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# First-principal investigations of electronic, structural, elastic and optical properties of the fluoroperovskite $\text{TlF}_3$ ( $L = \text{Ca}, \text{Cd}$ ) compounds for optoelectronic applications

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In this research work, the Tl-based fluoroperovskite compounds  $\text{TlF}_3$  ( $L = \text{Ca}, \text{Cd}$ ) were investigated computationally using density functional theory (DFT) to comprehend their structural, elastic, optical, and electronic properties. Computation of the tolerance factor and Birch–Murnaghan curve indicated that the compounds are cubic and structurally stable. The structurally optimized lattice constants and the optimum volume corresponding to the optimum energy were measured. Elastic properties were predicted using the IRelast package, and the results showed that the compounds of interest are mechanically stable, ductile, and anisotropic in nature. The electronic properties (band structures and density of states) show that  $\text{TlCaF}_3$  and  $\text{TlCdF}_3$  possess a wide direct bandgap from ( $X-X$ ) symmetry points of 5.7 eV and 5.6 eV, respectively. The contributions of different elemental states to the valence and conduction bands are evaluated from the total and partial density of states (TDOS & PDOS). Analysis of the optical properties showed that these compounds possess a high refractive index, absorption coefficient, and reflectivity at high energy ranges. The values of the direct bandgap indicated that these compounds are expected to be semiconducting in nature, and their use is primarily considered to be in the semiconductor industries and optoelectronic devices. These compounds are new and have been investigated for the first time using the computational approach, which provides comprehensive insight into their different properties; based on the results, they are recommended as industrial candidates.

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## 1. Introduction

Fluoroperovskites are well-known compounds with the typical chemical formula  $\text{ABF}_3$ , where A and B are cations (positive ions), and fluorine (F) is an anion (negative charge). Fluoroperovskite materials are an intriguing family, having a crystal structure that is mechanically stable while exhibiting excellent

optoelectronic properties ranging from semiconducting (1–4 eV) to insulating (above 4 eV) nature. Fluoroperovskites have garnered substantial attention in current years because of their industrial significance to lenses for optical lithography, radiation dosimeters, scintillation materials, and the semiconductor processing industry.<sup>1–4</sup> Due to their wide range of applications, fluoroperovskite compounds have been extensively investigated experimentally and computationally by a variety of authors, as detailed in ref. 5. Some investigations of fluoroperovskites under high pressure have been reported. It has been found that the materials are indirect bandgap insulators and have a stable equilibrium lattice constant, elastic constants, and bulk modulus.<sup>6–8</sup>

The energy band gap of fluoropyrocytes is often large, so these compounds are technologically important.<sup>9–11</sup>  $\text{KMgF}_3$  and  $\text{BaLiF}_3$  are utilized in optical lithography steppers as vacuum-ultraviolet materials for lenses<sup>11–13</sup> and when doped with lanthanide ions Ce and Er,  $\text{KMgF}_3$  also shows promise as a material for radiation dosimeters and scintillation.<sup>14</sup>

Study of the compound  $\text{SrCl}_2$  has found it to be an insulator with an indirect bandgap of 5.18 eV. The bandgap initially

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increases and then decreases as a function of pressure. The structural and electronic properties and elastic constants of cubic SrCl<sub>2</sub> also show its cubic structure.<sup>15</sup> Total energy and elastic constants, as well as the theoretical Young's modulus, shear modulus, Poisson's ratio, sound velocities, and Debye temperature, were calculated and examined. The bulk and shear moduli ratios reveal that Ti<sub>3</sub>AlN is ductile in nature, with its ductility expected to be greater than that of Ti<sub>3</sub>Al, but Ti<sub>3</sub>AlC is brittle.<sup>16</sup> A theoretical investigation of the Ag-based fluoroperovskites AgMgF<sub>3</sub> and AgZnF<sub>3</sub> was described by Murtaza *et al.*<sup>17</sup> The expected wide absorption energy range of the proposed materials makes them appropriate for various applications in modern devices. The optoelectronic characteristics of Sn-based fluoroperovskites were investigated; these compounds were discovered to be electronic insulators, and their Auger-Free luminescence was predicted (AFL).<sup>18</sup> The optoelectronic, structural, and magnetic properties of TMnX<sub>3</sub> (X = Cl, and F) were described by F. Hamioud, and his colleagues projected optical-spectrum-based uses for optics technology in ref. 19 and 20. Thallium-based compounds are being developed for radiation detection, and several investigations have shown their usefulness in this application.<sup>21,22</sup> Because these compounds contain a thallium atom, their effective atomic number is higher, which improves detection efficiency; because of their single growth requirement, the simple cubic structure of the compounds makes them a technologically promising candidate.

In this work, TlLF<sub>3</sub> (L = Ca, Cd) compounds are studied in depth utilizing the Tran and Blaha modified Becke–Johnson (TB-mBJ) technique, which provides insight into their structural, elastic, electrical, and optical properties. Using the computer simulation tool wien2k, we have discovered that TlCdF<sub>3</sub> is electrically insulating with high transparency over a wide range of energies, making it a good candidate for optical applications. To the best of our knowledge, there are no reports in the literature of research into Tl-based fluoroperovskites despite the interest in fluoroperovskite compounds for diverse applications.

There are four main components to this study. Section 2 covers the calculation methods, Section 3 covers the results and discussion, and Section 4 covers the conclusions of the study.

## 2. Methodology

The ternary fluoroperovskite TlLF<sub>3</sub> (L = Ca, Cd) compounds are characterized as cubic-structure-type perovskites with the space group *Pm* $\bar{3}$ *m* (#221). One molecule makes up the unit cell, and Tl and L (L = Ca, Cd) are located at Wyckoff coordinates of (0, 0, 0) and (0.5, 0.5, 0.5), respectively, while the three fluorine atoms in the molecule are located at (0.5, 0, 0.5), (0, 0.5, 0.5) and (0.5, 0.5, 0), as depicted (Fig. 1).

The typical structure of DFT was used for simulation studies to investigate their structural, elastic, electrical, and optical properties. In order to use the full potential linearized augmented plane wave (FP-LAPW) approach to its maximum extent, the WIEN2K code was used.<sup>23</sup> The energy *versus* volume curve was fitted using Murnaghan's equation of state to determine structural properties.<sup>24</sup> To ensure that there was no

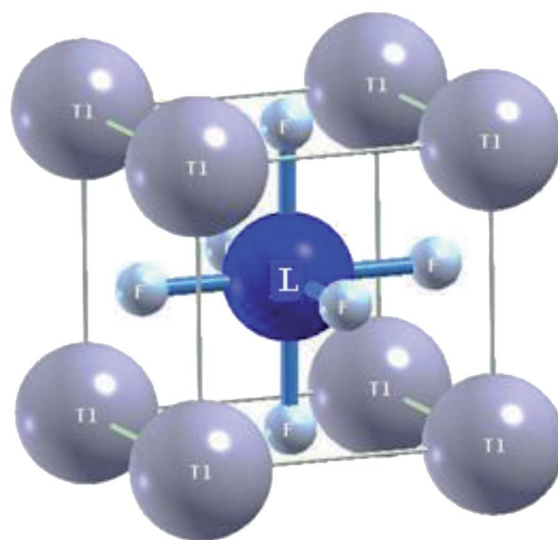


Fig. 1 Unit cell structure of TlLF<sub>3</sub> (L = Ca, Cd) fluoroperovskites.

charge leakage from either the total or the core energy, a suitable radius of muffin tin (RMT) value was selected during the investigation. The RMT values were 2.50, 2.01, and 2.12 for Tl, Ca, and Cd, respectively, while they were 2.01 and 1.88 for the F in TlCaF<sub>3</sub> and TlCdF<sub>3</sub>, respectively. There are 1000 K-points and 12 *G*<sub>max</sub> in the muffin tins; hence, the wave function was stretched in harmonics up to *l*<sub>max</sub> = 10 in the muffin tin spheres. The energy difference between the core and valence bands is 6 Ry.

## 3. Results and discussion

### 3.1 Electronic properties

The electronic properties of the compounds under investigation are shown with their energy, band structures and DOS. Fig. 2 shows that the computed band structure for TlLF<sub>3</sub> (L = Ca, Cd) at the equilibrium geometry falls in the first Brillouin zone, as well as the symmetry directions in that zone. There is no such thing as a zero; ideally, energy should be at or near the maximum point of the valence band. The bandgaps for the compounds TlCdF<sub>3</sub> and TlCaF<sub>3</sub> were calculated to be 5.7 and 5.6 eV using the TB-mBJ potential method. In the band structure, the conduction band minima of TlCdF<sub>3</sub> occur at the *X*-symmetry point and the maximum of the valence band Fig. 2(a), showing the direct (*X*–*X*) band nature by being at the *X* symmetry point, and is shown in Fig. 2(b). The conduction band minimum and a maximum of TlCaF<sub>3</sub> are both found at the exact *X*-axis symmetry location of the Brillouin zone point, which shows the direct bandgap nature. The band gap is not direct according to previous research; however, its nature is in agreement with TlCdF<sub>3</sub>, and the band gap properties of the proposed compound TlCaF<sub>3</sub> are of interest. The calculated band structure of TlLF<sub>3</sub> (L = Ca, Cd) along the symmetry directions of the equilibrium geometry in the first Brillouin zone are given in Fig. 2. The zero energy is set to correspond to the top of the valence bands. Using TB-mBJ, the values predicted for the



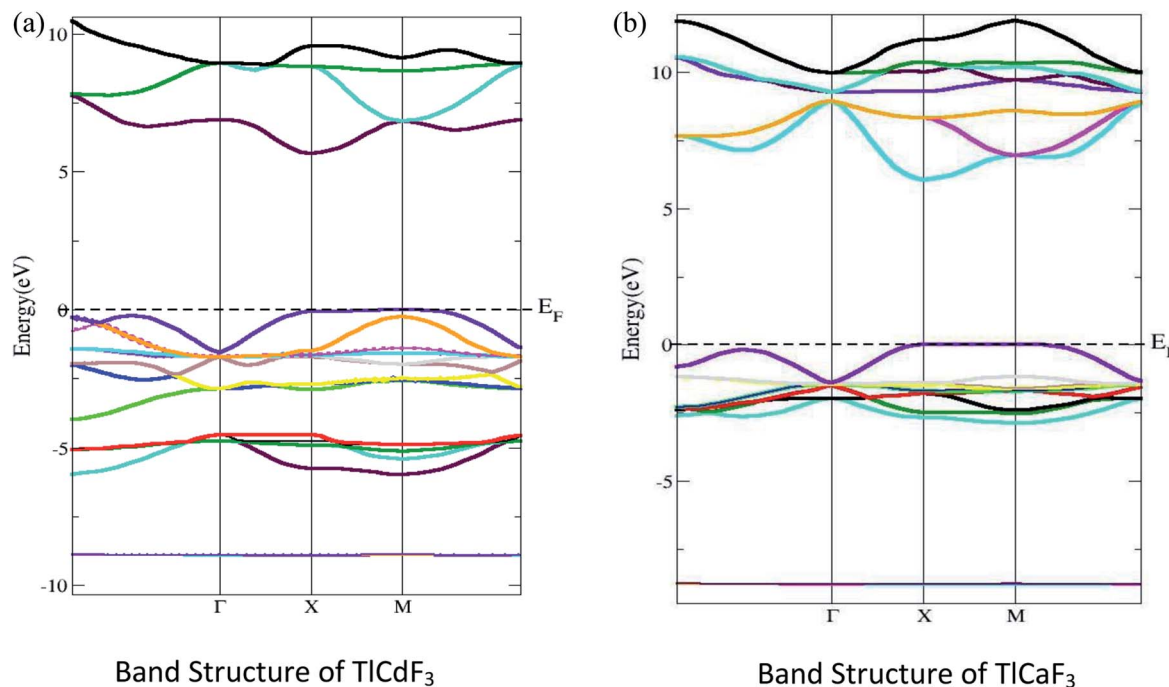


Fig. 2 (a) Computed band structures of TlCdF<sub>3</sub> and (b) TlCaF<sub>3</sub>.

Table 1 Computed bandgap energies  $E_g$  (eV) at high symmetry points for TlF<sub>3</sub> (X = Ca, Cd) using TB-mBJ potential

Compound	$E_g^{M-M}$	$E_g^{X-X}$	$E_g^{\Gamma-\Gamma}$	$E_g^{R-\Gamma}$	$E_g^{\Gamma-X}$
TlCdF <sub>3</sub>	5.24	5.7	6.94	5.82	6.69
TlCaF <sub>3</sub>	5.21	5.6	3.46	6.19	3.23

bandgaps of TlCdF<sub>3</sub> and TlCaF<sub>3</sub> are 5.7 eV and 5.6 eV, respectively, as shown in Table 1. The direct bandgap nature of TlCaF<sub>3</sub> is demonstrated in Fig. 2(a), in which both the CB minima and VB maxima are positioned at the same symmetry point, X, within the Brillion zone. The band structure of TlCaF<sub>3</sub> is depicted in Fig. 2(b), which shows that the CB minima and VC maxima lie at the M-symmetry point, indicating a direct band gap nature. The direct band gap nature is consistent with prior research on TlCdF<sub>3</sub>.<sup>20</sup>

### 3.2 The DOS (density of states)

The TDOS and PDOS (total and partial density of states) for both materials under investigation are presented in Fig. 3, which displays the contributions of the electronic states to the VB and CB bands. The Fermi level is shown as the vertical dotted line labelled  $E_F$ . The portion to the left of the Fermi level represents the VB, while the CB region is to the right. A deeper understanding is obtained from the different states of the constituent materials. The plot of the DOS for TlCaF<sub>3</sub> demonstrates that the VB is dominated by the F-p state with a slight contribution from the Tl-s state. In CB, the Tl-s state plays the key role. For TlCdF<sub>3</sub>, the VB is dominated by the F-p and Cd-d states, while the states of Cd dominate the CB.

### 3.3 Structural and elastic properties

The elastic constants  $C_{ij}$ , which are fundamental and indispensable, can be used to characterize elastic properties. The elastic constants are significant characteristics in determining the reaction when macroscopic stress is applied. The elastic constants can be used to describe how a material endures stress deformation and subsequently reinstates and reverts to its original form when the stress has passed.<sup>25</sup> The elastic constants can provide a valuable description of structural stability and bonding character within the atomic spheres in addition to the isotropic or anisotropic character.  $C_{11}$ ,  $C_{44}$ , and  $C_{12}$  are three discrete and self-governing elastic constants for cubic systems. To determine  $C_{ij}$ , the unit cell is slanted with a suitable strain tensor to obtain an energy-strain relationship. In the present work, the IRelast-package developed by Jamal Murtaza,<sup>26</sup> which is used and run within the WIEN2k code, was applied. Fig. 1 shows the ideal cubic crystal structures of the Tl-based fluoroperovskites in this work. Fitting the Birch-Murnaghan equation yields the equilibrium lattice constants.<sup>27</sup> Table 2 contains these values. The calculated lattice constant (4.3306 Å) for TlCdF<sub>3</sub> agrees reasonably well with the experimentally determined value (4.325 Å) in ref. 28.

When forces are applied to a crystal, its response is computed by its elastic constant, which reveals a lot about its mechanical properties. The elastic properties of cubic symmetries crystal, such as rigidity and stability, are calculated using  $C_{11}$ ,  $C_{12}$ , and  $C_{44}$  (three self-governing elastic constants).

Table 3 displays the calculated values of the elastic constants  $C_{ij}$ . The elastic constants found in ref. 29 for TlCdF<sub>3</sub> were compared with the present results. There is a good match between our computed values of  $C_{11}$ ,  $C_{12}$  and  $C_{44}$  for TlCdF<sub>3</sub>



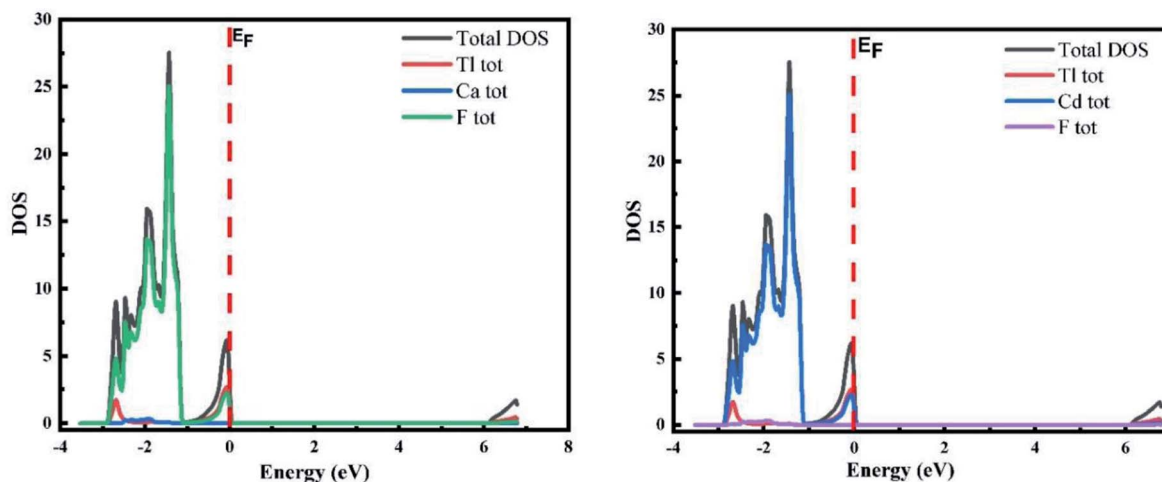


Fig. 3 TDOS and PDOS of ternary  $\text{TlLF}_3$  ( $L = \text{Ca}, \text{Cd}$ ) fluoroperovskites.

(114.34 GPa, 32.57 GPa, and 15.08 GPa) and the reported values (103.6 GPa, 39.6 GPa, and 18.08 GPa), while those for  $\text{TlCaF}_3$   $C_{11}$ ,  $C_{12}$  and  $C_{44}$  were 220.81 GPa, 40.32 and 30.28 GPa in ref. 30.

The elastic constants for  $\text{TlCdF}_3$  were also compared with the values obtained through DFT<sup>20</sup> using the generalized gradient approximation of Wu and Cohen (GGA-WC) for  $C_{11}$ ,  $C_{12}$ , and  $C_{44}$  (124.7 GPa, 39.59 GPa, and 11.53 GPa). In the case of  $\text{TlCdF}_3$ , the evaluation indicates that the TB-mBJ method delivers complete additional precise elastic properties closer to the experimental results than the GGA-WC approach.

Using established relationships, the ( $B$ ) bulk modulus may be determined from the elastic constant (ECs). All the ECs are positive and meet the requirements. For elastic stability,  $C_{11} > 0$ ;  $C_{44} > 0$ ;  $(C_{11} - C_{12}) > 0$ ;  $(C_{11} + 2C_{12}) > 0$ ;  $C_{12} > B > C_{11}$ .<sup>31</sup> Table 3 lists the resulting Young's modulus ( $E$ ), anisotropy factor ( $A$ ), Poisson ratio ( $\nu$ ), and Pugh index ratio ( $B/G$ ) values.<sup>30</sup> The  $G$  (shear modulus) signifies resistance to plastic deformation, whereas the bulk modulus  $B$  represents resistance to fracture. The  $B/G$  ratio determines whether a material is brittle or ductile.<sup>32</sup> If  $B/G$  is less than 1.75, the material is brittle, and if it is greater than 1.75, it has a ductile character. According to  $B/G$  (Pugh's criteria), both the materials are ductile. The Poisson ratio  $\nu$ <sup>33</sup> determines the brittleness of the material; if  $\nu$  is greater than 0.26, the material is ductile; else, it is brittle. Table 3 lists the values. " $A$ " is the elastic anisotropy factor, which is equal to one for an anisotropic material, with any number other than one indicating anisotropy. The degree of elastic anisotropy in the crystal is determined by the amount of deviation from 1. The value of  $A$  differs from one for all our compounds, indicating that they all have anisotropy.

Young's modulus ( $E$ ) is the best indicator of the stiffness of a material. When it is high for a given substance, the material becomes stiffer; both compounds are stiff.

The Poisson ratio, which indicates the bonding power, provides more data related to the uniqueness of the bonding power than some other elastic attributes. The Poisson ratio ( $\nu$ ) for covalent materials is very small (0.1), but the usual value for ionic materials is 0.25. Table 3 shows that the value for all compounds is more than 0.25, indicating that the nature of the bonding in all our materials is ionic.

### 3.4 Optical properties

All optical property calculations were carried out using the TB-mBJ potential approach. The optical properties of solids can be investigated by characterizing their electronic complex dielectric function, *i.e.*,  $\epsilon(\omega)$ . Intra-band and inter-band transitions play a significant role in this complicated function. Inter-band contributions are significant for metals<sup>34</sup> and can be divided into two types: direct transitions and indirect transitions. We disregard indirect inter-band transitions owing to their modest contribution to  $\epsilon(\omega)$  and their phonon-scattering character. The dielectric function  $\epsilon(\omega)$  for anisotropic materials is essentially a complicated kind of symmetrically second-order tensor that perfectly captures the linear response of the electronic system when an external electric field is applied. The complex dielectric function defined by Ehrenreich and Cohen's equation can be used to describe the optical characteristics of the compound.<sup>24</sup>

$$\epsilon(\omega) = \epsilon_1(\omega) + i\epsilon_2(\omega) \quad (1)$$

Table 2 Computed optimized structural properties of  $\text{TlLF}_3$  ( $L = \text{Cd}, \text{Ca}$ ) from the energy vs. volume results fixed by the Birch–Murnaghan technique

Compounds	$a_0$ (lattice constant in Å)	$B$ (bulk modulus in GPa)	$B'$ (derivative of bulk modulus in GPa)	$V_0$ (ground state volume in a.u. <sup>3</sup> )	$E_0$ (ground state energy in Ry)
$\text{TlCdF}_3$	4.33	79.64	3.41	548.13	-52333.15
$\text{TlCaF}_3$	4.37	63.74	3.48	563.83	-42509.21



Table 3 Simulated mechanical parameters of TILF<sub>3</sub> (L = Ca, Cd) using the IRelast package

Compounds	$C_{11}$	$C_{12}$	$C_{44}$	$B$	$G$	$E$	$A$	$\nu$	$B/G$
TlCdF <sub>3</sub>	114.34	32.57	15.08	172.03	22.78	72.631	0.369	0.43	7.549
TlCaF <sub>3</sub>	220.81	40.32	30.28	172.03	-346.54	5053.076	-0.016	-4.39	-0.96

Here,  $\varepsilon_1(\omega)$  and the  $\varepsilon_2(\omega)$  give the real and imaginary parts, respectively. One can calculate the real part, which can be derived from the imaginary part of the dielectric function and is dependent on  $\omega'^2 - \omega^2$ , which gives the integral denominator; additionally,  $\omega'$  has a direct relationship with  $E_g$ . The derivation of the real part  $\varepsilon_1(\omega)$  can be made by implementing the Kramers–Kronig expression:

$$\varepsilon_1(\omega) = 1 + \frac{2}{\pi} P \int_0^\infty \frac{\omega' \varepsilon_2(\omega')}{(\omega'^2 - \omega^2)} d(\omega') \quad (2)$$

From the real part, the imaginary part can be taken out using the Kramers–Kronig relationship:

$$\varepsilon_2(\omega) = \frac{8}{2\pi\omega} \sum_{m'} \int |P_{m'}(k)|^2 \frac{dS_k}{\nabla\omega_{m'}(k)} \quad (3)$$

For optical response, the photon energy range is 0–30 eV. The dispersive actions are shown from the real part of the surface of the material, and the absorption of light is shown from the imaginary part of the material.<sup>35</sup> The optical transitions and absorption of light within the energy bands are found from the imaginary part. Other optical parameters, comprising the refractive index, absorption coefficient, extinction coefficient, optical conductivity, and reflectivity, can be computed from the dielectric functions.<sup>36</sup>

**3.4.1 Refractive index.** The refractive index can be computed by means of equations.<sup>37</sup> In Fig. 4, we see that the refractive indices of the compounds TlCdF<sub>3</sub> and TlCaF<sub>3</sub> initially have approximately constant values and have peaks at about 6.4 and 6.8 eV. TlCdF<sub>3</sub> and TlCaF<sub>3</sub> have the highest refractive index peak ( $n$ ) is 6.3 at 6.4 eV, and the minimum refractive index ( $n$ ), is 13.50 eV each. Another study showed that at zero photon energy, the value of the static refractive index  $n(0)$  is approximately equal to 0 for many compounds. Minor peaks are also detected near 12 eV for the TlCaF<sub>3</sub> compound and at about 13 eV for TlCdF<sub>3</sub>. Their refractive indices show similar behavior.

**3.4.2 Absorption coefficient.** The absorption coefficient for each material offers information about how these compounds would behave when exposed to radiation. The frequency has a substantial impact on the interactions of the incoming photons with electrons based on the absorption coefficient and induces electrons to flow from the VB to the CB. The absorption coefficient indicates the ability of a material to absorb incident photons of a specific energy.<sup>28</sup> The absorption coefficients comprise components from both the real and imaginary parts of the dielectric functions. The absorption coefficient is given as follows:

$$I(\omega) = \sqrt{2\omega} \left[ \sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)} - \varepsilon_1(\omega) \right]^{1/2} \quad (4)$$

The calculated absorption coefficients of TlCdF<sub>3</sub> and TlCaF<sub>3</sub> are plotted in Fig. 5. The absorption part of the TlCdF<sub>3</sub> spectrum starts at 5.8 eV, and the maximum value for the compound occurs at 8.4 eV, as can be seen in Fig. 5. The absorption part of the TlCaF<sub>3</sub> spectrum starts at 5.6 eV, and the maximum absorption appears at 8.5 eV. The energy range of the incident photons for the ultraviolet region is 3 to 124 eV. The absorption coefficient values increase rapidly and reach their maximum when the energy of the radiation reaches a high level; that is, the absorption of the compounds begin, and it becomes high. The very sharp cut-off response, which occurs primarily in the high-energy region, indicates and confirms that these materials can be employed as possible materials for optoelectronic devices working in the UV range.

**3.4.3 Reflectivity.** Reflectivity is the property of a compound that describes that it reflects that, when light hits it. How much of the radiation will fall on the surface of those compounds? When it reaches a certain quantity, then these quantity will revert to its previous state. This is indicated by the term  $R(\omega)$  in optical physics. From the dielectric function's imaginary part, the reflectivity is calculated.

$$\delta(\omega) = \frac{2w_{ev}\hbar\omega}{E_0} \quad (5)$$

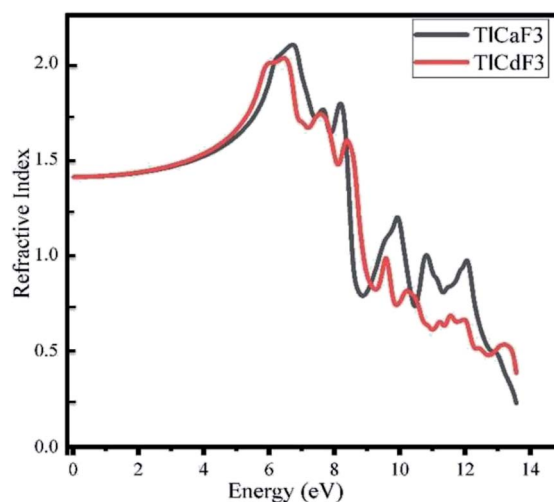


Fig. 4 Calculated refractive index  $n(\omega)$  for the TILF<sub>3</sub> (Ca, Cd) compounds.



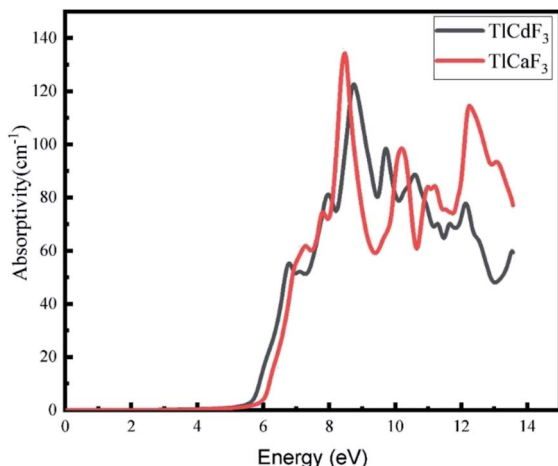


Fig. 5 Computed  $I(\omega)$  (absorption coefficient) for the fluoroperovskite  $\text{TlF}_3$  ( $L = \text{Ca}, \text{Cd}$ ) compounds.

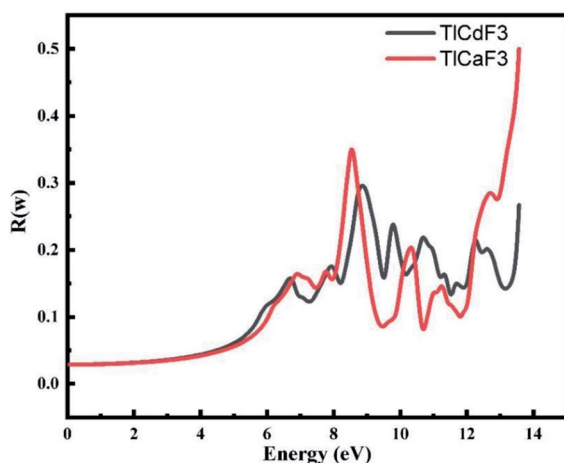


Fig. 6 Calculated reflectivity  $R(\omega)$  for the  $\text{TlF}_3$  ( $\text{Ca}, \text{Cd}$ ) compounds.

Fig. 6 demonstrates that low optical reflectivity is observed in the low-energy region up to 4 eV before the high energy region. High peaks at 13.8 eV occur for both compounds. Fig. 6 shows the static reflectivity  $R(\omega)$ , which is dependent on the frequency of the radiation or rays, for our two perovskite compounds  $\text{TlCdF}_3$  and  $\text{TlCaF}_3$ , which both have scores of 0.04, respectively. The maximum reflectivity  $R(\omega)$  scores for  $\text{TlCdF}_3$  and  $\text{TlCaF}_3$  are 0.5, 0.3, respectively, indicating that these materials have a low energy area. This is another important aspect to discuss, because it relates to the optical component. The low score of this term, particularly in the visible and infrared regions, indicates that these compounds have transparency in these regions, which is favorable because scientists use them as anti-reflecting coatings.

## 4. Conclusions

In the current work, the structural, optical, elastic, and electrical characteristics of  $\text{TlF}_3$  ( $L = \text{Ca}, \text{Cd}$ ) fluoroperovskites were

studied using the TB-mBJ potential approximation. The lattice constants at equilibrium were discovered to be between 4.15 Å and 4.94 Å. The inelastic properties, ECs, anisotropy factors, bulk moduli, Poisson's ratios, Young's moduli, and Pugh's ratios all indicate elastic characteristics. The Pugh ( $B/G$ ) ratio reveals that the investigated compounds are ductile. The obtained Poisson ratio values further support their ductile nature.  $\text{TlCaF}_3$  has direct bandgap behavior at the  $X$ -symmetry point, whereas  $\text{TlCdF}_3$  has an indirect band nature, according to our predictions. The predicted findings were compared to existing experimental and theoretical data and found to be consistent. The investigated compounds are exciting materials for scintillation detectors. They have simple cubic structures, large effective  $Z$  (atomic number) values due to the presence of the element thallium, and bandgaps within the range for insulator materials.

## Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have influenced the work reported in this paper.

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