

Enhanced thermoelectric properties in hybrid graphene/boron nitride nanoribbons

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The thermoelectric properties of hybrid graphene/boron nitride nanoribbons (BCNNRs) are investigated using the nonequilibrium Green's function approach. We find that the thermoelectric figure of merit (ZT) can be remarkably enhanced by periodically embedding hexagonal BN (h -BN) into graphene nanoribbons (GNRs). Compared to pristine GNRs, the ZT for armchair-edged BCNNRs with width index $3p + 2$ is enhanced 10–20 times, while the ZT of nanoribbons with other widths is enhanced by just 1.5–3 times. As for zigzag-edge nanoribbons, the ZT is enhanced 2–3 times. This improvement comes from the combined increase in the Seebeck coefficient and the reduction in the thermal conductance outweighing the decrease in the electrical conductance. In addition, the effect of the component ratio of h -BN on the thermoelectric transport properties is discussed. These results qualify BCNNRs as a promising candidate for building outstanding thermoelectric devices.

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I. INTRODUCTION

Thermoelectric materials that can convert heat into electric energy have attracted increasing attention from both theoretical and experimental research due to their potential technological applications.^{1–5} The thermoelectric conversion efficiency is measured by the figure of merit^{6–9}

$$ZT = \frac{\sigma S^2 T}{(\kappa_e + \kappa_p)}, \quad (1)$$

where σ , S , T , κ_e , and κ_p represent the electrical conductance, Seebeck coefficient, absolute temperature, and the electron and phonon contributions to thermal conductance, respectively. Clearly, an optimal thermoelectric material should bear high electrical conductance, high Seebeck coefficient, and correspondingly, low thermal conductance.

Graphene and hexagonal boron nitride (h -BN) are single-atomic-layer honeycomb materials.^{10–18} Because of the quantum confinement effect, graphene and h -BN exhibit very high electrical conductance and Seebeck coefficient.^{19–21} However, both pristine graphene and pristine h -BN are poor thermoelectric materials because their thermal conductance is also very high.^{22–25} Recently, a novel nanomaterial compounded of graphene and h -BN has been synthesized by the thermal catalytic chemical vapor deposition method.²⁶ This novel hybrid nanostructure possesses physical properties distinct from both pristine graphene and pristine h -BN.^{27–40} For example, a tunable electronic energy gap dependent on the concentration of h -BN emerges in the two-dimensional hybrid sheet,^{41,42} while in the one-dimensional hybrid nanoribbons, it is found that the armchair-edged graphene nanoribbons (A-GNRs) embedded in h -BN are always semiconducting while zigzag-edged structures become half-metallic;^{43,44} the thermal conductance of hybrid nanoribbons is obviously reduced by the interface scattering.⁴⁵ Previous studies only focused on electronic and thermal transport,^{43–45} respectively, while the thermoelectric properties of hybrid graphene and h -BN (BCN) have been left unexplored thus far. Because

ordered superlattices can form promising structures for improving thermoelectric performance,⁴⁶ it is natural to look into what happens if we construct a periodic structure composed of alternating h -BN and graphene.

In this paper, we investigate the thermoelectric properties of BCN nanoribbons (BCNNRs) by using the nonequilibrium Green's function (NEGF) approach.^{47–52} It is found that the figure of merit can be remarkably enhanced by periodically embedding h -BN into the GNRs. In particular, the ZT is enhanced 10–20 times for metallic A-GNRs with width index $3p + 2$ (where p is a positive integer), while the ZT of nanoribbons with other widths is enhanced ~ 1.5 –3 times. Regarding the zigzag-edge structure, the figure of merit for BCNNRs is improved 2–3 times compared to that of pristine zigzag-edged GNRs (Z-GNRs). The origin of this enhancement effect is analyzed: it is found that both the electrical conductance and thermal conductance decrease, whereas the Seebeck coefficient increases with the periodic number, N . In addition, the influence of the component ratio of h -BN on the thermoelectric transport properties is also discussed. Here, it is shown that the figure of merit of BCNNRs at component ratio near 0.5 assumes the largest value. These results indicate that the BCNNRs could be a very promising candidate for the design of excellent thermoelectric devices.

II. MODEL AND METHOD

Let us consider the armchair-edged BCNNRs (A-BCNNRs) shown in Figs. 1(a) and 1(b) and zigzag-edged BCNNRs (Z-BCNNRs) shown in Figs. 1(c) and 1(d), respectively. The systems are divided into three parts. The left and right leads composed of GNRs have perfect periodicity along the ribbon axis and the ribbon widths are labeled by W_A and W_Z for armchair and zigzag edges, respectively. The central scattering region is the hybridized structure consisting of alternating h -BN and graphene supercells. $L_S = (L_{BN} + L_C)$ and $L'_S = (L'_{BN} + L'_C)$ are the lengths of the supercell in armchair- and zigzag-edged nanoribbons, respectively, where

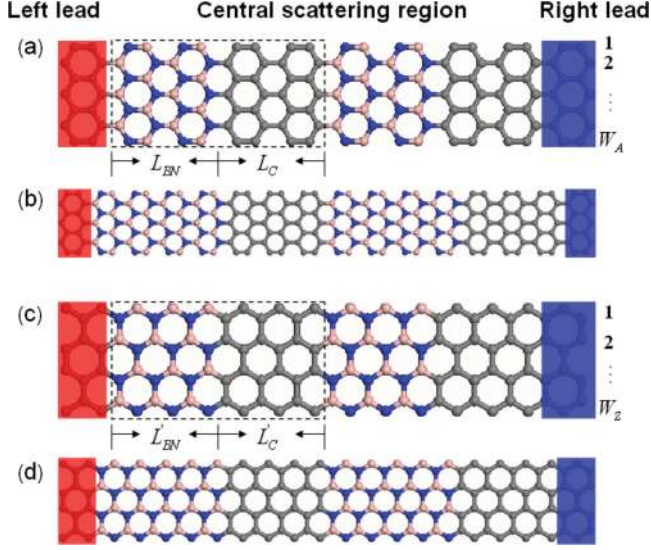


FIG. 1. (Color online) Schematics of (a) and (b) A-BCNNRs, and (c) and (d) Z-BCNNRs. The left and right leads made of GNRs have perfect periodicity along the ribbon axis. The central scattering region is formed by periodic alternating of *h*-BN and graphene. The dotted pane denotes a supercell in which L_{BN} (L'_{BN}) and L_C (L'_C) separately represent the lengths of *h*-BN and graphene for armchair- and zigzag-edged nanoribbons in the longitudinal direction, while W_A and W_Z are used to denote the width of the nanoribbons, respectively. Note that in (a) and (c) the length of *h*-BN equals the length of graphene, while in (b) and (d) they are different.

L_{BN} (L'_{BN}) and L_C (L'_C) separately represent the lengths of *h*-BN and graphene in the supercell in the longitudinal direction. We want to point out that in Figs. 1(a) and 1(c) the length of *h*-BN equals the length of graphene, while their lengths in Figs. 1(b) and 1(d) are different in the supercell. We use N for the number of supercells and the component ratio is defined as $\gamma = L_{BN}/L_S$ in A-BCNNRs and $\gamma' = L'_{BN}/L'_S$ in Z-BCNNRs. To investigate the thermoelectric properties of these structures, the NEGF method is employed. For the BCNNRs, the tight-binding Hamiltonian for electrons is written as

$$\begin{aligned} H &= \sum_{\alpha=L,C,R} H_\alpha + \sum_{\beta=L,R} (H^{\beta C} + H^{C\beta}) \\ &= \sum_{\alpha=L,C,R} H_\alpha + \sum_{i,j} t_{i,j}^{LC} (a_i^{\dagger L} a_j^C + \text{H.c.}) \\ &\quad + \sum_{i,j} t_{i,j}^{RC} (a_i^{\dagger R} a_j^C + \text{H.c.}), \end{aligned} \quad (2)$$

where $H_\alpha = \sum_i \varepsilon_i^\alpha a_i^\dagger a_i^\alpha + \sum_{i,j} t_{i,j}^\alpha a_i^\dagger a_j^\alpha$, a_i^\dagger and a_i^α are the creation and annihilation operators, ε_i^α is the site energy at the i th atom, and $t_{i,j}^\alpha$ is the hopping integral for nearest neighbors, respectively. We take the parameters of the Hamiltonian from Ref. 53 for graphene and Ref. 54 for *h*-BN, and take the average value for the hopping energy between carbon and boron or nitrogen atoms. The tight-binding parameters correspond to the $\pi-\pi^*$ chemical bonds. In general for *h*-BN, we should also consider the contribution of the $\sigma-\sigma^*$ bonds. However, as we will show in the following for calculating the figure of merit, the contribution of those bands can be

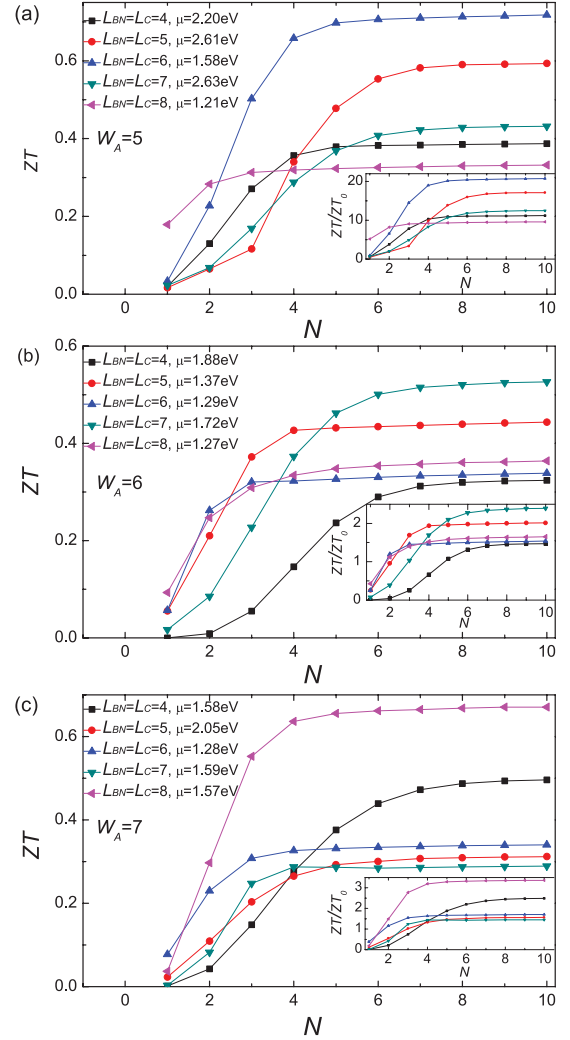


FIG. 2. (Color online) Thermoelectric figure of merit ZT for A-BCNNRs with different widths: (a) $W_A = 5$, (b) $W_A = 6$, and (c) $W_A = 7$, while the insets plot the ratio ZT/ZT_0 as a function of periodic number N for the nanoribbons, where ZT_0 corresponds to the result of perfect A-GNRs. The temperature $T = 300$ K and the component ratio $\gamma = 0.5$ (i.e., $L_{BN} = L_C$). For given lengths L_{BN} and L_C , we take the chemical potential (μ) corresponding to the absolute maximum of the figure of merit (a detailed discussion is given in the main text).

safely neglected. Starting from this Hamiltonian, the retarded Green's function is expressed as

$$G^r(E) = \left[(E + i0^+)I - H_C - \sum_L^r - \sum_R^r \right]^{-1}, \quad (3)$$

where E is the electron energy, I is the identity matrix, H_C is the Hamiltonian of the central scattering region, $\sum_\beta^r = H^{C\beta} g_\beta^r H^{\beta C}$ denotes the self-energy of the left (L) and right (R) regions, and g_β^r is the surface Green's function of the leads which can be computed by a recursive iteration technique.⁵⁵ Once the retarded Green's function is obtained, we can calculate the electronic transmission coefficient

$$T_e(E) = \text{Tr}[G^r(E)\Gamma_L G^a(E)\Gamma_R], \quad (4)$$

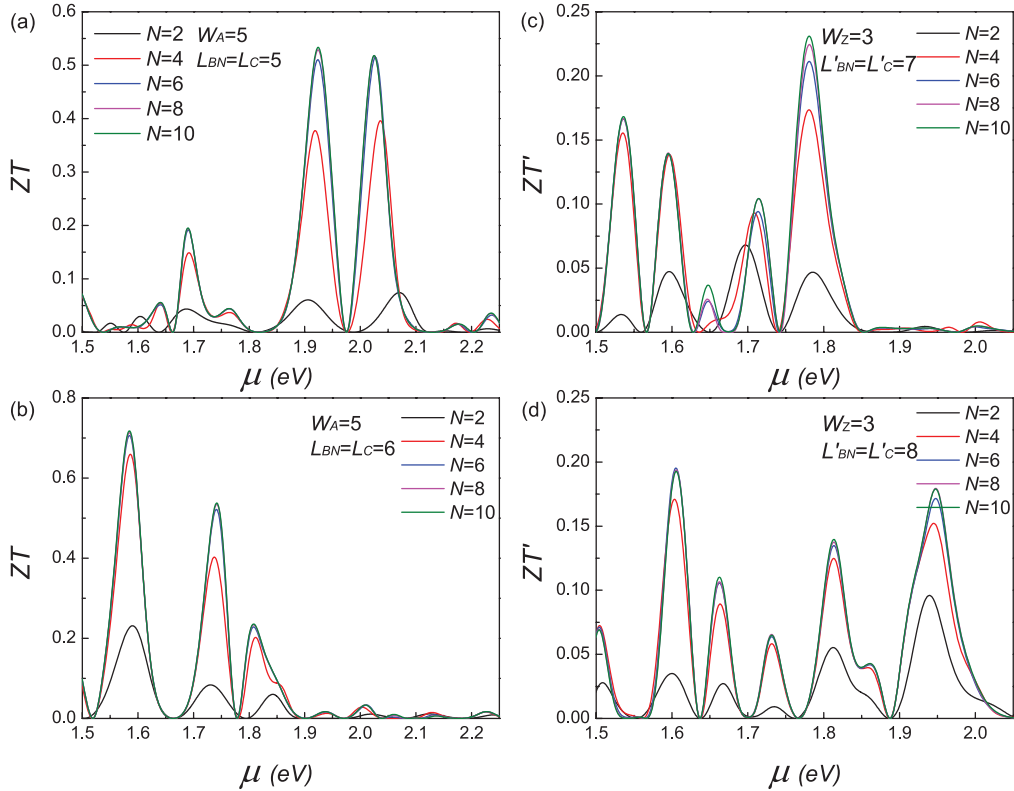


FIG. 3. (Color online) (a) and (b) Thermoelectric figure of merit (ZT) as a function of chemical potential (μ) for A-BCNNRs with width $W_A = 5$ at different periodic number (N) and different length of h -BN and graphene nanoribbons: (a) $L_{BN} = L_C = 5$ and (b) $L_{BN} = L_C = 6$. (c) and (d) Thermoelectric figure of merit (ZT) as a function of chemical potential (μ) for Z-BCNNRs with width $W_Z = 3$ at different periodic number (N) and different length of h -BN and graphene nanoribbons: (c) $L'_{BN} = L'_C = 7$ and (d) $L'_{BN} = L'_C = 8$. To evaluate the figure of merit, in all cases we have fixed the chemical potential by taking the absolute maximum of ZT as a function of the periodic number and length of the graphene and BN nanoribbons. The temperature is $T = 300$ K.

where $G^a(\omega) = [G^r(\omega)]^\dagger$ is the advanced Green's function, and $\Gamma_\beta = i(\sum_\beta^r - \sum_\beta^a)$ describes the interaction between the left or right lead and the central region. By using the transmission coefficient, the electrical conductance, the Seebeck coefficient, and the electron contributed thermal conductance can be calculated in linear response separately from

$$\sigma(\mu) = e^2 L_0(\mu, T), \quad (5a)$$

$$S(\mu) = \frac{1}{eT} \frac{L_1(\mu, T)}{L_0(\mu, T)}, \quad (5b)$$

$$\kappa_e(\mu) = \frac{1}{T} \left[L_2(\mu, T) - \frac{L_1(\mu, T)^2}{L_0(\mu, T)} \right], \quad (5c)$$

where $L_n(\mu, T) = (2/h) \int_{-\infty}^{\infty} dE T_e(E) (E - \mu)^n [-\partial f_e(E, \mu, T)/\partial E]$ is the Lorenz function, μ is the chemical potential, e is the electric charge of carriers, f_e is the Fermi-Dirac distribution function at temperature T , and h is the Planck constant. From the expression of the Lorenz function, it now becomes clear why we can neglect the contribution of the σ bands. At finite temperature, the derivative of the Fermi function effectively restricts the integration range to a small region close to the chemical potential. Normally, the chemical potential is small compared with the energy between the σ and π bands in h -BN.

On the other hand, for thermal transport, according to Refs. 48–50, the phonon-transmission coefficient $T_p(\omega)$ can be

calculated in the harmonic approximation with the same steps as for the electrons. Indeed, in Eq. (3) one only needs to change E into ω^2 , H_C into K_C , and correspondingly compute the self-energy, and then get the phonon retarded Green's function, where ω is the vibrational frequency of phonons and K_C is the spring constant matrix of the central region^{24,25,50,56–58} (the parameters are taken from Ref. 57 for graphene and Ref. 58 for h -BN). We take the average value for the force constant between graphene and h -BN. The presence of the square of the frequency, ω^2 , instead of the phonon energy stems from the different analytical properties of the phonon Green's functions. Finally, the phonons contributed thermal conductance is given by

$$\kappa_p(T) = \frac{h}{4\pi^2} \int_0^\infty \frac{\partial f_p(\omega)}{\partial T} T_p(\omega) \omega d\omega, \quad (6)$$

where $f_p(\omega)$ is the Bose-Einstein distribution function for heat carriers. We have therefore all the ingredients to calculate the figure of merit.

III. RESULTS AND DISCUSSION

In Fig. 2, we investigate the thermoelectric transport properties of A-BCNNRs with different widths as a function of the number of supercell N for $\gamma = 0.5$ ($L_{BN} = L_C$). Figure 2(a) shows ZT for A-BCNNRs with width $W_A = 5$, and the inset plots the ratio ZT/ZT_0 , where ZT_0 is the figure of merit of

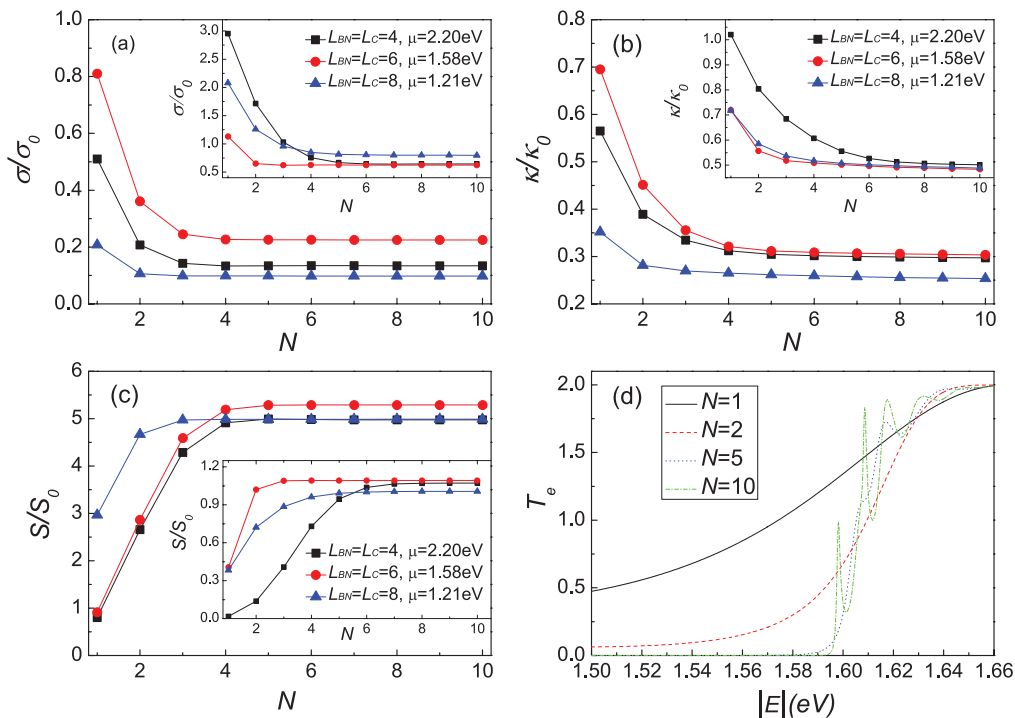


FIG. 4. (Color online) (a) Electrical conductance ratio σ/σ_0 , (b) thermal conductance ratio κ/κ_0 , and (c) Seebeck coefficient ratio S/S_0 as a function of periodic number N for A-BCNNRs with width $W_A = 5$, where σ_0 , κ_0 , and S_0 correspond to pristine A-GNRs; the insets show the results of nanoribbons with width $W_A = 6$. (d) Transmission coefficient T_e versus electronic energy E for A-BCNNRs with $W_A = 5$ at $L_{BN} = L_C = 6$ and $\mu = 1.58$ eV. The other parameters are the same as in Fig. 2.

pristine A-GNRs. In Fig. 2, the figure of merit plotted as a function of the geometrical properties of the nanoribbons for different lengths of the device are obtained by taking the maximum ZT as a function of the average chemical potential of the leads, where the chemical potential is taken to be small enough that the influence of interface states can be neglected. To do so, we have calculated ZT as a function of the chemical potential for all the geometries, and taken the *absolute* maximum of the figure of merit (see Fig. 3). For given lengths L_{BN} and L_C , we fix the chemical potential to the value that provides this absolute maximum also a function of the length N . This procedure fixes the chemical potential of the nanoribbon. We then use this value of the chemical potential to evaluate the figure of merit. We would like to point out that this procedure gives the maximum value of $ZT(ZT')$ for almost any N , as seen from Fig. 3. Only for small N , better values of $ZT(ZT')$ could be obtained by considering a different chemical potential.

One can find from Fig. 2(a) that when the periodic number $N = 1$, ZT is relatively small. Increasing N , ZT gradually increases for various values of L_{BN} and L_C . However, the magnitude of the enhancement is different, as can be seen from the inset of Fig. 2(a). At $L_{BN} = L_C = 4, 7$, and 8 , it is found that the figure of merit for A-BCNNRs is enhanced about ten times compared to that of pristine metallic A-GNRs. Interestingly, in the case of $L_{BN} = L_C = 5$ and 6 , a quite higher value $ZT/ZT_0 \approx 15$ – 20 is attained. When the periodic number is greater than 6 , we find that ZT/ZT_0 approaches almost a constant value. The reason for this behavior is easily explained: when the periodic number is sufficiently large, the whole nanoribbons can be approximately regarded as a superlattice structure composed of alternating h -BN and

graphene; then the transmission of electron and phonon in the A-BCNNRs is equivalent to that of the superlattice.

In Figs. 2(b) and 2(c), the thermoelectric properties of semiconducting A-GNRs embedded with h -BN are investigated. As a comparison, we plot the figure of merit for A-BCNNRs with width $W_A = 6$ and 7 . It can be seen that similarly to the case of metallic nanoribbons with width $W_A = 5$, ZT increases with the periodic number N and finally approaches a constant value for different L_{BN} and L_C . Nevertheless, compared to the perfect semiconducting A-GNRs, it is found that the figure of merit for A-BCNNRs is about 1.5–3.0 times larger than that of pristine graphene nanoribbons (see the insets). With further increasing the ribbon width, ZT is also increased with the periodic number and the enhanced magnitude is different for metallic and semiconducting A-GNRs. The ZT of metallic A-GNRs is enhanced 10–20 times, while the ZT of semiconducting A-GNRs is enhanced about 1.5–3 times.

To elucidate the origin of the thermoelectric figure of merit enhancement in A-BCNNRs, Figs. 4(a)–4(c) respectively depict the electrical conductance ratio (σ/σ_0), thermal conductance ratio (κ/κ_0), and Seebeck coefficient ratio (S/S_0) as a function of the periodic number N for the ribbon width $W_A = 5$, and accordingly, in the insets we plot the same quantities for nanoribbons with $W_A = 6$. The quantities with subscript 0 correspond to the parameters of pristine A-GNRs. It can be seen that both the electrical conductance and thermal conductance ratios decrease with increasing the periodic number, although the magnitude of the attenuation is different. In contrast to the perfect A-GNRs, the electrical conductance of A-BCNNRs is decreased nearly 2–3 times, while the thermal conductance is decreased 1–2 times, which suggests

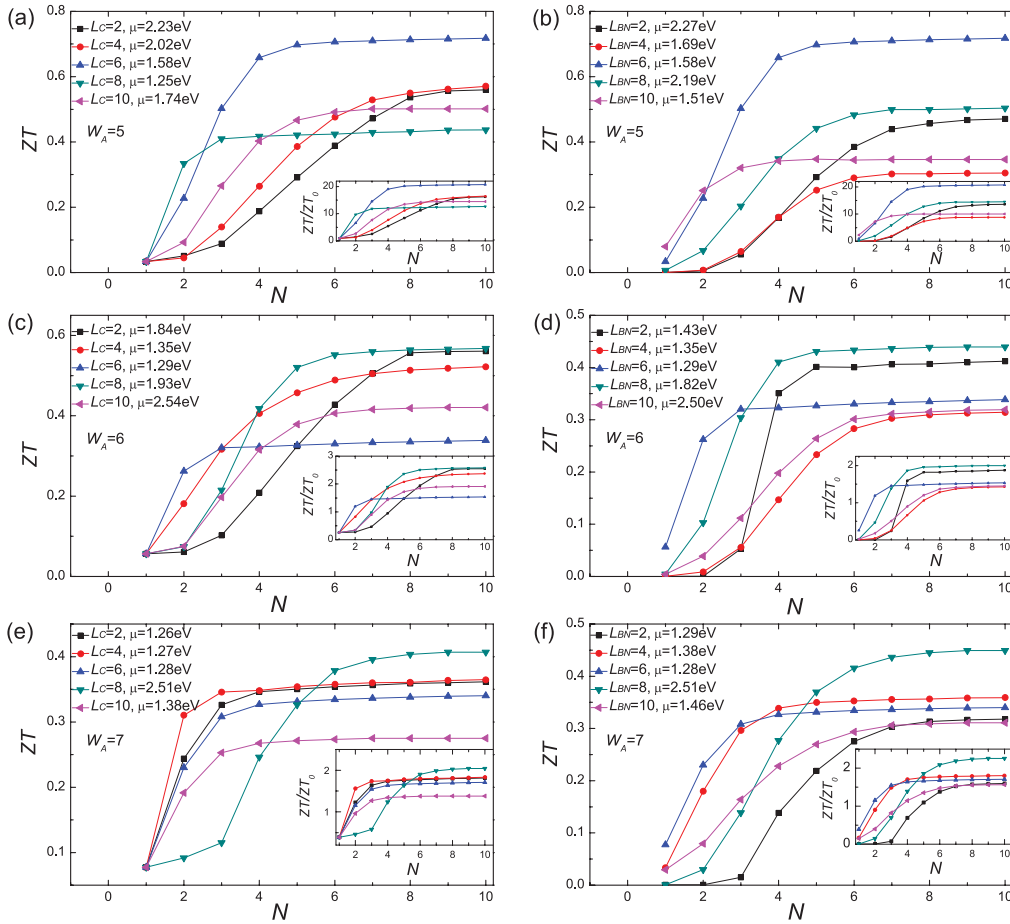


FIG. 5. (Color online) Thermoelectric figure of merit ZT for A-BCNNRs as a function of periodic number N . In the left column [(a), (c), and (e)] $L_{BN} = 6$, while in the right column [(b), (d), and (f)] $L_C = 6$. The insets plot the ratio ZT/ZT_0 of the nanoribbons. The chemical potential (μ) we have taken for different value L_{BN} (L_C) corresponds to the absolute maximum of the figure of merit and the other parameters are set the same as in Fig. 2.

that the hybrid structure has strong effects on the electronic conduction. The decreased electrical and thermal conductance ratios can be attributed to the destructive channels which block the carrier transmission. From Fig. 4(c), it is seen that the ratio of Seebeck coefficients increases with the periodic number N . The magnitude of the Seebeck coefficient for A-BCNNRs with width $W_A = 5$ is about five times larger than the Seebeck coefficient for pristine A-GNRs, and in any case larger than that of the nanoribbons with width $W_A = 6$ [see the inset of Fig. 4(c)]. Therefore, the enhancement of the figure of merit is mainly due to the increase of the Seebeck coefficient and the corresponding decrease of the thermal conductance. Together, they outweigh the decrease in the electrical conductance. From Eq. (5b), one can find that the Seebeck coefficient is dependent on the ratio of Lorenz function L_1/L_0 :

$$L_0(\mu, T) = \frac{2}{h} \int_{-\infty}^{\infty} \left[-\frac{\partial f_e(E, \mu, T)}{\partial E} \right] T_e(E) dE, \quad (7a)$$

$$L_1(\mu, T) = \frac{2}{h} \left\{ \int_{-\infty}^{E < \mu} \left[-\frac{\partial f_e(E, \mu, T)}{\partial E} \right] (E - \mu) T_e(E) dE + \int_{E > \mu}^{\infty} \left[-\frac{\partial f_e(E, \mu, T)}{\partial E} \right] (E - \mu) T_e(E) dE \right\}, \quad (7b)$$

where $\{-[\partial f_e(E, \mu, T)/\partial E]\}$ is a negative derivative of the Fermi function which effectively restricts the energy integration in a small region centered at the chemical potential as the temperature is finite. From Eq. (7a) it can be seen that L_0 is mainly determined by the electronic transmission coefficient $T_e(E)$: the larger $T_e(E)$, the larger L_0 . On the other hand, L_1 can be separated into two parts, for positive and negative values of $(E - \mu)$, respectively. By doing so we see that L_1 is proportional to the difference of $T_e(E)$ in the two parts. In Fig. 4(d), we report the electronic transmission coefficient T_e of A-BCNNRs with width $W_A = 5$ at $L_{BN} = L_C = 6$ and $\mu = 1.58$ eV as a function of the electron energy. It is found that T_e in the region $E < \mu$ decreases abruptly and an energy gap appears with increasing N , while T_e at $E > \mu$ varies slowly, leading to L_0 decreasing and L_1 increasing. Accordingly, the Seebeck coefficient increases with the periodic number.

In Fig. 5, the influence of the length of h -BN and graphene on the thermoelectric properties of A-BCNNRs is studied. In the left column [(a), (c), and (e)], we fix the length of h -BN L_{BN} and change L_C ; while in the right column [(b), (d), and (f)] we fix L_C and change L_{BN} . One can see that with the increase of the periodic number N , the figure of merit for nanoribbons with different widths gradually increases and

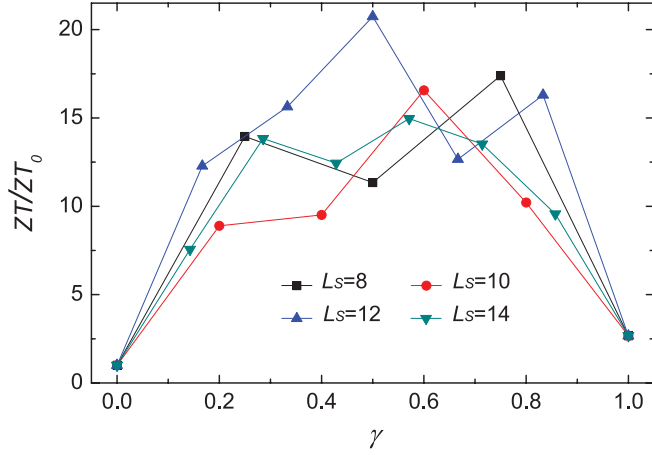


FIG. 6. (Color online) Thermoelectric figure of merit ratio ZT/ZT_0 of A-BCNNRs with width $W_A = 5$ versus component ratio γ , at periodic number $N = 10$ and temperature $T = 300$ K.

finally trends to a constant. Compared to pristine A-GNRs, the ZT of metallic nanoribbons is increased about 10–20 times [see the insets of Figs. 5(a) and 5(b)], while the ZT of semiconducting nanoribbons is enhanced about 1.5–3 times [see the insets of Figs. 5(c)–5(f)]. From these results we can conclude that the thermoelectric properties of A-GNRs can be tuned by embedding h -BN.

We next investigate the effect of the component ratio of h -BN on the thermoelectric transport properties in A-BCNNRs. Figure 6 shows the figure of merit ratio ZT/ZT_0 versus γ for nanoribbons with width $W_A = 5$ under different supercell length L_S . One can find that ZT/ZT_0 at $\gamma = 0$ and 1.0 is small, while in the range of γ between 0.2 and 0.8, in particular γ near 0.5, ZT/ZT_0 attains a relatively large value. The phenomenon can be qualitatively understood from the electronic energy band structure. For the pristine

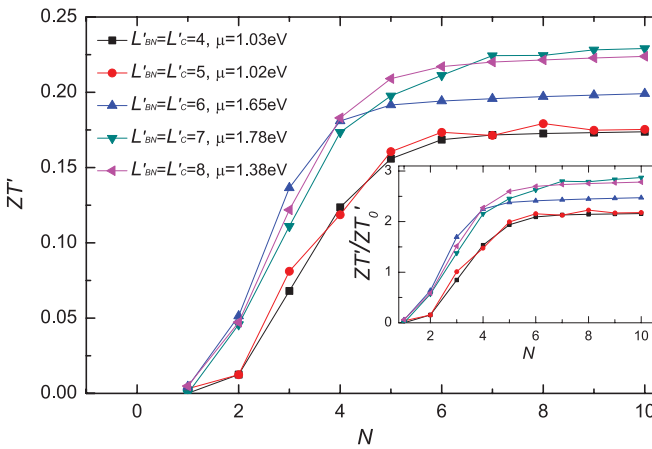


FIG. 7. (Color online) Thermoelectric figure of merit ZT' for Z-BCNNRs at room temperature, and the inset plots the ratio ZT'/ZT'_0 versus periodic number N , where ZT'_0 is the result of pristine Z-GNRs. Here we take $\gamma' = 0.5$ ($L'_{BN} = L'_C$) and $W_Z = 3$. For different lengths L'_{BN} and L'_C , we take the chemical potential (μ) corresponding to the absolute maximum of the figure of merit.

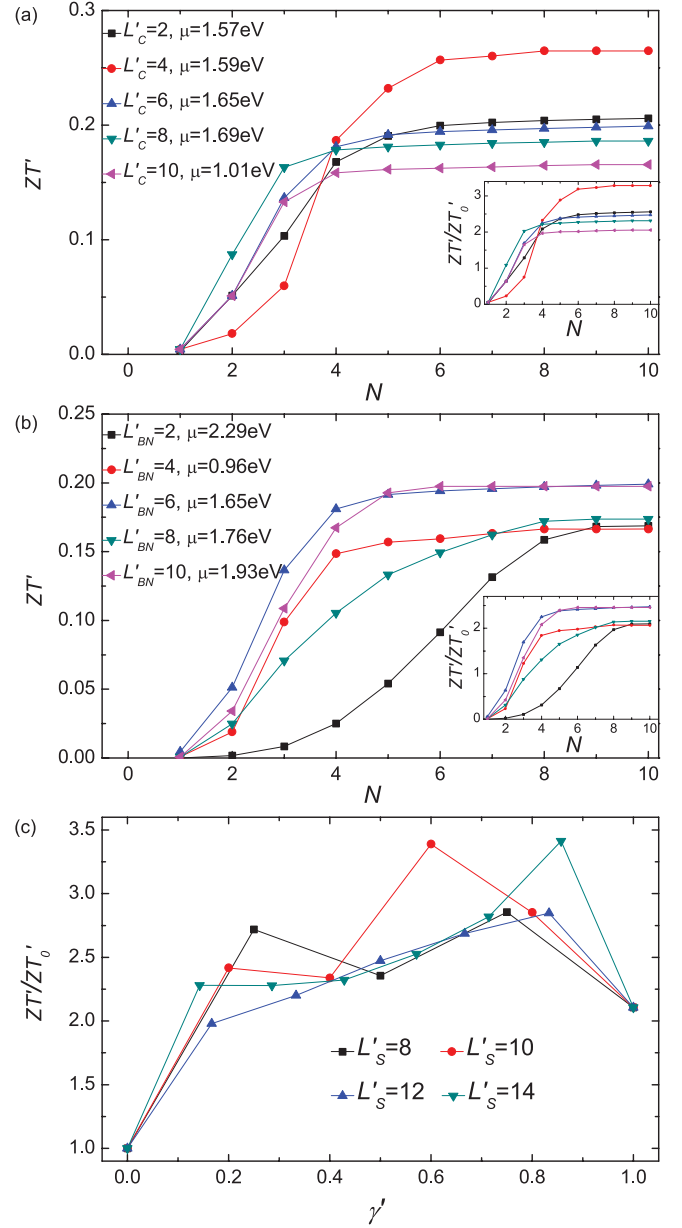


FIG. 8. (Color online) Thermoelectric figure of merit for Z-BCNNRs versus periodic number N under various component ratio. In (a) $L'_{BN} = 6$, (b) $L'_C = 6$, and the insets represent the ratio ZT'/ZT'_0 of the nanoribbons. (c) Figure of merit ratio versus component ratio γ' for different supercell length L'_S . We take the chemical potential (μ) in (a) and (b) corresponding to the absolute maximum of the figure of merit for given value L'_{BN} (L'_C) and the other parameters are the same as in Fig. 7.

A-GNRs, there exist perfect channels of carrier transport. By periodically embedding h -BN into the A-GNRs, on the one hand, the transport channels are destroyed, resulting in the thermal conductance and electrical conductance reducing, while on the other hand, a number of minigaps appear in the transmission spectrum. When the value of the component ratio γ gets close to 0.5, the width of some minigaps is sufficient to contribute to a high Seebeck coefficient. Moreover, the high Seebeck coefficient and decreased thermal

conductance outweigh the decreased electrical conductance. This implies that generating an effective electronic energy gap is important for improving the thermoelectric performance of materials.

The thermoelectric properties of Z-BCNNRs are discussed in Fig. 7 where we show the figure of merit ZT' as a function of the periodic number N for zigzag-edged nanoribbons with width $W_Z = 3$ at a fixed component ratio $\gamma' = 0.5$ ($L'_{BN} = L'_C$), and the inset plots the ratio ZT'/ZT'_0 , where ZT'_0 corresponds to the figure of merit of Z-GNRs. One can find that likewise to the case of A-BCNNRs, ZT' gradually increases with the periodic number and ultimately reaches a constant value. The figure of merit ZT' for Z-BCNNRs is improved 2–3 times compared to that of perfect Z-GNRs, as shown in the inset. The enhancement of the figure of merit for zigzag-edged nanoribbons is smaller than that of the metallic A-GNRs, while it is slightly larger than that of the semiconducting A-GNRs as seen by comparing Fig. 7 with Fig. 2. The origin of this behavior can be attributed to the zigzag-edged nanoribbons with higher thermal conductance than that of the armchair-edged nanoribbons of comparable width. As to the semiconducting A-GNRs, because of the energy gap, the thermoelectric figure of merit ZT_0 is higher than ZT'_0 , which outweighs the discrepancy of thermal conductance, resulting in a ZT/ZT_0 falling.

Figure 8 investigates the thermoelectric properties of Z-BCNNRs under different component ratios of h -BN and graphene. We first fix the length of h -BN in the supercell and change the length of graphene [Fig. 8(a)], and then fix the length of graphene and change the length of h -BN [Fig. 8(b)]. The insets correspond to the figure of merit ratio ZT'/ZT'_0 as a function of periodic number N , where ZT'_0 is the figure of merit of pristine Z-GNRs. One can find that the ZT' for Z-BCNNRs increases gradually with increasing the periodic number and finally reaches a constant. The enhanced magnitude, compared to the perfect Z-GNRs, is about 2–3 times (see the insets). The enhancement originates from the dramatically decreased thermal conductance and increased Seebeck coefficient outweighing the decreased electric conductance. In Fig. 8(c) the

figure of merit ratio ZT'/ZT'_0 versus the component ratio γ' for various supercell lengths L'_S is studied. It can be seen that ZT'/ZT'_0 in the range of γ' from 0.2 to 0.9 exhibits quite larger values and even exceeds 3.0. When γ' grows towards 1.0, ZT'/ZT'_0 decreases rapidly, while keeping a positive value in the tail because of the embedded h -BN.

IV. CONCLUSIONS

In summary, we have investigated the thermoelectric transport properties of BCNNRs using the NEGF technique. It is found that the thermoelectric figure of merit can be significantly enhanced by periodically embedding h -BN into GNRs. Compared to the pristine GNRs, the ZT of A-BCNNRs with the width index of $3p + 2$ is enhanced 10–20 times, while ZT of nanoribbons with other widths is enhanced about 1.5–3 times; as to zigzag-edge, the figure of merit for BCNNRs is improved 2–3 times. The strong enhancement can be justified by the combined increase in the Seebeck coefficient and decrease in the thermal conductance outweighing the decrease in the electrical conductance. In addition, the effect of the component ratio of h -BN on the thermoelectric properties is discussed. It is shown that the figure of merit in the case of γ (γ') nearing 0.5 exhibits a larger value. Our investigations demonstrate that the hybrid nanostructure composed of alternating h -BN and graphene is a very promising material for building highly efficient thermoelectric devices.

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