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## BRIEF COMMUNICATION OPEN

## Efficient first-principles prediction of solid stability: Towards chemical accuracy

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The question of material stability is of fundamental importance to any analysis of system properties in condensed matter physics and materials science. The ability to evaluate chemical stability, i.e., whether a stoichiometry will persist in some chemical environment, and structure selection, i.e. what crystal structure a stoichiometry will adopt, is critical to the prediction of materials synthesis, reactivity and properties. Here, we demonstrate that density functional theory, with the recently developed strongly constrained and appropriately normed (SCAN) functional, has advanced to a point where both facets of the stability problem can be reliably and efficiently predicted for main group compounds, while transition metal compounds are improved but remain a challenge. SCAN therefore offers a robust model for a significant portion of the periodic table, presenting an opportunity for the development of novel materials and the study of fine phase transformations even in largely unexplored systems with little to no experimental data.

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## INTRODUCTION

Reliably accurate first-principles stability calculations are critical to the studies of materials synthesis,<sup>1</sup> reactivity<sup>2,3</sup> and properties,<sup>4</sup> and essential for both the exploration of new chemical spaces and the study of difficult-to-observe phases. For known materials, the question of solids stability can be resolved experimentally through a variety of calorimetric techniques, which yield the enthalpy of formation ( $\Delta H_f$ ), defined as the energy released when a compound is formed out of the elemental constituents in their standard states. In the prediction of new materials however, the formation enthalpy must be calculated from first-principles, which is most commonly done using density functional theory (DFT)<sup>5–8</sup> with the well-established Perdew–Burke–Ernzerhof (PBE)<sup>9</sup> density functional. Furthermore, in the evaluation of solids differing in structure, but not chemistry, the differences between the formation enthalpies of competing phases often lie on a very fine energy scale, and are very difficult to measure experimentally, motivating their calculation from first-principles.<sup>10</sup>

Due to the diversity of chemical degrees of freedom, from subtle structural differences to changes in bonding characteristics, it is challenging for first-principles methods to tackle both facets of the stability problem at a reasonable computational cost. The chemical accuracy of  $\Delta H_f$ , typically quoted as 1 kcal/mol (0.04 eV/atom), can be achieved in experiments, usually limited by sample quality and measurement uncertainties. High-level wavefunction methods<sup>11</sup> (e.g., the configuration interaction or quantum Monte Carlo approaches) can achieve such accuracy, but they are limited to systems having a relatively small number of electrons per

periodic unit cell due to their high computational cost. DFT<sup>5–8</sup> with the PBE<sup>9</sup> generalized gradient approximation (GGA) to its exchange–correlation energy currently is the dominant calculation approach due to its relatively cheap computational cost and reasonable accuracy. Unfortunately, errors in formation enthalpy predicted by PBE are usually at the level of  $\sim 0.2$  eV/atom, which leads to significant errors in predicting phase stability among dissimilar chemistries. While error cancellation reduces the magnitude of enthalpy error when considering chemically similar phases, the remaining errors still result in difficulties in selecting the ground state structure among chemically similar phases (see Fig. 1). The three major sources of error in PBE are the self-interaction error, the incomplete error cancellation between the target compound and the elemental references, and the absence of van der Waals (vdW) interactions. Attempts have been made to improve upon PBE at the GGA level using non-empirical derivations, including the Wu–Cohen,<sup>12</sup> PBEsol<sup>13</sup> and AM05<sup>14</sup> GGAs. A GGA referred to as PBEfe has been specifically optimized for the formation enthalpies of solids.<sup>15</sup> However, none of the GGAs improve over PBE systematically and replace PBE as a general-purpose functional. For example, those GGAs such as PBEsol that are better than PBE for the lattice constants of solids are typically worse than PBE for the binding energies of molecules (relevant to the formation energy).

Moving beyond pure GGA functionals, the GGA+*U* approximation<sup>16,17</sup> is a common method which improves the representation of transition (with valence *d* electrons) metal elements at marginal extra computational cost. DFT+*U* is an empirical method for

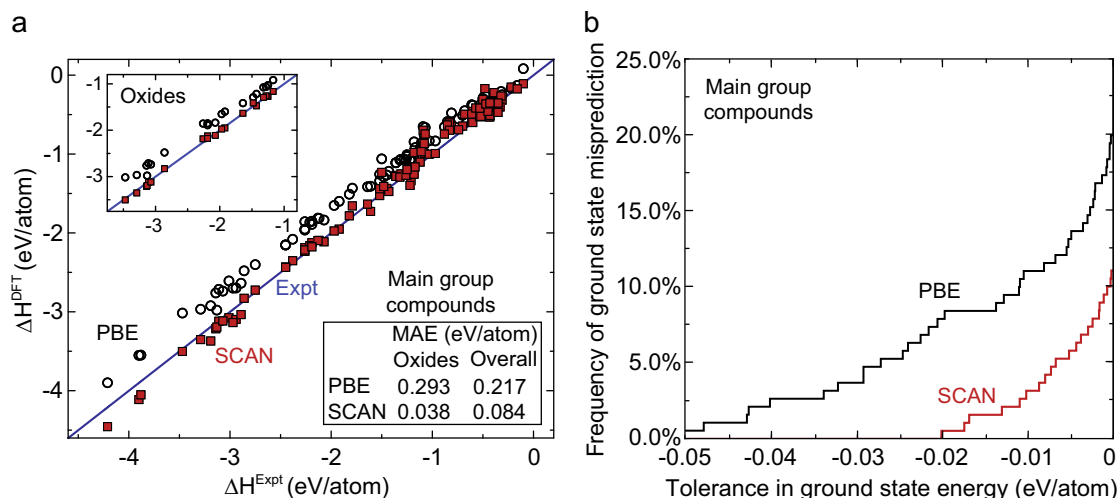
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**Fig. 1** SCAN provides a significant improvement over PBE in computing the absolute and relative stability of main group compounds, approaching the chemical accuracy of 0.04 eV/atom in formation enthalpy and providing reliable structure selection as an indicator of accurate relative stability. **a** The formation enthalpy of main group compounds, and oxides in particular. **b** The probability that the computed energy of a predicted structure with respect to that of the experimental ground state structure lies below a threshold energy or tolerance across main group compounds for a range of tolerances

modeling electron correlation effects in which one complements DFT with a model Hamiltonian by adding an on-site interaction given by the Hubbard  $U$  term. In the commonly used implementations,  $U$  is treated as an empirical scalar parameter to reduce the self-interaction error in the underlying density functional.<sup>18</sup> The DFT+ $U$  scheme reduces the self-interaction error by linearizing the electronic energy with respect to the occupation of a set of selected (for example,  $d$  or  $f$ ) orbitals and helps localize the associated electronic states, and improves energetic descriptions for many compounds involving transition metal elements.<sup>19</sup> However, the  $U$  parameter used in this approach is typically determined empirically and is system and property dependent, such that the simultaneous representation of, for example, chemical and structural stability would require different and non-generalizable  $U$  values.<sup>20,21</sup>

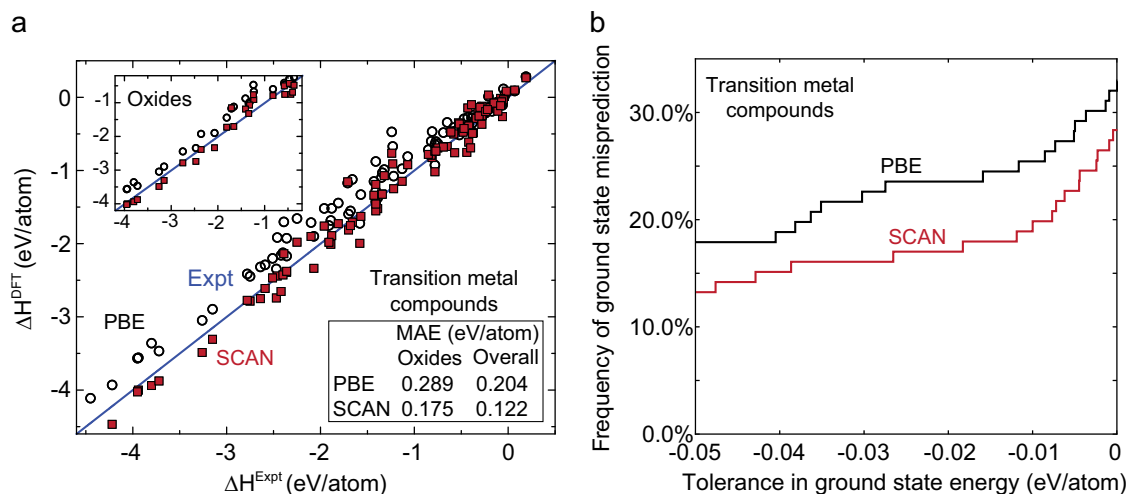
Similarly, an array of vdW corrections<sup>22–24</sup> have been proposed to PBE to improve the description of the vdW interactions. The vdW-corrected PBE is useful for systems where vdW interactions are important, which however faces the problem of double counting of the vdW interaction from both PBE (and in general the underlying density functional) and the vdW correction.<sup>25</sup> Furthermore, the vdW corrections are usually not helpful for dealing with the other two sources of error.

To deal with the imperfect error cancellation of PBE, a number of different approaches have been developed, such as the fitted elemental-phase reference energies (FERE) scheme,<sup>26,27</sup> and other schemes with fitted corrections to elemental and molecular reference states.<sup>28,29</sup> These corrections assume that most of the error in compound formation enthalpy depends only on the overall composition, and attempt to eliminate this error by using the total energies of elemental phases as fitting parameters. In the case of FERE, with about 30 fitting parameters, the PBE mean absolute error (MAE) is reduced from 0.250 eV/atom to 0.052 eV/atom for a set of 110 main-group binary solids that largely overlap our testing set to be discussed here. However, neither FERE nor the other composition-based schemes can provide the correction to PBE needed to predict the relative stability of different phases of a compound, which is critically important for structure selection, and by extension the prediction of properties that depend on local structural changes rather than simply the average composition. Furthermore, fitting schemes, based on common structures and thus common geometries and oxidation states, are

difficult to generalize outside of their initial fitting data, especially to situations where rare electronic configurations may give rise to unexpected errors not accounted for by the fitted correction.<sup>29</sup>

The corrections mentioned above were developed or motivated to solve one of the three major error sources of PBE. As a result, broad benchmarks across GGA-based functionals targeting solid formation enthalpies, crystal structure selection and various material properties continue to rely on PBE as a representative general-purpose GGA, offering a balanced performance in the representation of various characteristics of a solid.

Compared to GGAs, which are built using only the electron density and its gradient, meta-GGAs<sup>30–40</sup> add the electronic kinetic energy density as an additional ingredient. The kinetic energy density is semilocal in the occupied orbitals, immediately available from common DFT calculations, and thus only adds moderate extra computational cost. While a number of different meta-GGAs have been proposed based on both non-empirical derivations<sup>30,32,33,35–37</sup> and fitting schemes<sup>31,34,38–40</sup> the strongly constrained and appropriately normed (SCAN) meta-GGA<sup>30</sup> is unique in that it satisfies all known (17) exact constraints applicable to a meta-GGA. In contrast, the PBE GGA only satisfies 11 of the 17 exact constraints. By correctly building the kinetic energy density into a dimensionless orbital-overlap indicator, SCAN distinguishes between density regions characterizing different chemical bonds (including covalent, ionic, metallic, hydrogen and vdW), and treats them properly through appropriate GGA constructions, allowing SCAN to address diverse types of bonding in materials and systematically improve over PBE in general.<sup>41</sup> SCAN has been tested for different properties and systems with excellent performance in comparison with GGAs and other meta-GGAs, including molecules,<sup>42</sup> liquids,<sup>43,44</sup> surfaces<sup>45</sup> and solids.<sup>46–48</sup> For cohesive energies of solids tested in refs. 25 and 46 which might not be a good indicator for testing density functionals,<sup>49</sup> SCAN slightly improves over PBE,<sup>25,46</sup> while SCAN demonstrates significantly better accuracy than PBE and PBE-based density functionals for energetics and structures of a small set of binary oxides,<sup>47</sup> which is more relevant to experimental measurements. The unique and non-empirical derivation of SCAN suggests transferability in reproducing chemical and structural stability of compounds across the periodic table, as compared to PBE, motivating the present study.



**Fig. 2** SCAN considerably reduces the average error in formation enthalpy and structure selection error frequency relative to PBE in transition metal compounds that are more difficult to represent than the main group compounds due to an increased contribution of self-interaction error. **a** The enthalpy of formation of transition metal binary compounds, and oxides in particular. **b** The probability of incorrect structure selections by PBE or SCAN across transition metal compounds, for a range of tolerances

Here, we show that the SCAN<sup>30</sup> semilocal density functional halves the errors of PBE in predicting formation enthalpies of about 200 binary solids,<sup>15</sup> taking a significant step towards chemical accuracy while retaining a comparable efficiency to PBE. Remarkably, SCAN also yields a significant improvement in the reliability of crystal structure selection, consistently halving the error rate in ground state selection accuracy and reducing the error in the relative energies of crystal structures. While the computational cost of SCAN is modestly greater than (usually 2–3 times) that of PBE, it is much less (in general by an order of magnitude in plane-wave codes) than that of hybrid functionals, and very much less than that of wavefunction methods. We believe that this development is a very significant step for the ab initio prediction of novel compounds and their properties and one that will lend more credibility to such predictions when experimental efforts are needed to realize them.

## RESULTS AND DISCUSSION

To systematically compare the performance of SCAN and PBE, we group chemistries by how they are affected by known errors in semilocal density functionals. Self-interaction error, one of the three major error sources in PBE, is intrinsic to all computationally semilocal density functionals, among which are PBE and SCAN. Self-interaction error manifests itself in transition metal compounds, especially in semiconducting and insulating oxides, more than in main group compounds due to the presence of valence  $d$  electrons that localize more than valence  $sp$  electrons.<sup>50</sup> The late 3d elements Cr, Mn, Fe, Co, and Ni, are especially problematic. To resolve self-interaction error in a true first-principles spirit, nonlocal corrections are necessary, which are typically computationally expensive and scale poorly with system size. Therefore, we first address the behavior of main group compounds so as to characterize the performance of PBE and SCAN as efficient semilocal functionals largely in the absence of self-interaction error.

As shown in Fig. 1a, the MAE of SCAN in the formation enthalpy of 102 main group compounds is 0.084 eV/atom, about 2.5 times lower than that of PBE, while the MAE of SCAN for the 21 main group oxides (shown in the inset) is 0.038 eV/atom, impressively within the typically quoted chemical accuracy of 0.04 eV/atom. MAE is used here to characterize the accuracy of functionals, while further discussions about metrics for characterizing the accuracy can be found in refs. <sup>51</sup> and <sup>52</sup> The reduction in error afforded by

SCAN relative to PBE originates from the fact that PBE is not able to simultaneously and accurately treat the different types of chemical bonds<sup>30,41</sup> found in a compound and its constituent elemental phases, leading to the well-known imperfect error cancellation, e.g., between the molecular O<sub>2</sub> reference and metal oxides.<sup>28</sup> SCAN, on the other hand, is able to capture the behavior of all such interactions<sup>41</sup> by introducing the Kohn–Sham kinetic energy density into the functional in a way that satisfies all known limiting behaviors and constraints on electronic interaction appropriate to semilocal functionals.<sup>30</sup> This construction leads to a widely predictive functional.<sup>53</sup> In particular, without being fitted to any bonded system, SCAN captures even the intermediate-range vdW attraction between neighboring atoms in a solid, which PBE largely neglects. Furthermore, PBE underestimates the chemical stabilities of most solids, for example erroneously making InN chemically unstable, an error which SCAN avoids. These errors arise largely from PBE over-stabilization of reference molecules, and could not be fixed by simply adding a vdW correction to PBE.

An even stronger indication of the general reliability of SCAN for stability calculations is its superior performance in identifying ground state crystal structures. Here the intermediate-range vdW interaction, present in SCAN but not in PBE, can play a particularly important role, for example, by stabilizing the correct CsCl structure in the heavy halides CsCl, CsBr and CsI.<sup>54</sup> Based on 190 stoichiometric main group binary compounds with 1627 experimentally reported and predicted crystal structures, we identify the most stable low-temperature, low-pressure phases within PBE and SCAN. We then evaluate the frequency with which PBE or SCAN stabilizes an incorrect ground state structural polymorph in comparison to the experimental structure. The zero-temperature ground state crystal structure is thermodynamically defined as the phase with the lowest enthalpy, as, under these conditions, enthalpy is exactly equal to the system Gibbs free energy. Thus, the relative energies of competing crystal structures computed from DFT provide a nearly complete representation of their relative Gibbs free energies under these conditions. The missing terms in the Gibbs free energy difference are the change in zero-point vibrational energy, which is in the scale of 0.005 eV/atom,<sup>55–58</sup> ambient pressure effects, which lie in the scale of 0.001 eV/atom, and other contributions of smaller magnitudes. To account for these effects, as well as other potential noise in the calculations, we introduce a tolerance on structure selection  $\Delta E_{\text{tol}}$ , and count the frequency with which PBE or SCAN

**Table 1.** The PBE and SCAN errors in the structure selection and prediction of the formation enthalpy

|                                   | Error in structure selection (%) |    |                           |    |              |    | Error in formation enthalpy (eV/atom) |                           |              |
|-----------------------------------|----------------------------------|----|---------------------------|----|--------------|----|---------------------------------------|---------------------------|--------------|
|                                   | Main group binaries              |    | Transition metal binaries |    | All binaries |    | Main group binaries                   | Transition metal binaries | All binaries |
| $\Delta E_{\text{tol}}$ (eV/atom) | 0.01                             | 0  | 0.01                      | 0  | 0.01         | 0  |                                       |                           |              |
| PBE                               | 12                               | 21 | 26                        | 33 | 17           | 25 | 0.22                                  | 0.20                      | 0.21         |
| SCAN                              | 3                                | 11 | 19                        | 28 | 9            | 17 | 0.08                                  | 0.12                      | 0.10         |

SCAN systematically improves over PBE for predicting the chemical stability of solids and relative stability of different solid phases, halving the average error in formation enthalpy and the frequency of predicting the wrong ground state structure of binary solids

erroneously stabilize a crystal structure by more than  $\Delta E_{\text{tol}}$  with respect to the experimental ground state structure. As can be seen in Fig. 1b, SCAN provides a significant improvement over PBE in selecting the correct ground state structure, reducing the frequency of structure prediction error from 12% to just 3% at a 0.01 eV/atom tolerance, where the improvement in structure selection accuracy likely originates from the more accurate physical model provided by SCAN relative to PBE. Notably, the energy scale of competing crystal structures<sup>59–61</sup> and the associated tolerances are far below the average error suggested by total formation enthalpy statistics discussed earlier, which confirms the consensus that density functionals work better in predicting energy difference between chemically similar systems than that of dissimilar systems. Thus, it is evident that reliable structure selection is a sensitive indicator of how well a functional captures fine details in the relative stability of chemically similar phases.

Our structure selection results highlight the difficulty of reliable structure prediction by first-principles calculations. While we find that PBE yields close to a 21% error rate in structure selection in absolute terms and a 12% error rate with a 0.01 eV/atom tolerance, this error rate is likely a lower bound as no method that we are aware of can guarantee that no other crystal structures exist with a lower energy for any given chemistry. Conversely, experimental uncertainties in the ground state crystal structure originating from difficult-to-observe low-temperature phase transitions, small off-stoichiometries and other errors mean that even the exact functional would likely not be able to achieve complete agreement with experiment. In this light, the 3% error rate of SCAN within the 0.01 eV/atom tolerance is remarkable. The impact of this improvement is immediately visible—for example, in  $\text{SiO}_2$ , SCAN is able to reproduce the correct  $\alpha$ -quartz low-temperature, low-pressure ground state structure and, correspondingly, the pressure–temperature phase diagram, as PBE over-stabilizes the high-temperature  $\beta$ -cristobalite polymorph. The intermediate-range vdW interaction in SCAN stabilizes the correct, higher-density quartz phase of this earth-abundant material. Taken together with the promising performance in predicting formation enthalpy, these results suggest that SCAN is highly reliable for both the absolute and relative stability of the main group compounds.

We now turn to transition metal compounds, where self-interaction error poses a fundamental limitation on the performance of semilocal density functionals. For the formation enthalpies of 98 transition metal binaries (where we consider compounds consisting of a transition metal and an electronegative anion, excluding intermetallic compounds), Fig. 2a shows that SCAN still has an MAE of 0.122 eV/atom, which is significantly larger than that of the main group compounds. However, SCAN improves over PBE by about 0.08 eV/atom, or 40% of the total PBE error, for the transition metal compounds. Similarly, as can be seen in Fig. 2b, based on 106 transition metal binary compounds with 1336 experimental and predicted structures, SCAN also gives a

significant improvement relative to PBE in structure selection accuracy, although the absolute performance of both functionals is much worse than that in the main group compounds, both in the frequency of structure selection errors and their energetic magnitude. The fact that this discrepancy persists even at a 0.02 eV/atom tolerance, where the SCAN error rate in main group chemistries approaches zero, suggests that further improvements in functional performance would require a fundamentally different approach, moving to non-local functional forms or explicit self-interaction corrections.<sup>62–65</sup>

The development of semilocal exchange–correlation functionals in DFT has been driven by the promise of these approximations to efficiently evaluate the stability and properties of both known and predicted solid phases. As summarized in Table 1, the SCAN functional, without any fitted corrections, now approaches experimental accuracy in both total energy and the relative stability of solid phases across main group compounds, and the remaining challenges to DFT functionals appear to be self-interaction error dominated systems. Correspondingly, future improvement in general-purpose functional performance will require a solution to the self-interaction problem and a representation of non-local phenomena.

## METHODS

The calculations of enthalpy of formation are performed for 196 binary compounds with 101 main group systems (the chemistries are listed in Table S1 in Supplementary Information) and 95 systems containing transition metals (see Table S2). The structures and reference formation energies for these 196 compounds are based on the dataset reported by Sarmiento-Pérez et al.,<sup>15</sup> neglecting any potentially lower energy structures predicted by PBE or SCAN. The analysis of structure selection accuracy is based on 296 binary chemistries, of which 190 are main group compositions and 106 contain transition metals (see Table S4). In the choice of chemistries, we choose only compositions for which the low-temperature, low-pressure ground state crystal structure can be identified from experimental data to the best of our ability. The chemistries included in this sample are comprised of compositions previously chosen to benchmark formation enthalpy,<sup>15</sup> AB-type compounds and a selection of binary compositions previously enumerated in crystal structure prediction studies.<sup>66</sup> To the best of our knowledge, this selection of chemistries does not introduce any bias in the likelihood of structure selection error not present more generally in binary main group and transition metal compounds. For each chemistry, we consider experimentally reported crystal structures from the Inorganic Crystal Structure Database,<sup>67</sup> as well as likely structures predicted by data mined elemental substitution methods,<sup>66</sup> giving a total of 2963 crystal structures. All structures considered are available in POSCAR format in the supplementary materials, relaxed with PBE or SCAN. Finally, to determine whether or not a DFT-relaxed crystal structure matches the experimentally reported structure, we rely on a distortion-tolerant affine map, implemented as the Structure-Matcher algorithm in the Pymatgen package.<sup>68</sup>

All calculations are performed using the Vienna Ab Initio Simulation Package (VASP)<sup>69,70</sup> using the projector augmented wave (PAW) method<sup>71,72</sup> with a reciprocal space discretization of 25  $K$ -points per  $\text{\AA}^{-3}$  and a plane wave energy cutoff of 520 eV. Both the semilocal GGA in the



standard form of the PBE<sup>9,73</sup> and the SCAN<sup>30,41</sup> meta-GGA<sup>74</sup> exchange–correlation functionals are used (with VASP PAW potential version “PAW 52”). In magnetically active systems, the energy is taken as the lowest of a ferromagnetic and a sample of small-unit-cell antiferromagnetic orderings. For the structure selection, the calculations are converged to  $10^{-6}$  eV in total energy and 0.01 eV/Å on atomic forces. For computing formation enthalpies, all calculations are converged to  $10^{-7}$  eV on total energy and 0.01 eV/Å on atomic forces. Molecular reference states are used for H<sub>2</sub>, N<sub>2</sub>, O<sub>2</sub>, F<sub>2</sub> and Cl<sub>2</sub>, where the isolated molecule is represented by a dimer in a  $15 \times 15 \times 15 \text{ \AA}^3$  box. Experimental standard enthalpies of formation used to determine the error in formation energy are defined at 298 K and 1 atm of pressure.

#### Data availability

All structures considered in the structure selection dataset, and experimental reference structures, as well as their relative energies computed in PBE and SCAN, are available in POSCAR format in the supplementary information. All structures considered in the enthalpy of formation dataset are available in previous datasets referred to in the main text.

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#### AUTHOR CONTRIBUTIONS

J.S., H.P., J.P.P. and G.C. designed the project. Y.Z., D.A.K. and J.S. performed the calculations. Y.Z., J.S., R.A.S.-P. and M.A.L.M. constructed the formation enthalpy dataset. D.A.K., J.Y., T.C. and S.T.D. constructed the structure selection dataset. J.S., D.A.K. and Y.Z. wrote the paper. J.P.P. and G.C. edited the manuscript. All authors have discussed the results.

#### ADDITIONAL INFORMATION

**Supplementary information** accompanies the paper on the *npj Computational Materials* website (<https://doi.org/10.1038/s41524-018-0065-z>).

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