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Growth, characterization, and high-pressure optical studies of CuWO₄

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Copper tungstate (CuWO₄) crystals grown by the top-seeded solution growth method were characterized by X-ray diffraction, Raman scattering, and optical measurements. CuWO₄ has a triclinic structure ($P\bar{1}$) with a = 4.709 Å, b = 5.845 Å, c = 4.884 Å, $\alpha = 88.3^{\circ}$, $\beta = 92.5^{\circ}$, and $\gamma = 97.2^{\circ}$. It consists of cornerlinked CuO₆ and WO₆ octahedra, the former having a pseudo-teragonally elongated geometry caused by the Cu²⁺ Jahn–Teller effect. Fifteen out of the eighteen Raman modes of CuWO₄ are reported, discussed, and compared with those of other tungstates. We also determined the indirect band-gap energy of CuWO₄ (2.3 eV) and its negative pressure coefficient up to 25 GPa. The pressure evolution of the band-gap is discussed in terms of the electronic structure of CuWO₄. Finally, no clear evidence of structural changes were found in our high-pressure experiments, but large amounts of defects are apparent beyond 18 GPa.

Keywords: CuWO₄; tungstates; scintillating material; crystal structure; electronic properties

1. Introduction

Orthotungstates are technologically important materials with special prominence in their use in scintillation detectors, lasers, photoanodes, and optical fibers. Copper tungstate (CuWO₄) has attracted increasing interest due to these applications [1] and its use as a semiconducting photoelectrode for photoelectrolysis [2]. Kihlborg et al. [3] determined that the crystal structure of CuWO₄ is triclinic ($P\bar{1}$) and X-ray photoelectron spectroscopy studies revealed information on its band-structure [4]. However, many physical properties of CuWO₄ remain unknown. In this paper we report results on X-ray diffraction, Raman-effect, and optical-absorption measurements. We also performed high-pressure (HP) optical studies up to 25 GPa. Results are discussed and compared with those obtained in tetragonal and monoclinic orthotungstates; *e.g.*, PbWO₄ and ZnWO₄.

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2. Experimental details

 $CuWO_4$ crystals were prepared using the top-seeded solution growth technique. The growth was carried out in a vertical tubular muffle furnace using a Pt crucible and a $Na_2W_2O_7$ flux. The mixtures were synthesized according to the reactions:

$$Cu(OH)_2 + WO_3 \longrightarrow CuWO_4 + H_2O \uparrow$$

$$2WO_3 + Na_2CO_3 \longrightarrow Na_2W_2O_7 + CO_2 \uparrow$$

The starting materials (analytical grade) of 80 mol% CuWO₄ and 20 mol% Na₂W₂O₇ were mixed, slowly heated to 1000 °C, and kept at this temperature for two days to homogenize the solution. By seeding several times, the saturation temperature (T_S) was determined. The outer surface of the seed was dissolved placing it into the melt at a temperature 20 °C higher than T_S for half an hour. The growing crystal was rotated at 10–25 rpm, and the cooling rate was 1–3 °C/day. The as grown crystal was drawn out of the melt and cooled down to room temperature (RT) at a rate of 50 °C/h.

To identify the crystallization phase an X-ray diffraction analysis was performed with a powder diffractometer (RIGAKU, DMAX2500PC) operated at 40 kV and 100 mA using Cu K_{α} radiation $(\lambda = 1.5405 \text{ Å})$. The 2θ scan range was 5–90° with a step size of 0.05°. The diffraction patterns were analyzed using PowderCell. Raman measurements in single crystal CuWO₄ were performed in backscattering geometry with a Renishaw Invia micro-Raman setup equipped with notch filters. A 785 nm laser line at a power below 50 mW was used for excitation to avoid thermal effects. Raman spectra were obtained with a resolution of 3 cm^{-1} . The optical-absorption study was first carried out at ambient pressure to get a better estimation of the band-gap. The samples used were 10 to 30-µm thick platelets cleaved from the single crystal. HP optical-absorption measurements up to 25 GPa were carried out at RT in a membrane-type diamond-anvil cell equipped with 500- μ m culet anvils. For these measurements 5 to 10- μ m thick samples were loaded together with a ruby chip in a 200-µm diameter hole drilled in a 40-µm thick Inconel gasket. Methanol–ethanol– water (16:3:1) was used as a pressure transmitting medium and the pressure (P) was measured using the ruby fluorescence technique [5]. In both the ambient pressure and the HP experiments, the optical-absorption spectra were obtained from the transmittance spectra of the sample using the sample-in sample-out method in an optical setup similar to that described in [6]. Reflectance measurements were also performed in CuWO₄ working at normal incidence. Two independent sets of HP experiments were conducted in order to check the reproducibility of the reported results.

3. Results and discussion

Figure 1 shows an X-ray powder diffraction pattern obtained from CuWO₄. All the peaks on it can be indexed to a pure phase, corresponding to a triclinic structure with space group $P\bar{1}$. Its lattice parameters are a = 4.709(7) Å, b = 5.845(9) Å, c = 4.884(7) Å, $\alpha = 88.3(2)^{\circ}$, $\beta = 92.5(2)^{\circ}$, and $\gamma = 97.2(2)^{\circ}$. They are in good agreement with the literature [3,7]. The statistical parameters of the refinement are $R_{WP} = 2.04\%$, $R_P = 1.62\%$, and $R(F^2) = 1.85\%$ with background subtracted. There are 217 reflections. The structure of CuWO₄ is represented in Figure 1. The primitive cell contains two chemical formulas and consists of corner-linked CuO₆ and WO₆ octahedra, having the CuO₆ octahedra a pseudo-tetragonally elongated geometry caused by the Jahn–Teller effect of the Cu²⁺ cation. In the figure, it can be seen that although CuWO₄ is triclinic, its structure is topologically related to that of monoclinic wolframite (P2/c) [8]. The WO₆ octahedra are



Figure 1. Diffraction pattern of CuWO₄. Symbols: observations, line: refined model and residual, ticks: angular values of reflections. The inset shows the crystal structure of CuWO₄, different atoms are indicated and CuO₆ and WO₆ octahedra shown.

slightly distorted as in wolframite with W–O distances ranging from 1.8 to 2.2Å. However, the Jahn–Teller effect of Cu^{2+} , which reduces the degeneracy of 3*d* orbitals, is sufficient to cause a lowering of the monoclinic P2/c symmetry to the triclinic P1 symmetry. Consequently, the CuO₆ octahedra have a pseudo-tetragonally elongated geometry with four planar Cu–O distances close to 2Å and two axial Cu–O distances around 2.4Å.

The properties of CuWO₄ were also analyzed by means of Raman measurements. According to group theory, 18 zone-center Raman-active modes are expected in CuWO₄ ($\Gamma = 18A_g$). Figure 2 shows the Raman spectra of CuWO₄ measured at ambient pressure. Fifteen modes can be identified, one of them being observed as a shoulder of the peak near 398 cm⁻¹ (see inset). The remaining three modes cannot be seen most probably because they are located below 150 cm⁻¹



Figure 2. Raman spectrum of CuWO₄. The inset shows that the band near 398 cm^{-1} can be decomposed into two modes, located at $398 \text{ and } 405 \text{ cm}^{-1}$.

and consequently they are cut by the notch filter used to suppress the laser line and its Rayleigh scattering. The observed modes are located at 180, 192, 224, 283, 293, 315, 358, 398, 405, 479, 550, 676, 733, 779, and 906 cm⁻¹. The Raman spectra of CuWO₄ can be interpreted in terms of internal and external modes of the WO₆ octahedra. Internal modes correspond to vibrations of O atoms against W inside the WO₆ octahedra and external modes imply the movement of WO₆ octahedra as rigid units against Cu. Out of the internal modes of the WO₆ octahedra, we have shown in PbWO₄ [9] and ZnWO₄ [10], that according to Hardcastle and Wachs rules six modes can be considered as stretching modes related to the six W–O bonds inside the WO₆ octahedra. An estimation of these modes can be obtained by applying Hardcastle and Wachs formulae to check the total valence of W [9,10]. A calculation such as that performed for ZnWO₄ [10] suggests that the stretching modes of CuWO₄ are those at 405, 550, 676, 733, 779, and 906 cm⁻¹. Note that the stretching modes of CuWO₄ are very similar to those of wolframite ZnWO₄ (407, 546, 678, 709, 786, and 907 cm⁻¹) [10]. The similarity in the internal frequencies of both compounds can be attributed to the similarity in masses between Cu and Zn and the similarity of the triclinic $P\bar{1}$ and monoclinic wolframite structures.

Figure 3 shows optical-absorption spectra of CuWO₄ obtained up to 18 GPa. The spectra exhibit a slow rise and the maximum value reached by the absorption coefficient, α , is around 8000 cm⁻¹. The quadratic dependence of the absorption energy on the photon energy suggests that the fundamental absorption of CuWO₄ has indirect character, as proposed by Pandey et al. from thin film studies [1]. The indirect band-gap was then determined from the $\alpha^{1/2}$ versus $h\nu$ plot, yielding $E_g = 2.3 \text{ eV}$ at ambient conditions. Reflectance measurements also support the indirect band-gap hypothesis as the reflectance spectrum does not exhibit any intense structure in the spectral range around the band-gap.

In Figure 3, it can be seen that the only change induced by pressure in the absorption spectra is a monotonic red-shift. As the shape of the spectra does not change up to 18 GPa, we assumed that CuWO₄ keeps its indirect character in this pressure range. Under this hypothesis, we determined the pressure dependence of E_g shown in Figure 3. We found that E_g decreases with P following a quadratic function. At ambient pressure the pressure coefficient is $dE_g/dP = -11 \text{ meV/GPa}$, at 18 GPa it is -25 meV/GPa. In our analysis we only considered the absorption spectra collected up to 18 GPa because the quality of the spectra degrades beyond this pressure. A possible reason for this degradation is the introduction of defects in the sample that cause an irreversible change of the shape of the absorption edge, as observed in other tungstates. A consequence of the presence of defects is that the ambient-pressure absorption spectrum is not recovered when decompressing from P > 20 GPa; by contrast, the initial absorption spectrum is recovered when releasing



Figure 3. Absorption spectra of CuWO₄ single crystals for different pressures. The inset shows the pressure dependence of the absorption edge of CuWO₄. Symbols: experimental results, line: quadratic fit.

pressure from P < 20 GPa. It is important to note here that the defect appearance on the samples beyond 18 GPa might be due to the non-hydrostatic stresses applied to them at these pressures as a consequence of the pressure medium used in the experiments [11]. The influence of such stresses in the results should not be neglected in view of the use of single crystals in our experiments. Indeed non-hydrostatic stresses could be also responsible for similar changes observed in singlecrystalline CdWO₄ and ZnWO₄ [10]. To check this hypothesis, new experiments on CuWO₄ will be performed using neon as pressure medium.

According with the size criterion developed for AWO₄ compounds [12], a HP phase transition is predicted to take place in CuWO₄ beyond 21 GPa [13]. The HP phase would likely have a wolframite-type structure [13]. Up to 25 GPa we have not found any change in the absorption spectra or any shift in the evolution of the band-gap with *P* that pointed out to the occurrence of a phase transition. However, the defect creation observed beyond 20 GPa can be related to precursor effects of the predicted transition as observed in other semiconductors [14]. This conclusion about the HP structural stability of CuWO₄ is supported by a preliminary analysis of HP EXAFS measurements, which did not detect any phase transition up to 18 GPa.

To conclude, we will compare the pressure evolution of E_{σ} in CuWO₄ and other tungstates. Contrary to CuWO₄ ($dE_g/dP = -11 \text{ meV/GPa}$), wolframite ZnWO₄ and scheelite BaWO₄ ($I4_1/a$) have small but positive pressure coefficients ($\approx 9 \text{ meV/GPa}$) [15]. On the other hand, scheelitetype PbWO₄ has a large negative pressure coefficient of -75 meV/GPa [16]. Curiously, in the HP phases of BaWO₄ and PbWO₄ $dE_g/dP \approx -20$ meV, being similar to dE_g/dP in CuWO₄. Density-functional calculations on AWO₄ compounds (A = Ca, Cd, and Pb) [17] revealed that the bottom of the conduction band of orthotungstates are dominated by contributions of W 5*d* like states. On the other hand, the upper part of the valence band is mainly composed by O 2pstates. In those compounds where the valence shell of bivalent cation A contains only s states there is a small contribution of them to the valence and conduction bands, but if the bivalent cations have a different electronic configuration (e.g., Cu and Pb) a larger contribution of the metal A to the valence and conduction bands is expected [4,18,19]. Consequently, the decrease observed in E_g for CuWO₄ and PbWO₄ upon compression can be qualitatively explained by taking into account the pressure increase of the crystal field acting on W 5d and O 2p states and the consequent increase of their hybridization with metal A states [16]. In particular the valence band in CuWO₄ is expected to have a significant Cu 3d contribution, vanishing at the Γ point for symmetry reasons and being maximum at the zone edge. Apart from explaining the indirect character of CuWO₄, it explains the negative pressure coefficient of E_g , since O 2p states (with some Cu 3d hybridization) shift towards high energies faster than the W 2d states, causing a reduction of the energy difference between the bottom of the conduction band and the top of the valence band [16]. The similitude of the pressure effects on the band-gap of CuWO₄ and that of the HP phases of BaWO₄ and PbWO₄ is consistent with this picture. These HP structures have a W–O coordination more similar to the coordination six of CuWO₄ than the low-pressure phases [20]. This makes them less compressible than the low-pressure phase of PbWO₄ (W–O coordination = 4). Therefore E_g is expected to close with P in a similar way than in CuWO₄ and much less than in scheelite PbWO₄. Further experimental and theoretical studies are being conducted to clarify this interpretation of the reported results. An important issue to address by future studies is how the Jahn-Teller effect is affected by pressure in CuWO₄.

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