

# SCIENTIFIC REPORTS

OPEN

## Fully transparent high performance thin film transistors with bilayer ITO/Al-Sn-Zn-O channel structures fabricated on glass substrate

Yingying Cong<sup>1</sup>, Dedong Han<sup>1</sup>, Junchen Dong<sup>1</sup>, Shengdong Zhang<sup>1,2</sup>, Xing Zhang<sup>1</sup> & Yi Wang<sup>1</sup>

In this work, fully transparent high performance double-channel indium-tin-oxide/Al-Sn-Zn-O thin-film transistors (ITO/ATZO TFTs) are successfully fabricated on glass by radio frequency (RF) magnetron sputtering. The ITO layer acts as the bottom channel layer to increase the channel carrier concentration. The top ATZO channel layer, which is deposited via high oxygen partial pressure in the sputtering process, is useful to control the minimum off-state current. After annealing, the ITO/ATZO TFT demonstrates outstanding electrical performances, including a high ON/OFF current ratio ( $I_{on}/I_{off}$ ) of  $3.5 \times 10^8$ , a steep threshold swing (SS) of 142.2 mV/decade, a superior saturation mobility ( $\mu_{sat}$ ) of 246.0 cm<sup>2</sup>/Vs, and a threshold voltage  $V_T$  of 0.5V. The operation mechanisms for double-channel structures are also clarified.

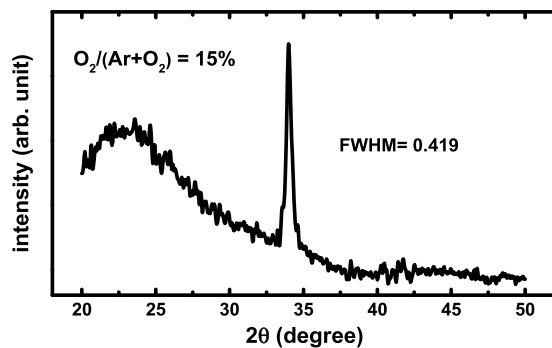
Transparent electronics is becoming a hot research topic, and the metal oxide thin film transistor is one of the key technologies<sup>1,2</sup>. Owing to the high mobility, low-temperature processing, transparency, and low cost, metal oxide thin-film transistors (TFTs) have become a promising candidate for driving Active-matrix organic light emitting diode (AMOLED) display panels and other high-performance display applications<sup>3-6</sup>. A series of the novel materials have been recommended as the channel materials to optimize the performances of the TFTs, including In-Ga-Zn-O (IGZO)<sup>7-9</sup>, Al-Zn-O<sup>10</sup>, Hf-In-Zn-O (HIZO)<sup>11</sup>, Al-Sn-Zn-In-O<sup>12</sup>, and so on. We have reported high electrical performances TFTs with Al-Sn-Zn-O (ATZO) channel materials, as the element Sn<sup>4+</sup> is helpful to increase the material mobility for the special electronic configuration of  $(n-1)d^{10}ns^0$  ( $n \geq 4$ )<sup>13</sup> and the element Al enhances the chemical bonds with oxygen<sup>14</sup> and improves the electrical characteristics at low temperature<sup>15</sup>.

Based on the novel ATZO TFT, modifying the device structures is an effective and convenient method to further improving the electrical performances. For example, we have improved the electrical performances of the ATZO TFTs with the double-channel structures in a previous work<sup>16</sup>. No complicated process is added to the fabrication flow. The double-channel structures are grown by successively depositing bilayer ATZO films with different oxygen partial pressures in the sputtering process. However, with the same kind of the channel materials, the double channel ATZO TFTs just keep the better performances of the single-channel TFTs without further improvements. Therefore in this letter, we adopt the indium-tin-oxide (ITO) material as the bottom channel layer to increase the channel carrier concentration and maximum channel current. The ATZO layer with higher channel resistivity is used as the top channel layer to control the minimum off-state current. Double-channel ITO/ATZO TFTs with excellent performances are achieved, and the operation mechanisms for double-channel structures are clarified.

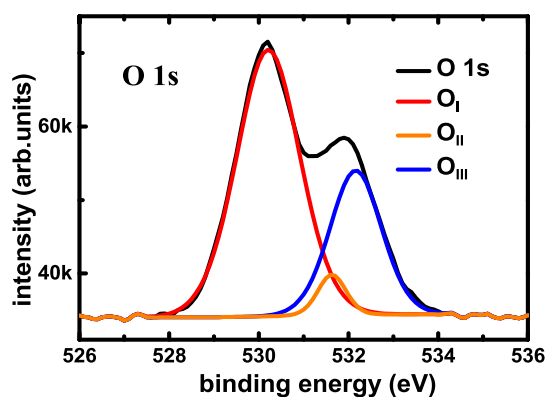
### Results and Discussion

The XRD spectra of the ATZO film with 15% oxygen partial pressures are shown in Fig. 1. The test sample was fabricated via the simultaneous deposition of channel layers in the sputtering chamber on glass. The spectra show that the ATZO film is polycrystalline in the hexagonal wurtzite structure with the preferred c-axis orientation. The extracted full width half maximum (FWHM) of the ATZO sample is 0.419. The grain size of the film is calculated by the Scherrer formula:

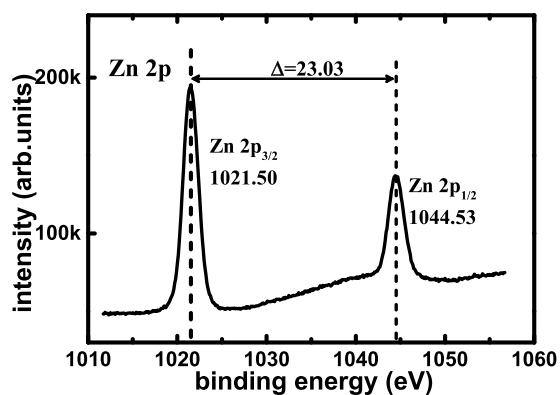
<sup>1</sup>Institute of Microelectronics, Peking University, Beijing, 100871, China. <sup>2</sup>Shenzhen Graduate School, Peking University, Shenzhen, 518055, China. Correspondence and requests for materials should be addressed to D.H. (email: [handedong@pku.edu.cn](mailto:handedong@pku.edu.cn))



**Figure 1.** XRD scan spectra of the ATZO films with 15% oxygen partial pressure. The thickness of the films is 40 nm.



(a)



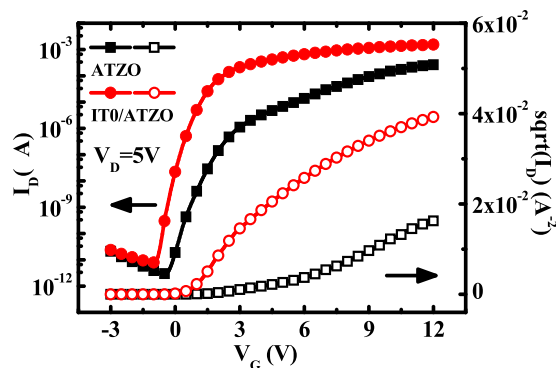
(b)

**Figure 2.** XPS scan spectra of ATZO films. (a) O 1s and (b) Zn 2p spectra of the ATZO films with oxygen partial pressure of 15%.

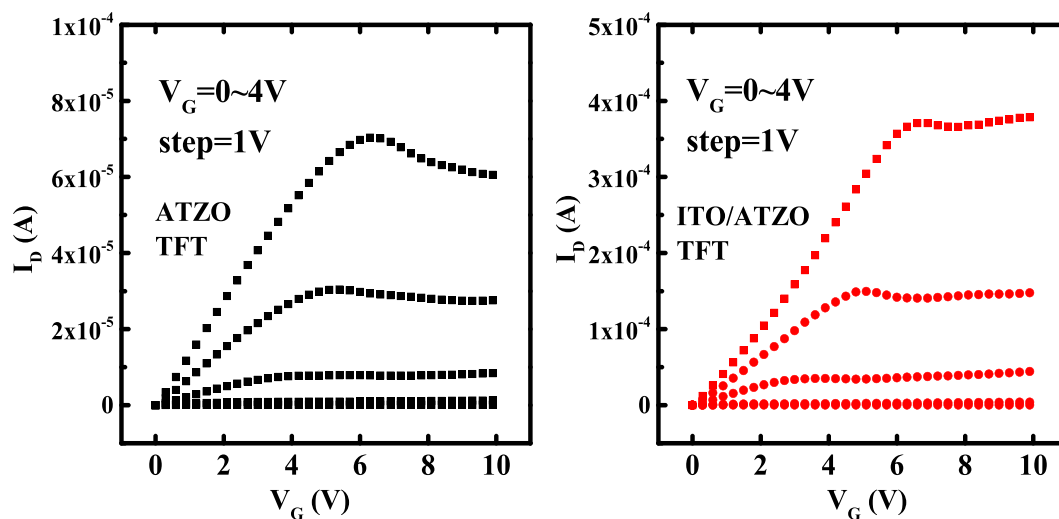
$$D = \kappa\lambda/\beta \cos \theta \quad (1)$$

where,  $D$  is the mean grain size,  $\kappa = 0.9$  is the dimensionless shape factor,  $\lambda$  is the X-ray wavelength,  $\beta$  is equal to the FWHM, and  $\theta$  is the diffraction angle. The estimated grain size of the ATZO film is 19.5 nm.

The O 1s XPS scan spectra of the ATZO film are shown in Fig. 2(a). To clarify the oxygen chemical bonds in the ATZO films, the asymmetric O 1s peaks are divided into three peaks, which are centered at 530.2 eV ( $O_I$ ), 531.6 eV ( $O_{II}$ ), and 532.2 eV ( $O_{III}$ ). The  $O_I$ ,  $O_{II}$ , and  $O_{III}$  peaks are generally attributed to  $O^{2-}$  bonded by metal ions (Zn–O, Al–O, and Sn–O),  $O^{2-}$  in the oxygen-deficient region (such as  $V_O$ ), and chemisorbed oxygen at grain boundaries or the surface of the film (such as metal hydroxide and oxy-hydroxide oxygen), respectively<sup>17, 18</sup>.



**Figure 3.** Transfer characteristics of the single-channel ATZO and double-channel ITO/ATZO TFTs.  $V_D = 5$  V.



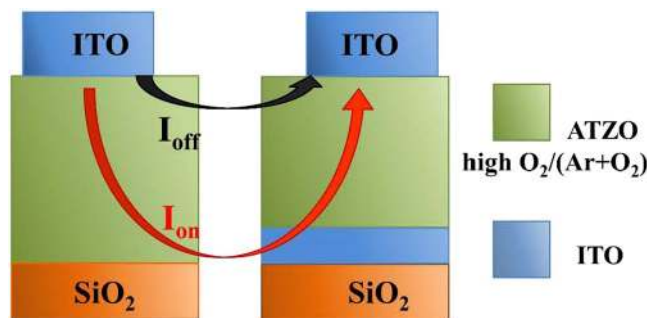
**Figure 4.** Output characteristics of (Left) single-channel ATZO and (Right) double-channel ITO/ATZO TFTs.

The relative quantities of the lattice oxygen, the  $V_O$ , and surface hydroxyl group can be achieved from the peak area ratio of the three peaks ( $O_p$ ,  $O_{II}$ , and  $O_{III}$ ). The relative  $O_{II}$  area ratio of the ATZO film is 4.47%. Figure 2(b) shows the Zn 2p peaks of the ATZO film. Due to the spin orbit split, the Zn 2p spectra have Zn  $2p_{1/2}$  and  $2p_{3/2}$  peak centered at 1021.50 eV and 1044.53 eV, respectively. It is generally accepted that the Zn  $2p_{3/2}$  peak centered at 1021.4 eV represents Zn interstitial in the oxygen-deficient regions<sup>19</sup>. The Zn interstitial and the  $V_O$  generate free carriers in n-type metal-oxide semiconductors.

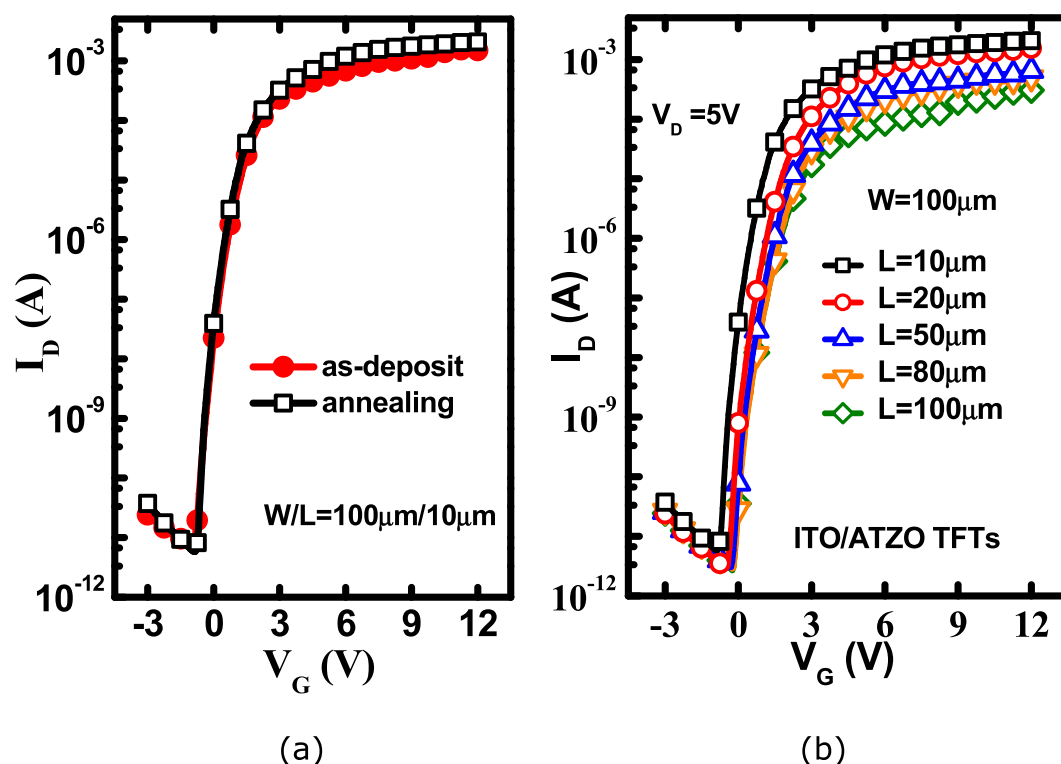
Figure 3 shows the transfer curves of the single-channel ATZO and double-channel ITO/ATZO TFTs. Because the high oxygen partial pressure suppresses the oxygen vacancies in the channel film, the ATZO channel has low carrier concentrations. The single-channel ATZO TFT demonstrates a low off-state current ( $I_{off}$ ), a low on-state current ( $I_{on}$ ), an on-to-off current ratio ( $I_{on}/I_{off}$ ) of  $8.9 \times 10^7$ , a  $V_T$  of 2.9 V, and a large subthreshold swing (SS) of 356.6 mV/dec. The extracted  $\mu_{sat}$  is  $26.6 \text{ cm}^2/\text{Vs}$ . To enhance the electrical characteristics of single-channel ATZO TFT, an ITO layer is inserted between the gate insulator and the ATZO channel and acts as the bottom channel. Due to the high-conductivity ITO film, the device performances is improved significantly compared with the single-channel ATZO device. The  $I_{on}$  obviously increases and the maximum value is an order of magnitude higher than that of the single-channel ATZO TFT. The  $V_T$  shifts to the left axis due to the increased carrier concentration in the total channels. The  $I_{off}$  increase slightly and the SS becomes steeper. The output curves of the single-channel ATZO and double-channel ITO/ATZO TFTs are shown in Fig. 4, respectively. Both the devices demonstrate n-type transistor characteristics.

The operation mechanism for double-channel TFTs can be explained by the schematic current paths shown in Fig. 5. In the on-state, the bottom ITO channel offers more free electrons and decreases the total channel resistance, which increases  $I_{on}$  and  $\mu_{sat}$  and decreases  $V_T$ , respectively. In the off-state, because the thick top ATZO channel layer with higher resistance controls the off-state performances, the double-channel ITO/ATZO TFT demonstrates a low  $I_{off}$ . Therefore, the  $I_{off}$  does not increase as much as the  $I_{on}$ , which is not influenced by the addition of high-conductivity ITO film. Thus, the ITO/ATZO TFT demonstrate a superior  $I_{on}/I_{off}$  of  $1.9 \times 10^8$  and the electrical performances of the ITO/ATZO TFT provides a clear evidence for the operation mechanism of the double-channel TFTs.

Figure 6(a) shows the changes in the electrical performances of the double-channel ITO/ATZO TFT after thermal annealing in a vacuum at 150 °C for 30 min,  $V_D = 5$  V. The device demonstrates superior performances



**Figure 5.** Schematic operation mechanism for (Left) single-channel ATZO and (Right) double-channel ITO/ATZO TFTs.



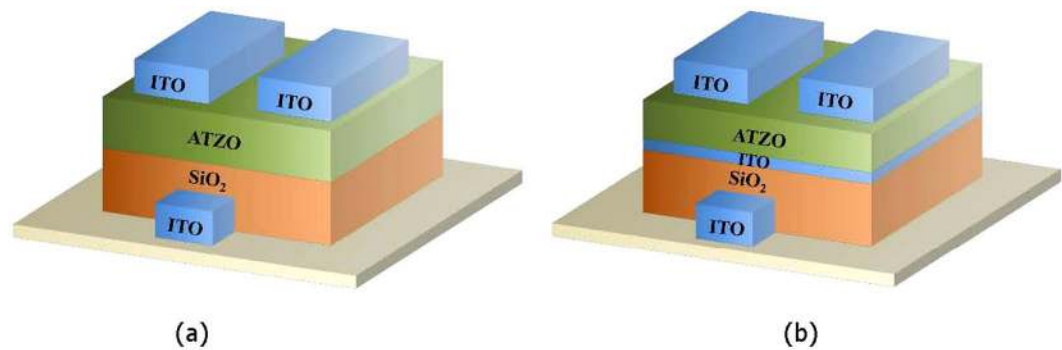
**Figure 6.** (a) Comparison of the transfer curves of the device after annealing. (b) Transfer characteristics of the double-channel ITO/ATZO TFTs with various width-to-length ratios.

after annealing. The  $I_{on}$  increases and  $I_{off}$  decreases slightly. The  $I_{on}/I_{off}$  increases to  $3.5 \times 10^8$ . The  $\mu_{sat}$  is  $246.0 \text{ cm}^2/\text{Vs}$  and the  $V_T$  is  $0.5 \text{ V}$ . The subthreshold region becomes much steeper and the SS is  $142.2 \text{ mV}/\text{dec}$ . The transfer characteristics of the ITO/ATZO TFTs with various width-to-length ratios after thermal annealing are shown in Fig. 6(b). All the devices demonstrate good electrical performance. As the channel length decreases, the  $I_{on}$  gradually increases and  $V_T$  shifts to the negative axis.

In conclusion, we successfully fabricate double-channel ITO/ATZO TFTs. The high-conductivity ITO film is adopted as the bottom channel layer to improve the  $I_{on}$  and  $\mu_{sat}$  of ATZO TFTs and the top ATZO channel layer with higher resistance controls the off-state characteristics. The double-channel ITO/ATZO TFTs shows excellent electrical characteristics, including a high ON/OFF current ratio of  $3.5 \times 10^8$ , a high saturation mobility  $\mu_{sat}$  of  $246.0 \text{ cm}^2/\text{Vs}$ , a low threshold voltage ( $V_T$ ) of  $0.5 \text{ V}$ , and a steep threshold swing SS of  $142.2 \text{ mV}/\text{decade}$ . Thus, with a simple fabrication process and superior electrical characteristics, the ITO/ATZO TFTs show promising applications in the high-performance display fields.

### Methods Section

In this work, the ATZO TFTs were fabricated on glass substrates with a conventional inverted-staggered structure. Briefly the fabrication process is as follows. First, an ITO gate electrode was deposited on glass by RF magnetron sputtering at room temperature with  $100 \text{ nm}$  film thickness. Second, a  $150\text{-nm}$ -thick  $\text{SiO}_2$  film, which was served



**Figure 7.** The schematic structures of (a) the single-channel ATZO and (b) the double-channel ITO/ATZO TFTs.

as the gate insulator layer, was formed by plasma enhanced chemical vapor deposition (PECVD). The deposition temperature was 80 °C. To create better interface properties between the insulator and channel layer, the ATZO channel layer was deposited by RF magnetron sputtering successively after formation of the SiO<sub>2</sub> film at room temperature. The wt% of SnO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> in the ATZO sputtering target are both 2%. The oxygen partial pressure [O<sub>2</sub>/(O<sub>2</sub> + Ar)] in the sputtering chamber was 15% and the channel layer was 40-nm-thick. Finally, an ITO source/drain electrode was deposited using the same process conditions with the gate electrode after formation of the channel layer. For the double-channel structure, we introduced the ITO film as the bottom channel layer between the insulator and ATZO channel layer to optimize the electrical performances. The oxygen partial pressure of the ITO film was 5%. The total channel thickness was 40 nm, including a 5-nm-thick ITO and 35-nm-thick top ATZO channel layer. Figure 7 shows the schematic structures of the single-channel ATZO and the double-channel ITO/ATZO TFTs.

During the fabrication process, we adopted standard photolithography and lift-off technique. The electrical performances of the ATZO TFTs was measured using a semiconductor parameter analyzer (Agilent 4156 C) at room temperature. The chemical states of atoms in the ATZO films were investigated using X-ray photoelectron spectroscopy (XPS, Axis Ultra). The crystal quality conditions of ATZO films were tested using X-ray diffraction (XRD, Rigako).

## References

1. Wager, J. F. Transparent electronics. *Science* **300**(5623), 1245–1246, doi:10.1126/science.1085276 (2003).
2. Nomura, K. *et al.* Thin-Film Transistor Fabricated in Single-Crystalline Transparent Oxide Semiconductor. *Science*. **300**(5623), 1269–1272 (2003).
3. Kenji, N. *et al.* Room-temperature fabrication of transparent flexible thin-film transistors using amorphous oxide semiconductors. *Nature* **432**, 488–492 (2004).
4. Park, J. *et al.* Review of recent developments in amorphous oxide semiconductor thin-film transistor devices. *Thin solid films* **6**, 1679–1693 (2012).
5. Fortunato, E., Barquinha, P. & Martins, R. Oxide Semiconductor Thin-Film Transistors: A Review of Recent Advances. *Adv. Mater.* **22**, 2945–2986 (2012).
6. Goto, T. *et al.* Application of Rotation Magnet Sputtering Technology to a-IGZO Film Depositions. *SID Symp. Dig. Tech. Pap.* **4** (2014).
7. Kim, S. I. *et al.* High performance oxide thin film transistors with double active layers. *IEDM Tech. Dig.* 73–76 (2008).
8. Chen, Y. C. *et al.* The suppressed negative bias illumination-induced instability in In-Ga-Zn-O thin film transistors with fringe field structure. *Appl. Phys. Lett.* **101**, 223502 (2012).
9. Liu, K. H. *et al.* Investigation of on-current degradation behavior induced by surface hydrolysis effect under negative gate bias stress in amorphous InGaZnO thin-film transistors. *Appl. Phys. Lett.* **104**, 103501 (2014).
10. Cai, J. *et al.* High-Performance Transparent AZO TFTs Fabricated on Glass Substrate. *IEEE Trans. Electron Devices* **60**, 2432–2435 (2013).
11. Kim, S. *et al.* Carrier Transport at Metal/Amorphous Hafnium-Indium-Zinc Oxide Interfaces. *ACS Appl. Mater. Interfaces* **40**, 22385–22393 (2015).
12. Yang, S. *et al.* High-Performance Al-Sn-Zn-In-O Thin-Film Transistors: Impact of Passivation Layer on Device Stability. *IEEE Electron Device Lett.* **2**, 144–146 (2010).
13. Hosono, H. Ionic amorphous oxide semiconductors: Material design, carrier transport, and device application. *J. Non-Cryst. Solids* **9–12**, 851–858 (2006).
14. Bak, J. Y. *et al.* Transparent Al-In-Zn-O Oxide semiconducting films with various in composition for thin-film transistor applications. *Ceram. Int.* **3**, 2561–2566 (2013).
15. Cho, D.-H. *et al.* Transparent Al-Zn-Sn-O thin film transistors prepared at low temperature. *Appl. Phys. Lett.* **93**, 142111 (2008).
16. Cong, Y. *et al.* High-Performance fully transparent Al-Sn-Zn-O Thin-Film Transistors using double-channel structures. *IEEE Electron. Lett.* **12**, 1069–1070 (2016).
17. Nomura, K. *et al.* Depth analysis of subgap electronic states in amorphous oxide semiconductor, a-In-Ga-Zn-O, studied by hard x-ray photoelectron spectroscopy. *J. Appl. Phys.* **7**, 073726-1–073726-8 (2011).
18. Kumar, B., Gong, H. & Akkipeddi, R. A study of conduction in the transition zone between homologous and ZnO-rich regions in the In<sub>2</sub>O<sub>3</sub>-ZnO system. *J. Appl. Phys.* **97**, 063706-1–063706-5 (2005).
19. Pan, Z. C. *et al.* Investigation of structural, optical and electronic properties in Al-Sn co-doped ZnO thin films. *Applied Surface Science* **265**, 870–877 (2013).

## Acknowledgements

This work was partly supported by the National Basic Research Program of China (973 program, Grant No. 2013CBA01604) and by the National Natural Science Foundation of China (Grant No. 61275025).

## Author Contributions

Y.Y.C. and D.D.H. wrote the main manuscript text. J.C.D. organized the data and created the figures. The project was guided by S.D.Z., X.Z., and Y.W., who provided suggestions about content and structure. All authors reviewed the manuscript.

## Additional Information

**Competing Interests:** The authors declare that they have no competing interests.

**Publisher's note:** Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



**Open Access** This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2017