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Growth of sillenite $\text{Bi}_{12}\text{FeO}_{20}$ single crystals: structural, thermal, optical, photocatalytic features and first principle calculations

Durga Sankar Vavilapalli¹, Ambrose A. Melvin², F. Bellarmine³, Ramanjaneyulu Mannam⁴, Srihari Velaga⁵, Himanshu K. Poswal⁵, Ambesh Dixit⁶, M. S. Ramachandra Rao³ & Shubra Singh¹✉

Ideal sillenite type $\text{Bi}_{12}\text{FeO}_{20}$ (BFO) micron sized single crystals have been successfully grown via inexpensive hydrothermal method. The refined single crystal X-ray diffraction data reveals cubic $\text{Bi}_{12}\text{FeO}_{20}$ structure with single crystal parameters. Occurrence of rare Fe^{4+} state is identified via X-ray photoelectron spectroscopy (XPS) and X-ray absorption spectroscopy (XAS). The lattice parameter (a) and corresponding molar volume (V_m) of $\text{Bi}_{12}\text{FeO}_{20}$ have been measured in the temperature range of 30–700 °C by the X-ray diffraction method. The thermal expansion coefficient (α) $3.93 \times 10^{-5} \text{ K}^{-1}$ was calculated from the measured values of the parameters. Electronic structure and density of states are investigated by first principle calculations. Photoelectrochemical measurements on single crystals with bandgap of 2 eV reveal significant photo response. The photoactivity of as grown crystals were further investigated by degrading organic effluents such as Methylene blue (MB) and Congo red (CR) under natural sunlight. BFO showed photodegradation efficiency about 74.23% and 32.10% for degrading MB and CR respectively. Interesting morphology and microstructure of pointed spearhead like BFO crystals provide a new insight in designing and synthesizing multifunctional single crystals.

Sillenite is a rich family consisting of more than 60 individual compounds and solid solutions. Sillenite structured complex metal oxides exist with general formula $\text{Bi}_{12}\text{MO}_{20}$, where M is a tetravalent cation (M: Si, Ti, Ge etc.) constituting discrete MO_4 tetrahedra, separated by Bi-O frameworks to form a body centered cubic lattice with space group $I23$ ^{1–6}. These Bi rich compounds are attractive materials for practical applications owing to their photoconductive, photorefractive, piezoelectric, pyroelectric and nonlinear optical (NLO) properties⁷. The catalytic properties encourage potential use of these compounds for photocatalytic applications in organic effluent treatment and water splitting. One such material, $\text{Bi}_{24}\text{Ga}_2\text{O}_{30}$, plays the role of an efficient photocatalyst in degrading Methylene blue⁸. Few sillenite compounds also possess nonlinear optical (NLO) properties with a potential to be used further for second harmonic generation (SHG) applications. Such wide scale potential applications have led researchers to propose sillenite structures based on transition-metal ions, like Mn and Fe, exploring the role of these multivalent cations. The fact that these cations can exhibit tetravalency, as in Mn^{4+} and Fe^{4+} , has led us to investigate them further for their influential properties. Tetravalency of Fe and Mn is rarely observed in perovskites and related complex oxides. These tetravalent cations (Fe^{4+} and Mn^{4+}) form an ideal sillenite structure (e.g. $\text{Bi}_{12}\text{FeO}_{20}$, $\text{Bi}_{12}\text{MnO}_{20}$ etc.)⁹. Until now researchers have investigated and proposed Fe based non-ideal sillenite single crystals such as $\text{Bi}_{12}\text{Fe}_{0.63}\text{O}_{18.945}$ and $\text{Bi}_{25}\text{FeO}_{40}$, constituting Fe^{3+} oxidation states with Bi-rich content which can be used for photoactive applications^{10,11}. These sillenite compounds often occur as intermediate products in the preparation of BiFeO_3 . Proximity to the class of Bi-rich iron-based oxides makes these compounds attractive with wide interests. As compared to widely used UV operated photocatalysts such as ZnO and TiO_2 , sillenites offer optical band gap values between 2.5 and 2.8 eV which lie in the visible region of

¹Crystal Growth Centre, Anna University, Chennai 600025, India. ²University of Bordeaux, ISM UMR CNRS 5255, Bordeaux INP, ENSCBP, 16 Avenue Pey Berland, Bordeaux 33607, Pessac, France. ³Nano Functional Materials Technology Centre, Materials Science Research Centre and Department of Physics, Indian Institute of Technology Madras, Chennai 600036, India. ⁴Division of Physics, Department of Science and Humanities, Vignans Foundation for Science, Technology and Research, Guntur 522213, India. ⁵High Pressure and Synchrotron Radiation Physics Division, Bhabha Atomic Research Centre, Mumbai 400085, India. ⁶Department of Physics and Centre for Solar Energy, Indian Institute of Technology Jodhpur, Jodhpur 342 037, India. ✉email: shubra6@gmail.com

solar spectra making them photoactive with potential applications in water splitting and dye degradation under visible light^{12, 13}. Not much research has been carried out on related ideal sillenite type $\text{Bi}_{12}\text{FeO}_{20}$ compound.

In this communication, to the best of our knowledge, we report for the first time a spearhead like ideal sillenite structured $\text{Bi}_{12}\text{FeO}_{20}$ single crystals. Non-ideal Fe based Bi-rich sillenite materials, such as crystallites of $\text{Bi}_{12}\text{Fe}_{0.63}\text{O}_{18.945}$ and $\text{Bi}_{25}\text{FeO}_{40}$ single crystals, have been investigated before^{14, 15}. An ideal Sillenite structure of polycrystalline $\text{Bi}_{12}\text{FeO}_{20}$, synthesized at elevated temperatures by typical solid state reaction method, has also been reported by Elkhoun et al¹⁶. First report on low temperature synthesis of ideal structured $\text{Bi}_{12}\text{FeO}_{20}$ single crystals with rare Fe^{4+} state via inexpensive hydrothermal process is revealed here, along with systematic characterization and application related studies.

Experimental

Crystal growth. High pure $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ and $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ as starting precursors and 0.2 M concentration each were dissolved in 50 ml of deionized water. Few drops of HNO_3 were added to get clear transparent solution. After vigorous stirring, 50 g of KOH was added and the temperature of solution was cooled down to room temperature before transferring it into a 100 mL Teflon-lined autoclave, upto 70% of its maximum capacity. Crystallization takes place at 200 °C for 72 h. Post this, the autoclave was cooled and depressurized, product was washed with distilled water, sonicated and then the sample was harvested as fine, reddish brown crystals.

Characterization. For structural analysis, Single crystal X-ray diffraction (SC-XRD, Bruker Kappa ApexII) was performed and the lattice parameters were obtained by refining Single Crystal XRD data by SHELXTL refining software^{17, 18}. High Resolution transmission electron microscopic (HRTEM) images and selected area electron diffraction patterns (SAED) were recorded on an FEI Tecnai TF-20 operating at 200 kV. X-ray photoelectron spectroscopic (SPECS GmbH, Germany) measurements were conducted to investigate the Fe and Bi oxidation states. X-ray absorption spectroscopy (XAS) measurements were carried out to examine the valence state of Fe in $\text{Bi}_{12}\text{FeO}_{20}$ (BFO). The experiments were performed at the Energy-Scanning EXAFS beamline (BL-9) at the Indus-2 Synchrotron Source (2.5 GeV, 200 mA), Raja Ramanna Centre for Advanced Technology (RRCAT), Indore, India^{19, 20}.

Synchrotron based powder X-ray diffraction measurements. Synchrotron based powder X-ray diffraction measurements were carried out on well ground powder samples of $\text{Bi}_{12}\text{FeO}_{20}$ single crystals at Extreme Conditions X-ray diffraction (EC-XRD) beamline (BL-11) at Indus-2 synchrotron source, Raja Ramanna Centre for advanced Technology (RRCAT), Indore, India. High temperature measurements were carried out on STOE high temperature attachment 0.65.3 with Eurotherm 2416 controller. Desired wavelength (0.6285 Å) for ADXRD diffraction experiments was selected from white light from the bending magnet using a Si(111) channel cut monochromator. The monochromatic beam is then focused on to the sample with a Kirkpatrick-Baez mirror or K-B mirror. A MAR345 image plate detector (which is an area detector) was used to collect 2-dimensional diffraction data. Sample to detector and the wavelength of the beam were calibrated using NIST standards LaB_6 and CeO_2 . Calibration and conversion/integration of 2D diffraction data to 1D, intensity vs 2θ , was carried out using FIT2D software^{21, 22}.

Electronic structure calculations. The density functional calculations are carried out under full potential linear augmented plane wave (FP-LAPW), as implemented in Wien2K²³. The modified Becke-Johnson parameterization is used as an exchange correlation function²⁴. The unit cell is divided in muffin tin region (with R_{mt} as radius) and interstitial region (IR). The muffin tin radii for bismuth, iron and oxygen atoms are chosen in such a way that there is no overlap among different atomic elements. The plane wave cut off parameters $R_{\text{mt}} \times K_{\text{max}} = 7$ and $G_{\text{max}} = 12$ are used for structural, electronic and optical properties of $\text{Bi}_{12}\text{FeO}_{20}$. The maximum value of l (l_{max}) is considered 10 and cut-off energy is at -7.0 Ry, defining the separation between the core and valence states. The self-consistent calculations are carried out under total energy convergence of ~ 0.001 Ry. A large plane wave cut-off of 150 Ry is used throughout the calculation and initially 125 K-points are considered in Brillouin zone for optimization while 1000 K points are used for computing the other properties like electronic and optical properties of the material.

Results and discussion

Single crystal X-ray diffraction data for $\text{Bi}_{12}\text{FeO}_{20}$ (BFO) crystals were measured at 296 K using a Bruker Kappa Apex II with a wavelength of $\text{MoK}\alpha$ radiation of 0.7107 Å. The structure was refined using SHELXTL refining software^{17, 18}. The refined XRD pattern and crystal structure is shown in Fig. 1a. The lattice parameters of BFO are a , b and $c = 10.1713(10)$ Å and α , β and $\gamma = 90^\circ$ and the cell volume is $V = 1052.28(3)$ Å³. Single crystal structure refinement data is presented in Table S1 and the refined data indicates the formation of $\text{Bi}_{12}\text{FeO}_{20}$ with body centered cubic crystal system and $I23$ space group. The optimization of crystal structure is performed and the calculated variation of volume versus energy is shown in Fig. S1. A clear minimum noticed at $-523,513.3$ eV, correspond to the minimum energy structure. The computed lattice parameter is $a = 10.2707$ (19.4154 Bohr) Å, and is in good agreement with experimental refined XRD data.

Figure 1b shows the HRTEM image of BFO single crystals, with clear resolved crystalline domains and an interplanar distance 0.71 nm, 0.5 nm and 0.415 nm corresponding to (110), (200) and (211) planes respectively. In Fig. 1c the SAED pattern reveals the presence of (200), (220) and (222) planes. Fast Fourier transform of HRTEM image gives an evidence of the crystallinity of BFO as shown in Fig. 1d.

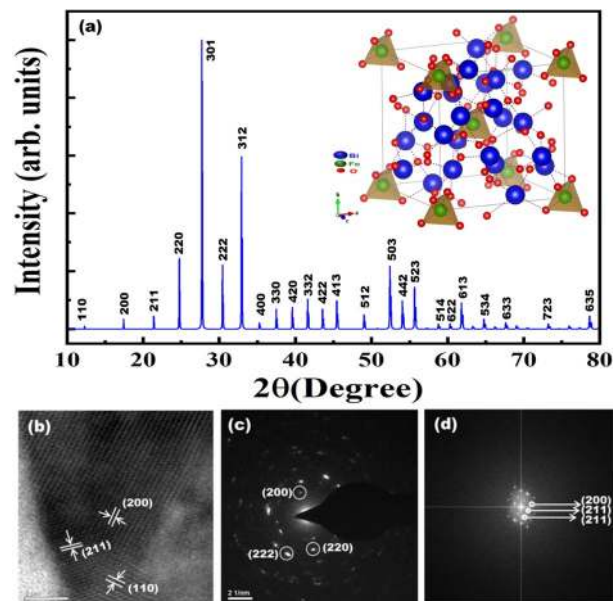


Figure 1. (a) Single crystal XRD pattern of $\text{Bi}_{12}\text{FeO}_{20}$ (inset crystal structure of $\text{Bi}_{12}\text{FeO}_{20}$ comprising of body-centered cubic FeO_4 tetrahedra) (b) HR-TEM image of BFO (c) SAED pattern of BFO (d) Corresponding fast fourier transform (FFT) image.

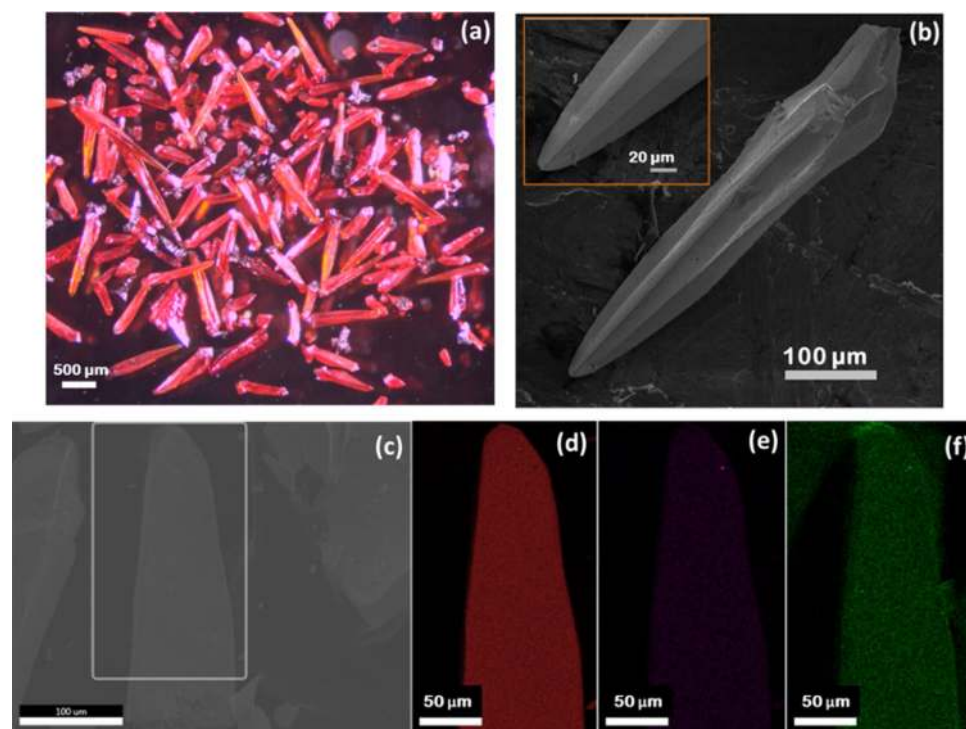


Figure 2. (a) Optical microscope image of $\text{Bi}_{12}\text{FeO}_{20}$ crystals (b) SEM image and the EDS elemental mapping of the BFO crystals (c) SEM image of EDS mapped area (d) Bi, (e) Fe, (f) O elements.

Optical and scanning electron microscopy images of BFO in Fig. 2a,b reveal interesting spearhead like morphology with an average size ~ 1 mm. EDS mapping of BFO crystals showed a uniform distribution of Bi, Fe and O throughout the crystal Fig. 2c–f.

A possible growth mechanism of spearhead like BFO crystals based on microscopic revelations has been illustrated and represented with a schematic in Fig. 3. SEM images recorded at various stages of crystallization of BFO proves that the growth mechanism involves various steps starting from nucleation to single crystal formation.

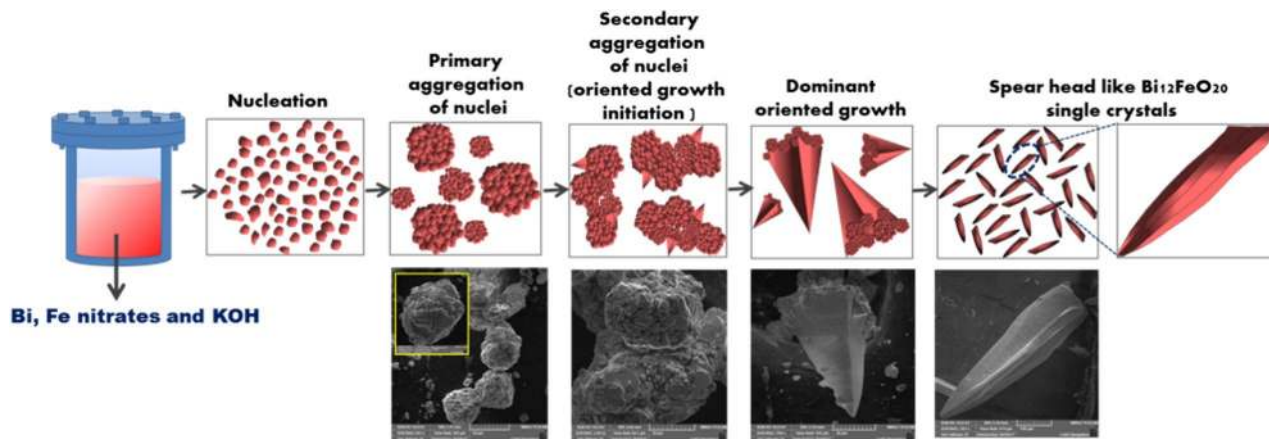


Figure 3. Schematic illustrations of hydrothermally grown spear head like $\text{Bi}_{12}\text{FeO}_{20}$ single crystals with corresponding SEM images.

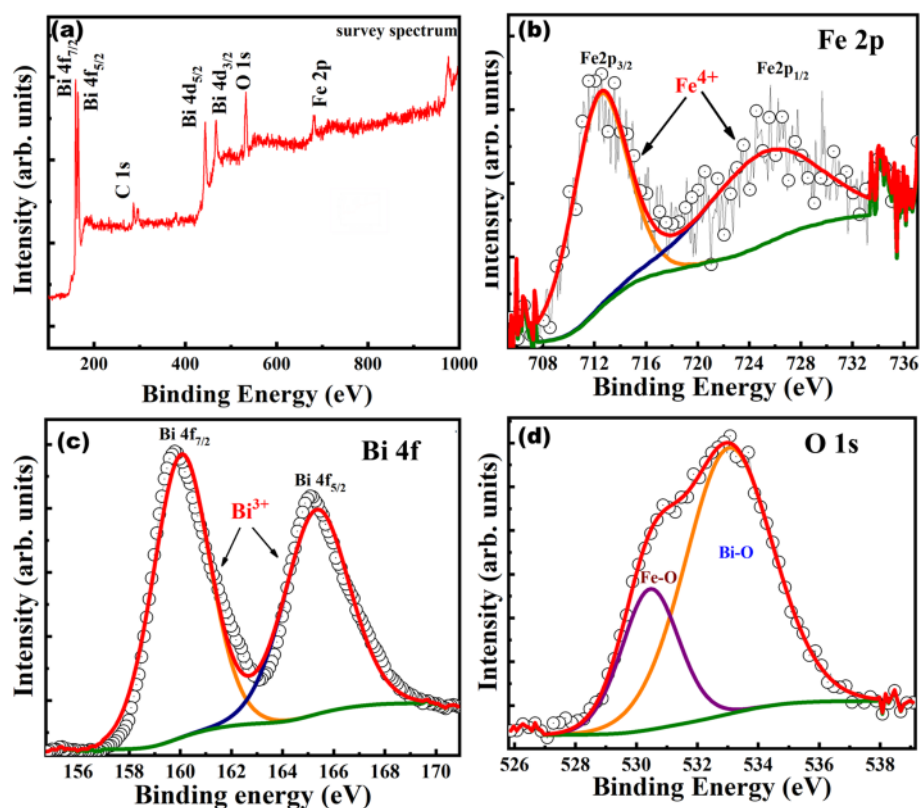


Figure 4. XPS spectra of (a) full survey spectrum, (b) Fe 2p, (c) Bi 4f, and (d) O 1s spectra of as synthesized BFO crystals.

During the hydrothermal growth at applied temperature and autogenerated pressure, supersaturation takes place and initiates nucleation in the precursor solution. An increase in growth time tends to aggregate the nuclei to form clusters. This step is followed by secondary aggregation of clusters initiating oriented growth which is desirable in single crystal. At an optimized growth time and temperature complete growth of crystal takes place.

XPS survey spectrum in Fig. 4a, calibrated using carbon binding energy of 284.6 eV, indicates the presence and oxidation states of constituent Bi, Fe and O elements in $\text{Bi}_{12}\text{FeO}_{20}$ single crystals. XPS has been extensively used to determine the oxidation state of b-site cation, which is expected to possess tetravalency in an ideal sillenite structured $\text{Bi}_{12}\text{FeO}_{20}$. In Fe-based non-ideal sillenite structure such as $\text{Bi}_{12}\text{Fe}_{0.63}\text{O}_{18.945}$ and $\text{Bi}_{25}\text{FeO}_{40}$, the tetravalency is non-existent⁴. Studies have revealed that $\text{Bi}_{12}\text{Fe}_{0.63}\text{O}_{18.945}$ constitutes Bi and Fe in 3+ oxidation state, whereas $\text{Bi}_{25}\text{FeO}_{40}$ possesses Bi with multiple oxidation states of 3+ and 5+ and Fe with 3+ oxidation state for charge compensation²⁵. The XPS spectra was fitted to Lorentzian-Gaussian utilizing Shirley background

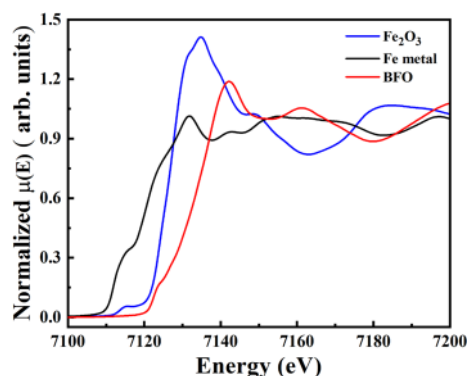


Figure 5. X-ray absorption spectra of $\text{Bi}_{12}\text{FeO}_{20}$ near Fe-K edge position as compared with standard metallic Fe and Fe_2O_3 .

method and a spin splitting of 13.4 eV between $\text{Fe } 2p_{3/2}$ and $\text{Fe } 2p_{1/2}$ was observed corresponding to Fe 2p spectra in Fig. 4b. Fe 2p was fitted to a single peak at 712.6 eV and 726.04 eV for $\text{Fe } 2p_{3/2}$ and $\text{Fe } 2p_{1/2}$ spin orbits respectively. For $\text{Fe } 2p_{3/2}$ the binding energy (B.E) of Fe^{2+} typically exists between 709 eV–710 eV, whereas for Fe^{3+} it exists between 710 eV–711 eV²⁶. However, in the present case, it was observed that $\text{Fe } 2p_{3/2}$ and $\text{Fe } 2p_{1/2}$ correspond to B.Es ~ 712.6 eV and ~ 726.04 eV respectively. The $\text{Fe } 2p_{3/2}$ and $\text{Fe } 2p_{1/2}$ peaks shift towards higher energies as compared to typical binding energy of Fe^{2+} and Fe^{3+} oxidation states clearly indicating the existence of higher oxidation state (Fe^{4+}) constituting $\text{Fe } 2p$ ^{27–31}. With regard to Bi, the Bi 4f XPS spectra shown in Fig. 4c exhibit two main peaks centered at 165.34 eV (Bi $4f_{5/2}$) and 160.03 eV (Bi $4f_{7/2}$) with spin–orbit splitting difference of 5.3 eV corresponding to the binding energy of Bi^{3+} ³². The XPS spectra of O 1s shown in Fig. 4d also revealed two peaks at 530.43 eV and 533.11 eV corresponding to Fe–O and Bi–O bonds respectively^{33,34}. Hence the role of Bi^{3+} and rare occurrence of Fe^{4+} in constituting the ideal $\text{Bi}_{12}\text{FeO}_{20}$ sillenite single crystal was confirmed. The valence state Fe in BFO was also confirmed by XAS studies. The Fe K-edge position of BFO was recorded and compared with standard metallic Fe(0) and Fe_2O_3 (Fe^{3+}) as shown in Fig. 5. Fe absorption edge for Fe_2O_3 was observed at ~ 7120 eV, which corresponds to valence state of Fe^{3+} ³⁵. The absorption edge of Fe in BFO is shifted to a higher energy as compared to Fe_2O_3 , implying that Fe in BFO possesses a higher valence state of Fe^{4+} . The XAS studies are in good agreement with XPS studies supporting the existence of Fe with 4+ oxidation state in sillenite $\text{Bi}_{12}\text{FeO}_{20}$.

Sillenites are also referred to as pyroelectric materials, which can enable them to generate voltages when they experience heat energy. Hence, these materials can be used for thermal sensing devices³⁶ and determination of thermal properties of such materials can give new insights into their behavior. Thermal expansion of these materials can be analyzed by finding the coefficient of thermal expansion / thermal expansivity (α). Temperature dependent XRD is the one of the accurate methods for determining the molar volume (V_m) of a solid because thermal expansion can be calculated per unit cell (atomic level). In the present work, we have attempted to find the thermal expansivity of sillenite BFO using ADXRD within a temperature range of 30–700 °C shown in Fig. 6.

Thermal expansion coefficient (α) is the function of change in molar volume (V_m) with respect to temperature, as shown below

$$\alpha = \frac{1}{V_m} \left(\frac{\partial V_m}{\partial T} \right) \quad (1)$$

where α is the thermal expansion coefficient, V_m is the molar volume and T is the absolute temperature. The value of ' α ' can be determined from the temperature dependent values of V_m ³⁷. Experimentally, this can be done by using temperature dependent XRD to measure the unit cell parameters at various temperatures. For cubic crystal structure materials, the molar volume V_m is defined as

$$V_m = \frac{a^3 N_A}{Z} \quad (2)$$

where a is the lattice parameter, N_A is the Avogadro's number (6.022×10^{23}) and Z is the number of formula units per unit cell for body centered cubic cells ($Z = 2$). Cell parameters, as for example d-spacing (d) and lattice constant, were calculated by Reitveld refinement for XRD patterns recorded at various temperatures. Refined XRD pattern is shown in the supporting information (Figs. S2–S16).

The molar volume V_m as a function of temperature can be expressed as

$$V_m = V_{m,0} (1 + AT + BT^2), \quad (3)$$

where $V_{m,0}$ is the initial molar volume at 30 °C. Fitting the measured values of V_m in Eq. (3), provides V_m as a function of temperature. First order differentiation of Eq. (3) and substitution into Eq. (1) provides the value for thermal expansivity (α).

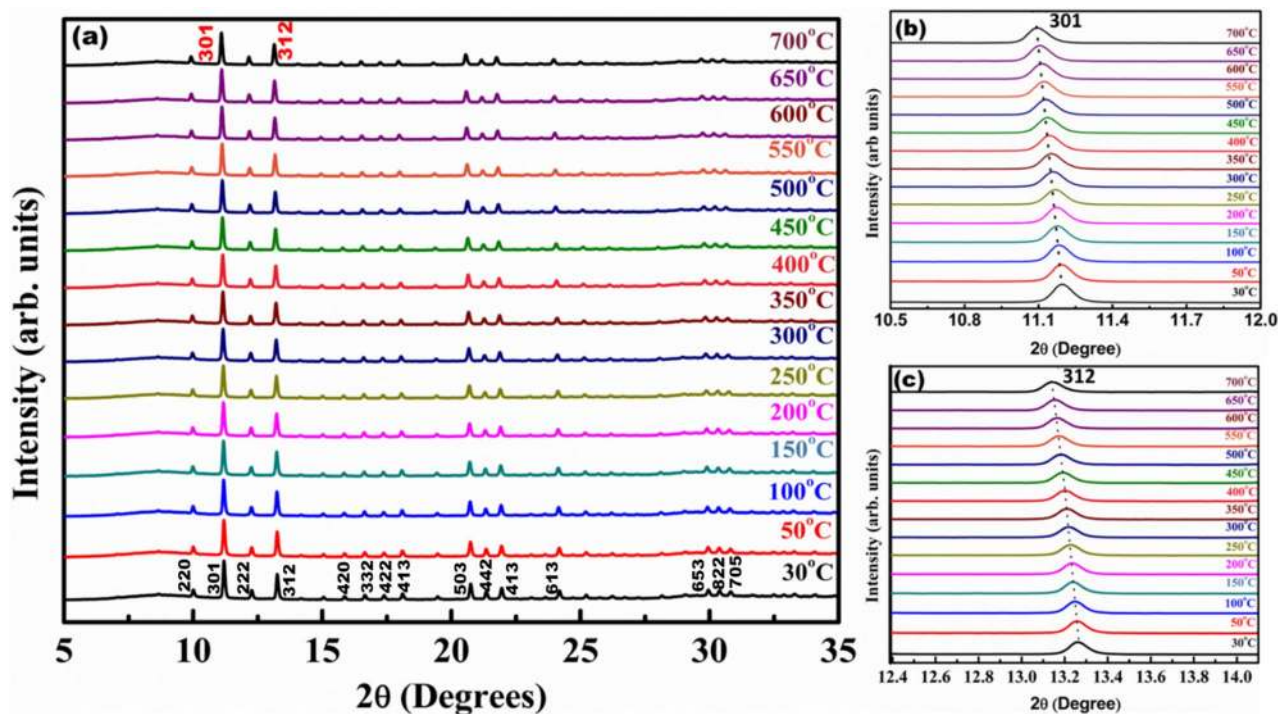


Figure 6. (a) Temperature dependent XRD patterns of BFO from 30 °C to 700 °C (b,c) shift in 2θ with respect to temperature.

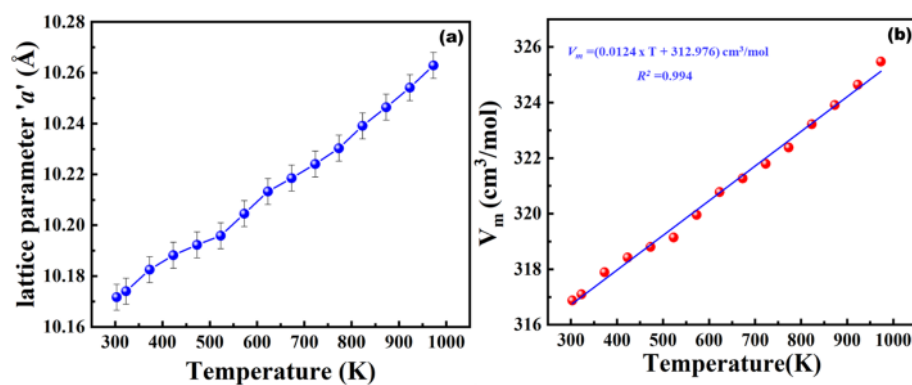


Figure 7. (a) The lattice parameter ‘ a ’ as a function of temperature (b) molar volume (V_m) of BFO as a function of temperature.

From the temperature dependent XRD, we found an increase in lattice parameters and shifting of 2θ value towards lower degree with respect to increase in temperature from 30 °C to 700 °C (Fig. 6). Upon refining the XRD pattern, lattice parameters at corresponding temperatures were calculated. The measured ‘ a ’ value was substituted in Eq. (2) and then molar volume (V_m) was calculated. The measured ‘ a ’ value as the function of temperature is shown in Fig. 7a.

From the fitting of Eq. (3), volumetric changes with respect to temperature was calculated (Fig. 7b) and substituted in Eq. (1). The coefficient of thermal expansion (α) was found to be $3.93 \times 10^{-5} \text{ K}^{-1}$. Differential scanning calorimetry (DSC) measurements (Fig. S18) (in order to determine the phase transition states) were performed up to 1000 °C. Major phase transitions were observed at 770 °C and 827 °C. The endothermic peak at 770 °C and 827 °C might be caused by partial and complete decomposition of the material. Transition above 900 °C is due to melting of decomposed materials.

The absorption spectra of BFO along with the corresponding Tauc plot (inset Fig. 8a) reveal an effective optical bandgap of 2 eV, which falls under visible region, satisfying a major requirement for photoactive applications like photocatalysis in degradation of organic dyes and water splitting etc.

Any possibility of emission of photons of characteristic wavelengths from ideal $\text{Bi}_{12}\text{FeO}_{20}$ sillenite single crystals can also be probed by Cathodoluminescence (CL) (Fig. 7b) under high-energy electron bombardment. In CL the excitation source can be focused to a probe in an electron microscope providing us with luminescence

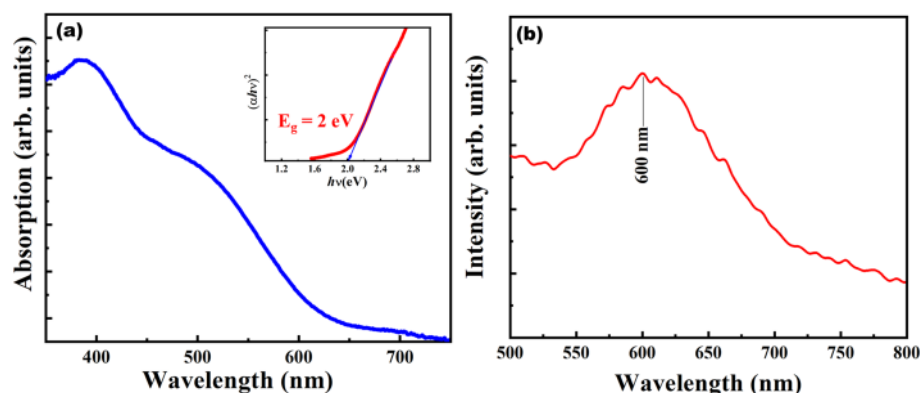


Figure 8. (a) Optical absorption spectra (inset Tauc plot to measure bandgap) (b) CL spectra of BFO.

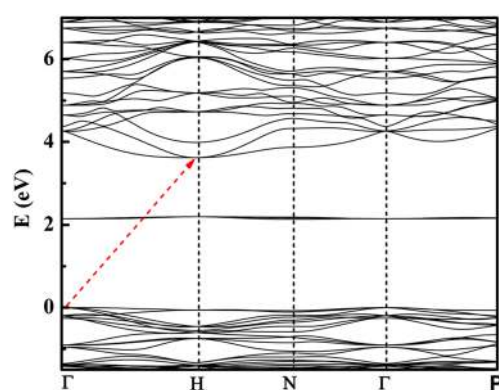


Figure 9. Bandgap energy calculation for sillenite type $\text{Bi}_{12}\text{FeO}_{20}$ (cubic) using density functional calculations.

information having spatial resolution orders of magnitude higher as compared to other techniques. CL on perovskite materials can provide evidence regarding the photoactivity of these materials as it has the potential to resolve emission characteristics in nanoregime³⁸. The spatial distribution of emission recorded at 300 K reveals that the entire $\text{Bi}_{12}\text{FeO}_{20}$ crystal radiates predominantly in the visible region. CL spectrum in Fig. 8b obtained at 20 kV for as grown $\text{Bi}_{12}\text{FeO}_{20}$ single crystals, reveals strong band edge emission at about 600 nm (2.04 eV) which is in good agreement with the measured bandgap of BFO single crystals.

Further, we compute the electronic band structure shown in Fig. 9. The electronic band structure clearly suggests that $\text{Bi}_{12}\text{FeO}_{20}$ is a wide bandgap indirect semiconductor. The valence band maxima and conduction band minima lie at Γ and H in Brillouin zone with bandgap ~ 3.17 eV, as marked with a red dashed arrow (Fig. 9). The mismatch in experimental and calculated bandgap values is ascribed to superposition. However, it is known that DFT calculations do not reproduce the bandgap values.

The computed total and partial density of states are plotted in Fig. 10. We find that valence band consists of oxygen (O) s and p orbitals mainly with partial contribution from iron(Fe) s, p and d atomic orbitals, whereas bismuth(Bi) p and oxygen(O) s orbitals contribute mainly to the conduction band. Iron is coordinated as FeO_4 tetrahedra in $\text{Bi}_{12}\text{FeO}_{20}$ lattice, causing the crystal field splitting in t_{2g} (d_{xy} , d_{xz} and d_{yz}) and e_g ($d_{x^2-y^2}$ and d_{z^2}) orbitals, Fig. 10. The tetrahedral splitting difference is about 2.27 eV with e_g orbitals lying inside the valence band and t_{2g} orbitals lying within the bandgap, giving rise to the intra band states near 2.13 eV, with very large contribution to the density of states. We also computed the absorption spectra of single crystals (Fig. S18). The large absorption coefficient $\sim 10^4 \text{ cm}^{-1}$ is noticed with a small peak ~ 2.13 eV, superimposed with bandgap absorption. This is attributed to the presence of iron intra band states within the bandgap, as noticed in partial density of states, and in experimentally recorded optical absorption spectra of BFO.

Being a light sensitive compound, sillenite BFO has the potential to act as a photoactive material. A detailed analysis of photoresponse as well as PEC (Photoelectrochemical) measurements can help us to evaluate its photoactive behavior. In the PEC measurement setup, the working electrodes were prepared as follows: BFO crystals were ground to fine powder which was used to form slurry using distilled water. The slurry was coated on FTO and the photoresponse of BFO sample was measured under Xenon lamp source (100 W/cm^2 , AM1.5). Dark and light current measurements at fixed potential (0.5 V) reveal significant photoresponse with proper ON/OFF response observed by chronoamperometry studies Fig. 11a,b. Impedance spectroscopic measurements under dark and light illumination conditions on the sample showed a rapid decrease in impedance under light

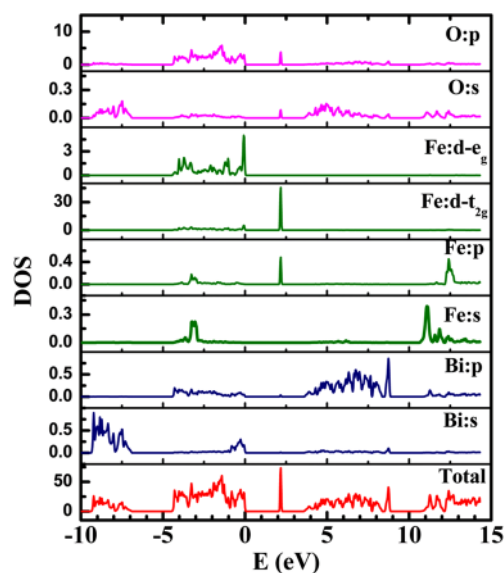


Figure 10. Partial density of Bi (s and p), Fe (s, p, and d) and O (s and p) states in sillenite $\text{Bi}_{12}\text{FeO}_{20}$.

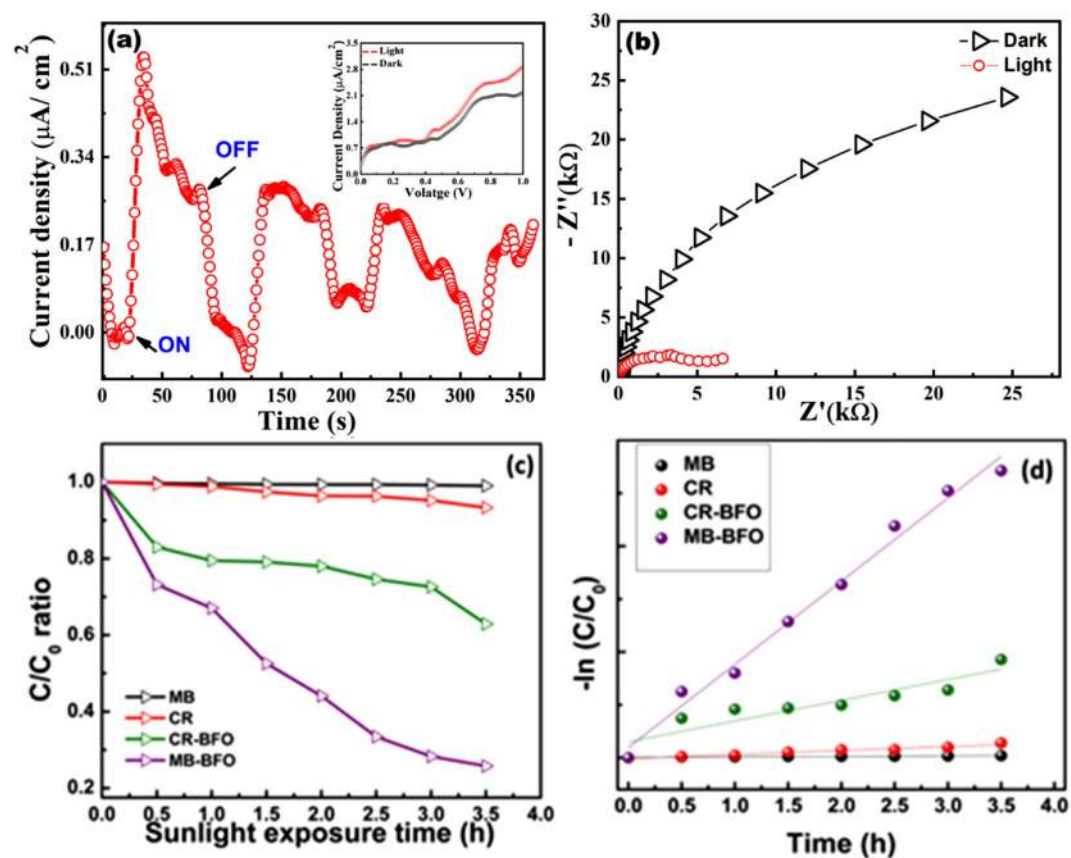


Figure 11. PEC studies on $\text{Bi}_{12}\text{FeO}_{20}$ (a) Chronoamperometry curve, shows current response for ON/OFF states of light (inset dark and light I-V characteristics) (b) Impedance measurements under dark and light illumination (c) C/C_0 ratio graph of MB and CR dyes with and without catalyst (d) First order reaction kinetics for finding rate constant of MB, CR, MB-BFO and CR-BFO under sunlight exposure.

illumination Fig. 11b. The results imply an efficient and rapid separation of photogenerated charge carriers under light irradiation which leads to photoconductivity as well as a rapid decrease in impedance of the sample.

A direct evidence of photocatalytic activity of as synthesized BFO crystals was found from dye degradation studies carried out with fine ground powder of BFO. BFO was tested to degrade of methylene blue (MB) and Congo red (CR) organic dyes under direct sunlight irradiation. The initial dye concentration of MB and CR was kept at C_0 : 3.5 mg/L and 10 mg/L respectively. The catalyst loaded dye solution was thoroughly blended using ultra sonication under dark condition. The experiment was conducted for different intervals of time under natural sunlight and dyes were found to readily degrade. The sunlight driven degradation of both MB and CR was recorded at regular intervals using UV–visible absorption spectroscopy. After 3.5 h of sunlight exposure MB and CR degraded by 74.23% and 32.10% of their initial concentration respectively. The degradation profile is shown in Fig. S17 along with C/C_0 ratio graphs and photocatalytic reaction kinetics of MB and CR in Fig. 11c,d. The degradation rate of MB by BFO was found $\sim 0.392 \text{ h}^{-1}$, whereas for CR it was $\sim 0.098 \text{ h}^{-1}$. Photodegradation kinetics MB and CR in presence of BFO are compared with few reported semiconductor nanoparticles based photocatalyst. BFO showed better performance as compared to some of the existing semiconductor photocatalysts [as tabulated in Table S2] pointing towards the efficacy of BFO as a good photocatalyst material for organic effluent treatment.

Conclusion

Ideal sillenite spearhead type $\text{Bi}_{12}\text{FeO}_{20}$ single crystals were grown by hydrothermal method for the first time with an average size of $\sim 1 \text{ mm}$. The refined single crystal XRD data along with supporting XPS analysis confirms the crystallinity and presence of Bi^{3+} and rare Fe^{4+} oxidation states in BFO respectively. The thermal expansion coefficient was calculated from temperature dependent XRD studies. The bandgap energy from the optical absorption data was found to be $\sim 2 \text{ eV}$ and the electronic structure was also investigated using first principle calculations. The photoactivity of as grown crystals was proved beyond doubt by PEC measurements, revealing significant photoresponse. CL and photodegradation studies of prepared BFO crystals revealed the luminescence and photocatalytic behavior respectively with promising applications.

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Author contributions

D.S.V. conceived and planned the present idea of work. D.S.V. developed the material and performed the experimental work as well as analyzed its structural, optical and catalytic studies. S.S. supervised the findings of this work and helped shape the manuscript. XPS and SEM characterization was performed by F.B. and R.M., A.M. performed PEC studies. V.S. and H.K.P. performed high-temperature XRD studies using synchrotron source at RRCAT. Indore. A.D. has contributed to DFT studies. D.S.V., M.S.R.R. and S.S. contributed to the design and implementation of the research, analysis of the results and writing of the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

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Correspondence and requests for materials should be addressed to S.S.

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