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Gate-controlled reversible rectifying behaviour in tunnel contacted atomically-thin MoS₂ transistor

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Atomically thin two-dimensional semiconducting materials integrated into van der Waals heterostructures have enabled architectures that hold great promise for next generation nanoelectronics. However, challenges still remain to enable their applications as compliant materials for integration in logic devices. Here, we devise a reverted stacking technique to intercalate a wrinkle-free boron nitride tunnel layer between MoS₂ channel and source drain electrodes. Vertical tunnelling of electrons therefore makes it possible to suppress the Schottky barriers and Fermi level pinning, leading to homogeneous gate-control of the channel chemical potential across the bandgap edges. The observed features of ambipolar *pn* to *np* diode, which can be reversibly gate tuned, paves the way for future logic applications and high performance switches based on atomically thin semiconducting channel.

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decade after the first isolation and study of twodimensional (2D) materials, their atomically precise integration into van der Waals (vdW) planar heterostructures^{1, 2} is now forming an outstanding platform for developing novel nanoelectronic devices³⁻⁵. Such platform has been the source of many recent advances in electrical engineering that takes the advantages of the coupling of mono- or few-layered two-dimensional (2D) materials such as graphene, hexagonal boron nitride (h-BN), and transition metal dichalcogenides (TMDCs). It has thus far thrived a rich variety of physical phenomena, including metal oxide semiconductor field effect transistors (FETs)¹, spintronics memory devices⁶, photovoltaics⁵, and atomically thin superconductors⁷. Although doping control by an electrostatic gate in those devices has enabled tremendous opportunities, the lack of gapped 2D channel with complementary (p and n) polarities has hampered its application in logic units based on the co-manipulation of diodes and field effect transistors, each has been the core of modern electronics. MoS₂ is among the most studied TMDC compounds for both its outstanding electronics and optoelectronics properties as it combines well-defined bandgap, stability in ambient conditions and relatively high charge carrier mobility. Indeed 2H-type molybdenum disulfide (2H-MoS₂) has a thickness-dependent bandgap of 1.3 eV indirect gap ~1.9 eV direct gap from bulk down to single layer, respectively⁸. It therefore holds great promise not only for fun-damental studies^{7, 9, 10}, but also for future applications such as high performance FETs and opto-electronics^{11–17}. Field effect transistors involving atomically thin MoS₂¹¹ as the active channel have enabled original architectures which unlock new features such as sub-thermionic inter-band tunnelling exhibiting unprecedented minimum sub-threshold swing¹², or ultra-short gatelength FETs¹³, opening promising pathways for further enhanced integration.

To fulfill the desired performances of CMOS-type logics using MoS_2 FETs, one of the key (yet evasive) goals has been achieving

programmable ambipolar operation (i.e., obtaining easily reconfigurable same-chip *n*- and *p*-doping in MoS_2 FETs). However, to date, only few experiments^{18–20} reported hole transport in MoS_2 , which was achieved through gate dielectric engineering with high gate voltage operation¹⁹ or in an ionic liquid gating environment²⁰. Great efforts have been conducted to pursue ambipolar field effect and further gate tunable rectifying characteristics in MoS_2 based heterostructures, including MoS_2 coupled with other materials such as carbon nanotube films²¹. Similar effects can also be found in *n*-type TMDCs vdW interfaced with *p*-type TMDCs^{22, 23} or with organic crystal thin films²⁴.

Here, we show an alternative route based on architectureengineering: on the basis of the well known technique of vdW heterostructure but with a crucial refinement of the so-called reverted transfer, we enable the fabrication of very reliable high quality h-BN tunnel barriers which gives rise to gate tunable rectification and reversible pn to np diode behaviuor in tunnelcontacted few layer MoS₂ transistors.

Results

h-BN as an ultra-thin dielectrics for carrier injection via tunelling. In this work, we demonstrate the design and room temperature operation of FETs based on a tunnel-contacted (TC) MoS_2 channel. The tunnel barrier insulating layer is implemented by an ultra-thin capping layer that enables the vertical tunnelling of electrons from the top deposited electrodes. Ultra-thin (one or few monolayer) BN has been identified in the past as an efficient dielectric essential to a number of vertical transport devices, including graphene tunnel transistors^{25–30}, and excitonic superfluid double layer systems^{31, 32}.

As the few-layers h-BN is used as the top most layer, it assumes the role of an atomically uniform potential barrier, across which electrons are coupled through the tunnelling process. For that purpose it is required to be contaminant- and wrinkle-free.



Fig. 1 Comparison between metal-contacted and metal/insulator tunnel-contacted MoS_2 FETs. **a** Schematics of a metal-contacted MoS_2 film leading to a Schottky barrier field effect transistor (SB-FET). **b** Schematics of a tunnel-contacted MoS_2 field effect transistor (TC-FET). **c**, **d** Semiconductor representation of the energy levels respectively for SB-FET and TC-FET showing the absence of band bending in TC-FETs. **e** Schematics cross section of the device showing SB-FET and TC-FET side by side on the same MoS_2 flake. **f** Optical micrograph of a typical TC-FET device. Red dashed line highlights the two to four layer tunnel top h-BN, which covers half the MoS_2 . Scale bar is 10 μ m. **g** Scanning electron microscopy (SEM) image of the cross-section of the graphite-gated MoS_2 vertical tunnel device, with its boxed area zoomed in transmission electron microscopy (TEM) images in **h**, **i**



Fig. 2 Transport characteristics of MoS₂ TC-FETs. **a** Color map of output curves (l_{ds} vs V_{ds}) at different gate voltages for a typical tunnel-contacted device (room temperature operation). Red and blue-boxed areas highlight the operation range in gate voltage for *pn* and *np* diodes, respectively. **b** Typical ambipolar field effect curve at $V_{ds} = +2$ V measured in samples fabricated by the reverted vdW stacking method. Inset: same data in a semilog plot

Recent results on shot noise measurements in metal-hBN-metal tunnel junctions confirm that h-BN behaves as an ideal tunnel barrier³³.

With the conventional scheme of metal/MoS₂ contact, Fermi level pinning at the contact interface usually leads to a gate-dependent Schottky barrier (SB)34, which results in extra contact resistance that interferes with device performance (Fig. 1a, c). Here, by using the reverted vdW stacking method (Supplementary Figs. 1 and 2 show the experimental details of the reverted vdW process), large-area and wrinkle-free few-layered h-BN can be inserted between metal contacts and 2D semiconductor channel. We found that the presence of tunnel barriers in the form of two to four layer h-BN can suppress the SB, and chemical potential of the MoS₂ layer can be adjusted in a uniform manner across the entire channel, achieving precise electrostatic control of the Fermi level of the 2D layer (Fig. 1b, d). Ambipolar field effect at finite source-drain bias, and consequently fully reversible pn to np diodes by gating was obtained.

Schematic together with an optical image of a typical TC device is shown in Fig. 1e, f. Same flake of few-layered MoS₂ is contacted by normal metal contacts, and TC electrodes. Atomic force microscopy image confirms that devices made by our reverted vdW stacking method exhibit atomically flat top tunnel laver, which is free of wrinkles nor ruptures over $10 \times 10 \,\mu\text{m}^2$ area (Supplementary Fig. 3). The cross-sectional transmission electron microscopy specimen prepared by focused ion beam of the sample in a local area under metal electrodes is shown in Fig. 1g-i. Typical width of the electrodes are around 1 µm, with the MoS₂ channel beneath having dimensions $L \times W$ of $1 \,\mu\text{m} \times 1-5 \,\mu\text{m}$ for the tested devices. The multi-layered vdW heterostructure can be clearly seen with a tunnel h-BN on top of few-layered MoS₂. To improve the gate efficiency and uniformity³⁵, graphite flakes with thickness of about 4-6 nm are used as electrostatic gate spaced by a ~10 nm h-BN under the MoS₂ channel (Fig. 1i).

Ambipolar field effect at certain bias condition in MoS₂ TC-FETs. First, we characterize the MoS₂ FET with conventional Au (50 nm) electrodes. As shown in Supplementary Fig. 4, transport measurements of them show typical n-type FET behaviour. Color map of *IV* characteristics at fixed gate voltages (V_g) indicates ON states at positive and negative bias voltages (V_{ds}) on the electron side, while the channel turns off on the hole side. *IV* curves at fixed V_g slightly deviate from linear behaviour, while the transfer curves at fixed bias voltage V_{ds} show typical *n*-type unipolar field effect (Supplementary Note 1). These behaviours are standard in MoS₂ FET, agree with previously reported^{9, 11, 15}.

A striking consequence of the insertion of an ultra-thin h-BN below metal contact is the dramatic change in the color map of IV curves at fixed gate voltages, as shown in Fig. 2a. Instead of the rather symmetric V_{ds} polarization with ON state only seen in the electron side for metal-contacted MoS₂ FET, the vertical TC-FET on the same piece of MoS₂ flake, as well as in the same gate range, features strongly asymmetric V_{ds} polarization in the whole gate range. Surprisingly, when V_{ds} is larger than a threshold value of about 1 V, the device starts to exhibit ambipolar transfer curves, with ON state observed on both electron and hole sides at source-drain bias above +1 V. A typical such ambipolar field effect curve is shown in Fig. 2b. A detailed comparison of transfer curves between MoS₂ normal FET and TC-FET, as well as data from various samples are given in Supplementary Figs. 5-7 and Supplementary Note 2. We note that recent report shows that a monolayer chemical vapor deposited h-BN spacing layer can diminish SB at the metal contact, giving rise to a tripled output current in the transistor³⁶. However, we did not see such behaviour in our vertical tunnel devices, which may be a result of the less-defected and thicker tunnel h-BN crystals used in this work.

To better understand the obtained result in Fig. 2a, we now plot the line cuts of IV along fixed $V_{\rm g}$. It is found that, at the largest negative gate voltage of about -3 V (all $V_{\rm g}$ and $V_{\rm ds}$ in the measurements were pushed to the limit which keeps gate leakage negligible), the output curves behave as typical *pn* diode with rectification characteristics (Fig. 3a), and on/off ratio over 10^5 (Fig. 2b). When gate voltage is brought into the range of -1to +2 V, it is seen that the diode behaviour is inverted into *np* type by solely tuning the gate (Fig. 3b). The ON side is now in the negative bias voltage direction, as marked by boxes in Fig. 2a. Upon further doping to the electron side, i.e., at larger positive gate voltages, the output curves gradually shift from the diode behaviour into an asymmetric IV with the low bias range following the conventional semiconducting trend, but rather



Fig. 3 Room temperature gate-controlled reversible rectifying diode in a TC-FET. **a**-**c** *IV* curves showing perfect rectifying behaviour with reversible polarity characteristics of MoS₂ TC-FETs. **a**-**c** are linecuts of Fig. 2a, with output curves along fixed gate voltages of -3, 0, and 4-7 V, respectively. While **d**-**f** are the corresponding schematic band alignment pictures. **g**-**I** Simulations of rectifying characteristics of tunnel-contacted MoS₂ FET. **g**-**i** Simulated *IV* characteristics of the MoS₂ vertical tunnel FET at hole doping, neutral, and electron doping, respectively. At these corresponding doping level, their simulated PLDOS at $V_{ds} = +1$ V are shown in **j**-**I**

linear at large positive bias. Strikingly, the linear parts can be extrapolated into a single crossing point on the zero-current axis, with a crossing voltage $V_{\rm C}$ of about -1.25 V (Fig. 3c). This extrapolated crossing point of *IV* curves is not readily understood and provides food for further experimental and theoretical studies.

Discussion

We propose a simple band alignment model to explain the observed behaviour of gate-induced switching between pn to np diodes. In conventional metal-contacted MoS₂ devices, due to the work function mismatch, SB forms at the interface of metal and 2D materials, as a result of Fermi level pinning and band bending near the interface (Fig. 1a, c). However, tunnel h-BN in our case overcomes this problem, leading to a relatively free moving conduction and valence bands (Fig. 1b, d). At each stage of electrostatic doping in Fig. 3a–c, Fermi level sits at a fixed energy between the minimum of conduction band and the maximum of valence band, respectively. This free band alignment model offers

a good description of the *pn* to *np* diode inversion in a V_{ds} range of ± 2 V, as illustrated in Fig. 3d–f. Moreover, when Fermi level enters conduction band from the band gap, a slope of unity in V_{ds} vs V_g can be extracted in Fig. 2a in the negative V_{ds} regime, indicating a strong energetic coupling of chemical potential from the electrostatic gate to the electronic band in the few-layered MoS₂ channel. Once the Fermi level enters the conduction band, the gate becomes capacitively coupled owing to the large density of states, giving rise to a significantly reduced slope of V_{ds} vs V_g .

In the following, we compare the measured data with firstprinciples simulations. For simplicity, we consider the simplest scenario of monolayer MoS_2 tunnel device with a channel length of about 6 nm and two-layered tunnel h-BN (computational details can be found in Supplementary Figs. 8–11 and Supplementary Note 3). Compared to Fig. 3a–c, first-principles calculations based on the simplified model give qualitative agreement with experimental observations. As shown in Fig. 3g–i, the two-layered h-BN TC MoS_2 FET in our calculated model shows pn, np, and asymmetrical full pass rectifying characteristics at hole doping, neutral, and electron doping, respectively.



Fig. 4 Temperature dependence of transfer curves in a MoS_2 TC-FET. **a**, **b** Transfer curves at different temperatures for the device shown in Fig. 2, at drain source voltages $V_{ds} = +2$ and -2 V, respectively. Inset in **a** is a log scale of the field effect curve. **c** Line traces of temperature dependence of I_{ds} at fixed gate voltage along the blue and red dashed lines in Fig. 5a, b, respectively. Gray solid line indicates the gate leakage current during the same measurement. Solid fitting line in Fig. 5c is fitted using Eq. (1) in the main text

Their corresponding projected local density states (PLDOS) at $V_{\rm ds} = +1$ V are shown in Fig. 3j–l. One can see in the LDOS that the effective transmission forbidden region Δ in TC device is about 2.5 eV, which is largely enhanced due to the existence of h-BN tunnel barrier ($\Delta \sim 1.8$ eV in normal contacted device, shown in Supplementary Fig. 10). The simulated results echo our hypothesis of free band alignment model in Fig. 3d–f. Fermi level pinning in metal-contacted devices are suppressed by ultra thin tunnel contact, resulting in the observed finite-bias ambipolar field effect, as well as gate tunable rectifying characteristics with multiple operation states.

It is of fundamental interest to study the temperature dependence of tunnelling current in the MoS₂ TC-FETs. Figure 4a, b plots the transfer curves of the same device in Fig. 2a, with $V_{ds} = \pm 2$ V at different temperatures from 300 K down to 5 K. It can be seen in Fig. 4a that bipolar transfer curves at $V_{ds} = +2$ V show very weak temperature dependence. A plot of the transfer curve at 300 K is plotted in the inset of Fig. 4a, the sub-threshold swing is extracted on the hole side to be about 230 mV decade⁻¹, higher than the 60 mV decade⁻¹ theoretical limit³⁷. On the contrary, at $V_{ds} = -2$ V, the transfer curves show rather strong temperature dependence (Fig. 4b), with the I_{ds} decreasing upon lowering the temperature. Single traces of $I_{ds}-T$ monitored at $+2V_{ds}$ with $-5V_{g}$, and $-2V_{ds}$ with $+7V_g$ are plotted in Fig. 4a, b, respectively. The

negatively source–drain biased I_{ds} –T curve at +7 V_g (red curve) can be fitted by a phonon-assisted tunnelling model³⁸:

$$I \propto \frac{eE}{(8m^*\varepsilon_{\rm T})^{1/2}} [\Omega - \gamma]^{1/2} [1 + \gamma^2]^{-1/4} \exp\left\{-\frac{4}{3} \frac{(2m^*)^{1/2}}{eE\hbar} \varepsilon_{\rm T}^{3/2} [\Omega - \gamma]^2 \left[\Omega + \frac{1}{2}\gamma\right]\right\},$$
(1)

where $\gamma = \alpha \sqrt{2m^*/\varepsilon_{\rm T}} \frac{\hbar\omega^2}{eE} \left(2 \left[\exp(\hbar\omega/k_{\rm BT}) - 1\right]^{-1} + 1\right)$, and $\Omega = (1 + \gamma^2)^{1/2}$, with α being a fitting parameter, *E* the electrical field strength, $\varepsilon_{\rm T}$ the tunnel energetic depth, m^* the electron effective mass, $\hbar\omega$ the energy of the phonon taking part in the tunneling process, *e* and $k_{\rm B}$ the element charge and Boltzmann's constant, respectively. Using an effective mass of about $0.018m_e^{39}$, the best fit in the black solid line Fig. 4c gives $\varepsilon_{\rm T} = 0.6 \, {\rm eV}$ and $\hbar\omega \sim 11 \, {\rm meV}$.

Finally, as a proof of principle for realizing gate-tunable rectifier in the MoS_2 TC-FET, we used a simple diode circuit with load resistor of 1 M Ω and output to a 100 M Ω impedance voltage amplifier (1× amplification was used in the measurement), as illustrated in the schematics in Fig. 5a. As seen in Fig. 5b, when a sinusoidal wave is input in the MoS₂ TC-FET, output wave starts from a positively rectified half wave in the largest hole doping side, and can be first gate tuned into an intermediate OFF state,



Fig. 5 Reversal rectification of an analog harmonic signal in MoS₂ TC-FET. **a** Schematics of the gate-control rectifier device placed in a measurement and biasing circuit. The MoS₂ TC-FET is symbolized as a polarityswitchable diode. **b** Input (harmonic signal ~13 Hz) and output waves of the gate-controlled diode. A π phase shift, together with multiple states of output level (e.g., *pn* diode, OFF, *np* diode, and full pass), in the rectified output wave can be seen via gating. Each measured curve was averaged over 150 recorded traces

followed by a negatively rectified half wave in the electron doping side. Further electron doping recovers both positive and negative half output wave, with different amplitude. This gate-tunable rectification inversion with a π phase shift phenomenon, together with multiple states of output level (e.g., pn diode, OFF, np diode, and full pass), has not been reported before, and can be of great use in future gate-tunable logic circuits with atomically thin conduction channels. As discussed in Supplementary Note 4, it is noteworthy that in a device directly fabricated on SiO₂, we obtained a cut-off frequency in such MoS2 TC-FET of about 20 kHz when the Si gate is heavily doped (Supplementary Fig. 12). Moreover, stability and reliability in 2D materials based devices have been a timely topic^{40, 41}, which is crucial from the application point of view. For example, the thin h-BN layer intercalated between the metal contacts and MoS₂ channel can cause extra charge trapping that may lead to inferior reliability as compared to conventional metal-contacted MoS₂ FETs (Supplementary Note 5). We rule out this possibility based on the hysteresis measurements, as shown in Supplementary Fig. 13.

To conclude, we have developed a reverted vdW stacking method for high yield fabrication of resist-free pristine vdW heterostructure with ultra-thin top layer. This method itself opens new routes to a number of applications such as scanning tunnelling microscope on pristine 2D materials supported by another, as well as the high quality spacing layer for tunnelling electrodes. Using this technique, we have demonstrated a vertical TC MoS₂ transistor, in which suppression of band bending and Fermi level pinning is realized. The so called TC Field Effect Transistor hence gives rise to gate tunable rectification with fully reversible *pn* to *np* diode, leading to multiple operation states of output level (e.g., positive-pass, OFF, negative-pass, and fullpass). The observed ambipolar field effect at finite positive V_{ds} shows on/off ratio up to 10⁵ in such MoS₂ FETs, with an output current reaching the order of 100 nA on both electron and hole sides. We proposed a free moving band alignment model to explain the behaviour of the MoS₂ TC-FET, which is further qualitatively supported by a simplified first-principles simulation model. This work paves the way for future application in gatetunable logic devices with atomically thin semiconducting channels.

Methods

Reverted vdW heterostructures fabrication process. In order to have resist-free pristine vdW heterostructures, one of the limitations is its stacking sequence: a thick enough h-BN has to be picked up first by polymer (Propylene-Carbonate, PPC, for example) to serve as a top layer. When the top layer is too thin (<5 layers), ruptures and wrinkles increase significantly, thus reduce the quality of the final device. We solved this problem by developing a reverted vdW stacking method: few-layered MoS2 is sandwiched by a thick (~10 nm) BN (crystals from HQ Graphene) and thin (two to four layer) BN, respectively, with the resulted top later picked up lastly (Supplementary Methods). vdW heterostructures were fabricated using an integrated system E-Stack-One from Eoulu Co., Ltd., Suzhou, China. When the whole stack is collected, the PPC stamp will be flipped upside down, peeled off with care from the PDMS substrate, and slowly landed onto a hot plate of about 100 °C (Supplementary Methods). At this stage, the stack will be floating on the PPC film, which can be completely evaporated in a vacuum annealer at 350 °C for around 20 min. Followed by standard lithography and metallization. MoS₂ flake is half covered by two to four layer h-BN, and Au electrode with thickness of 20 nm is deposited onto the stack, forming conventional direct contacts and tunnel contacts, respectively. Electronic transport was measured on a Cascade probe station at room temperature, and in a Quantum Design PPMS system with a home-made sample probe interfaced with external measurement setup at low temperatures, respectively.

First-principles simulations. The device simulations in this work are carried out by using the first-principles software package Atomistix ToolKit, which is based on density-functional theory in combination with the non-equilibrium Greens function⁴². The exchange-correlation potential is described by the local density approximation and the wave function is expanded by the

Hartwigsen–Goedecker–Hutter (HGH) basis in this work. More computational details are discussed in Supplementary Note 3. The real space grid techniques are used with the energy cutoff of 150 Ry in numerical integrations. The geometries are optimized until all residual force on each atom is smaller than $0.05 \text{ eV} \text{ Å}^{-1}$. The current can be calculated by the Landauer formula⁴³:

$$I(V_{\rm ds}) = \frac{2e}{h} \int T(E, V_{\rm ds}) [f_S(E, V_{\rm ds}) - f_D(E, V_{\rm ds})] dE.$$
(2)

Here, V_{ds} is the bias voltage between the drain and the source, $T(E, V_{ds})$ is the transmission coefficient, $f_S(E, V_{ds})$ and $f_D(E, V_{ds})$ are the Fermi-Dirac distribution functions of the source and drain, respectively. The transmission coefficient $T(E, V_{ds})$ as a function of the energy level *E* at a certain V_{ds} can be calculated by the formula:

$$T(E, V_{\rm ds}) = \operatorname{Tr} \left[\Gamma_{S}(E) G^{\rm R}(E) \Gamma_{D}(E) G^{\rm A}(E) \right], \tag{3}$$

where $G^{R}(E)$ and $G^{A}(E)$ are the advanced and retarded Greens functions of the scattering region, respectively.

Data availability. The data that support the findings of this study are available from the corresponding authors upon reasonable request.

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Author contributions

Z.H. and Z.-D.Z. conceived the experiment and supervised the overall project. X.-X.L. fabricated the samples. X.-X.L., V.B., and Z.H. carried out experimental measurements; D.-M.S. and M.-L.C. provided clean room support for the experiment; XL. and J.-H.C. contributed to electron beam lithography and device fabrication. X.-W.J. and Z.-Q.F. conducted the theoretical simulations. J.-J.G. and P.-Z.L. carried out the TEM characterizations. Data analysis and interpretation were done by Z.H., X.-X.L., V.B., and C.-K.J.; the manuscript was written by Z.H. with discussion and inputs from all authors.

Additional information

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