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# Investigation of the properties of alkali-activated slag mixes involving the use of nanoclay and nucleation seeds for 3D printing

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## Abstract:

This study investigated the properties of alkali activated slag (AAS) binders formulated for extrusion-based 3D printing. The fresh properties of AAS mixes were tailored through the use of nanoclay (NC) and nucleation seeds. The printability criteria employed were the ease of extrusion (extrudability) and the stability of the layered structure (buildability). Introduction of 0.4% NC in AAS mixes led to improved thixotropic properties due to the flocculation effect, which accounted for the extrudability and shape fidelity of the binder. Inclusion of 2% hydromagnesite seeds in this mix design provided additional nucleation sites for the increased precipitation of hydrate phases, resulting in denser microstructures. This enhanced the hydration reaction and improved the structural build-up rate necessary for large-scale 3D printing. The developed AAS mix containing 0.4% NC and 2% hydromagnesite seeds was used in the printing of an actual 3D structure to demonstrate its feasibility to be used in 3D printing applications.

**Keywords:** Alkali-activated slag; 3D printing; rheology; hydration; strength

## 1 **1. Introduction**

2

3 Extrusion-based additive manufacturing, also known as 3D printing, enables the mould-free  
4 fabrication of complex customized parts, which cannot be easily processed by other conventional  
5 manufacturing methods [1-4]. This technology has been successfully applied in aerospace,  
6 automotive and biomedical fields, while it is still being researched in the construction sector. The  
7 expected benefits of 3D concrete printing are higher productivity, shorter construction periods,  
8 higher geometrical freedom and more efficient use of natural resources [5]. The use of 3D printing  
9 can also present advantages in terms of reduced costs in the case of complex structures, in  
10 comparison to the conventional construction methods.

11

12 A pivotal approach in the success of 3D printing in construction is the use of Building Information  
13 Modelling (BIM). Since BIM already serves as a rich source of geometric information for existing  
14 structures, on site 3D concrete printing will eventually need scheduling and assembly sequence  
15 information to maintain safety and productivity [6, 7]. Previous studies [8] that shared this vision  
16 proposed a shift to a digital construction organization by combining existing technologies such as  
17 rapid digital mapping, BIM, digital collaboration, internet of things and design to construction. To  
18 further digitalise the construction industry, digital twin technology can be used to continuously  
19 monitor progress against the schedule laid out in the BIM model. Furthermore, approaches such  
20 as the digital twin technology can be combined with 3D concrete printing to reduce the volume of  
21 trial and error testing, reduce defects and shorten the time between the design and production  
22 processes [9].

23

24 While the use of cement-based materials in 3D printing presents several advantages, certain  
25 challenges need to be resolved for a successful printing process. To be used for printing  
26 applications, the rheology of cement-based mixes must meet certain requirements. Accordingly,  
27 during extrusion, the material must be fluid to prevent any blocking, bleeding or segregation.  
28 However, once they are printed, each layer must be able to harden quickly to support the  
29 superposed layers [10]. Another parameter to be considered is the time gap between printed layers.  
30 Long time gaps can provide an adequate mechanical strength for supporting the weight of the  
31 subsequently deposited layers, while short time gaps ensure optimized bonding strength.

32 Therefore, for successful printing, a narrow process window in terms of material yield stress exists,  
33 as shown in Fig. 1, where a schematic of extrusion-based additive manufacturing of cement-based  
34 mixes is demonstrated. Accordingly, the rheology of the developed mixes needs to be adjusted to  
35 achieve a minimum yield stress for smooth extrudability. After extrusion, the yield stress must  
36 evolve faster than the stress acting on the bottom layer to avoid strength-based failures.  
37 Alternatively, materials with a high yield stress can result in poor interlayer bonds, despite their  
38 ability to support additional layers.

39

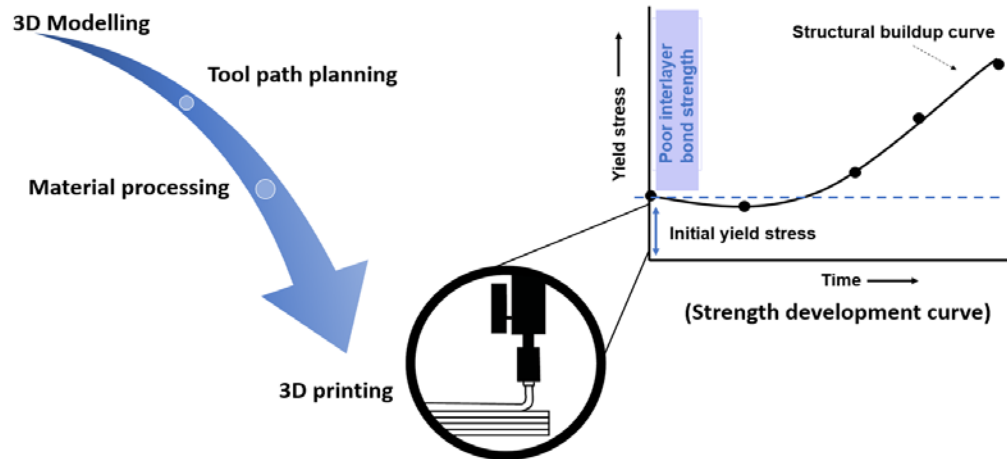
40 Incorporation of reinforcement into the concrete matrix during the printing process is one of the  
41 most challenging issues needed to be dealt with to enable structural applications of 3D printing  
42 technology [11, 12]. In terms of materials, most of the printable mixtures contain ordinary Portland  
43 cement (PC) as the prime binder material due to its inherent thixotropic properties that originate  
44 from the combined effect of interparticle, gravitational and inertial forces [13-17]. However, as  
45 the production of PC accounts for 5-7% of the total anthropogenic CO<sub>2</sub> emissions, alternative  
46 cementitious materials are being investigated for their suitability to be used in 3D printing. Recent  
47 studies [18-21] have demonstrated the use of cementitious industrial by-products to reduce the  
48 carbon footprint of PC in 3D printing applications. As a part of these initiatives, the importance of  
49 proper rheology control for smooth extrusion and higher buildability properties was highlighted.

50

51 In line with these initiatives, this study aims to investigate the use of alkali-activated slag (AAS)  
52 mixes in 3D printing applications to contribute to the development of a sustainable built  
53 environment. To enable this, the prepared AAS formulations included nanoclay due to its  
54 thixotropic properties [23]. Furthermore, nucleation seeds (i.e. hydromagnesite) were also  
55 included to improve the rheological and mechanical properties of AAS [24, 25]. Accordingly,  
56 previous research [26] has shown that the use of seeds can provide additional nucleation sites for  
57 the increased precipitation of hydrate phases, thereby enhancing the rate and degree of the  
58 hydration reaction. To study the effect of these additives, nanoclay was first introduced into AAS  
59 mixtures for improved printability. An initial assessment of the fresh properties led to the  
60 determination of the mix with the highest yield stress and lowest viscosity. Once this mix was  
61 determined, different dosages (i.e. 1-2% by mass of slag) of hydromagnesite seed was introduced  
62 into the mix design. X-ray diffraction (XRD) and field emission scanning electron microscopy

63 (FESEM) were employed to analyse the formation of hydration products and investigate the  
64 microstructural development at the end of the curing process.

65



66

67

68 **Fig. 1** Schematic of extrusion-based additive manufacturing of cement-based mixes

69

70

## 71 **2. Materials and Methodology**

72

### 73 **2.1 Raw materials and sample preparation**

74

75 The primary material utilized in this study was ground granulated blast-furnace slag (GGBS),  
76 provided by EnGro Corporation (Singapore). The chemical and physical properties of GGBS are  
77 shown in Table 1. The alkali activator was solid sodium metasilicate ( $\text{Na}_2\text{SiO}_3 \cdot 5\text{H}_2\text{O}$ ), obtained  
78 from VWR (Singapore). Nanoclay (NC), a thermal treated, highly purified attapulgite clay  
79 supplied by Actigel (USA), was used to improve the thixotropy of the AAS binder. As can be seen  
80 from Fig. 2(a), NC particles agglomerated into clusters of different size composed of a pleated  
81 spherical morphology. Hydromagnesite ( $4\text{MgCO}_3 \cdot \text{Mg}(\text{OH})_2 \cdot 4\text{H}_2\text{O}$ ) seeds, obtained from Fisher  
82 Scientific (UK) with a specific surface area of  $43.5 \text{ m}^2/\text{g}$ , were used as nucleation seeds to stimulate  
83 the nucleation and growth of hydration products. The SEM image of hydromagnesite (Fig. 2(b))  
84 demonstrated these seeds as spherical agglomerations with a diameter of  $\sim 1\text{-}7 \mu\text{m}$ , composed of

85 ~0.5  $\mu\text{m}$  diameter disks [24]. Fine aggregates with a maximum particle size of 1.18 mm was used  
 86 in a saturated surface dry (SSD) condition to formulate the AAS mortars.

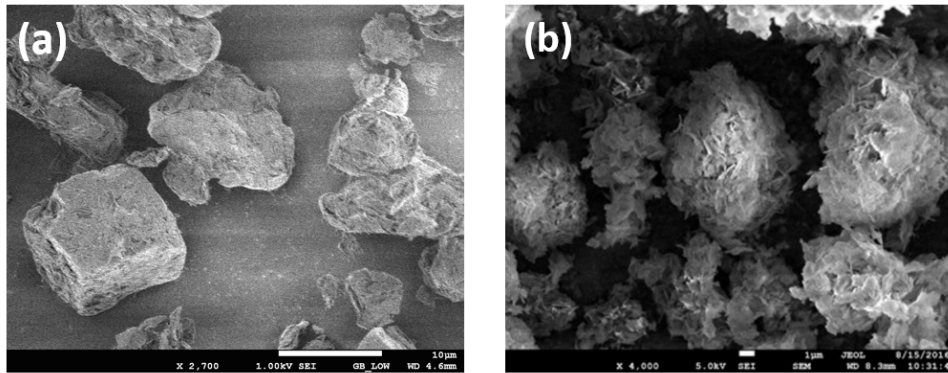
87

88 **Table 1** Chemical composition and physical properties of GGBS.

89

	Chemical composition (%)							Physical properties	
	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	CaO	MgO	SO <sub>3</sub>	TiO <sub>2</sub>	LOI	Specific gravity (g/cm <sup>3</sup> )	Blaine surface area (m <sup>2</sup> /g)
GGBS	29.65	15.56	39.37	7.54	4.32	1.75	4.0	2.85	> 300

90



91

92

93 **Fig. 2** SEM images of (a) nanoclay and (b) hydromagnesite seed [20]

94

95 The AAS used in this study was composed of only GGBS and 10% activator (i.e. by mass of slag)  
 96 in line with the findings of a previous study [26]. Mixtures were prepared using water to binder  
 97 (w/b) ratios of 0.35 and 0.40 for all pastes and mortars (i.e. at a sand/binder ratio of 0.83),  
 98 respectively. AAS mortars without NC were prepared by adding water to the slag and sand mixture  
 99 and mixed until a homogenous blend was obtained. For the preparation of mixes containing NC,  
 100 NC was first blended into the predetermined amount of water for 3-4 minutes, after which it was  
 101 added into the slag and sand mix and further mixed for another 2 minutes to ensure effective  
 102 dispersion. For mixes involving the use of seeds, the seeds were first dispersed in half of the  
 103 required total water and added into the mix, followed by the addition of NC, which was mixed  
 104 with the remaining water.

105

106

## 107 **2.2 Methodology**

108

### 109 **2.2.1 Static yield stress**

110

111 A commercial rheometer (Anton-Paar MCR 102) was used to measure the static yield stress of the  
112 prepared AAS mortars. To initiate the study, freshly prepared AAS mortars were loaded in a  
113 measuring cup. Stress growth test was then performed by applying deformation at a constant shear  
114 rate of  $0.1 \text{ s}^{-1}$ . The shear stress progressively developed to a maximum value, followed by its  
115 decline to reach an equilibrium value. The static yield stress was defined as the peak shear stress  
116 value [27].

117

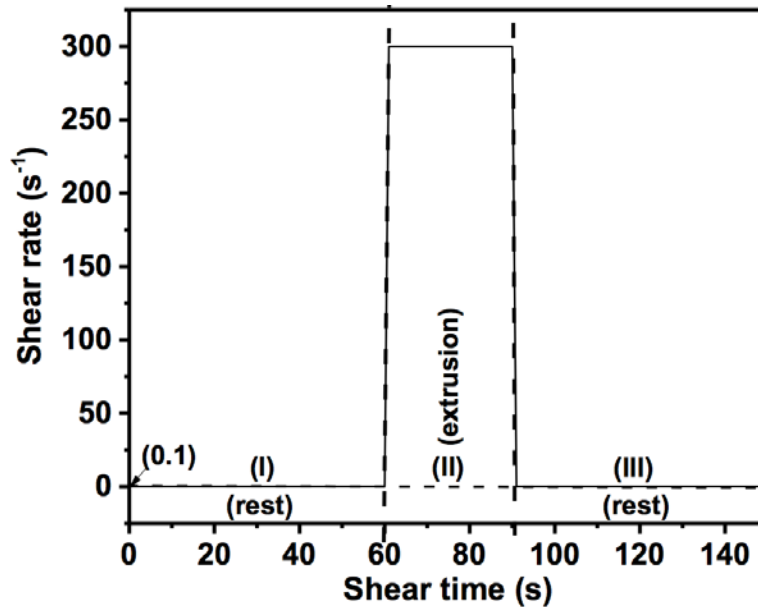
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### 119 **2.2.2 Thixotropy (shear thinning and viscosity recovery)**

120

121 Thixotropy is an important property of printable materials, which can be characterized by a high  
122 viscosity at low stress and vice versa [39]. While there are various methods to quantify thixotropy,  
123 the “viscosity recovery” test was used to measure the thixotropy of AAS mixtures in this paper.  
124 Shear thinning property was measured by applying a constant shear rate of  $300 \text{ s}^{-1}$ , while viscosity  
125 recovery was measured by following a three-stage protocol, as previously described in [28]. The  
126 three stages and their respective shear rates and shearing timings were decided by mimicking the  
127 concrete printing process, where the state of the material starts from rest (i.e. at hopper) followed  
128 by high shear (i.e. extrusion) and finally ends at rest (i.e. on print bed). Fig. 3 shows the schematic  
129 of the protocol used for the evaluation of viscosity recovery. In addition to thixotropy, the  
130 structural build-up rate was also calculated from the evolution of static yield stress after 0, 5, 10  
131 and 15 minutes of rest, which was used to assess the buildability property of the AAS mortars.

132



**Fig. 3** Viscosity recovery protocol for measuring the thixotropy of AAS mortars

### 2.2.3 3D printing

The printability criteria employed in this study were: (i) ease of extrusion through a rectangular nozzle (i.e. extrudability) and (ii) stability of the layered printed filament (i.e. buildability). A 4-axis gantry printer [29] was used to test the extrudability of AAS mortars using a 30 (L) x 15 (W) mm nozzle. The printer was connected to a screw pump and the print speed was varied from 60 to 100 mm/sec at a constant flow rate to find the optimum speed for better shape stability of the filaments. This was followed by the printing of a hollow cylinder with the optimum combination of flow rate and speed to demonstrate the buildability of the AAS mortar.

### 2.2.4 Microstructural characterization (XRD and FESEM)

Samples extracted from AAS pastes were stored in acetone to stop hydration, followed by vacuum drying in preparation for XRD and FESEM analyses. XRD was recorded on a Philips PW 1800 spectrometer using Cu K $\alpha$  radiation (40 kV, 30 mA), with a scanning rate of 0.04° 2 $\theta$ /step from



154 10 to 70° 2θ. FESEM was carried out with a Zeiss Evo 50 microscope to investigate the  
155 morphologies of the hydration products. The vacuum dried samples were mounted onto aluminium  
156 stubs using double-sided adhesive carbon disks and coated with gold before FESEM analysis.

157

158

### 159 **3. Results and Discussion**

160

#### 161 **3.1 Effect of nanoclay on extrusion rheology**

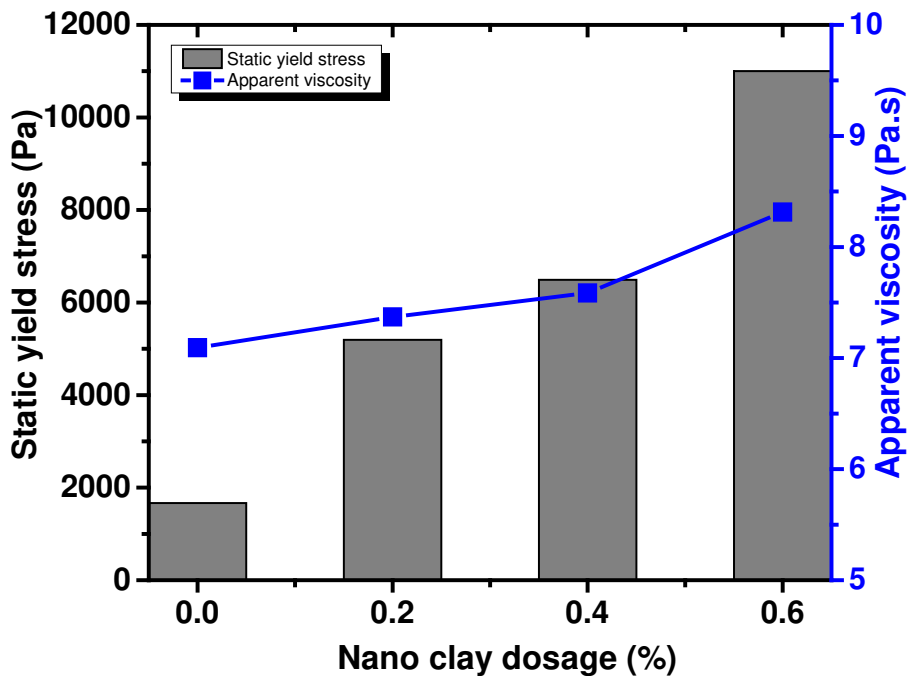
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163 Printability is defined as the ability of a mixture to be extruded (i.e. extrudability) and maintain its  
164 structural integrity when built in layers (i.e. buildability) [10, 32]. In this respect, yield stress is an  
165 important rheological parameter that influences printability. Similarly, viscosity of a mixture is  
166 also related to its extrudability. Therefore, the yield stress and viscosity values obtained from a  
167 rotational rheometer were synergistically considered to analyse the printability of the prepared  
168 AAS mixes. Fig. 4 shows the effect of the inclusion of different amounts of NC on the static yield  
169 stress and apparent viscosity of AAS mortars. AAS mixes usually demonstrate low yield stress  
170 values due to the absorption of silicate anions (i.e. from the activator) on the slag surfaces, which  
171 results in strong double-layer repulsive forces between the slag particles, thereby causing particle  
172 separation and a low yield stress [31]. However, when compared to the control mix (0% NC), an  
173 increase in the NC content led to a higher static yield stress. Accordingly, the inclusion of a small  
174 amount of NC (0.4%) was sufficient to increase the yield stress by higher than three times when  
175 compared to the control mix. Therefore, the addition of NC enabled rapid flocculation, making it  
176 an ideal thixotropic material for 3D printing applications.

177

178 While the use of NC increased the yield stress of AAS mixes, the apparent viscosity remained  
179 relatively constant under different NC contents. This could be useful in providing a smooth  
180 extrusion and deposition of the prepared mix, maintaining the shape fidelity of the filaments. Such  
181 unique behaviour of NC is attributed to the flocculation phenomena that controls the rheology,  
182 depending on the applied shear forces at different stages of concrete printing [32]. Accordingly,  
183 NC particles carry a negative and positive charge on their opposite ends. They tend to associate  
184 with each other by electrical attraction when the material is at rest. NC also presents a higher initial

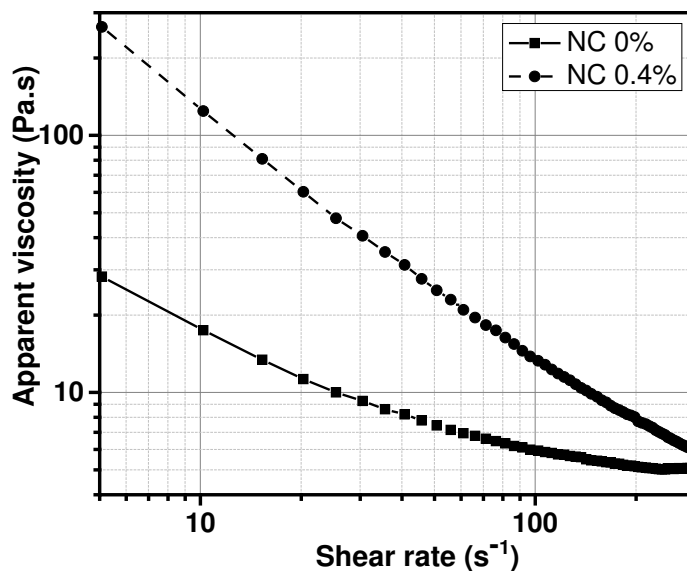
185 viscosity at rest, which then decreases under strong pre-shearing as the structure breaks down [33].  
186 Amongst the mixes prepared, since the addition of 0.6% NC resulted in the maximum yield stress  
187 that was measurable by the rheometer, this was the highest amount of NC introduced into the  
188 prepared AAS mortars in this study. However, as it provided the highest yield stress within the  
189 extrudable limit (< 8 KPa), a NC dosage of 0.4% was selected as the optimum amount for  
190 subsequent analyses. Alternatively, the addition of 0.6% NC led to the extrusion of a discontinuous  
191 filament, which could negatively affect the interlayer bond strength due to the high yield stress of  
192 the mixture.  
193



194  
195  
196 **Fig. 4** Static yield stress and apparent viscosity as a function of NC dosage  
197

198 Fig. 5 shows the flow curves over a range of shear rates ( $1-300\text{ s}^{-1}$ ) for AAS mortars with and  
199 without the addition of 0.4% NC. A shear-thinning behaviour was observed in both samples, which  
200 indicated increased viscosities with decreasing shear rates. When compared to the control mix, the  
201 NC modified mix resulted in higher viscosities at low shear rates, which could be helpful in  
202 maintaining the shape of the filament and support more layers. Alternatively, both mixtures  
203 demonstrated comparably low viscosities at high shear rates, which could enable a smooth

204 concrete flow without any discontinuity in the filament deposition. These results have  
205 demonstrated the role of NC in contributing to the development of more rigid AAS mixes via the  
206 flocculation of clay, resulting in mixes with high viscosities at low shear rates and vice versa.  
207



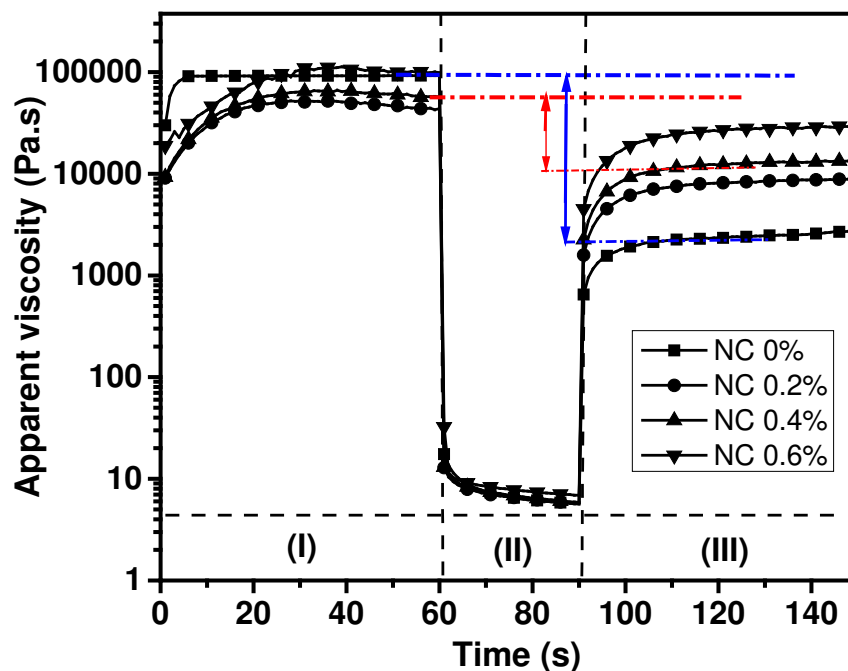
208  
209

210 **Fig. 5** Viscosity as a function of shear rate for the control (0% NC) and 0.4% NC modified AAS  
211 mortars

212

213 In order to be ideal for extrusion based printing purposes, a material should be highly thixotropic  
214 [10, 34]. In this respect, a low viscosity under a high shear force and the recovery of the viscosity  
215 to its original value after the removal of the shear force are preferable. Therefore, it is important  
216 to know how fast a deposited AAS mortar can recover its initial viscosity before the next layer is  
217 printed. Accordingly, Fig. 6 shows the viscosity recovery curves of AAS mortars including  
218 different amounts of NC. Although the initial viscosity (i.e. at stage I) of the control sample was  
219 higher, its recovery ability after strong shearing (i.e. at stage III) was much lower. This difference,  
220 shown by the blue arrow mark (Fig. 6), was an indication of the low recovery property of the  
221 control sample, which would negatively affect its shape retention after deposition as the viscosity  
222 after extrusion reduced due to poor recovery ability. On the other hand, mixes including NC  
223 revealed significantly better recovery properties, as shown by the red arrow mark (Fig. 6). The  
224 improved recoveries of these samples were attributed to the thixotropic property of NC. Amongst

225 the prepared samples, AAS including 0.4% NC was chosen for the assessment of its printability  
 226 and structural build-up property due to its extrudable nature and 25% higher recovery ability than  
 227 the control mix (0% NC).  
 228



229

230

231 **Fig. 6** Effect of the NC on the viscosity recovery of AAS mixtures

232

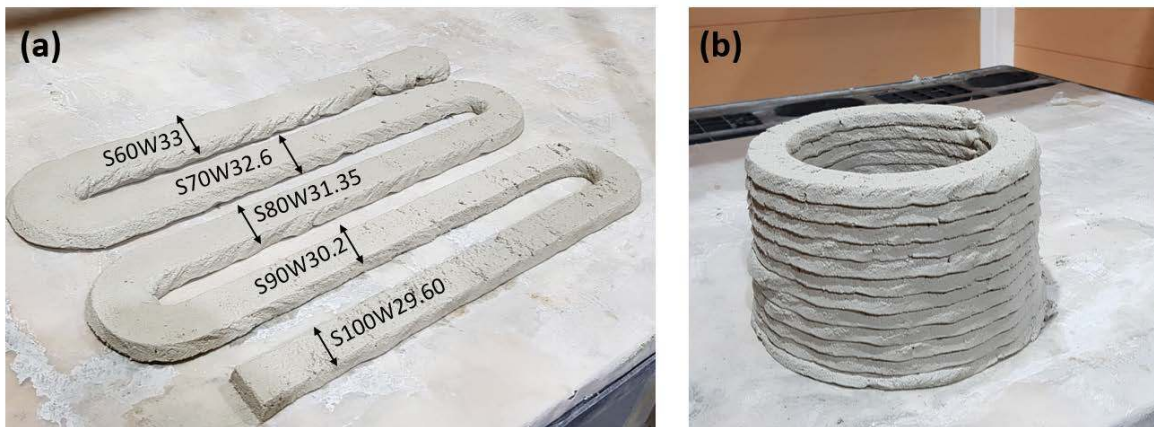
233

234 **3.2 3D printing**

235

236 3D printing of cementitious materials is a complex process, during which material properties, part  
 237 design and process parameters interact with each other and define the characteristics of the final  
 238 product. Process parameters like printing speed, nozzle shape and flow rate have a strong impact  
 239 on the dimensional accuracy and mechanical properties of the final product [3, 29]. Fig. 7(a) shows  
 240 the effect of printing speed (i.e. ranging between 60-100 mm/sec) on the dimensional accuracy of  
 241 the filaments, while flow rate was kept constant. The printing process revealed that the printing  
 242 speed and flow rate were dependant on each other. Accordingly, at each flow rate, there was an  
 243 optimum speed that enabled the production of filaments with the same dimension as the nozzle

244 inlet. An increase or decrease in the speed beyond this optimum value would cause extra or  
245 insufficient material deposition, respectively. Out of the trials performed, setting the printing speed  
246 to 90 mm/s resulted in extrudates with a width (W) that was equivalent to the nozzle dimension  
247 (30 mm). Therefore, to demonstrate the possibility of using the developed AAS mortars in 3D  
248 printing applications, this speed was used in the printing of a cylinder with a diameter of 20 cm.  
249 Accordingly, the AAS mix containing 0.4% NC was printed up to 15 layers without the  
250 deformation of the bottom layer, as shown in Fig. 7(b). The deformation during printing can also  
251 be simulated by using optical techniques based on digital photogrammetry or terrestrial laser  
252 scanning, as discussed in previous studies [34].  
253



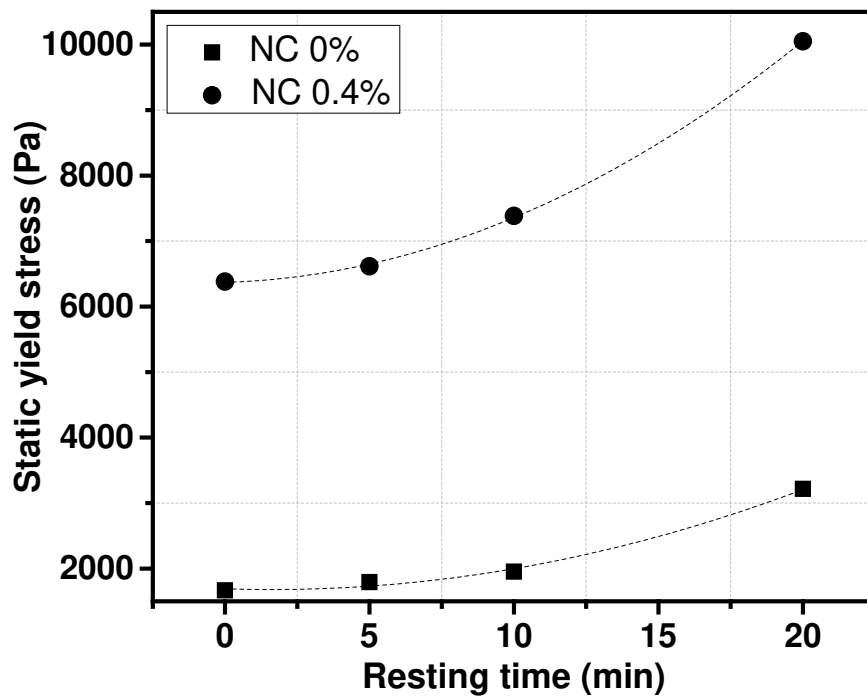
254  
255  
256 **Fig. 7** 3D printing of AAS mix containing 0.4% NC, showing the effect of: (a) printing speed on  
257 filament dimension and (b) buildability of several layers

258  
259  
260 **3.3 Time-dependent rheological properties**

261  
262 The rheology of the material changes with time according to binder chemistry and presence of  
263 additives. The evolution of rheology with time is an important parameter for the printing process  
264 since fresh concrete is deposited in a layer-by-layer manner. This necessitates the material to  
265 possess fresh properties that can provide a 3D printed structure with structural stability [35].  
266 Depending on the printing volume and speed, the material can be tailored to gain sufficient strength

267 before the printing of subsequent layers. The early strength of the material can be indirectly  
268 measured from its yield strength. Accordingly, Fig. 8 shows the yield stress values of the control  
269 AAS mix and the AAS mix containing 0.4% NC at different time intervals, obtained directly from  
270 the stress growth test. The inclusion of NC was found to increase the static yield stress of AAS  
271 mixes, revealing values that were three times higher than those of the control mix. This  
272 improvement in the yield stress was associated with the flocculation of NC particles, as discussed  
273 earlier in Section 3.1.

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275

276

277 **Fig. 8** Evolution of the static yield stress of AAS mortars with and without 0.4% NC

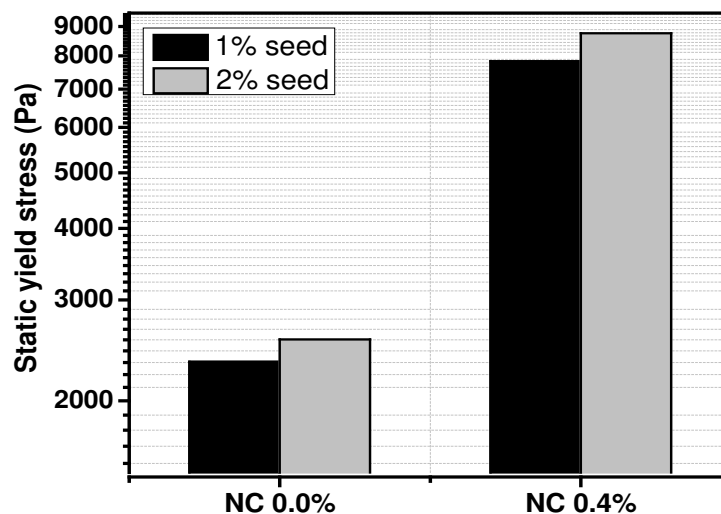
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279 Alternatively, the rate of change of yield stress over time, also known as the structural build-up  
280 rate, was not significantly affected by the addition of clay. AAS mix containing 0.4% NC  
281 progressively gained yield stress, similar to the control sample. This observation was in line with  
282 the findings of previous studies [32, 33], where the addition of NC was reported to have an  
283 immediate impact on the yield stress, but have little or no influence on the rate of change of the  
284 yield stress over time. To overcome this issue, an accelerator can be added into AAS mixtures,  
285 which will enhance the build-up rate by accelerating the chemical (i.e. hydration) reaction.

286 Previous studies [24-26] have shown that nucleation seeds such as Ca- or Mg-based  
287 hydrate/carbonate phases can provide additional nucleation sites for the rapid and increased growth  
288 of hydration products. Therefore, different amounts of hydromagnesite  
289 ( $4\text{MgCO}_3 \cdot \text{Mg}(\text{OH})_2 \cdot 4\text{H}_2\text{O}$ ) seeds were introduced into the AAS mix containing 0.4% NC to  
290 investigate their effectiveness in improving the build-up rate and early age mechanical properties.  
291

292 Fig. 9 shows the effect of 1% and 2% seed addition into the AAS mortar with and without the use  
293 of NC. Along with the findings of previous studies, where it was revealed that high seed contents  
294 were not effective due to dispersion issues [24, 25], only up to 2% seed addition was investigated  
295 in this study. Although the seed dosage had a little effect on the yield stress of both mixes, its  
296 influence on increasing the rate of change of yield stress over time, which can enhance the  
297 buildability of the printed structure without the need for any additional admixtures, should be  
298 evaluated before any final conclusions on the effectiveness of seeds could be made. Accordingly,  
299 Fig. 10 shows a successfully printed twisted column involving the use of the AAS mix containing  
300 0.4% NC and 2% hydromagnesite seeds. As a part of this approach, a slender column was printed  
301 directly from the CAD model demonstrated in Fig. 10(a). This process was completed without any  
302 significant bottom layer deformation, as can be seen in Fig. 10(b). While this was an indication of  
303 the necessary stiffness of the AAS mix in resisting deformation imposed by top layers, its higher  
304 stiffness resulted in a rough surface texture, which could be improved for better printing results.

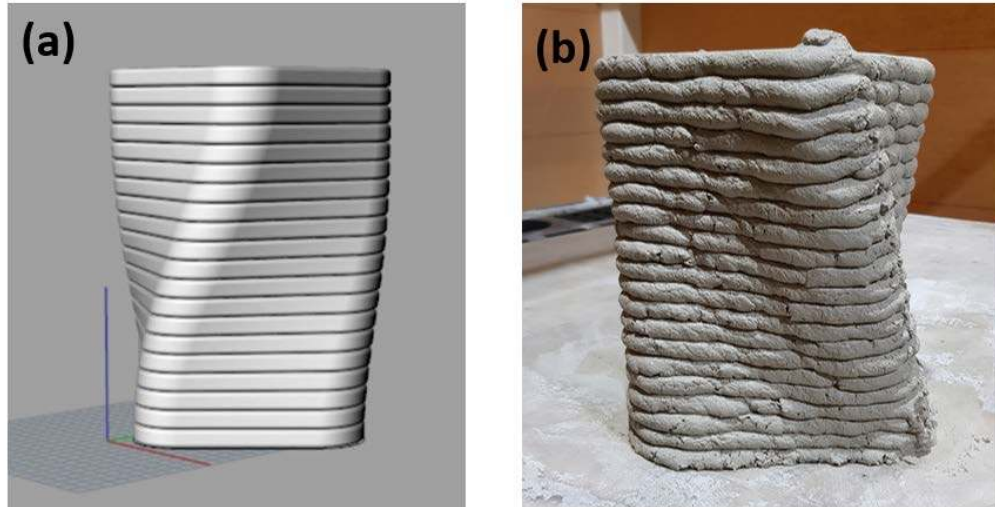
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307

308 **Fig. 9** Effect of nucleation seeds on the yield stress of AAS mixes containing 0.4% NC, obtained  
309 just after mixing  
310



311  
312  
313 **Fig. 10** 3D twisted column generated by using the AAS mix containing 0.4% NC and 2%  
314 hydromagnesite seeds, showing: (a) 3D CAD model and (b) actual printed structure  
315  
316

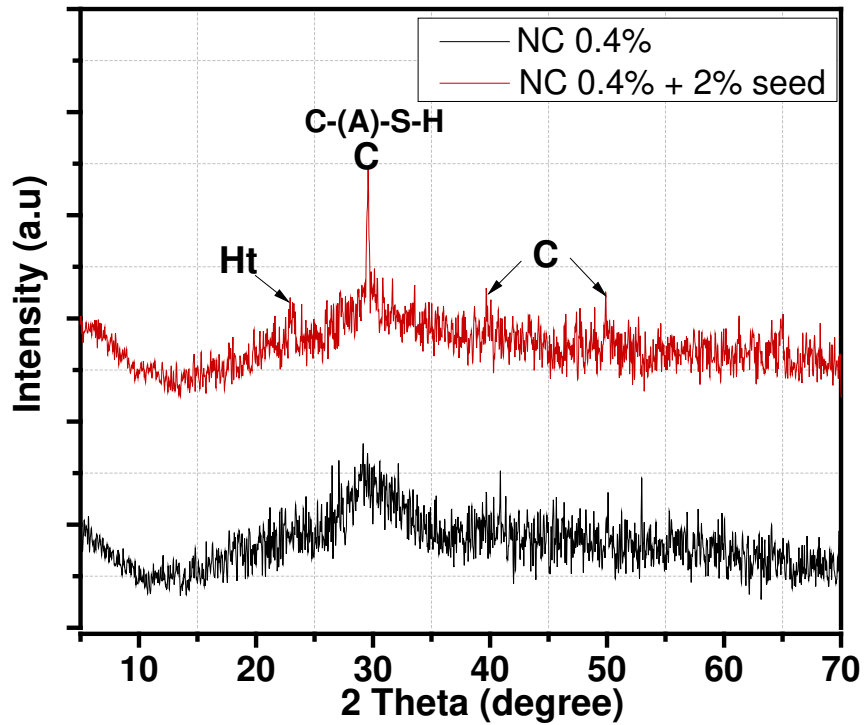
### 317 **3.4 Microstructural characterization**

318  
319 The feasibility of using the developed AAS mix in large-scale printing applications was further  
320 evaluated via the assessment of its microstructural properties Fig. 11 presents the XRD patterns of  
321 the AAS mix containing 0.4% NC and 0-2% hydromagnesite seeds after 14 days of curing. The  
322 main hydrate phase observed in both samples was C-(A)-S-H gel (i.e.  $30-31^\circ 2\theta$ ; PDF #00-033-  
323 0306), which was in line with the findings of previous studies [37, 38]. A comparison of the XRD  
324 patterns of both samples revealed the higher intensity of C-(A)-S-H in the seeded sample when  
325 compared with the unseeded sample. Along with C-(A)-S-H, a small amount of calcite ( $\text{CaCO}_3$ ;  
326 PDF #01-071-3699) and hydrotalcite ( $\text{Mg}_6\text{Al}_2\text{CO}_3(\text{OH})_{16}\cdot 4(\text{H}_2\text{O})$ ) were also observed.  
327 Furthermore, the microstructures of both samples, shown in Fig. 12, revealed the denser  
328 microstructure of the seeded sample in comparison to the unseeded sample. The wide-spread  
329 growth of C-(A)-S-H gel on the surfaces of the rosette-like hydromagnesite particles confirmed



330 the role of these seeds in providing additional nucleation sites for the increased precipitation of  
331 hydration products, which could lead to fast setting and improved early mechanical performance  
332 [36, 40].

333

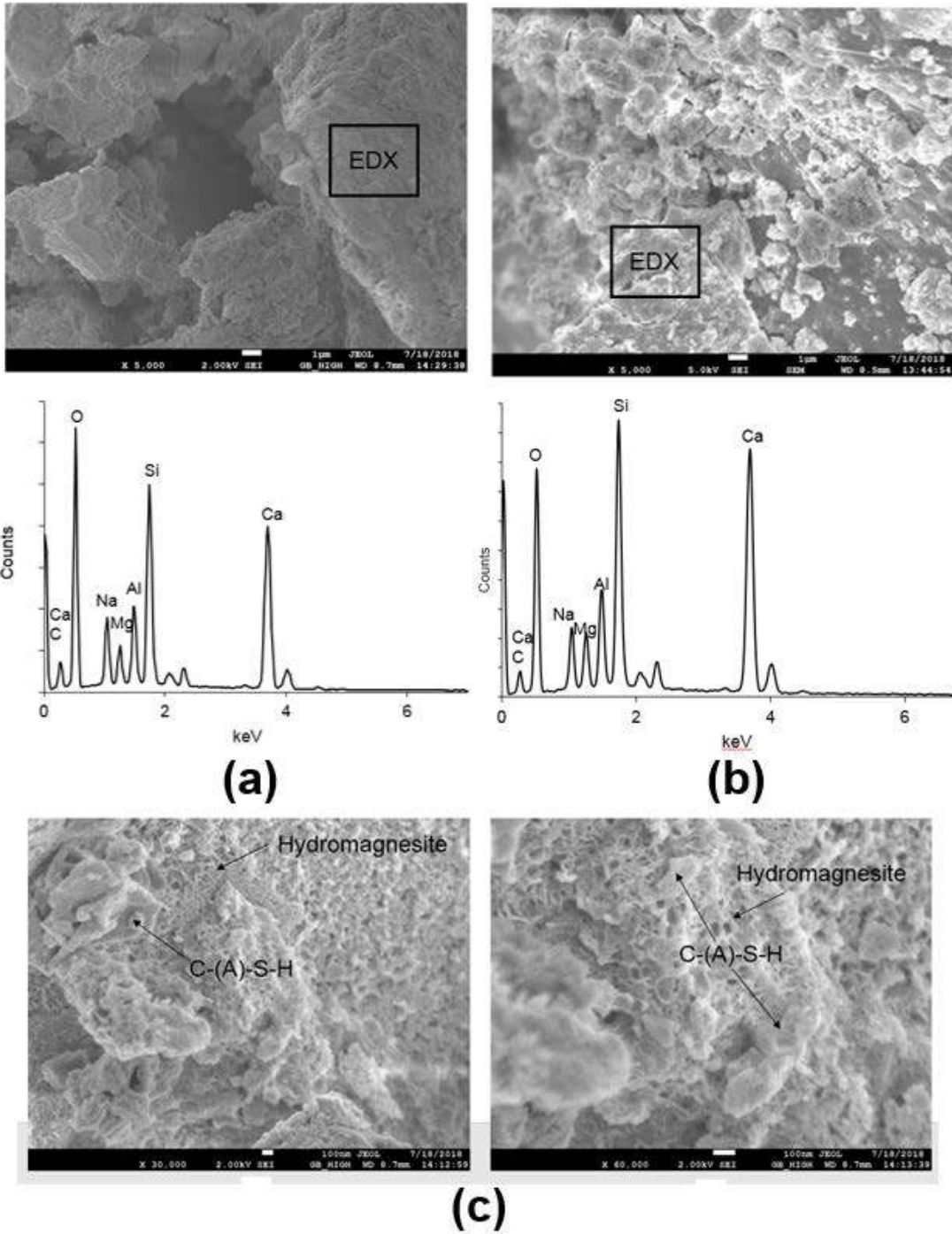


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335

336 **Fig. 11** XRD patterns of the AAS mix containing 0.4% NC and 0-2% hydromagnesite seeds after  
337 14 days of curing (Ht: hydrotalcite and C: calcite)

338



339  
 340  
 341  
 342  
 343  
 344

**Fig. 12** SEM images and elemental composition of the AAS mix containing 0.4% NC and (a) 0% seeds and (b)-(c) 2% seeds

#### 345 **4. Conclusions**

346

347 This paper aimed to improve the rheological properties of alkali-activated slag (AAS) binders for  
348 extrusion-based 3D printing. While slag was used as the main binder, different amounts of  
349 nanoclay (NC) were introduced to enable the development of extrudable and buildable (i.e.  
350 printable) mixes. Selected mixes were subjected to a range of rheological measurements to  
351 determine their yield stress, viscosity and thixotropy recovery required for 3D printing  
352 applications. The AAS mix by itself was found to exhibit a low yield stress due to the plasticizing  
353 and deflocculating effects of the silicate particles. Inclusion of 0.4% NC in these AAS mixes  
354 significantly improved the initial yield stress due to the flocculation effect. However, structural  
355 rebuilding rate remained unaffected even in the presence of NC and this was resolved via the  
356 addition of 2% hydromagnesite seeds, which can increase the rate of the hydration reaction and  
357 early strength development of AAS mixes, contributing to the buildability property necessary for  
358 large-scale concrete printing.

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360 While this study has paved the way for the development of alternative binders with improved  
361 properties for 3D printing applications, further studies focusing on the optimization of the mix  
362 design in terms of the NC content and seed type could enable the end users to fully harvest the  
363 benefits of each additive in creating sustainable mixes with an enhanced performance.

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