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In Situ Growth and UV Photocatalytic Effect of ZnO Nanostructures on a Zn Plate Immersed in Methylene Blue

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Abstract: Nanostructures of zinc oxide (ZnO) are considered promising photocatalysts for the degradation of organic pollutants in water. This work discusses an in situ growth and UV photocatalytic effect of ZnO nanostructures on a Zn plate immersed in methylene blue (MB) at room temperature. First, the Zn surfaces were pretreated via sandblasting to introduce a micro-scale roughness. Then, the Zn plates were immersed in MB and exposed to UV light, to observe ZnO nanostructure growth and photocatalytic degradation of MB. Scanning electron microscopy, energy dispersive spectroscopy, X-ray diffraction, X-ray photoelectron spectroscopy, and UV-Vis diffuse reflectance spectroscopy were used to characterize the Zn surfaces. We observed the growth of stoichiometric and crystalline ZnO with a nano-leaf morphology and an estimated bandgap of 3.08 eV. The photocatalytic degradation of MB was also observed in the presence of the ZnO nanostructures and UV light. The average percentage degradation was 76% in 4 h, and the degradation rate constant was 0.3535 h⁻¹. The experimental results suggest that room temperature growth of ZnO nanostructures (on Zn surfaces) in organic dye solutions is possible. Furthermore, the nanostructured surface can be used simultaneously for the photocatalytic degradation of the organic dye.

Keywords: ZnO; nanostructure; photocatalyst; water treatment

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1. Introduction

Semiconductor photocatalysis and its application in water pollutant degradation have been the subject of extensive research over the past two decades [1,2]. It involves the absorption of light by a semiconductor, the generation of electron-hole pairs, and subsequent redox reactions that produce free radicals also known as reactive oxygen species (ROS) [3–5]. ROS are utilized to mineralize water contaminants; hence the process aids the complete degradation of contaminants rather than just physical separation [6]. Nano-sized metal oxides have been widely studied for this application [7,8]. TiO₂ is the most popular metal oxide photocatalyst owing to its optoelectronic properties, stability, and relative non-toxicity [9–11]. Another metal oxide that is comparable to TiO₂ is ZnO [5,12]. ZnO is an n-type wide bandgap semiconductor (3.37 eV) with a high exciton binding energy at room temperature, and high electron mobility. It is also non-toxic, low-cost, and abundant [5].

Researchers have used various methods to synthesize ZnO nanostructures [13]. These methods can be broadly categorized into solution-based and vapor phase methods. Some examples of solution-based methods are sol-gel [14], hydrothermal [15,16], electrochemical deposition [17,18], and coprecipitation [19]. In these methods, the morphology and size of the nanostructures can be controlled by various experimental parameters, types of solvents, and reaction conditions [12,17]. Solution-based methods are simple and less energy consuming than the vapor phase methods, such as physical vapor deposition, chemical vapor deposition, thermal evaporation, and pulsed laser deposition [12]. However,

both categories involve precursor chemicals, high temperatures, and/or vacuum systems, which can be costly, energy consuming, and toxic to the environment.

In recent years, there has been much interest in environmentally benign nanostructure synthesis methods [20]. Especially in nano-photocatalysis, where the intended application is environmental remediation, it is desired that their synthesis approaches do not involve the use of toxic chemicals [21]. Additionally, ZnO nano-photocatalyst synthesis methods need to be cost-effective to promote their wide-spread use in water treatment. This means the synthesis methods should be simple and not require expensive equipment [21]. One such green nanostructure approach is the hot water treatment (HWT) method which simply involves the immersion of metal surfaces in hot deionized (DI) water [22]. In a previous work, we reported the synthesis of ZnO nanostructures by HWT of Zn and its use for the photocatalytic degradation of methylene blue (MB) [23]. One of the significant advantages of this method is that it does not require any chemical additives [22]. In addition, the temperature used in HWT (75–95 °C) is much lesser than the conventional nanostructure synthesis methods [22]. Another precursor-free and water-based ZnO nanostructure synthesis method is the submerged photosynthesis of crystallites (SPSC) reported by Jeem et al. [24] and Zhang et al. [25]. They showed that ZnO nanostructures could be synthesized just by the immersion of Zn plates in room temperature DI water and exposure to UV light. They used a submerged liquid plasma process as a surface pretreatment to introduce ZnO seed layers, which initiated the nanostructure growth process [24,25]. The growth process involved hydrothermal as well as photoinduced reactions [24,25].

This work is inspired by the SPSC process and investigates if ZnO nanostructure growth can occur in situ in the organic dye solution, which one wants to degrade via photocatalysis. In other words, we designed a single-step approach where the photocatalyst synthesis and its application happen simultaneously. For this, we immersed a Zn plate in an aqueous MB solution at room temperature and exposed it to UV light. We used a simple and low-cost sandblasting process as a surface pretreatment, which introduced micro-scale roughness on our Zn surfaces [26]. Our results show that leaf-shaped ZnO nanostructures are formed when Zn surfaces are immersed in MB. We did not observe a significant difference in the nanostructure growth with or without UV light. Nevertheless, we observed the degradation of MB due to the photocatalytic activity of the nanostructures. The main advantages of this nano-photocatalyst synthesis method are that it does not require high temperatures or any chemical additives, and it is simple, low-cost, and environmentally friendly. Moreover, the ZnO nanostructures in our photocatalyst design are attached to the Zn surface. Researchers have given much emphasis to the immobilization of photocatalysts to avoid agglomeration of the nanostructures, secondary pollution, and nanostructure reclamation from the water, which could be expensive [27,28]. Our method achieves photocatalyst immobilization in a single step. The materials characterization, possible growth mechanisms, and photocatalytic performance of the ZnO nanostructures are discussed.

2. Results and Discussion

2.1. Materials Characterization

Scanning electron microscopy (SEM) was used to investigate the Zn plate surface before and after its immersion in MB solution. The micro-scale roughness introduced on the Zn surface by sandblasting can be observed in Figure 1a. Figure 1b,c show the formation of leaf-shaped ZnO nanostructures on the Zn surface after it was immersed in MB for 4 h, with and without UV irradiation, respectively. In both samples, the nanostructures had approximate lengths ranging from 300 to 600 nm and widths ranging from 150 to 250 nm. Low magnification images of both samples show some irregular structures, which could be amorphous Zn(OH)₂. Overall, the sample surfaces are uniformly covered with leaf-shaped nanostructures. Figure 1d,e show the ZnO nanostructures grown on Zn surfaces immersed in pure DI water, with and without UV light, respectively.

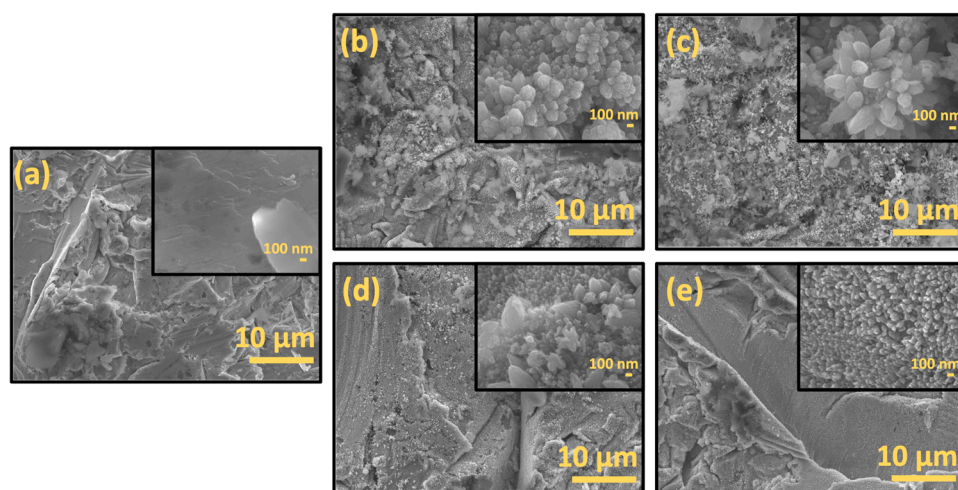


Figure 1. SEM images of (a) Zn surface after sandblasting, ZnO nanostructures on Zn surface grown (b) in MB with UV light, (c) in MB without UV light, (d) in DI water with UV light, and (e) in DI water without UV light. The insets show higher magnification images of the same.

The morphology of the nanostructures grown in DI water is the same as in MB. However, the density of nanostructure growth is less in DI water, both in the presence and absence of UV light. Most of the DI water nanostructures had lengths ranging from 100 to 200 nm and widths ranging from 50 to 200 nm, but some structures with sizes similar to those grown in MB were also observed. The comparison of ZnO nanostructure growth in the two mediums (MB and DI water) and the presence and absence of UV light are further discussed in Section 2.2.

Energy dispersive X-ray spectroscopy (EDS) was used to analyze the chemical composition of the Zn surfaces. EDS mapping of the ZnO nanostructures grown in MB with UV light showed an even distribution of Zn and O (Figure 2a–d), indicating the growth of stoichiometric ZnO. The higher atomic percentage of Zn is most likely due to the underlying Zn plate (Figure 2e). The compositional analysis of the irregularly shaped structures showed a higher O ratio, possibly due to Zn(OH)₂ (Figure S1). The EDS results of Zn surfaces immersed in MB in the dark, and DI water with and without UV light showed a similar chemical composition (Figures S2–S4).

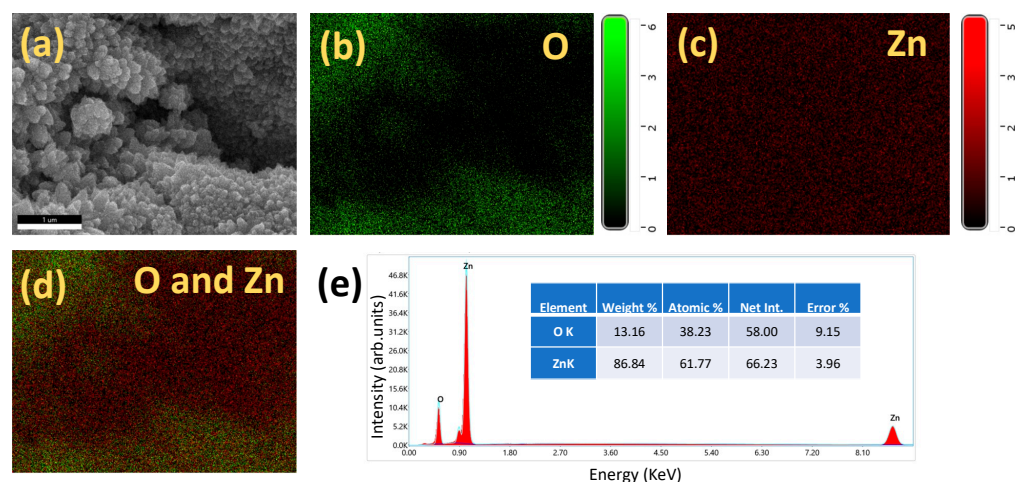


Figure 2. (a) SEM image and (b–d) corresponding EDS maps of ZnO nanostructures grown on Zn surface in MB and with UV light, and (e) elemental composition of the same.

X-ray diffraction (XRD) was used to analyze the crystal structure of the ZnO nanostructures. Figure 3a shows the XRD profile of the sandblasted Zn surface before its immersion

into MB. The peaks centered at 36.2° , 38.9° , 43.1° , and 54.2° could be attributed to (002), (100), (101), and (102) crystal planes of Zn (crystallography open database entry 9008522). After 4 h of immersion in MB, we observed the appearance of a peak at 34.4° , which can be attributed to the (002) crystal plane of hexagonal wurtzite ZnO (crystallography open database entry 1011258). No other peaks were observed; hence, we could not confirm the presence of $\text{Zn}(\text{OH})_2$ from XRD.

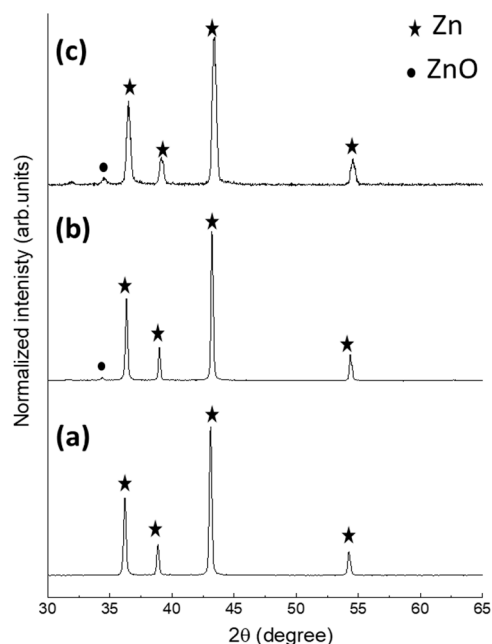


Figure 3. XRD profiles of Zn surface (a) before immersion in MB, (b) after immersion in MB without UV light, and (c) with UV light.

X-ray photoelectron spectroscopy (XPS) was used to analyze the surface chemical composition of the samples. Figure 4a shows the XPS survey spectrum of the Zn surface immersed in MB and exposed to UV light for 4 h. Figure 4b shows the Zn 2p XPS spectrum. Since the Zn 2p_{3/2} peaks of elemental Zn and ZnO are very close, it is difficult to distinguish between elemental Zn and ZnO on the sample. Hence, Auger Zn LMM spectrum of the sample was obtained (Figure 4c) [29]. The Auger peak could be fitted to two peaks centered at kinetic energies of 987.6 eV and 990.4 eV [29]. The former is attributed to ZnO and the latter to elemental Zn. Thus, XPS also confirmed the formation of ZnO nanostructures on the surface of the Zn plate. The O 1s spectrum (Figure 4d) of the sample could be fitted to two peaks, one centered at 530.29 eV and the other at 532.11 eV. The lower binding energy peak is attributed to O^{2-} in the stoichiometric wurtzite ZnO lattice, whereas the higher binding energy peak can be attributed to OH or H_2O groups attached to Zn [30]. Hence, the XPS analysis also indicates the presence of $\text{Zn}(\text{OH})_2$ on the sample, which could be the irregularly shaped white layers observed in the SEM images [25,29].

The optical bandgap of the ZnO nanostructures was estimated using UV-Vis diffuse reflectance spectroscopy, Kubelka-Munk function ($F(R)$), and Tauc plot [31,32]. The diffuse reflectance measurement of the ZnO nanostructures synthesized by the immersion of Zn plates in MB and exposed to UV light for 4 h is shown in Figure 5a. $F(R)$ is given by Equation (1), where α is the absorption coefficient, s is the scattering coefficient, and R is the diffuse reflectance. Here, we assume that s does not depend on wavelength and is, therefore, a constant. The absorption coefficient is related to the bandgap according to the Tauc equation (Equation (2)). Here, h is Planck's constant, ν is the frequency, A is a constant, E_g is the bandgap energy, and n is $\frac{1}{2}$ for direct bandgap. Hence, the Tauc equation could be rewritten as Equation (3). Figure 5b shows the graph plotted according to Equation (3). The optical bandgap of the ZnO nanostructures was estimated by extrapolating the linear

portion of the plot to the x-axis, where $(F(R).h\nu)^2$ is zero. The value of the bandgap obtained from the plot was 3.08 eV, which is lower compared to the bandgap of bulk ZnO (3.37 eV). Further investigation needs to be done to determine the reason for the lower bandgap; however, it could be due to defects in the ZnO nanostructures [33,34].

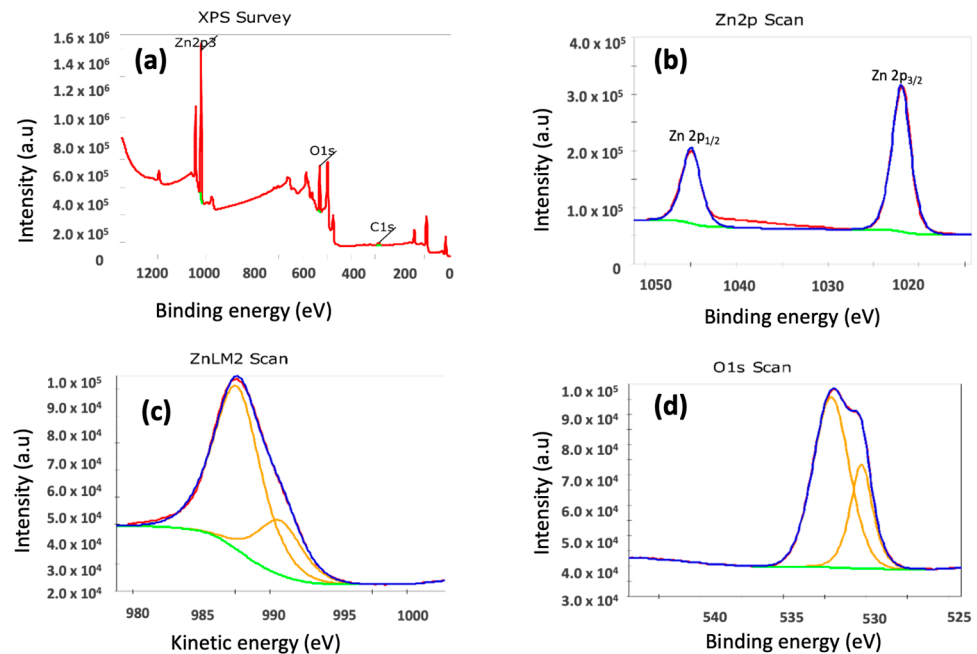


Figure 4. XPS analysis of ZnO nanostructures grown on Zn surface (a) the survey spectrum, (b) Zn 2p peaks, (c) Auger spectra with distinct peaks corresponding to Zn and ZnO, and (d) O 1s spectrum.

$$F(R) = \frac{\alpha}{s} = \frac{(1-R)^2}{2R} \quad (1)$$

$$(\alpha h\nu)^{1/n} = A(h\nu - E_g) \quad (2)$$

$$(F(R).h\nu)^2 = A(h\nu - E_g) \quad (3)$$

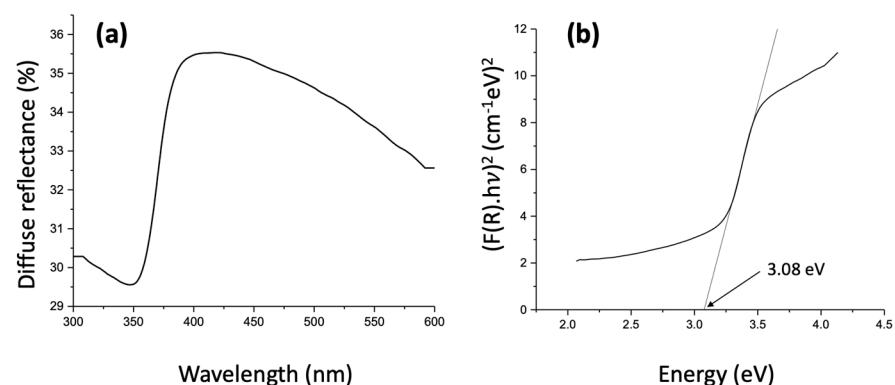
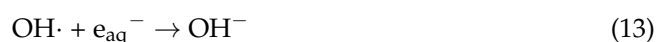
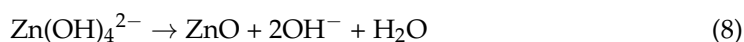
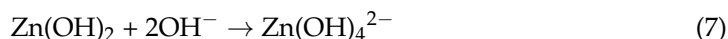


Figure 5. (a) Diffuse reflectance spectrum and (b) $(F(R).h\nu)^2$ vs. energy ($h\nu$) plot of ZnO nanostructures grown on Zn plate by immersion in MB and exposure to UV light for 4 h.

2.2. Nanostructure Growth Mechanism

The chemical reactions involved in the room-temperature growth of ZnO nanostructures on a Zn plate are explained by Jeem et al. [24] and Zhang et al. [25]. In their work, both hydrothermal (Equations (4)–(8)) and photoinduced reactions (Equations (9)–(13))

were involved in the synthesis of ZnO nanostructures. The latter produced $\text{OH}\cdot$, which converted into OH^- , which is important for the formation of ZnO [25].



In our work, the SEM images show that ZnO nanostructure growth in MB is very similar in the presence and absence of UV light. Based on this, we assume that hydrothermal reactions are dominant during ZnO nanostructure growth in our study. Reactions (1) and (2) are the anodic (oxidation) and cathodic (reduction) reactions, respectively, involved in the corrosion of Zn metal in water [25]. We believe that the micro-rough surface created by sandblasting acts as a corrosion micro-cell, as described by Zhang et al. [25]. In the experiments involving UV light, we observe the degradation of MB. This indicates that light does produce hydroxyl radicals. However, they do not seem to participate in the growth of ZnO, at least in this experiment duration (4 h). Longer treatment times may yield different results [24,25], which will be explored in future studies.

We also want to address two other observations from our experiments. The first one is the presence of $\text{Zn}(\text{OH})_2$ on our sample surfaces as indicated by EDS and XPS analysis. According to the chemical reactions listed above, $\text{Zn}(\text{OH})_2$ is formed as an intermediate during the formation of ZnO. The conversion of $\text{Zn}(\text{OH})_2$ to ZnO requires an alkaline environment. Hence, $\text{Zn}(\text{OH})_2$ could persist on the sample surface if enough OH^- were not available for it to form ZnO [25,35]. Another observation is the enhancement in nanostructure growth in MB compared to in DI water. Wang et al. have reported that the presence of organic dyes can affect the growth direction of ZnO nanostructures during hydrothermal synthesis [36]. Their results show that the presence of anionic dyes during the growth of ZnO nanowires led to a reduction in the growth of ZnO nanowires along the c-axis. This was due to the electrostatic attraction between the dye anions and the Zn^{2+} ions [36]. We hypothesize that the presence of dissolved MB cations in the vicinity of the Zn surface could be responsible for the electrostatic attraction of more OH^- ions. This could increase the interactions between Zn^{2+} and OH^- and thus enhance the growth of ZnO.

2.3. Photocatalytic Degradation of Methylene Blue

Figure 6 shows the normalized concentration (C/C_0) of MB vs. time plot during different degradation experiments. The curves show average C/C_0 values from three experiments, and their standard deviations are shown in the error bars. Please see Figures S5–S7 for the UV-Vis profiles from all experiments. There was no change in MB concentration before and after the 30 min stirring in the dark. Two control experiments were performed, one where only MB was exposed to UV light and the other where the Zn plate was immersed in MB but not exposed to UV light. MB degradation was negligible in both scenarios. These results show that the degradation observed when Zn plates were immersed in MB and exposed to UV light is due to the photocatalytic effect of the ZnO nanostructures on the surface of the Zn plate. The average percentage degradation of MB is 76% in 4 h. According

to the Langmuir-Hinshelwood mechanism, photocatalytic degradation can be modeled as pseudo-first order reaction described by the equation $\ln(C_0/C) = kt$, where k is the degradation rate constant. Hence, $\ln(C_0/C)$ was plotted against time and linearly fitted to find the degradation rate constant from the slope, which is 0.3535 h^{-1} .

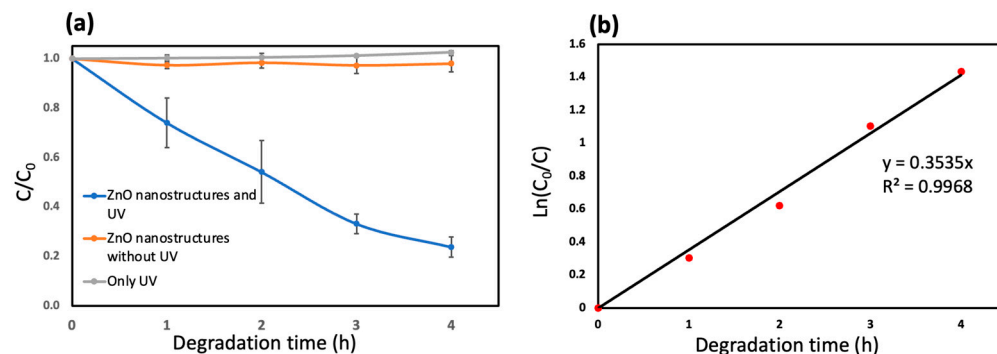


Figure 6. (a) Normalized degradation of MB (C/C_0) vs. time plot and (b) $\ln(C_0/C)$ vs. time plot of the degradation of MB with ZnO nanostructures and UV light.

Florica et al. [37] reported a similar study in which ZnO nanowires were synthesized by a precursor-free thermal oxidation method on Zn foils and were used for the photocatalytic degradation of MB. They reported a percentage degradation of 57% in 320 min and a degradation rate constant of 0.1542 h^{-1} . However, it is worth mentioning that comparing photocatalyst performances based only on degradation percentage and rate constant is not the best practice. Various other factors affect a photocatalyst's performance, such as its morphology and crystal structure, the wavelength and intensity of the light source, and the type, concentration, pH, and volume of the organic dye used in the experiments [17,38–41]. Table 1 shows some works that reported the synthesis of ZnO nanostructures that are immobilized on various surfaces. The photocatalytic degradation of MB and some other organic dyes are also shown. We have limited this table to pure ZnO nanostructures and excluded reports that modified ZnO nanostructures by doping or forming heterojunctions with other materials.

Here, we acknowledge that various methods can control the morphology [42], crystal structure [17,43,44], bandgap [45–47], and defects [48,49] of ZnO nanostructures by altering the reaction conditions, thereby fine-tune its photocatalytic performance. The main distinction of this work is that ZnO nanostructures are synthesized on a micro-rough Zn plate without using any chemical additives, precursors, or high temperatures. Moreover, photocatalyst synthesis and its application for photocatalytic dye degradation co-occur, which makes this process extremely simple and less energy consuming. This result is of significant importance in the current scenario of increased interest in cost-effective, energy-efficient, and environmentally friendly nanostructure synthesis methods for photocatalytic applications.

One of the limitations of this work is that the ZnO nanostructure growth mechanism is not fully understood. In a future study, we will investigate how the ZnO nanostructure growth is affected by the time of immersion of the Zn plates in MB. It is important to understand if the nanostructures continue to grow during longer immersion times, which may alter their photocatalytic performance. We will also investigate the effect of concentration, pH, and type of organic dye on the photocatalytic activity, as well as the photocatalytic degradation mechanism. We also plan to test the durability of the nanostructures against photocorrosion. Understanding the growth mechanism and durability determines the long-term reusability of the ZnO nanostructures which is crucial for photocatalytic water treatment applications.

Table 1. Different methods to synthesize ZnO nanostructures immobilized on different surfaces and their photocatalytic degradation performances.

Method	ZnO Synthesis			Organic Dye Solution (Type, Volume, Concentration)	Light Source (Wavelength and Power)	Best Photocatalytic Performance or Rate Constant	Ref.
	Nanostructures immobilized on	Materials used	Reaction parameters				
Hydrothermal	Si and glass	Zinc acetate, zinc nitrate, hexamethylenetetramine	60–150 °C, 13 h (approx.)	Methylene Blue, 12 mL, 10 mM	Fluorescent lamp, 375 nm, 18 Watts	45% in 6 h, 0.0021 min ⁻¹	[50]
Hydrothermal	PDMS/PET	Zinc acetate, zinc nitrate, hexamethylenetetramine	60–90 °C, 10 h (approx.)	Methylene Blue, 40 mL, 10 mM	Fluorescent lamp, 375 nm, 18 Watts	50% degradation in 6 h, 0.0018 min ⁻¹	[51]
Thermal oxidation	Zn	Nil	500 °C, 12 h	Methylene Blue, 35 × 10 ⁻⁶ M	UV lamp, 312 nm, 15 Watts	57% degradation in 320 min, 0.1542 h ⁻¹	[37]
Spray pyrolysis	FTO coated glass	Zinc acetate, methanol	425 °C	Methylene Blue, volume not reported, 1 mM	UV light, wavelength and power not reported	98.1% degradation in 120 min, rate constant not reported	[52]
Anodization	Zn	Potassium bicarbonate, orthophosphoric acid, hydrochloric acid, oxalic acid, and sodium hydroxide	10–40 V DC for anodization, pre and post treatment annealing at 300 °C.	Phenol, 50 mL, 5 ppm	UV lamp, 254 nm, 6 Watts	82% degradation in 4 h, 0.44 h ⁻¹	[17]
Anodization and O ₂ plasma annealing	Zn	Potassium bicarbonate	1 V DC, pretreatment annealing at 300 °C and post treatment thermal and O ₂ plasma annealing (200–350 °C)	Phenol, 50–80 mL, 5–10 ppm	UV lamp, 254 nm, 6 Watts	70% degradation in 4 h, 0.31 h ⁻¹	[35]
Hydrothermal	Zn	Zinc nitrate, hexamethylenetetramine, methylamine, potassium hydroxide	120–180 °C, 2–12 h	Rhodamine B, 10 mL, 1 × 10 ⁻⁵ M	UV lamp, wavelength and power not reported	80% degradation of rhodamine B in 3 h, rate constant not reported	[15]

3. Materials and Methods

3.1. ZnO Nanostructure Synthesis

Zn plates (Purchased from [amazon.com](https://www.amazon.com), USA, manufacturer name: LTKJ Ltd.) of 99.9% purity and 0.3 mm thickness were cut into pieces of dimension 25 mm × 25 mm. Sandpaper of size p3000 was used to clean any surface contamination on the plates. Then, the Zn plates were sandblasted using Al₂O₃ abrasives of size 120 grit at a pressure of 80 psi to introduce micro-scale roughness on the plate surface. After sandblasting, they were cleaned by ultrasonication in acetone, isopropyl alcohol, and DI water for 10 min each. After this, the Zn plates were immersed in an aqueous solution of methylene blue (SPI supplies, West Chester, PA, USA) (50 mL) at a concentration of 2.5 ppm (mg/L) contained in a glass beaker. The solution was stirred for 30 min in the dark using a magnetic stirring bar to establish adsorption–desorption equilibrium. Then, the Zn plates and MB were

irradiated with 254 nm UV light from an 8-Watt UV lamp fixed at a vertical distance of 15 cm, for 4 h. Three control experiments were conducted to investigate the effects of UV light and MB on the ZnO nanostructure growth: (1) Zn plates were immersed in MB and left in the dark for 4 h, (2) Zn plates were immersed in pure DI water and exposed to UV light, and (3) Zn plates were immersed in pure DI water and left in the dark for 4 h. In all experiments, MB/DI water was continuously stirred throughout the experiment and a cold-water bath was used to prevent the increase in water temperature in the beaker.

3.2. Materials Characterization

SEM (JOEL JSM-7000F, Tokyo, Japan) along with EDS (EDAX Element, Pleasanton, CA, USA) were used to analyze the morphology and map the chemical composition of the Zn surfaces. XRD (Rigaku Miniflex 600, Tokyo, Japan) was used to analyze the crystal structure, XPS (Thermo Scientific K-Alpha, Waltham, MA, USA) was used to investigate the surface chemical states of the samples. The sample surfaces were etched for 10 s, to clean the surfaces before XPS analysis. UV-Vis diffuse reflectance spectroscopy (Shimadzu UV-3600, Kyoto, Japan) was used to estimate the bandgap of the ZnO nanostructures.

3.3. In Situ Photocatalytic Degradation Experiment

The concentration of MB at regular intervals of time was measured using UV-Vis absorbance spectroscopy (Agilent Carry 60, Santa Clara, CA, USA). A total of 3 mL of MB was retrieved from the as-prepared solution, after 30 min of stirring in dark, and every hour after the UV light was switched on. Beer-Lambert law was used to estimate the concentration of MB from the UV-Vis peak (at 664 nm) absorbance values. If C_0 represents the initial concentration of MB and C , the concentration at a particular time, the percentage degradation of MB was calculated using $((C - C_0)/C_0) \times 100$ [53].

4. Conclusions

Zn plates were immersed in an aqueous solution of MB, and an in situ ZnO nanostructure growth and UV photocatalytic effect were investigated. Leaf-shaped ZnO nanostructures were formed on the surface of the Zn plate both in the presence and absence of UV light. In the presence of UV, the ZnO nanostructures acted as a photocatalyst and degraded MB. The average MB degradation was 76% in 4 h. The results show that nanostructured ZnO photocatalysts could be synthesized simply by the immersion of Zn plates in MB solutions, without any chemical additives or heating. Hence, this is a low-cost, low-energy, and environmentally benign method to synthesize ZnO nanostructures for photocatalytic applications.

Supplementary Materials: The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/catal12121657/s1>, Figure S1: (a) SEM image and (b–d) corresponding EDS maps of irregularly shaped ZnO nanostructures grown on Zn surface in MB and with UV light, and (e) elemental composition of the same. Figure S2: (a) SEM image and (b–d) corresponding EDS maps of ZnO nanostructures grown on Zn surface in MB and without UV light, and (e) elemental composition of the same. Figure S3: (a) SEM image and (b–d) corresponding EDS maps of ZnO nanostructures grown on Zn surface in DI water and with UV light, and (e) elemental composition of the same. Figure S4: (a) SEM image and (b–d) corresponding EDS maps of ZnO nanostructures grown on Zn surface in DI water and without UV light, and (e) elemental composition of the same. Figure S5: Time dependent UV-Vis spectra of MB with ZnO nanostructures and UV light. Figure S6: Time dependent UV-Vis spectra of MB with ZnO nanostructures without UV light. Figure S7: Time dependent UV-Vis spectra of MB with UV light only.

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Data Availability Statement: All data are contained in the article and the supplementary material.

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