

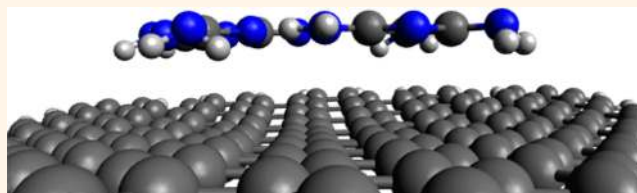
Exfoliation of Graphite with Triazine Derivatives under Ball-Milling Conditions: Preparation of Few-Layer Graphene *via* Selective Noncovalent Interactions

Verónica León,^{†,‡} Antonio M. Rodríguez,[†] Pilar Prieto,[†] Maurizio Prato,[‡] and Ester Vázquez^{†,*}

[†]Departamento de Química Orgánica, Facultad de Ciencias y Tecnologías Químicas-IRICA, Universidad de Castilla-La Mancha, 13071 Ciudad Real, Spain and

[‡]Department of Chemical and Pharmaceutical Sciences, University of Trieste, Piazzale Europa 1, 34127 Trieste, Italy

ABSTRACT A ball-milling treatment can be employed to exfoliate graphite through interactions with commercially available melamine under solid conditions. This procedure allows the fast production of relatively large quantities of material with a low presence of defects. The milling treatment can be modulated in order to achieve graphene flakes with different sizes. Once prepared, the graphene samples can be redispersed in organic solvents, water, or culture media, forming stable dispersions that can be used for multiple purposes. In the present work, we have screened electron-rich benzene derivatives along with triazine derivatives in their respective ability to exfoliate graphite. The results suggest that the formation of a hydrogen-bonding network is important for the formation of multipoint interactions with the surfaces of graphene and that can be used for the exfoliation of graphite and the stabilization of graphene in different solvents. Aminotriazine systems were found to be the best partners in the preparation and stabilization of graphene layers in different solvents, while the equivalent benzene derivatives did not show comparable exfoliation ability. Computational studies have also been performed to rationalize the experimental results. The results provide also the basis for further work in the preparation of noncovalently modified graphene, where derivatives of aminotriazines can be designed to form extensive hydrogen-bond networks on the graphene surface with the aim of manipulating their electronic and chemical properties.



KEYWORDS: triazine · graphene · noncovalent interactions · ball milling · density functional calculations

Graphene has emerged as a new material, with outstanding mechanical and electronic properties that will permit a broad range of applications, from microelectronics to composites or even medicine.^{1,2} After a few years of intense research, devoted mainly to the preparation of samples of high quality, though in a very small scale, the attention has now turned to transferring these extraordinary features to real world applications. For this reason, scientists and engineers are working together to develop ways to make high-quality graphene in sufficient quantities and at reasonable cost.

Many ingenious methods have been described for the preparation of graphene, leading to materials with different properties, suitable for diverse applications.³ The

chemical vapor deposition (CVD) methods produce, in general, graphene with a low number of defects that can be very useful for highly demanding electronic applications, in which small quantities of graphene are still sufficient.⁴ However, for the development of large-scale applications, ranging from conductive inks and fillers in composites to sensors or batteries, cost-effective production methods would be advisable with a good balance between ease of fabrication and/or manipulation and preservation of graphene properties. Moreover, besides biomedical applications,⁵ considering that graphene could be integrated into new electronics or composites, it would be fundamental to evaluate the impact of this new material on health and environment.^{6,7} All these studies require graphene dispersions

* Address correspondence to ester.vazquez@uclm.es.

Received for review October 3, 2013 and accepted December 31, 2013.

Published online 10.1021/nn405148t

© XXXX American Chemical Society

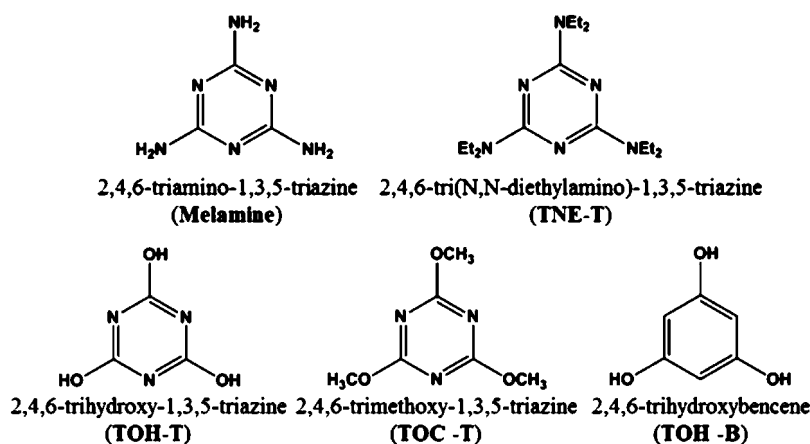


Chart 1. Chemical Structure of the Exfoliating Agents Used in This Study with Their Names and Corresponding Acronyms As Used in the Text

48 in aqueous media and controlled chemical modifica-
49 tion.⁸

50 In principle, exfoliation from bulk graphite is the
51 most economical way to achieve large quantities of
52 graphene. In addition, liquid-phase exfoliation techniques⁹
53 present several advantages because stable suspen-
54 sions of graphene can be used for various processing
55 steps of the material such as film deposition, surface
56 modification, and chemical functionalization. The ex-
57 foliation of graphene into solution requires breaking
58 the enormous van der Waals-like forces between gra-
59 phite layers, which can be achieved by sonication of
60 graphite in solvents¹⁰ and by chemical and electro-
61 chemical oxidation.¹¹ However, these techniques also
62 have some limitations. Oxidation techniques are very
63 effective but disruptive and, even after reduction,
64 produce high defective graphene.¹² Sonication in or-
65 ganic solvents gives graphene of better quality, though
66 still defective, but usually the flake size is relatively small.

67 Moreover, graphene layers tend to aggregate in
68 order to re-establish the graphitic structure and to
69 minimize surface free energy. This can be avoided by
70 covalent functionalization^{13,14} or by noncovalent inter-
71 action with stabilizers, such as surfactants, polymers,
72 and aromatic molecules.¹⁵ Though covalent techni-
73 ques may alter significantly the electronic structure
74 of graphene, the absorption of molecules on graphene
75 can serve as a mere protecting coat of the graphene
76 sheets. In other cases, the adsorption of certain mol-
77 ecules can also induce a band gap in graphene.¹⁶ Thus,
78 the understanding of noncovalent interactions of dif-
79 ferent molecules with graphene is a very interesting
80 topic for the development of new derivatives for
81 desirable applications, including graphene-based chem-
82 ical detectors, field-effect transistors, or organic op-
83 toelectronic devices.

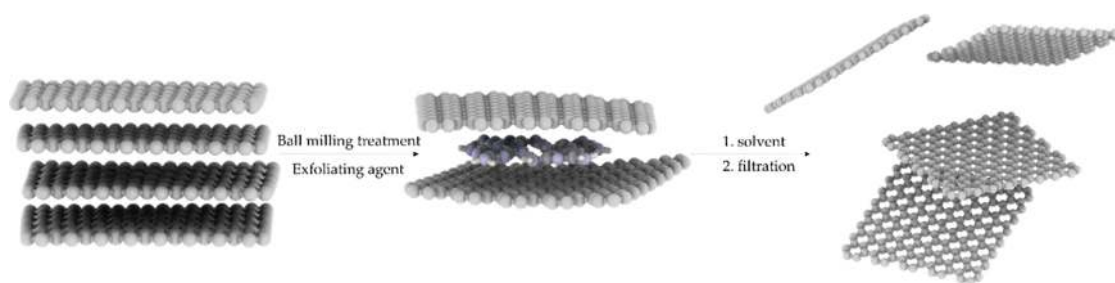
84 Recently, we described an interesting alternative for
85 the preparation of stable dispersions of graphene in
86 different solvents, driven by an easy and eco-friendly
87 ball-milling approach.¹⁷ Mechano-chemical activation

of carbon nanostructures has generated great interest
in recent years,¹⁸ and it has lately been used for
the selective functionalization of graphene nano-
platelets.¹⁹ In our work, we used a ball-milling treat-
ment to exfoliate graphite through interactions with
commercially available melamine under solid condi-
tions. This procedure allows the fast production of
relatively large quantities of material with a low pre-
sence of defects. The milling treatment can be modu-
lated in order to achieve graphene flakes with different
sizes. Once prepared, the graphene samples can be
redispersed in organic solvents, water, or culture med-
ia, forming stable dispersions that can be used for
multiple purposes.

There are several interesting aspects in the use of
melamine as dispersing agent in ball-milling experi-
ments. First of all, melamine can be easily washed
away, leaving pure graphene flakes in organic or water
dispersions. Another question relates to the size of
melamine, which is a small molecule, compared with
the structure of polyaromatic hydrocarbons used for
the exfoliation of graphite. In order to explain these
results and especially to understand the role of mela-
mine in the exfoliation, we have performed further
experimental and computational investigations. We
have modified the structure of the exfoliating agent
from melamine to other triazine/benzene derivatives.
This study aims to respond to many fundamental
questions: Is this exfoliating process only induced by
melamine? Can we use other triazine systems to obtain
the exfoliation and, at the same time, stabilize gra-
phene flakes in different solvents? Does this work
disclose new concepts for the noncovalent modifica-
tions of graphene?

RESULTS AND DISCUSSION

Experiment. The compounds depicted in Chart 1
were commercially available, except 2,4,6-tri(*N,N*-
diethylamino)-1,3,5-triazine, which was prepared follow-
ing published procedures.²⁰ We have used a ball-milling



Scheme 1

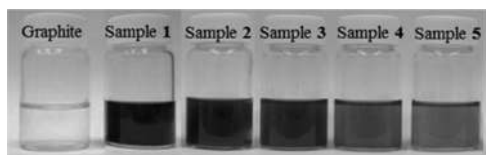


Figure 1. Photograph of graphite and graphene/exfoliating agent solutions in DMF.

TABLE 1. Comparison of Effectiveness of Exfoliating Agent as Stabilizer of Graphene in DMF

sample	exfoliating agent	concentration of exfoliating agent (mg/mL) ^a	concentration of graphene (mg/mL) ^a	graphene concn/ exfoliating agent concn
1	melamine	1.13	0.37	0.33
2	TNE-T	2.28	0.12	0.05
3	TOH-T	0.98	0.07	0.07
4	TOC-T	1.33	0.05	0.04
5	TOH-B	0.95	0.05	0.05

^a Calculated by thermogravimetric analysis.

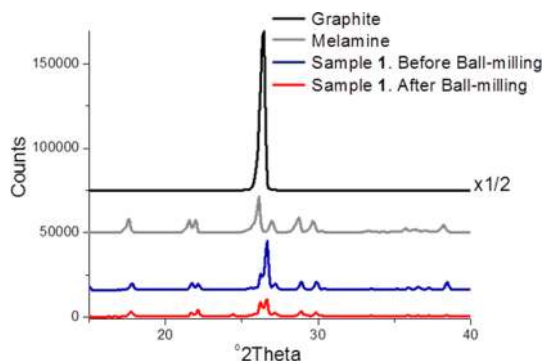


Figure 2. XRD patterns of graphite, melamine, and sample 1 before and after the ball-milling treatment.

of graphite after ball-milling treatment with the triazine/benzene derivatives present a weight loss at around 300 °C, which corresponds to the amount of exfoliating agent used in each sample. No further loss takes place above or below this temperature, showing that no oxidative defects have been generated around the graphite flakes.²¹ TGA (Figure S2) has also been used to calculate the concentration of pure exfoliating agent and graphene in dispersions of DMF (Table 1) and water (see Supporting Information, Table S1).

The final concentration of graphene in DMF depends on the substituents in the triazine derivatives. The highest graphene concentration is obtained using melamine; in fact, the graphene to melamine ratio is higher than all the tested compounds and is higher than the one described for other polymers and surfactants.¹⁵ While the possibility of forming hydrogen bonds between molecules leads to higher graphene concentrations (1 vs 2 and 3 vs 4 in Table 1), amino substituents in the triazine rings lead to the highest graphene concentrations (entries 1 and 2 vs 3 and 4 in Table 1), which suggests that amino substituents provide more stabilization than hydroxyl groups. Water dispersions can also be prepared (Table S1, Supporting Information), in which case the solubility of the exfoliating agent seems to be the key factor in the stabilization of the graphene layers: derivatives such as 2,4,6-trihydroxy-1,3,5-triazine, with a very high solubility in water, do not disperse the graphene flakes in water.

The as-prepared dispersions can be filtered and washed to remove the triazine/benzene derivatives, and the graphene samples can then be redispersed in

127 approach to exfoliate graphite through interactions with
128 different 1,3,5-triazine/benzene derivatives (Chart 1) under
S1 129 solid conditions (Scheme 1). In a typical procedure,
130 7.5 mg of graphite and 0.16 mmol of the triazine/benzene
131 derivative were ball-milled at 100 rpm during 30 min
132 under air atmosphere. After the milling treatment, the
133 resulting solid mixtures were dispersed in 20 mL of water
134 or DMF to produce black suspensions. After letting the
135 solutions rest for 5 days, the precipitate was removed,
136 and the resulting dispersions were stable at room tem-
137 perature within weeks. A comparison between the dis-
F1 138 persions obtained in DMF, using the different exfoliating
T1 139 agents, is presented in Figure 1 and Table 1. No dispersion
140 but only a black solid was obtained when graphite was
141 ball-milled in the absence of exfoliating agent.

142 Typical X-ray diffraction (XRD) patterns of solid
143 samples before and after ball-milling treatment with
F2 144 melamine are reported in Figure 2, showing that the
145 sharp graphitic (002) reflection around 25° clearly
146 decreases after the milling.

147 Thermogravimetric analysis (TGA) provides insights
148 into the composition of graphite and the exfoliating
149 agent after the ball-milling process (Supporting Infor-
150 mation, Figure S1). Graphite is thermally stable when
151 heated up to 900 °C under inert atmosphere. Samples

TABLE 2. Final Graphene Concentration, In-Plane Crystallite Sizes (L_a), and Distance between Defects (L_D) of Graphene Samples Prepared Using Different Exfoliating Agents

sample	exfoliation agent	concentration of graphene (mg/mL) ^a	I_D/I_G ^b	$I_D/I_{G'}$ ^b	L_a (nm)	L_D (nm)
1	melamine	0.38	0.97, 2.09, 2.25	0.42, 0.44, 0.49	39, 40, 44	16, 17, 17
2	TNE-T	0.18	2.47, 2.60, 2.86	0.37, 0.47, 0.56	30, 35, 46	15, 16, 19
3	TOH-T	0.07	1.40, 1.99, 2.32	0.48, 0.53, 0.57	30, 32, 35	15, 15, 16
4	TOC-T	0.07	1.42, 1.53, 1.96	0.31, 0.33, 0.35	47, 51, 54	19, 20, 20
5	TOH-B	0.06	2.66, 2.90, 3.00	0.71, 0.79, 0.82	20, 21, 24	12, 13, 13

^a Calculated by UV–vis–NIR absorption spectra. ^b The different I_D/I_G and $I_D/I_{G'}$ values are from different locations in the sample.

184 fresh solvents, forming stable suspensions, which can be
185 characterized by UV–vis–NIR absorption spectroscopy,
T2 186 showing Lambert–Beer behavior. Table 2 shows the final
187 graphene concentrations of free exfoliating graphene
188 dispersions in DMF calculated by UV–vis–NIR absorption
189 (see Supporting Information for details, Figure S3).

190 In order to demonstrate the exfoliation of graphite
191 and to analyze the quality of the produced graphene
192 using the different exfoliating agents, fresh graphene
193 dispersions of samples 1–5 were drop-cast onto sili-
194 con oxide surfaces and studied by Raman spectroscopy.
195 This technique helps to identify graphene from
196 graphite and few-layer graphene and has become a
197 key technique to probe disorder in graphene through
198 defect-activated peaks.^{22,23} Graphene exhibits G and
199 2D (also called G') modes around 1580 and 2700 cm^{-1} ,
200 respectively, that always satisfy the Raman selection
201 rules. However, when graphene is affected by defects,
202 the Raman features at 1345 (D band) and 1626 cm^{-1}
203 (D' band) appear in the spectrum. Recently, the evolu-
204 tion of the intensity ratio I_D/I_G between the G and the D
205 band has been used to provide a method to quantify
206 the density of defects in monolayers and few-layer
207 graphene.^{24,25} Moreover, at moderate defect concen-
208 tration, the D' peak can be clearly distinguished from
209 the G peak and the intensity ratio of the D and D' peaks
210 can be used to get information of the nature of the
211 defects for a moderate amount of disorder.²⁶

F3 212 Figure 3 compares the 514 nm Raman spectra of
213 bulk graphite and representative flakes from a DMF
214 dispersion of samples 2 and 5. Even though the final
215 concentrations of graphene are not as high as using
216 melamine as exfoliating agent, Raman spectroscopy
217 shows that the graphene produced in the different
218 samples consists of few layers. The 2D band of the
219 diverse samples is quite different from bulk graphite, it
220 can be deconvoluted into four bands (Lorentzian
221 peaks), which has been reported as a characteristic
222 feature of the Raman spectra of bilayer graphene.²⁷ In
F4 223 addition, Figure 4 shows a representation of the in-
224 tensity ratio between G and 2D bands, and Raman
225 spectra were taken in different areas of the sample. It
226 has been described that the intensity of the G band
227 increases with the number of layers,²⁸ thus from this
228 figure, it seems that triaminotriazine derivatives pro-
229 duce better exfoliation of graphite.

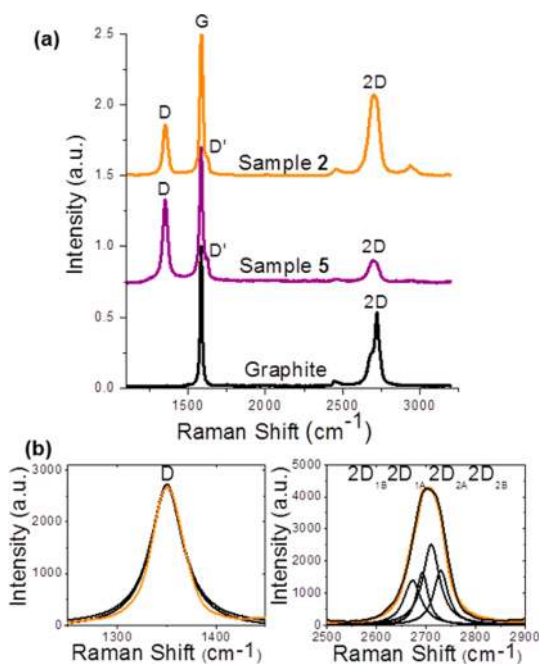


Figure 3. (a) Comparison of Raman spectra at 514 nm for graphite, sample 2, and sample 5. (b) D band in sample 2, fitted to one component. The four components of the 2D band in sample 2.

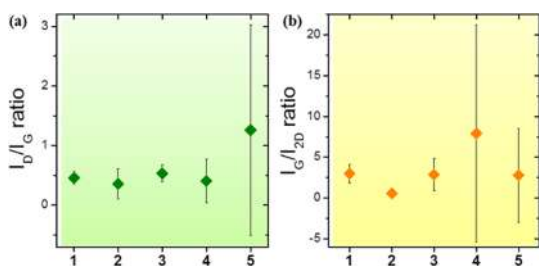


Figure 4. Statistical representation of intensity ratios between (a) G and D bands and (b) G and 2D bands in Raman spectra of samples 1–5. Data are taken from 10 different locations in the sample and are plotted with standard deviations.

We have also calculated the in-plane crystallite size
 (L_a) ²⁹ (amount of border with respect to the total
crystallite area) and the distance between defects in
the sp^2 lattice L_D .^{24,30} Table 2 clearly shows smaller
values for the benzene derivative, while similar data are
observed for the triazine derivatives. The nature of
these defects can be estimated by the intensity ratio of

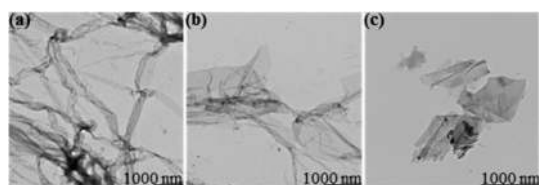


Figure 5. TEM images of graphene in (a) sample 1, (b) sample 3, and (c) sample 5.

the D and D' peaks, using the model described by Eckmann *et al.*²⁶ This ratio is maximum, around 13, for sp³ defects, and it decreases to 7 for vacancy-like defects and reaches the minimum for boundary-like defects. Based on the data reported in Table 2 and Figure 4, these latter are the type of defects that we expect to find in our samples.

Graphene flakes can also be observed by transmission electron microscopy (TEM). Figure 5 shows representative images of graphene flakes, free from the exfoliating agents, produced by immersion of the respective dispersions onto a TEM grid. The flakes appear very large and extremely flexible, with many visible wrinkles.

Computations. In order to rationalize the experimental results, and especially to evaluate the relative ability of the studied compounds to exfoliate graphite and stabilize graphene, we performed computational calculations. Different factors must be taken into account: the structure of the individual components, the interactions between the adsorbed molecules, and their affinity for the graphene surface.

Recently, several authors have performed theoretical studies on the interactions of small molecules and polycyclic aromatic compounds with graphene.^{31–35} The capability of tuning the graphene band structure is a hot topic, and the absorption of these molecules could be a simple and effective method to control the electronic properties of graphene systems.

From a theoretical point of view, the quantification and identification of the nature of interactions of adsorbed molecules on graphene present several challenges. One important aspect in this kind of system is the calculation of the dispersion interactions, which are ubiquitous weak attractive forces between molecules. It has become clear that they turn to be one of the dominant forces for large molecules and in supramolecular chemistry. The calculation of these dispersion forces is complex from a computational perspective. Quantum mechanical calculations based on density functional theory (DFT), with the most widely used exchange-correction DFT functionals, that is, local density approximation (LDA)³⁶ and generalized gradient approximation (GGA),^{37,38} are often inadequate. Some studies have recently shown that the absorption strength of molecules on graphene is governed by dispersive interactions,³³ and the omission of nonlocal electron correlations can strongly affect the calculated adsorption energy. A number of dispersion-corrected

DFT methods have been developed. The most widely used are probably the DFT-D family of methods by Grimme (mostly at the B97-D level)³⁹ and fully nonlocal and computationally expensive methods.^{40–42}

In this paper, all the calculations were performed using GAUSSIAN 09⁴³ suites of programs. In order to reduce the computational cost, we have preoptimized all the structures using the ONIOM^{44–46} method with DFT⁴⁷ B3LYP/3-21G*^{48–50} and B97-D as the low- and high-theory levels, respectively. A graphene sheet consisting of 170 carbon atoms and 36 peripheral hydrogen atoms was used in the calculations, with C–C and C–H bond distances set at 1.46 and 1.01 Å, respectively. Preoptimized structures were fully optimized with a semilocal density functional with dispersion correction (B97-D) level of theory developed by Grimme.³⁹ This method renders reasonable results at a very low computational cost in comparison to CCSD(T),⁵¹ and it has already proven to be useful in noncovalent interactions between graphene sheets.⁵² For the adsorbents, the molecular structures were fully optimized with B97-D method until the minima were localized or when the gradient convergence factor was better than 10^{–6} hartree/bohr. During the optimization steps, all species were free to move.

The absorption of different triazine and benzene derivatives on a graphene sheet has been studied. The geometry features of some of these compounds are depicted in Figure S5 (Supporting Information), showing the most stable conformation, in which the adsorbate atoms are located at the positions where the carbon atoms of the next second graphite-like sheet should be found, in agreement with similar studies.³³ The adsorption energy, E_{ads} , for the exfoliating agents on the graphene sheet is defined according to the following equation:

$$E_{\text{ads}} = E_{\text{Ea/Graphene}} - (E_{\text{Ea}} + E_{\text{Graphene}})$$

where $E_{\text{Ea/Graphene}}$ is the energy of the system with the exfoliating agent (Ea) adsorbed on graphene and E_{Ea} and E_{Graphene} are the energies of the isolated exfoliating agent and graphene, respectively. With this definition, the adsorption energy is negative if the complete system is stabilized with the adsorption of the exfoliating agent on graphene.

Table 3 shows the adsorption energies of some triazine and benzene derivatives. These results are in concordance with previous computational studies.⁵³ The stabilization energies of benzene and 1,3,5-triazine derivatives are similar, although slightly higher in the case of benzene rings (Table 3, entries 3, 8 and 4, 9). As expected, the presence of NH₂ and OH groups in the aromatic core increases the adsorption energy, which is also higher with the number of groups, providing NH₂ substituents with more stabilization than OH groups (Table 3, entries 4, 6 and 9, 11).

Close examination of the geometry features of the optimized structures reveals that the atoms of

TABLE 3. Adsorption Energies (E_{ads}) for Each Exfoliating Agent on the Graphene Sheet Calculated at B97-D Theory Level

entry	exfoliating agent	E_{ads}^a (kcal/mol)
1	1,3,5-triazine	-13.70
2	2-amino-1,3,5-triazine	-17.18
3	2,4-diamino-1,3,5-triazine	-20.86
4	melamine	-24.34
5	2,4,6-tri(<i>N,N</i> -dimethylamino)-1,3,5-triazine	-38.03
6	TOH-T	-18.75
7	benzene	-14.33
8	1,3-diaminobenzene	-21.86
9	1,3,5-triaminobenzene	-25.69
10	1,3,5-tri(<i>N,N</i> -dimethylamino)benzene	-39.28
11	TOH-B	-20.23

^aThe adsorption energy for the exfoliating agents on the graphene sheet.

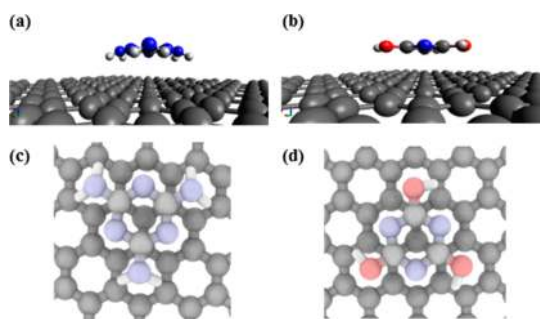


Figure 6. Calculated geometries of melamine adsorbed on a graphene sheet: (a) side view, (c) top view, and TOH-T adsorbed on a graphene sheet (b) side view, (d) top view.

TABLE 4. Adsorption Energies (E_{ads}) of Melamine and 1,3,5-Triaminobenzene Dimers on the Graphene Sheet Calculated at B97-D Theory Level

entry	exfoliating agent	E_{ads}^a (kcal/mol)
1	melamine	-24.34
2	melamine dimer	-44.60
3	two melamine molecules	-44.94
4	1,3,5-triaminobenzene	-25.69
5	1,3,5-triaminobenzene dimer	-34.97
6	two 1,3,5-triaminobenzene molecules	-48.60

^aThe adsorption energy for the exfoliating agents on the graphene sheet.

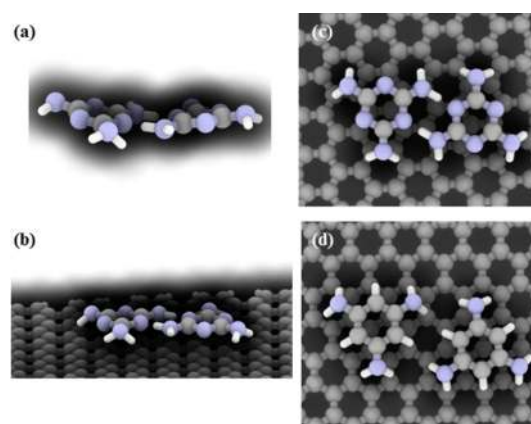


Figure 7. Calculated geometries of (a) melamine dimer, (b) melamine dimer on graphene side view, (c) melamine dimer on graphene top view, (d) 1,3,5-triaminobenzene dimer on graphene top view.

hydrogen in the NH_2 group are directed toward the graphene surface (Figure 6a). As previously reported, there is a great degree of interaction between the NH_2 group and the graphene surface, which requires distortion of NH_2 groups out of the triazine plane. In fact, the calculated distance from the graphene surface to the hydrogen atom of the NH_2 (2.7 Å) group in melamine is smaller than that for the hydrogen atom of the OH group in the 2,4,6-trihydroxy-1,3,5-triazine (3.1 Å). It has been previously described⁵³ that the adsorption energies of aminotriazines on graphene are directly related to the ability of these molecules to accept charge transfer from graphene to aminotriazines, in part through the presence of hydrogen atoms in the substituents. This fact explains why hydrogen atoms are directed toward the graphene surface and could also explain a higher adsorption energy for NMe_2 substituents, which provides a larger number of hydrogen atoms.

However, these results do not agree with the experimental findings. As described above, the highest concentration of graphene was obtained using melamine as exfoliating agent and not using TNE-T. This last molecule has more hydrogen atoms that can interact with the graphene surface and, following this model, should present higher adsorption energy. This means

that the adsorption energy of discrete molecules on graphene is not sufficient by itself to justify the exfoliation and stabilization of graphite in good yields.

It is known that triaminotriazines can form hydrogen bonds that can create 2D molecular assemblies on different surfaces.^{54,55} Thus, we performed the computational calculations of these possible aggregations. The calculated results at B97-D level of theory for melamine and benzene derivative aggregates are collected in Table 4. The geometry features of some aggregates are depicted in Figure 7. It can be pointed out that the adsorption energy of the melamine dimer on the graphene surface is higher than the one of the 1,3,5-triamine benzene dimer. It is interesting to note that, in the case of melamine, this energy is similar when considering either a dimer or two individual molecules. Its behavior differs in the case of the benzene derivative (Table 4, entries 2, 3 and 5, 6). In the case of the benzene ring, the adsorption of two individual molecules is more favorable than the adsorption of the dimer.

Similar results were observed by Rochefort *et al.*,⁵⁶ who studied the adsorption of benzoic acids on graphene. Thus, whereas the absorption of one molecule of benzoic acid presents both CO and OH units in the COOH group bent toward the graphene sheet, the

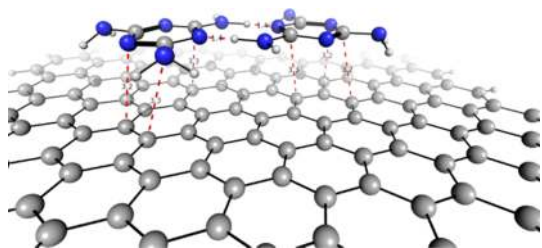


Figure 8. Melamine dimer on graphene.

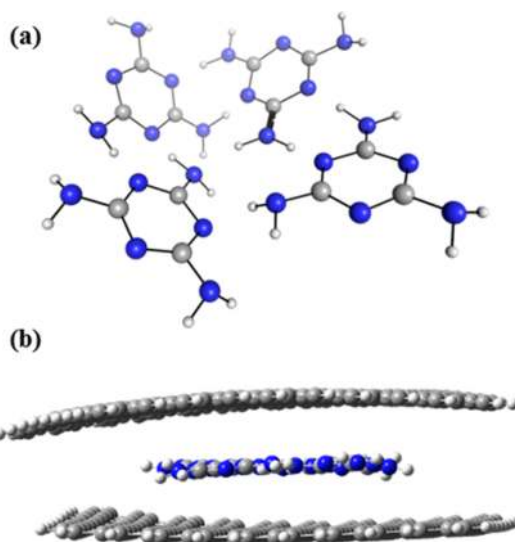


Figure 9. (a) Melamine tetramer, (b) melamine tetramer inside two graphene layers calculated at PM6⁵⁷ theory level.

These results provide also the basis for further experimental work for the preparation of noncovalently modified graphene, where derivatives of aminotriazines can be designed to form extensive hydrogen-bond 2D networks on the graphene surface with the aim of manipulating their electronic and chemical properties.

CONCLUSIONS

We have studied, both experimentally and theoretically, the interactions of a series of aromatic derivatives with graphene layers. We have found that, although melamine performs better in the exfoliation of graphite, triaminotriazine derivatives are also useful systems for the exfoliation and stabilization of graphene flakes in different solvents. The reason for this behavior can be attributed to two main features: (1) analogously to all the other investigated systems, melamine has an aromatic nucleus able to interact with the π -system of graphene; (2) in addition to this, melamine is able to form extended 2D networks, based on the presence of hydrogen bonds. These results can be usefully exploited toward the ideal system for the exfoliation of graphite, in which triaminotriazine derivatives can be designed to form extensive hydrogen-bond 2D networks on the graphene surface and, at the same time, manipulate their electronic and chemical properties by noncovalent interactions.

COOH in the dimer remains parallel to the underlying surface. The energy of adsorption of the dimer was smaller than the energy of adsorption for two isolated molecules. The authors attributed this behavior to an increased π - π repulsion. In our case, benzene derivatives do not form hydrogen bonds among themselves, while in the melamine dimer, the distance between two melamine molecules confirms the formation of hydrogen bonds. Moreover, the optimized structure of the dimer alone presents a 3D structure but, when it lays on graphene, forms a 2D structure by formation of hydrogen bonds on the graphene layer (Figure 8).

If we go further to a four melamine system, two geometrically different tetramers can be formulated, a linear one and a nonlinear one (Supporting Information, Figure S6). The two possible aggregation patterns have similar energy values, though the most favorable is the nonlinear one (Figure 9a). The adsorption of this 3D structure on a graphene sheet has also been studied (Supporting Information), following a similar behavior to the adsorption of the melamine dimer, in terms of stability (see above). Finally, in order to simulate the graphite exfoliation process, we placed the tetramers, both melamine and 1,3,5-triaminobenzene, inside two graphene layers (Supporting Information). In this case, due to the high molecular weight of this complex, the calculation was performed at PM6⁵⁷ theory level, which allows a higher number of atoms to be evaluated. Under these conditions, it can be observed that the melamine tetramer, from a 3D pattern, becomes a 2D structure, while the graphene layers tend to separate (Figure 9b).

These results strongly suggest that the formation of a hydrogen-bonding network makes the formation of multipoint interactions with the surfaces of graphene possible and can be used for the exfoliation of graphite and the stabilization of graphene in different solvents.

MATERIALS AND METHODS

Solvents were purchased from SDS and Fluka. Chemicals were purchased from Sigma-Aldrich and used as received without further purification. Graphite was purchased from Bay Carbon, Inc. (SP-1 graphite powder, www.baycarbon.com) and used without purification. The thermogravimetric analyses were performed with a TGA Q50 (TA Instruments) at 10 °C/min under

nitrogen atmosphere. Raman spectra were recorded with an InVia Renishaw microspectrometer equipped with a He-Ne laser. Raman samples were prepared from stable diluted dispersions of graphene/aromatic molecules by drop-casting of silicon oxide surfaces (Si-Mat silicon wafers, CZ). For the TEM analyses, concentrated dispersions of graphene/aromatic molecules were filtrated on a Millipore membrane (PTFE, 0.45 μ m),

472 paying special attention to keep the samples wet during the
473 filtration processes. Graphene samples were redispersed in
474 fresh water or DMF, forming stable dispersions. These disper-
475 sions were placed on a copper grid (3.00 mm, 200 mesh), coated
476 with carbon film by immersion, and dried under vacuum, and
477 the sample was investigated by a TEM Philips EM 208 at an
478 accelerating voltage of 100 kV. The milling treatments were
479 carried out in a Retsch PM100 planetary mill at 100 rpm during
480 30 min under air atmosphere.

481 **Preparation of Graphene Dispersions.** Graphite (7.5 mg) and
482 exfoliating agent (0.16 mmol) were introduced in a stainless
483 steel grinding bowl with 10 stainless steel balls (1 cm diameter).
484 The bowl was closed and placed within the planetary mill. The
485 ball-milling treatment conditions were 100 rpm during 30 min
486 under air atmosphere. After the treatment, the resulting solid
487 mixture was suspended in 20 mL of water or DMF and sonicated
488 for 1 min. After letting the solutions rest for 5 days, the pre-
489 cipitate was removed, and the resulting dispersions were stable
490 at room temperature for weeks.

491 **Conflict of Interest:** The authors declare no competing
492 financial interest.

493 **Acknowledgment.** This work has been supported, in part,
494 by the Spanish Ministerio de Economía y Competitividad
495 (project CTQ2011-22410 and project IPT-2012-0429-420000),
496 the University of Trieste and the Italian Ministry of Education
497 MIUR (cofin Prot. 2010N3T9M4 and Fibr RBAP11C58Y). A.M.R.
498 would like to acknowledge MEC/MICINN for the FPU fellowship.
499 Technical support from the High Performance Computing
500 Service of University of Castilla-La Mancha and from Claudio
501 Gamboz (University of Trieste, TEM images) is gratefully
502 acknowledged.

503 **Supporting Information Available:** TGA of samples 1–5.
504 Figures S1–S6, Tables S1 and S2, and Cartesian coordinates of
505 all the stationary points (B97-D) and adsorption geometries
506 discussed in the main text. This material is available free of
507 charge via the Internet at <http://pubs.acs.org>.

508 REFERENCES AND NOTES

509 1. Palermo, V. Not a Molecule, Not a Polymer, Not a Substrate
510 ... the Many Faces of Graphene as a Chemical Platform. *Chem. Commun.* **2013**, 49, 2848–2857.
511
512 2. Novoselov, K. S.; Fal'ko, V. I.; Colombo, L.; Gellert, P. R.;
513 Schwab, M. G.; Kim, K. A Roadmap for Graphene. *Nature*
514 **2012**, 490, 192–200.
515 3. Edwards, R. S.; Coleman, K. S. Graphene Synthesis: Rela-
516 tionship to Applications. *Nanoscale* **2013**, 5, 38–51.
517 4. Wei, D.; Wu, B.; Guo, Y.; Yu, G.; Liu, Y. Controllable Chemical
518 Vapor Deposition Growth of Few Layer Graphene for
519 Electronic Devices. *Acc. Chem. Res.* **2013**, 46, 106–115.
520 5. Mao, H. Y.; Laurent, S.; Chen, W.; Akhavan, O.; Imani, M.;
521 Ashkarran, A. A.; Mahmoudi, M. Graphene: Promises, Facts,
522 Opportunities, and Challenges in Nanomedicine. *Chem. Rev.*
523 **2013**, 113, 3407–3424.
524 6. Bianco, A. Graphene: Safe or Toxic? The Two Faces of the
525 Medal. *Angew. Chem., Int. Ed.* **2013**, 52, 4986–4997.
526 7. Bussy, C.; Ali-Boucetta, H.; Kostarelos, K. Safety Considera-
527 tions for Graphene: Lessons Learnt from Carbon Nano-
528 tubes. *Acc. Chem. Res.* **2013**, 46, 692–701.
529 8. Rodriguez-Perez, L.; Angeles Herranz, M.; Martin, N. The
530 Chemistry of Pristine Graphene. *Chem. Commun.* **2013**, 49,
531 3721–3735.
532 9. Cui, X.; Zhang, C. Z.; Hao, R.; Hou, Y. L. Liquid-Phase
533 Exfoliation, Functionalization and Applications of Graphene.
534 *Nanoscale* **2011**, 3, 2118–2126.
535 10. Coleman, J. N. Liquid Exfoliation of Defect-Free Graphene.
536 *Acc. Chem. Res.* **2013**, 46, 14–22.
537 11. Chen, D.; Feng, H.; Li, J. Graphene Oxide: Preparation,
538 Functionalization, and Electrochemical Applications. *Chem. Rev.*
539 **2012**, 112, 6027–6053.
540 12. Boukhvalov, D. W.; Katsnelson, M. I. Modeling of Graphite
541 Oxide. *J. Am. Chem. Soc.* **2008**, 130, 10697–10701.

13. Quintana, M.; Vazquez, E.; Prato, M. Organic Functionaliza-
542 tion of Graphene in Dispersions. *Acc. Chem. Res.* **2013**, 46,
543 138–148.
544
545 14. Georgakilas, V.; Otyepka, M.; Bourlinos, A. B.; Chandra, V.;
546 Kim, N.; Kemp, K. C.; Hobza, P.; Zboril, R.; Kim, K. S. Func-
547 tionalization of Graphene: Covalent and Non-covalent
548 Approaches, Derivatives and Applications. *Chem. Rev.*
549 **2012**, 112, 6156–6214.
550 15. Parviz, D.; Das, S.; Ahmed, H. S. T.; Irin, F.; Bhattacharia, S.;
551 Green, M. J. Dispersions of Non-covalently Functionalized
552 Graphene with Minimal Stabilizer. *ACS Nano* **2012**, 6,
553 8857–8867.
554 16. Kozlov, S. M.; Vines, F.; Goerling, A. Bandgap Engineering of
555 Graphene by Physisorbed Adsorbates. *Adv. Mater.* **2011**,
556 23, 2638.
557 17. Leon, V.; Quintana, M.; Antonia Herrero, M.; Fierro, J. L. G.;
558 de la Hoz, A.; Prato, M.; Vazquez, E. Few-Layer Graphenes
559 from Ball-Milling of Graphite with Melamine. *Chem. Com-
560 mun.* **2011**, 47, 10936–10938.
561 18. Vázquez, E.; Giacalone, F.; Prato, M. Non-conventional
562 Methods and Media for the Activation and Manipulation
563 of Carbon Nanoforms. *Chem. Soc. Rev.* **2013**, 43, 58–69.
564 19. Jeon, I.-Y.; Choi, H.-J.; Jung, S.-M.; Seo, J.-M.; Kim, M.-J.; Dai,
565 L.; Baek, J.-B. Large-Scale Production of Edge-Selectively
566 Functionalized Graphene Nanoplatelets via Ball Milling
567 and Their Use as Metal-Free Electrocatalysts for Oxygen
568 Reduction Reaction. *J. Am. Chem. Soc.* **2013**, 135, 1386–1393.
569 20. Kolesinska, B.; Kaminski, Z. J. The Umpolung of Substituent
570 Effect in Nucleophilic Aromatic Substitution. A New Ap-
571 proach to the Synthesis of N,N-Disubstituted Melamines
572 (Triazine Triskelions) under Mild Reaction Conditions.
573 *Tetrahedron* **2009**, 65, 3573–3576.
574 21. Marcano, D. C.; Kosynkin, D. V.; Berlin, J. M.; Sinitskii, A.; Sun,
575 Z.; Slesarev, A.; Alemany, L. B.; Lu, W.; Tour, J. M. Improved
576 Synthesis of Graphene Oxide. *ACS Nano* **2010**, 4, 4806–
577 4814.
578 22. Ferrari, A. C.; Basko, D. M. Raman Spectroscopy as a
579 Versatile Tool for Studying the Properties of Graphene.
580 *Nat. Nanotechnol.* **2013**, 8, 235–246.
581 23. Dresselhaus, M. S.; Jorio, A.; Hofmann, M.; Dresselhaus, G.;
582 Saito, R. Perspectives on Carbon Nanotubes and Graphene
583 Raman Spectroscopy. *Nano Lett.* **2010**, 10, 751–758.
584 24. Cancado, L. G.; Jorio, A.; Martins Ferreira, E. H.; Stavale, F.;
585 Achete, C. A.; Capaz, R. B.; Moutinho, M. V. O.; Lombardo, A.;
586 Kulmala, T. S.; Ferrari, A. C. Quantifying Defects in Gra-
587 phene via Raman Spectroscopy at Different Excitation
588 Energies. *Nano Lett.* **2011**, 11, 3190–3196.
589 25. Jorio, A.; Lucchese, M. M.; Stavale, F.; Martins Ferreira, E. H.;
590 Moutinho, M. V. O.; Capaz, R. B.; Achete, C. A. Raman Study
591 of Ion-Induced Defects in N-Layer Graphene. *J. Phys.:
592 Condens. Matter* **2010**, 22, 334204.
593 26. Eckmann, A.; Felten, A.; Mishchenko, A.; Britnell, L.; Krupke,
594 R.; Novoselov, K. S.; Casiraghi, C. Probing the Nature of
595 Defects in Graphene by Raman Spectroscopy. *Nano Lett.*
596 **2012**, 12, 3925–3930.
597 27. Cancado, L. G.; Reina, A.; Kong, J.; Dresselhaus, M. S.
598 Geometrical Approach for the Study of G' Band in the
599 Raman Spectrum of Monolayer Graphene, Bilayer Gra-
600 phene, and Bulk Graphite. *Phys. Rev. B* **2008**, 77, 245408–
601 245417.
602 28. Xu, M.; Fujita, D.; Gao, J.; Hanagata, N. Auger Electron
603 Spectroscopy: A Rational Method for Determining Thick-
604 ness of Graphene Films. *ACS Nano* **2010**, 4, 2937–2945.
605 29. Cancado, L. G.; Takai, K.; Enoki, T.; Endo, M.; Kim, Y. A.;
606 Mizusaki, H.; Jorio, A.; Coelho, L. N.; Magalhaes-Paniago, R.;
607 Pimenta, M. A. General Equation for the Determination of
608 the Crystallite Size L_a of Nanographite by Raman Spec-
609 troscopy. *Appl. Phys. Lett.* **2006**, 88, 163106.
610 30. Lucchese, M. M.; Stavale, F.; Ferreira, E. H. M.; Vilani, C.;
611 Moutinho, M. V. O.; Capaz, R. B.; Achete, C. A.; Jorio, A.
612 Quantifying Ion-Induced Defects and Raman Relaxation
613 Length in Graphene. *Carbon* **2010**, 48, 1592–1597.
614 31. Lu, Y. H.; Chen, W.; Feng, Y. P.; He, P. M. Tuning the
615 Electronic Structure of Graphene by an Organic Molecule.
616 *J. Phys. Chem. B* **2009**, 113, 2–5.

- 617 32. Zhang, Y.-H.; Zhou, K.-G.; Xie, K.-F.; Zeng, J.; Zhang, H.-L.;
618 Peng, Y. Tuning the Electronic Structure and Transport
619 Properties of Graphene by Noncovalent Functionalization:
620 Effects of Organic Donor, Acceptor and Metal Atoms.
621 *Nanotechnology* **2010**, *21*, 065201.
- 622 33. Kozlov, S. M.; Vines, F.; Goerling, A. On the Interaction of
623 Polycyclic Aromatic Compounds with Graphene. *Carbon*
624 **2012**, *50*, 2482–2492.
- 625 34. Lazar, P.; Karlicky, F.; Jurecka, P.; Kocman, M.; Otyepkova, E.;
626 Safarova, K.; Otyepka, M. Adsorption of Small Organic
627 Molecules on Graphene. *J. Am. Chem. Soc.* **2013**, *135*,
628 6372–6377.
- 629 35. Brunetti, F. G.; Isla, H.; Arago, J.; Orti, E.; Perez, E. M.; Martin,
630 N. Exploiting Multivalent Nanoparticles for the Supramole-
631 cular Functionalization of Graphene with a Nonplanar
632 Recognition Motif. *Chem.—Eur. J.* **2013**, *19*, 9843–9848.
- 633 36. Becke, A. D. Density-Functional Exchange-Energy Approx-
634 imation with Correct Asymptotic-Behavior. *Phys. Rev. A*
635 **1988**, *38*, 3098–3100.
- 636 37. Burke, K. Perspective on Density Functional Theory.
637 *J. Chem. Phys.* **2012**, *136*, 150901.
- 638 38. Cohen, A. J.; Mori-Sanchez, P.; Yang, W. Challenges for
639 Density Functional Theory. *Chem. Rev.* **2012**, *112*, 289–320.
- 640 39. Grimme, S. Semiempirical GGA-Type Density Functional
641 Constructed with a Long-Range Dispersion Correction.
642 *J. Comput. Chem.* **2006**, *27*, 1787–1799.
- 643 40. Dion, M.; Rydberg, H.; Schroder, E.; Langreth, D. C.; Lundq-
644 vist, B. I. van der Waals Density Functional for General
645 Geometries. *Phys. Rev. Lett.* **2004**, *92*, 246401.
- 646 41. Eshuis, H.; Bates, J. E.; Furche, F. Electron Correlation
647 Methods Based on the Random Phase Approximation.
648 *Theor. Chem. Acc.* **2012**, *131*, 1084.
- 649 42. Ren, X.; Rinke, P.; Joas, C.; Scheffler, M. Random-Phase
650 Approximation and Its Applications in Computational
651 Chemistry and Materials Science. *J. Mater. Sci.* **2012**, *47*,
652 7447–7471.
- 653 43. Frisch, M. J.; Trucks, G. W.; Schlegel, H. B.; Scuseria, G. E.;
654 Robb, M. A.; Cheeseman, J. R.; Scalmani, G.; Barone, V.;
655 Mennucci, B.; Petersson, G. A.; et al. *Gaussian 09*, revision C;
656 Gaussian, Inc.: Wallingford, CT, 2009.
- 657 44. Vreven, T.; Morokuma, K. On the Application of the IMOMO
658 (Integrated Molecular Orbital Plus Molecular Orbital)
659 Method. *J. Comput. Chem.* **2000**, *21*, 1419–1432.
- 660 45. Vreven, T.; Morokuma, K.; Farkas, O.; Schlegel, H. B.; Frisch,
661 M. J. Geometry Optimization with QM/MM, ONIOM, and
662 Other Combined Methods. I. Microiterations and Con-
663 straints. *J. Comput. Chem.* **2003**, *24*, 760–769.
- 664 46. Senn, H. M.; Thiel, W. QM/MM Methods for Biomolecular
665 Systems. *Angew. Chem., Int. Ed.* **2009**, *48*, 1198–1229.
- 666 47. Parr, R. G.; Yang, W. *Density-Functional Theory of Atoms and*
667 *Molecules*; Oxford University Press: New York, 1994.
- 668 48. Kohn, W.; Becke, A. D.; Parr, R. G. Density Functional Theory
669 of Electronic Structure. *J. Phys. Chem.* **1996**, *100*, 12974–12980.
- 670 49. Ditchfie, R.; Hehre, W. J.; Pople, J. A. Self-Consistent
671 Molecular-Orbital Methods. Extended Gaussian-Type
672 Basis for Molecular-Orbital Studies of Organic Molecules.
673 *J. Chem. Phys.* **1971**, *54*, 724.
- 674 50. Hehre, W. J.; Ditchfie, R.; Pople, J. A. Self-Consistent Mo-
675 lecular-Orbital Methods. Further Extensions of Gaussian-
676 Type Basis Sets for Use in Molecular-Orbital Studies of
677 Organic-Molecules. *J. Chem. Phys.* **1972**, *56*, 2257.
- 678 51. Scuseria, G. E.; Janssen, C. L.; Schaefer, H. F. An Efficient
679 Reformulation of the Closed-Shell Coupled Cluster Single
680 and Double Excitation (CCSD) Equations. *J. Chem. Phys.*
681 **1988**, *89*, 7382–7387.
- 682 52. Grimme, S.; Muck-Lichtenfeld, C.; Antony, J. Noncovalent
683 Interactions between Graphene Sheets and in Multishell
684 (Hyper) Fullerenes. *J. Phys. Chem. C* **2007**, *111*, 11199–11207.
- 685 53. Wuest, J. D.; Rochefort, A. Strong Adsorption of Amino-
686 triazines on Graphene. *Chem. Commun.* **2010**, *46*, 2923–
687 2925.
- 688 54. Zhang, X.; Chen, T.; Chen, Q.; Wang, L.; Wan, L.-J. Self-
689 Assembly and Aggregation of Melamine and Melamine-
690 Uric/Cyanuric Acid Investigated by STM and AFM on Solid
691 Surfaces. *Phys. Chem. Chem. Phys.* **2009**, *11*, 7708–7712.
- 692 55. Llanes-Pallas, A.; Palma, C.-A.; Piot, L.; Belbakra, A.; Listorti,
693 A.; Prato, M.; Samori, P.; Armaroli, N.; Bonifazi, D. Engineer-
694 ing of Supramolecular H-Bonded Nanopolygons via Self-
695 Assembly of Programmed Molecular Modules. *J. Am.*
696 *Chem. Soc.* **2009**, *131*, 509–520.
- 697 56. Rochefort, A.; Wuest, J. D. Interaction of Substituted Aro-
698 matic Compounds with Graphene. *Langmuir* **2009**, *25*,
699 210–215.
- 700 57. Stewart, J. J. P. Optimization of Parameters for Semiempir-
701 ical Methods V: Modification of NDDO Approximations
702 and Application to 70 Elements. *J. Mol. Model.* **2007**, *13*,
703 1173–1213.