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# Enhancement of fluorescence-based sensing using microstructured optical fibres

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**Abstract:** We develop a generic model of excitation and fluorescence recapturing within filled microstructured optical fibres (MOFs) with arbitrary structure and demonstrate that the light-matter overlap alone does not determine the optimal fibre choice. Fibre designs with sub-wavelength features and high-index glasses exhibit localised regions of high intensity, and we show that these regions can lead to approximately two orders of magnitude enhancement of fluorescence recapturing. Here we show how this regime can be exploited for sensing and demonstrate experimentally in-fibre excitation and fluorescence recapturing within a filled, solid-core MOF.

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**OCIS codes:** (060.2370) fiber optics sensors; (060.4005) microstructured fibers; (280.1415) biological sensing and sensors

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## 1. Introduction

Microstructured optical fibres (MOFs) have the potential to dramatically improve the performance of fibre optic sensors based on absorption and fluorescence spectroscopy and have recently attracted considerable interest [1, 2, 3, 4, 5, 6, 7], since a significant portion of the guided light can be located in holes within the fibre. In the literature, the sensitivity of optical fibre sensors based on absorption and fluorescence spectroscopy is usually considered to be related solely to the power fraction of the guided mode field that is available for overlap with the ma-

terial to be sensed. Such sensors are generally referred to as evanescent-field sensors, reflecting the fact that only the evanescent tail of the guided mode is available for light-matter interactions. Examples of fibre geometries used for evanescent-based sensing include; tapered fibres [8, 9], D-shaped fibres [10], optical nanowires [11], solid-core MOFs [5, 12], multi-core MOFs [3, 4] and hollow-core MOFs [13]. Most evanescent-based sensing geometries only allow a limited light-matter overlap and thus are restricted in the sensitivity that they can achieve. Some typical mode-matter overlaps that have been reported include 0.1 – 0.2% for D-shaped fibres [10], 5.2% in an MOF [13], and multi-core MOFs with 6.5% [4]. Fibres with sub-wavelength cores including optical nanowires and MOFs with embedded nanowire-like cores [14, 15] allow overlaps as large as 99% as is evident from literature reports of 40% [12] and 99% [16]. These advances allow solid-core fibres to achieve light-matter overlaps as high as other field-based sensors which have access of up to 97% [17] or higher of the guided mode power such as capillary tubes [18], index-guiding filled hollow-core MOFs [19] and hollow-core photonic band gap fibres [20].

Although different MOF-variants of fluorescence-based sensors have been reported [2, 3, 4, 7, 10], the benefits that can be obtained using MOFs are far from being realised, largely due to the lack of a formalism for predicting and thus optimizing the measurable fluorescence power. Although models of the efficiency of fluorescence-based optical fibre sensors have been developed for simple structures such as tapered or D-shaped fibres, they have limited applicability because; 1) it is assumed that the modes of the fibre are the same at both the absorption and fluorescence wavelengths and 2) they are based on ray-optics [1, 21] or scalar electromagnetic fields (without including the effect of absorption loss) [8]. These models do not work well for fluorescent dyes with significant excitation-fluorescence wavelength separation (such as quantum dots), MOFs with wavelength-scale features and high contrast refractive indices, or complex sensing geometries when the hole surface of MOFs are coated (functionalized) with chemical-biological materials.

Here, we develop a general model of excitation and fluorescence recapturing within an MOF's modes based on guided mode solutions of vectorial form of Maxwell's equations and considering their losses at both emission and fluorescent frequencies. By evaluating the modal characteristics of a range of fibres at both wavelengths, we explore ways of enhancing the sensing sensitivity by maximising the fraction of the fluorescent photons that are coupled to the guided mode(s) of the filled microstructured fibres (the fluorescence capture fraction, *FCF*). We demonstrate that the light-matter overlap alone does not determine the optimal fibre choice as is usually assumed. Fibre designs with sub-wavelength features and high index glass exhibit localised regions of high intensity [22, 23], and we show that these regions lead to enhanced fluorescence recapturing and can be exploited for sensing. The significance of this parameter regime becomes clearer considering; 1) recent indication of the existence of such narrow-width high intensity regions at the interface of two different dielectrics, due to discontinuity of the electric field, in both waveguides [25] and MOFs [22, 23] and 2) sensing configurations based on surface excitation of chemical and biological layers. Recently, there have been reports on the importance of the above-mentioned parameter regime in related fields. The enhancement of fluorescent scattering of semiconductor quantum dots deposited upon photonic crystals slabs [24] and the enhancement (by few orders of magnitude) of the efficiencies of conventional surface spectroscopy by using tapered fibres [6] are two examples.

Experimentally, chemical and biological sensing have been demonstrated using absorption spectroscopy in D-shaped fibres [10] and MOFs [4, 5] and captured fluorescence-based sensing in tapered fibres [26], liquid-filled hollow-core MOFs [19, 7], side excited MOFs [2, 17], and double-clad and multi-core (liquid filled) MOF [3]. Here, we demonstrate experimentally both in-fibre excitation and fluorescence recapturing within a liquid-filled, solid-core, index-guiding

MOF [Fig. 1(b)] through its core guided modes. The experimental methods based on selective hole filling of MOFs (c.f. Ref. [19, 7] for example) or propagating modes of high index liquids (c.f. Ref. [3] for example) may be limiting in some applications. Thus, the use of high-index glass (SF57 here) solid-core MOFs allows access to high light-matter overlaps without necessitating selective hole filling while relying on solid-core modes even when the MOF holes are filled with high index liquid.

## 2. Theory

To develop the model, we assume that the propagating modes of an absorbing MOF are the same as nonabsorbing ones except that their powers decay with an attenuation factor of  $\gamma$  as they propagate. The excitation electromagnetic power in the  $j$ th mode at excitation frequency  $\omega_E$  can then be expressed as: [27]

$$P_{Ej}(z) = |a_{Ej}|^2 N_{Ej} \exp(-\gamma_{Ej}z); \quad N_{Ej} = \frac{1}{2} \text{Re} \left\{ \int_{A_\infty} (\mathbf{e}_{Ej} \times \mathbf{h}_{Ej}^*) \cdot \hat{\mathbf{z}} dA \right\} \quad (1)$$

$$\gamma_{Ej} = k \left( \frac{\epsilon_0}{\mu_0} \right)^{1/2} \frac{\int_{A_\infty} n_E n_E^i |\mathbf{e}_{Ej}|^2 dA}{N_{Ej}}, \quad (2)$$

where  $a_{Ej}$  is the expansion coefficient for mode  $j$ ,  $\mathbf{e}_{Ej}(x, y)$ ,  $\mathbf{h}_{Ej}(x, y)$ ,  $\beta_{Ej}$ , and  $\gamma_{Ej}$  are the  $j^{\text{th}}$  mode electric and magnetic field distributions, propagation constant and power decaying factor due to absorption, respectively. Here, we assume that  $\gamma_j$  represents all absorption mechanisms in the MOF, including absorption due to the Beer-Lambert law [28].

For an arbitrary filled MOF both  $n_E(x, y)$  and  $n_E^i(x, y)$  (real and imaginary parts of refractive indices) are functions of transverse coordinates and hence the piece-wise integral in Eq. (2) can be integrated over the glass and hole (filled) regions. Eq. (1) indicates that although the absorption of the excitation mode occurs in the filled region, through Beer-Lambert law, the peak intensity also reduces, keeping the shape of the mode and the hole power fraction constant.

Upon absorbing the excitation photons, the fluorescent species in the holes behave as sources and emit fluorescent photons in all directions. Similar to the excitation field, the emission of this new fluorescent source can in general be written [27] as the sum of forward, backward, and radiation modes of the non-absorbing MOF with the consideration of power decay due to loss at the fluorescence frequency. Based on the formalisms developed in Ref. [27, 29], we find the fluorescent power contribution to the  $j$ th forward mode of the MOF at the end of the filled region  $z = L$ , due to a small section  $\Delta z = z_2 - z_1$  [see Fig. 1(c)], and including its loss as:

$$dP_{Fj}(z') = \frac{\pi \exp[-\gamma_{Fj}(L - z')]}{4\omega_F \mu_0 n_F^H k_F N_{Fj}} \int_H \int_{z_1}^{z_2} |\mathbf{e}_{Fj}|^2 P_D(\mathbf{r}) dz'' dA. \quad (3)$$

Here,  $P_D(\mathbf{r})$  is the radiation power density of any sources within the MOF, which for the case considered here is due to the fluorescent emission of the filling material. The density of fluorescent emission at point  $\mathbf{r}$  depends on the absorption of excitation field from the beginning of the filled area up to the point  $\mathbf{r}$ , see Fig. 1(a). Using equations (1) and (2), assuming that the fluorescent power density is proportional to the density of excitation power loss due to Beer-Lambert law in the filled region (proportionality constant  $\xi$ ), and taking into account energy conservation, we have found  $P_D(\mathbf{r})$  as:

$$P_D(\mathbf{r}) = \frac{1}{2} \xi \alpha_B n_E^H (\epsilon_0 / \mu_0)^{1/2} |a_{Ej}|^2 \delta_{Ej}^H \text{Re}[(\mathbf{e}_{Ej} \times \mathbf{h}_{Ej}^*) \cdot \hat{\mathbf{z}}] \exp(-\gamma_{Ej}z''), \quad (4)$$

$$\delta_{Ej}^H = \int_H |\mathbf{e}_{Ej}|^2 dA / \int_H (\mathbf{e}_{Ej} \times \mathbf{h}_{Ej}^*) \cdot \hat{\mathbf{z}} dA. \quad (5)$$

Here,  $\alpha_B = \varepsilon_\lambda C$  is the absorption coefficient due to Beer-Lambert law [28], where  $\varepsilon$  is the molar extinction coefficient of the filling material,  $C$  is the molar concentration, and superscript  $H$  refers to hole regions. Substituting Eq. (4) into Eq. (3), and taking the integral over  $z''$  and the limit of  $z_1 \rightarrow z_2$ , we find

$$dP_{Fj}(z') = \frac{\pi \xi \alpha_B n_E^H (\varepsilon_0 / \mu_0)^{1/2} |a_{Ej}|^2 \delta_{Ej}^H}{8 \omega_F \mu_0 n_F^H k_F N_{Fj}} \exp[-\gamma_{Fj}(L-z')] \exp(-\gamma_{Ej}z') dz' \quad (6)$$

$$\times \int_H |\mathbf{e}_{Fj}|^2 \text{Re}[(\mathbf{e}_{Ej} \times \mathbf{h}_{Ej}^*) \cdot \hat{\mathbf{z}}] dA.$$

Integrating the fluorescent contributions of the elements from  $z = 0$  to  $z = L$ , the fluorescence capture fraction ( $FCF$ ) into the  $j$ th guided mode of the MOF can be expressed as

$$FCF = P_{Fj}(L)/P_{Ej}(0) = AB_j \frac{\exp(-\gamma_{Fj}L)}{(\gamma_{Ej} - \gamma_{Fj})} \{1 - \exp[(\gamma_{Fj} - \gamma_{Ej})L]\} \quad (7)$$

$$A = \frac{\xi \alpha_B \lambda^2}{8\pi (n_F^H)^2}; \quad B_j = n_F^H n_E^H \left( \frac{\varepsilon_0}{\mu_0} \right) \delta_{Ej}^H \frac{\int_H |\mathbf{e}_{Fj}|^2 \text{Re}[(\mathbf{e}_{Ej} \times \mathbf{h}_{Ej}^*) \cdot \hat{\mathbf{z}}] dA}{4N_{Fj}N_{Ej}}. \quad (8)$$

In this equation  $A$  is a constant coefficient and  $P_{Ej}(0)$  is the input excitation field power at the beginning of the filled part of the fibre, whose length is shown by  $L$  [see Fig. 1(a)]. It should be noted that throughout the text we only consider  $FCF$  into the fundamental guided mode propagating in the forward direction. To find  $FCF$  into backward propagating modes, different loss calculations should be included in the above formalism which is beyond the scope of this paper. Also, in derivation of Eq. (6) and (7), it is assumed that the whole fibre length is completely filled. For fibres that are only partially filled, the mode mismatch between the filled and unfilled sections should be considered for both excitation and fluorescence frequencies, which is beyond the scope of this paper.

### 3. Modelling results and discussion

The model developed in Section 2 is general and can be applied to any filled MOF with arbitrary cross section structure. Here, we consider an MOF [shown in Fig. 1(b)], which consists of a core surrounded by three large, non-circular, air holes creating a somewhat triangular core supported by three struts. This type of fibre geometry has been studied [30, 31] and is the simplest fibre geometry that can be fabricated giving rise to a well defined air-suspended core with large surrounding (fillable) air holes. To find the propagation constant and field distributions for the MOF, we solve the full vectorial form of Maxwell's equations since, for the subwavelength scales considered here, a scalar approximation gives inaccurate results [32].

We define the MOF core diameter,  $d$ , to be the diameter of a circle with area equal to that of the largest equilateral triangle that fits wholly within the substrate core region, shown as the solid circle in Fig. 1(b). For simplicity we consider an idealized MOF structure, which closely matches that of the SEM image [Fig. 1(b)]: the curvature of the core is approximated by the edges of the three dashed circles connected by the bases of three rectangles approximating the struts. To solve Maxwell's equations for this geometry we use the Finite Element Modelling (FEM) technique instantiated in the commercial FEM package COMSOL 3.2. The accuracy of the modal parameters depends heavily on the densities of the mesh in different regions, which have been set separately to achieve converged values. The parameters used for the simulations are; excitation wavelength  $\lambda_E = 532 \text{ nm}$ , fluorescence wavelength  $\lambda_F = 590 \text{ nm}$ , filling material: Rhodamine B dissolved in isopropanol, and five different glass materials [silica, lead silicates (LLF1, F2, and SF57), and bismuth] whose refractive indices are indicated on the figures.



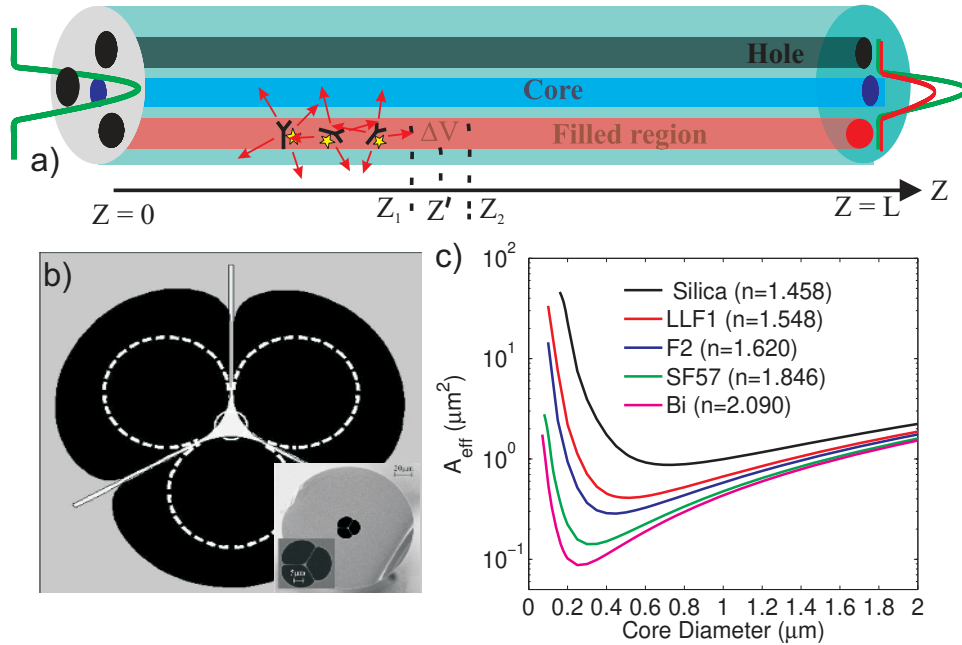


Fig. 1. Schematic of a filled MOF showing the parameters used in modelling (a) and the SEM image of the cross section of the MOF used for the modelling and experiment (b). Dashed circles in (b) show the idealized geometry used for modelling. The effective area of the fundamental mode for the geometry shown in (b) when the holes are filled with Rhodamine B in an isopropanol solution (c). The wavelength is  $590\text{ nm}$ , refractive index of isopropanol is 1.3774 and different substrate glasses are marked.

Simulation results of FCF as a function of fibre length for constant core diameter and concentration, as shown in Fig. 2(a), indicate that there is an optimum fibre length  $L_{opt} = \ln(\gamma_F/\gamma_E)/(\gamma_F - \gamma_E)$ , which leads to maximal FCF for any fibre geometry [see Eq. (7)]. For  $L < L_{opt}$  increasing the fibre length increases the absorption in the filled region via the Beer-Lambert law, and thus increases FCF. Beyond this optimum length, fibre attenuation dominates and the fluorescent power decays as  $\exp(-\gamma_F L)$ . Unsurprisingly, as the results in Fig. 2(a) show, the use of lower index glasses results in a higher FCF since the relatively low core-cladding index contrast leads to a higher light-matter overlap within the holes [this is also evident in the behaviour of  $A_{eff}$  in Fig. 1(c)].

Numerical simulations of the FCF also identify a less obvious and particularly interesting regime [see Fig. 2(b)]. For small core diameters ( $d < 0.8\ \mu\text{m}$ ), the FCF can be significantly enhanced by employing high index (soft) glasses. For example, the maximum FCF (FCF at fibre length  $L_{opt}$ ) for bismuth-oxide fibres at  $d \approx 0.18\ \mu\text{m}$ , is 2.2%, 10 times larger than the maximum FCF value for silica fibres (0.22%) at  $d \approx 0.52\ \mu\text{m}$ . Also, at the core size of  $d \approx 0.2\ \mu\text{m}$  the maximum FCF value for bismuth fibres is 2.1%, 88 times larger than that of silica fibres (0.024%). This is contradictory to the usual assumption that sensitivity is proportional to power fraction in the holes, since high index glasses result in lower power fraction in the holes compare to that of low index glasses at small core diameter. For example, at the core diameter of  $d \approx 0.2\ \mu\text{m}$  hole power fraction at excitation frequency (defined as  $\eta_E^H = n_E^H (\epsilon_0/\mu_0)^{1/2} (1/2N_{Ej}) \int_H |\mathbf{e}_{Ej}|^2 dA$ ) for silica and bismuth fibres are 0.97 and 0.43 re-

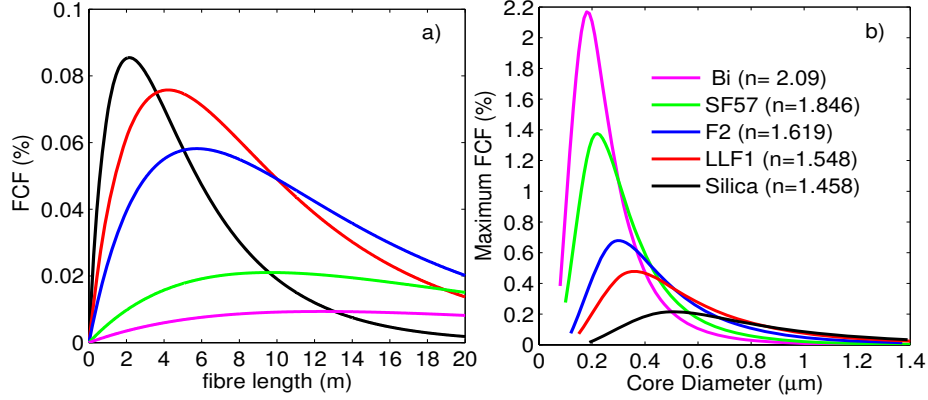


Fig. 2. Numerical results of the fluorescence capture fraction (FCF) as a function of fibre length (a) and core diameter (b) for different substrate glasses. Other parameters are; core diameter  $1.0 \mu\text{m}$ , in (a) and concentration  $5 \times 10^{-5} \text{ Mol}$  in (a) and (b). Maximum FCF in (b) corresponds to optimum fibre length.

spectively.

To understand this effect, we examine coefficient  $B_j$  in Eq. (8), which depends on the field distributions of the guided modes of the fibre and their overlap with the materials within the holes. We assume that the mode profiles of the excited and fluorescent fields are the same (i.e.,  $N_{Fj} = N_{Ej}$ ), which although not strictly true especially for filling materials such as quantum dots with large separation of absorbing and fluorescent wavelengths, can help provide physical insight. We rewrite coefficient  $B_j$  as  $B_j = NOI_j/A_{eff}$ , where;

$$NOI_j = n_E^H n_F^H \left( \frac{\epsilon_0}{\mu_0} \right)^{1/2} \frac{\delta_{Ej}^H \int_H |\mathbf{e}_j|^2 \text{Re}[(\mathbf{e}_j \times \mathbf{h}_j^*) \cdot \hat{\mathbf{z}}] dA}{\int_{A_\infty} |\text{Re}[(\mathbf{e}_j \times \mathbf{h}_j^*) \cdot \hat{\mathbf{z}}]|^2 dA}; \quad A_{eff} = \frac{(\int_{A_\infty} \text{Re}[(\mathbf{e}_j \times \mathbf{h}_j^*) \cdot \hat{\mathbf{z}}] dA)^2}{\int_{A_\infty} |\text{Re}[(\mathbf{e}_j \times \mathbf{h}_j^*) \cdot \hat{\mathbf{z}}]|^2 dA}$$

Here  $NOI_j$  is a normalized field-matter overlap integral, which approaches 1 when the core diameter becomes very small and most of the light is located outside the core [see Fig. 3(a)].  $A_{eff}$ , defined based on  $z$  component of the Poynting vector, is a generalised form of the usual definition of  $A_{eff}$  [33].

Inspecting  $A_{eff}$  [Fig. 1(c)] and  $NOI$  [Fig. 3(a)] at a core size of  $d = 0.2 \mu\text{m}$  for both silica and bismuth, reveals that the field-matter overlap  $NOI$  for silica is 3.6 times larger than that of bismuth. However the effective area,  $A_{eff}$ , of the propagating mode for bismuth is 230 times smaller than that of silica for this core diameter, resulting in higher intensity values for bismuth and thus a larger  $FCF$ . Examining the intensity profiles of the fundamental mode for these silica and bismuth fibres [Fig. (4)], clearly shows that while the mode is well expanded into the hole region in the case of silica, it is well confined within the core for bismuth and forms a high intensity, thin layer at the core-hole interface within the filled region.

These localised high intensity regions are formed due to the discontinuity of the electric field at the interface of two dielectric media, as recently reported in slab waveguides [25] and MOFs [22, 23]. The magnitude of the discontinuity is proportional to the ratio of the dielectric



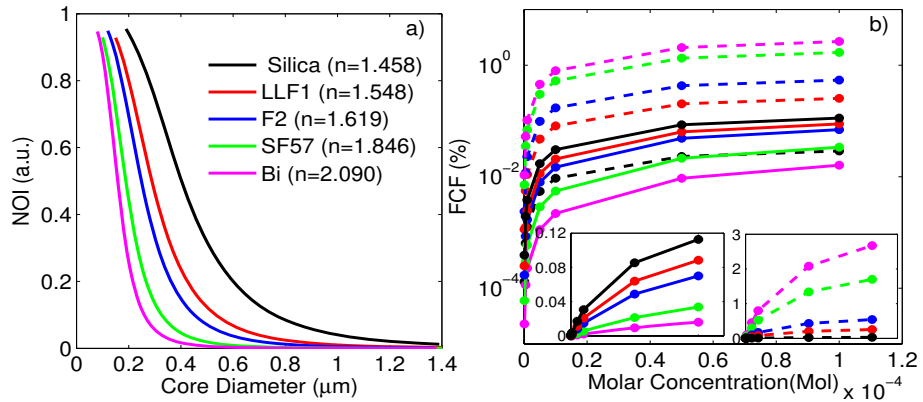


Fig. 3. Numerical results of Normalized Overlap Integral (NOI), defined in the text and calculated at the wavelength of  $590\text{ nm}$  (a), and the fluorescent capture fraction (FCF) as a function concentration (b). In (b) dashed and solid lines correspond to core diameters  $0.2\ \mu\text{m}$  and  $1.0\ \mu\text{m}$  respectively, and the insets show the linear scale plot of the main graph over the same concentration range.

constants of the two media and hence soft glasses with higher refractive indices result in higher intensities at the glass-hole interface.

The results presented here demonstrate for the first time that overlapping localised high intensity regions in the modal field with a fluorescent material is an effective way of enhancing the performance of a sensing fibre. This enhancement is significant not only for situations where the holes are filled with liquids, but is also particularly relevant for sensing configurations where samples are coated or functionalized onto the walls of the MOF. This enhanced *FCF* regime,

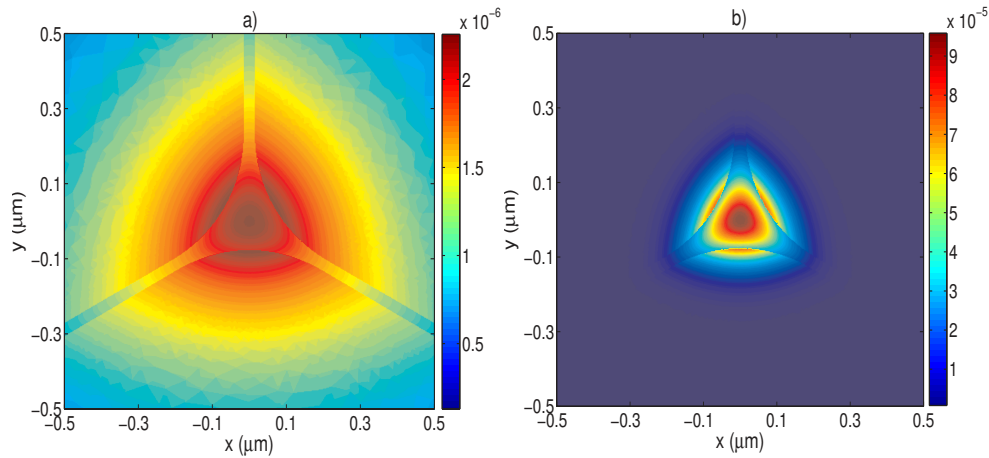


Fig. 4. Intensity distribution of the fundamental mode for Silica (a) and Bismuth (b). In both (a) and (b) core diameters are  $0.2\ \mu\text{m}$  and the wavelengths are  $590\text{ nm}$ . The mode is more confined in (b) and a thin layer of high intensity region is formed at the glass-hole interface.

achieved by the use of small core dimensions and high index glasses, also enhances the sensitivity of  $FCF$  to concentration variations in the material to be sensed. Using Eq. (7), the  $FCF$  can be approximated as  $[1 - \exp(-\varepsilon_\lambda C \eta_{E_j}^H)]$  as a function of concentration  $C$ , which simplifies to  $\varepsilon_\lambda C \eta_{E_j}^H$  in the limit of small concentration. This behaviour of  $FCF$  is confirmed as shown in Fig. 3(b) for two core diameters  $d = 1.0$  (solid lines) and  $d = 0.2 \mu m$  (dashed lines) and different glasses. However, the behaviour of  $FCF$  as a function of concentration  $C$  is opposite for small and large core regimes. Fig. 3(b) and its insets show that for large core diameters (e.g.,  $d = 1.0 \mu m$ ), silica has the largest asymptotic  $FCF$  value ( $\sim 0.11\%$ ) and  $FCF$  slope ( $\partial(FCF)/\partial C \sim 0.0027\%/ \mu M$ ). Whereas, for small core diameter ( $d = 0.2 \mu m$ ), bismuth has the largest corresponding  $FCF$  value of ( $\sim 2.67$ ) and  $FCF$  slope ( $\sim 0.069\%/ \mu M$ ). As a result, both  $FCF$  and its sensitivity to small variations in concentration are enhanced in the small core, high index glass regime. This regime is expected to be of particular practical benefit in allowing observations of captured fluorescence to occur at extremely low sample concentrations, allowing for the development of sensors with competitive detection limits.

#### 4. Experimental results

We have demonstrated experimentally in-fibre excitation and fluorescence recapturing within a filled, solid-core MOF. The holes of the MOF, shown in Fig. 1(b), were filled with Rhodamine B dissolved in isopropanol ( $n = 1.3774$ ) using capillary action. Using the experimental setup in Fig. 5(a), Fig. 5(b) shows the experimental measurements and theoretical prediction of the filling rate. We use the filling rate equation developed in Ref. [34] for a circular capillary tube, considering the following parameters for isopropanol, its interaction with glass, and the fibre geometry; density  $785 \text{ kgm}^{-3}$ , surface tension  $0.022 \text{ Nm}^{-1}$ , viscosity  $2.27 \times 10^{-3} \text{ Nsm}^{-2}$ , effective radius  $6.11 \times 10^{-6} \text{ m}$ , contact angle  $0^\circ$ , coefficient of slip  $0 \text{ m}$ , and external pressure  $0 \text{ Pa}$ . To find the effective radius  $r_{eff}$ , we have assumed that the holes of the MOF in Fig. 1(b) are circular with the same area as that of the real fibre. This, strictly speaking, is inaccurate because capillary forces are mainly a surface effect, which depends on the radius of curvature of the different corners in the geometry, and we believe that this assumption is the main reason for the discrepancy between the theoretical and experimental results in Fig. 5(b). For these experimental measurements the position of the liquid in the holes was recorded by observing the fluorescent emission at the liquid interface in the backward direction of the laser beam in Fig. 5(a).

The setup sketched in Fig. 5(c) was used to excite the Rhodamine B molecules filled into the holes of the fibre and measure the captured fluorescence emission. The outer surface of the fibre was coated with an index matching liquid, DAG, to strip any fluorescent emission coupled to the cladding modes. This ensured that all the measured fluorescence had been captured by the relatively low-loss, core-guided modes of the fibre, as assumed by the theoretical model. The absorption and low concentration fluorescent peaks of Rhodamine B are at  $540 \text{ nm}$  [35] and  $570 \text{ nm}$ , respectively. The MOF used in the experiment has a core diameter of  $d = 1.8 \mu m$ , core material of SF57 and its cross section is shown in Fig. 1(b). A CW laser at  $532 \text{ nm}$  was coupled into the MOF using an aspheric lens of  $f = 2.75 \text{ mm}$  and  $NA = 0.65$  and maximum coupling efficiency of around 19% was measured. The loss of the unfilled fibre at  $532 \text{ nm}$  is  $5.5 \pm 0.5 \text{ dB/m}$ , using the standard cutback method. At the output, we used a long pass filter to exclude the excitation frequency components. The fluorescence emission is then coupled into a single mode (SM) fibre, which was connected to an optical spectrum analyser for spectrum measurement.

The experimental results are presented in Fig. 5(d) clearly showing the expected fluorescence and also significant decay of the measured fluorescence over 960 seconds exposure to the excitation field. The decay in fluorescence was due to photobleaching, the photo-induced

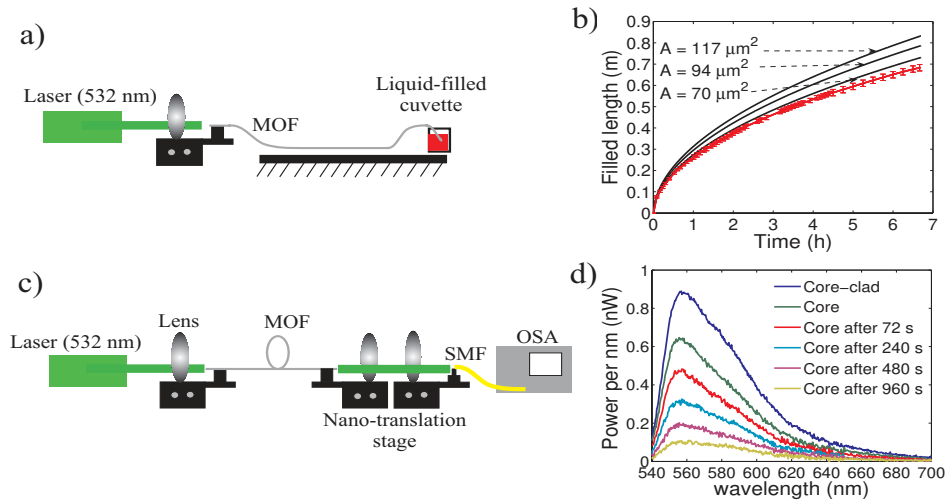


Fig. 5. Experimental setup for filling an MOF (a). The experimental and theoretical predictions of filling time as a function of filled length (b). For the theoretical predictions it is assumed that the holes of the fibre are circles whose area are 1, 0.8, or 0.6 times of that of the real fibre in Fig. 1 (117, 94, and  $70 \mu\text{m}^2$  respectively). Experimental set up (c) and results (d) for capturing the fluorescent emission by the core of the MOF.

destruction of the fluorophore [28]. While partial recovery is possible, photobleaching has the potential to be problematic. For example, to measure FCF as a function of fibre length we fill a fibre and use a cut-back method to measure FCF at different lengths. However due to photobleaching effect and the time that takes to cleave, align, and couple the fluorescence beam into the OSA, it is very difficult to measure FCF as a function of length for Rhodamine B. A promising alternative is to replace organic dyes with quantum dots, which experience negligible photobleaching and have already found use in sensing such as for biological and medical applications [26, 36].

## 5. Discussion and conclusion

To the best of our knowledge this is the first time that a general model of both in-fibre excitation and fluorescence recapturing of filled solid-core MOFs by their core guided modes has been developed and demonstrated experimentally. An expression for the efficiency of fluorescence capture fraction has been developed for an arbitrary MOF based on vectorial solutions of Maxwell's equations and considering the modal behaviour of the MOF for distinct excitation and fluorescence frequencies. We have predicted between one to two orders of magnitude improvement in the fluorescence capture fraction (FCF) of MOFs with high index substrate glasses and small core diameters in comparison with those with low index glasses. This parameter regime results in localised, high intensity electromagnetic fields at the interface of the glass and hole regions, making it an ideal regime for thin layer sensing where chemical or biological substances are coated onto the interface.

The fluorescence capture fraction is normalised to the input power in the fibre and hence, although the small core parameter regime degrades the coupling efficiency, higher incident power (below the damage threshold of the glass) can be used to attain certain power in the fibre. Additionally, by using advanced coupling techniques such as tapers or high numerical aperture buffer fibres [37], one should be able to minimise the coupling loss into small core

fibres.

Our recent progress in fabricating soft glass MOFs with small cores, and the evidence that with careful fabrication processes fibre loss of order of  $< 0.5 \text{ dB/m}$  can be achieved [38], provide an attractive new route towards the development of highly-sensitive fluorescence sensors.

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