

Research Article

Evaluation of Corrosion Inhibition Efficiency of Aluminum Alloy 2024 by Diaminostilbene and Azobenzene Schiff Bases in 1M Hydrochloric Acid

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The Schiff base compounds N,N'-bis(salicylidine)-4,4'-diaminostilbene(SDS) and N,N'-bis(salicylidine)-4,4'-diamino azobenzene(SDA) were synthesized, and their molecular structure was determined by FT-IR and 1H NMR. The corrosion inhibitions of Schiff base compounds on aluminum alloy 2024 in 1 M hydrochloric acid were evaluated by potentiodynamic polarization, impedance techniques, weight loss method, and scanning electron microscopic technique. The potentiodynamic polarization (PDP) studies revealed that SDS and SDA compounds acted predominantly as cathodic inhibitors. The electrochemical impedance spectroscopic (EIS) parameters confirmed the adsorption of SDS and SDA molecules over the surface of aluminum alloy 2024 alloy by forming an inhibitive layer. The weight loss studies showed that the inhibition efficiency of these compounds increases directly with concentration and decreases with an increase in solution temperature and immersion time. The thermodynamic parameters were calculated to investigate the mechanism of corrosion inhibition. The SDA was found to be more effective than SDS and followed the Langmuir adsorption isotherm model. The scanning electron microscopy (SEM) results revealed that the deterioration of the alloy surface is minimal in the presence of an inhibitor. Both Schiff base molecules exhibited superior corrosion inhibition for aluminum alloy 2024 alloy in HCl medium.

1. Introduction

Aluminum alloy 2024 is a high strength alloy that finds its utility in products that require a high strength-to-density ratio and good resistance to fatigue. Because of its very good durability and tolerance towards damage, toughness, fractures, fatigue, crack propagation, and low cost to manufacture, it is often being used in automotive, transportation, and aerospace industries [1, 2]. Its high strength is due to the presence of copper, magnesium, and manganese. The higher percentage (3.8–4.9%) of copper in 2024 aluminum alloy greatly reduces its resistance to corrosion [3]. AA2024 alloy is very much susceptible to localized corrosive attack since the intermetallic compounds at the boundary get segregated. The localized pitting corrosion starts where the most predominant intermetallic inclusions exist. Al2MgCu is one such inclusion and reveals cathodic potential concerning the alloy matrix. Magnesium and aluminum get chemically and electrochemically dissolved from Al2MgCu particles at the beginning of corrosion which leads to dealloying enriching them with copper. The intermetallic precipitates become active anodes and cathodes, resulting in deep dealloying of the matrix [4].

The corrosion inhibition by triazole and thiazole derivatives against aluminum-based copper-containing alloys was studied earlier [5, 6]. 2-Mercaptobenzothiazole, 8-hydroxyquinoline, and benzotriazole were used as corrosion inhibitors on AA 2024-T3 [7]. 2-(4-(Diethylamino)-2hydroxybenzyledene) hydrazinecarboamide was also applied as corrosion inhibitor for AA2024-T3 alloy in acid solution [8]. Organic inhibitors were also studied as inhibitors of corrosion for aluminum and its alloys in both acidic and basic solutions [9]. *Thymus algeriensis* extract was experimented as corrosion inhibitor for AA2024 alloy in an acidic medium [10]. Many Schiff bases were also applied as corrosion inhibitors for aluminum alloys in acidic solution [11–14]. In general, various organic and inorganic inhibitors were utilized by past researchers, as corrosion inhibitors for AA2024 alloy in acidic and neutral chloride solutions [15–27], but no significant study has been conducted on the evaluation of corrosion inhibition of AA2024 alloys in acidic media by organic inhibitors.

The AA2024 alloy was found to exhibit a high degree of susceptibility to severe corrosion in acid media. The electrochemical etching of aluminum and its alloys usually involves the application of HCl. As the dissolution rate of AA2024 alloy in HCl solution is very high, it is necessary to avoid the dissolution of alloy by adding inhibitors as additives in the etching medium. Organic inhibitors have the ability to control the corrosion in metal alloys by forming an insoluble layer on the surface of the alloys. In general, organic inhibitors get adsorbed on metal alloy surfaces by chemisorption by forming coordination bonds with the atoms or ions by physisorption through electrostatic interaction between the alloy and inhibitors. Electronegative groups found in inhibitor molecules such as S, P, O, and N atoms assist in strong bond formation on the metal surface [28].

However, investigation of inhibitive action of N,N' -bis(salicylidine)-4,4'-diaminostilbene (SDS) and N,N' -bis(salicylidine)-4,4-diamino azobenzene(SDA) in acid medium on the corrosion behavior of AA2024 alloy has not yet been explored. Thus, we present this work targeting the evaluation of corrosion inhibitive effect of N,N'-bis(sa-licylidine)-4,4'-diaminostilbene (SDS) and N,N'-bis(salicy-lidine)-4,4'-diaminoazobenzene (SDA) on AA2024 alloy in presence of 1 M HCl using PDP, EIS, and weight loss methods.

2. Materials and Methods

2.1. Test Specimens. Aluminum alloy 2024 ingots were procured from M/s. Fenfee Metals, Bengaluru, India. The aluminum alloy 2024 ingots were melted and moulded into 26 mm diameter $\times 170$ mm length cylindrical rods using the gravity die casting method at Metallurgical Lab, Chemistry Department, HKBK College of Engineering, Bengaluru, India. The elemental composition of alloy was analyzed using Optical Emission Spectrometry (as per IS: 7658) at CMTI Lab, Bengaluru. The composition was found to be within the range of ASTM specifications as shown in Table 1.

The test specimen for weight loss studies and the cylindrical rods were cut into circular discs using the abrasive cutting wheel. Each circular disc measuring a height of 2 mm and a diameter of 24 mm and having a 2 mm mounting hole at the center was prepared. The test specimens for electrochemical studies were prepared by moulding the

TABLE 1: The elemental composition of AA 2024 alloy.

Element	Wt.%
Al	90.7-94.7
Cu	3.8-4.9
Mg	1.2–1.8
Si	Maximum 0.5
Fe	Maximum 0.5
Mn	0.3-0.9
Ti	Maximum 0.15
Cr	Maximum 0.1
Zn	Maximum 0.25
Others	Maximum 0.15

cylindrical rods of 20 mm length in cold setting acrylic resin. In such a way, leaving the alloy surface of 1.0 cm^2 area is exposed on one end, and another end is connected to copper wire for electrical contact. Before each experimental trial, the electrodes were polished with emery papers of various grades ranging from 6 to 12 hundred for a smooth surface. The specimens were washed with double distilled water and acetone for degreasing and finally drying at $353 \pm 1 \text{ K}$ for over half-hour in a thermostat electric oven. The alloy specimens were placed in a moisture-free desiccator.

2.2. Test Solution—Corrosion Medium. The corrosive medium 1 M HCl was prepared by diluting analytical grade 37% acid (SDFCL) in double-distilled water. The stock solution of 1000 ppm Schiff bases was prepared in 1 M HCl, and 25, 50, 75, and 100 ppm solutions were prepared by further dilution in 1 M HCl.

2.3. Synthesis and Characterization of Schiff Base Inhibitors. The Schiff base compounds under investigation were synthesized in accordance with the method used by Hossein Naeimi and Arash Heidarnezhad [29]. The diaminostilbene and azobenzene Schiff bases were prepared by condensing the corresponding aromatic aldehydes with 4,4'-diaminostilbene and 4,4'-diaminoazobenzene as shown in Scheme 1. The structural elucidation of the synthesized compound was confirmed by FT-IR and ¹H NMR spectral studies along with its melting points (Table 2).

3. Experimental Methods

3.1. Weight Loss Measurement. The weight loss experiments were performed as per ASTM [30] under different parameters like duration of a specimen immersed, the solution temperature, and the concentration of inhibitor. Before each experiment, the following cleaning actions were performed:

- (i) Electrodes were abraded using emery papers (grades 600, 800, and 1200)
- (ii) Electrodes were washed with double-distilled water and degreased using acetone



N,N'-bis (salicylidine)-4,4' –diaminostilbene(SDS)



N,N'-bis (salicylidine) -4,4' -diaminoazobenzene(SDA)

SCHEME 1: Synthesis of diaminostilbene and azobenzene Schiff bases.

TABLE 2: ((FT-IR), ¹H NMR), molecular formula, and melting point data of SDS and SDA.

No.	Diamines	IR and ¹ H NMR data
(1)	N,N'-bis(Salicylidine)-4,4' -diaminostilbene (SDS) Molecular formula: C ₂₈ H ₂₂ N ₂ O ₂ Molecular mass: 418.28 g/Mol Melting point: 240-244°C	IR/ ν (KBr) cm ⁻¹ : 3320-3500, 2940-2990, 1605, 1150, 1130 ¹ H NMR (DMSO-d6, 500 MHz) δ (ppm): 13 (s, 2H), 8.5 (s, 2H), 7-7.9 (m, 16H), and 6.8 (d, 2H). MS (EI, m/z): 418 (M+, 7) and 392 (11)
(2)	N,N'-bis(Salicylidine)-4,4' -diaminoazobenzene (SDA) Molecular formula: C ₂₆ H ₂₀ N ₄ O ₂ Molecular mass: 420.26 g/Mol Melting point: >300°C	IR/ ν (KBr) cm ⁻¹ : 3310-3450, 3030, 1609, 1170, and 1150 ¹ H NMR (DMSO-d6, 500 MHz), δ (ppm): 13.31 (s, 2H), 8.73 (s, 2H), 8.07 (d, 4H), 7.42(m, 4H), 7.36 (m, 2H), 7.27 (q, 4H), and 6.22 (t, 2H). MS (EI, m/z): 420 (M+, 40) and 392 (18).



FIGURE 1: Tafel plots for (a) N,N'-bis(salicylidine)-4,4'-diaminostilbene (SDS). (b) N,N'-bis(Salicylidine)-4,4'-diaminoazobenzene (SDA).

TABLE 3: PDP parameters of N,N'-bis(salicylidine)-4,4'-diaminostilbene (SDS) and N,N'-bis(salicylidine)-4,4'-diaminoazobenzene(SDA).

Tafel data							
Inhibitor	Concentration (ppm)	OCP (mV)	$-E_{\rm corr}$ (mV vs. SCE)	$i_{\rm corr} ({\rm mA}{\rm cm}^{-2})$	$b_{\rm c} \ ({\rm mV} \ {\rm dec}^{-1})$	$b_{\rm a} \ ({\rm mV} \ {\rm dec}^{-1})$	$\mu_{\rm p}\%$
	0	-796.5	741	4.091	179	65	_
	25	-805.8	769	1.105	201	68	73
SDS	50	-810.6	778	1.023	209	69	75
	75	-815.9	781	0.859	212	72	79
	100	-821.8	790	0.736	215	74	82
SDA	25	-810.9	780	0.941	211	76	77
	50	-817.3	785	0.859	214	78	79
	75	-818.8	788	0.736	216	82	82
	100	-823.2	792	0.532	221	89	87



FIGURE 2: Nyquist plots. (a) N,N'-bis(salicylidine)-4,4'-diaminostilbene (SDS). (b) N,N'-bis(Salicylidine)-4,4'-diaminoazobenzene (SDA).

- (iii) The electrodes were dried at 80° C for a half-hour in a thermostatic electric oven
- (iv) The electrodes were preserved in a moisture-free desiccator

Prior to each weight loss experimental trial, the test specimens were weighed in Shimadzu AY220 electronic analytical weighing balance and the weight of the test specimens were recorded. After each experimental trial, the corroded test specimens were washed under running water and placed in 70% HNO_3 for 180 sec. The corroded test specimens were gently rubbed with a soft nylon brush for removing the corrosion product from the alloy surface. The corroded specimens were dried at 353 K temperature and were weighed up to 0.1 mg accuracy and the loss in weight was noted down. Weight loss experiments were carried out in a constant temperature water bath fitted with an auto thermostat.



FIGURE 3: Bode plots. (a) N,N'-bis(salicylidine)-4,4'-diaminostilbene (SDS). (b) N,N'-bis(Salicylidine)-4,4'-diaminoazobenzene (SDA).

The weight loss experiment was carried out by immersing the test specimens' in

- (i) 100 mL 1 M HCl solution
- (ii) 100 mL 1 M HCl solution with various concentrations of Schiff bases-25 ppm through 100 ppm with increments of twenty-five ppm
- (iii) At the temperatures ranging from 303 through 333 ± 1 K with increments of 10 K
- (iv) At different exposure periods ranging from 2 through 8 hours with increments of two hours

The weight loss experiments were conducted in triplicate, and the average weight loss was calculated.



FIGURE 4: Equivalent circuit model used to fit EIS experimental data for AA2024 in 1 M HCl (a) Nyquist and (b) Bode plot.

3.2. Evaluation of Corrosion Rate (CR). The average corrosion rate (CR) was evaluated as per ASTM equation (1) in mmpy, where K is a constant (8.76×10^4), W is the weight loss in grams, D is the density in g/cm³, A is the surface area of test specimen in cm², and T is the time of exposure in hours.

$$CR = \frac{W \times K}{TAD}.$$
 (1)

3.3. Calculation of Inhibition Efficiency. Inhibition efficiency percentage ($\mu_{WL\%}$) and degree of surface coverage (θ) were calculated using the weight loss of uninhibited W_0 and W_i inhibited test specimen using equations (2) and (3), respec-

tively,

$$\mu_{\rm WL}\% = \frac{W_0 - W_i}{W_0} \times 100, \tag{2}$$

$$\theta = \frac{W_0 - W_i}{W_0}.$$
(3)

3.4. Electrochemical Measurements. The electrochemical measurements were performed in a 3-electrode pyrex glass electrochemical cell using CHI660c electrochemical workstation. The aluminum alloy 2024 test specimen with a surface area of 1 cm^2 , a saturated calomel electrode (SCE) attached to a Luggin capillary to keep the IR drop minimized, and platinum electrode has been used as working electrode, reference,

TABLE 4: EIS data using SDS and SDA as inhibitors.

Electroche	emical impedance	e spectroscopy	data							
Inhibitor	Concentration (ppm)	$R_{\rm S}~(\Omega~{\rm cm}^{-2})$	$R_{\rm ct} \; (\Omega \; {\rm cm}^{-2})$	$R_L \ (\Omega \ {\rm cm}^{-2})$	$L (\text{Hcm}^{-2})$	ω (Hz)	CPE (μ F cm ⁻²)	Ν	Cdl (μ F cm ⁻²)	$\mu_{\rm Rct}$ %
	0	1.14	09	1.41	26.1	3.7	119	0.8805	67	_
	25	1.19	34.0	22.1	113	27	34	0.9138	24.1	73.5
SDS	50	1.29	38.5	30.2	141	35	29	0.9216	22.3	76.6
	75	1.34	45.2	45.3	167	46	25	0.9281	19.1	80.1
	100	1.38	52.9	52.4	291	51	22	0.9299	14.3	83.0
	25	1.29	40.5	11.3	161	44	29	0.9251	25.7	77.8
SDA	50	1.32	44.6	32.9	227	55	17	0.9273	13.5	79.8
	75	1.38	52.0	39.3	281	62	15	0.9288	12.1	82.8
	100	1.43	66.7	40.3	301	71	13	0.9332	9.8	86.5

and auxiliary electrodes, respectively. The surface area of the test specimen was cleaned as explained under material and methods. The open circuit potential, potentiodynamic polarization (PDP), and electrochemical impedance spectroscopy (EIS) experiments were carried out in triplicates, and average results were considered for analysis.

3.5. Potentiodynamic Polarization (PDP) Measurements. The AA2024 test specimens were polarized in the potential range between -250 and +250 mV with respect to OCP at 1 mV/s scan rate to obtain Tafel plots. The corrosion current densities I°_{corr} and I_{corr} in the absence and presence of inhibitor, respectively, were obtained by extrapolating linear segments of anodic and cathodic curves from plots to corrosion potential (E_{corr}). The efficiency of inhibition was calculated using equation (4),

$$\mu_P \% = \frac{I_{\text{corr}}^\circ - I_{\text{corr}}}{I_{\text{corr}}^\circ} \times 100.$$
(4)

3.6. Electrochemical Impedance Spectroscopy (EIS) Measurements. The electrochemical impedance spectroscopy experiments were conducted with 10 mV peak-peak amplitude and AC signal at OCP, and frequency varied between 100 kHz and 1.0 MHz. The impedance data was analyzed using Nyquist plot and Bode plots. Data fitting for obtaining an equivalent circuit model was performed using Echem software ZSimpWin v3.21.

The inhibition efficiency μ_{Rct} % was calculated using charge transfer resistance data in inhibited R^{i}_{ct} and uninhibited R^{o}_{ct} solutions, respectively, using equation (5).

$$\mu_{Rct}\% = \frac{R_{ct}^{i} - R_{ct}^{\circ}}{R_{ct}^{i}} \ge 100$$
(5)

3.7. Evaluation of Thermodynamic and Adsorption Isotherm *Parameters*. The thermodynamic activation parameters were obtained using weight loss experimental data.

3.8. Activation Parameters. Activation parameters like activation energy (E_a) , enthalpy of activation (ΔH^*) , and entropy of activation (ΔS^*) for the dissolution of metal in the acid medium in the presence of various concentrations

of inhibitor were calculated from the Arrhenius equation (6) and transition state equations (7).

$$\log (CR) = \frac{-E_a}{2.303RT} + A,$$
 (6)

$$CR = \frac{RT}{Nh} \exp\left(\frac{\Delta S^*}{R}\right) \exp\left(-\frac{\Delta H^*}{RT}\right).$$
 (7)

3.9. Adsorption Isotherms. The surface coverage (θ) values derived from weight loss data for different concentrations of Schiff bases were tested graphically to best fit one of the adsorption isotherms from among Langmuir, Freundlich, and Temkin isotherms. With $C_{(inh)}$ indicating inhibitor concentration, Langmuir's isotherm ($C_{(inh)}/\theta$ vs. $C_{(inh)}$) with a linear regression coefficient value nearest to one was found to be the fitting best. Langmuir adsorption isotherm is expressed as follows equation (8) with $K_{(ads)}$ representing equilibrium constant for adsorption and desorption:

$$\frac{C_{\rm inh}}{\theta} = \frac{1}{K_{\rm (ads)}} + C_{\rm inh}.$$
 (8)

3.9.1. Standard Free Energy of Adsorption (ΔG°_{ads}). The ΔG°_{ads} was calculated using $K_{(ads)}$ that was derived from straight-line intercepts on the $C_{(inh)/\theta}$ axis and equation (9).

$$\Delta G_{\rm ads}^{\circ} = -RT\ln(55.5 \, K_{\rm ads}). \tag{9}$$

3.10. Standard Enthalpy and Entropy of Adsorption (ΔH°_{ads}) and ΔS°_{ads}). The standard enthalpy of adsorption (ΔH°_{ads}) and standard entropy of adsorption (ΔS°_{ads}) were derived by plotting a graph ln $K_{(ads)}$ vs. 1/*T* and by measuring straight lines with a slope equal to ($-\Delta H^{\circ}_{ads}/R$) and intercept ($\Delta S^{\circ}_{ads}/R + \ln 1/55.5$).

3.11. Scanning Electron Microscopic (SEM) Studies. The surface morphology of corroded and inhibited aluminum alloy 2024 test specimens was analyzed by JSM-840A-JEOL model scanning electron microscope. Following cases of Al alloy specimen were considered for the studies:

International Journal of Corrosion

Weight loss data						
Inhibitor	Temperature	Concentration	Weight loss	C.R.	Илля %	Θ
	(K)	(ppm)	(mg)	(mmpy)	L WL	
		Blank	128.7	197.0		_
		25	36.0	55.1	72	0.72
	303	50	30.9	47.3	76	0.76
		75	25.7	39.4	80	0.80
		100	21.9	33.5	83	0.80
		Blank	288.0	440.8	—	_
		25	99.6	152.5	63	0.63
	313	50	90.8	139.0	65	0.65
		75	79.1	121.1	68	0.68
NN' kig(Calingliding) 4.4^{\prime} diamin actillance (CDC)		100	67.4	103.1	72	0.73
IN,IN -DIS(Salicylidine)-4,4- diaminostilbene (SDS)		Blank	578.2	884.9	—	_
		25	242.8	371.6	56	0.56
	323	50	221.7	339.2	59	0.59
		75	203.7	311.7	64	0.64
		100	185.7	284.2	67	0.67
	333	Blank	989.7	1514.6	_	_
		25	445.4	681.6	51	0.51
		50	425.6	651.3	54	0.54
		75	386.0	590.7	57	0.57
		100	356.3	545.3	62	0.62
		25	33.5	51.2	75	0.75
	303	50	25.7	39.4	80	0.80
		75	21.9	33.5	83	0.83
		100	18.0	27.6	86	0.86
		25	87.9	134.5	68	0.68
		50	76.2	116.6	70	0.70
	313	75	67.4	103.1	73	0.73
N N'-bis(Salicylidine)-4 4'-diamino azobenzene		100	58.6	89.7	76	0.76
(SDA)		25	219.7	336.3	59	0.59
		50	202.4	309.7	64	0.64
	323	75	185.0	283.2	67	0.67
		100	150.3	230.1	72	0.72
		25	425.6	651.3	54	0.54
		50	376.1	575.6	57	0.57
	333	75	346.4	530.1	61	0.61
		100	306.8	469.5	66	0.66

TABLE 5: Weight loss data related to SDS and SDA.

- (i) Polished samples
- (ii) Samples corroded in 1 M HCl with no inhibitors
- (iii) Samples inhibited in 1 M HCl with 100 ppm of Schiff bases

4. Results and Discussion

4.1. Potentiodynamic Polarization Studies. The potentiodynamic polarization studies of AA2024 alloy were conducted in 1 M HCl at 303 ± 1 K in the absence and presence of SDS and SDA as inhibitors, with 25, 50, 75, and 100 ppm concentrations. The experimental data were used to plot the Tafel plots as depicted in Figures 1(a) and 1(b) for SDS and SDA, respectively. The corresponding PDP parameters, corrosion potential ($E_{\rm corr}$), corrosion current density ($I_{\rm corr}$), and cathodic and anodic Tafel slopes (b_a and b_c) derived from the Tafel plots are presented in Table 3. For each given concentration of SDS and SDA, the inhibition efficiency (μ p %) of inhibitors was derived using equation (4).

The PDP results showed the curves being shifted to the lower current density side when inhibitor concentration



FIGURE 5: (a) Influence of inhibitor concentration, (b) influence of immersion time, and (c) influence of temperature on the % IE.



FIGURE 6: Arrhenius plot related to (a) SDS and (b) SDA.

was increased and the corrosion potential (E_{corr}) values to not having shown any major shift(within 85 mV with respect to E_{corr} of blank solution). This confirms the characteristics of mixed type inhibitors [31, 32].

The PDP results also show the cathodic curves which have been more polarized. This suggests that the cathodic reduction rate was retarded, and diamines were the influencers of cathodic reactions compared to anodic and hence suggest the characteristics of cathodic inhibitors [33, 34] which could have occurred because of protonated diamine molecules getting adsorbed on cathodic and anodic sites. The inhibition efficiency of SDA was found to be the most efficient than SDS at different concentrations.

4.2. Electrochemical Impedance Spectroscopy Studies. The Nyquist and Bode plots for the corrosion inhibition of

Thermodynamic data				
Schiff base inhibitors	Concentration (ppm)	<i>Ea</i> (kJmol ⁻¹)	ΔH^* (kJmol ⁻¹)	$\Delta S^* $ (J mol ⁻¹ K ⁻¹)
	Blank	58.12	53.09	-25.15
	25	73.64	72.37	25.23
N,N'-bis(Salicylidine)-4,4-'diaminostilbene (SDS)	50	75.22	73.89	29.42
	75	77.53	74.94	30.15
	100	81.17	75.57	32.54
	25	75.87	72.74	27.71
NNI his(Calindidina) 4.4 diamina anahangana	50	79.78	74.28	29.94
in,in -bis(Sancyndine)-4,4 diamino azobenzene	75	82.19	75.63	30.74
	100	85.48	77.31	37.56

TABLE 6: Thermodynamic parameters related to SDS and SDA.

AA2024 alloy in 1 M HCl without inhibitors and with different concentrations of SDS and SDA are presented in Figures 2(a), 2(b), 3(a), and 3(b), respectively. The Nyquist diagrams showed semicircular curves which reveal that the corrosion process is under charge transfer control. In general, the shape of the Nyquist plots is with a large capacitive loop at higher frequencies and an inductive loop at lower frequencies. Some researchers have also reported similar EIS plots [35–41] for aluminum alloys in hydrochloric acid. The depressed semicircular curves having center under real axis behavior termed as dispersing effect [42, 43] are the property of solid electrode.

An equivalent circuit model fitted for simulating the measured impedance data of AA2024 are depicted with standard errors in Figures 4(a) and 4(b) for Nyquist and Bode plot, respectively. The equivalent circuit model which contains R_s , R_{ct} , R_L , and L correspondingly represent solution resistance, charge transfer resistance, and inductance resistance and inductance L, respectively. The circuit model also contains the constant phase element CPE (Q).

The constant phase element (CPE) in the equivalent circuit can be substituted by the respective capacitor of C_{dl} in order to give a more accurate fit, which is a frequencydependent element and related to surface roughness. The impedance function of a CPE is calculated using equation (10) [44],

$$Z_{\rm CPE} = \frac{1}{\left(Y_0 j\omega\right)^n},\tag{10}$$

where Y_0 is the CPE magnitude. *j* is the CPE exponent (frequency-independent). ω is the angular frequency for which -Z'' reaches its maximum value. *n* is the value that depends on surface morphology having range: $-1 \le n \le 1$. Y_0 and *n* are found from the equation shown by Mansfield et al. [45].

The double-layer capacitance $(C_{\rm dl})$ is derived using angular frequency ω_{max} with imaginary component of impedance reaching a maximum at $Z_{\rm imax}$ by equation (11) [46]:

$$C_{\rm dl} = Y_0(\omega_{\rm max})^{n-1}.$$
 (11)

The EIS spectra of uninhibited and inhibited aluminum alloy AA2024 are depressed. This kind of deviation is referred to as frequency dispersion and has been attributed to nonhomogeneous solid alloy surfaces. Assumption of a simple $(R_{\rm ct} - C_{\rm dl})$ is usually a poor approximation for systems showing depressed semicircle behavior due to nonideal capacitive behavior of solid electrodes surface [47]. The CPE [48], $C_{\rm dl}$, and $\mu R_{\rm ct}$ % were evaluated by equations (5), (10), and (11). The electrochemical impedance parameters R_s , $R_{\rm ct}$, R_L , L, ω , CPE, n, and the inhibition efficiency evaluated by R_{ct} and $C_{\rm dl}$ values are tabulated in Table 4.

The charge transfer resistance (R_{ct}) of the inhibited alloy increased and double-layer capacitance (C_{dl}) decreased with an increase in inhibitor concentration. This is because the adsorption of inhibitor molecules on to the alloy which forms physical barrier and block either or both anodic and cathodic reactions thereby reducing the alloy reactivity. This is maybe due to changes in the electric double layer at the solution and alloy surface interface. The reduction in double-layer capacitance (C_{dl}) can be caused by a decrease in the local dielectric constant and or an increase in the thickness of the electrical double layer. This suggests that the inhibitor molecules inhibit the aluminum alloy surface by adsorption at the alloy acid interface.

It is perceptible that the inhibition efficiency increases with an increase in the concentration of inhibitor, which is in concurrence with the potentiodynamic polarization outcome. The charge transfer resistance (R_{ct}) and double-layer capacitance (C_{dl}) values indicate that the AA2024 alloy surface is more efficiently inhibited by SDA than SDS.

4.3. Weight Loss Studies. The weight loss studies of AA2024 alloy in 1 M HCl in the absence and presence of SDS and SDA were carried out at different inhibitor concentrations, immersion time, and solution temperature. The weight loss experimental parameters weight loss (Δ w), percentage of inhibition efficiency (μ_{WL} %), corrosion rate (C.R.) in mmpy,



FIGURE 7: Thermodynamic plot related to (a) SDS and (b) SDA.

and degree of surface coverage (θ) calculated for 2 hours exposure time and at different temperatures are presented in Table 5 for SDS and SDA correspondingly.

4.3.1. Effect of Inhibitor Concentration. The study revealed greater inhibition efficiency at higher inhibitor concentrations, which could be attributed to the formation of persistent film and covering a large surface area of the alloy by

the inhibitor molecules. The maximum inhibition was achieved at 100 ppm and beyond 100 ppm concentration, and the inhibition efficiency was found to remain almost constant as shown in Figure 5(a).

4.3.2. Effect of Immersion Time. The inhibition efficiency of SDA and SDS decreased with an increase in immersion time from two to eight hours at 100 ppm concentration is shown



FIGURE 8: (a) Langmuir isotherm plot for SDS and SDA. (b) Heat of adsorption isotherm plot for SDS and SDA.

in Figure 5(b). This suggests desorption of inhibitor molecule over the surface of the alloy due to longer exposure time, resulting in the formation of lesser persistent film and an increase in cathodic reaction rate [49].

4.3.3. Effect of Temperature. The inhibition efficiency of the SDS and SDA was found to decrease with an increase in temperature between 303 and 333 ± 1 K is shown in Figure 5(c). This could be attributed to inhibitor molecules

getting desorbed from the alloy surface and aluminum alloy getting dissolved at faster rates with a rise in temperatures.

4.4. Thermodynamic Activation Parameters. The activation energy (E_a) values were calculated from Arrhenius equation ((6)) using the measured slope values of graphs as presented in Figures 6(a) and 6(b), and the corresponding data is presented in Table 6.

The ΔS^* and ΔH^* values were calculated using the measured slope and intercept values of graphs shown in

Thermodynamic adsorption parameters								
Schiff base inhibitor	Concentration (ppm)	Temperature (K)	K_{ads} (10 ³ M ⁻¹)	ΔG°_{ads} (kJmol ⁻¹)	∆H° _{ads} (kJmol⁻¹)	ΔS°_{ads} (Jmol ⁻¹)		
		303	15.8	-34				
N,N'-bis(Salicylidine)-4,4-' diaminostilbene (SDS)	100	313	11.2	-34				
		323	8.34	-35	29.2	203		
		333	6.56	-35				
N,N'-bis(Salicylidine)-4,4' -diamino azobenzene (SDA)		303	24.7	-35				
	100	313	17.3	-35				
		323	11.5	-36	30.6	207		
		333	9.6	-36				

TABLE 7: Thermodynamic parameters related to SDS and SDA.

Figures 7(a) and 7(b) by transition state equation (7) and tabulated in Table 6.

The E_a values of AA2024 in 1 M HCl in the presence of inhibitors SDS and SDA were found to be higher than those values obtained in the absence, depicting the increase in energy barrier for the corrosion reaction. The obtained E_a values for the corrosion inhibition process are being above 20 kJ mol⁻¹ and indicate that the inhibition process is wholly controlled by the surface reactions [50]. The inhibitor molecules get adsorbed onto the electrode surface leading to the development of a physical barrier being formed between the metal and corrosion media. This blocks the transfer of charge and reduces the reaction of alloy. An increase in inhibitor concentration leads to an increase in E_a values due to physical adsorption on the metal surface by the inhibitor molecules [51]. In a sense, a physical barrier is formed due to the adsorption of inhibitor molecules on the surface of the electrode which reduces the alloy dissolution in electrochemical reactions [52].

The decrease in the inhibition efficiency of SDS and SDA inhibitor compounds with the increase in temperature can be attributed to a decrease in the extent of adsorption of the inhibitor on the alloy surface at elevated temperatures. The corresponding increase in corrosion rate is due to a larger area of matrix surface that gets exposed to the corrosion medium. The decrease in inhibition efficiency and increase in corrosion rate at elevated temperatures support the view that the inhibitor is adsorbed on the alloy surface through physisorption [53, 54]. The endothermic nature of the corrosion process is indicated by the positive values of ΔH^* . The rise in values of ΔS^* in presence of inhibitor concentration reveals that an increase in randomness occurred on going from reactants to the activated complex [55-57]. The adsorption of SDS and SDA inhibitor molecules on the AA2024 alloy surface could be regarded as quasisubstitution between the inhibitors molecules in the aqueous phase and water molecules on the electrode surface.

4.5. Adsorption Isotherms. In general, it is presumed that the adsorption of the inhibitor is the first step in the mechanism of inhibition in aggressive media at the alloy-solution interface. The adsorption isotherms furnish useful understandings into the mechanism of corrosion inhibition.

Theoretically, the adsorption process has been well regarded as a simple substitution adsorption process, in which an organic molecule in the aqueous phase substitutes the water molecules adsorbed on the alloy surface [58]. $H_2O_{(sol)}$, $Org_{(sol)}$, $H_2O_{(ads)}$, and $Org_{(ads)}$ are represented in equation (12), water and organic molecules present in solution, and those adsorbed with *x* representing replaced water molecules count.

$$xH_2O_{(ads)} + Org_{(sol)} \longrightarrow Org_{(ads)} + xH_2O_{(sol)}.$$
 (12)

Understanding the adsorption mode and adsorption isotherm becomes necessary which provide deep insight into the interaction happening between inhibitor and alloy surface. From the weight loss experimental data obtained for different concentrations of inhibitor compounds, the surface coverage (θ) values were utilized for explaining the adsorption isotherm that best fitted. These values were tested graphically to fit into Langmuir, Freundlich, or Temkin isotherms. Among them, Langmuir's isotherm equation (8) fitted the best as seen in Figure 8(a). The linear regression coefficient values (R^2) were found to be between 0.9990 and 0.9992.

The equilibrium constant $K_{(ads)}$ was derived from the graph (Figure 8(b)) using equation (8). The values of standard free energy of adsorption of inhibitor (ΔG°_{ads}), the heat of adsorption (ΔH°_{ads}), and entropy of adsorption (ΔS°_{ads}) were derived from Figure (8) using equation (9). All these data were tabulated as shown in Table 7.

Inhibitor compounds SDS and SDA under investigation are spontaneously getting adsorbed on alloy surface, and the layer being very stable is indicated by the negative values of standard free energy of adsorption. The standard free energy of adsorption values $\langle -20 \text{ kJ mol}^{-1}$ indicates physisorption, and $\rangle -40 \text{ kJ mol}^{-1}$ indicates chemisorption [59]. The calculated experimental values were found between -34 and -37 kJ mole⁻¹. Hence, the adsorption of SDS and SDS compounds on AA2024 alloy surface may be attributed to be both physical and chemical where chemisorption was found to be predominating. The positive values of ΔH°_{ads} and Δ S°_{ads} represent substitutional adsorption. This can be attributed to rise in entropy of solvent and more positive water



(a)





(c) FIGURE 9: Continued.



(d)

FIGURE 9: SEM image of (a) polished AA2024 alloy. (b) Rough surface in the presence of 1 M HCl. (c) Presence of 1M HCl and 100 ppm SDS. (d) Presence of 1MHCl and 100 ppm SDA.

desorption enthalpy. The rise in entropy is the driving force for SDS and SDA molecules getting adsorbed on AA2024 alloy surface.

4.6. Mechanism of Inhibition. In highly acidic solutions, the inhibitor molecules undergo protonation and exist as protonated-positive species [60]. These protonated-positive species adsorb on cathodic sites of alloy surface by electrostatic interaction and resulting in decreased rate for the cathodic reaction. The anions present in the acidic solution and their adsorption on the alloy surface play an important role in the mechanism of inhibition exhibited by inhibitor molecules.

The anions found in the corrosive medium solution assist in the inhibition process by organic compounds by getting adsorbed on alloy surfaces [61]. The corrosive medium used in this investigation is highly acidic, and the alloy surface is positively charged. Because of this, the negative Cl⁻ ions get adsorbed onto the surface of the alloy which makes the alloy surface negatively charged. This results in the physisorption of protonated inhibitor compound molecules which are positively charged interacting with Cl⁻ ions that are adsorbed on alloy surface through electrostatic interaction. The negative charge centers of the inhibitor molecules that contain lone pair of electrons and π -electrons get involved in an interaction with anodic sites on the alloy surface resulting in their adsorption. The neutral inhibitor molecules occupy the sites on the surface of alloy that is vacant through chemisorption by displacing the water molecules from the alloy surface and sharing of electrons by the heteroatoms like S, N, and O. Inhibitor molecules existing in protonated form and presence of negatively charged centers on the inhibitor molecules are also the reason for inhibitor molecules interacting mutually on the surface of the alloy.

In acidic solution, nitrogen and oxygen atoms from inhibitor molecules may get adsorbed on cathodic sites of AA2024 alloy surface competing with H⁺ ions. This is a result of organic compounds getting adsorbed via phenolic, amine, and imine groups through the delocalized π -electrons of the phenolic group, amine and imine nitrogen, and also via electron getting released by –OH groups. The inhibition efficiency of SDA was found to be higher than that of SDS since it contains more electronegative imine group -N=N-.

4.7. Scanning Electron Microscopic (SEM) Study. The surface morphology of AA2024 alloy was analyzed using the recorded SEM images of alloy samples polished, exposed for 2 hours in 1 M HCl in the absence and the presence of 100 ppm SDS and SDA inhibitors. The SEM images of polished AA2024 alloy with smooth surface, with rough surface in presence of 1 M HCl, and less rough surface in presence 1 M HCl of and 100 ppm of SDS and SDA inhibitors are shown in Figures 9(a)–9(d), respectively.

It is noticed that the alloy surface is damaged to a lesser extent in presence of an inhibitor. The alloy surface is smoother and uniform in the case of SDA than SDS. Thus, it is concluded that both SDA and SDS were proved to exhibit inhibitor mechanism AA2021 alloy surface. In addition, it was also found that SDA protects the AA2024 alloy more effectively than the SDS.

5. Conclusions

The inhibition efficiencies obtained by potentiodynamic polarization, electrochemical impedance spectroscopy, and weight loss techniques of Schiff base inhibitors at all concentrations on the AA 2014 alloy in 1 M HCl are in good agreement.

The inhibition efficiencies of Schiff bases increase in the order: N, N' – bis(salicylidine) – 4, 4 – 'diaminostilbene < N , N' – bis(salicylidine) – 4, 4' – diamino azobenzene.

The increase in inhibition efficiency of Schiff bases is attributed to

- (1) The inhibition efficiency of N,N'-bis(salicylidine)-4,4-'diaminostilbene (SDS) is due to delocalized lone pair of electrons on the electronegative nitrogen atoms in NH₂, π -electrons of the -C=C- group and delocalized π -electrons of the phenolic group. In addition to the above, groups in Schiff base molecules undergo protonation. The cathodic reaction rate goes down due to protonated species undergoing adsorption over the cathodic sites via electrostatic interaction. Two delocalized π -electrons of the phenolic groups and electronegative oxygen atoms in the -OH group also contribute to the inhibition of alloy surfaces through adsorption
- (2) The inhibition efficiency of N,N'-bis(salicylidine)-4,4-diaminoazobenzene is greater than N,N'-bis(salicylidine)-4,4-diaminostilbene that is due to delocalized lone pair of electrons on the electronegative nitrogen atoms in NH₂, delocalized π -electrons of the phenolic group, and delocalized electronegative nitrogen atoms in the imine group (-N=N-). In addition to the above, groups in the Schiff base molecule undergo protonation. The cathodic reaction rate goes down due to protonated species undergoing adsorption over the cathodic sites via electrostatic interaction. Two delocalized π -electrons of the phenolic groups and electronegative oxygen atoms in -OH group also contribute to the inhibition of alloy surface through adsorption

Finally, it can be concluded that both SDA and SDS inhibitors were proved to be potential corrosion inhibitors for AA2021 alloy.

Data Availability

The [Excel] data used to support the findings of this study are available from the corresponding author upon request.

Conflicts of Interest

The authors declare that they have no conflicts of interest.

Authors' Contributions

The concept and experiments were designed and conducted by Mrs Shobha Bhaskara. The original draft of the manuscript was written by Dr Sanaulla Pathapalya Fakrudeen. Editing and compiling of the final draft were done by Dr Tegene Desalegn and Dr H C Ananda Murthy. The complete supervision of the work was done by Dr Bheemaraju V. All authors read and approved the final manuscript.

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