatigue tests.

Declaration of conflicting interests

The author(s) declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article.

Funding

The author(s) disclosed receipt of the following financial support for the research, authorship, and/or publication of this article: This work was supported by the National Nature Science Foundation of China (No. 51405029), Shaanxi Natural Science Foundation (No. 2016JQ5020), and Chang'an University (Nos 310825153510 and 310825151038).

References

- Pandey K and Chand S. Fatigue crack growth model for constant amplitude loading. *Fatigue Fract Eng M* 2004; 27: 459–472.
- 2. Glinka G. A cumulative model of fatigue crack growth. *Int J Fatigue* 1982; 4: 59–67.
- Paris PC and Erdogan F. A critical analysis of crack propagation laws. J Basic Eng: T ASME 1963; 85: 528–535.
- Chen L, Cai LX and Yao D. A new method to predict fatigue crack growth rate of materials based on average cyclic plasticity strain damage accumulation. *Chinese J Aeronaut* 2013; 26: 130–135.
- Kujawski D and Ellyin F. A fatigue crack growth model with load ratio effects. *Eng Fract Mech* 1987; 28: 367–378.
- 6. Pandey K and Chand S. An energy based fatigue crack growth model. *Int J Fatigue* 2003; 25: 771–778.

- Li D, Nam W and Lee C. An improvement on prediction of fatigue crack growth from low cycle fatigue properties. *Eng Fract Mech* 1998; 60: 397–406.
- Rajabipour A and Melchers RE. Application of Paris' law for estimation of hydrogen-assisted fatigue crack growth. *Int J Fatigue* 2015; 80: 357–363.
- Shi KK, Cai LX and Bao C. Crack growth rate model under constant cyclic loading and effect of different singularity fields. *Proced Mater Sci* 2014; 3: 1566–1572.
- Forman RG, Kearney VE and Engle RM. Numerical analysis of crack propagation in cyclic-loaded structures. *J Basic Eng: T ASME* 1967; 89: 459–464.
- Jones R, Barter S, Molent L, et al. Crack growth at low ΔK's and the Frost-Dugdale law. *J Chin Inst Eng* 2004; 27: 871–877.
- 12. Barter SA, Molent L, Goldsmith N, et al. An experimental evaluation of fatigue crack growth. *Eng Fail Anal* 2005; 12: 99–128.
- Jones R, Barter S, Molent L, et al. Crack patching: an experimental evaluation of fatigue crack growth. *Compos Struct* 2004; 67: 226–238.
- Elber W. The significance of fatigue crack closure. In: Damage tolerance in air craft structure, paper no. ASTM STP 486, 1971, pp.230–242, http://www.astm.org/DIGITAL_ LIBRARY/STP/SOURCE_PAGES/STP486_foreword.pdf
- 15. Ukadgaonker VG, Bhat S, Jha M, et al. Fatigue crack growth towards the weld interface of alloy and maraging steels. *Int J Fatigue* 2008; 30: 689–705.
- Tsay LW, Chen YC and Chan SLI. Sulfide stress corrosion cracking and fatigue crack growth of welded TMCP API 5L X65 pipe-line steel. *Int J Fatigue* 2001; 23: 103–113.
- Tsay LW, Chern TS, Gau CY, et al. Microstructures and fatigue crack growth of EH36 TMCP steel weldments. *Int J Fatigue* 1999; 21: 857–864.
- Zhang CG, Yang JZ, Hu XZ, et al. Microstructure characteristics and fatigue properties of welded HSLA with and without buffer layer. *Mat Sci Eng A: Struct* 2012; 546: 169–179.
- Namakian R, Shodja HM and Mashayekhi M. Fully enriched weight functions in mesh-free methods for the analysis of linear elastic fracture mechanics problems. *Eng Anal Bound Elem* 2014; 43: 1–18.
- Zhang CG, Van der Vyer S, Hu XZ, et al. Fatigue crack growth behavior in weld-repaired high-strength low-alloy steel. *Eng Fract Mech* 2011; 78: 1862–1875.