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**LIGHT-RESPONSIVE SELF-ASSEMBLED MATERIALS BY
SUPRAMOLECULAR POST-FUNCTIONALIZATION VIA
HYDROGEN BONDING OF AMPHIPHILIC BLOCK
COPOLYMERS**

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3 ABSTRACT
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7 A new class of light-responsive supramolecular amphiphilic block copolymers (BCs) based on
8 the association through multiple H-bonding between 4-isobutyloxyazobenzene motifs and 2,6-
9 diacylaminepyridine units is reported. Block copolymers containing 2,6-diacylaminopyridine
10 side units, as hydrophobic block, and poly(ethylene glycol), as a hydrophilic segment, were
11 functionalized with either a carboxylic acid azodendron, *via* double H-bonding, or a thymine
12 azobenzene, *via* triple H-bonding. The structural and thermal characterization of these
13 supramolecular azo-copolymers in bulk and solution is presented. The work emphasizes the self-
14 assembly of these supramolecular polymers in water and the study of their UV-light responsive
15 properties by UV-vis spectroscopy, dynamic light scattering (DLS), transmission electron
16 microscopy (TEM), and synchrotron small-angle X-ray (SAXS). The present noncovalent post-
17 polymerization functionalization strategy has provided stable self-assemblies in water with light
18 responsive properties that can be used to load and trigger the delivery of small fluorescent
19 molecules.
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3 INTRODUCTION
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6 The design of efficient drug delivery vehicles remains a true challenge in polymer and materials
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8 science. A variety of polymeric nanocarriers have been described in recent years based on the
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10 ability of amphiphilic block copolymers (BCs) to spontaneously generate assemblies in water,
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12 such as micelles or vesicles, whose size and morphology are highly dependent on the
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14 composition, molecular geometry and relative block lengths of the BCs.¹⁻⁵ The incorporation of
15
16 pH-, temperature- or light-responsive moieties into these amphiphilic BCs makes them potential
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18 candidates as systems for controlled release.⁶⁻⁸ The use of light as an external stimulus is
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20 particularly advantageous as provides with temporal and spatial control in the material
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22 response.⁹⁻¹² Among other photocromic moieties, azobenzene presents rapid, reversible, and
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24 high quantum yield photoisomerization, and is undoubtedly the most widely investigated group
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26 in the design of light-responsive systems based on amphiphilic BCs.¹³
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33 In previous contributions, we have reported on photoresponsive vesicles based on amphiphilic
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35 linear-dendritic block copolymers, LDBC, with hydrophilic/hydrophobic weight ratios of
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37 approx. 20/80, and bearing 4-cyanoazobenzene moieties.¹⁴ Theoretical simulations on these
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39 systems predicted the disruption of macromolecular aggregates by photoisomerization that could
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41 ultimately trigger the release of encapsulated molecules.¹⁵ Alternatively, we investigated the use
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43 of 4-alkoxyazobenzene photoresponsive units that facilitate the disruption of the assemblies
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45 under UV irradiation.¹⁶ The resulting vesicles were loaded with both hydrophobic and
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47 hydrophilic fluorescent probes to demonstrate that the distortion of the bilayer membrane under
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49 low intensity UV irradiation increases its permeability to the encapsulated molecules.^{17, 18} Those
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51 light-responsive vesicles were based on covalent polymeric systems, and the time-consuming
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53 synthetic procedures involved reduce their practical feasibility in controlled drug release.^{19, 20}
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3 The introduction of supramolecular polymers by Kato and Fréchet,²¹ and Lehn and
4 coworkers,²² broadened the playground for polymer chemists, which is no longer restricted to
5 macromolecular species based on the repetition of monomeric units governed by covalent
6 bonding. Supramolecular polymeric materials combine many of the attractive features of
7 conventional covalent polymers with the properties resulting from the reversibility of
8 noncovalent interactions.²³⁻²⁶ In this context, several works have reported on the preparation and
9 application of noncovalent azobenzene-containing polymers. Ionic interactions were used by
10 Bazuin and coworkers for grafting azobenzene derivatives to methylated poly(4-vinylpyridine),
11 yielding liquid crystalline polyelectrolytes²⁷, and by Marcos and coworkers for the preparation of
12 azobenzene containing poly(propyleneimine) codendrimers by simple acid-base titration.²⁸⁻³⁰
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28 Alternatively, hydrogen bonding combines ease of preparation with high specificity and
29 directionality between components. Priimagi and coworkers pioneered the research on hydrogen
30 bonded supramolecular side chain azopolymers by complexing poly(4-vinylpyridine) with a
31 variety of azobenzene molecules.³¹⁻³⁵ This strategy can be extended to build supramolecular
32 block copolymers (BCs) yielding materials with multiple-level hierarchical self-assembly. Thus,
33 we investigated a series of photo-responsive supramolecular polymers prepared by mixing
34 carboxylic terminated promesogenic azobenzene derivatives with commercially available
35 poly(styrene)-*block*-poly(4-vinylpyridine) copolymers.^{36,37} In these systems, the azo
36 chromophore was linked to the polymer *via* hydrogen bonding through molecular recognition
37 between the carboxylic groups (H-donor) and the pyridine rings (H-acceptor). However, even
38 though the materials exhibited good photoresponsive properties, high levels of complexation led
39 to some degree of macroscopic segregation between the components (*i.e.*, for 100% substitution
40 of the pyridine pendant units). In an attempt to overcome such limitation and prepare
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3 quantitative complexated supramolecular BCs, we have used functional groups capable of
4 forming multiple hydrogen bonds. With that aim, we have reported on diblock copolymers
5 having a poly(methyl methacrylate) block bearing 2,6-diacylaminopyridine pendant moieties,
6 which were fully complexated with 4-cyanoazobenzene derivatives to obtain homogeneous
7 materials.³⁸
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16 Recently, we have described the preparation and self-assembly of analogous amphiphilic BCs
17 and described their potential as polymeric-based nanocarriers for hydrophobic drugs.³⁹ These
18 amphiphilic BCs contained a polymethacrylate block bearing pendant 2,6-diacylaminopyridine
19 units (**PDAP**) and a poly(ethylene glycol) block (**PEG**) of two different average molar masses,
20 2000 and 10000 g mol⁻¹ (**PEG₂-b-PDAP** and **PEG₁₀-b-PDAP**, respectively) (**Figure 1a**).
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In the search of highly functionalized high-responsive nanocarriers, and encouraged by the ability of the 2,6-diacylaminopyridine moiety to form multiple hydrogen bonds, here we report on new amphiphilic supramolecular BCs obtained by complexation of **PEG₂-b-PDAP** and **PEG₁₀-b-PDAP** with two 4-isobutyloxyazo derivatives, namely **dAZO_i** and **tAZO_i**. **dAZO_i** is a dendron that contains three peripheral 4-isobutyloxyazobenzene moieties and a carboxylic acid focal point, while **tAZO_i** contains a thymine head linked to one 4-isobutyloxyazobenzene unit (**Figure 1b**). It is envisaged that the 2,6-diacylaminopyridine pendant groups will form either two (**dAZO_i**) or three (**tAZO_i**) hydrogen bonds (**Figure 1c**) with the azobenzene derivatives.

The present work describes the preparation, characterization and thermal properties of these new light-responsive, amphiphilic supramolecular polymers. Since these materials are targeted as controlled release systems, we also present a detailed characterization of the corresponding assemblies in water and their light-responsive behavior by transmission electron microscopy (TEM), dynamic light scattering (DLS) and synchrotron small-angle X-ray (SAXS), as well as encapsulation-release studies using fluorescent probes.

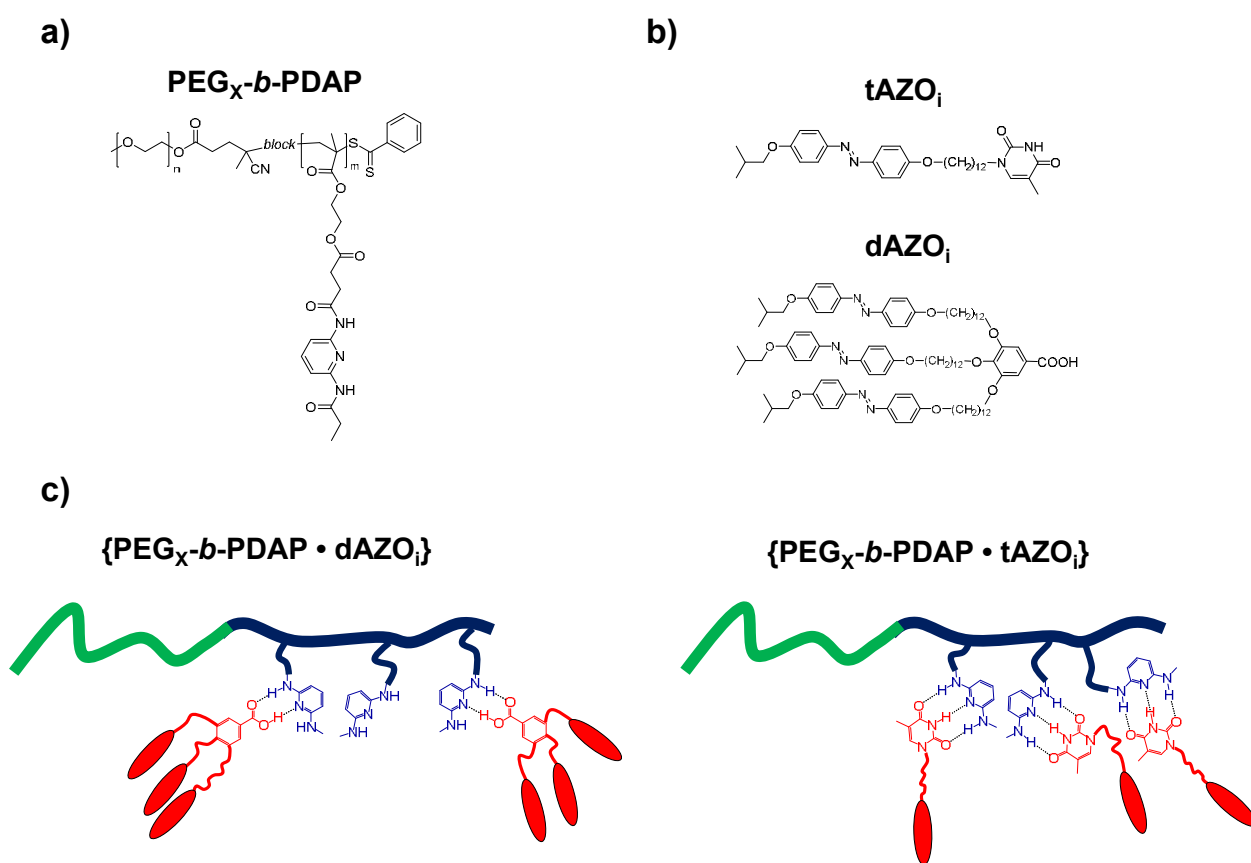


Figure 1. Chemical structure of (a) PEG_x-b-PDAP block polymers, (b) tAZO_i and dAZO_i compounds and, schematic representation of (c) azobenzene-containing supramolecular BCs.

EXPERIMENTAL SECTION

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3 **Materials.** The preparation of the **PDAP** homopolymer and **PEG_x-*b*-PDAP** amphiphilic
4 block copolymers have been reported elsewhere,^{38, 39} and details of the synthesis and
5
6 characterization of **tAZO_i** and **dAZO_i** can be found in the Supporting Information.
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11 **Preparation of supramolecular polymers.** Required amounts of the polymer and the
12 corresponding azocompounds were weighted and dissolved in THF followed by slow
13 evaporation under stirring at room temperature. The supramolecular polymers were dried under
14 vacuum at 40°C for at least 2 days. For characterization in the bulk, polymers were first heated
15 up to 175°C for 5 min and rapidly quenched to room temperature to have a controlled thermal
16 history.
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21 **Preparation of self-assemblies in water.** Milli-Q water was gradually added to a solution of
22 5 mg mL⁻¹ of the corresponding BC in tetrahydrofuran, THF, and the self-assembly process was
23 followed by measuring the loss of intensity of transmitted light at 650 nm due to scattering
24 (turbidimetry) as a function of the water content. At a critical water value, a sudden increase of
25 turbidity occurs coinciding with the onset of polymer self-assembly. When a constant value of
26 turbidity was reached, the mixture was dialyzed against water to remove THF using a
27 Spectra/Por[®] dialysis membrane (MWCO 1000) for 4 days. Water suspensions of the polymeric
28 self-assemblies were obtained with concentrations around 1.7 mg mL⁻¹.
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47 **Determination of the Critical Aggregation Concentration (CAC).** Critical aggregation
48 concentration (CAC) was determined by fluorescence spectroscopy using Nile Red as the probe
49 as follows. 119 μL of a solution of Nile Red in dichloromethane (5×10⁻⁶ M) was added into a
50 series of flasks and then the solvent evaporated. Afterwards, water suspensions of the self-
51 assemblies, prepared by diluting the former 1.7 mg mL⁻¹ suspension, were added to each flask
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3 with concentrations ranging from 1.0×10^{-4} to 1.0 mg mL^{-1} . In each flask a final concentration of
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5 1.0×10^{-6} M of Nile Red was reached and the resulting suspensions were stirred overnight at
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7 room temperature to reach equilibrium before fluorescence was measured. The emission spectra
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9 of Nile Red were registered from 560 to 700 nm while exciting at 550 nm.
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14 **Loading of Rhodamine B into the vesicle.** Loaded vesicles were formed following the same
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16 procedure described for the preparation of self-assemblies but using a solution of Rhodamine B
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18 in Milli-Q water. The BC was dissolved in THF and the aqueous solution of Rhodamine B was
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20 gradually added. The charge ratio was 1:5 mol copolymer/mol dye. Self-assembly was
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22 monitored by turbidimetry until an almost constant value was reached. Finally, the mixture was
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24 dialyzed against water to remove THF and non-encapsulated Rhodamine B, and the absorbance
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26 of the resulting solution was measured. The absorption values were compared to a calibration
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28 curve of Rhodamine B, previously obtained, in order to determine the quantity of non-
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30 encapsulated dye molecules and therefore the number of molecules encapsulated in the vesicle.
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37 **Irradiation experiments.** The water dispersions of the self-assemblies were irradiated with a
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39 compact low-pressure fluorescent lamp Philips PL-S 9W emitting between 350 and 400 nm. The
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41 samples were placed at a distance of 10 cm from the light source in quartz cuvettes at room
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43 temperature. After irradiation, the water suspensions were kept in the dark.
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47 **Techniques.** Fourier transform infrared spectroscopy (FTIR) was applied using a Bruker
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49 Vertex 70 FT-IR spectrophotometer and KBr pellets or films deposited by casting over BaF_2 .
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51 Solution NMR experiments were carried out on Bruker Avance spectrometers operating at 400
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53 or 300 MHz for ^1H , and 100 or 75 MHz for ^{13}C , using standard pulse sequences. Chemical shifts
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55 are given in ppm relative to TMS and the solvent residual peak was used as internal reference.
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3 Thermogravimetric analysis (TGA) were performed using a Q5000IR module from TA
4 instruments at a heating rate of 10 °C min⁻¹ under a nitrogen atmosphere. Thermal transitions
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6 were determined by differential scanning calorimetry (DSC) using a Q2000 calorimeter from TA
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8 instruments on powdered samples (2–5 mg) sealed in aluminum pans. Glass transition
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10 temperatures (T_g) were determined at the half height of the baseline jump, and first order
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12 transition temperatures were read at the maximum of the corresponding peak for the polymers
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14 and at the onset for low molar mass compounds. The homogeneity of the samples was inspected
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16 by polarizing optical microscopy (POM) using an Olympus BH-2 polarizing microscope fitted
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18 with a Linkam THMS600 hot stage. UV-Vis absorption spectra were recorded in an ATI-
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20 Unicam UV4-200 spectrophotometer. Fluorescence measurements were performed using a
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22 Perkin Elmer LS 50B fluorescence spectrophotometer. Dynamic light scattering (DLS)
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24 measurements were carried out in a Malvern Instrument Nano ZS using a He–Ne laser with a
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26 633 nm wavelength, a detector angle of 173° at 25 °C using a He–Ne laser with a 633 nm
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28 wavelength. The self-assemblies concentrations were 0.075 mg mL⁻¹ and size measurements
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30 were performed at least three times on each sample to ensure reproducibility.
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40 **Transmission electron microscopy (TEM) and cryoscopic transmission electron**
41 **microscopy (Cryo-TEM).** The morphology of the block copolymer vesicles was studied by
42 transmission electron microscopy (TEM) using JEOL-2000 FXIII and TECNAI G² 20 (FEI
43 COMPANY) electron microscopes operating at 200 kV. For the preparation of TEM samples, 5
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45 μL of a 0.5 mg mL⁻¹ water dispersions of the self-assemblies were deposited onto carbon-coated
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47 copper grid, and the water was then removed by capillarity using filter paper. The samples were
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49 stained with uranyl acetate and the grid was left to dry overnight under vacuum. The 0.5 mg mL⁻¹
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51 suspension was prepared by diluting the former 1.7 mg mL⁻¹ suspension with Milli-Q water.
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3 Cryo-TEM experiments were performed using a JEM-2011 electron microscope, and samples for
4 inspection were prepared by casting 5 μL of a 1.7 mg mL^{-1} water dispersion of self-assemblies
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8 on a suitable grid and then quenched in liquid ethane.
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11 **Confocal Microscopy.** Samples were observed with an Olympus FV10i confocal scanning
12 microscope. Images were collected using a 60 \times oil immersion lens (lens specification, Plan S-
13 APO 60 \times O, NA 1.35), a line average of 8 and a format of 1024 \times 1024 pixels. The confocal
14 pinhole was 1 Airy unit. Samples preparation consisted on casting 5 μL of a 1.7 mg mL^{-1} water
15 dispersions of self-assemblies with encapsulated Rhodamine B on a glass slide, followed by
16 covering with a slip on the top. The edges were sealed to avoid solvent evaporation during
17 measurement.
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29 **Small Angle X-Ray Scattering (SAXS).** X-ray scattering techniques provide information
30 about the structure and molecular conformations at different length scales. SAXS is particularly
31 suitable when studying supramolecular organization in the length scale of tens of nanometers,⁴⁰
32 and in combination with synchrotron radiation allows monitoring changes in real time in both
33 conformation and structure.⁴¹ Experiments were performed at beamline BL11-NCD (ALBA,
34 Spain) using a wavelength of 0.1 nm. The SAXS detector (ADSC, Quantum 210r CCD, pixel
35 size 102 μm) was located at 6430 mm distance from the sample position. Experiments under UV
36 irradiation were performed at room temperature, and SAXS patterns were collected while the
37 sample was irradiated with a UV lamp (Phillips PL-S 9W) placed at a distance of 10 cm. The
38 angular (q -axis) was calibrated using standard samples of Silver Behenate.
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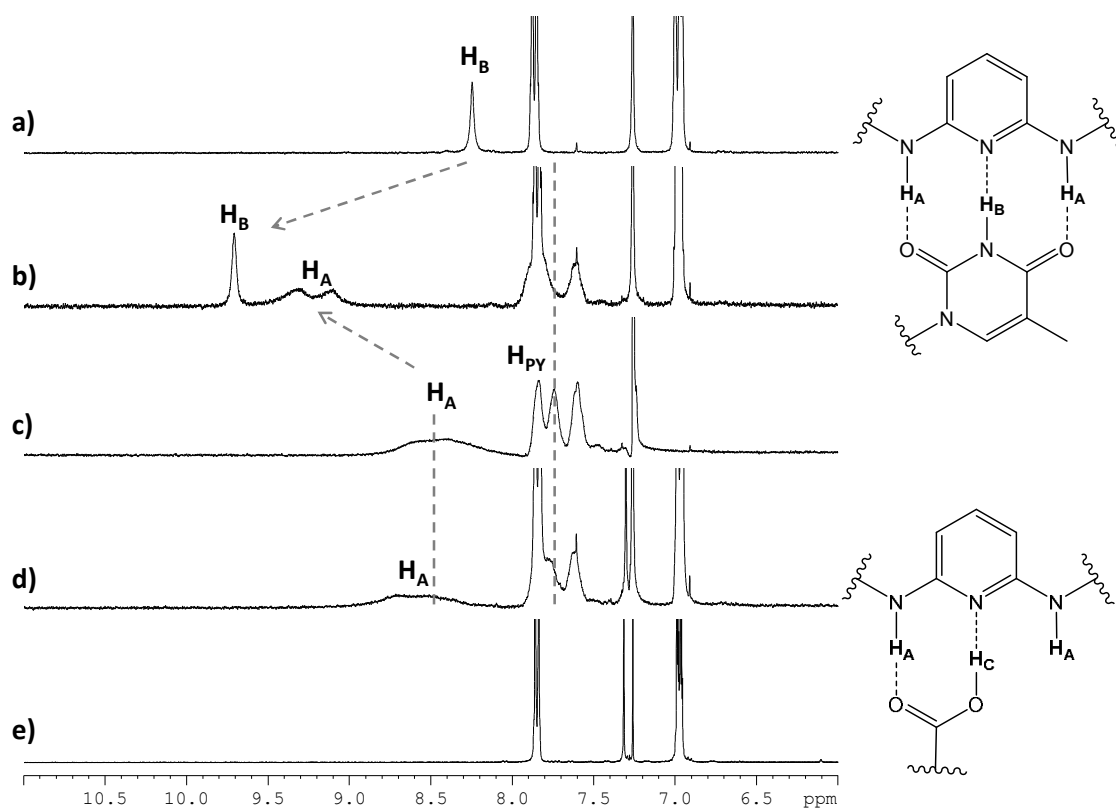
57 RESULTS AND DISCUSSION
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3 **Preparation and characterization of supramolecular polymers.** The supramolecular block
4 copolymers were prepared by **dissolving** corresponding amounts of the parent BCs (**PEG_x-b-**
5 **PDAP**) and the azocompounds (**dAZO_i** or **tAZO_i**) in THF, followed by slow evaporation of the
6 solvent under continuous stirring at room temperature. According to our previous results on
7 covalent LDBC, the degree of complexation (number of azo molecules per 2,6-
8 diacylamino-pyridine repeating unit) was adjusted to achieve hydrophilic/hydrophobic weight
9 ratios of approx. 20:80 in order to predominantly yield vesicles.¹⁴ Therefore, a degree of
10 complexation of 1 was used with **tAZO_i**, while 0.30 was used with **dAZO_i** (see **Table 1**). For
11 comparative purposes, fully complexed supramolecular polymers (degree of complexation = 1)
12 of the homopolymer **PDAP**³⁸ with either **dAZO_i** or **tAZO_i** were also prepared, namely
13 **{PDAP•dAZO_i}** and **{PDAP•tAZO_i}**.
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30 The ¹H NMR spectra of the supramolecular BCs in CDCl₃ proved the formation of hydrogen
31 bonded complexes, assuming that there is a rapid equilibrium between the complex and its
32 components. In general, proton signals involved in the hydrogen bonds, as well as those that are
33 close to the complexing groups, experienced correlated changes in their chemical shifts. As a
34 representative example, the ¹H NMR spectra of the **{PEG₂-b-PDAP•tAZO_i}** and **{PEG₂-b-**
35 **PDAP•dAZO_i}** supramolecular polymers are shown in **Figure 2**, along with those of their
36 respective components.
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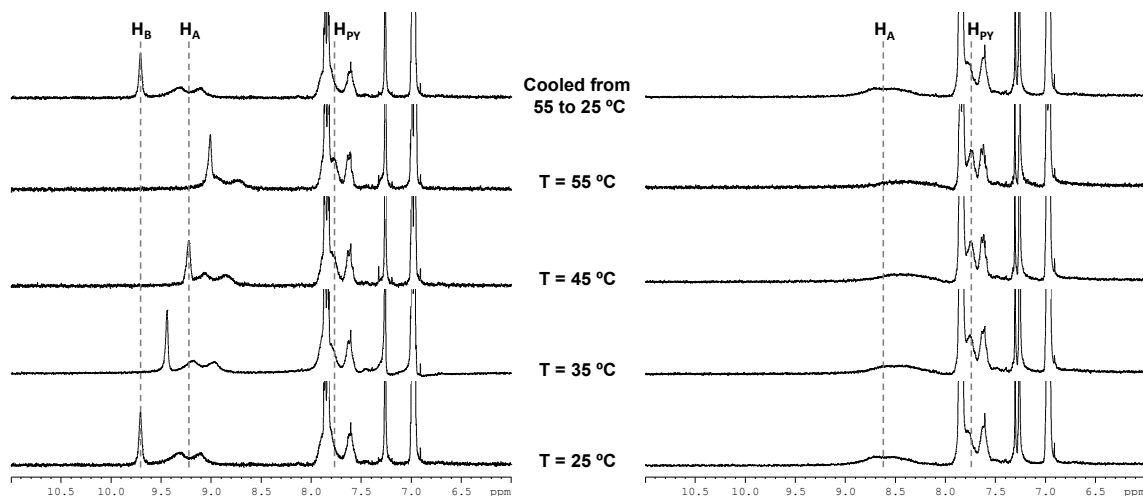
48 The formation of hydrogen bonds between complementary **tAZO_i** and 2,6-
49 diacylamino-pyridine units in **{PEG₂-b-PDAP•tAZO_i}** was assessed by the simultaneous
50 downfield shifts of the N–H signals of the PDAP block (*H_A*, from 8.45 to 9.23 ppm) and the
51 thymine unit (*H_B*, from 8.25 to 9.71 ppm). In the case of supramolecular BCs containing
52 **dAZO_i**, the 2,6-diacylamino-pyridine N–H protons (*H_A*) experienced slight downfield
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3 displacements (from 8.45 to 8.58 ppm) also attributed to hydrogen bonding interactions with the
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5 **dAZO_i** carboxylic acid. Such small shift is explained by the low complexation degree of these
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7 supramolecular BCs, see **Table 1**. The acidic proton signal (H_C) was very broad and was not
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9 visible in the ¹H NMR spectrum. It is particularly noteworthy that in both systems the protons
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11 close to the hydrogen bonds experienced slight displacements. For example, protons of the
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13 pyridine ring of the 2,6-diacylaminopyridine units (H_{PY}) are shifted by around +0.05 ppm.
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15 NOESY was also applied to further evaluate hydrogen bonding in these materials. The ¹H-¹H
16
17 NOESY spectrum of {PEG₂-*b*-PDAP•tAZO_i} is shown in **Figure S6**, and depicts significant
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19 cross-peaks between H_A and H_B , thus indicating that these groups are close in space because of
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21 H-bonding interactions. However, NOESY experiments of BCs with **dAZO_i** did not show any
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23 additional information.
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3 **Figure 2.** ^1H NMR spectra of (a) tAZO_i , (b) $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$, (c) $\text{PEG}_2\text{-}b\text{-PDAP}$, (d)
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5 $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{dAZO}_i\}$, and (e) dAZO_i , in CDCl_3 solution at 25°C .
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10 Temperature dependent NMR experiments can also provide with valuable information about
11 the strength of hydrogen bonding, and therefore the ^1H NMR spectra of the supramolecular BCs
12 were also recorded in solution in the 25 to 55°C range (**Figure 3**). On heating the tAZO_i
13 complexes, the H_A and H_B signals moved steadily from 9.23 and 9.71 to 8.95 and 9.01 ppm,
14 respectively, while for the dAZO_i complexes, the H_A signal shifted from 8.61 to 8.36 ppm.
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16 When the solutions were cooled from 55 to 25°C , all displaced signals went back to their
17 original chemical shifts. These gradual displacements are attributed to dissociation and recovery
18 of the hydrogen bonds between complementary units on heating and cooling, respectively, and
19 highlight their thermoreversible character.⁴²
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49 **Figure 3.** ^1H NMR spectra in CDCl_3 of $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ (left) and $\{\text{PEG}_2\text{-}b\text{-}$
50 $\text{PDAP}\cdot\text{dAZO}_i\}$ (right) taken at different temperatures (bottom to top: spectra registered on
51 heating and after cooling down again to room temperature).
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3 An overview of the IR response of {PDAP•tAZO_i} and {PDAP•dAZO_i}, in the complete
4 frequency range, is shown in **Figure S7** (Supporting Information). The FTIR spectra of the
5 complexes display several changes respect to the azocompounds and polymer precursors due to
6 the formation of specific interactions between the PDAP units and the azocompounds. In the
7 case of supramolecular polymers containing tAZO_i (**Figure S8a**), the significant modifications
8 of the stretching vibration modes of the carbonyl and amide groups, C=O_{st} and N-H_{st}, as well as
9 of the bending bands associated to the amide groups, N-H_δ, are additional signatures of
10 hydrogen bonding between the 2,6-diacylaminopyridine cores and the thymine units. The IR
11 profiles do not vary to a great extent with temperature, as a consequence of the stability of these
12 interactions in the bulk, and only slight variations are observed in the C=O_{st} and N-H_δ regions,
13 (see **Figure S9a**). The IR C=O_{st} region of the dAZO_i complexes (**Figure S8b**), is dominated by
14 two main bands at around 1690 cm⁻¹ (attributed to hydrogen bonded acid groups of dAZO_i and
15 to amide groups of 2,6-diacylaminopyridine) and 1730 cm⁻¹ (assigned to ester and free dAZO_i
16 acid groups). On heating above the melting temperature (**Figure S9b**), the 1690 cm⁻¹ region
17 undergoes a relative decrease while the 1730 cm⁻¹ band broadens, and this is accompanied with a
18 noticeable decrease of the N-H_{st} signals (below 3400 cm⁻¹). These simultaneous variations
19 indicate partial breakage of hydrogen bonds between the acid groups of dAZO_i and the amide
20 groups of the 2,6-diacylaminopyridine cores. The previous changes are reversed on cooling, and
21 the behavior is reproducible on further heating and cooling cycles, in consistency with the NMR
22 results shown above.
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51 **Thermal Characterization of Precursors and Supramolecular BCs.** The azocompound
52 precursors and derived supramolecular polymers showed good thermal stability, as determined
53 by TGA, with onset temperatures (T_{onset}) associated to mass loss above 200°C (**Table 1**).
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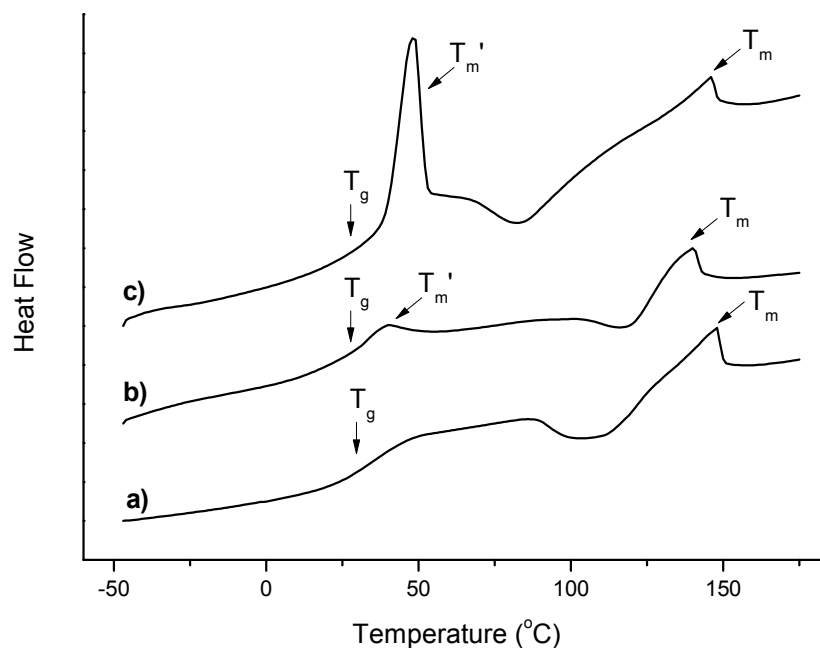
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3 Evolution of volatiles due to the presence of residual solvents or water was not observed. The
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5 phase behavior of the building components and supramolecular polymers was evaluated by DSC
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7 (Table 1). Properties of the PDAP homopolymer, the PEG macro-CTAs, PEG_x-CTA, and the
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9 PEG_x-*b*-PDAP BCs were previously reported³⁹ and relevant data are included in Table 1 as
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11 reference. The PEG_x-*b*-PDAP BCs are amorphous and only exhibit a single glass transition (no
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13 melting transitions were observed) indicating compatibility between blocks. The low molar
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15 mass azocompound tAZO_i is a crystalline material that melts to give an isotropic liquid phase at
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17 166°C, while dAZO_i exhibits semicrystalline behavior, showing a glass transition at $T_g=68$ °C
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19 and a melting point at $T_m=141$ °C (see Figure S10 in Supporting Information).

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22 All the supramolecular polymers appear homogeneous under POM, with no signs of
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24 segregation between the individual components upon subsequent heating and cooling cycles,
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26 corroborating the formation of single-like materials.

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Complexation promotes notable variations in the thermal properties of the tAZO_i-containing
supramolecular polymers compared to their precursors (Figure 4). The heating curve of
{PDAP•tAZO_i} is shown as a reference in Figure 4(a), and displays a T_g at around 35°C
followed by cold crystallization and subsequent melting. {PEG_x-*b*-PDAP•tAZO_i} polymers
also depict the previous transitions related to the crystalline complexed PDAP block,
accompanied by melting of the PEG block at 40–48 °C, see Figure 4(b) and 4(c). The
appearance of several thermal events ascribable to the different blocks suggests that
complexation induces microphase separation by decreasing the miscibility between blocks.

On the other hand, the {PDAP•dAZO_i} homopolymer is a semicrystalline material with $T_g=$
77°C and $T_m=136$ °C, (Figure 5, trace a). The corresponding supramolecular BCs, {PEG_x-*b*-

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3 **PDAP•dAZO_i**}, show an additional glass transition at lower temperatures in the 4-15°C range,
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5 together with a weak thermal event related to melting of the PEG blocks at 40-52°C (see inset in
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7 **Figure 5**). These results seem to confirm that complexation promotes microphase separation
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9 similarly to the **tAZO_i** containing BCs. The additional glass transition observed at lower
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11 temperatures ($T_g \sim 4-15^\circ\text{C}$) can be attributed to the presence of uncomplexed sample, **{PEG_x-b-**
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13 **PDAP}**}, caused by the low complexation degree yielded in these **dAZO_i**-based materials (0.30).
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45 **Figure 4.** DSC traces of (a) **{PDAP•tAZO_i}**, (b) **{PEG₂-b-PDAP•tAZO_i}** and (c) **{PEG₁₀-b-**
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47 **PDAP•tAZO_i}** corresponding to the second heating scans (10 °C min⁻¹, Exo down)
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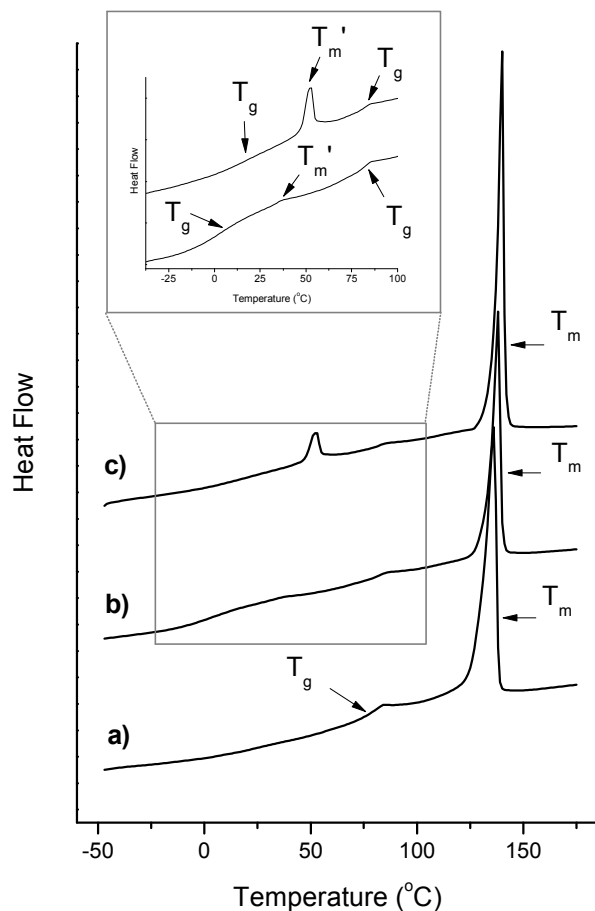


Figure 5. DSC traces of (a) {PDAP•dAZO_i}, (b) {PEG₂-*b*-PDAP•dAZO_i} and (c) {PEG₁₀-*b*-PDAP•dAZO_i} corresponding to the second heating scans (10 °C min⁻¹, Exo down)

Self-Assembly of the supramolecular BCs in water. We have recently reported that the uncomplexed copolymers under study form spherical micellar self-assemblies with a diameter of approx. 21 nm, for PEG₂-*b*-PDAP, and 32 nm, for PEG₁₀-*b*-PDAP, as determined by DLS (18 and 25 nm as determined by TEM).³⁹ We now investigate the self-assembled structures of the supramolecular BCs, which were prepared by the co-solvent method using THF/water, and whose formation was monitored by turbidity (see **Figure S11** in Supporting Information).

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3 While **dAZO_i**-containing supramolecular BCs precipitated during dialysis, indicating the
4 collapse of the assemblies in water, we obtained stable dispersions of **{PEG₂-*b*-PDAP•tAZO_i}**
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6 and **{PEG₁₀-*b*-PDAP•tAZO_i}**
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11 Rotello and co-workers showed that the efficiency of H-bonding post-functionalization on
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13 polymers containing pendant nucleobase analogues rely on the balance between intramolecular
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15 polymer-polymer binding and intermolecular polymer-guest binding.^{43, 44} Therefore, in solution
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17 recognition will depend on the tendency of the pendant diacylaminopyridines to self-
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19 dimerization compared to the binding affinity of the complementary pair diacylaminopyridine-
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21 carboxylic acid or diacylaminopyridine-thymine. However, in DAP polymers, the DAP-DAP
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23 association is low and hardly competitive for instance in the DAP-thymine recognition
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25 process.^{44, 45}
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31 Carboxylic acid-diacylaminopyridine recognition was used by Kato to prepare side chain
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33 liquid crystalline polymers demonstrating the feasibility of the functionalization.⁴⁶ However,
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35 even if it is stabilized by the liquid crystal state, this interaction is probably not strong enough in
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37 aqueous solution (water is a competitive solvent) to form stable self-assemblies. Among other
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39 factors, the strength of the association increases with the number of hydrogen bonds²³ so the use
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41 of thymine instead of the carboxylic acid renders stable polymeric nanoparticles. To confirm
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43 this point and discard a steric effect on the dendron complexation, a new acarboxylic acid
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45 analogue of tAZO, i.e. with one azobenzene (**acAZO_i**), was synthesized (see Supporting
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47 Information). Two new supramolecular complexes with **PEG₁₀-*b*-PDAP** with degrees of
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49 complexation of 0.30 and 1 were prepared but with none of them stable self-assemblies were
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51 obtained. This finding stresses our hypothesis that the stability of self-assemblies is influence by
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53 the H-bond strength more than by the steric hindrance imposed by the dendron.
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3 Thus, all further studies were performed for {**PEG₂-*b*-PDAP•tAZO_i**} and {**PEG₁₀-*b*-**
4 **PDAP•tAZO_i**}. The CAC of these supramolecular BCs in water was determined by
5 fluorescence, using Nile Red (**Figure S12** in Supporting Information). Values of 28 and 42
6 $\mu\text{g/mL}$ were obtained for {**PEG₂-*b*-PDAP•tAZO_i**} and {**PEG₁₀-*b*-PDAP•tAZO_i**}, respectively,
7 as the threshold concentration of the amphiphilic polymer above which the polymer chains start
8 to associate. These values correlate with their thermodynamic stability and are typical for
9 amphiphilic BCs.⁴⁷

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21 The stability of the self-assemblies suspensions was followed by temperature dependent DLS
22 experiments. The resulting self-assemblies were stable over 8 days at 25 °C, after 30 days not
23 precipitation was observed but polydispersity was high to obtain a stable measurement (see
24 **Figure S13** in Supporting Information). DLS experiments were also recorded at 25, 35, 45, and
25 55°C, which were the temperatures selected for NMR experiments, and self assemblies were
26 stable at 25 and 35°C but not at 45 and 55°C. Reformation of the self assemblies was not
27 observed upon cooling.

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38 The morphology of the self-assemblies was firstly investigated by TEM. It was found that
39 {**PEG₂-*b*-PDAP•tAZO_i**} self-assembled into vesicles with a deflated appearance, which it was
40 attributed to water removal during preparation for sample inspection (**Figure 6a**). The aqueous
41 suspension of the vesicles was then inspected by cryo-TEM, by quenching the sample into liquid
42 ethane at -170 °C, showing spherical vesicles having a membrane thickness around 10 nm
43 (**Figure 6b**). The mean average hydrodynamic diameter (D_h) of these vesicles evaluated by DLS
44 was 345 nm (**Figure 7**), which is in good agreement with cryo-TEM observations. It is worth
45 emphasizing that the parent **PEG₂-*b*-PDAP** BC formed spherical micelles and upon
46 functionalization *via* H-bonding with **tAZO_i**, vesicles are formed.

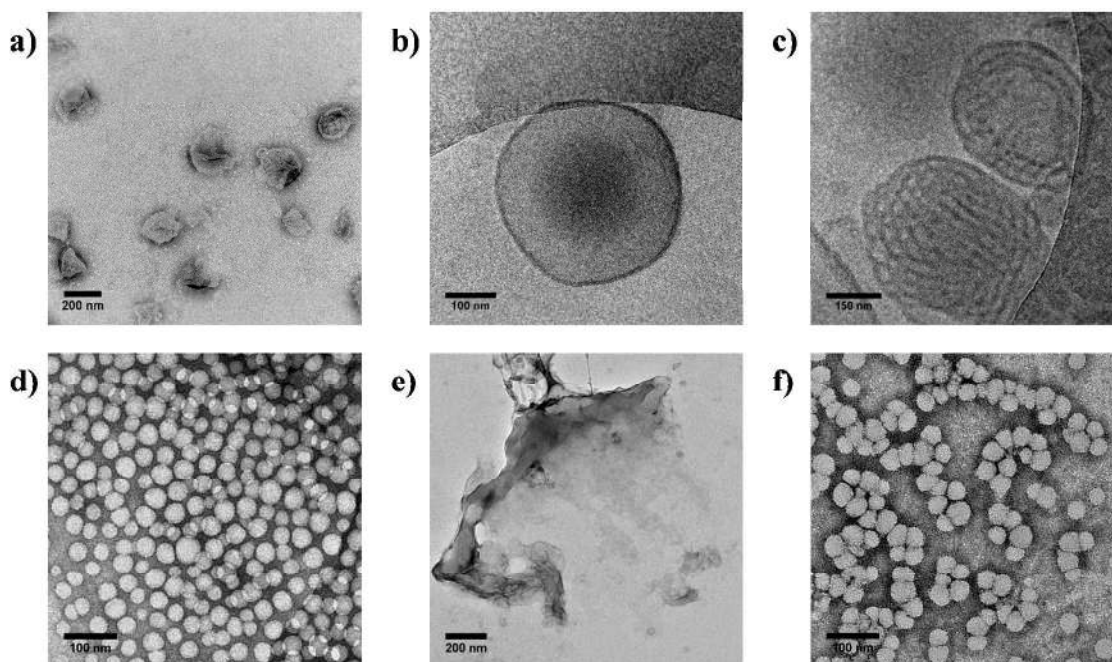


Figure 6. (a) TEM image of $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ non-irradiated vesicles. Cryo-TEM images of $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ vesicles (b) before and (c) after UV irradiation. TEM images of $\{\text{PEG}_{10}\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ micelles (d) before, (e) after irradiation and (f) 24h after irradiation.

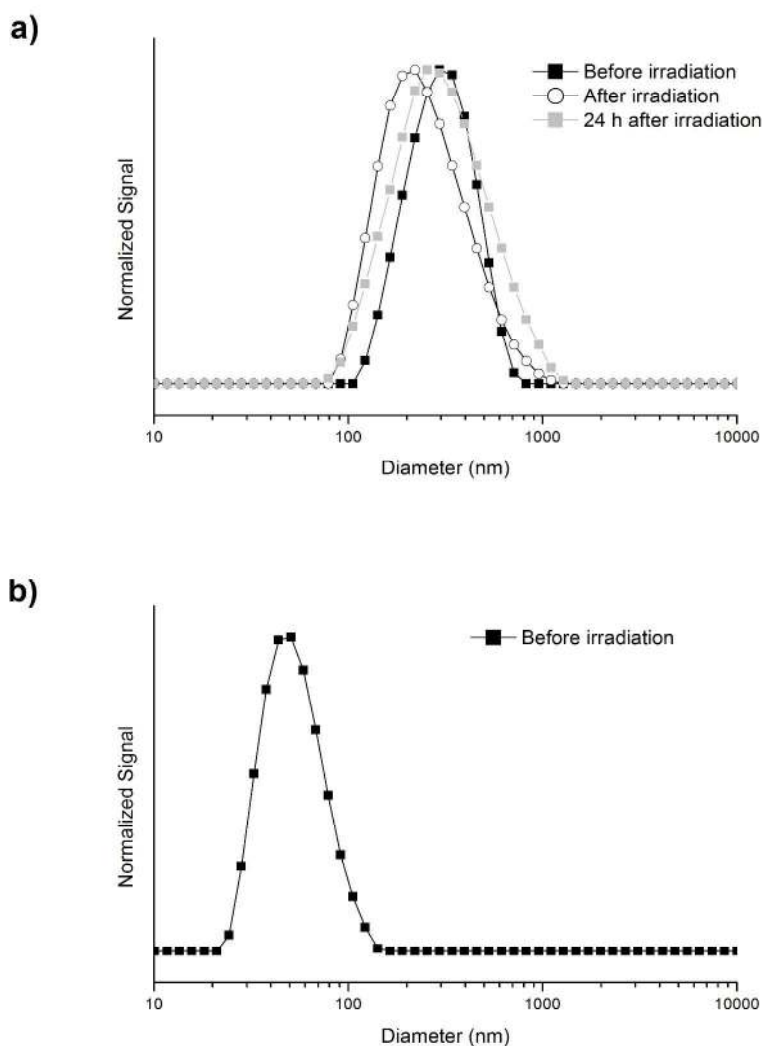


Figure 7. DLS measurements of a water suspension of (a) $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ vesicles before and after irradiation for 10 min and (b) $\{\text{PEG}_{10}\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ non-irradiated micelles (after irradiation, accurate DLS measurements were not possible due to a very high polydispersity).

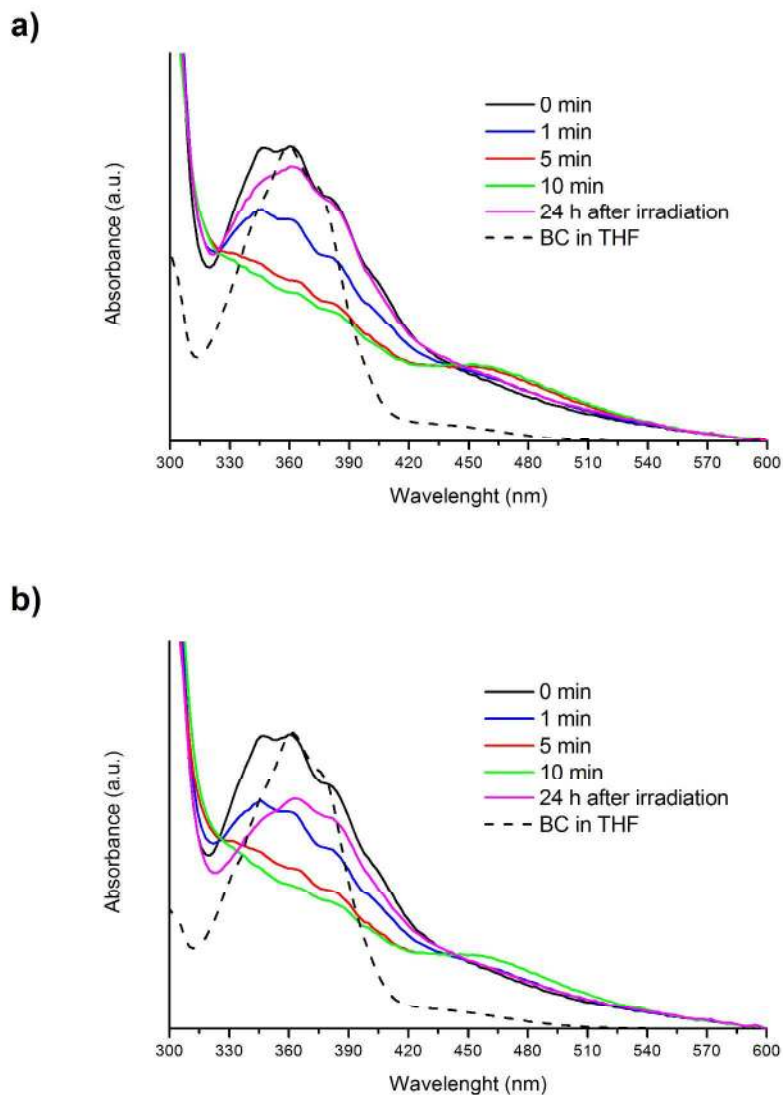
TEM images of $\{\text{PEG}_{10}\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ show the formation of spherical micelles with a diameter of approx. 40 nm (Figure 6d). The average D_h determined by DLS was 51 nm (Figure 7). Therefore, and unlike its analogue with the shorter PEG block, complexation of $\text{PEG}_{10}\text{-}b\text{-}$

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3 **PDAP** with **tAZO_i** does not promote a change of the self-assembled structures from spherical
4 micelles to vesicles, albeit larger micelles (20 nm increase on diameter) are observed respect to
5 the parent BC. The previous results can be rationalized in terms of the macromolecular structure
6 and the hydrophobic/hydrophilic balance. In **{PEG₂-*b*-PDAP•tAZO_i}**, the lower
7 hydrophilic/hydrophobic weight balance, which change from approx. 35/65 to 18/82 upon
8 complexation, seems to stabilize the vesicular morphology, while the higher molecular weight of
9 the hydrophilic blocks in **{PEG₁₀-*b*-PDAP•tAZO_i}** allows to retain the spherical micellar
10 morphology under the present experimental conditions.
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23 **Light-responsive behavior of the supramolecular BCs.** The photoresponse of **{PEG₂-*b*-**
24 **PDAP•tAZO_i}** and **{PEG₁₀-*b*-PDAP•tAZO_i}** was evaluated by UV-vis spectroscopy. The
25 spectra of the amphiphilic BCs in THF solution showed the characteristic profile of the *E*-isomer
26 of the azobenzene chromophore. This includes an intense band at 362 nm, related to the π - π^*
27 transition, together with a weak absorption band at about 450 nm, corresponding to the symmetry
28 forbidden n - π^* transition of the *E*-azobenzene. The spectra of micelles and vesicles in water
29 evidenced broadening and hypsochromic shifting of the π - π^* band (**Figure 8**) compared to the
30 results of the supramolecular BCs in THF. The absorption band shifted down to 348 nm, which
31 indicates the predominant formation of azobenzene H-aggregates. Furthermore, two additional
32 contributions at higher wavelengths were observed: one at 362 nm, corresponding to the value
33 determined for the non-aggregated *E*-azobenzene detected in solution, and other at 378 nm,
34 characteristic of J-aggregates.
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52 The sensitivity of the self-assemblies to UV-light was assessed by UV illuminating the
53 respective suspensions and simultaneously recording changes in their UV-vis spectra (**Figure 8**).
54 A remarkable decrease in π - π^* absorbance was observed accompanied by a notable increase of
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3 the absorbance at 450 nm due to the photoinduced *E*-to-*Z*-azobenzene isomerization. After 10
4 min of light exposure, only slight changes were observed in the UV-vis spectrum indicating that
5 a photostationary state was reached. After 24h in the dark, UV-vis spectra started to recover the
6 initial shape due to thermal *Z*-to-*E* back isomerization.
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53 **Figure 8.** UV-Vis spectra of (a) $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}\}_i$ and (b) $\{\text{PEG}_{10}\text{-}b\text{-PDAP}\cdot\text{tAZO}\}_i$ in a
54 5×10^{-6} M THF solution and UV irradiated self-assemblies (1 mg mL^{-1} suspensions) for different
55 times.
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Potential morphological changes upon 10 min UV irradiation were further studied by TEM, Cryo-TEM and DLS. For {PEG₂-*b*-PDAP•tAZO_i}_j, cryo-TEM images show the presence of wrinkled vesicles (Figure 6c) of smaller average D_h respect to the sample prior to exposure, around 225 nm, as calculated by DLS (Figure 7). TEM images of irradiated {PEG₁₀-*b*-PDAP•tAZO_i}_j micelles taken immediately after UV illumination show disruption of the assemblies and regions of material without clear morphology (Figure 6e). Accordingly, accurate measurements by DLS were not possible due to a very high increase of polydispersity of the self-assemblies size in solution. TEM images of irradiated suspensions taken after maintaining the sample 24 h in the dark at room temperature were additionally registered. Under these conditions, thermal *Z*-to-*E* back isomerization of the azobenzene takes place and images show that micelles are at least partially recovered (Figure 6f). To gain additional information about the process, ¹H NMR of a solution of {PEG₁₀-*b*-PDAP•tAZO_i}_j in CDCl₃ was registered after 15 min irradiation (Figure S14 in supporting information). The only observed differences were the presence of new peaks at around 6.88 and 6.75 ppm corresponding to aromatic protons of the *Z*-azobenzene. However, peaks corresponding to the protons involved in the H-bonding were not shifted. This seems to probe that isomerization only influences the azobenzene packing and/or hydrophobic/hydrophilic balance and not H-bonding association.

Self-assemblies of the parent PEG_{*x*}-*b*-PDAP copolymers and the supramolecular BCs {PEG_{*x*}-*b*-PDAP•tAZO_{*i*}}_{*j*}, as well as their light responses, were investigated with detail by SAXS. Figure 9a shows the SAXS profile as a function of the module of the scattering vector q , for different aggregates in water ($q=4\pi/\lambda\sin(\theta)$, being 2θ the scattering angle). In the case of the PEG_{*x*}-*b*-PDAP BCs, the SAXS curves can be described by the scattering of a distribution of

spheres with different diameters.^{48, 49} The scattering form factor for a sphere of radius R is given by **Eq 1**,

$$F_s(q, R) = \left[\frac{3[\sin(qR)] - qR\cos(qR)}{(qR)^3} \right]^2 \quad (1)$$

For **PEG₂-b-PDAP** the distribution was centered around diameters (2R) of approx. 18 nm and for **PEG₁₀-b-PDAP** around 25 nm (**Figure 9b**), which is consistent with DLS analysis and TEM observations.³⁹

However, the scattering from the complexed **{PEG_x-b-PDAP•tAZO_i}** assemblies cannot be fitted with a simple sphere distribution model. In the case of **{PEG₂-b-PDAP•tAZO_i}** cryo-TEM images (**Figure 6b**) showed that the assemblies are vesicles with a diameter larger than 300 nm and a wall thickness of just a few nanometers. Therefore, due to the vesicles dimensions, SAXS profiles only give information about the their walls. Taking this under consideration and using the Guinier approximation for planar objects, it is possible to obtain the radius of gyration (**Eq 2**) from the scattering curve of the supramolecular assembly **{PEG₂-b-PDAP•tAZO_i}**.⁵⁰ The radius of gyration for a planar object with large lateral dimensions is related to its thickness (**Eq 3**),

$$I(q) \sim q^{-2} \exp(-q^2 R_g^2) \quad (2)$$

$$d_t^2 = 12R_g^2 \quad (3)$$

Figure 9c shows the Guinier plot for **{PEG₂-b-PDAP•tAZO_i}** in the dark. Fitting the scattering to the Guinier approximation, provides with a vesicle wall thickness of around 10 nm, which is consistent with our cryo-TEM observations (**Figure 6b**).

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With the aim to monitor the modifications induced on the morphology of the supramolecular assemblies by UV light exposure, real time SAXS experiments were also performed on the **tAZO_i** containing supramolecular assemblies. **Figure 10a** shows the scattering curves corresponding to **{PEG₂-*b*-PDAP•tAZO_i}** vesicles for selected UV illumination times. The UV light induced subtle changes in the scattering curves, especially at large values of q , although these could not be quantified due to excessive noise in the measurements. Nevertheless, Guinier analysis of the curves allowed to estimate the thickness of the vesicle wall as a function of the UV exposure time, and the results are presented in the inset of **Figure 10a**. UV illumination induced thinning of the vesicles wall, and at long exposure times, the slope at low q increases probably as consequence of their further disruption.

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On the other hand, we have seen earlier by TEM than **{PEG₁₀-*b*-PDAP•tAZO_i}** assembled into micelles of mean radius around 20 nm, and thus a different scattering pattern is expected for this supramolecular BC. Although an estimation of the radius of the spherical objects can be obtained by the Guinier approximation, in this case, the whole scattering curve of these assemblies can be fitted using the form factor of a micelle model with a spherical core and Gaussian polymer chains attached to the surface (Eq 4),^{51, 52}

$$F_{mic}(q) = N_{agg}^2 \rho_s^2 F_s(q, R) + N_{agg}^2 \rho_c^2 F_c(q, R_{gauss}) + N_{agg}(N_{agg} - 1) \rho_c^2 S_{cc}(q) + 2N_{agg}^2 \rho_c \rho_s S_{sc}(q) \quad (4)$$

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considering the self-correlation term F_s of the spherical core, of radius R , the self-correlation of the Gaussian chains F_c , characterized by their radius of gyration R_{gauss} , and the crossed terms between the core and the Gaussian chains, S_{sc} , where the fact that the chains cannot penetrate the core is considered, and finally a crossed term between the different Gaussian chains decorating

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3 the core S_{CC} .⁵¹ N_{agg} is the aggregation number of the micelle, *i.e.* the number of molecules per
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6 micelle, ρ_C and ρ_S are the total scattering length excess of the blocks in the spherical core and in
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8 the chains respectively.
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11 **Figure 10b** shows the scattering of the {**PEG₁₀-*b*-PDAP-tAZO_i**} assemblies in water, initially
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13 in the dark and after 15 min of UV illumination, and only slight changes can be observed.
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15 However, fitting their scattering to **Eq 3** allowed to obtain the dependence of several parameters
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17 with the UV illumination time, and the results are depicted in **Figure 10c**. As the sample was
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19 illuminated, the number of chains per micelle decreased slightly, whereas the core radius
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21 increased slightly and the radius of gyration of the chains at the surface remained nearly
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23 constant. Since the photoresponsive groups are located within the core of the micelles, this
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25 region undergoes the strongest effects under UV exposure. **Qualitatively the SAXS results are in**
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27 **agreement with the disruption upon UV irradiation of the micelles shown by TEM results**
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29 **(Figure 6e).** However, it is worth to emphasize on the differences in sample environment
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31 **between the two experiments. While quartz cuvettes were used for the UV irradiation of the**
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33 **samples subsequently investigated by TEM and DLS, glass capillaries were used for SAXS**
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35 **experiments, which has influence in the light absorption by the walls container being higher for**
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37 **glass than for quartz. This effect may provoke significant differences upon comparing**
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39 **quantitatively the kinetics of micelle disruption by UV illumination.**
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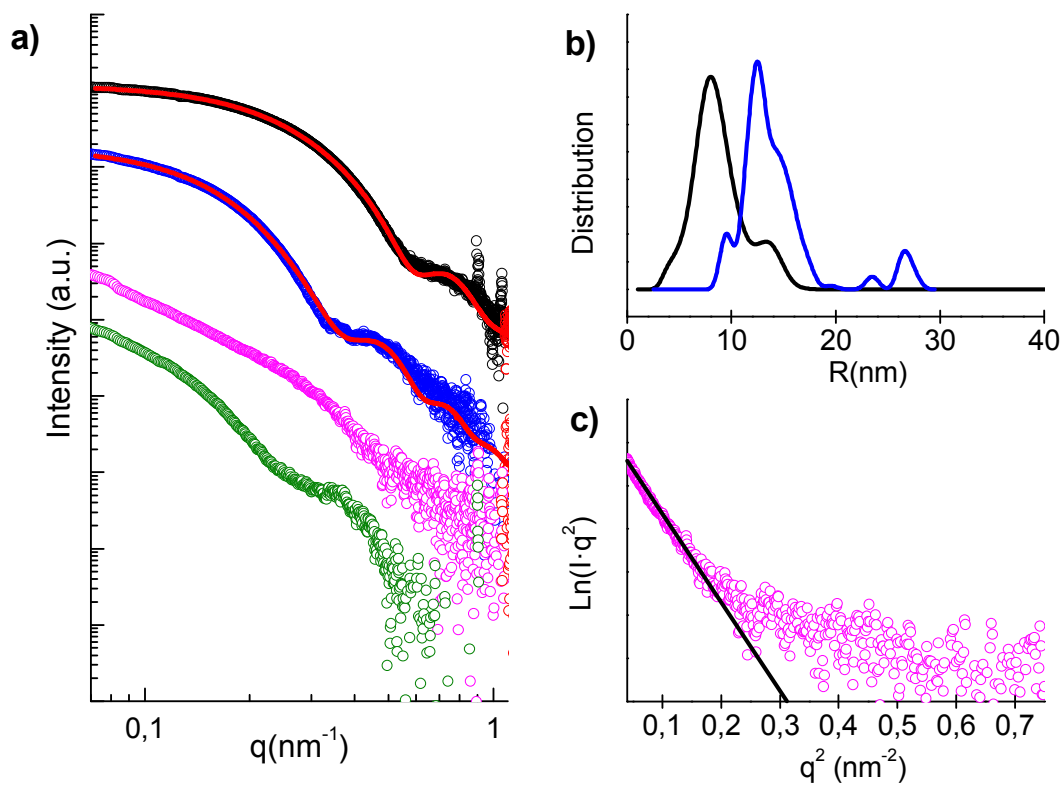


Figure 9. (a) SAXS profiles from PEG₂-b-PDAP (o), PEG₁₀-b-PDAP (o), {PEG₂-b-PDAP•tAZO_i} (o) and {PEG₁₀-b-PDAP•tAZO_i} (o) in the dark. Continuous red lines are fits to the scattering from a distribution of individual spheres. (b) Radius distribution of the spheres (according to the fit in figure a) for PEG₂-b-PDAP (black) and PEG₁₀-b-PDAP (blue). (c) Guinier plot of the scattering from the supramolecular polymer {PEG₂-b-PDAP•tAZO_i} in the dark. The solid line corresponds to the fit to Eq 2.

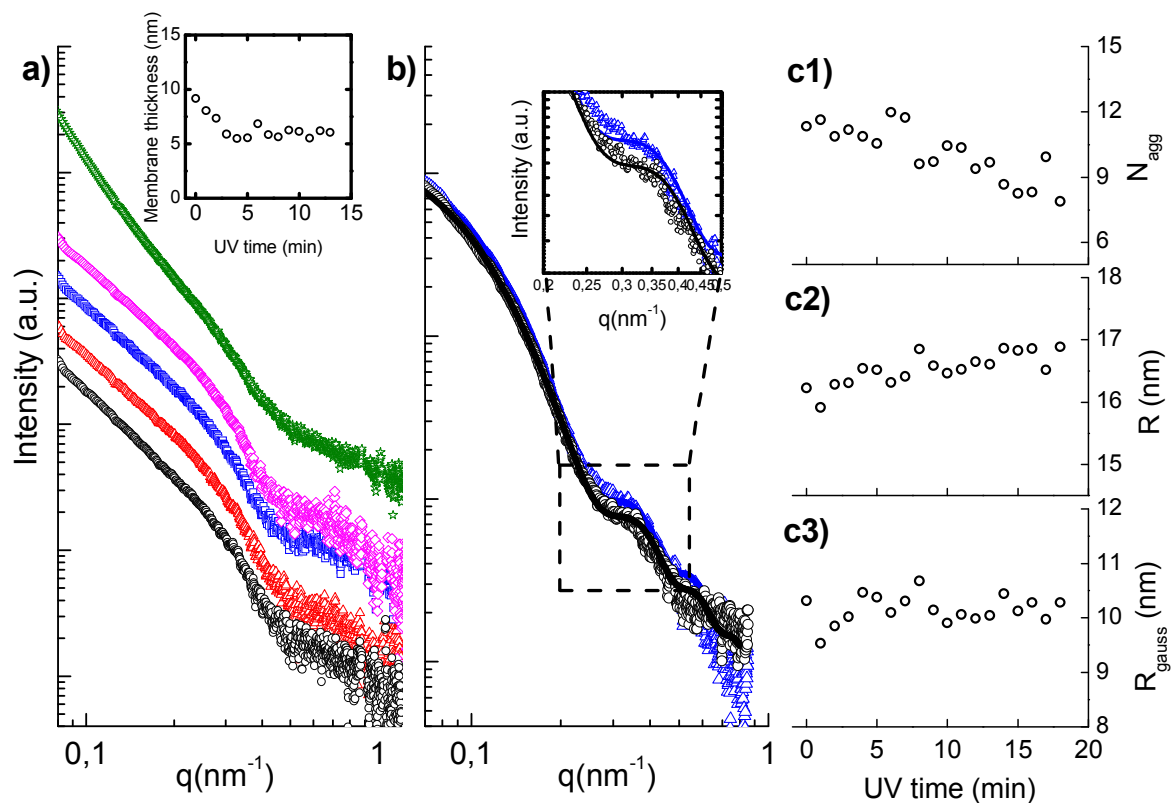


Figure 10. (a) SAXS profiles from {PEG₂-b-PDAP•tAZO_i} assemblies in dark (o), and after 5 (Δ), 10 (□), 15 (◇) and 20 (☆) min of UV illumination. The inset in (a) shows the evolution of the vesicle membrane thickness as obtained by the Guinier approximation in Eq 2 and Eq 3. (b) SAXS profiles from {PEG₁₀-b-PDAP•tAZO_i} assemblies in dark (o) and after 15 min of UV illumination (Δ). Continuous lines correspond to fit to Eq 4. (c) Variation of the relevant structural parameters included in Eq 4 related to the micelles of {PEG₁₀-b-PDAP•tAZO_i} under UV illumination: (c1) Number of chain per micelle, (c2) Radius of the core, and (c3) Radius of gyration of the Gaussian chains forming the shell.

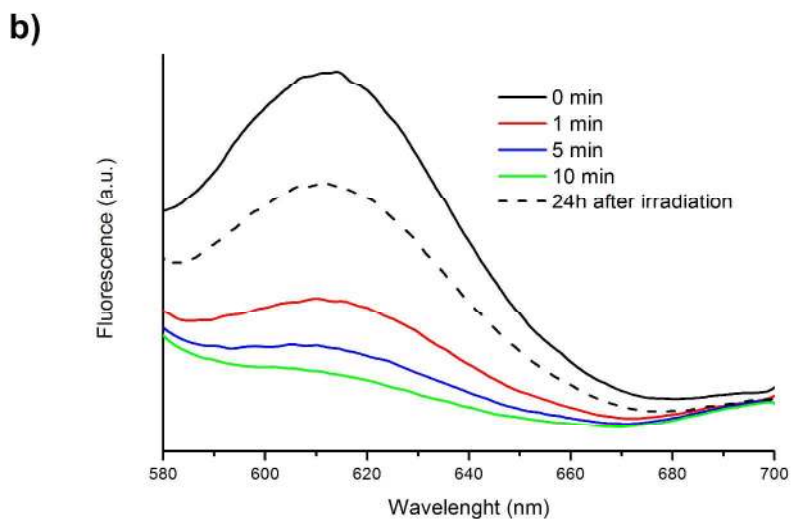
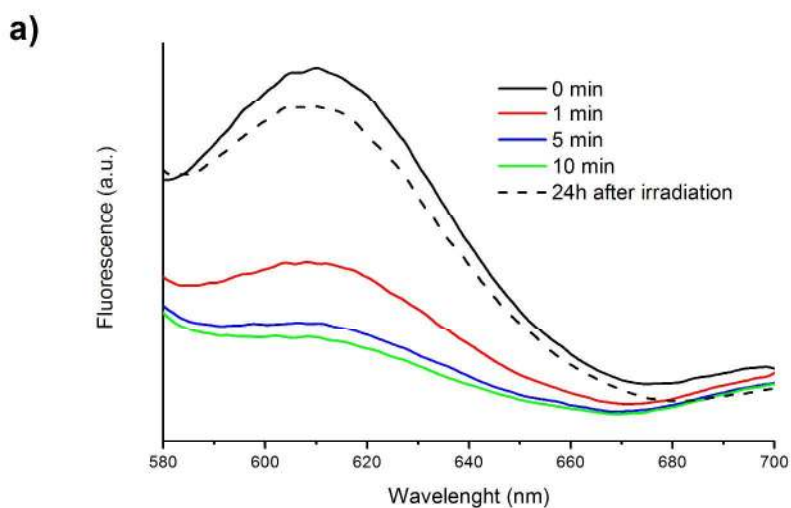
Light stimulated release of fluorescence probes from supramolecular BCs. The potential use of the present vesicles and micelles as stimulus responsive nanocontainers was investigated

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3 by encapsulation and subsequent release of fluorescent probes. $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$
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5 vesicles were tested with two fluorescent probes, since molecules with different nature can be
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7 trapped either into the hydrophilic hollow cavity or the hydrophobic membrane formed by the
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9 azobenzene moieties. Therefore, the ability to encapsulate Nile Red or Rhodamine B, which are
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11 respectively of hydrophobic and hydrophilic nature, was evaluated for these systems.
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16 Nile Red loaded $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ shows a strong emission from 560 to 700 nm while
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18 exciting at 550 nm. Upon irradiation, an abrupt decrease on the initial fluorescence intensity at
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20 606 nm was observed (**Figure 11a**), indicating a shift to a less hydrophobic environment. This
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22 can be due to Nile Red migrating from the membrane to the aqueous media and also to the
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24 increase in the polarity in the inner membrane caused by photoisomerization.¹⁷ When the
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26 irradiated vesicles were kept in the dark for 24 h, the fluorescence was partially recovered, which
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28 could be attributed to partial release of the fluorescent probe.
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33 $\{\text{PEG}_2\text{-}b\text{-PDAP}\cdot\text{tAZO}_i\}$ vesicles loaded with Rhodamine B were also prepared by forming
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35 the assemblies in the presence of the probe, and it was estimated that 0.8 molecules of
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37 Rhodamine B were trapped per macromolecule of BC (using a feed ratio of 5:1
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39 Rhodamine/macromolecule). Encapsulation of Rhodamine B was confirmed by observation of
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41 fluorescence dots in a dark background under confocal microscopy (**Figure 12a**). After 10 min
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43 of UV light irradiation, fluorescence dots were still visible but the fluorescence of background
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45 was more intense. These results strongly suggest that Rhodamine B is released from the interior
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47 of the vesicles to the aqueous surrounding media (**Figure 12b**), and evidence that the vesicle
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49 membrane became permeable to the probe after UV irradiation.
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3 {PEG₁₀-*b*-PDAP•tAZO_i} micelles were only loaded with Nile Red, which should be
4 accommodated within their hydrophobic cores. Again, an abrupt decrease of the initial
5 fluorescence was observed after UV irradiation, indicating that the environment of the probe
6 becomes more hydrophilic (**Figure 11b**). After 24h, the Nile Red fluorescence was not
7 completely recovered, suggesting that the Nile Red was, at least partially, released into the
8 aqueous medium.
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Figure 11. Emission spectra of the Nile Red encapsulated self-assemblies of (a) $\{\text{PEG}_2\text{-}b\text{-}\text{PDAP}\cdot\text{tAZO}_i\}$ and (b) $\{\text{PEG}_{10}\text{-}b\text{-}\text{PDAP}\cdot\text{tAZO}_i\}$, recorded after irradiation for different time intervals.

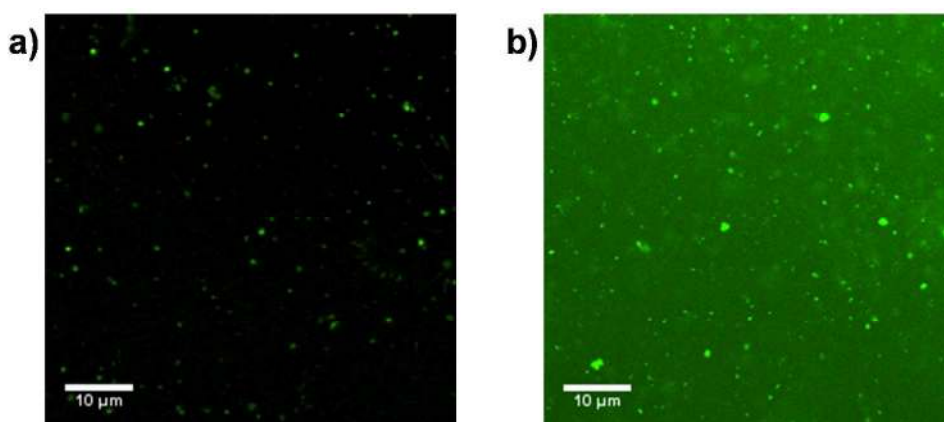


Figure 12. Fluorescence microscopy images of the water suspension of loaded $\{\text{PEG}_2\text{-}b\text{-}\text{PDAP}\cdot\text{tAZO}_i\}$ vesicles (a) before and (b) after irradiation for 10 min.

CONCLUSIONS

Supramolecular amphiphilic BCs based on the H-bond complexation of side chain 2,6-diacylaminopyridine units with thymine functionalized 4-isobutyloxybenzene have been investigated as nanocarriers for capturing and light stimuli delivery of small molecules. These supramolecular BCs, in which PEG is used as the hydrophilic block, have been prepared by selectively embedding azobenzene units into the hydrophobic block through triple H-bonding. While for the parent BCs, $\text{PEG}_{10}\text{-}b\text{-}\text{PDAP}$ and $\text{PEG}_2\text{-}b\text{-}\text{PDAP}$, core-shell spherical micelles were formerly described,³⁹ spherical micellar aggregates have been obtained for the

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3 supramolecular {PEG₁₀-*b*-PDAP•tAZO_i} and vesicles for {PEG₂-*b*-PDAP•tAZO_i} upon self-
4 assembly in water.
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9 Low intensity UV illumination of the self-assemblies in water triggered *E*-to-*Z* isomerization
10 of the azobenzene resulting in morphological changes that were observed by DLS and TEM, and
11 were further confirmed by real time SAXS analyses. More concretely, TEM and SAXS provided
12 evidence of distortion and thinning of the {PEG₂-*b*-PDAP•tAZO_i} vesicles walls after UV
13 exposure. These effects modify the permeability of the membrane towards guest molecules,
14 either hydrophobic cargo molecules entrapped at the membrane, or hydrophilic ones entrapped at
15 the internal aqueous cavity. On the other hand, light provokes the destruction of the core-shell
16 {PEG₁₀-*b*-PDAP•tAZO_i} micelles, which endows in the release of hydrophobic molecules
17 entrapped at the micellar hydrophobic core.
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31 Finally, it should be highlighted that in general terms, the present supramolecular systems
32 present similar light responsive capacities as those previously obtained for analogous covalent
33 amphiphilic BCs.^{17, 18, 53, 54} Thus it can be concluded that hydrogen bonding is an effective and
34 flexible strategy to obtain stimuli responsive assemblies by avoiding the time-consuming
35 procedures associated to the preparation of covalent amphiphilic BCs.
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8 studies.
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19 **Supporting Information.** Synthesis details and characterization of **dAZOi** and **tAZOi**. ^1H - ^1H
20 NOESY spectrum of **{PEG₂-*b*-PDAP•tAZOi}**. FTIR spectra of **{PDAP•dAZOi}** and
21 **{PDAP•tAZOi}**. Self-Assembly of the supramolecular BCs in water and determination of CAC.
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27 This material is available free of charge via the Internet at <http://pubs.acs.org>.
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Table 1. Thermal parameters obtained for the azocompounds, parent polymers and supramolecular polymers

	Degree of complexation ^a	of Hydrophilic/Hydrophobic wt. ratio ^b	$T_{\text{onset}}^{\text{c}}$ (°C)	T_{g}^{d} (°C)	T_{m}^{e} (°C)
tAZO_i	–	–	360	–	166
dAZO_i	–	–	350	68	141
PDAP	0	–	255	73	–
PEG₂-CTA	0	–	210	–	48
PEG₂-<i>b</i>-PDAP	0	35/65	250	6	–
PEG₁₀-CTA	0	–	310	–	58
PEG₁₀-<i>b</i>-PDAP	0	37/63	255	-3	–
{PDAP • tAZO_i}	1	–	265	35	148
{PEG₂-<i>b</i>-PDAP • tAZO_i}	1	18/82	250	33	40, 140
{PEG₁₀-<i>b</i>-PDAP • tAZO_i}	1	19/81	270	38	48, 146
{PDAP • dAZO_i}	1	–	270	77	136
{PEG₂-<i>b</i>-PDAP • dAZO_i}	0.30	20/80	255	4, 80	40, 139
{PEG₁₀-<i>b</i>-PDAP • dAZO_i}	0.30	20/80	265	15, 79	52, 140

^a Number of **tAZO_i** or **dAZO_i** molecules per 2,6-diacylaminopyridine repeating unit

^b Hydrophobic/hydrophilic ratio is given in weight percentage considering the PEG block as hydrophilic the PDAP block (complexed with **dAZO_i** or **tAZO_i**) as hydrophobic.

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3 ^c Onset temperature associated to mass loss detected in the thermogravimetric curve.
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5 ^d Glass transition temperature determined at the half height of the baseline jump on the second heating scan at 10
6 °C·min⁻¹.
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8 ^e Melting temperature(s) read at the maximum of the peak on the second heating scan at 10 °C min⁻¹.
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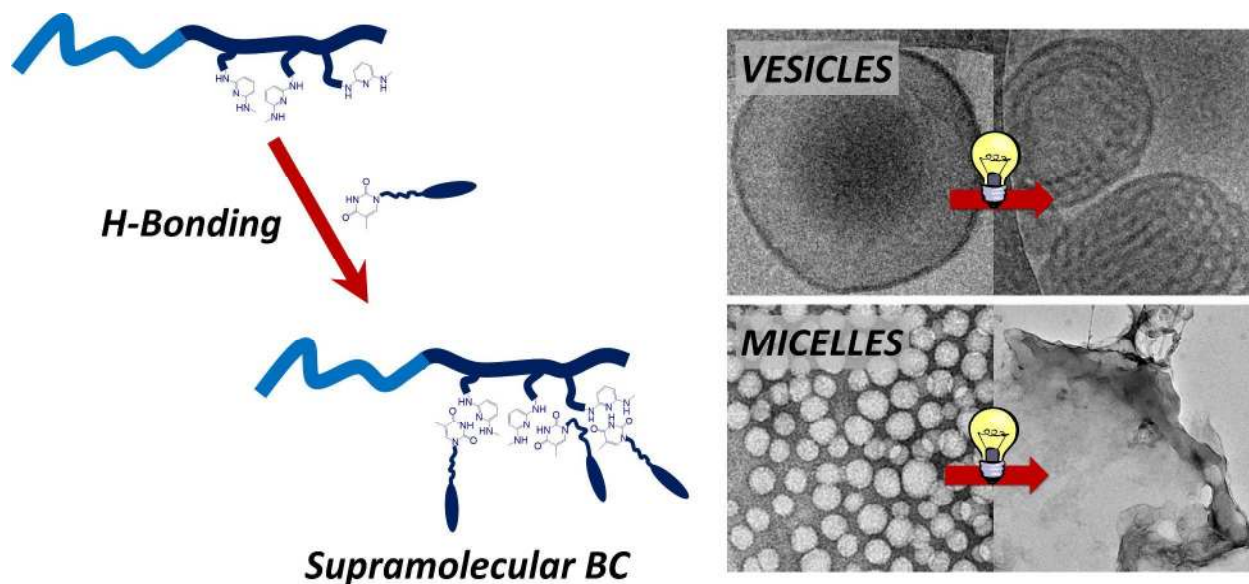
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TABLE OF CONTENTS GRAPHIC



KEYWORDS

Amphiphilic block copolymers; supramolecular post-functionalization; light-responsive nanoself-assemblies