

# Electrochemically Generated Peryleniumyl-Hexafluorophosphate and Hexafluoroarsenate: New One-Dimensional Metals

H. J. KELLER, D. NÖTKE, H. PRITZKOW, D. WEHE, and M. WERNER

*Anorganisch-Chemisches Institut der Universität,  
Im Neuenheimer Feld 270, 6900 Heidelberg, West Germany.*

and

P. KOCH and D. SCHWEITZER

*Max-Planck-Institut für Medizinische Forschung,  
Abteilung für Molekulare Physik, Jahnstr. 29, 6900 Heidelberg, West Germany.*

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Compounds of stoichiometry  $(pe)_2(PF_6)_{1.1} \times 0.8 CH_2Cl_2$  (**1**) ( $pe =$  perylene),  $(pe)_2(AsF_6)_{1.1} \times 0.7 CH_2Cl_2$  (**2**),  $(pe)_2(PF_6)_{1.4} \times 0.6 THF$  (**3**),  $(pe)_2(AsF_6)_{1.5} \times 0.5 THF$  (**4**) and  $(pe)_3(SbF_6)_2 \times 0.75 CH_2Cl_2$  (**5**) have been obtained as crystalline samples by electrochemical deposition from  $CH_2Cl_2$  [(**1**), (**2**) and (**5**)] or from THF [(**3**) and (**4**)] solutions of perylene, containing the appropriate counterion. The three compounds (**1**)–(**3**) crystallize in isomorphous orthorhombic lattices. (**1**) forms black needles: space group Pnmm with  $a = 4.285 \text{ \AA}$ ,  $b = 12.915 \text{ \AA}$  and  $c = 14.033 \text{ \AA}$ ,  $z = 1$ . (**2**) gives black needles, orthorhombic space group Pnmm with  $a = 4.294 \text{ \AA}$ ,  $b = 13.077 \text{ \AA}$ , and  $c = 14.132 \text{ \AA}$ ,  $z = 1$ . The structures of (**1**) and (**2**) were solved by direct methods and refined by least squares to final  $R = 0.148$  and  $R = 0.088$  based on 476 and 322 observed reflections. The perylene forms segregated stacks in direction of the  $a$ -axis with interplanar distances of  $3.40 \text{ \AA}$  and an angle of  $37.7^\circ$  between the  $bc$ -plane and the perylene. The channels between the segregated stacks are filled by anions and solvent molecules. The d.c. conductivities (four probe measurements) of (**1**)–(**4**) fall in the range of  $70\text{--}1200 \Omega^{-1} \cdot \text{cm}^{-1}$  at room temperature. The conductivities show a metallic regime down to about  $200^\circ\text{K}$  and drop off below that temperature.

## 1 INTRODUCTION

Electrochemically induced redox reactions in organic solvents are a common tool to generate and investigate open shell organic ions.<sup>1-3</sup> In several cases this procedure led to the isolation of solid products containing radical ions with remarkable solid state properties.<sup>4-8</sup> The recent interest in crystallized organic radical ions is connected mainly with their high electrical conductivities.<sup>9-11</sup> Some of the very promising materials have recently been obtained electrochemically.<sup>8</sup> The electrochemical method is especially well suited in the search for highly conducting (metallic) radical salts. This is because electrochemical deposition of solid organic radical salts from solutions "inevitably" selects in favor of the highly conducting crystals. From all thermodynamically favorable compounds which could crystallize from solution the metallic ones grow fastest. After nucleation at the electrode surface those crystals with the lowest resistivity will enable the most intense electron flow. Only those compounds which allow electron transport for the redox reaction over a distance of several millimeters and in a high density flux can be deposited as reasonable crystals. The advantages of electrocrystallization have been demonstrated earlier by different groups growing single crystals of one-dimensional metals of the mixed valence tetracyanoplatinate (II, IV) type electrochemically.<sup>12,13</sup>

Electrochemical oxidation reactions on several aromatic hydrocarbons have been carried out in this laboratory, following the above mentioned lines. By varying experimental conditions, using different hydrocarbons (like coronene, anthracene, pyrene, triphenylene, etc.) and supporting electrolytes like  $(C_4H_9)_4N^+PF_6^-$ ,  $(C_4H_9)_4N^+AsF_6^-$ , etc., it was found as a preliminary result that *perylene* is especially well suited to form highly conducting solid radical species. Structural and physical investigations on some of these solids obtained by electrocrystallization procedures are presented here.

## 2 EXPERIMENTAL

### 2.1 Preparation of compounds

#### 2.1.1 Apparatus for electrochemical syntheses

The electrolyses were carried out in a 100 ml U-tube with a glass filter or in a 75 ml electrochemical cell of 3 compartments separated by 2 glass filters from each other. In some cases the crystal size of the reaction products depend on the used electrochemical cell. Platinum wire was used as electrode material. Ni, Ag and Cu can be used as cathode material, too. The potential was selected to give a current of 20-50  $\mu A$ . Depending on the cell, solvent

and concentration of the supporting electrolyte, potentials between 2.5 and 6 V have been used. This results in a current density of 2–5  $\mu\text{A}/\text{cm}^2$ . This current density is conclusive for the velocity of the crystal growth.

### 2.1.2 Chemicals

Methylenechloride ( $\text{CH}_2\text{Cl}_2$ ) was dried over  $\text{CaCl}_2$  and a molecular sieve 4 Å and saturated with nitrogen. Tetrahydrofurane (THF) 99.9% (Gold-Marke) saturated with nitrogen was purchased from EGA Chemie. Tetrabutylammoniumhexafluorophosphate ( $\text{Bu}_4\text{PF}_6$ ) was obtained by mixing equimolar amounts of  $\text{Bu}_4\text{N}^+\text{J}^-$  and  $\text{KPF}_6$  in aqueous solution.<sup>14</sup>  $\text{Bu}_4\text{NASF}_6$  was obtained in a similar fashion by using potassium hexafluoroarsenate instead of  $\text{KPF}_6$ .  $\text{Bu}_4\text{NSbF}_6$  was prepared in a similar manner as  $(\text{CH}_3)_4\text{NSbF}_6$ .<sup>15</sup> To a solution of 13.15 g Potassiumhexahydroxoantimonate (V),  $\text{K}[\text{Sb}(\text{OH})_6]$  in 100 ml 40% HF a 40% aqueous solution of Tetrabutylammoniumhydroxide was added. The mixture was evaporated to half of its original volume, the precipitate was filtered and washed with a few milliliters of water. Recrystallization from isopropanol gave white needles, Fp.: 220°C, Yield: 11 g  $\cong$  46% of the theoretical yield.

Analyses:  $(\text{C}_4\text{H}_9)_4\text{NSbF}_6$  Molemass: 478.22

Calc.: C 40.19% H 7.59% N 2.93%

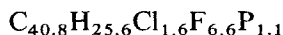
Found: C 40.19% H 7.79% N 2.80%

All these supporting electrolytes were dried under high vacuum. Perylene 99+ % Gold label was purchased from Aldrich Chemical Company and used without further purification.

### 2.1.3 Preparation of *pe*- $\text{PF}_6$ compounds

2.1.3.1  $(\text{pe})_2(\text{PF}_6)_{1.1} \times 0.8 (\text{CH}_2\text{Cl}_2)$ . 0.1 g perylene and 0.5 g  $\text{Bu}_4\text{NPF}_6$  were dissolved in 100 ml absolute  $\text{CH}_2\text{Cl}_2$ , filtrated into the electrochemical cell and electrolyzed by applying about 3 V and 30  $\mu\text{A}$  for about 2 days under exclusion of light. Crystal growth starts after a few minutes and after two days black needles (which appear green by looking through the crystals), of several cm in length, were obtained. A typical experiment is shown in Figure 1. The crystals decompose above 180°C, giving up neutral perylene.

Analyses:  $(\text{pe})_2(\text{PF}_6)_{1.1} \times 0.8 \text{CH}_2\text{Cl}_2$ :



Molemass: 732.06

Calc.: C 66.94 H 3.52 Cl 7.75 F 17.13 P 4.65

Found: C 66.71 H 3.66 Cl 8.05 F 17.00 P 4.50

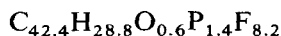
C 66.57 H 3.72 P 4.53

Closer inspection of the prepared samples showed, that principally two different types of crystals grow simultaneously at the surface of the anode: *thinner* ones with a rectangular or trapezoid cross-section ("type one") while the *thicker* ones have rectangular, quadratic or preferably hexagonal cross-sections ("type two"), (for typical dimensions see Section 2.3. "Conductivity measurements"). The latter crystals are the main product during electrocrystallization.

Analytical data suggest that type one crystals contain much less  $\text{CH}_2\text{Cl}_2$  but more  $\text{PF}_6^-$  ions than the main product. This results in a higher formal oxidation number for the peryleniumyl stacks in "type one" crystals compared to the "type two" modification. The analytical data of the mixture which are given above are very close to the values of the main product (type two).

2.1.3.2  $(pe)_2(\text{PF}_6)_{1.4} \times 0.6 (\text{THF})$ . By using THF instead of  $\text{CH}_2\text{Cl}_2$  as solvent the same condition with a current of 20  $\mu\text{A}$  leads after two days to long but very *thin* black needles. No "type two" crystals have been observed. The crystals decompose above 170°C. Above this temperature perylene crystals grow at the surface of the black needles.

Analyses:  $(pe)_2(\text{PF}_6)_{1.4} \times 0.6 \text{ THF}$



Molemass: 751.46

Calc.: C 67.77 H 3.86 F 21.24 O 1.28 P 5.77

Found: C 67.48 H 3.96 F 20.62 O - P 5.77

#### 2.1.4 Preparation of *pe-AsF<sub>6</sub>* compounds

2.1.4.1  $(pe)_2(\text{AsF}_6)_{1.1} \times 0.7 (\text{CH}_2\text{Cl}_2)$ . The electrocrystallization was carried out with the same apparatus and under the same conditions just using  $\text{Bu}_4\text{NAsF}_6$  instead of  $\text{Bu}_4\text{NPF}_6$ . Long black needles are obtained (green in absorption) which decompose above 260°C giving up neutral perylene.

Analyses:  $(pe)_2(\text{AsF}_6)_{1.1} \times 0.7 \text{ CH}_2\text{Cl}_2$



Molemass: 771.90

Calc. C 63.34 H 3.32 As 10.62 Cl 6.42 F 16.25

Found: C 63.30 H 3.44 As 10.8 Cl 6.20 F 17.0

Again two different types of crystals (thinner and thicker ones) have been obtained simultaneously at the anode. The results are very similar to those described for the  $\text{PF}_6^-$ -derivatives (see Sections 2.1.3.1. and 2.3).

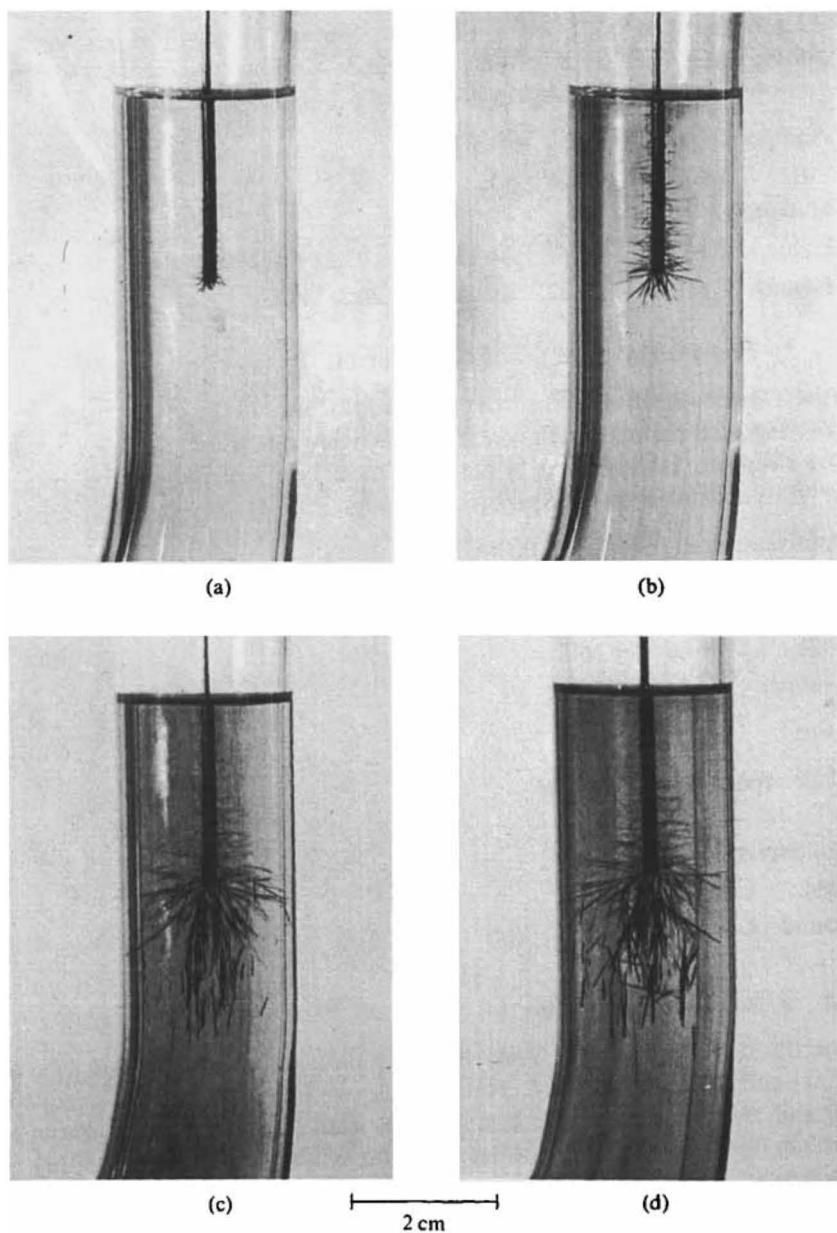
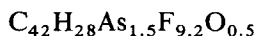


FIGURE 1 Electrocrystallization of (*I*) at a platinum anode: (a) after 30 min. (b) after 2 h, (c) after 12 h, (d) after 20 h. (Only the anode half cell is shown).

2.1.4.2  $(pe)_2(AsF_6)_{1.5} \times 0.5$  (THF). Similar procedures as described above, using THF as a solvent, give black long but *thin* needles which decompose above 260°C. Again no second phase has been found.

Analyses:  $(pe)_2(AsF_6)_{1.5} \times 0.5$  THF



Molemass: 824.08

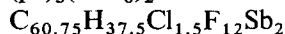
Calc.: C 61.22 H 3.42 As 13.67 F 21.21 O 0.97

Found: C 60.92 H 3.32 As 14.1 F 20.38 O -

### 2.1.5 Preparation of $(pe)_3(SbF)_2$ compounds

Electrocrystallization from  $CH_2Cl_2$  and from THF gives small black platelets and prisms which cover the electrode but which do not grow into the solution. Only C and H analyses have been carried out because the crystals are not usable for physical investigations.

Analyses:  $(pe)_3(SbF_6)_2 \times 0.75 CH_2Cl_2$



Molemass: 1292.16

Calc.: C 56.67 H 2.93

Found: C 56.74 H 3.14

from

THF:  $(pe)_3(SbF_6)_2 \times THF$



Molemass: 1300.57

Calc.: C 59.10 H 3.41

Found: C 58.93 H 3.62

## 2.2 X-Ray investigations

For the collection of the X-ray intensity data only the type two crystals of (1) and (2) could be used. Nevertheless we investigated *thin* crystals of (3) and showed them to be isomorphous with the type two phases of (1) and (2) (see Table I). There seems to be no doubt that all phases of (1)-(4) are isostructural.

Lattice parameters were determined from Weissenberg films and refined with 22 for (1) respectively 19 for (2) reflections measured on a Syntex R3 diffractometer. Intensity data were collected on the diffractometer (MoK $\alpha$  radiation,  $\lambda = 0.7107 \text{ \AA}$ , graphite monochromator). Reflections were measured by  $\theta/2\theta$  scan with a scan range of 1.9°. Crystal data and details of

TABLE I

Compound	(1)	(2)	(3)
Crystal System	orthorhombic	orthorhombic	orthorhombic
Space group	Pnmm	Pnmm	Pnmm
Cell parameters a (Å)	4.285	4.294	4.26 <sup>a</sup>
b (Å)	12.915	13.077	12.79 <sup>a</sup>
c (Å)	14.033	14.132	13.89 <sup>a</sup>
Cell volume (Å <sup>3</sup> )	776.56	793.55	756.80 <sup>a</sup>
z	1	1	1
Density calc. (g/cm <sup>3</sup> )	1.57	1.65	1.61
Linear absorption factor (cm <sup>-1</sup> )	13.0	12.1	—
Size of crystal (mm)	0.5 × 0.2 × 0.1	0.3 × 0.2 × 0.1	0.3 × 0.2 × 0.05
2θ max.	67.6	69.6	—
Number of measured refl.	781	1226	—
Unobserved refl.	305	904	—

<sup>a</sup> Cell parameters from Weissenberg films were not refined by diffractometer data.

measurements are given in Table I). LP and absorption corrections were applied to the data. Reflections with  $I < 3 \sigma(I)$  were considered unobserved and not included in the refinement. The calculations were carried out on an IBM 370/168-computer (Universitäts-Rechenzentrum, Heidelberg).

### 2.3 Conductivity measurements

The dc-conductivity was measured by the standard four-probe method<sup>16</sup> along the needle axis (length 2–4 mm) parallel to the perylene stacks. The samples were mounted on gold wires in a Teflon block as described by Coleman.<sup>17</sup> The current and voltage contacts were made with silver-(Dupont 4929) or gold-paint (Doduco Auromal GM7). The current contacts covered the ends of the crystal to assure uniform current distribution. Currents from 0.5–10  $\mu\text{A}$  (HP 4329A) were used for the measurements. The voltage contacts were carefully painted completely around the crystal. The voltage was measured with a Digital Nanovoltmeter (Keithley Model 180). The samples were placed in a continuous flow cryostat (Oxford Instruments CF 204 with Digital Temperature Controller DTC-2) for lowering the temperature.

The temperature dependence of the conductivity was measured on about twenty  $(\text{pe})_2(\text{PF}_6)_{1.1} \times 0.8 \text{CH}_2\text{Cl}_2$  crystals. The room temperature conductivity  $\sigma_{(300)}$  for these crystals varied between  $160 (\Omega \text{cm})^{-1}$  and  $1.100 (\Omega \text{cm})^{-1}$ . A typical value for the majority of the samples was about  $600 (\Omega \text{cm})^{-1}$ . Very similar results were obtained for the  $(\text{pe})_2(\text{AsF}_6)_{1.1} \times 0.7 \text{CH}_2\text{Cl}_2$  crystals. The room temperature conductivity  $\sigma_{(300)}$  varied between  $210 (\Omega \text{cm})^{-1}$  and  $1200 (\Omega \text{cm})^{-1}$ , a typical value was about  $700 (\Omega \text{cm})^{-1}$ .

For (1) as well as for (2) two types of crystals were measured. The first type of crystals was very thin ( $\sim 0.005$ – $0.01$  mm) and at least 10 to 50

times as broad, while the second type of crystals was thicker ( $\sim 0.05$  mm) and only 1 to 3 times as broad. The room temperature conductivities of both types of crystals are comparable but the temperature dependence is somewhat different. Figure 2 shows the typical temperature dependence of the conductivity normalized to the room temperature value  $\sigma_{(300)}$  for the type one crystals of (1) and (2). Figure 3 shows the same temperature dependence for the second type of crystals. The main difference of both conductivity characteristics is that in the second case the conductivity makes a sudden jump at a temperature of about 110–120 K. In the moment it is not clear if this change in conductivity is due to a phase transition.

In the case of (3) and (4) only one type of crystal is found [similar to the type one crystals of (1) and (2)]. The room temperature conductivity  $\sigma_{(300)}$  of these crystals of (3) and (4) varied between 70 and 700  $(\Omega \text{ cm})^{-1}$  and is therefore somewhat lower than the r.t. conductivity of both forms of (1) and (2).

The temperature dependence of the normalized conductivity of (3) and (4) is shown in Figure 4 and is similar to the one of type one crystals of (1) and (2) in Figure 2. Nevertheless, in the semiconducting region below 180 K the observed activation energy for (3) and (4) is only about half as large as for type one of (1) and (2).

## 2.4 EPR measurements

X-Band EPR spectra were taken on a Bruker B-ER 420 spectrometer equipped with a Bruker B-NM 12 NMR Gaussmeter for field calibration. Absorption intensities were obtained by double integration of the usual first derivative lines by means of a Bruker ASPECT 2000 computer.

EPR spectra of three *thick* (type two) single crystals of (1) show one strong, exchange narrowed line of Lorentzian shape. The peak to peak linewidth of only 0.6 Gauss remains constant between 300 K and 190 K. Below that temperature the linewidth slightly increases with decreasing temperature, being 0.8 Gauss at 100 K. The EPR absorption intensity is constant within  $\pm 2\%$  from 300 K down to 190 K. At lower temperatures one observes an activated paramagnetism. Because of the small temperature interval in which that paramagnetism could be observed in our experiment the activation energy can be given only very roughly. We estimate the value as 0.025 eV.

## 3 STRUCTURE DETERMINATION

The structure determination of (2) is described in detail. The structure was solved in the space group Pnmm by direct methods using the SHELX 76 program. The highest peaks in the *E*-map are assigned to the carbon atoms of



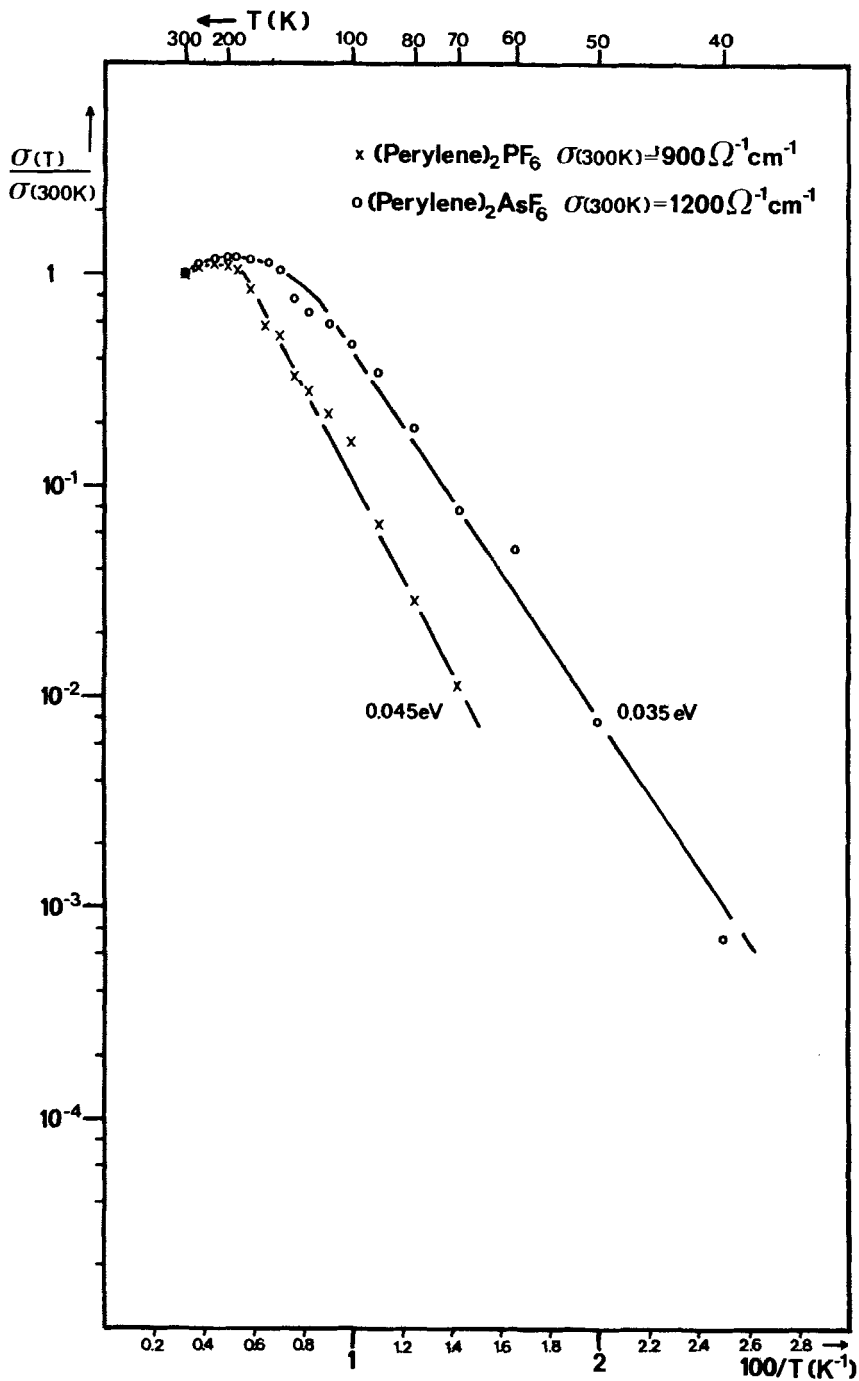


FIGURE 2 Temperature dependence of the normalized d.c. conductivity of "thin" crystals (first type: see text) of (1) ["(Perylene)<sub>2</sub>PF<sub>6</sub>"] and (2) ["(Perylene)<sub>2</sub>AsF<sub>6</sub>"].

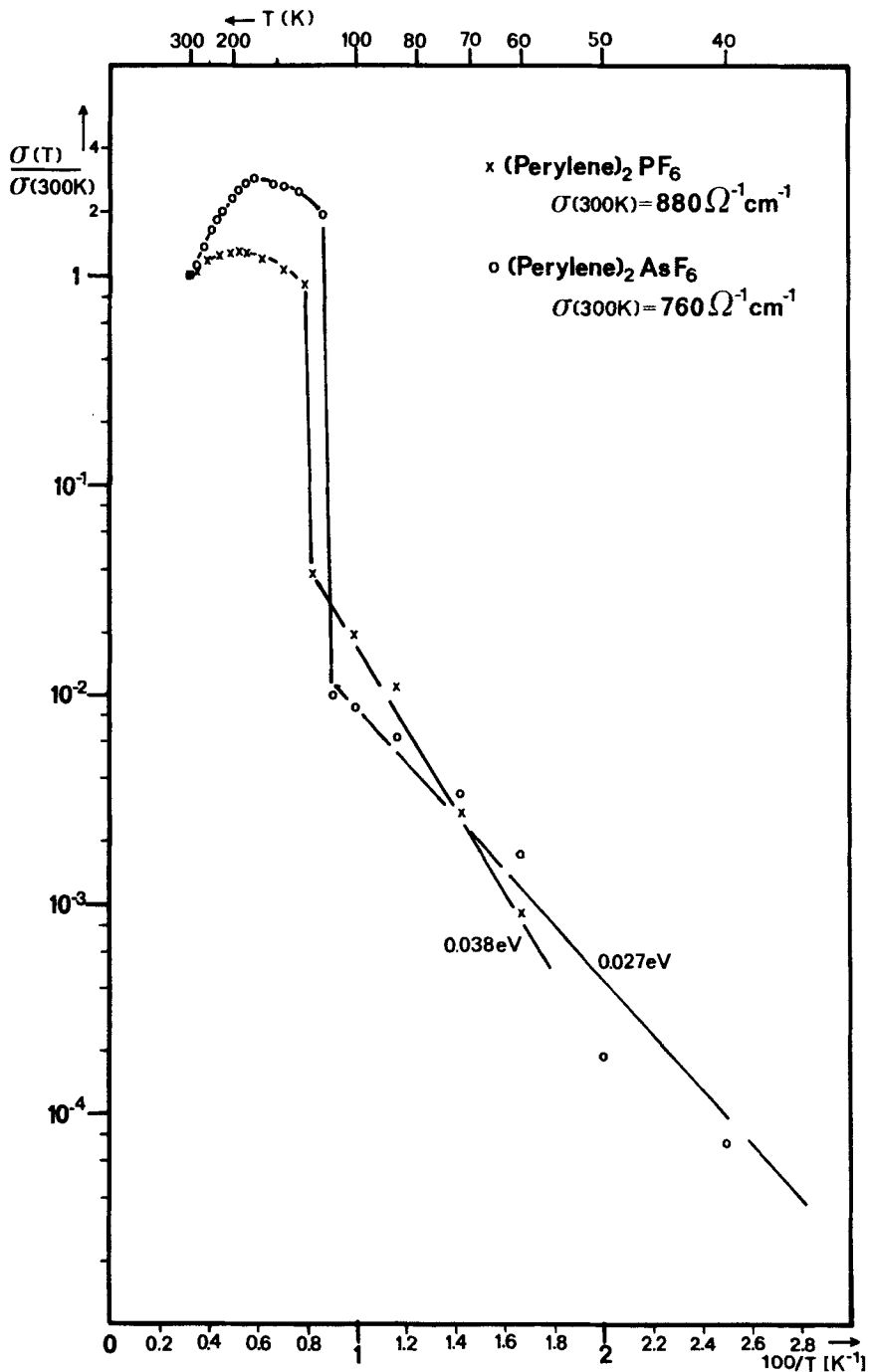


FIGURE 3 Temperature dependence of the normalized d.c. conductivity of "thick" (second type: see text) of (1) and (2).

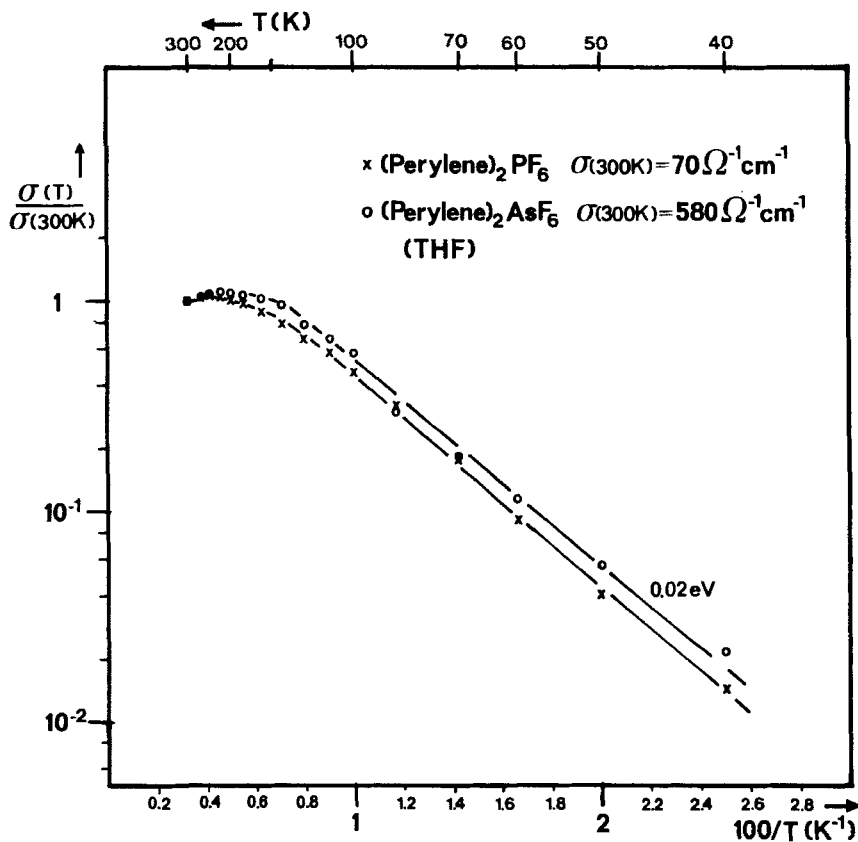


FIGURE 4 Temperature dependence of the normalized d.c. conductivity of (3) [(Perylene) $_2$ PF $_6$ (THF)] and (4) [(Perylene) $_2$ AsF $_6$ (THF)].

the perylene, which lies in the special position 2b  $(0, \frac{1}{2}, 0)$ . The perylene molecules form isolated stacks in the direction of the a-axis with a translational period of  $4.29 \text{ \AA}$ . The channels between these stacks are filled by anions and solvent molecules. The localization of these molecules is more difficult. Another strong peak in the special position 2a  $(0, 0, 0)$  gives the position of the As atom, but the peak height is too small and no F atoms can be found. As the chemical analysis gives a perylene: AsF $_6$  ratio of 2:1.1, the special position 2a can only partially be occupied by anions. Considering the cell and the form of the anion this is sensible. In the direction of the a-axis the distance between anions in the same channel would come to  $4.29 \text{ \AA}$ , less than the shortest extension of the anion. Therefore, only every second cell in a channel is occupied by an anion, the holes between are filled by solvent molecules. This distribution gives a perylene: AsF $_6$  ratio of 2:1. With an

analysis ratio of 2:1.1, some anions have to occupy adjacent cells in the same channel. This is only possible, if they are shifted from one another in the direction of the *a*-axis. The very big anisotropic temperature factor in this direction for As (see Table III) may be produced by this displacement. In the case of an ordered arrangement the cell parameter has to be doubled and in the reciprocal space intermediate layers must appear, to which only the anions and solvent molecules would contribute. On rotation photographs of (2) diffuse streaks appear in the middle between the layers perpendicular to *a*. We conclude from this that the distribution in the channels has a translational period of  $2a$ , but without correlation between different channels in the *bc* plane. A refinement with all carbon atoms and the As atom gives  $R = 0.27$ . The difference Fourier map shows several peaks around the As atom. Three of these form with the As atom an octahedron with distances of about  $1.7 \text{ \AA}$  and angles of  $90^\circ$ . The mirror plane of the space group goes through the  $\text{AsF}_6$ -octahedron, but doesn't coincide with the

TABLE II  
Coordinates and temperature factors for  $(\text{pe})_2\text{PF}_6(\text{CH}_2\text{Cl}_2)$

	<i>x</i>	<i>y</i>	<i>z</i>	$U_{11}$
C(1)	0.0936 (0.0050)	0.5000	0.8314 (0.0014)	0.0463 (0.0052)
C(2)	0.2999 (0.0047)	0.5000	0.9189 (0.0013)	0.0387 (0.0046)
C(3)	0.3985 (0.0029)	0.5955 (0.0010)	0.9592 (0.0008)	0.0392 (0.0028)
C(4)	0.2901 (0.0036)	0.6871 (0.0011)	0.9204 (0.0010)	0.0507 (0.0039)
C(5)	0.0800 (0.0040)	0.6853 (0.0012)	0.8394 (0.0011)	0.0577 (0.0043)
C(6)	-0.0094 (0.0034)	0.5938 (0.0012)	0.7999 (0.0009)	0.0559 (0.0037)
H(1)	0.2450	0.7685	0.9025	0.05
H(2)	0.3143	0.7009	0.8086	0.05
H(3)	-0.1626	0.6169	0.7403	0.05
P	0.0000	0.0000	1.0000	(1)
F(1)	0.2577 (0.0145)	0.0816 (0.0045)	0.9676 (0.0040)	0.0438 (0.0133)
F(2)	0.1330 (0.0143)	-0.0776 (0.0053)	0.4263 (0.0046)	0.0238 (0.0146)
F(3)	0.1858 (0.0155)	-0.0412 (0.0047)	0.5623 (0.0047)	0.1675 (0.0264)
Cl(1)	-0.0361 (0.0099)	0.0397 (0.0026)	0.5979 (0.0026)	0.0724 (0.0097)

(1) Temperature factors for <i>P</i> :					
$U_{11}$	$U_{22}$	$U_{33}$	$U_{23}$	$U_{13}$	$U_{12}$
0.1375 (0.0209)	0.0419 (0.0084)	0.0637 (0.0103)	0.0	-0.0488 (0.0130)	0.0

mirror plane of the octahedron, thus producing two orientations for  $\text{AsF}_6^-$ . Two other peaks are assigned to the chlorine atoms of  $\text{CH}_2\text{Cl}_2$ . A refinement with all non-hydrogen atoms (As anisotropic) lowers the  $R$ -values to  $R = 0.102$  and  $R_w = 0.077$ , adding the H-atoms in calculated positions gives the final values of  $R = 0.088$  and  $R_w = 0.061$ .

For (1) similar results were obtained. But the assignment of the peaks around the P gave no reasonable results. Thus we assume that the smaller  $\text{PF}_6^-$  octahedron is not as well fixed in the available space between the cation stacks and thus no special orientation for  $\text{PF}_6^-$  could be found.

Rotation- and Weissenberg-photographs of (3) agreed exactly with that of (1).

#### 4 DESCRIPTION OF THE STRUCTURE

Tables II and III show the atom coordinates and temperature factors for (1) and (2) respectively. The coordinates for the carbon atoms show the similarity of the structures of both compounds (1) and (2). The perylene molecule occupies the special position 2b ( $0, \frac{1}{2}, 0$ ). The angle between the perylene and the bc-plane is  $37.7^\circ$  and the intermolecular distance is  $3.40 \text{ \AA}$ . Atomic distances and angles of the perylene ion in (2) with the standard deviations are shown in Figure 5. Figure 6 shows a projection of two adjacent perylenium cations perpendicular to the perylene plane. A projection of the

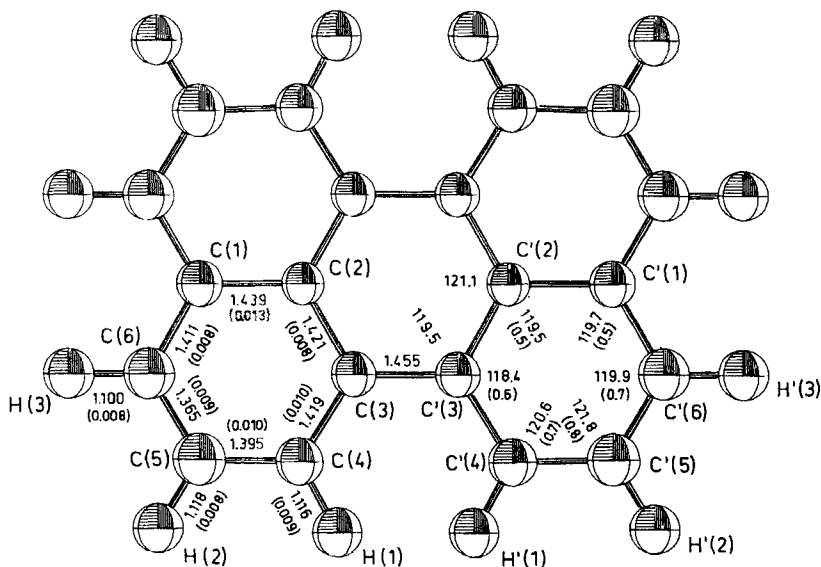


FIGURE 5 Atom distances and angles of the peryleniumyl ion in (2).

TABLE III

Coordinates and temperature factors for (pe)<sub>2</sub>AsF<sub>6</sub>(CH<sub>2</sub>Cl<sub>2</sub>)

	<i>x</i>	<i>y</i>	<i>z</i>	<i>U</i> <sub>11</sub>		
C(1)	0.0960 (0.0025)	0.5000 (0.0)	0.8386 (0.0008)	0.0427 (0.0028)		
C(2)	0.2999 (0.0025)	0.5000 (0.0)	0.9194 (0.0008)	0.0367 (0.0028)		
C(3)	0.3967 (0.0016)	0.5946 (0.0006)	0.9592 (0.0005)	0.0407 (0.0020)		
C(4)	0.2863 (0.0019)	0.6865 (0.0006)	0.9177 (0.0006)	0.0462 (0.0023)		
C(5)	0.0895 (0.0020)	0.6838 (0.0006)	0.8392 (0.0006)	0.0563 (0.0024)		
C(6)	-0.0074 (0.0021)	0.5937 (0.0006)	0.8002 (0.0005)	0.0520 (0.0021)		
H(1)	0.3639	0.7617	0.9468	0.05		
H(2)	0.0053	0.7575	0.8082	0.05		
H(3)	-0.1686	0.5942	0.7397	0.05		
As	0.0000	0.0000	1.0000	(1)		
F(1)	0.2641 (0.0037)	0.0549 (0.0019)	0.9259 (0.0041)	0.1668 (0.0042)		
F(2)	0.6913 (0.0032)	0.0837 (0.0016)	0.9601 (0.0013)	0.1207 (0.0041)		
F(3)	0.0906 (0.0032)	0.0723 (0.0014)	1.0902 (0.0012)	0.2853 (0.0037)		
Cl(1)	-0.0553 (0.0042)	-0.0958 (0.0031)	0.9478 (0.0028)	0.1386 (0.0043)		
Cl(2)	-0.0248 (0.0040)	-0.0217 (0.0023)	1.1071 (0.0017)	0.0735 (0.0041)		
H(4) <sup>a</sup>	-0.2311	0.0280	0.9846	0.05		
H(5) <sup>a</sup>	-0.1699	0.0565	0.9791	0.05		
(1) Temperature factors for As:						
	<i>U</i> <sub>11</sub>	<i>U</i> <sub>22</sub>	<i>U</i> <sub>33</sub>	<i>U</i> <sub>23</sub>	<i>U</i> <sub>13</sub>	<i>U</i> <sub>12</sub>
	0.1668 (0.0037)	0.0341 (0.0023)	0.0507 (0.0026)	0.0	0.0249 (0.0033)	0.0

<sup>a</sup> H-atoms for CH<sub>2</sub>Cl<sub>2</sub> in the AsF<sub>6</sub> compound.

unit cell on the bc-plane shows Figure 7, the anions, which lie between the perylene cations are omitted. A cation stack shows Figure 8. This corresponds to a projection onto the ac-plane. The distances and angles of the perylene ion are in agreement with the values found for perylene.<sup>18</sup> Because of the agreement of the structures of perylene [Pt(mnt)<sub>2</sub>]<sup>19</sup> and the present compound it must be assumed that the perylene determines the structure by making segregated stacks, whereas the remaining place is filled by anions. In this case the place is filled by AsF<sub>6</sub> and CH<sub>2</sub>Cl<sub>2</sub>, where the AsF<sub>6</sub> octahedron has an orientation between the cation stacks, which gives a suitable

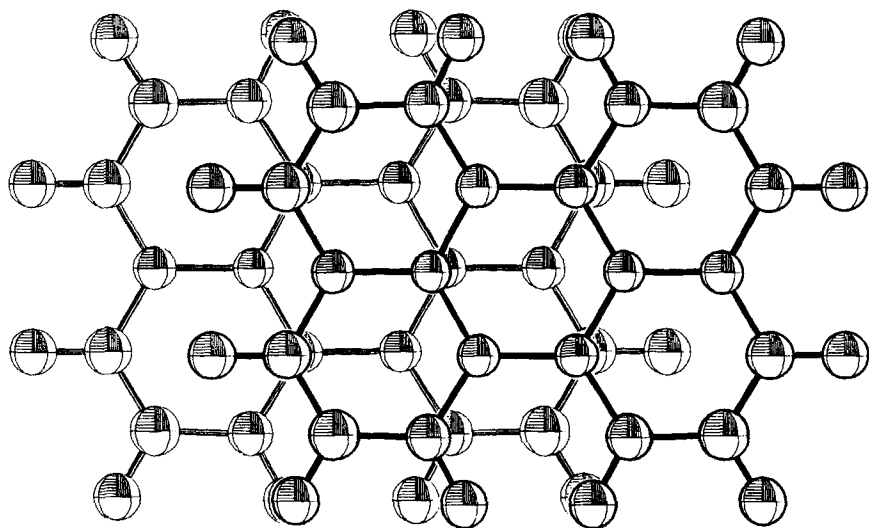


FIGURE 6 Projection of two adjacent perylenium ions perpendicular to the perylene plane.

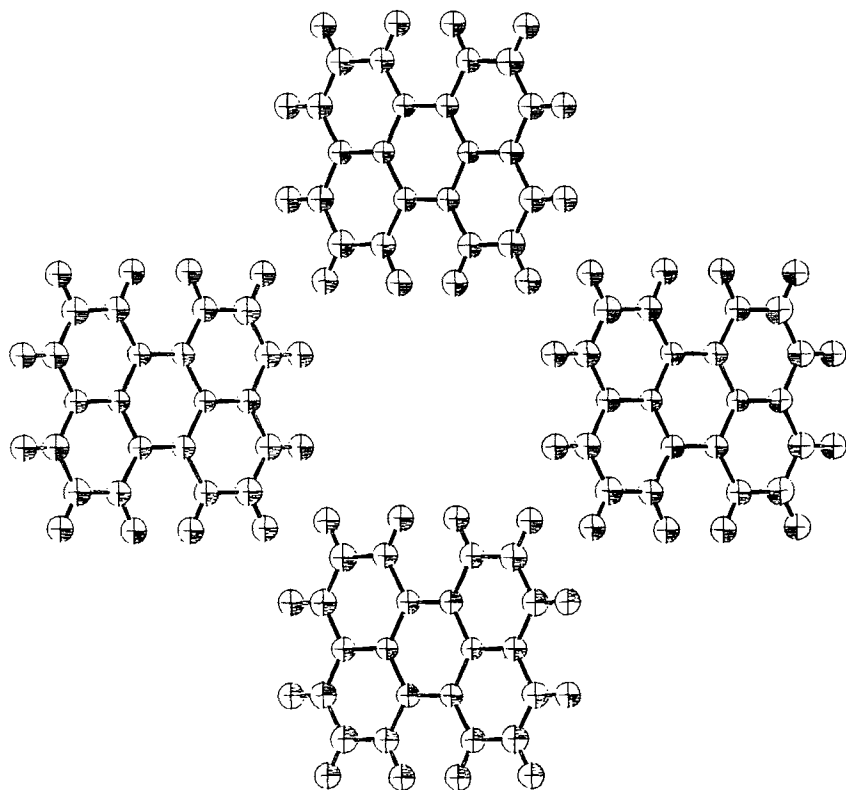


FIGURE 7 Projection of the unit cell on the bc-plane (anions are omitted).

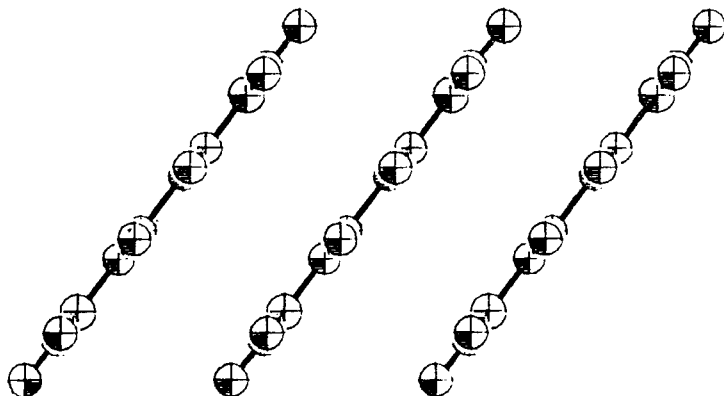


FIGURE 8 Projection of a cationic stack onto the *ac* plane.

filling of the space in the channels. As the anions are disordered, the found distances and angles (AsF: 1.63, 1.70, 1.81 and F-As-F: 92.3, 94.4, 93.9) are not very reliable.

## 5 DISCUSSION

It is well known that perylene can easily be oxidized by halogens—especially bromine<sup>20–24</sup> and iodine<sup>20,23,25–31</sup>—by different metal salts<sup>27,32–34</sup> like antimonyhexachloride,<sup>32–35</sup> silverperchlorate<sup>34</sup> and electrochemically.<sup>4,6</sup> Though it has been pointed out very early that some of these oxidized solids are quite good electrical conductors (at least compared to other typical organic compounds, which are insulators),<sup>22</sup> detailed physical and structural data on single crystals, so far have only been obtained for  $(pe)_2 Pt(mnt)_2$  (*mnt* = anion of dithiomaleodinitrile).<sup>19</sup> The latter solid was obtained by oxidizing perylene with iodine in the presence of tetra-*n*-butylammonium- $Pt(mnt)_2$ ,<sup>36</sup> as well as electrochemically.<sup>6</sup> The principle arrangement of peryleniumyl cations in the hexafluoroarsenate (V) and hexafluorophosphate (V) is very similar to the columns in  $(pe)_2 [Pt(mnt)_2]$ .<sup>19</sup>

The d.c.-conductivities parallel to the *a*-axes of all crystals of (1), (2), (3) and (4) in the temperature range between 300 and 190 K show a metallic-like behaviour. Below 190 K the conductivities of the crystals decrease with decreasing temperatures. In this *semiconducting* region the conductivity of the crystals of (1) and (2) depend on the “shape” of the crystals. In the case of the thicker crystals (type two) of (1) and (2) a sharp drop of the conductivity at a temperature of about 110–120 K is observed. The thinner crystals (type one) of (1), and (2) show a smooth decrease in the conductivity with decreasing temperature below 190 K. The observed activation energies in the semiconducting region are smaller for (3) and (4) than for (1) and (2) (see Figure 2–4). Nevertheless, this activation energies are very similar to



the one estimated from the ESR-measurements. Temperature cycling experiments between 300 and 50 K have shown a degradation of the room temperature conductivity which was particularly sensitive for those crystals of (1), (2), (3), and (4) which primarily showed the highest conductivities. Similar observations were made earlier for other one-dimensional organic metals.<sup>37</sup>

The relatively high conductivities of these compounds are possibly connected with the nonintegral formal oxidation numbers on the organic part of the structure. The "substitution" of some of the counter anions (like  $\text{PF}_6^-$  or  $\text{AsF}_6^-$ ) in the channels by *neutral* solvent molecules ( $\text{CH}_2\text{Cl}_2$  or THF) leads to a mixed valence state for the cationic stacks. This mixed valence state which is recognized as a prerequisite for high conductivities,<sup>9-11</sup> is normally brought about by partial charge transfer but in the cases discussed here determined by stoichiometry. In this respect it is not surprising that crystals with different  $\text{CH}_2\text{Cl}_2/\text{PF}_6^-$  or different  $\text{CH}_2\text{Cl}_2/\text{AsF}_6^-$  ratios show different d.c. conductivity behaviour (type one and type two crystals of (1) and (2)), (see Sections 2.1.3.1; 2.1.4.1 and 2.3).

In by far the most highly conducting organic charge transfer complexes one observes very sharp epr lines, only very few exceptions are known.<sup>38,39</sup> A linewidth of 0.6 Gauss as in our case thus clearly hints to a rather strong exchange interaction between neighbouring molecules. This interaction apparently leads to a metallic-like behaviour between 300 K and 190 K as can be seen from temperature independent paramagnetism and conductivity data. Such a temperature independent paramagnetism has been likewise observed in NMP-TCNQ,<sup>40</sup> a well-known organic metal. The similarity of the magnetic properties of NMP-TCNQ and (1) is also apparent from the temperature dependence of the linewidth. We are thus quite sure that (1) can be considered as an organic metal at higher temperatures whereas it behaves as a paramagnetic semiconductor at temperatures below 190 K.

The hexafluoroantimonate (V) salt (5) which differs in composition from (1)–(4) shows a d.c. conductivity which is about two orders of magnitude lower than those of (1)–(4). Consistent with this observation is the result of electrocrystallization experiments using  $\text{SbF}_6^-$  as counterion: only small black prisms covering the electrode have been obtained (see Section 2.1.5).

## 6 CONCLUSION

Electrocrystallization is a valuable tool in the search for new highly conducting solids. This method selects crystals with the highest electrical conductivities and not those with the lowest free energies under the experimental conditions as found by using chemicals as oxidation agents.

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