

Review Article

High-Power ZBLAN Glass Fiber Lasers: Review and Prospect

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Received 15 October 2009; Accepted 7 January 2010

Academic Editor: Iyad Dajani

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ZBLAN ($\text{ZrF}_4\text{-BaF}_2\text{-LaF}_3\text{-AlF}_3\text{-NaF}$), considered as the most stable heavy metal fluoride glass and the excellent host for rare-earth ions, has been extensively used for efficient and compact ultraviolet, visible, and infrared fiber lasers due to its low intrinsic loss, wide transparency window, and small phonon energy. In this paper, the historical progress and the properties of fluoride glasses and the fabrication of ZBLAN fibers are briefly described. Advances of infrared, upconversion, and supercontinuum ZBLAN fiber lasers are addressed in detail. Finally, constraints on the power scaling of ZBLAN fiber lasers are analyzed and discussed. ZBLAN fiber lasers are showing promise of generating high-power emissions covering from ultraviolet to mid-infrared considering the recent advances in newly designed optical fibers, beam-shaped high-power pump diodes, beam combining techniques, and heat-dissipating technology.

1. Introduction

Since the first demonstration of laser emission from a ruby crystal (chromium-doped corundum) in 1960 [1], hundreds of crystals and glasses doped with rare-earth ions have been fabricated and utilized in solid-state lasers to generate coherent emissions at different wavelengths. In contrast to crystals, glasses do not only have broad laser transitions which are essential conditions for wavelength tuning and ultrashort pulses generation but also have broad absorption spectra that relieve the wavelength tolerance for the pump sources. Most importantly, single-mode optical fibers, as the most flexible and compact gain media for high-efficiency and excellent beam-quality laser generation, are mostly drawn from glasses. Although crystalline fibers can be drawn using techniques of edge-defined film-fed growth [2], micropulling-down [3], and laser heated pedestal growth [4], their cores cannot be precisely controlled to be small enough to ensure exclusive single-transverse-mode guiding and their lengths are also technically limited. To date, silicate, phosphate, fluoride, and chalcogenide glasses can be drawn into single-mode fibers. A variety of lasers have also been demonstrated in these glass fibers. The spectral range of glass fiber lasers can cover from ultraviolet (UV) to mid-

infrared and the output power of a single-element fiber laser can be up to 6 kilo-watts [5]. In contrast to other lasers, the attractive features of fiber lasers include outstanding heat-dissipating capability, excellent beam quality, high optical conversion efficiency, simplicity and compactness, high single-pass gain, low laser threshold, and broad gain bandwidth.

Silicate glasses are outstanding hosts for rare-earth ions and most of recent fiber lasers are constructed with silica fibers due to their low loss, high tenability, and strong strength. Phosphate glass as the host for the fiber lasers has a high solubility that enables extremely high doping level of rare-earth ions (up to 20 wt%) and now is mostly used for high-gain fibers. Very high-gain per unit fiber length (~ 5 dB/cm) can be obtained and watt-level outputs have been delivered from short-length (< 10 cm) fiber cavities that support oscillation with single longitudinal mode [6, 7]. Fluoride and chalcogenide glasses have drawn much attention because they are found to have low phonon energy and mid-infrared transparency. These glasses are excellent candidates for fiber lasers in visible and mid-infrared regions where emissions are hard to be obtained from silicate and phosphate fibers. Comparing to chalcogenide fibers, fluoride fibers have been studied for lasing actions with more

significant efforts due to their high allowable doping levels (up to 10 mol%), relatively high strength, high stability, and low background loss (<0.05 dB/m). Chalcogenide fibers are generally investigated for nonlinear wavelength transfer due to their high nonlinearity [8, 9]. Though there are some studies on rare-earth-doped chalcogenide fiber lasers and amplifiers [10–13], their efficiencies and output powers are relatively low due to the low available rare-earth doping level (~ 0.1 mol%), the large background loss (0.5–1 dB/m), and the fragility of chalcogenide glasses. Significant long-term efforts are needed before the realization of high output power chalcogenide fiber lasers for practical applications. In contrast to chalcogenide fiber lasers, a plenty of fluoride glass fiber lasers have been reported in the past two decades and ten-watt-level output powers have already been demonstrated [14–16]. So far there are only a few reviews on the progress of fluoride fiber lasers and they are dated before 1995 [17, 18]. In this paper we give an overview of fluoride glass fiber lasers and show their promise for generating high-power emissions covering from UV to mid-infrared.

The paper is organized as follows. Progresses and properties of fluoride glasses and fabrication of fluoride fibers are briefly described in Section 2. Reviews on mid-infrared lasers, upconversion UV and visible lasers, and supercontinuum generation are presented in Section 3. Analyses and discussions on power scaling of fluoride fiber lasers are completed with calculations in Section 4. Summary is given at the end of this paper.

2. Fluoride Glass Synthesis and ZBLAN Fiber Fabrication

Since detailed descriptions of heavy metal fluoride (HMF) glasses and complete investigations of their properties can be found in the literature [19–28], we only briefly discuss here the synthesis of fluoride glasses and the fabrication of ZBLAN ($\text{ZrF}_4\text{-BaF}_2\text{-LaF}_3\text{-AlF}_3\text{-NaF}$) fibers. Basic properties of ZBLAN glass are also summarized in this section as a preparation for the discussion in Section 4.

2.1. Fluoride Glass Synthesis. In the history of science and technology, unexpected outcome of an experiment usually leads to a fantastic discovery. Poulain's discovery of the fluorozirconate glasses was entirely a "surprise". In 1974, he was working on the synthesis and characterization of ZrF_4 -containing crystalline compounds doped with rare earths in the laboratory of Jacques Lucas at the University of Rennes. When NaF was added to fill the voids in the crystal structure of a $\text{ZrF}_4\text{-BaF}_2$ phase, the mixture of $\text{ZrF}_4\text{-BaF}_2\text{-NaF-NdF}_3$ system was unexpectedly a glass. One year later, the first description of "fluorozirconate" glasses was produced with additional vitreous compositions [29]. ZBLAN glass, referring to a mixture of $\text{ZrF}_4\text{-BaF}_2\text{-LaF}_3\text{-AlF}_3\text{-NaF}$ with a molar composition of 53% ZrF_4 , 20% BaF_2 , 4% LaF_3 , 3% AlF_3 , and 20% NaF, considered as the most stable glass practical for optical fiber applications, was reported first by Ohsawa et al. of the Furukawa Electric Co. in 1981 [30]. Although Poulain's $\text{ZrF}_4\text{-BaF}_2\text{-NaF}$ ternary system was not

a very stable glass, it represents 90–95 percent of the most common standard composition of ZBLAN glass.

Compared to silica glasses, it is more complicated to make fluoride glasses. This is due to the fact that to achieve high quality fluoride glasses, not only extremely pure starting materials with infrared-absorbing impurities below 1 ppb are needed but also absolute absence of water and prevention of contamination are required at all synthesis stages. Ammonium hydrogen difluoride ($\text{NH}_4\text{F}\cdot\text{HF}$) is extensively used as a fluorinating agent for conversion of metal oxides to fluorides and also as an additive to fluoride mixtures to provide a fluorinating atmosphere during meltdown. The first step of manufacturing fluoride glass is to synthesize and purify the metal fluorides as precursors. After weighing and mixing the various components, the main work consists in melting the batch and pouring the melt onto a hot mold. Generally, any liquid is likely to form a glass if it is cooled fast enough. However, the HMF glasses are particularly prone to crystallization because of the closeness of their glass transition temperature to their melting points and their low melt viscosities. Therefore, most fluoride glasses must be cooled at rates in excess of 0.1 K/s to completely avoid the occurrence of crystallization. After synthesis, fluoride glasses need to be completely annealed before they are exposed in common environments.

It should be noted that the basic ZBLAN composition is frequently modified to change the glass properties. For instance, viscosity and refractive index may be modified by the addition of small quantities of ZnF_2 , CaF_2 , and PbF_2 ; infrared edge can be shifted from 6–7 μm to 7–8 μm and 8–9 μm by substituting ZrF_4 by HfF_4 and ThF_4 , respectively; AlF_3 may be substituted by GaF_3 or InF_3 and NaF by LiF to control the glass stability.

2.2. Properties of ZBLAN Glass. Because of their broad transmission window, low optical dispersion, low refractive index, ease of machining and polishing, and small thermal dependence of the optical properties, HMF glasses have drawn significant attention since Poulain's unexpected discovery. ZBLAN glass as the most stable HMF glass was extensively investigated for their ease of fiber drawing. In this subsection, properties of ZBLAN glass are summarized with a comparison to those of silica glass.

Because the fluoride ion is singly charged, bond strengths are lower in ZBLAN glass than in silica glass. The weaker bonding leads to greater infrared transparency and higher chemical reactivity. Simply speaking, infrared edge of ZBLAN glass is much longer than that of silica glass, but stability and hardness of ZBLAN glass is much lower than those of silica glass. This suggests that ZBLAN glass will be susceptible to handling damage and ZBLAN fibers need special coating to improve their strength for practical applications.

Optical glasses, especially those used for fiber fabrication, should have a background loss as small as possible due to the relative long length of optical fibers (from meters to kilometers). Attenuation of light propagating in an optical glass comes from intrinsic and extrinsic processes.

TABLE 1: Comparison of basic properties between silica and ZBLAN glasses [20–28].

Glass property	Silica	ZBLAN
Approximate transmission range (1 mm thickness, $T > 10\%$) (μm)	0.16–4.0	0.22–8.0
Maximum phonon energy (cm^{-1})	1100	600
Transition temperature ($^{\circ}\text{C}$)	1175	260
Specific heat ($\text{J}/(\text{g}\cdot\text{K})$)	0.179	0.151
Thermal conductivity, $\text{W}/(\text{m}\cdot\text{K})$	1.38	0.628
Expansion coefficient ($10^{-6}/\text{K}$)	0.55	17.2
Density (g/cm^3)	2.20	4.33
Knoop hardness (kg/mm^2)	600	225
Fracture toughness ($\text{MPam}^{1/2}$)	0.72	0.32
Poisson's ratio	0.17	0.17
Young's modulus (Gpa)	70	58.3
Shear's modulus (Gpa)	31.2	20.5
Bulk's modulus (Gpa)	36.7	47.7
Refractive index (@ 0.589 μm)	1.458	1.499
Abbe number	68	76
Zero material dispersion wavelength (μm)	1.3	1.6
Nonlinear index (10^{-13} esu)	1	0.85
Thermo-optic coefficient ($10^{-6}/\text{K}$)	11.9	-14.75

Intrinsic processes include band-gap absorption, Rayleigh scattering, and multiphonon absorption. Because band-gap absorption and Rayleigh scattering are only significant at short wavelengths, lower intrinsic loss can be obtained by shifting infrared edge of multiphonon absorption to longer wavelengths. It is straightforward that materials with the lower bond strength and the higher reduced mass would be expected to have fundamental absorptions at longer wavelengths. Both these criteria are met in ZBLAN glass since fluoride anion is only single charged and the average cation mass is typically 90. Consequently, the minimum loss coefficient for ZBLAN glass is predicted as low as 0.01 dB/km at 2.5 μm [24], which is much less than silica's 0.2 dB/km at 1.5 μm . This stimulated a tremendous interest on ZBLAN fiber as an alternative fiber for long-haul optical communications in 1980s. The attenuation coefficient of less than 1 dB/km has been demonstrated on ZBLAN fibers fabricated in labs [31]. However, the typical background loss of commercial ZBLAN fibers (10–100 dB/km) is still much larger than that of commercial silica fibers (0.2 dB/km) due to the extrinsic scattering and absorption that are relatively difficult to control in the fabrication process. Extrinsic absorption originates from impurities such as transition metals (Fe, Cu, Co, Ni), rare earths (Nd, Tb, Er), OH and H₂O, and other molecular species (CO, CO₂, NH₄⁺) and extrinsic scattering is resulted from bubbles, crystals, inclusions, and geometrical defects impurities.

Physical properties of ZBLAN glass and silica glass are listed in Table 1. Obviously, the mechanical, optical, and thermal resistibility of ZBLAN glass is much lower than that of silica glass. As a result, output levels of ZBLAN fiber lasers would be much lower than those of silica fiber lasers.

2.3. ZBLAN Fiber Fabrication. Since ZBLAN glass can be cooled more slowly than 1 K/s without noticeable homogeneous nucleation, it has been considered to be as the most stable HMF glass and the most resistant to crystallization under optical fiber preform-making and drawing condition. Three years after Pollain's discovery, fabrication of fluoride glass fibers was performed at the Center National d'Etudes des Telecommunications (CNET), France, and the Nippon Telephone and Telegraph Public Corporation (NTT), Japan [32]. Since then, substantial progress has been made in improving the optical attenuation, mechanical strength, chemical durability, and manufacturing processing of fluoride glass fibers. So far, the background loss of a commercial ZBLAN fiber can be <50 dB/km in the 0.5–3.5 μm wavelength region. UV-curable poly-acrylate is usually used to coat ZBLAN fiber so that the mean tensile strength can be up to 700 MPa. Although poor chemical durability of ZBLAN inevitably leads to the degradation of fiber strengths by moisture attack, coating with various metals or highly stable oxides and fluorides was suggested to significantly improve the durability of a ZBLAN fiber.

Like the fabrication of silica fibers, the preform method in which the fiber is drawn from a preformed glass rod at a temperature below the glass melting temperature is mostly used for ZBLAN fiber fabrications. ZBLAN fiber preforms can be prepared by different casting processes including cladding-over-core casting (Mitachi et al. [33]), build-in casting (Mitachi et al. [33, 34]), and rotational casting (Tran et al. [35]). During the casting processes the critical temperature range between liquidus and transition temperature must be passed through rapidly because of the large crystallization tendency of ZBLAN glass. Normally the

preform is made in separated glove box that allows complete processing of the glasses with an inert environment. In other words, ambient condition has to be avoided at all times and only the completed annealed preform is allowed to come into contact with air.

For a fiber preform, refractive indices of the core and the cladding must be precisely controlled to obtain a required index difference. In typical fluoride glasses, PbF_2 and BiF_3 raise the index, and LiF and AlF_3 lower the index. Replacing ZrF_4 by HfF_4 in a ZBLAN glass also leads to a small reduction in index; so the latter is often used in cladding glass.

During the drawing process, preform and drawing speed must be precisely controlled because of the narrow working range influenced by the ZBLAN glass viscosity and the crystallization rate. In order to reduce the crystallization rate ZBLAN fibers should be drawn with a high drawing speed at temperatures as low as possible. The drawing temperature is usually in the 340–400°C range. Because moisture induces devitrification through hydration reactions, all fibers guiding devices are flushed by dry nitrogen before the fiber is coated with a UV-curable acrylate polymer.

It should be noted that, for each particular glass composition, the largest piece that can be made is determined by its crystallization rate during cooling. Therefore, the dimension of a preform is limited and consequently so is the total length of a ZBLAN fiber. This consideration should be included in the design of double-clad ZBLAN fibers for high-power operations. Although ZBLAN fibers with unlimited length can be directly drawn utilizing the crucible technique [36], it is hard to eliminate crystallization during the fiber drawing. Moreover, this method cannot be used for specially-designed fibers such as double-clad fibers with D-shaped or rectangular pumping cladding and microstructured fibers.

3. High-Power ZBLAN Fiber Lasers

Because of their wide transparency window, ZBLAN fibers have been suggested for a lot of applications including infrared imaging, remote infrared spectrometry, remote infrared spectrometric imaging, remote thermometry, laser power delivery, and fiber lasers. In the past decade, with advances in high-brightness semiconductor diode lasers, novel fiber designs, and new pumping schemes, fiber lasers have been developed with an tremendous speed due to the increasing demand of high-power fiber lasers in a variety of applications, such as material processing, telecommunications, spectroscopy, laser pumps, directed energy weapons, and medicine. ZBLAN fiber lasers have also experienced elevation of the output power level and expansion of the operation spectral region. In this section, ZBLAN fiber lasers are reviewed with three categories: infrared, upconversion UV and visible, and supercontinuum.

3.1. Infrared ZBLAN Fiber Lasers. Shortly after the emergence of fluoride glasses it was suggested that they may make particularly good hosts for a lasing material due

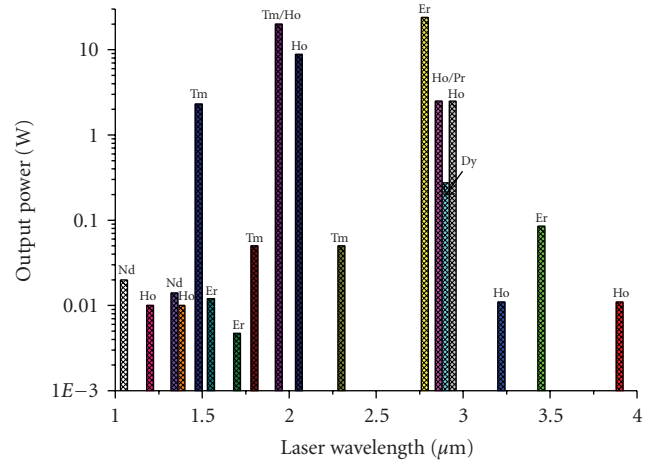


FIGURE 1: Highest output powers of infrared rare-earth-doped ZBLAN fiber lasers at different emission wavelengths.

to the extended infrared edge and low phonon energy [37]. Although many researchers had studied fluorescence in fluoride glasses it was not until 1987 when the first fluoride glass laser was developed on Nd^{3+} -doped multimode fluoride fiber pumped by a argon ion laser at 514 nm [38]. Laser emission at 1.05 μm was measured with a threshold launched pump power of 300 mW and an efficiency of about 0.03%. Er^{3+} -doped ZBLAN fiber laser at 1.56 μm [39] and Nd^{3+} -doped ZBLAN fiber laser at 1.35 μm [40, 41] were soon followed. The 1.35 μm emission is significant because it cannot be obtained from a silica host. Shortly after various rare-earth-doped ZBLAN fiber lasers were demonstrated in different wavelength regions where silica fiber lasers are absent. For instance, 2.9 μm, 3.22 μm, and 3.9 μm emissions were obtained from Ho^{3+} -doped ZBLAN fiber lasers [42–44], 2.7 μm and 3.45 μm emissions from Er^{3+} -doped ZBLAN fiber lasers [45–47], and 2.3 μm emission from Tm^{3+} -doped ZBLAN fiber lasers [48].

Activated by the increasing demand of high-power lasers for various applications, power scaling of fiber lasers has been mainly driven by the progresses of high-brightness semiconductor diode pumps, novel pump schemes, and new fiber designs. Rare-earth-doped ZBLAN fiber lasers have also benefited from these progresses. The highest output powers of infrared ZBLAN fiber lasers at different wavelengths are summarized in Figure 1. Critical parameters of these fiber lasers are listed in Table 2. Progress of infrared ZBLAN fiber lasers and their lasing mechanisms are described sequentially as follows.

3.1.1. Infrared Er^{3+} -Doped ZBLAN Fiber Lasers. The partial energy level diagram of Er^{3+} ions to describe the transitions involving in infrared emissions is plotted in Figure 2. When the Er^{3+} ions are excited to the upper energy levels via ground-state absorption (GSA) and excited-state absorption (ESA) at pump wavelengths of 655 nm, 790 nm, and 975 nm, radiative transition $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{9/2}$ produces an emission of

TABLE 2: Summary of infrared rare-earth-doped ZBLAN fiber lasers with the highest output powers so far.

Rare-earth	Laser wavelength (μm)	Pump wavelength (nm)	Output power (W)	Rare-earth Concentration (mol%)	Slope efficiency (%)	Reference
Er	1.55	980	0.012	0.5	30	[49]
	1.7	791	0.007	0.5	1.8	[50]
	2.7	975	24	6	14.5	[16]
	3.45	640	0.085	1	2.8	[47]
Tm	1.48	1064	2.3	0.2	65	[51]
	1.94	792	20	2.5	49	[15]
	2.3	790	0.001	0.1	10	[48]
Ho	2	806	8.8	0.4	36	[52]
	2.86	1100	2.5	3	29	[53]
	3.22	532	0.011	0.2	2.8	[43]
	3.9	885	0.011	0.2	1.5	[54]
Dy	2.9	1100	0.275	0.1	4.5	[55]
Pr	1.3	1064	0.0045	0.09	0.45	[56]
Yb	1	0.911	0.09	1.8	56	[57]
Nd	1.05	514.5	0.02	1.5	2.5	[38]
	1.34	800	0.0136	0.2	12	[58]

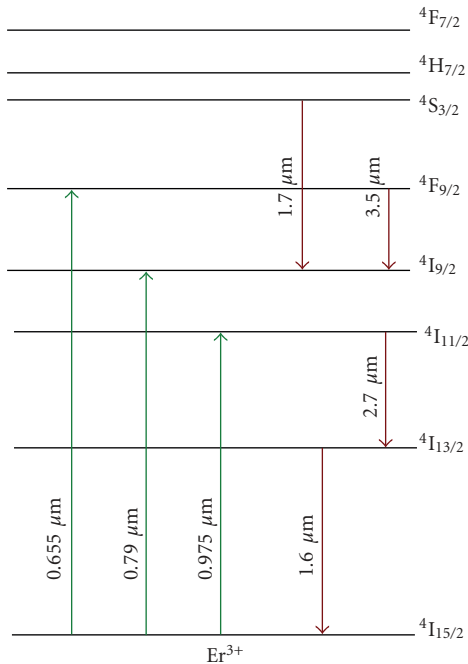


FIGURE 2: Partial energy level diagram of Er^{3+} ion in ZBLAN related to infrared laser emissions. Pump absorption transitions are indicated with green upward arrows. Radiative emission transitions are indicated with red downward arrows.

$1.7 \mu\text{m}$, ${}^4\text{I}_{11/2} \rightarrow {}^4\text{I}_{13/2}$ produces an emission of $2.7 \mu\text{m}$, ${}^4\text{I}_{13/2} \rightarrow {}^4\text{I}_{15/2}$ produces an emission of $1.55 \mu\text{m}$, and ${}^4\text{F}_{9/2} \rightarrow {}^4\text{I}_{9/2}$ produces an emission of $3.45 \mu\text{m}$.

Although $1.55 \mu\text{m}$ emission was first demonstrated in Er^{3+} -doped ZBLAN fiber [39], there is few further investigations at this wavelength because Er^{3+} -doped ZBLAN fiber is not as competitive as stable and low-cost Er^{3+} -doped silica fiber which has been extensively used for optical amplifiers at $1.5 \mu\text{m}$ telecommunication window. Most recently, some investigations [49, 59] have been conducted on an $\text{Er}^{3+}/\text{Ce}^{3+}$ -codoped ZBLAN fiber laser that has a wide tuning range of $1490 \text{ nm} - 1610 \text{ nm}$ compared to $1520 \text{ nm} - 1600 \text{ nm}$ of silica fiber laser. However, the output power was only 12 mW and the slope efficiency was approximate 30% . Nevertheless, $\text{Er}^{3+}/\text{Ce}^{3+}$ -codoped ZBLAN fiber may be promising for ultrafast pulse generation because of its wide and flat gain profile.

Since $1.7 \mu\text{m}$ emissions in Er^{3+} -codoped ZBLAN fiber generally result from the accessorial transitions in the cascade lasing of $2.7 \mu\text{m}$ laser [50, 60, 61] and there is few report on independent operation at the this wavelength [62], we will not address this transition here.

Because $2.7 \mu\text{m}$ emission is close to the highest absorption peak of water around $3 \mu\text{m}$ and thereby can be extensively used in laser surgery, high-power laser emission at this wavelength has been pursued with significant efforts [14, 16, 45, 50, 60–91].

Generally, the $2.7 \mu\text{m}$ lasing is a self-terminating transition, because the longer natural lifetime of the lower laser level (${}^4\text{I}_{13/2}$, $\tau_1 = 9 \text{ ms}$) relative to that of the upper laser level (${}^4\text{I}_{11/2}$, $\tau_2 = 6.9 \text{ ms}$) of erbium ions in ZBLAN glass may restrict lasing to a pulsed mode. However, continuous-wave (CW) multimode emission was observed in the first $2.7 \mu\text{m}$ Er^{3+} -doped ZBLAN fiber laser pumped by an argon

ion laser at 476.5 nm [45]. This unusual CW lasing of the self-terminating transition was explained by ESA depleting the lower level $^4I_{13/2}$ and maintaining the population inversion. The first single-mode Er^{3+} -doped ZBLAN fiber laser was demonstrated by Allain et al. [64] and an output power of 250 μW was measured. Soon several low-efficiency diode-pumped fiber lasers were reported [65–67]. Compared to the low efficiency (<10%) and low power operations (<10 mW) of these singly Er^{3+} -doped ZBLAN fiber lasers, higher output power of 30 mW with an efficiency more than 13% was obtained in an $\text{Er}^{3+}/\text{Pr}^{3+}$ -codoped ZBLAN fiber laser [68]. Since these CW operations were demonstrated at relatively low power levels, the population bottleneck caused by the shorter lifetime of the upper laser level relative to the lower laser level was not noticed until Bedo [69, 70] reported the saturation of the 2.71 μm laser output from Er^{3+} -doped ZBLAN fibers with different concentrations pumped by a Ti:sapphire laser at 791 nm. Colasing at 1.7 μm via transition $^4S_{3/2} \rightarrow ^4I_{9/2}$ was suggested to suppress the competitive lasing on the 850 nm transition $^4S_{3/2} \rightarrow ^4I_{13/2}$ and relieve the population bottleneck [50, 60, 61]. Thereafter, 150 mW unsaturated output was achieved from a cascade Er^{3+} -doped ZBLAN fiber laser [71]. However, this type of cascade-lasing is only efficient in low-doped, core-pumped fiber lasers and cannot be used for low-brightness diode-pumped high-power (watt-level) fiber lasers. Jackson recently demonstrated a novel cascade fiber laser in which the 1.6 μm transition was utilized to prevent population bottleneck by radiative decay and watt-level output was obtained [72].

In order to elevate the output power of 2.7 μm fiber lasers, significant investigations on lasing mechanisms and power scaling schemes have been completed. So far, energy-transfer process between Er^{3+} ions and codoped Pr^{3+} ions and energy-transfer upconversion process between Er^{3+} and Er^{3+} ions have been proven most efficient to depopulate the lower laser level $^4I_{13/2}$ and solve the population bottleneck even with low-brightness pumping. Watt-level and 10-watt-level 2.7 μm lasers have been demonstrated in high-concentration singly Er-doped (>1 mol%) ZBLAN double-clad fibers [14, 16, 78, 87, 88, 90, 91] and Er/Pr codoped ZBLAN double-clad fibers [76, 77, 89, 90]. It should be noted that, the output of the first 10-watt-level 2.7 μm laser [14] was only constrained by the thermo-optic effects at high-power operation. Much stronger emissions are recently achieved when a large-core fiber is used and efficient heat dissipation is employed [16]. Increasing the efficiency of a laser can boost up the output power without increasing the pump power and some theoretical calculations suggest that heavily Er-doped (>10 mol.%) 2.7 μm fiber lasers possibly make a slope efficiency close to or even larger than the Stokes limits [84, 85]. However, much work is still needed to experimentally determine the optimum concentration for maximum efficiency and to fabricate low-loss ZBLAN fibers with such high-concentration. Technically, crystallization is an obstacle of fabricating high concentration ZBLAN fibers and 10 mol% is the maximum concentration of rare-earth ions in ZBLAN without inducing detrimental crystallization. Though zero-gravity environment has been proven to significantly reduce the crystallization during ZBLAN fabrication

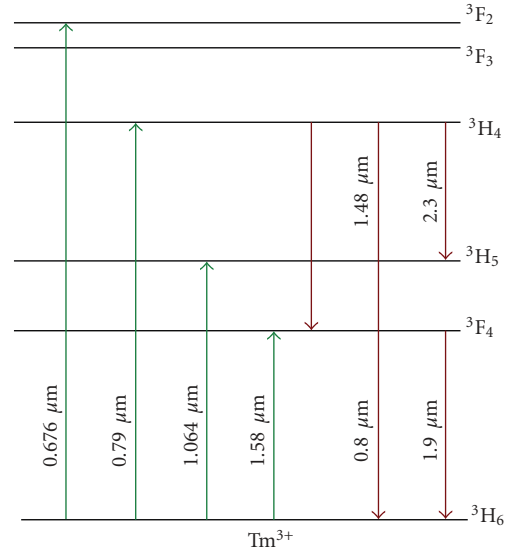


FIGURE 3: Partial energy level diagram of Tm^{3+} ion in ZBLAN related to infrared laser emissions. Pump absorption transitions are indicated with green upward arrows. Radiative emission transitions are indicated with red downward arrows.

process [92], there is little feasibility of fabricating heavily Er-doped ZBLAN fibers (>10 mol%) for high efficiency lasers in the near future. Therefore, using large-mode-area core or multicore ZBLAN fibers will be the effective approach to obtain 100-watt-level output at 2.7 μm in the future.

In 1999, Tobben reported the laser emission of 3.5 μm in a 12-cm-long Er^{3+} -doped ZBLAN fiber pumped with a DCM dye laser at 655 nm [46]. However, the operation temperature was 77 K. Room temperature CW operation of this fiber laser was demonstrated shortly after with an output power of 2.5 mW and a slope efficiency of 2.8% [47]. But the threshold is much larger than the cooled operation. Since emission of 3.5 μm can be obtained only when Er^{3+} -doped ZBLAN is pumped with a short wavelength pump at 655 nm, it is challenging to achieve high-power output at this wavelength due to the small Stokes efficiency and the unavailability of high-power laser diode at 655 nm. Nevertheless, 3.5 μm fiber laser is of much interest because various hydrocarbon and hydrochloride groups and several commonly used solvents show strong absorption in the wavelength range from 3.2 μm to 3.6 μm . Furthermore, the atmosphere has a minimum attenuation in the window between 3 μm and 4 μm . Watt-level output at 3.5 μm can be achieved in Er^{3+} -doped ZBLAN fibers in the future by using novel fiber designs and new pump schemes.

3.1.2. Infrared Tm^{3+} -Doped ZBLAN Fiber Lasers. The partial energy level diagram of Tm^{3+} ions to describe the transitions producing infrared emissions is plotted in Figure 3. Three radiative de-excitation processes from the 3H_4 level emit at 0.8 μm , 1.47 μm , and 2.3 μm with branching ratio of 0.893, 0.083, and 0.024. Emission of 1.9 μm can be obtained from the transition $^3F_4 \rightarrow ^3H_6$.

The $1.47\ \mu\text{m}$ laser oscillation cannot be realized in silica glass fibers because the transition ${}^3\text{H}_4 \rightarrow {}^3\text{F}_4$ is a fast nonradiative relaxation which rapidly reduces the population of excited ${}^3\text{H}_4$ level. However, like the CW operation of $2.7\ \mu\text{m}$ in Er^{3+} -doped ZBLAN, due to the longer lifetimes of ions in ZBLAN relative to those in silica, the $1.47\ \mu\text{m}$ emission can be achieved in Tm^{3+} -doped ZBLAN even it is a self-terminating transition. In 1989, Allain et al. [93] demonstrated laser oscillation around $1.48\ \mu\text{m}$ using a Tm^{3+} -doped multimode fluoride fiber for the first time using a krypton ion laser operating at $676\ \text{nm}$ as a pump source. Since the transition ${}^3\text{H}_4 \rightarrow {}^3\text{F}_4$ is a self-terminating system, the slope efficiency of this fiber laser was very low and the output power was a few tens of μW . Three approaches have been proven able to solve the self-terminating problem. One is colasing at $1.9\ \mu\text{m}$ via the transition ${}^3\text{F}_4 \rightarrow {}^3\text{H}_6$ that can depopulate the ${}^3\text{F}_4$ level efficiently [94, 95]. Another is to shorten the actual lifetime of the lower level by energy transfer to codopant such as Tb^{3+} or Ho^{3+} ions [96]. The most efficient way is upconversion pumping at $1064\ \text{nm}$ using ESA from the ${}^3\text{F}_4$ to the ${}^3\text{F}_2$ level to depopulate the lower laser level [97–99].

In 1992, Komukai et al. [97] observed highly efficient $1.47\ \mu\text{m}$ continuous-wave oscillation in Tm^{3+} -doped ZBLAN fiber using $1064\ \text{nm}$ upconversion pumping. A watt-level fiber laser was soon demonstrated with this scheme [100]. The upconversion pumping process can be described as follows. First ground state ions are excited to the ${}^3\text{H}_5$ level and decay to the metastable ${}^3\text{F}_4$ level by multiphonon relaxation. These ions are then reexcited to the ${}^3\text{F}_2$ level by ESA process and rapidly decay to the ${}^3\text{H}_4$ level. This ESA process ${}^3\text{F}_4 \rightarrow {}^3\text{F}_2$ maintains the population inversion by reducing the ${}^3\text{F}_4$ population and increasing the ${}^3\text{H}_4$ population. In order to avoid the effect of cross-relaxation process that starts from the upper laser level ${}^3\text{H}_4$, a large core area ZBLAN fiber doped with low-concentration Tm^{3+} ions (0.1 wt%) was used in a $1.56\ \text{W}$ $1.47\ \mu\text{m}$ fiber laser pumped at $1064\ \text{nm}$ [101]. Most recently, $1.48\ \mu\text{m}$ single-mode fiber lasers based on fiber Bragg gratings directly inscribed in the ZBLAN fiber core by femtosecond pulses at $800\ \text{nm}$ were demonstrated [51, 102]. Maximum output power of $2.3\ \text{W}$ and a conversion efficiency of 65% have been achieved when the Tm^{3+} -doped ZBLAN fiber was pumped with an Yb^{3+} -doped silica fiber laser at $1064\ \text{nm}$. Since the output power was only limited by the available pump power, ten-watt or even 100-watt all-fiber laser at $1.48\ \mu\text{m}$ can be realized in the near future.

In 1989, Allen and Esterowitz [48] reported the first successful operation of a GaAlAs diode ($790\ \text{nm}$) pumped CW Tm^{3+} -doped ZBLAN fiber laser at $2.3\ \mu\text{m}$. Output power of $1\ \text{mW}$ and slope efficiency of 10% were demonstrated. In the same year, a tunable CW lasing around $0.82\ \mu\text{m}$, $1.48\ \mu\text{m}$, $1.88\ \mu\text{m}$, and $2.35\ \mu\text{m}$ was demonstrated in a Tm^{3+} -doped ZBLAN fiber [93]. In 1991, Percival et al. [103] demonstrated a wavelength tunable fiber laser pumped by a $790\ \text{nm}$ Ti:sapphire laser. The wavelength was continuously tunable from $2.25\ \mu\text{m}$ to $2.5\ \mu\text{m}$. For the $2.3\ \mu\text{m}$ emission, the transition ${}^3\text{H}_4 \rightarrow {}^3\text{H}_5$ is followed by a rapid nonradiative decay to the ${}^3\text{F}_4$ level. Since the lifetime of the ${}^3\text{H}_4$ level (1.5 ms)

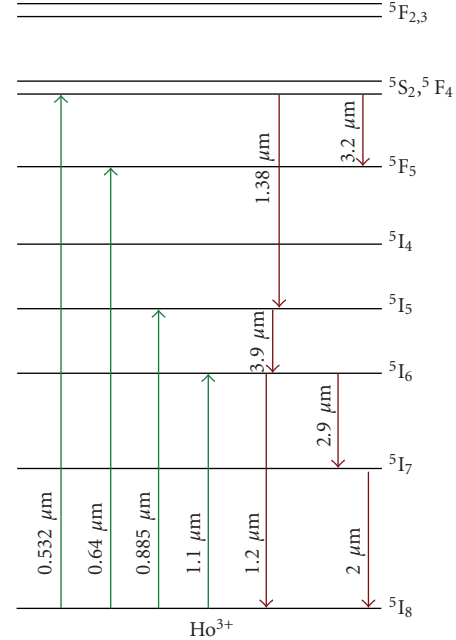


FIGURE 4: Partial energy level diagram of Ho^{3+} ion in ZBLAN related to infrared laser emissions. Pump absorption transitions are indicated with green upward arrows. Radiative emission transitions are indicated with red downward arrows.

is shorter than that of the ${}^3\text{F}_4$ level (6.8 ms), population builds up in the ${}^3\text{F}_4$ level and consequently the performance of the laser is severely curtailed. This undesirable effect can be eliminated by enforcing simultaneous lasing at $1.9\ \mu\text{m}$ on the ${}^3\text{F}_4 \rightarrow {}^3\text{H}_6$ transition. Due to the increasing demand of fiber lasers beyond $2\ \mu\text{m}$, watt-level $2.3\ \mu\text{m}$ fiber laser should emerge very soon.

In 2008, Eichhorn and Jackson [15] did a comparative study of CW Tm^{3+} -doped silica and fluoride fiber lasers at $1.9\ \mu\text{m}$. Ten-watt-level Tm^{3+} -doped ZBLAN fiber laser pumped by two laser diodes at $792\ \text{nm}$ was demonstrated. Output power of $20\ \text{W}$ with a slope efficiency of 49% was obtained in a $30\ \mu\text{m}$ core ZBLAN fiber highly doped with Tm^{3+} ions concentration of 2.5 mol%. Although the output of the fluoride fiber laser is not comparable with that of the silica fiber laser, his experimental results indicate that tens of watts ZBLAN fiber lasers are feasible when newly designed fibers and novel pumping schemes are utilized with optimized operation conditions.

3.1.3. Infrared Ho^{3+} -Doped ZBLAN Fiber Lasers. The partial energy-level diagram of Ho^{3+} ions to describe the transitions involving in infrared emissions is plotted in Figure 4. Ho^{3+} -doped ZBLAN has more infrared transitions comparing to other rare-earth-doped ZBLANs. $1.2\ \mu\text{m}$, $1.38\ \mu\text{m}$, $2\ \mu\text{m}$, $2.9\ \mu\text{m}$, $3.2\ \mu\text{m}$, and $3.9\ \mu\text{m}$ emissions can be achieved from the transitions, ${}^5\text{I}_6 \rightarrow {}^5\text{I}_8$, ${}^5\text{F}_4 \rightarrow {}^5\text{I}_5$, ${}^5\text{I}_7 \rightarrow {}^5\text{I}_8$, ${}^5\text{I}_6 \rightarrow {}^5\text{I}_7$, ${}^5\text{F}_4 \rightarrow {}^5\text{F}_5$, and ${}^5\text{I}_5 \rightarrow {}^5\text{I}_6$, respectively.

In 1988, Brierley et al. [104] observed the lasing at $2.08\ \mu\text{m}$ and $1.38\ \mu\text{m}$ from a Ho^{3+} -doped ZBLAN fiber for the first time using an argon ion pump laser at $488\ \text{nm}$.

However, the efficiency is very low (<0.3%) and the operation at $2\ \mu\text{m}$ was not CW at room temperature. Higher efficient (30%) CW operation was demonstrated when a Ho^{3+} -doped ZBLAN fiber was pumped by a Ti:sapphire laser at 890 nm [105]. Allain et al. [106] demonstrated a high-efficiency CW $2\ \mu\text{m}$ ZBLAN fiber laser in which Ho^{3+} ions are sensitized with Tm^{3+} ions. Tm^{3+} ions are excited in the $^3\text{H}_4$ level by a pump of wavelength around $0.8\ \mu\text{m}$. If the Tm^{3+} concentration is high enough, cross relaxation process between two neighboring Tm^{3+} ions can double the pumping efficiency because one Tm^{3+} ion in the $^3\text{H}_4$ level relaxes to the $^3\text{F}_4$ level with simultaneous excitation of another Tm^{3+} ion from the $^3\text{H}_6$ level to the $^3\text{F}_4$ level. Energy transfer from Tm^{3+} ions to Ho^{3+} ions leads to the population of the $^5\text{I}_7$ level. Therefore, using Tm^{3+} codoping can significantly improve the performance of $2\ \mu\text{m}$ Ho^{3+} -doped ZBLAN fiber laser. An output power of 250 mW, a slope efficiency of 52%, and a pump efficiency of 130% were achieved in this $\text{Tm}^{3+}/\text{Ho}^{3+}$ codoped system. In 2001, Jackson [52] demonstrated an 8.8 W $\text{Tm}^{3+}/\text{Ho}^{3+}$ codoped fiber laser at this wavelength by using two high-power pump diodes at 800 nm.

Because of the larger water absorption coefficient at $3\ \mu\text{m}$ ($\sim 10^4\ \text{cm}^{-1}$) relative to $2\ \mu\text{m}$ ($\sim 10\ \text{cm}^{-1}$), $3\ \mu\text{m}$ band of Ho^{3+} ions has been of much interest for laser surgery. The first $3\ \mu\text{m}$ Ho^{3+} -doped ZBLAN fiber laser was demonstrated by two pump lasers at 640 nm and 750 nm [107]. Because the lifetime of the $^5\text{I}_6$ level (3.5 ms) is shorter than that of the $^5\text{I}_7$ level (12 ms), the $3\ \mu\text{m}$ emission is a self-terminating transition. The CW operation was explained by ESA of the pump starting from the $^5\text{I}_7$ level. When pumped at 640 nm, maximum output power of 12.6 mW and the highest slope efficiency of 4.4% were achieved. In order to solve the self-terminating and improve the efficiency of $3\ \mu\text{m}$ transition, a cascade lasing with transitions from $^5\text{I}_6$ to $^5\text{I}_7$, to $^5\text{I}_8$, where the $2\ \mu\text{m}$ oscillation depletes the $^5\text{I}_7$ population, was employed [108, 109]. Maximum total output power of 3 W with a slope efficiency of 65% was obtained in a Ho^{3+} -doped ZBLAN fiber pumped by a Raman fiber laser at 1150 nm [109].

In order to improve the efficiency of the $3\ \mu\text{m}$ emission and eliminate the $2\ \mu\text{m}$ lasing, Ho^{3+} -doped fibers were codoped with Pr^{3+} ions as deactivators because strong energy transfer from the energy level $^5\text{I}_7$ of Ho^{3+} to the energy level $^3\text{F}_2$ of Pr^{3+} allows fast depopulation of the lower laser level $^5\text{I}_7$ [53, 110–113]. So far, the highest output power of 2.5 W with a slope efficiency of 32% was achieved in a $\text{Ho}^{3+}/\text{Pr}^{3+}$ codoped ZBLAN fiber pumped by a Yb^{3+} -doped silica fiber laser at 1100 nm [53]. Alternatively, the $3\ \mu\text{m}$ oscillation can be enhanced while the $2\ \mu\text{m}$ lasing is circumscribed by doping the fiber core with high Ho^{3+} ion concentration to allow significant energy transfer upconversion process between ions and thus depopulate the lower laser level $^5\text{I}_7$. Exclusive operation at $2.92\ \mu\text{m}$ with maximum output power of 340 mW and a slope efficiency of $\sim 5\%$ was achieved in a singly Ho^{3+} -doped ZBLAN fiber with a concentration of 4.3 mol% [114]. It was recently found that highly stable emission at $2.95\ \mu\text{m}$ can be obtained without colasing at $2\ \mu\text{m}$ when Ho^{3+} -doped ZBLAN fiber was pumped at a Raman fiber laser at 1175 nm [115], because the ESA of

pump photons in a transition $^5\text{I}_7 \rightarrow ^5\text{I}_4$ can depopulate the lower laser level and maintain the population inversion in the transition of interest.

Because the strong absorption band of hydrocarbon and hydrochloride groups and a transparency window of atmosphere locate in the $3\ \mu\text{m}$ – $4\ \mu\text{m}$ wavelength region, emissions beyond $3\ \mu\text{m}$ are of much interest. In 1998, Carbonnier et al. [43] reported a room temperature CW Ho^{3+} -doped ZBLAN fiber laser operating at $3.22\ \mu\text{m}$ with an output power of 11 mW and a slope efficiency of 2.8%. This fiber laser was pumped by an Nd:YAG laser at 532 nm. The input coupler had a reflectivity of 100% at the lasing wavelength and output coupler had a reflectivity of 90% at the lasing wavelength.

In 1993, Tobben [116] observed the fluorescence at $3.9\ \mu\text{m}$ in an Ho^{3+} -doped ZBLAN fiber for the first time. This experiment suggested that a transition near $3.9\ \mu\text{m}$ is possible even though the lasing conditions are worse compared with other radiative transitions due to the longer wavelength and the increased multiphonon rates. In 1995, the first $3.9\ \mu\text{m}$ fiber laser was realized in an Ho^{3+} -doped ZBLAN fiber pumped by a DCM dye laser at 640 nm [44]. The $3.9\ \mu\text{m}$ emission is maintained by a cascade lasing at $1.2\ \mu\text{m}$ that depletes the lower laser level $^5\text{I}_6$ to the ground state $^5\text{I}_8$. The output is about 1 mW and the slope efficiency is 1.5%. Shortly after an 11 mW output was achieved with an optimization of the mirror design and the fiber [54]. However, in order to achieve efficient CW $3.9\ \mu\text{m}$ emission, the fiber has to cool down to 77 K because the multiphonon rate increases strongly for temperature above 100 K and for wavelengths above $3\ \mu\text{m}$. Nevertheless laser oscillation without additional cooling (at room temperature) is possible if the gain is large enough at $3.9\ \mu\text{m}$.

3.1.4. Infrared Dy^{3+} -Doped ZBLAN Fiber Lasers. The partial energy-level diagram of Dy^{3+} ions is plotted in Figure 5. Infrared Laser emission at $2.9\ \mu\text{m}$ can be obtained from the radiative transition $^6\text{H}_{13/2} \rightarrow ^6\text{H}_{15/2}$.

There are few reports on Dy^{3+} -doped ZBLAN fiber lasers. The first CW Dy^{3+} -doped ZBLAN fiber laser was demonstrated in 2003 [55]. Maximum output power of 0.275 W was generated at a slope efficiency of 4.5% when the ZBLAN fiber laser was pumped with a diode-cladding-pumped Yb^{3+} -doped silica fiber laser at $\sim 1100\ \text{nm}$. In the continued investigation [117], a slope efficiency was increased to 20% when the ZBLAN fiber laser was pumped by a $1.3\ \mu\text{m}$ Nd:YAG laser because of the increased Stokes efficiency and the reduced ESA of pump photons. However, the output power was 0.18 W limited by the available pump power. High efficiency Raman fiber laser at $1.29\ \mu\text{m}$ was suggested as a powerful pump for the $2.9\ \mu\text{m}$ Dy^{3+} -doped ZBLAN fiber laser. With recent advances in $1\ \mu\text{m}$ silica fiber laser, watt-level or ten-watt-level Dy^{3+} -doped ZBLAN fiber laser can be developed in the near future.

3.1.5. Infrared Nd^{3+} -Doped ZBLAN Fiber Lasers. The partial energy-level diagram of Nd^{3+} ions to describe the transitions involving in infrared emissions is plotted in Figure 6. When

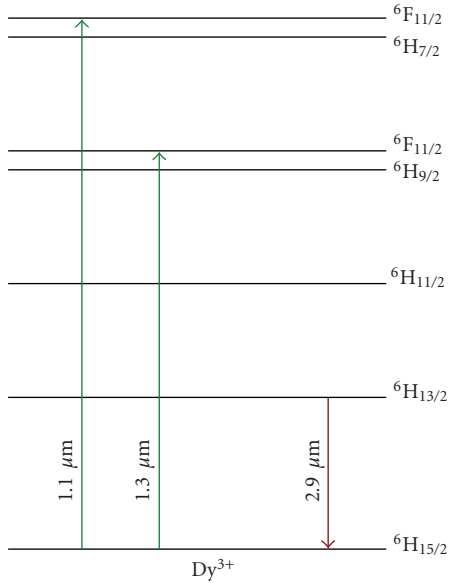


FIGURE 5: Partial energy level diagram of Dy^{3+} ion in ZBLAN related to the $2.9 \mu\text{m}$ laser emission. Pump absorption transitions are indicated with green upward arrows. Radiative emission transition is indicated with red downward arrow.

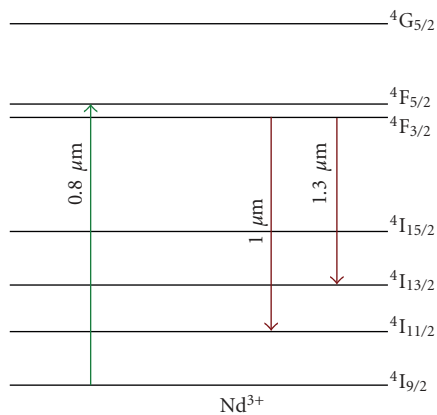


FIGURE 6: Partial energy level diagram of Nd^{3+} ion in ZBLAN related to infrared laser emissions. Pump absorption transition is indicated with green upward arrow. Radiative emission transitions are indicated with red downward arrows.

the ions are excited to the upper energy levels via GSA ${}^4\text{I}_{9/2} \rightarrow {}^4\text{F}_{5/2}$, radiative transition ${}^4\text{F}_{3/2} \rightarrow {}^4\text{I}_{13/2}$ produces an emission at $1.3 \mu\text{m}$, and ${}^4\text{F}_{3/2} \rightarrow {}^4\text{I}_{11/2}$ produces an emission at $1.0 \mu\text{m}$.

As mentioned above, the first fluoride fiber laser was demonstrated on a multimode Nd^{3+} -doped ZBLAN fiber [38]. Laser emission at $1.05 \mu\text{m}$ was observed when the fiber was pumped by an argon ion laser at 514 nm . This ZBLAN fiber laser was considered as significant because the emission wavelength is beyond the typical emission wavelength range ($1.06 \mu\text{m}$ – $1.08 \mu\text{m}$) of widely used Nd^{3+} -doped silica fiber lasers. When high reflection mirror at $1.3 \mu\text{m}$ band was used, $1.33 \mu\text{m}$ – $1.34 \mu\text{m}$ range laser was obtained from the same Nd^{3+} -doped ZBLAN fiber [118]. An output power

of larger than 10 mW and a slope efficiency of 2.6% were obtained. Essentially, there are two constraints for power scaling of Nd^{3+} -doped ZBLAN fiber lasers operating at $1.3 \mu\text{m}$ band. One is the ESA transition ${}^4\text{F}_{3/2} \rightarrow {}^4\text{G}_{7/2}$ which absorbs the signal light around $1.31 \mu\text{m}$. The other is the strong competitive $1.05 \mu\text{m}$ transition which can easily suppress $1.3 \mu\text{m}$ band oscillation. A more efficient Nd^{3+} -doped ZBLAN fiber laser operating in the $1.3 \mu\text{m}$ band pumped at 800 nm was demonstrated by Komukai et al. [58]. The slope efficiency and the maximum output power are 15.7% and 13.6 mW , respectively. This fiber laser was of practical interest due to the wavelength tunable range $1.315 \mu\text{m}$ – $1.348 \mu\text{m}$ in the $1.3 \mu\text{m}$ telecommunication window.

3.1.6. Other Infrared ZBLAN Fiber Lasers. Compared to the rare-earth ions described above, there are few studies on Pr^{3+} and Yb^{3+} singly-doped ZBLAN fiber lasers. However, Pr^{3+} and Yb^{3+} ions have been extensively used as codopants in ZBLAN fibers to effectively enhance the desired transitions including laser emission and pump absorption and suppress competitive emissions [53, 68, 76, 77, 79–81, 84–87, 89, 90, 110–113, 119].

Pr^{3+} -doped ZBLAN fiber has been intensively investigated for $1.3 \mu\text{m}$ optical amplifiers during the 1990's bloom of optical communications. CW lasing action in Pr^{3+} -doped ZBLAN fiber was reported by Durteste in 1991 [56]. Due to the broad gain band, ultra-short pulse generation from Pr^{3+} -doped ZBLAN fibers was then demonstrated by using nonlinear optical loop mirrors [120, 121].

Yb^{3+} -doped ZBLAN fiber laser was only demonstrated by Allain et al. [57] and proposed as a potential pump for Pr^{3+} -doped ZBLAN fiber amplifier at $1.3 \mu\text{m}$ because efficient emission can be obtained at 1020 nm in a ZBLAN fiber but not in a silica fiber.

3.2. Upconversion Ultraviolet and Visible ZBLAN Fiber Lasers. Solid-state lasers operating in the UV and visible spectral region have a lot of applications including laser lighting displays, photolithography, optical data storage, holography, microscopy, and spectroscopy. So far, there are three general methods to generate UV and visible laser emissions. One is using nonlinear frequency doubling or tripling processes. Another is developing short wavelength semiconductor laser diodes such as GaN and ZnSe. The third is employing upconversion emissions in fluoride glasses and crystals in which the nonradiative decay probabilities are relatively low due to the small phonon energy. Because the relatively long effective lifetimes of excited states facilitate a sequential absorption of pump photons either by a single ion or via energy transfer between excited ions, two or more incoming photons can be absorbed by the materials and can be reemitted as a single higher energy photon. Thus, UV and visible emission can be generated by pumping upconversion materials with high intensity pumps at near infrared.

Upconversion emission was observed for the first time in flash-lamp-pumped Er^{3+} - Yb^{3+} and Ho^{3+} - Yb^{3+} codoped BaY_2F_8 crystals by Johnson and Guggenheim [122], who

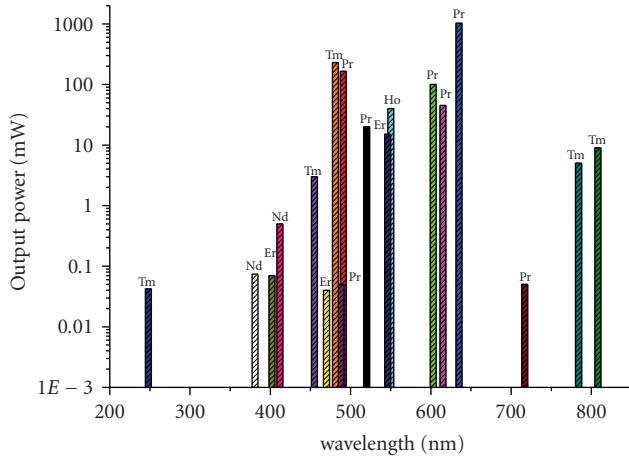


FIGURE 7: Highest output powers of upconversion rare-earth-doped ZBLAN fiber lasers at different emission wavelengths.

obtained stimulated emission at 670 nm and 551 nm. Quimby et al. [123] reported infrared ($1.06 \mu\text{m}$) to visible (550 nm) upconversion in a fluoride glass containing Yb^{3+} and Er^{3+} ions. Okada et al. [124] observed both red (650 nm) and green (550 nm) fluorescence from an $\text{AlF}_3\text{-ZrF}_4$ -based glass containing ErF_3 when pumped at an 807 nm laser diode. The two experiments on fluoride glasses suggested the feasibility of compact and efficient upconversion ZBLAN fiber lasers.

Upconversion ZBLAN fiber lasers are noted for their high efficiency, low threshold, compactness, and tenability. The confinement of both the pump and signal within the fiber core over the long distances reduces the pump threshold and increases output power significantly. High intensity coupled with the long interaction lengths results in optical-to-optical conversion efficiencies larger than 20% and threshold below 10 mW. Moreover, because rare-earth-doped ZBLAN glasses exhibit broad absorption and emission spectral band, it is possible to construct all-fiber wavelength-tunable UV and visible fiber lasers.

In the past twenty years, various upconversion ZBLAN fiber lasers have been demonstrated at UV and visible. Output power over 1 W has already been achieved. The spectrum of upconversion ZBLAN fiber lasers with the highest output powers is summarized in Figure 7. Critical parameters of these fiber lasers are listed in Table 3. Progress of upconversion fiber lasers and their lasing mechanisms are described as follows.

3.2.1. Upconversion Er^{3+} -Doped ZBLAN Fiber Lasers. The partial energy-level diagram of Er^{3+} ions to describe the transitions involving in upconversion emissions is plotted in Figure 8. When Er^{3+} ions are excited to higher energy levels via upconversion pumping, short wavelength emissions at 544 nm, 470 nm, 402 nm, and 317 nm are possible via transitions from $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$, $^2\text{P}_{3/2} \rightarrow ^4\text{I}_{11/2}$, $^2\text{P}_{3/2} \rightarrow ^4\text{I}_{13/2}$, and $^2\text{P}_{3/2} \rightarrow ^4\text{I}_{15/2}$, respectively.

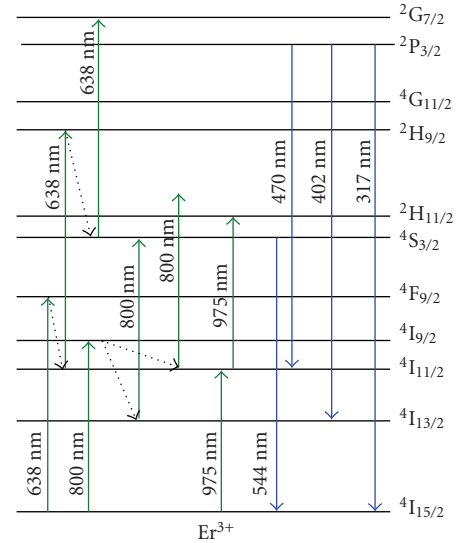


FIGURE 8: Partial energy level diagram of Er^{3+} ion in ZBLAN related to upconversion laser emissions. Pump absorption transitions are indicated with green upward arrows. Radiative transitions are indicated with blue downward arrows. Nonradiative decays are indicated with dotted arrows.

The first demonstration of upconversion lasing in an Er^{3+} -doped ZBLAN fiber was realized using a pump at 801 nm of Ti:sapphire laser [138]. Maximum output of 23 mW CW emission at a wavelength of 546 nm has been achieved with a slope efficiency of 11%. The dominant excitation mechanism is attributed to sequential ESA of pump photons. The Er^{3+} ions are excited into the $^4\text{I}_{9/2}$ level via GSA and some of which then branch into the $^4\text{I}_{11/2}$ and $^4\text{I}_{13/2}$ levels. These ions are then further excited by pump photons into the higher levels, from which a large proportion of the ions then relax into the $^4\text{S}_{3/2}$ level. A direct transition $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ is responsible for the green emission. However, a saturation behavior was observed in this experiment and was thought to be caused by competing laser transition $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{13/2}$ at 850 nm. Allain et al. [126] demonstrated a 50 mW tunable (540 nm–545 nm) green fiber laser in which the drawback of favoring oscillation of 850 nm was avoided by shifting the pump wavelength to 970 nm. Pumping at 970 nm eliminates the competitive lasing because there is no ESA from $^4\text{I}_{13/2}$ at this wavelength. Diode-pumped green fiber lasers at 800 and 971 nm were demonstrated by Massicott et al. [139] and Piehler and Craven [140], respectively.

Aside from pumping at a wavelength of GSA, an upconversion laser can be excited with a wavelength far away from GSA due to the photon avalanche phenomenon which refers to the change in the order of magnitude of the fluorescence when the pump intensity is over a certain critical threshold. Chen et al. [141] demonstrated a room-temperature photon avalanche up-conversion emission in an Er^{3+} -doped ZBLAN fiber. A strong emission at 550 nm as well as a series of other wavelengths has been observed when the excitation wavelength is about 700 nm.

TABLE 3: Summary of upconversion rare-earth-doped ZBLAN fiber lasers with the highest output powers so far.

Rare-earth	Laser Wavelength (nm)	Pump Wavelength (nm)	Output Power (mW)	Rare-earth Concentration (mol%)	Efficiency (%)	Reference
Er	402	638	0.070	0.1	1.6	[125]
	470	638	0.040	0.1	3	[125]
	544	970	50	0.1	11	[126]
Tm	248	1064	0.042	1	9%	[127]
	455	645+1064	3	0.1	1.5%	[128]
	481	1123	230	0.1	18.5%	[129]
	784	1120	5	0.1	0.7%	[130]
	808	1120	9	0.1	2.5%	[130]
Pr/Yb	491	840	165	0.3/2	12.1	[131]
	520	860	20	0.3/2	12.4	[132]
	605	840	55	0.3/2	19	[133]
	615	860	45	0.3/2	11.5	[132]
	635	850+823	1020	0.3/2	19	[134]
Ho	550	645	40	0.12	16.8	[135]
Nd	381	590	0.074	0.1	—	[136]
	412	590	0.5	0.1	1.5	[137]

Generally, efficient shorter-wavelength emissions can be achieved if Er^{3+} ions are excited at a shorter pump wavelength. When the ZBLAN fiber is pumped with red lasers, Er^{3+} ions are excited to the $^4\text{F}_{9/2}$ level and then relax to the $^4\text{I}_{11/2}$ and $^4\text{I}_{13/2}$ levels from which sequential ESA with pump photons results in populating on $^2\text{P}_{3/2}$ and $^4\text{S}_{3/2}$ levels. Green, blue, and violet emissions can be achieved through the radiative decays from the $^4\text{S}_{3/2}$ and $^2\text{P}_{3/2}$ levels [142]. Pope et al. [143] observed an efficient violet upconversion signal from an Er^{3+} -doped ZBLAN fiber pumped at 633.5 nm. Violet 402 nm and blue 470 nm upconversion-emission with power of several $10 \mu\text{W}$ from Er^{3+} -doped ZBLAN fibers have been obtained by Ferber et al. [125] using a laser diode at 638 nm. In the energy upconversion and spectroscopic studies of Er^{3+} -doped ZBLAN, Bullock et al. [144] pointed out that infrared to green, red to blue and violet, and blue to UV energy upconversion processes are able to occur by pumping the glass with different lasers. Most remarkably, when pumped at 488 nm, UV emissions at 322 nm ($^2\text{I}_{11/2} \rightarrow ^4\text{I}_{11/2}$), 352 nm ($^2\text{I}_{11/2} \rightarrow ^4\text{I}_{9/2}$), 275 nm ($^2\text{H}_{9/2} \rightarrow ^4\text{I}_{15/2}$), and 380 nm ($^2\text{H}_{9/2} \rightarrow ^4\text{I}_{11/2}$) were observed. This suggests that efficient UV emission can be achieved with tandem configurations, that is, using visible upconversion fiber lasers to pump UV upconversion lasers.

3.2.2. Upconversion Tm^{3+} -Doped ZBLAN Fiber Lasers. The partial energy-level diagram of Tm^{3+} ions to describe the transitions involving in upconversion emissions is plotted in Figure 9. Usually visible and UV upconversion emissions at 515 nm, 480 nm, 455 nm, 365 nm, 350 nm, and 284 nm can occur in a Tm^{3+} -doped ZBLAN fiber through transition $^1\text{D}_2 \rightarrow ^3\text{H}_5$, $^1\text{G}_4 \rightarrow ^3\text{H}_6$, $^1\text{D}_2 \rightarrow ^3\text{F}_4$, $^1\text{D}_2 \rightarrow ^3\text{H}_6$, $^1\text{I}_6 \rightarrow ^3\text{F}_4$, and, $^1\text{I}_6 \rightarrow ^3\text{H}_6$, respectively.

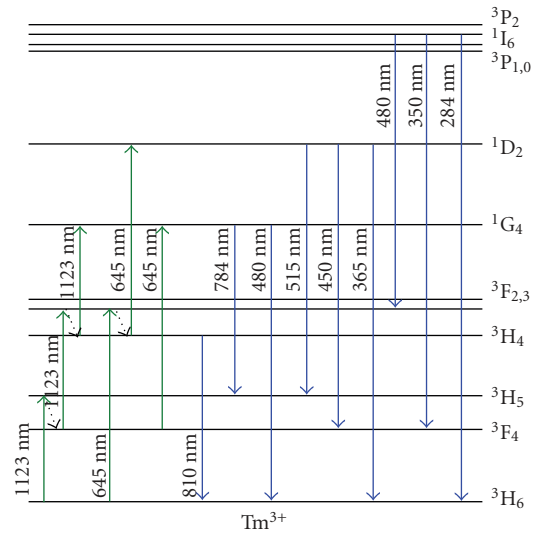


FIGURE 9: Partial energy level diagram of Tm^{3+} ion in ZBLAN related to upconversion laser emissions. Pump absorption transitions are indicated with green upward arrows. Radiative transitions are indicated with blue downward arrows. Nonradiative decays are indicated with dotted arrows.

In 1990, Allain et al. [145] demonstrated the first upconversion laser emission at 480 nm in a Tm^{3+} -doped ZBLAN fiber copumped at 676.4 nm and 647.1 nm. 676.4 nm is used to populate the level $^3\text{H}_4$ by GSA and fast nonradiative decay processes since this wavelength is efficiently absorbed by levels $^3\text{F}_2$ and $^3\text{F}_3$. 647.1 nm is used for ESA to populate the $^3\text{H}_4$ level and then the $^1\text{G}_4$ level. In the experiment, enhanced upconversion emissions at 365 nm, 455 nm, 480 nm, and 515 nm were measured. Weak emissions at 288 nm and

350 nm have also been detected. However, the fiber was immersed in liquid nitrogen at 77 K. The first single-wavelength pumped room temperature blue upconversion laser was demonstrated by Grubb et al. in 1992 [146]. The 1G_4 level is populated by three sequential absorptions of pump photons $^3H_6 \rightarrow ^3H_5$, $^3F_4 \rightarrow ^3F_{2,3}$, and $^3H_4 \rightarrow ^1G_4$. 480 nm laser was obtained in Tm^{3+} -doped ZBLAN glass fiber pumped by an Nd^{3+} :YAG laser at 1120 nm. Up to 60 mW of output power has been measured with a slope efficiency of 18% with respect to coupled pump power. Infrared laser-diode-pumped Tm^{3+} -doped ZBLAN fiber laser was demonstrated by Sanders et al. [147]. Maximum output of 106 mW with a conversion efficiency of 30% was achieved by using two 600 mW laser diodes at 1137 nm. In 1997, Zellmer et al. [148] demonstrated a 100 mW all fiber laser system in which a Tm^{3+}/Yb^{3+} -doped ZBLAN fiber was pumped with a 2.8 W diode pumped Nd^{3+} -doped silica fiber laser at 1065 nm. In the same year, Paschotta et al. [129] achieved the highest output (230 mW) of 480 nm light from a Tm^{3+} -doped ZBLAN fiber laser pumped by a 1.6 W Nd:YAG laser at 1123 nm. However, the fiber laser performance degraded quickly at high-power operation. It is believed that the excitation of higher levels of Tm^{3+} ions via energy transfer processes leads to the creation of color centers that significantly increase the absorption in visible band. Although a few investigations of photodegradation and photobleaching have been conducted [149–152], so far there still has no effective method to prevent photodegradation occurring in high-power operation of Tm^{3+} -doped ZBLAN fiber lasers. Nevertheless, using new pumping schemes [153–156] can improve the CW performance of Tm^{3+} -doped ZBLAN fiber lasers by reducing the pump threshold and increasing the conversion efficiency. A pump threshold as low as 5 mW [155] was demonstrated by using pump sources at 1210 nm and 649 nm and a slope efficiency as high as 48% [156] has been achieved by using Raman-generated pumps at 1117 nm and 1175 nm, respectively. Since 480 nm Tm^{3+} -doped ZBLAN fiber laser is a promising alternative for argon ion laser, a lot of theoretical studies [157–159] have also been carried out to help people understand and improve the performance of this fiber laser.

Another blue emission at 455 nm can be obtained from the transition $^1D_2 \rightarrow ^3F_4$. The 455 nm radiation is self-terminated; therefore the lower laser level 3F_4 is required to be emptied for achieving high-efficiency CW operation. Three different approaches can be utilized to solve the problem: (1) enforcing a cascade laser operation on the $^3F_4 \rightarrow ^3H_6$ transition at $1.9 \mu m$, (2) introducing cross-relaxation processes to relieve the 3F_4 population efficiently by codoping with terbium or europium, and (3) pumping near $1.06 \mu m$ in resonance with the ESA transition $^3F_4 \rightarrow ^3F_{2,3}$. The first room-temperature CW upconversion laser at 455 nm in a Tm^{3+} -doped ZBLAN fiber was demonstrated by Le Flohic et al. [128] using two pump sources at 645 nm and 1064 nm. This pumping scheme populates the 1D_2 level and depopulates the 3H_4 level efficiently. As much as 3 mW of blue laser light has been obtained.

Aside from the blue band, upconversion emissions at other wavelengths have also been observed in Tm^{3+} -doped

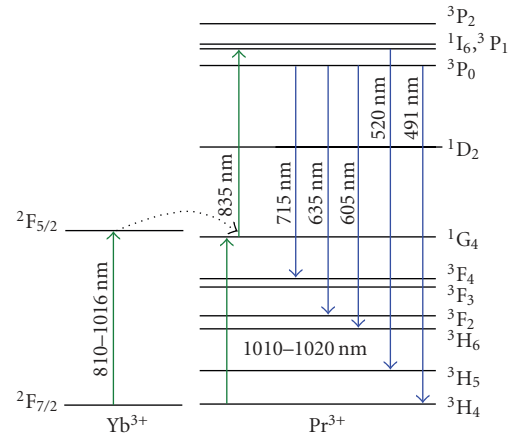


FIGURE 10: Partial energy level diagram of Pr^{3+} ion in ZBLAN related to upconversion laser emissions (Partial energy level diagram of Yb^{3+} ion is also plotted to indicate pump absorption and energy transfer from Yb^{3+} to Pr^{3+}). Pump absorption transitions are indicated with green upward arrows. Radiative emission transitions are indicated with blue downward arrows.

ZBLAN fibers. Single wavelength (784 nm), two-wavelength (785 nm + 805 nm or 788 nm + 793 nm), three-wavelength (787 nm + 794 nm + 802 nm) upconversion laser has been demonstrated in a Tm^{3+} -doped ZBLAN fiber pumped by a 1120 Raman fiber laser [130]. Ultraviolet upconversion signals at 293 nm ($^1I_6 \rightarrow ^3H_6$), 351 nm ($^1I_6 \rightarrow ^3F_4$), and 366 nm ($^1D_2 \rightarrow ^3H_6$) were observed from Tm^{3+} -doped ZBLAN fiber pumped with an argon ion laser at 458 nm [160]. When the fiber was simultaneously pumped by 458 nm and 585 nm laser, enhanced upconversion signals at 293 nm and 351 nm were observed. Most recently, El-Agmy [127] demonstrated a successful CW ultraviolet laser operating at 284 nm in a Tm^{3+} -doped ZBLAN fiber which was pumped with a 1064 nm Nd:YAG laser. A four-step upconversion pumping was involved in exciting Tm^{3+} ions to the upper laser level 1I_6 . Due to the high Tm^{3+} concentration (1 wt%), the favoring cross relaxation process $^1G_4 + ^1G_4 \rightarrow ^1I_6 + ^3F_4$ that populates the 1I_6 level and depletes the 1G_4 level enhances the ultraviolet emission. A laser output power of $42 \mu W$ was obtained for 590 mw of the launched pump power. The slope efficiency with respect to the launched pump power was measured to be 9%.

3.2.3. Upconversion Pr^{3+} -Doped ZBLAN Fiber Lasers Upconversion. The partial energy-level diagram of Pr^{3+} ions to describe the transitions involving in the upconversion emissions is plotted in Figure 10. Pr^{3+} -doped fluoride glass is notable for the white light fluorescence consisting of blue (491 nm), green (520 nm), orange (605 nm), red (635 nm), and infrared (715 nm) emission bands, which are resulted from radiative transitions $^3P_0 \rightarrow ^3H_4$, $^3P_1 \rightarrow ^3H_5$, $^3P_0 \rightarrow ^3H_6$, $^3P_0 \rightarrow ^3F_2$, and $^3P_0 \rightarrow ^3F_4$, respectively.

The first visible oscillation in Pr^{3+} -doped ZBLAN fiber was demonstrated by Allain et al. [161]. Operation at orange and red wavelengths was achieved by pumping the fiber with an argon laser at 476.5 nm. However, it was not upconversion

lasing. Shortly after they obtained upconversion laser at 635 nm in a $\text{Pr}^{3+}/\text{Yb}^{3+}$ -codoped ZBLAN fiber pumped by a Ti:sapphire laser at 850 nm [162]. The first upconversion lasing in a singly Pr^{3+} -doped ZBLAN fiber was reported by Smart et al. [163, 164]. Laser oscillation at 491 nm, 520 nm, 605 nm, and 635 nm was realized by simultaneously pumping the fiber with two Ti:sapphire lasers at 1010 nm and 835 nm. The top energy levels ($^3\text{P}_0$, $^1\text{I}_6$, and $^3\text{P}_1$) are populated by a two-stage pumping: the GSA of 1010 nm light excites Pr^{3+} ions to an intermediate level $^1\text{G}_4$ and the ESA of 835 nm light further elevates excited Pr^{3+} ions to the top levels. An output of 185 mW has been measured for the 635 nm transition with an optical power conversion efficiency of approximate 7%.

Using alternative pumping scheme can improve the performance of Pr^{3+} -doped ZBLAN fiber and enhance the desired upconversion emission. When pumped at 835 nm and 1017 nm, an efficient blue Pr^{3+} -doped fluoride fiber upconversion laser operating at 492 nm was demonstrated with a slope efficiency of more than 13% and output power more than 9 mW [165]. Diode-pumped blue Pr^{3+} -doped ZBLAN fiber laser was demonstrated by Baney [166] using two laser diodes at 830 nm and 1020 nm. Because 492 nm is close to the wavelength of an argon ion laser, this pumping scheme has been further studied [167, 168]. Pask et al. [169] used a Yb^{3+} -doped silica fiber laser pumped at 840 nm to provide two pump wavelengths, 840 nm and 1020 nm that are required for pumping a Pr^{3+} -doped ZBLAN fiber upconversion laser. 54 mW output at 635 nm with a slope efficiency of 23%, 18 mW output at 520 nm with a slope efficiency of 8.5%, and 7 mW output at 491 nm with a slope efficiency of 6% have been achieved.

For these singly Pr^{3+} -doped ZBLAN fiber lasers, however, two pump lasers are required to excite Pr^{3+} ions to the upper laser level. In an effort to relieve the constraint of two-wavelength pumping, Yb^{3+} ions was added as a sensitizer for Pr^{3+} upconversion emission because the broad absorption band of Yb^{3+} ions permits a wide pump wavelength, strong absorption of Yb^{3+} ions permits shorter length fiber that still yield adequate pump absorption, and the simple energy structure of Yb^{3+} ions reduces the possibility for backwards energy transfer from the activator ion.

The first upconversion laser operating at 635 nm was demonstrated in a $\text{Pr}^{3+}/\text{Yb}^{3+}$ -codoped ZBLAN fiber pumped by a Ti:sapphire laser at 850 nm [162]. Yb^{3+} ions are excited by GSA and transfer energy to the $^1\text{G}_4$ level of Pr^{3+} ions, which is then brought into the $^3\text{P}_0$ upper laser level by ESA. Extracted powers in excess of 20 mW at 635 nm and a slope efficiency of 10% were measured. Xie and Gosnell [132] demonstrated room-temperature upconversion fiber laser tunable in the red, orange, green, and blue spectral regions in $\text{Pr}^{3+}/\text{Yb}^{3+}$ codoped ZBLAN fiber with a single Ti:sapphire laser that can be tuned in the 780–880 nm wavelength region. Tunable laser operation over the 635 nm–637 nm, 605 nm–622 nm, 517 nm–540 nm, and 491 nm–493 nm wavelength bands was obtained. At a pump wavelength of 860 nm, 300 mW output at 635 nm, 45 mW at 615 nm, 20 mW at 520 nm, and 4 mW at 493 nm were measured. $\text{Pr}^{3+}/\text{Yb}^{3+}$ -codoped ZBLAN fiber upconversion lasers pumped with a

single semiconductor laser were reported by Baney et al. [170, 171]. Red and orange or green and blue upconversion lasing was observed. Sandrock et al. [134] realized a high-power CW red radiation in a $\text{Pr}^{3+}/\text{Yb}^{3+}$ -codoped ZBLAN fiber by tuning the wavelength of the Ti:sapphire pump laser to 850 nm. A maximum output power of 675 mW was obtained at an incident pump power of 3.37 W. When the fiber was pumped by two Ti:sapphire lasers at 852 nm and 826 nm, the highest output at 635 nm (1020 mW) so far was achieved. Using a 1.6 W Ti:sapphire laser as the pump source, Zellmer et al. [131] obtained more than 165 mW at 491 nm in a $\text{Pr}^{3+}/\text{Yb}^{3+}$ -codoped ZBLAN fiber laser. Due to the high efficiency of $\text{Pr}^{3+}/\text{Yb}^{3+}$ -codoped ZBLAN system, different fiber laser configurations such as actively mode-locked [172], all fiber tunable [133], and all-fiber ring [173] have also been constructed.

Goh et al. [174] demonstrate an alternative approach to achieve blue and red laser action in Pr^{3+} -doped ZBLAN fiber using single-wavelength pump by sensitizing Pr^{3+} with Nd^{3+} ions which enable a strong absorption at 796 nm. Nd^{3+} ions are excited to the $^2\text{D}_{5/2}$ level via upconversion pumping process. Nonradiative decay from the $^2\text{D}_{5/2}$ level populates the lower excited states $^4\text{G}_{11/2}$ and $^4\text{G}_{9/2}$, which are in close resonance with the $^1\text{I}_6$ and $^3\text{P}_1$ levels of Pr^{3+} ions. Energy transfer processes between Nd^{3+} and Pr^{3+} ions result in highly excited Pr^{3+} ions in the $^3\text{P}_0$, $^3\text{P}_1$, and $^1\text{I}_6$ levels. Laser at blue (488 nm) and red (635 nm, 717 nm) has been observed in the $\text{Pr}^{3+}/\text{Nd}^{3+}$ -codoped ZBLAN fiber laser with 796 nm pump.

3.2.4. Upconversion Nd^{3+} -Doped ZBLAN Fiber Lasers. The partial energy-level diagram of Nd^{3+} ions to describe the transitions involving in upconversion emissions is plotted in Figure 11. When pumped at 590 nm, Nd^{3+} ions are excited to the $^4\text{G}_{5/2}$ level via GSA and then decay to the $^4\text{F}_{3/2}$ level, from which Nd^{3+} ions are excited to the upper laser levels by ESA. Radiative transition $^2\text{P}_{3/2} \rightarrow ^4\text{I}_{11/2}$ produces emission at 412 nm and transition $^4\text{D}_{3/2} \rightarrow ^4\text{I}_{11/2}$ produces emission at 381 nm.

In 1994, Funk et al. [136] reported the first ultraviolet room-temperature Nd^{3+} -doped ZBLAN fiber laser. Output power of 74 μW at 381 nm was obtained with pumping with a 275 mW 590 nm dye laser. One year later, a room temperature ZBLAN fiber laser in the violet band was demonstrated with the same experiment setup. Output power of 0.5 mW at 412 nm was produced for 320 mW of pump power at 590 nm [137].

3.2.5. Upconversion Ho^{3+} -Doped ZBLAN Fiber Lasers. The partial energy-level diagram of Ho^{3+} ions to describe the transitions involving in the emission around 550 nm is plotted in Figure 12. Ho^{3+} ions are first excited to the intermediate $^5\text{F}_5$ level via GSA $^5\text{I}_8 \rightarrow ^5\text{F}_5$. The lower levels are then fed due to the fast nonradiative decay. Ho^{3+} ions in the metastable $^5\text{I}_6$ and $^5\text{I}_7$ levels are excited to higher levels via ESA and consequently populated the upper laser level $^5\text{F}_4$. Radiative transition $^5\text{F}_4 \rightarrow ^5\text{I}_8$ produces emission in the blue band.

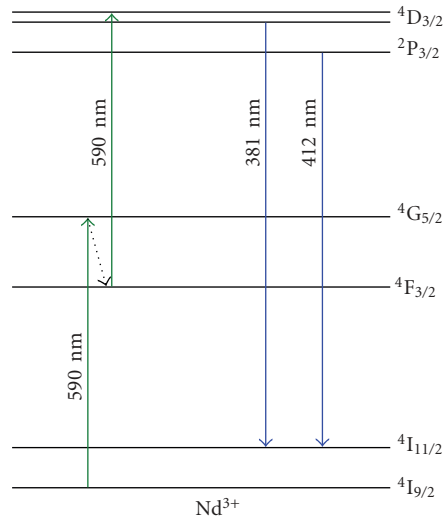


FIGURE 11: Partial energy level diagram of Tm^{3+} ion in ZBLAN related to upconversion laser emissions. Pump absorption transitions are indicated with green upward arrows. Radiative transitions are indicated with blue downward arrows. Nonradiative decays are indicated with dotted arrows.

In 1990, Allain et al. [175] reported a room-temperature CW tunable green upconversion Ho^{3+} -doped ZBLAN fiber laser between 540 nm and 553 nm for the first time. The pump source was a krypton-ion laser at 647.1 nm. Output powers in excess of 10 mW with slope efficiency of 20% were achieved. Diode-pumped Ho^{3+} -doped ZBLAN fiber laser was demonstrated by Funk et al. [176] in 1997 using a 30 mW InGaAlP laser at 643 nm. Excitation spectra, tunability, temporal and spectral characteristics of green Ho^{3+} -doped ZBLAN fiber laser were investigated in detail by Funk et al. [177, 178]. In his continued experiment [135], power conversion efficiency, pump acceptance bandwidth, and excited-state kinetics were studied with a series of experiments in which the Ho^{3+} -doped ZBLAN fiber laser has been characterized with regard to fiber length and core diameter, threshold pump power and slope efficiency, and pump acceptance bandwidth. The highest slope efficiency of 22%, the maximum output power of 40 mW, and an absorbed pump power threshold of less than 1 mW were obtained.

3.3. Supercontinuum Generation Using ZBLAN Fibers. Supercontinuum generation is a physical process leading to dramatic spectral broadening of laser as it propagates through a nonlinear medium. Though the resulted beam has a very broad spectral bandwidth (i.e., low temporal coherence), its spatial coherence usually remains high. This particular property leads to extensive applications including coherence tomography, fluorescence microscopy, flow cytometry, optical devices characterization, dense wavelength division multiplexing in optical fiber communications systems, light-imaging detection and ranging, and optical frequency metrology.

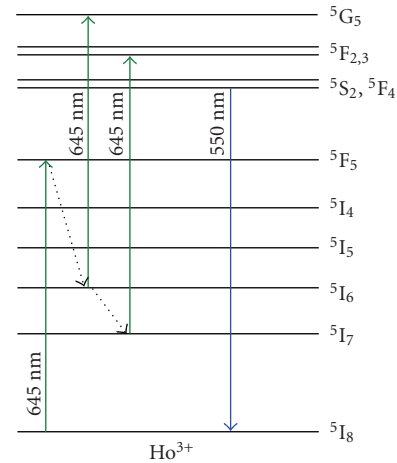


FIGURE 12: Partial energy level diagram of Ho^{3+} ion in ZBLAN related to upconversion laser emission in the blue band. Pump absorption transitions are indicated with green upward arrows. Radiative transition is indicated with blue downward arrow. Nonradiative decays are indicated with dotted arrows.

Since the first demonstration in bulk glass [179, 180], supercontinuum generation has been intensively investigated in a wide variety of nonlinear media including solids, organic and inorganic liquids, gases, and various types of waveguide. Optical fibers were suggested as a promising alternative for supercontinuum generation [181] because they can provide a significant length for nonlinear interaction. Especially, with the emergence of microstructured fibers [182] whose designs can be tailored to provide controllable dispersion and highly confined single-mode core, supercontinuum generation utilizing optical fibers has drawn tremendous interest in the past ten years. However, most of the investigations were carried out on silica fibers and the long wave edge of supercontinuum is limited to $\sim 2 \mu m$, the multiphonon edge of silica glass. As continuum at mid-infrared wavelength region is of much interest, ZBLAN fiber is a preferred media due to its low background loss and wide transparency window.

The first supercontinuum generation in ZBLAN fiber was demonstrated by Hagen et al. [183]. A commercial mode-locked Er^{3+} -doped fiber laser at 1550 nm with an energy of $1.5 \mu J$ per pulse, a duration of 900 femtosecond, and a repetition of 200 kHz was used to pump a 21-cm long standard single-mode silica fiber that was connected by a 91 cm long common ZBLAN fiber with a core diameter 8.5 μm and a NA of 0.21. By cascaded Raman soliton self-shifting, a continuum extending to a wavelength of $\sim 3 \mu m$ was generated. The total average power emitted in the 1.8 μm –3.4 μm range is 5 mW. The average spectral power is $\sim 4 \mu W/nm$ at the range of 1.4 μm –2.9 μm .

Using a similar fiber construction, Xia et al. [184] demonstrated an alternative approach to generate supercontinuum beyond 4.5 μm from a ZBLAN fiber pumped by amplified nanosecond laser diode pulses. The supercontinuum generation is initiated by the breakup of nanosecond

laser diode pulses into femtosecond pulses through modulation instability, and the spectrum is then broadened primarily through self-phase modulation and stimulated Raman scattering in the following ZBLAN fiber. A supercontinuum with a spectrum from $0.8\ \mu\text{m}$ to $4.5\ \mu\text{m}$, an average power of 23 mW, and a spectral power density of $-20\ \text{dBm/nm}$ was achieved. In contrast to femtosecond pulse, nanosecond pulse can be readily amplified by multiple stages of fiber amplifiers. Therefore, using nanosecond pulse to generate supercontinuum is more scalable with recent advances of high-power fiber amplifiers. The average power of the supercontinuum was scaled up to 1.3 watts when a three-stage fiber amplifier was used [185] and up to 10.5 watts when a more powerful power amplifier was used [186, 187]. However, the short wave edge is limited to $0.8\ \mu\text{m}$ and the plug-in efficiency is still low due to the small efficiency of erbium fiber amplifiers at $1.55\ \mu\text{m}$. The two problems may be solved by pumping ZBLAN fiber with $1\ \mu\text{m}$ laser. Recently, when a 2.5-cm-long fluoride fiber was pumped by a 1450 nm femtosecond laser, supercontinuum from $0.35\ \mu\text{m}$ to $3.85\ \mu\text{m}$ was generated [188]. However, all these supercontinua mentioned above are temporally incoherent. A method to generate coherent mid-infrared broadband continuum in nonuniform ZBLAN fiber taper was proposed [189]. It is believed that high efficiency and high-power supercontinuum covering a spectral region from $0.2\ \mu\text{m}$ to $4.5\ \mu\text{m}$ or a longer wavelength will emerge in the near future considering recent progress of high-power $1\ \mu\text{m}$ fiber laser and amplifiers and microstructured ZBLAN fibers [190]. It should be noted that other soft-glass fibers with high nonlinearities [191–193] have also been demonstrated and proposed to generate ultra-broad supercontinuum. Though ZBLAN glass has the nonlinearity even less than that of silica glass, ZBLAN fibers are superior to other soft-glass fibers for high-power supercontinuum generation due to their much lower background loss and relatively higher strength.

4. Discussion and Prospect

With the advent of reliable and high-brightness diode pump lasers and double-clad fibers that facilitate coupling the highly elliptical pump light into the fiber, output power of single-mode fiber laser has been significantly increased over the past decade. Output power of 6 kW has been successfully demonstrated in the Yb^{3+} -doped single-mode silica fiber [5]. Now ZBLAN fiber lasers can provide $>20\ \text{W}$ output at mid-infrared [16]. Evolutions of Yb^{3+} -doped silica and Er^{3+} -doped ZBLAN fiber lasers are illustrated in Figure 13. Although fragility and low transition temperature restrict the extractable output of ZBLAN fiber lasers, there are still some space for power scaling and output power over 100 watts is possible.

Basically, nonlinear phenomena such as stimulated Raman scattering (SRS) and stimulated Brillouin scattering (SBS), thermal effects, and optical damage are major constraints on power scaling of CW fiber lasers. SRS and SBS resulting from stimulated inelastic scattering processes transfer a part of the energy of the incident light to the glass

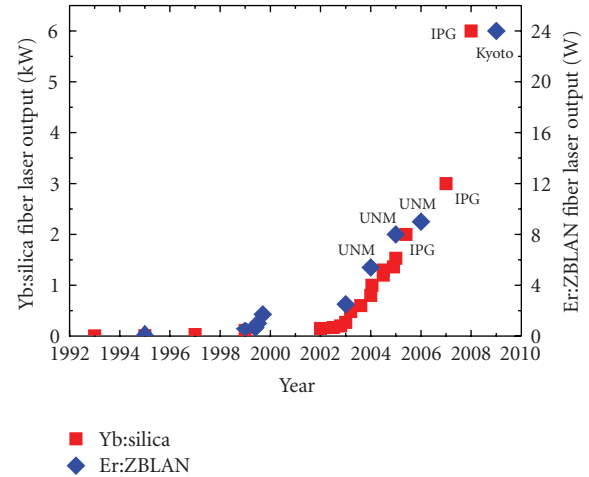


FIGURE 13: Evolution of continuous-wave output of single-mode $1\ \mu\text{m}$ silica fiber laser and $2.7\ \mu\text{m}$ ZBLAN fiber laser over the past decade.

host in the form of excited vibrational modes and red-shift the wavelength of the light. Both processes lead to significant power loss and deteriorate the performance of high-power fiber lasers. Though the chief advantage of fiber lasers over bulk solid-state lasers is their outstanding heat-dissipation capability due to the large ratio of surface area to volume of the fiber allowing effective heat dissipation with minimal heat-sinking requirements, heat generation during a very high-power operation can still destroy an optical fiber, via thermal damage of the coating, fracture, or even melting of the core. Due to the occurrence of self pulsing, optically induced damage is also a hindrance of power scaling of fiber lasers [14]. Generally, for silica fiber lasers, nonlinearity limits the performance of the fiber systems before limits due to thermal effects or fracture of the fiber are reached. For ZBLAN fiber lasers, however, thermal effects and optical damage always happen at a power level much lower than that for the occurrence of nonlinear phenomena due to the low transition temperature ($\sim 260^\circ\text{C}$) and the fragility of ZBLAN glass. Therefore, avoiding optical damage and relieving thermal effects are two routes to break through the power limits of ZBLAN fiber lasers.

Self-pulsing has been frequently observed in high-power fiber lasers [14, 15, 194]. It can be induced by relaxed oscillation, saturated absorption effect, SRS, and SBS. The pulses caused by these instabilities carry sufficient optical energy to cause catastrophic damage to the fiber facets. The damage threshold of rare-earth-doped ZBLAN glass is about $25\ \text{MW/cm}^2$ as measured by using 10-ms long pulses at $2.8\ \mu\text{m}$ [14]. The damage threshold of ZBLAN fiber with respect to the core diameter is plotted in Figure 14. Large-mode-area fiber can effectively increase the extractable power of a ZBLAN fiber laser. It is also found that dual-end pumping configuration can effectively suppress the self-pulsing in ZBLAN fiber lasers [194].

With recent ZBLAN fiber technology, single-mode infrared fiber with a core diameter of tens of microns can be fabricated

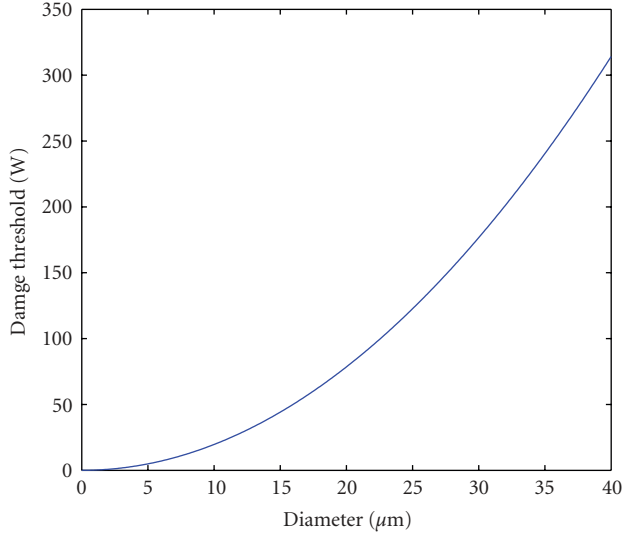


FIGURE 14: Optical damage threshold of a ZBLAN fiber with respect to its core diameter.

and thereby is capable of delivering tens of or even over 100 watts output. However, as the core size increases, thermal effects are enhanced and actively cooling is required for efficient heat dissipation. For a common rare-earth-doped fiber, the temperature distribution in the core and the cladding region can be calculated by the following equations [195]:

$$T_{\text{core}}(r) = T_c + \frac{Q_0 a^2}{4k} \left[2 \ln\left(\frac{b}{a}\right) + \frac{2k}{bh} \right] + \frac{Q_0 a^2}{4k} \left[1 - \left(\frac{r}{a}\right)^2 \right], \quad (1)$$

$$T_{\text{cladding}}(r) = T_c + \frac{Q_0 a^2}{2bh} - \frac{Q_0 a^2}{2k} \ln\left(\frac{r}{b}\right),$$

where T_c is the coolant temperature, Q_0 is the heat power density, k is the thermal conductivity, h is the convective heat transfer coefficient, a and b are the radii of the core and the cladding, respectively. For the actively cooled Er^{3+} -doped ZBLAN fiber laser described in [14] pumped with a 100 W laser diode at 975 nm, the temperature distribution at the pumped fiber facet is plotted in Figure 15. The fiber core center has the highest temperature. Therefore, in order to ensure the stable performance of a fiber laser, effective cooling is required to keep the core center temperature much smaller than the transition temperature of ZBLAN glass ($\sim 260^\circ\text{C}$). Moreover, the spatial variation in temperature can lead to internal stresses within the fiber and result in ultimately catastrophic failure as a fracture when the thermally induced stress exceeds the tensile strength. High efficiency heat dissipation is also necessary to avoid large temperature variation. The distribution of the core center temperature along the fiber is plotted in Figure 16. The pumping end has the highest temperature and the temperature decreases with the reduced absorbed pump to the minimum value at the other end, thereby the pumping end needs more efficient

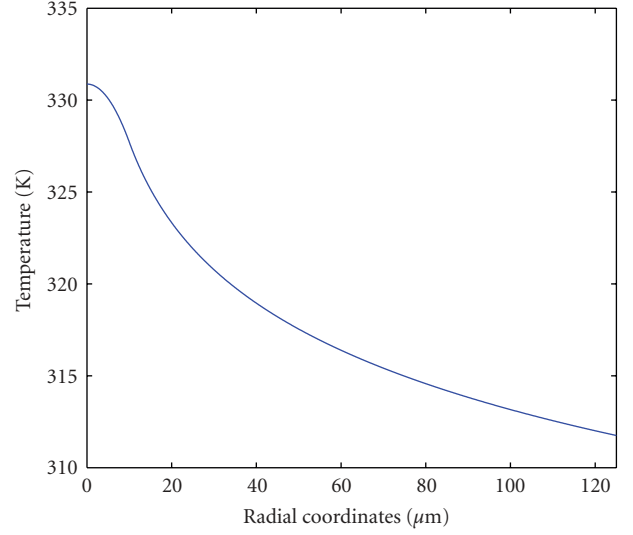


FIGURE 15: Temperature in the pumping end of the Er^{3+} -doped ZBLAN fiber as a function of radial coordinates for a 100 W 975 nm pump.

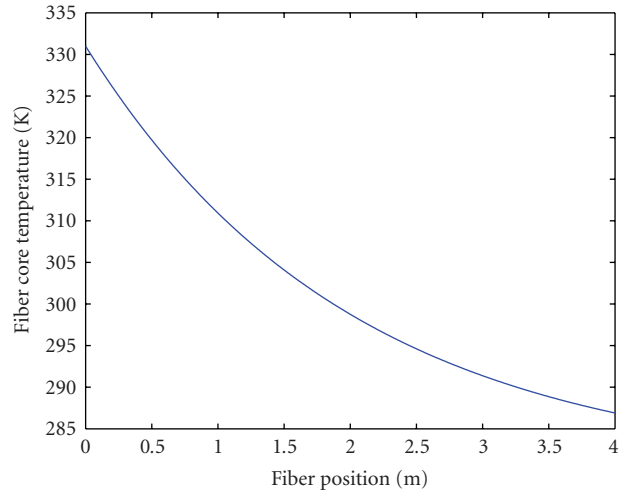


FIGURE 16: Core center temperature in the Er^{3+} -doped ZBLAN fiber as a function of fiber position for a 100 W 975 nm pump.

heat dissipation. The core center temperature of the pumping end as a function of the heat transfer coefficient is shown in Figure 17. When the pumping end is not actively cooled ($h < 100$), the core temperature largely exceeds the melt point of ZBLAN glass and consequently the fiber will be readily melt down. Therefore, sufficient cooling is crucial for high-power ZBLAN fiber lasers. For instance, in the 24 W diode-pumped Er^{3+} -doped ZBLAN fiber laser, a liquid-cooling system was utilized to efficiently remove the heat [16].

Aside from optical damage and thermal effects, photodarkening, a phenomenon with significant increase of the core absorption at visible and near-infrared wavelengths when the fiber is pumped with high-power infrared light, also need to be taken into account for high-power upconversion ZBLAN fiber lasers. In 1995, Barber [196] observed

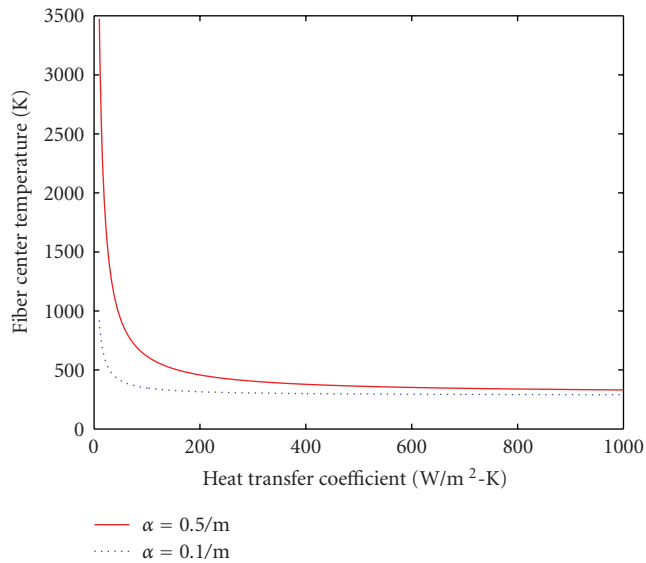


FIGURE 17: Core center temperature in the Er^{3+} -doped ZBLAN fiber as a function of heat transfer coefficient for a 100 W 975 nm pump (α is the absorption coefficient).

a phenomenon of infrared-induced photodarkening in Tm^{3+} -doped fluoride fibers for the first time. The loss induced in the visible region by 1140 nm radiation is very strong (as high as 25 dB in a 50 cm piece) and broadband. Laperle et al. [149] found that the darkening rate is shown to follow a fourth-power dependence on the pumping intensity and a strong dependence on the Tm^{3+} ion concentration. The photodegradation of upconversion Tm^{3+} -doped ZBLAN fiber lasers was also observed [150]. After upconversion lasing at 482 nm, the fiber absorbs strongly at wavelength below 650 nm affecting further upconversion lasing. The physical details of photodarkening are very complicated and have so far not been fully understood. The formation of color centers or other microscopic structural transformations in the fiber core would be the main cause. Since photodarkening can lead to serious performance degradations of upconversion fiber lasers, it would be particularly important to fully understand the mechanisms and minimize the effect to a large extent by optimizing the compositions of ZBLAN glass and the laser operation conditions.

On the other hand, a fiber laser can produce more output power with less heat generation when the efficiency of optical conversion is increased. Using pump with a wavelength close to the desire laser wavelength can increase the conversion efficiency and reduce the heat generation accordingly. Upconversion energy transfer and cross-relaxation processes between ions in high-concentration fibers can be used for efficiency improvement due to the doubled quantum efficiency. Cascade lasing is also an effective method to enhance the desired laser emission and reduce the heat generation by replacing the nonradiative multiphonon process with radiative emission. For upconversion fiber lasers, proper pumping scheme can also increase the efficiency and effectively avoid nonradiative transitions.

It should be mentioned here that Bragg gratings can be directly inscribed on the ZBLAN fibers with femtosecond pulses now [197, 198]. This enables the construction of a ZBLAN fiber laser in a monolithic all-fiber configuration [51, 102]. Narrow line-width mid-IR sources which can be used for spectroscopic sensing can also benefit from the convenient writing of ZBLAN fiber Bragg grating.

Following the development of high-power silica fiber lasers, performance of ZBLAN fiber lasers can be further improved and 100-watt-level output is possible with large-core ZBLAN fibers or beam combining techniques [199] in the near future. Due to the increasing demand of laser sources at wavelengths where silica fiber lasers cannot operate, high-power ZBLAN fiber lasers covering from ultraviolet to mid-infrared are promising for further development in the next decade.

5. Summary

Fluoride glass properties and ZBLAN fiber fabrication are introduced briefly in this paper to give readers a general knowledge of this remarkable gain medium. Detailed progress of infrared, upconversion UV and visible, and supercontinuum ZBLAN fiber lasers were reviewed. Optical damage, thermal effects, and photodarkening are the three major constraints on power scaling of ZBLAN fiber lasers. Using novel fiber design and high-brightness diode pumps, output power of >20 W has recently been obtained by sufficiently cooling the ZBLAN fiber laser system. In the next decade, a variety of high-power ZBLAN fiber lasers covering from ultraviolet to mid-infrared will emerge.

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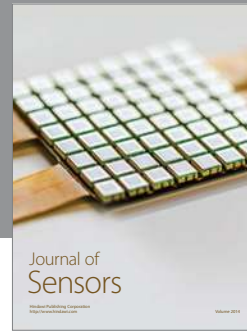
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