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Life cycle strain monitoring in glass fibre reinforced polymer laminates using embedded fibre Bragg grating sensors from manufacturing to failure

Michael W Nielsen¹, Jacob W Schmidt², Jacob H Høgh³,
Jacob P Waldbjørn², Jesper H Hattel¹, Tom L Andersen³ and
Christen M Markussen³

Abstract

A holistic approach to strain monitoring in fibre-reinforced polymer composites is presented using embedded fibre Bragg grating sensors. Internal strains are monitored in unidirectional E-glass/epoxy laminate beams during vacuum infusion, curing, post-curing and subsequent loading in flexure until failure. The internal process-induced strain development is investigated through use of different cure schedules and tool/part interactions. The fibre Bragg grating sensors successfully monitor resin flow front progression during infusion, and strain development during curing, representative of the different cure temperatures and tool/part interfaces used. Substantial internal process-induced strains develop in the transverse fibre direction, which should be taken into consideration when designing fibre-reinforced polymer laminates. Flexure tests indicate no significant difference in the mechanical properties of the differently cured specimens, despite the large differences in measured residual strains. This indicates that conventional flexure testing may not reveal residual strain or stress effects at small specimen scale levels. The internal stresses are seen to influence the accuracy of the fibre Bragg gratings within the loading regime. This study confirms the effectiveness of composite life cycle strain monitoring for developing consistent manufacturing processes.

Keywords

Strain monitoring, fibre Bragg grating sensors, curing, process-induced strains, life cycle, composite

Introduction

Fibre-reinforced polymer (FRP) composites are widely used in various industrial applications. Such materials provide low weight, excellent corrosion resistance and high strength, which are beneficial properties in, for instance, the construction of commercial wind turbine blades and civil structures. The manufacturing process of the composite structure can, however, greatly influence the performance of the part, which is dependent on the presence of voids, dry spots, matrix cracks, waviness of the ply stack and other manufacturing-related defects. Such defects can result in crack initiation when the structure is loaded and premature failure at lower loads than expected when designing the structure.

During the manufacture of thermosetting matrix composites, a material state transition of the resin

occurs, from a viscous, to rubbery and glassy state during curing. It is within this transition that various mechanisms result in the development of process-induced stresses in the composite structure, which when unconstrained result in shape distortions.^{1–3}

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In cases where these process-induced stresses are sufficiently large and unavoidable, for instance due to strong internal or external constraints, they can result in various strength reducing mechanisms that support premature failure. For instance, large process-induced residual stresses are known to induce matrix cracking, interfacial debonding and delamination in fibre-reinforced composites.¹ In other works,^{4–6} demonstrated that for thick-walled composite tubes, residual stresses can be so great that matrix cracking was promoted. At a micromechanic level, Nedele and Wisnom⁷ showed that process-induced stresses at the fibre–matrix interface can be as large as 30 MPa, due to the combination of hoop and radial stress components. This stress magnitude is substantial considering that the typical tensile strength of epoxy matrix materials is approximately 60–65 MPa. Furthermore, due to the viscoelastic nature of the polymer matrix material, large process-induced residual stresses can also influence the long-term dimensional stability of the part, due to creep or stress relaxation behaviour.

The main mechanisms responsible for process-induced shape distortions and residual stresses are: (a) the thermal expansion mismatch of the constituent materials; (b) matrix chemical shrinkage during curing; and (c) interactions at the tool/part interface.^{2,3,8–11} Of particular interest, when considering thick laminate composites is also the manufacturing cure cycle design. High cure temperatures and fast heating/cooling rates can result in highly non-uniform temperature and cure gradients within the part resulting in differential curing and subsequent development of internal stresses.^{3,4–6,11} Similar effects in the form of a non-uniform distribution of in-plane shear stresses can arise due to constraints posed by tooling during processing.^{10,12} A combination of the above-named residual stress mechanisms and manufacturing defects can cause local premature failure at lower load magnitudes than would otherwise be predicted in a defect-free composite structure.

In order to experimentally monitor the internal strain development during manufacturing, the use of fibre optic sensors has gained approval as a desirable in situ method.^{13–24} Apparent advantages include their ease of integration structurally during the composite manufacturing phase and the minimal influence to the host part structural properties. In recent works by the authors, fibre Bragg grating (FBG) sensors were successfully embedded in a laminate composite for cure strain monitoring purposes.¹⁹ As opposed to the use of conventional non-intrusive strain/deformation monitoring methods (e.g. using strain gauges (SGs) or digital image correlation (DIC)), embedded optical fibre sensors offer the advantage of being able to monitor local internal strains in the composite readily. Moreover,

FBG sensors offer the capability to accurately monitor resin flow front progression during infusion.^{14,23,24}

This study focuses on investigating the development of internal strains in glass FRP (GFRP) composites in a holistic manner during the composite life cycle using embedded FBG sensors. The term *life cycle* in this context refers to the composite life time from manufacturing using vacuum infusion and until mechanical testing, as illustrated in Figure 1. Thus, using this approach, we wish to monitor the development of process-induced internal strains and investigate the influence of these strains and residual stresses in the composite during manufacturing and testing. This study provides an approach to monitor and gain knowledge during the manufacturing procedure of FRP composites. Such manufacturing control is important if consistency in the quality of the composite is desired. In addition, information of the residual strain and stress state in the structure is important if these are to be accounted for in the design of a structural component.

FBG sensor principle

Strain monitoring using optical fibres with FBG sensors requires a light source, an interrogator unit and an appropriate data acquisition software tool. The light source transmits a broadband light spectrum through the optical fibre and the axially placed FBG sensor back-reflects light within a specific pre-defined wavelength. This back-reflected light is fed into the interrogating unit designed to measure the reflected light signal. Modulation of the reflected signal, centred along the Bragg wavelength, is linearly related to mechanical or thermal loads. The FBG wavelength variation is hence determined using¹³

$$\frac{\Delta\lambda_B}{\lambda_B^0} = K_\epsilon \left[\Delta\epsilon_{app}^{host} + \Delta\epsilon_{th}^{host} \right] + K_T \Delta T = K_\epsilon \Delta\epsilon_{tot} + K_T \Delta T \quad (1)$$

where $\Delta\lambda_B$ is the change in Bragg grating wavelength and λ_B^0 the initial unstrained grating wavelength, K_ϵ and K_T are the optical fibre mechanical strain and temperature sensitivities, specific for the type of sensor used. $\Delta\epsilon_{tot}$ is the change in total strain – a summation of the increments from the applied strain $\Delta\epsilon_{app}^{host}$ and the host thermal strain $\Delta\epsilon_{th}^{host}$. Equation (1) describes the linear mechanical and thermal response of the FBG sensor. When the FBG is embedded in a composite material, here on termed the host material, complete strain transfer is achieved once bonding between the optical fibre and the host material occurs. Subsequently, any mechanical straining and thermal expansion of the host material would govern the response of the FBG sensor. Knowing the temperature

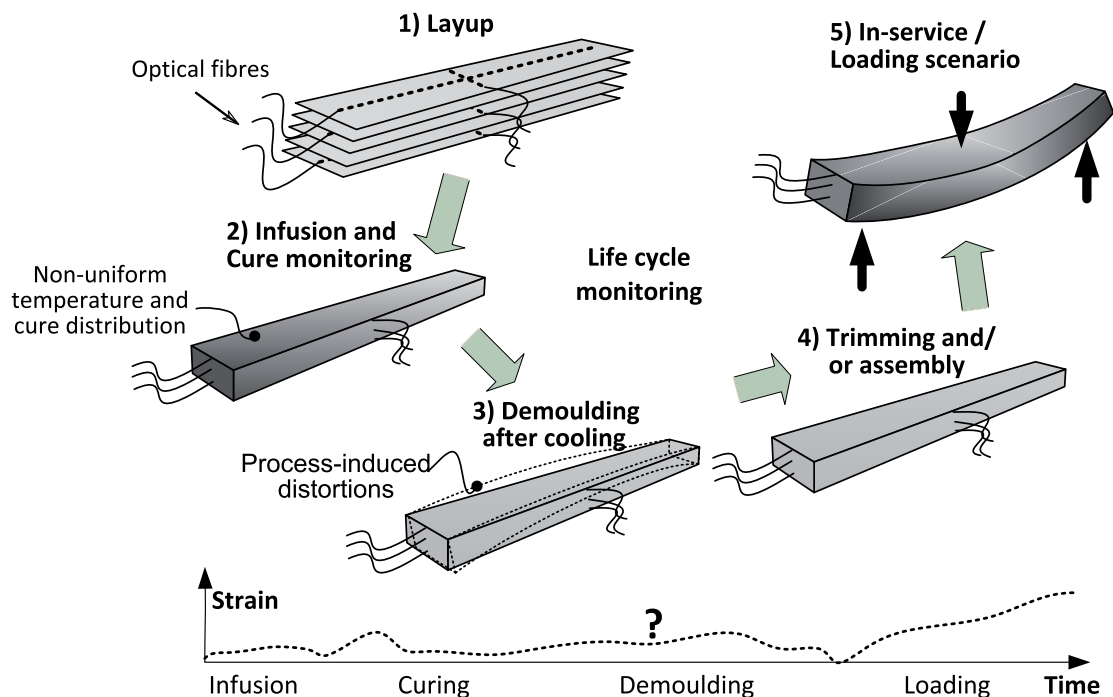


Figure 1. Schematic of life cycle monitoring approach for a GFRP laminate using embedded optical fibres where internal strains are monitored from manufacture to loading.

GFRP: glass fibre reinforced polymer.

and sensor wavelength variation, the corresponding change in host material total strain in the optical fibre axial direction can be determined using equation (1) by isolating $\Delta\varepsilon_{tot}$.

In this study, silica optical fibres provided by FOS&S, Belgium, were used, with multiple draw tower grating (DTG[®]) FBG sensors with anOrmocer[®] coating (cladding diameter 125 μm). The sensitivity coefficients K_ε and K_T are equal to $0.775\text{E} - 6 \mu\text{e}^{-1}$ and $6.27\text{E} - 6^\circ\text{C}^{-1}$, respectively. When considering the magnitudes of the mechanical strain and temperature sensitivity coefficients, it is clear that accurate temperatures at the vicinity of the sensors is needed in order to determine precise strains. It has been shown that the strain sensitivity remains constant below 180°C for this optical fibre.²⁰ Furthermore, DTG FBGs are known to exhibit little variation in strain-optic coefficients when loaded which is advantageous. The Ormocer coating helps reduce the sensor sensitivity to external transverse loading, hence increasing the accuracy of the axial strain measurements. An I-MON E interrogator, from Ibsen Photonics, Denmark, was used for data acquisition. The I-MON E is a stand-alone interrogation monitor with a wavelength bandwidth of 85 nm, capable of assessing <70 FBG sensor signals. The interrogator wavelength accuracy lies within 5 pm with a sampling rate of up to 970 Hz. The grating length used in this work, i.e. the effective strain

measurement region, spans 4 mm of the optical fibre length at each sensor positioned along the optical fibre. In other works, where a similar length is used, no peak splitting of the back-reflected wavelength spectrum was experienced.^{21,22} Peak-splitting is generally attributed to the existence of transverse loading of the sensor causing a local distortion of the reflected signal spectra. This is most likely to occur when using long gratings where a large transverse local load is present in the vicinity of the FBG sensor. By having a short grating length and using a coated fibre, this problem is greatly reduced. Studies found that good predictions of process strains in neat resin and composite laminates were achievable using a similar setup.^{13,16,19}

Experimental procedure

Materials

The composite constituent materials used in this study comprise of unidirectional (UD) fibre reinforcement fabric type L1200/G50F-E06-A, from Devold AMT, with a nominal area weight of 1246 g/m² and a thermoset epoxy resin matrix material type Airstone 760E mixed 100:32 parts-by-weight with Airstone 776H hardener, from Dow Chemicals Company. The reinforcement fibre and resin matrix material used in this study are representative of materials commonly

used in commercial wind turbine blades. Material properties are summarised in Table 1. As mentioned in the introduction, epoxy resins exhibit volumetric chemical shrinkage during cross-linking, generally known to lie between 3% and 7%.^{9–11} Both constituent materials used exhibit isotropic thermal expansion behaviour, as observed from Table 1.

Manufacturing procedure

In order to investigate the influence of the cure cycle profile and boundary conditions on the process-induced strain development and subsequent part performance, two different manufacturing procedures were employed, as summarised in Figure 2 and Table 2. The tool used was a transparent tempered glass plate, chosen in order to visibly ensure full wetting of the reinforcement fibres during infusion. Vacuum infusion was performed as edge infusion with resin flow in the longitudinal fibre direction in the entire length of the specimens. An Enka channel mesh and highly porous distribution net (Figure 2) are used in order to allow fast flow along the top surface of the reinforcement fabric and subsequent through-thickness wetting. Fibre layup was established using 22 layers of UD glass fibre fabric each cut to $500 \times 100 \text{ mm}^2$ ($l \times b$) and stacked symmetrically on the glass tool with the fibre backing material faced outwards. During the layup procedure, placement of optical fibres and

thermocouples was conducted as described in the subsequent section. After layup, a vacuum leak test was carried out prior to infusion. The epoxy resin was mixed with the hardener and degassed in a vacuum chamber with a vacuum pressure above 95 kPa, until all foaming had subsided. Infusion was carried out at ambient temperature with a vacuum pressure of 95 kPa. The inlet hose was closed once resin flowed through the outlet hose and full impregnation of the reinforcement fibres was visible. Full wetting of the reinforcement fibres took approximately 45 min. The vacuum pressure was then reduced to 60 kPa and pressure equalisation and further compaction of the laminate was allowed for 15 min before the outlet was closed. The vacuum infused specimens were placed into pre-heated ovens at 50°C and 80°C for approximately 5 and 3 h, respectively, for the different cure cycles investigated (Table 2). The in-mould cure periods were chosen such that the resin was nearly fully cured based on differential scanning calorimetry isothermal scan data at various temperatures. Different cure temperatures were chosen in order to achieve different thermal residual stresses.

Table 1. Material properties of E-glass reinforcement fibre and epoxy resin.

	Density (kg/m^3)	Young's modulus (GPa)	Poisson's ratio	CTE ($\times 10^{-6} \text{ } ^\circ\text{C}^{-1}$)
UD E-glass fibre	2600	72.0 ^a	0.22	5.04 ^a
Epoxy resin (cured)	1040	3.116	0.32 ^b	65.41 ^b

CTE: coefficient of thermal expansion; UD: unidirectional.

^aIsotropic for E-glass.

^bApproximated value from the literature.

^cResin supplier data sheet.²⁸

Table 2. Experimental design matrix showing layup, cure schedules and boundary conditions.

	Type A	Type B
Layup	[0] ₂₂	[0] ₂₂
Cure temperature ^a (°C)	50	80
Temperature hold ^b (min)	300	180
Cooling	Natural convection	Natural convection
Post-cure temperature ^a (°C)	90	–
Post-cure hold ^b (min)	120	–
Tool/part interface	Release agent ^c	Perforated release foil

^aPre-heated oven temperature.

^bTemperature hold period includes heating from ambient- to pre-heated oven temperature.

^cThe tool surface was treated with release agent prior to the experiments.

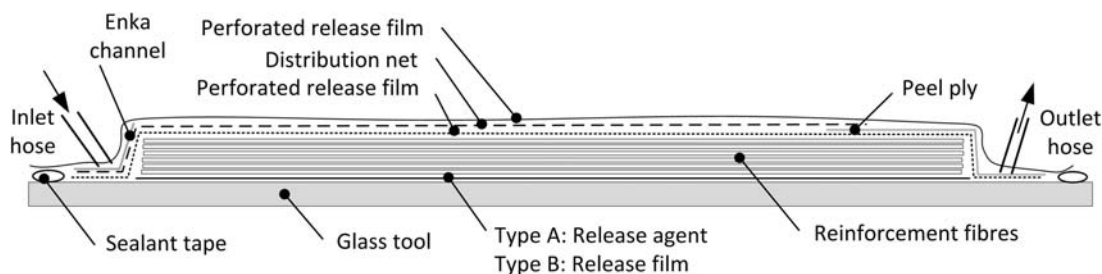


Figure 2. Schematic of vacuum infusion setup.

The cured in-mould laminates were cooled by natural convection to ambient temperature. Cooling in this manner took approximately 3.5 h and 5.5 h, respectively, for the 50°C and 80°C cure cycles. After cooling, demoulding was carried out and type A specimens were post-cured in a pre-heated oven at 90°C for 2 h. Type B specimens were not post-cured. All specimens were then cut into $500 \times 45 \text{ mm}^2$ ($l \times b$) beams. Trimming of the beam sides was done in order to ensure a constant cross section of each specimen, necessary in the mechanical tests. The manufactured specimens had a final average thickness of 19.04 mm with a variation of approximately 0.20 mm, and a fibre volume fraction of approximately 55%.

During manufacturing, two different tool/part interfacial interactions were investigated using release agent and release film (Table 2). This was done to investigate non-uniform through-thickness strain variations during manufacturing as well as the possible occurrence of locking.¹¹ A total of 10 laminates were manufactured using the different cure cycle and boundary conditions summarised in Table 2 evenly divided between the specimens.

Optical fibre and thermocouple sensor embedment

During the layup process, optical fibres consisting each of three FBG sensors were placed in the laminate, as illustrated in Figure 3. The optical fibres were placed along the UD reinforcement fibre direction (i.e. longitudinally) as well as perpendicular to the reinforcement fibres (transversely). This was done at three different interlayer locations along the ply stack thickness direction: between the 1st and 2nd, 11th and 12th and 21st and 22nd plies representing the laminate beam specimen top, middle and bottom planes (*Top*, *Mid* and *Bot*

henceforth). In order to ensure that the optical fibres remained in place, they were *sewn* in place using the backing fibres of the reinforcement fibre fabrics, as an alternative to using adhesives, etc. In this manner, the optical fibres remained aligned but at the same time not entirely constrained. Temperature monitoring during processing was conducted using type-J (Fe-CuNi) $2 \times 0.5 \text{ mm}$ thermocouples, placed at various regions on the outer surface and within different layers approximately 10 mm from the edge of the laminate. Temperature data was collected using an eight-channel DaqPRO data acquisition system (Fourier systems, Barrington, USA), with a sampling frequency of 10 Hz. Data logging were commenced once infusion was initiated, as well as at different intervals at ambient temperatures between the various manufacturing steps. Hence, information revealing the internal strain development at discrete times throughout the manufacturing process was obtained from the specimens at different through-thickness layers in different reinforcement fibre directions. Strain and temperature monitoring was carried out in three 'Type A' and two 'Type B' laminate beam specimens, respectively. The remaining laminate beams without embedded sensors were solely used for mechanical test verification purposes.

Temperature logging conducted during the experiments served two purposes. First, the temperatures are used in equation (1) in order to compensate for the thermal expansion of the optical fibre and determine the total mechanical strains from the FBG data. The thermocouples are also used to monitor the exothermic reaction of the resin during curing and ensure that the specimens were cooled to the ambient temperature at the data sampling intervals. Figure 4 shows some of the specimens manufactured at various stages during the experiments.

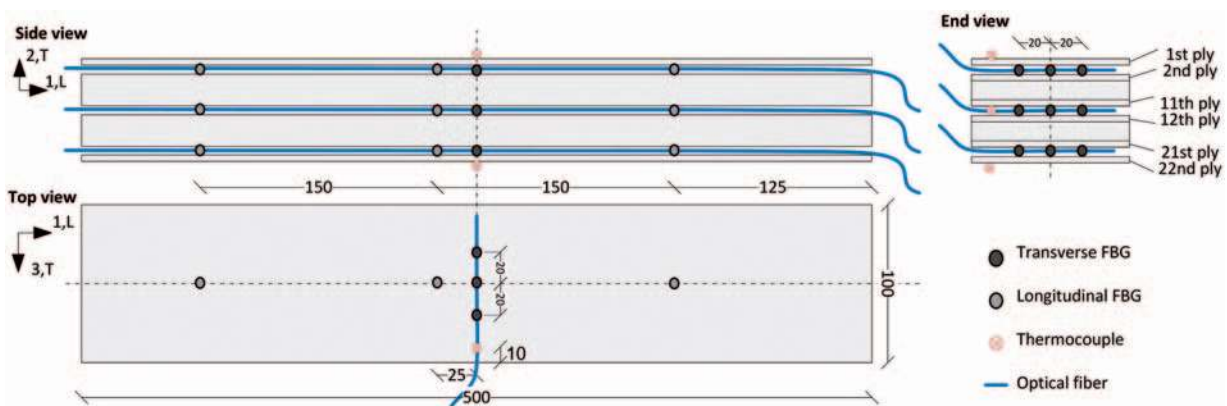


Figure 3. Schematic of laminate beam layup with FBG and thermocouple sensor placement. FBG: fibre Bragg grating. All dimensions are in mm.

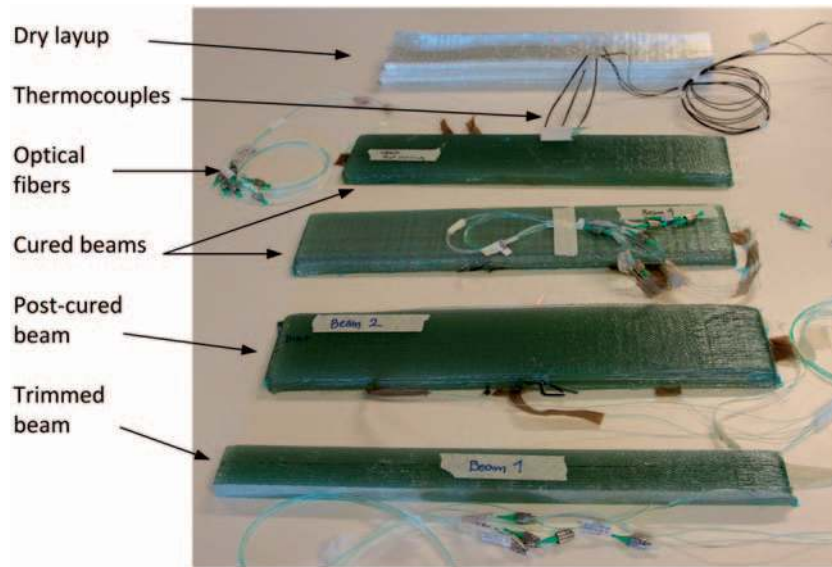


Figure 4. Laminate beam specimens at different stages in the manufacturing procedure, seen with embedded FBG sensors and thermocouples.

FBG: fibre Bragg grating.

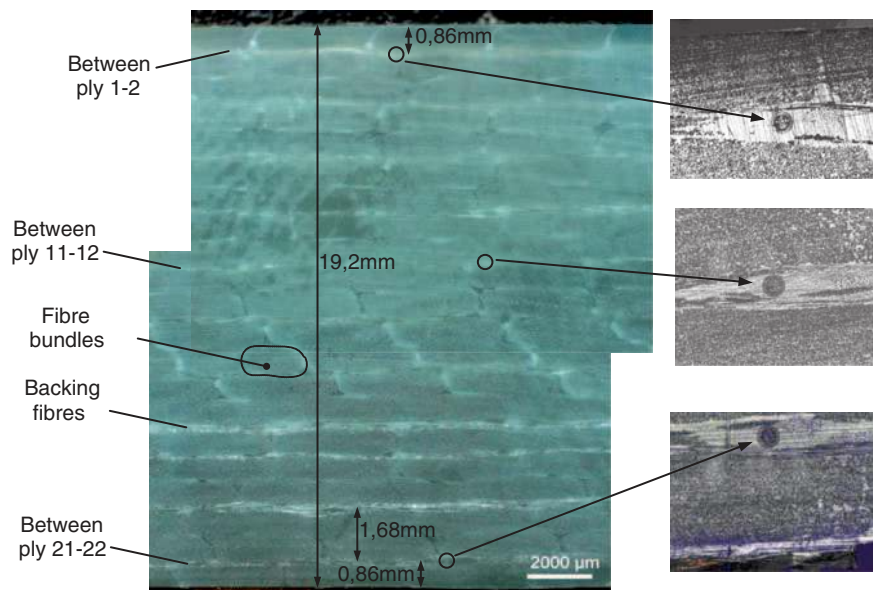


Figure 5. Image of laminate specimen cross section with close-up of inter-ply placement of optical fibres.

To ensure that the optical fibres remained at their respective laminate inter-ply locations during layup and vacuum infusion, a cross-sectional microscope analysis was conducted, after mechanical testing. Knowing the exact depth of each optical fibre is, for instance, vital for correct comparison of surface mounted SGs and embedded optical fibres. Figure 5 shows microscope images of the laminate beam cross section, showing placement of optical fibres after manufacturing. The optical fibres were found to be situated at the same inter-ply regions as during layup. It has been reported

in some studies that standard 125 µm optical fibres produce a insignificant perturbation of the host material when embedded parallel to the reinforcing fibres in laminates.^{25,26}

Experimental results and discussion

Strains at discrete periods during manufacturing

Figures 6 and 7 show measured total strains at discrete periods during manufacturing for type A and B

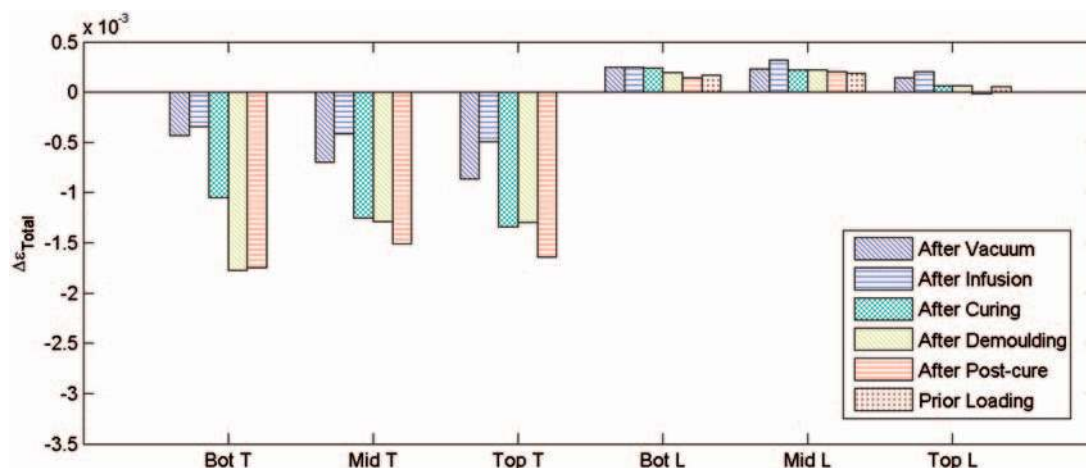


Figure 6. Average total strains at manufacturing intervals, measured at ambient temperature for type A laminate beam specimens.

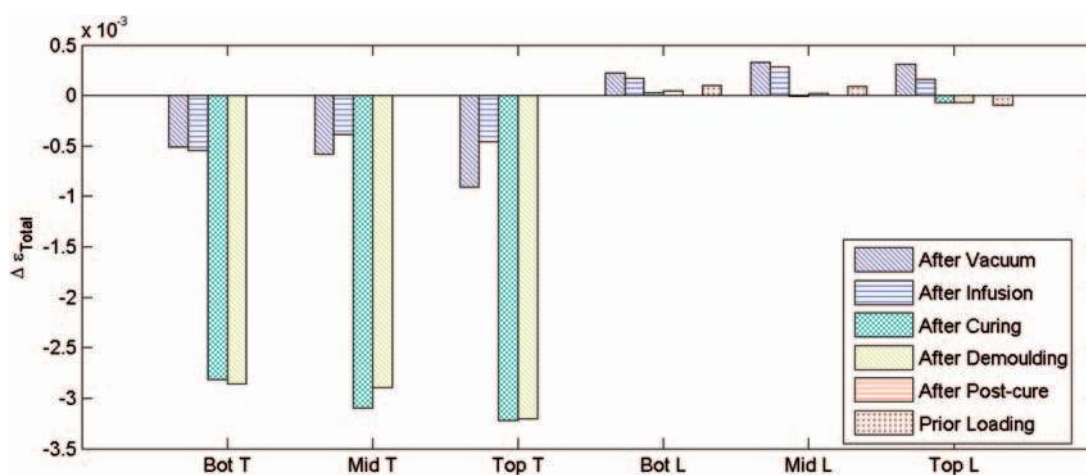


Figure 7. Average total strains at manufacturing intervals, measured at ambient temperature for type B laminate beam specimens.

laminate beams, respectively, at ambient temperature. The values presented are averages of type A and B specimens for all three FBG sensors on each optical fibre at discrete stages during manufacturing. Total strains are shown: (a) after vacuum pressure is applied; (b) after infusion (resin flow); (c) after curing; (d) after demoulding; (e) after post-curing (type A only); and finally (f) prior loading. Note that the transversely embedded optical fibres were no longer usable after cutting, hence no transverse measurements are conducted prior loading.

In both cure cycle cases shown (Figures 6 and 7), transverse total strains which are matrix dominated are seen to have higher magnitudes as compared to the longitudinal strains, which are fibre dominated, i.e. more constrained. General tendencies are as follows: upon applying vacuum pressure, a negative increment in strains are seen in the transverse direction, while a small positive increment in tensile strains arise

in the longitudinal direction (here onwards *T*- and *L*-directions). The elongation longitudinally is believed to be due to re-alignment of fibres from a wavy to straight state when under pressure effectively straining the optical fibre in tension. After infusion, a slight decrease in strain magnitudes is exhibited in general. After curing, a significant negative strain increment in the *T*-direction occurs in type A and B specimens throughout the beam thickness, as well as a slight negative strain increment in the *T*-direction. This is an indication of prevailing compression/shrinkage due to the inherent mechanisms present, i.e. matrix shrinkage and thermal compression upon cooling. A significant difference in strain magnitudes after curing is seen between type A and B specimens, with larger negative strains in type B specimens. Assuming that both specimens are fully cured such that total chemical shrinkage has taken place, the difference in magnitudes owes to larger thermally induced strains in type B beams. The through-thickness strain

distribution in both cases is largest near the top surface as compared to at the tool/part interface (Bot) where tool/part interaction constrains movement. Upon demoulding, a larger strain release is exhibited at Bot T in type A specimens, as opposed to Bot T in type B beams. This is an indication that better bonding at the tool/part interface existed during curing when release agent is used at the tool/part interface as opposed to when using the perforated release foil.

Figure 8 shows the total strains for type A and B beams in the L - and T -directions compared to each other. Results are shown for stages after curing, where it is assumed that perfect bonding between the composite and optical fibre exists. After curing, a non-uniform strain distribution through the thickness is seen in both L - and T -directions. After demoulding, negative strain increments are seen at the laminate bottom plane in the T -direction for both beam types (Figure 8 (T-2)), indicating a tensile stress state at the bottom laminate plane after curing. This implies that a compressive stress state exists at the laminate centre while tensile stresses exist at the surfaces, which is a

common stress state when an inside-out cure occurs due to exothermic peak temperatures at the centre, as also seen in Bogetti and Gillespie.³ Moreover, for type B specimens, this is also seen at the top plane, while at the middle plane a positive strain increment is seen. When considering the through-thickness strain gradients, a more uniform distribution is generally seen in type B specimens. This indicates that weaker bonding at the tool/part interface exists when using release film, allowing for more sliding friction behaviour. This observation is supported by the fact that all type B specimens were significantly easier to separate from the glass tool during demoulding.

When considering the free-standing change in type B longitudinal strains from demoulding and prior loading (Figure 8 (L-2 to L-4)), an increase in tensile strains is seen near the beam bottom plane. This is due to viscoelastic creep, exhibited by the composite during the 6 week period in time before loading tests were commenced. Contrary to this, a decrease in tensile strains (compressive creep) is seen for type A beams after post-curing and prior loading. Positive creep strains indicate

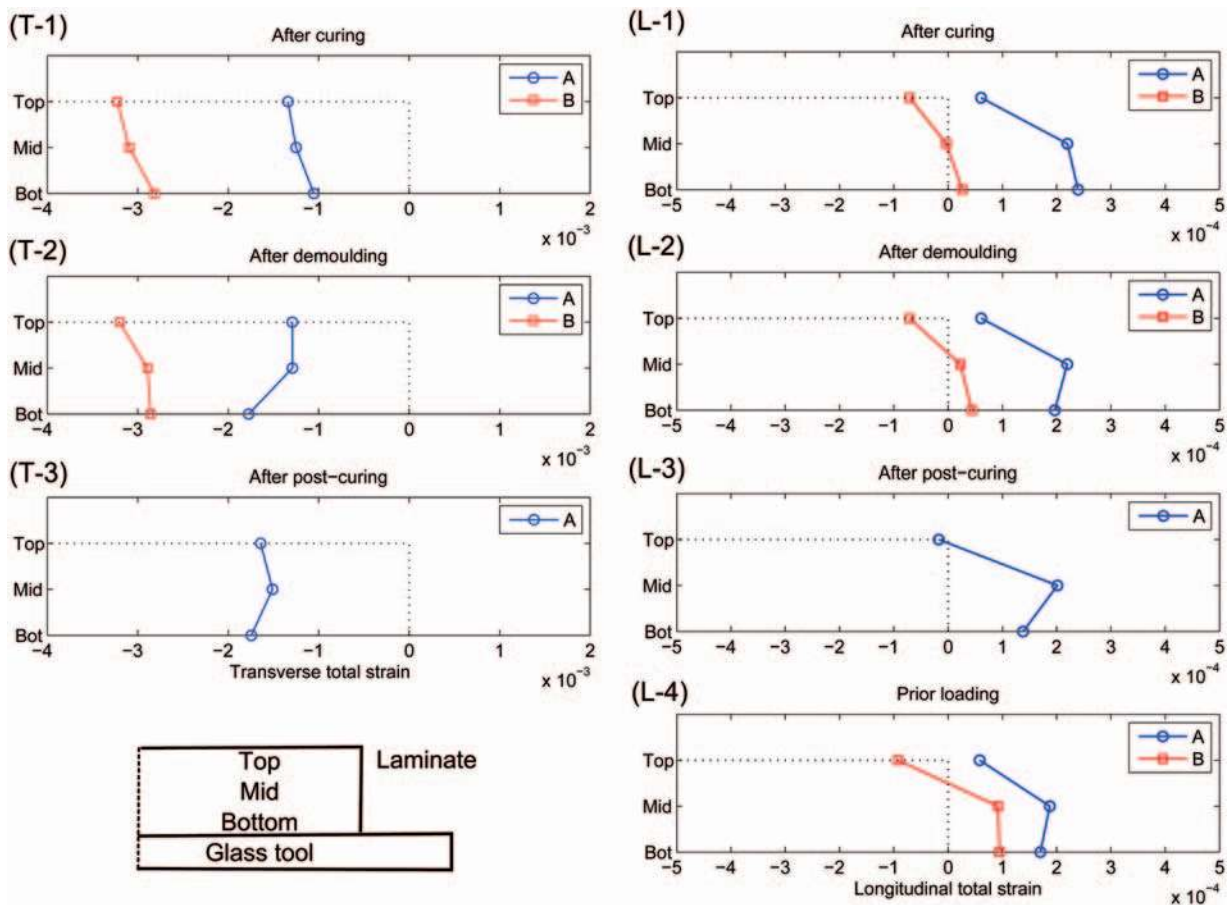


Figure 8. Average laminate type A and B transverse (T) and longitudinal (L) strains after various manufacturing stages compared to non-strained (dashed line) laminate beam state.

that a compressive stress state exists, that is not readily released as distortions after curing, but gradually with time. Nedele and Wisnom⁷ and Wisnom et al.¹¹ found that residual stresses at a micromechanic level may not normally readily cause distortions because they may arise at a very local scale, where any deforming effects are averaged out over the larger volume of material. Hence inter-ply or microscale compressive stresses may still exist in the beams.

Strain monitoring signal quality

The quality of the strain measurements was monitored during this study by constant evaluation of the reflected light signal spectrum. This is presented for a type B

laminates after curing at ambient temperature in Figure 9, compared to the reflected signal prior vacuum-bagging and infusion for longitudinally and transversely embedded FBG sensors. Due to the large compressive strains in the transverse direction after curing, the signal peaks have been shifted left to lower wavelength values. Furthermore, some distortion of the FBG spectrum is seen in the form of birefringence (peak splitting) of the Bragg peak. The peak-splitting seen indicates development of transverse loads at the vicinity of the sensors.^{13,18} This could be the result of backing-fibres or perpendicular UD fibre bundles that are locally loading the optical fibres not to mention resin shrinkage stresses around each optical fibre (Figure 10). Broadening of the spectrum is also

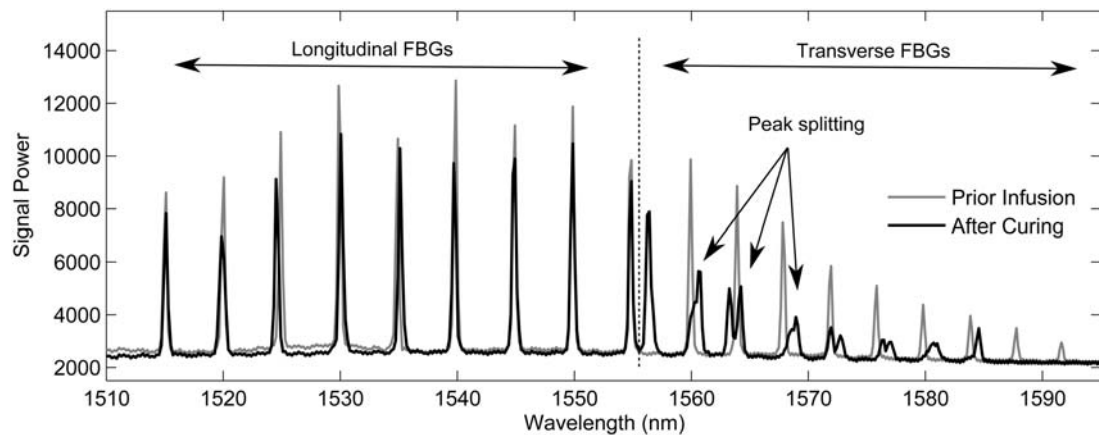


Figure 9. Wavelength signal spectrum for type B laminate after curing, showing clear signs of birefringence of the light signal (peak splitting).

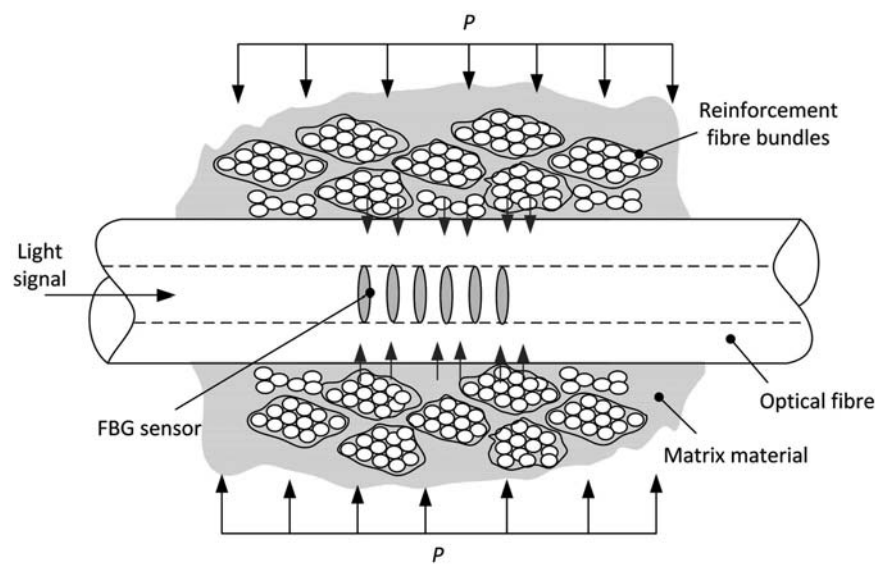


Figure 10. Illustration of mechanisms causing transverse loading of optical fibre from reinforcement fibre tows perpendicular to the optical fibre during processing.

observed, which is also an indication that transverse stress may exist.¹³ Distortion is not seen in any of the signals from longitudinal sensors, but only in Type B transverse signals. Acquiring the correct discrete wavelength shift in order to determine the mechanical strain variation must, therefore, be interpreted with caution where peak-splitting of the signal is present. The data acquisition software seeks only the signal peaks above a user-defined threshold found using an n th-order polynomial fit for a prescribed number of expected peak signals (here 18). In this study, in order to account for these signal distortions, an approach is used where averaging of multiple peaks within a common base is performed for the discrete ambient strains presented in the previous section. The spectrum data shows that larger stresses are generally present in Type B specimens as opposed to Type A.

In situ strains during infusion

Flow front monitoring is conducted during resin infusion by monitoring the shift in wavelength of the embedded FBG sensors during vacuum infusion. Figures 11 and 12 show the strain variations during infusion for a beam specimen representative of general tendencies occurring in the L - and T -directions, respectively.

After approximately 100 s, infusion is commenced and the resin flow front reaches the L-embedded sensors near the inlet first, seen as changes in sensor 3 at the top, mid and bottom (Figure 11). As the flow front progresses along the laminate top surface and gradually through the thickness towards the outlet, changes in wavelengths are seen in the L-embedded sensors and the corresponding T-embedded sensors (Figure 12). Hence, at different times during

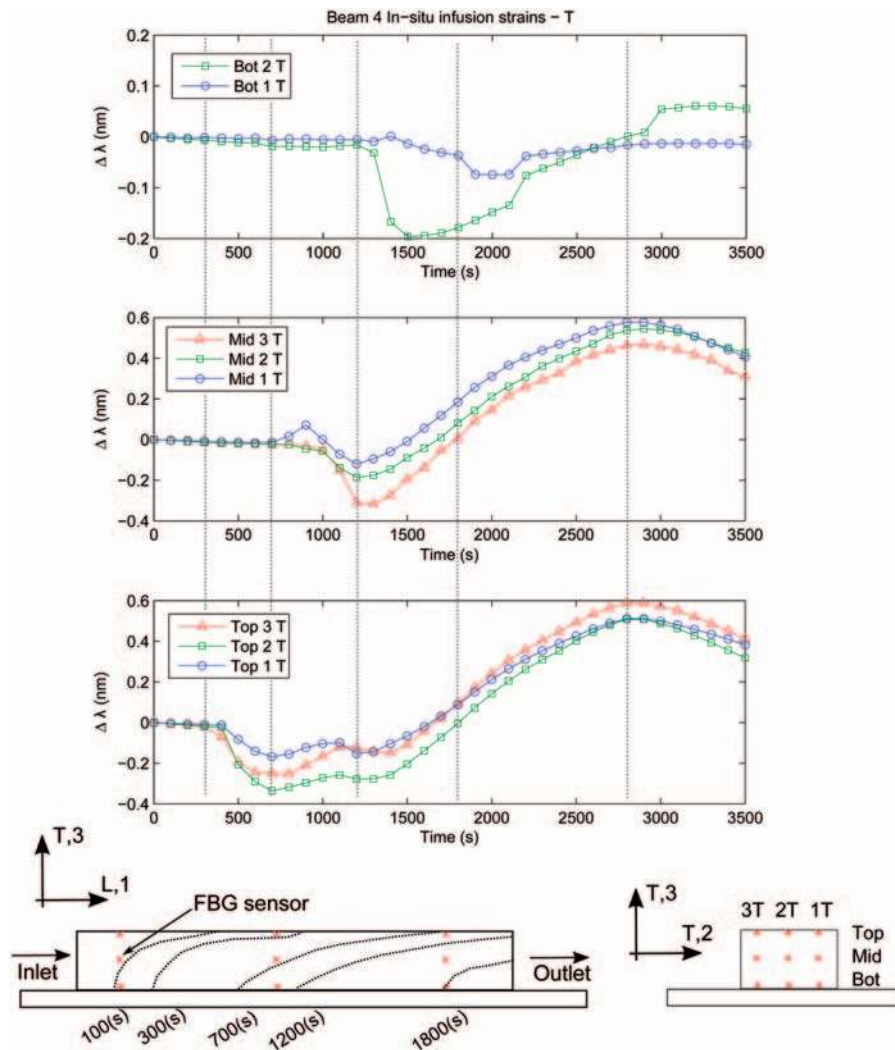


Figure 11. Variations in longitudinally embedded FBG sensor wavelengths during infusion. Also seen is a schematic of the sensor placement from inlet to outlet and the expected flow front development. FBG: fibre Bragg grating.

infusion, through-thickness tracking of the resin flow front in different directions is possible. After approximately time $t = 2800$ s, the inlet hose is closed, marked by changes in wavelength in all embedded sensors as the flow speed is drastically reduced.

In situ strains during curing

In Figure 13, experimentally measured temperatures for type A and B cure cycles are presented. A gap in the

acquired type A temperature data is seen between time $t = 180$ and 300 min. For the sake of continuity, estimated temperature development curves are drawn, following tendencies from type B data. In both cases, the exothermic reaction during resin cross-linking as the matrix cures results in peak temperatures that are higher than the prescribed oven cure temperatures of 50°C and 80°C for type A and B, respectively. The large increase in temperatures is accredited to high heat generation rates as the matrix cures. The peak exothermic

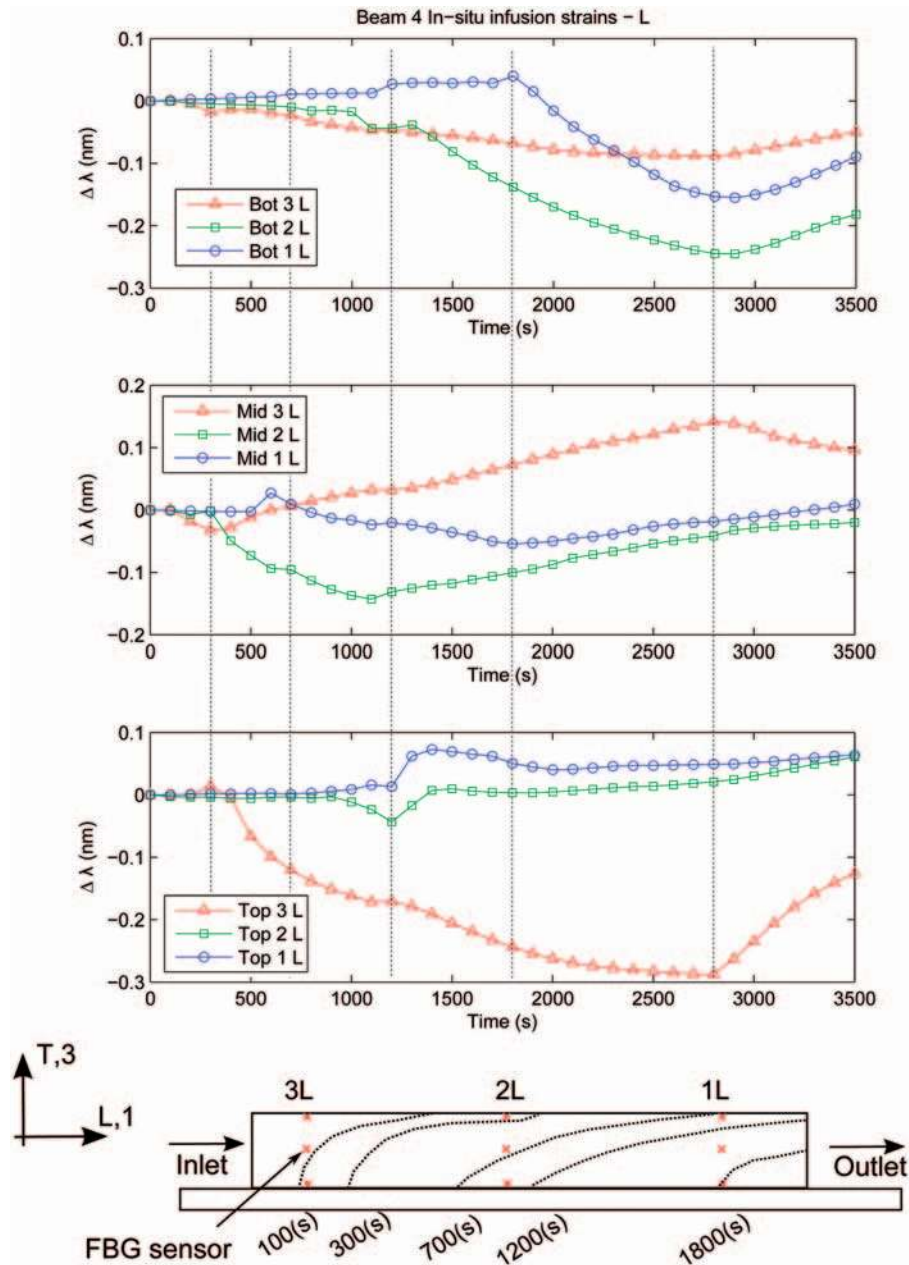


Figure 12. Variations in transversely embedded FBG sensor wavelengths during infusion.

Similarly, a schematic of the sensor placement is given and the expected flow front development, as shown in Figure 11. FBG: fibre Bragg grating.

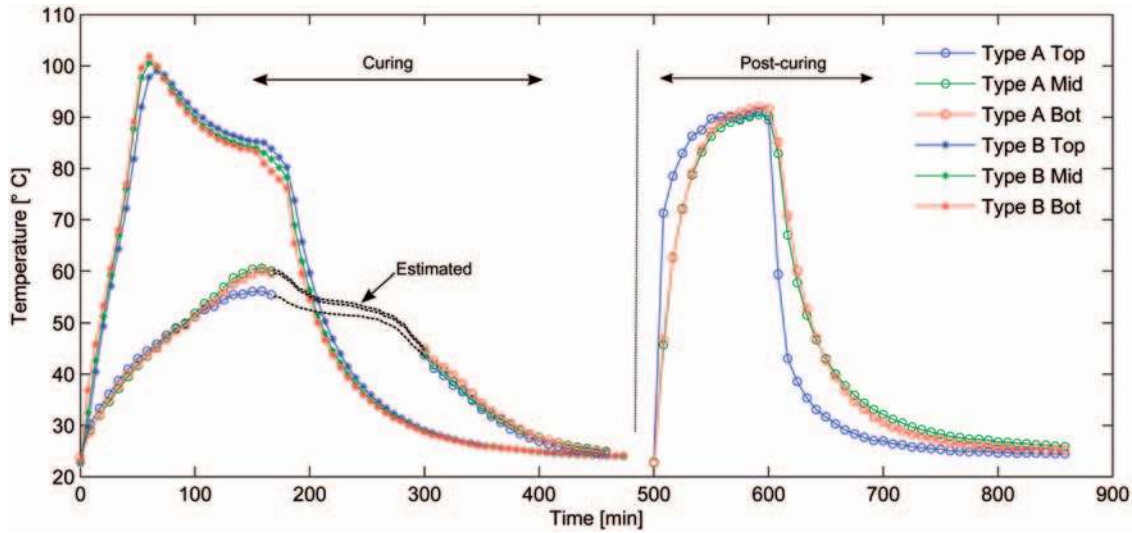


Figure 13. Temperature profiles for type A and B cure cycle laminate beam samples during curing. Note that only type A specimens are post-cured.

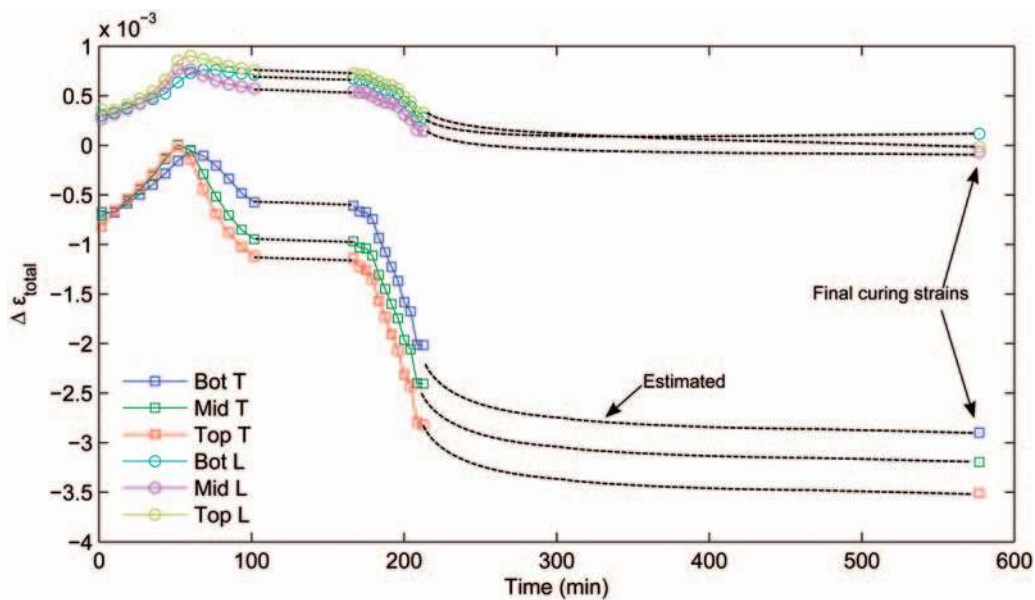


Figure 14. In situ total strain measurements during curing for type B specimen. Dotted lines are added for continuity where loss of signal occurred.

temperatures are approximately 60.1°C and 100.6°C during in mould cure, while the peak post-cure temperature is 98.3°C measured towards the end of the 90°C post-cure temperature hold for type A beam specimens. The peak temperatures are measured at the laminate mid-layer. The temperature peaks during post-curing is due to the release of residual latent heat which was not fully released during the first heating cycle.

In situ strain monitoring during curing is shown in Figure 14 for a type B beam using averaged values for

top, mid and bot (T and L) sensor measurements as calculated using equation (1). Where loss of signal from the interrogator unit was encountered, dotted lines are added for the sake of continuity based on the temperature profiles observed from Figure 13 during cooling. The in situ strain development generally follows the temperatures for type B specimens well (Figure 13) depicting an increase upon heating, a peak, dwell and decrease in strains during cooling. Early in the process, thermal expansion- and chemical cure shrinkage strains occur simultaneously, resulting

in negative strains after the exothermic reaction has occurred. This is mainly seen in the transverse (matrix driven) direction. In both the L - and T -directions, the higher mechanical constraints at the bottom layer, in both T - and L -directions, result in lower negative and positive strains, respectively, as compared to strain magnitudes at mid and top. Upon the onset of cooling after approximately 160 min, negative strain increments in the T - and L -directions are seen, owing primarily to thermal contraction, assuming the resin is fully cured and total chemical cure shrinkage has occurred. The strains after cooling are also shown, corresponding to the ‘after curing’ strains previously shown in Figures 6 and 7.

Simple analytic calculations based on the self-consistent field micromechanic model are made to validate the cure strain magnitudes found. The effective laminate longitudinal and transverse thermal strain increments can be calculated using³

$$\Delta\varepsilon_L^{th} = \left[\frac{\alpha_f E_f V_f + \alpha_m E_m (1 - V_f)}{E_f V_f + E_m (1 - V_f)} \right] \cdot \Delta T \quad (2)$$

$$\begin{aligned} \Delta\varepsilon_T^{th} = & \left\{ (\alpha_f + \nu_{12f}\alpha_f) V_f + (\alpha_m + \nu_m\alpha_m)(1 - V_f) \right. \\ & \left. - [\nu_{12f}V_f + \nu_m(1 - V_f)] \right. \\ & \left. \times \left[\frac{\alpha_f E_f V_f + \alpha_m E_m (1 - V_f)}{E_f V_f + E_m (1 - V_f)} \right] \right\} \cdot \Delta T \quad (3) \end{aligned}$$

where E , α and ν are the respective constituent modulus, coefficient of thermal expansion (CTE) and Poisson’s ratio and V_f the reinforcement fibre volume fraction. The subscripts f and m denote fibre and matrix specific quantities and ΔT the change in temperature. Assuming that the resin is in glassy state the instant the peak exothermic temperature is achieved, the temperature variation during the cooling phase (i.e. from the peak cure temperature to ambient) can be used to calculate the thermal strain increments. Results are summarised in Table 3.

The analytically determined transverse thermal strains coincide well with the experimentally determined FBG strain values, measured at the least constrained laminate layer, namely the top layer. A large discrepancy is seen when comparing the longitudinal strains, which could be due to the larger influence of the tool interaction along the length of the beam. It is believed that the poor agreement in analytical and experimental strains in the longitudinal fibre direction may generally owe to the positive pre-straining of the fibres in this direction once the vacuum pressure was applied and during infusion (Figures 6 and 7). Note that the analytical calculations do not take into account gelation, differential resin material CTE in rubbery and

Table 3. Analytical thermal strain increment during cooling compared to measured total strains at the laminate beam top plane.

	Type A	Type B
ΔT (°C)	−38.6	−79.6
$\Delta\varepsilon_L^{th}$	−0.281E−3	−0.579E−3
$\Delta\varepsilon_T^{th}$	−1.80E−3	−3.60E−3
$\Delta\varepsilon_L^{tot} - \text{FBG}$	−0.06E−3	−0.001E−3
$\Delta\varepsilon_T^{tot} - \text{FBG}$	−1.34E−3	−3.50E−3

FBG: fibre Bragg grating.

glassy state or chemical shrinkage strains. If one wishes to accurately predict these effects, numerical process models could be used.

Flexural tests – Three-point bending

Although large process-induced strains were seen to develop in the T -direction during curing, only small strains were measured in the L -direction. Hence, large residual stresses in the L -direction may still exist that are not released into distortions upon demoulding or relaxed during post-curing. At a micromechanical level, resin cure shrinkage and contraction upon cooling result in a tensile stress state in the L -direction in the matrix, due to constraints posed by the reinforcement fibres. In turn, the reinforcement fibres are in a compressive stress state upholding equilibrium. Whether these residual stresses influence the composite mechanical behaviour during loading is investigated using simple flexure tests. Figure 15 shows a schematic of the flexure test beam samples and sensor placement.

The tests were conducted in a custom made three-point test rig with adjustable rolling cylindrical supports (Figure 15), using an MTS 810 axial servo-hydraulic table top test, with a ± 180 mm stroke and force capacity of 25 kN. The loading and support nose cylinder diameters are 40 and 25 mm, respectively. A large nose radius was used to avoid excessive indentations and failure due to stress concentrations directly under the loading point. The test rig is connected to the load cell and hydraulic piston by serrated grips. The piston is controlled through the MTS FlexTest system by a servo valve model 525-15A-04 with a maximum flow of 10 L/s and load cell model 662.10A-05. A cross-head rate of 0.5 mm/min is used and tests are conducted until a substantial drop in the specimen load-carrying capacity is exhibited. The strain at the top and bottom surface of the test specimen is acquired from four SG located at the top and bottom surface along the same beam cross section as the embedded FBG sensors (Figure 15). Strains are measured along the laminate

laminate/loading nose contact point. The LVDT and DIC displacements correspond well throughout the loading range, which indicates that no significant deformations occur in the test rig itself. FBG strains measured at the laminate top centre (and bottom centre for validation) are converted to vertical displacements using the following geometric relations

$$\delta = \rho[1 - \cos(\theta)]$$

where

$$\rho = \frac{h/2}{\Delta\varepsilon_{\{FBG\}}} \text{ and } \theta = \sin^{-1}\left(\frac{L/2}{\rho}\right) \quad (4)$$

In equation (4) δ is the vertical displacement, ρ the chord length and θ is the angle of curvature. A good agreement between the FBG, DIC and LVDT displacements is seen in the early stages of loading, albeit gradually diverging. The FBG displacements increase linearly with the load until a threshold strain is reached (ε_{FBG} in Figure 16) after which the signal becomes sporadic and inconsistent, marking sensor failure. The strain is also seen to be lower in type B beams due to the higher initial process-induced strains which lower

the fracture strain of the optical fibre. The strain range of the FBG is given by the supplier to be approximately 5000 $\mu\varepsilon$, which both type A and B specimens in Figure 16 lie below, more so for type B beams. This may indicate that the slight compressive pre-straining of the FBG sensors in the L -direction during processing (see Figure 8 (L-4)) decreases the strain to failure of the sensors.

Flexural test results from all beam specimens were collected using load–DIC displacement data. From the test results, the flexural modulus E_f of each beam is found using the slope of the load–displacement curves along the linear regime as²⁷

$$E_f = \frac{S^3 m}{4bh^3} \quad (5)$$

where S is the support span, b the width and h the thickness of the beams. The flexural strength of the respective specimens is the stress at the surface upon failure, calculated assuming a linear stress–strain relationship up to failure as²⁷

$$\sigma_{cr} = \frac{3P_{cr}S}{2bh^2} \quad (6)$$

Table 4. Flexural test modulus, failure load and flexural strength for all tested specimens.

	Type A					Type B				
	A 1	A 2	A3	A 4	Average	B 1	B 2	B 3	B4	Average
h (mm)	19.10	18.56	19.17	19.10	18.98	19.56	19.45	19.05	19.42	19.37
E_f (GPa)	39.26	40.60	39.36	38.06	39.32	37.86	38.45	38.65	38.70	38.23
P_{cr} (kN)	17.583	16.593	17.198	15.992	16.84	18.312	17.156	17.468	17.930	17.72
σ_{cr} (MPa)	708.04	722.55	701.89	657.54	697.53	717.92	680.24	721.88	712.92	708.32

Respective beam specimen thickness (h) measured at the laminate centre, are also seen.

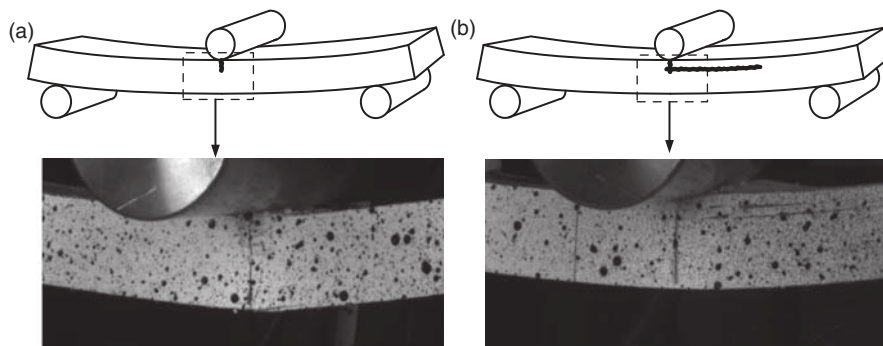


Figure 17. Images capturing failure progression using DIC during fracture. Failure starts for most specimens as compressive fracture at the top laminate layer (a), and later develops to compressive fracture with interlaminar shear (b). DIC: digital image correlation.

where P_{cr} is the load at the moment of fracture. The experimentally determined flexural modulus and failure load is presented for all specimens in Table 4 as well as the flexural strength of each specimen, determined using equation 6. From Table 4, beam type A and B average failure loads are 16.84 and 17.72 kN, respectively, resulting in average flexural strength values of 697.5 and 708.3 MPa, respectively. As these results highly depend on the thickness of the specimens (as h is squared in equation (6)) substitution of P_{cr} in equation (6) with normalised failure loads for 20 mm thick beam specimens is conducted resulting in average flexural strength values of 771.1 and 716.5 MPa for type A and B specimens, respectively. Note that in Table 4, no distinct differences are seen between beam specimens with embedded optical fibres (i.e. A1, A2, B1 and B2) and without.

DIC was also utilised to monitor the onset of failure (Figure 17). For all 10 beam specimens, failure modes were first visible after the maximum load was reached. From the DIC images, the first failure modes seen for a majority of the specimens was compressive fracture at the top surface, followed second by a number of specimens exhibiting compressive fracture with interlaminar shear. No distinct failure mode type accredited to only type A or B specimens was visible.

Conclusion

A life cycle approach where strain monitoring using embedded FBG sensors in E-glass/epoxy composite laminate beam specimens from manufacturing to failure in three-point bending was conducted. The embedded FBG sensors successfully monitored resin flow front progression during infusion and internal strain development during curing as well as at subsequent intervals until loading to failure. Process-induced strains were seen to reflect the differences in processing temperatures and tool/part interface interactions well. Laminates cured at 80°C for 3 h induced larger internal strains than those cured at 50°C for 5 h. It was shown that process-strains in the transverse reinforcement fibre direction were large enough to result in a decrease in FBG signal quality due to peak splitting and signal distortion for specimens cured at 80°C. No such signal distortions were seen in specimens cured at 50°C for 5 h with subsequent post-curing at 90°C for 2 h. Flexural tests were conducted showing no significant effect of the process-induced stresses and strains on the longitudinal flexural stiffness or failure strength of the specimens. This could owe to the simplicity of the analysed UD laminate beam specimens, which readily releases most residual stresses due to lack of significant internal or external constraints. Furthermore, it can be questioned whether small specimens, similar to the beams used in

this study, are representative measures for the behaviour of large FRP structures. In such structures, thicker laminate sections and complex geometries incorporating internal constraints could play a role, not to mention the likelihood of including manufacturing defects such as voids, dry spots or matrix cracks to name a few. However, a difference in the failure strain of the embedded FBG sensors was seen for specimens cured at 50°C and post-cured at 90°C as compared to specimens cured at 80°C with the latter resulting in lower sensor failure strains. Hence, if embedded FBG sensors are to be used for in-service strain monitoring of composite structures where large deformations are expected, process-induced pre-straining of the FBG sensors should be taken into account.

The manufacturing and loading strain monitoring conducted in this study can be utilised to validate an integrated numerical process and loading model where the effects of process-induced residual stresses on the composite response under loading and failure strength can be analysed. Moreover, use of embedded FBG sensors offers the ability to measure and track part consistency during processing. The procedure presented in this article is applicable for strain monitoring of composite structures from manufacture to in-service use.

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Conflict of interest

None declared.

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