

Low-temperature crystal and magnetic structures of the magnetoelectric material $\text{Fe}_4\text{Nb}_2\text{O}_9$

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$\text{Fe}_4\text{Nb}_2\text{O}_9$ was recently reported to be a new magnetoelectric material with two distinct dielectric anomalies located at $T_N \approx 90$ K for an antiferromagnetic transition and $T_{\text{str}} \approx 77$ K of unknown origin, respectively. By analyzing low-temperature neutron-powder-diffraction data, here we determined its magnetic structure below T_N and uncovered the origin of the second dielectric anomaly as a structural phase transition across T_{str} . In the antiferromagnetically ordered state below T_N , both Fe1 and Fe2 magnetic moments lying within the weakly and strongly buckled honeycomb layers are arranged in a fashion that the three nearest neighbors are directed oppositely. Upon cooling below T_{str} , the symmetry of crystal structure is lowered from trigonal $P\bar{3}c1$ to monoclinic $C2/c$, in which a weak sliding of the metal octahedral planes introduces a monoclinic distortion of $\sim 1.7^\circ$. The magnetic structure is preserved in the low-temperature monoclinic phase, and the Fe magnetic moment increases from $2.1(1)\mu_B$ at 95 K to $3.83(4)\mu_B$ at 10 K assuming an equal moment configuration at Fe1 and Fe2 sites. The magnetic point group and linear magnetoelectric tensor at each temperature region are determined. From a symmetry-related tensor analysis, the microscopic origins of the magnetoelectric effects between T_N and T_{str} are proved to be spin-current and d - p hybridization mechanisms.

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I. INTRODUCTION

Multiferroic materials have been extensively studied over the last two decades since they hold great application possibilities in rapidly growing information technology such as the storage memory or logic devices. In multiferroic materials, ferroelectric and magnetic orders can coexist and have a mutual coupling between them. Therefore, the electric ordering can be controlled by changing the magnetic field and vice versa. Because these two ordering parameters are usually mutually exclusive, multiferroics are not commonly found in nature. Thus, an enormous effort has been devoted to predict and design multiferroic materials with outstanding magnetoelectric (ME) coupling. In this regard, spin-driven (or type-II) multiferroics are promising candidates in which some specific types of magnetic structure break the inversion symmetry and generate polarization. Several type-II multiferroics such as TbMnO_3 , TbMn_2O_5 , CuO , and GdFeO_3 have been found to exhibit strong ME coupling [1–4].

Apart from this class of multiferroics, another group of materials possessing centrosymmetric crystal structure, e.g., Cr_2O_3 , FeSb_2O_4 , NdCrTiO_5 , and MnTiO_3 [5–8], also exhibit strong cross coupling between electric and magnetic order via the linear magnetoelectric (LME) effect. Unlike type-II multiferroics, they are not ferroelectrics at the ground state, and become ferroelectric only under an external magnetic field above some critical value. In addition, their polarizations linearly increase with magnetic field. Although the LME effect was first observed in antiferromagnetic (AF) Cr_2O_3 in 1961 [5], so far only a limited number of LME materials are available due to symmetry restriction.

Recently, the family of $A_4M_2O_9$ ($A = \text{Fe}, \text{Co}, \text{Mn}$ and $M = \text{Nb}, \text{Ta}$) materials has attracted profound research interest owing to the observation of novel LME effect [9–17]. At ambient conditions, these materials crystallize in a trigonal $P\bar{3}c1$ centrosymmetric space group, analogous to the magnetoelectric corundum-type Cr_2O_3 [18]. In a more recent study on $\text{Fe}_4\text{Nb}_2\text{O}_9$ (FNO), LME coupling and spin-driven ferroelectricity were reported below its Néel temperature, T_N , under magnetic field above 6 T [9]. Its magnetic susceptibility $\chi(T)$ exhibits an anomaly at the paramagnetic to AF transition of $T_N \approx 90$ K. Interestingly, its dielectric permittivity $\epsilon'(T)$

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displays two successive anomalies at about 90 and 77 K, respectively. While the former anomaly corresponds to the AF transition seen in $\chi(T)$, the origin for the second anomaly of $\varepsilon'(T)$ around 77 K remains elusive. In Ref. [9], it was speculated that the latter anomaly might arise from either a spin reorientation or even a structural transition. To address this issue, it is essential to obtain the information about low-temperature crystal and magnetic structures of FNO, which remain unknown to our knowledge. In this work, we have carried out low-temperature neutron-powder-diffraction (NPD) measurements on polycrystalline FNO. Detailed data analyses allowed us to determine its AF spin structure below T_N , and to reveal a structural phase transition across T_{str} (denoted as T_{N2} in Ref. [9]) from high-temperature trigonal $P-3c1$ to low-temperature monoclinic $C2/c$ phase.

II. EXPERIMENTAL

Polycrystalline FNO samples were prepared in a conventional solid-state reaction route as described in Ref. [9]. Stoichiometric amounts of Fe, Fe_2O_3 , and Nb_2O_5 powders were thoroughly mixed and then pelletized in a glove box filled with argon. The pellets were then sealed under vacuum in a quartz tube and heated at 1000 °C for 20 h. This process was repeated once to improve the sample quality. Powder x-ray diffraction (XRD) was carried out at room temperature to verify the phase purity using $\text{Cu K}\alpha$ radiation. NPD measurements were carried out on the High Resolution Powder Diffractometer at China Advanced Research Reactor, using neutron wavelength $\lambda = 1.888 \text{ \AA}$ in the angular range 7° – 152° with a step of 0.07° . The data have been collected at 150, 110, 95, 85, 60, and 10 K. Temperature-dependent dielectric permittivity was measured with Agilent E4980A Precision LCR meter at different frequencies. The low-temperature environment was realized in a Physical Property Measurement System (PPMS) from Quantum Design (QD). Magnetic susceptibility was measured with a QD Magnetic Property Measurement System (MPMS3) magnetometer.

III. RESULTS

Room-temperature XRD pattern on FNO confirms that we have obtained a nearly single-phase sample with $\text{Nb}_2\text{Co}_4\text{O}_9$ -type structure having a trigonal $P-3c1$ space group. A small amount of FeNb_2O_6 secondary phase, $\sim 1.3 \text{ wt\%}$, was identified via analyzing the XRD pattern, similar as reported in the previous study [9]. The calculated lattice parameters $a = 5.2231(1) \text{ \AA}$ and $c = 14.2018(4) \text{ \AA}$ at room temperature are in excellent agreement with those of $a = 5.2253(1) \text{ \AA}$ and $c = 14.2058(2) \text{ \AA}$ reported previously [9].

The obtained sample is further characterized by measurements of magnetic susceptibility $\chi(T)$ and dielectric permittivity $\varepsilon'(T)$. The $\chi(T)$ was measured upon warming up under an external field of 0.01 T after both zero-field cooled (ZFC) and field cooled (FC) from room temperature. As shown in Fig. 1(a), a small kink observed in both ZFC and FC $\chi(T)$ curves suggests the AF transition around $T_N \approx 94 \text{ K}$, which is slightly higher than that reported in Ref. [9]. The increase of $\chi(T)$ below T_N upon cooling should arise from the field-induced spin canting along some easy axis, which can

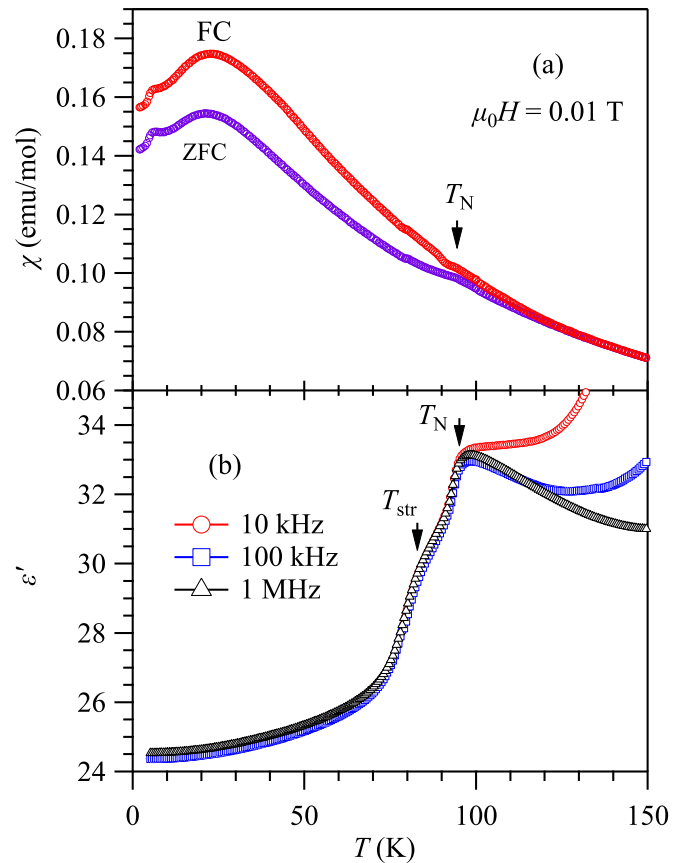


FIG. 1. Temperature dependence of (a) magnetic susceptibility $\chi(T)$ measured under 0.01 T and (b) dielectric permittivity $\varepsilon'(T)$ at different frequencies ($f = 10 \text{ kHz}$, 100 kHz , and 1 MHz). The Néel temperature $T_N \approx 94 \text{ K}$ and the structural transition temperature $T_{str} \approx 80 \text{ K}$ are indicated by the arrows in the figure.

be determined from measurements of $\chi(T)$ along different directions of single-crystal sample. An additional peak around 6 K should be attributed to the FeNb_2O_6 impurity. Similar as the previous report [9], two anomalies are evidenced in $\varepsilon'(T)$ at $T_N \approx 94 \text{ K}$ and $T_{str} \approx 80 \text{ K}$ in the absence of magnetic field. These anomalies show no frequency dependence up to 1 MHz as seen in Fig. 1(b). As mentioned above, the magnetic structure of FNO below T_N and the origin of the second $\varepsilon'(T)$ anomaly around T_{str} remain elusive. Thus, the main purpose of this work is to determine the low-temperature crystal and magnetic structures of FNO.

Figure 2 shows the low-angle regions of NPD patterns measured at various temperatures for FNO sample. For the sake of clarity, the patterns are offset by 0.6° and 600 counts with respect to each other. It can be clearly seen that two consecutive phase transformations take place upon cooling. First, a magnetic transition occurs between 110 and 95 K as indicated by the emergence of several peaks at low angles around 17° , 26° , and 33° . Then, a pronounced peak splitting at 85 K and below suggests the presence of a structural phase transition between 95 and 85 K. It should be noted that the magnetic and structural transition temperatures seen from NPD in Fig. 2 are slightly higher than those determined from anomalies in $\chi(T)$ and $\varepsilon'(T)$ shown in Fig. 1. This is

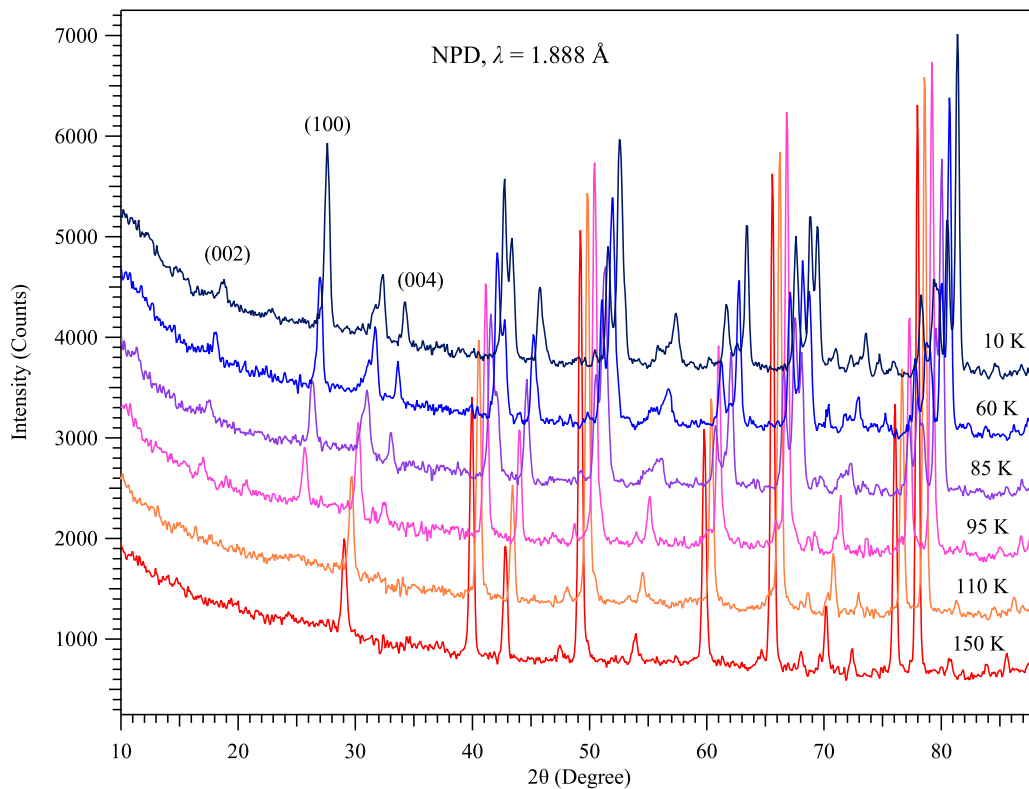


FIG. 2. Neutron-diffraction patterns of $\text{Fe}_4\text{Nb}_2\text{O}_9$ at low-angle regions measured at different temperatures. For the sake of clarity, the patterns are offset by 0.6° and 600 counts with respect to each other. The (hkl) indices of the magnetic peaks in the figure are given for the high-temperature trigonal structure setting.

presumably due to the distant location of thermometer from sample in the NPD measurements. Thus, the transition temperatures $T_N \approx 94$ K and $T_{\text{str}} \approx 80$ K determined from $\chi(T)$ and $\varepsilon'(T)$ should be regarded as the real values.

A. Crystal structure at 150 K

Before solving the low-temperature magnetic and crystal structures, we first determined the structural parameters of FNO at 150 K, which retains the centrosymmetric $P\bar{3}c1$ symmetry with two Fe atoms located at $4d$ sites $(1/3, 2/3, z)$, one Nb atom at $4c$ site $(0, 0, z)$, and two O atoms at $6g$ and $12f$ positions, respectively. Figure 3(a) displays the Rietveld refinement plot of the NPD data at 150 K. The obtained structural parameters are listed in Table I, and the refined crystal structure is depicted in Fig. 4(a).

As can be seen, the layered trigonal crystal structure represents an ordered stacking of the honeycomb layers of edge-sharing metal-oxygen octahedra, where (Fe2)Nb bilayers are interleaved with a single layer composed of solely Fe1-oxygen octahedra. In this arrangement, every metal-oxygen octahedron shares three edges with neighboring metal-oxygen octahedra within the plane, and one face with an octahedron on a neighboring layer along the c axis, while the opposite face borders the vacant space in the middle of the honeycomb hexagons in the neighboring layer of the opposite direction. Along the c axis, Fe1 octahedra are exclusively facing the Fe2 octahedra and vice versa, and the Nb octahedra are exclusively facing the Nb octahedra. In the (Fe2)Nb

bilayers, all honeycomb units are exclusively composed of the sequentially positioned Fe2 and Nb atoms, such that every Nb atom borders only the Fe2 octahedra in the plane, and every Fe2 atom borders the octahedra around Nb atoms. The fact that every metal octahedron along the c axis is face-sharing connected to another octahedron in one direction and to the empty space in the middle of a honeycomb unit in another leads to a rather pronounced buckling of the layers. Thereby, the “thickness” of the buckled (Fe2)Nb layers, amounting to $0.715(11)$ Å, is significantly larger than the buckling of the layers composed of solely Fe1 atoms that amounts to $0.420(9)$ Å. The in-plane interatomic separations between the metal atoms in the buckled planes are slightly longer than the separations through the shared faces along the c axis. For example, the Nb–Fe2 bonds in the Fe(2)Nb planes are $3.098(2)$ Å, while the Nb–Nb bonds along the c axis are $3.081(10)$ Å, respectively. Similarly, the Fe1–Fe1 bonds in the Fe1 planes are $3.043(1)$ Å, while the Fe1–Fe2 bonds along the c axis are $2.931(9)$ Å.

B. Magnetic structure at 95 K

Upon cooling to 95 K, we observe the appearance of (002), (100), (004), and some other diffraction peaks as shown in Fig. 2. All these are actually allowed diffraction peaks for the $P\bar{3}c1$ crystal structure, yet they have diminishingly weak intensities in neutron diffraction on the purely chemical structure. Thus, they should be attributed to the development of long-range AF ordering with propagation vector $\kappa = 0$. By

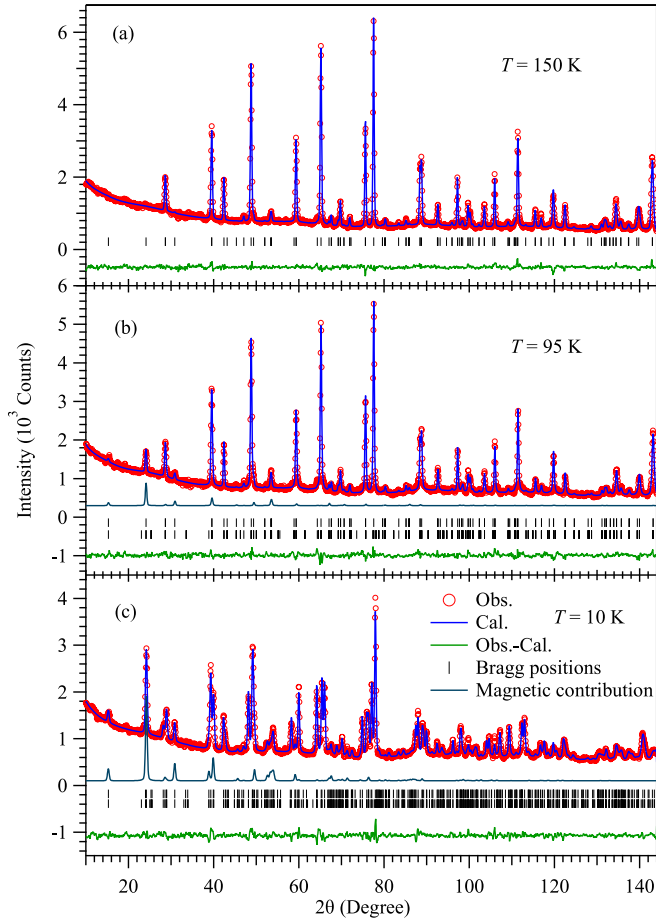


FIG. 3. Rietveld refinement plots of the neutron-powder-diffraction data at (a) 150, (b) 95, and (c) 10 K. The ticks below the graphs indicate the calculated Bragg positions of the crystal structure (upper row) and magnetic structure (lower row). The magnetic contributions at 95 and 10 K are also shown under the observed/calculated profiles.

using the program SARAH [19], we carried out the symmetry analysis of possible magnetic ordering of the Fe1 and Fe2 ions for a propagation vector $\kappa = 0$, based on the NPD at 95 K and high-temperature trigonal crystal structure. Decomposition of the magnetic representation Γ_{Mag} and the possible magnetic basis vectors are summarized in Supplemental

TABLE I. Crystal structural parameters of $\text{Fe}_4\text{Nb}_2\text{O}_9$ refined from neutron-powder-diffraction data at 150 K^a in space group $P\text{-}3c1$ (No. 165, $Z = 2$) with the lattice parameters $a = 5.2205(2)$ Å, $c = 14.1857(5)$ Å, and $V = 334.81(2)$ Å³.

Atom	Site	x	y	z	B (Å ²)
Nb	4c	0	0	0.1414(5)	0.61(11)
Fe1	4d	1/3	2/3	0.0148(4)	0.59(13)
Fe2	4d	1/3	2/3	0.3082(5)	0.82(12)
O1	12g	0.3403(6)	0.3158(11)	0.0849(4)	0.66(7)
O2	6f	0.2824(10)	0	1/4	0.58(12)

^aReliability factors: $\chi^2 = 1.58$, $R_{\text{Bragg}}(\%) = 5.14$.

Material, Table S1 [20]. After a careful examination of all possible magnetic-ordering models against the NPD pattern in the Rietveld refinements, we conclude that the magnetic ordering occurs according to the irreducible representation Γ_6 , and the magnetic structure can be described by a linear combination of either the basis vectors ψ_9 and ψ_{10} , or ψ_{11} and ψ_{12} . The coefficients in these linear combinations are identical, i.e., either of the two combinations $1^*\psi_9 + 1/3^*\psi_{10}$ or $1^*\psi_{11} + 1/3^*\psi_{12}$, provides an equivalently satisfying fit to the experimental data, and gives an essentially identical magnetic structure.

An illustration of the final Rietveld refinement of the crystal and magnetic structures based on the NPD data at 95 K is shown in Fig. 3(b). For a better understanding of the magnetic structure, it is instructive to map out the magnetic Fe sublattices. The two distinct honeycomb Fe layers are highlighted in Fig. 4(c). The Fe1 atoms form a weakly buckled Fe1 layer with the interatomic separations of 3.043(1) Å and an effective buckling thickness of 0.420(9) Å, while the Fe2 atoms belonging to two adjacent (Fe2)Nb layers form a strongly buckled Fe2 layer, in which both the interatomic Fe2-Fe2 distances of 3.437(5) Å, and the buckling thickness of 1.651(11) Å are much larger than those in the Fe1 layer. In addition, while the Fe1 layer is indeed a unique layer formed by the edge-sharing octahedra around the Fe1 atoms, the exact exchange paths in the severely buckled Fe2 layer are more complex, because these Fe2 atoms are consecutively belonging to different neighboring (Fe2)Nb layers.

The refined magnetic structure at 95 K is depicted in Fig. 4(d). As can be seen, the Fe1 and Fe2 magnetic moments are lying in the ab plane, and are arranged antiferromagnetically inside the planes. For every Fe atom in either the weakly or strongly buckled honeycomb layers, the moments of the three nearest-neighbor Fe atoms in the same layer are directed oppositely. Along the c axis, the moments of Fe1 and Fe2 atoms are coupled ferromagnetically. Assuming an equal moment configuration at the Fe1 and Fe2 sites, the magnitude of Fe magnetic moment is refined to be 2.1(1) μ_B at 95 K.

C. Structural phase transition at 85 K

Upon further cooling to 85 K and below, the nuclear diffraction peaks get split (Fig. 2), which indicates that the sample undergoes a structural phase transition. A closer inspection of the NPD patterns at 85 K and below indicates that nuclear-diffraction peaks display a specific splitting pattern, whereas the magnetic peaks seem to stay unchanged in their positions except for a gradual enhancement of intensity upon cooling. It was thus assumed that the structure transition should happen into the symmetry-related subgroup, while the magnetic structure is preserved. Indeed, a direct subgroup of the $P\text{-}3c1$ —a monoclinic $C2/c$ space group—immediately delivers a solution that can satisfactorily describe the observed pattern.

In the low-temperature monoclinic $C2/c$ structure, the positions of O1 and O2 atoms of the high-temperature trigonal structure are splitting into three and two independent positions, respectively. Except for O2_1 located at the 4e (0, y , 1/4) site, all other atoms are located in the general 8f (x , y , z) positions. Refinement of the crystal-structure

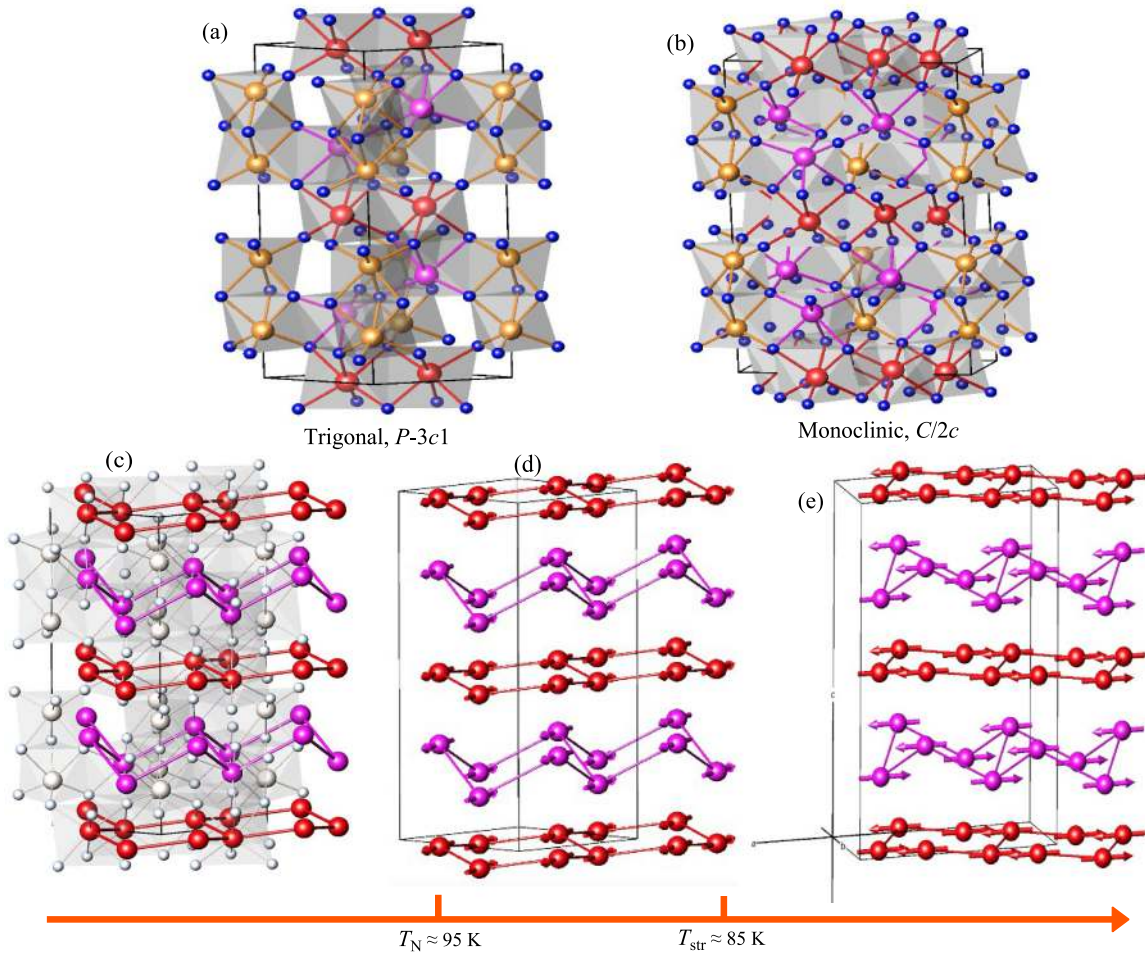


FIG. 4. Illustrations of the crystal structure (a) at 150 K in trigonal $P-3c1$ space group and (b) at 10 K in monoclinic $C/2c$ space group. Fe atoms are plotted in red (Fe1, in the layers formed solely by Fe) and in pink (Fe2, in the FeNb layers), Nb atoms in orange, and oxygen atoms in blue. (c) Schematic view of Fe sublattices in a direction nearly perpendicular to the c axis. The weakly buckled honeycomb layers of Fe1 atoms (red) are interleaved by the strongly buckled honeycomb layers of Fe2 atoms (pink). (d) Schematic view onto the magnetic structure at 95 K in trigonal phase. Fe moments are directed along the projections of two of the bonds of every honeycomb onto the $x-y$ plane, and 60° inclined to the projections of the other four bonds. (e) The magnetic structure at 10 K in the monoclinic phase.

parameters was done based on the neutron data at 10 K. An illustration of the Rietveld refinement plot is shown in Fig. 3(c), and a schematic representation of the crystal structure at 10 K is given in Fig. 4(b). The refined crystal-structure parameters are summarized in Table II. As shown in Fig. 4(b), while the spatial ordering of the Fe and Nb octahedra remains essentially same as the trigonal phase, the low-temperature crystal structure features some differences. The a/b ratio of the resulting rectangle built on the a and b lattice constants displays a weak deviation from $\sqrt{3}$, namely $a/\sqrt{3}b \approx 1.013$. Additionally, a rather weak sliding of the metal octahedral planes with respect to each other introduces a monoclinic distortion of $\sim 1.7^\circ$.

For this low-temperature monoclinic cell, all magnetic peaks fit nicely into the $\kappa = 0$ assumption, thus confirming that the magnetic ordering is preserved through the structural phase transition. Based on the NPD data at 10 K, the symmetry analysis of possible magnetic ordering of the Fe1 and Fe2 ions, both located in the $8f$ (x, y, z) positions for a propagation vector $\kappa = 0$, has been carried out again using

the program SARAH [19]. The decomposition of the magnetic representation Γ_{Mag} and the possible magnetic basis vectors are summarized in Table S2 [20]. After checking all possibilities, we conclude that the only irreducible representation that succeeds to fit the data is the Γ_2 with the ψ_4 basis function. The resulting magnetic structure is presented in Fig. 4(e). The in-plane contradiction between the trigonal symmetry and the unique direction of magnetic moments along one of the three Fe–Fe bonds' directions at 95 K is being taken off by a slight deviation from trigonal symmetry in the xy plane. Under assumption of the equal moment configuration, the magnetic Fe moments amount to $3.83(4)\mu_B$ at 10 K.

IV. DISCUSSION

The magnetic structure of $A_4M_2O_9$ ($A = \text{Co, Mn and } M = \text{Nb, Ta}$) materials and the origin of spin-induced polarization have been subjected to extensive study. Bertaut *et al.* [18] first reported the magnetic structure of $\text{Mn}_4\text{Nb}_2\text{O}_9$ and $\text{Co}_4\text{Nb}_2\text{O}_9$ using NPD experiments. According to their study, the spins

TABLE II. Crystal structural parameters of $\text{Fe}_4\text{Nb}_2\text{O}_9$ refined from neutron-powder-diffraction data at 10 K^a in space group $C2/c$ (No. 15, $Z = 4$) with the lattice parameters $a = 9.0916(6)$ Å, $b = 5.1825(3)$ Å, $c = 14.1983(9)$ Å, $\beta = 91.677(1)^\circ$, and $V = 668.71(7)$ Å³.

Atom	Site	x	y	z	B (Å ²)
Nb	8f	-0.0007(11)	-0.000(3)	0.1421(8)	0.31(18)
Fe1	8f	0.1645(8)	0.501(3)	0.0132(5)	0.65(13)
Fe2	8f	0.1722(9)	0.505(2)	0.3066(6)	0.97(16)
O1_1	8f	0.1698(16)	0.850(3)	0.084(1)	0.77(12) ^b
O1_2	8f	0.3462(16)	0.3192(19)	0.087(1)	0.77(12) ^b
O1_3	8f	0.4951(17)	0.825(2)	0.085(1)	0.77(12) ^b
O2_1	4e	0	0.720(4)	1/4	1.1(2) ^c
O2_2	8f	0.1365(12)	0.145(3)	0.250(1)	1.1(2) ^c

^aReliability factors: $\chi^2 = 1.68$, $R_{\text{Bragg}}(\%) = 5.20$.

^bAtomic displacement parameters for atoms O1_1, O1_2, and O1_3 were refined with constraint to equality.

^cAtomic displacement parameters for atoms O2_1 and O2_1 were refined with constraint to equality.

are ordered along the c axis within two ferromagnetic chains along $(1/3, 2/3, z)$ (+spin) and $(2/3, 1/3, z)$ (-spin). These two chains are antiferromagnetically coupled resulting in a net zero magnetization. Magnetic susceptibility measurements on $\text{Mn}_4\text{Nb}_2\text{O}_9$ single crystal confirmed the collinear magnetic structure along the c axis in this system [21]. Similar magnetic structure was also reported in $\text{Mn}_4\text{Ta}_2\text{O}_9$ single crystal [22]. In contrast, measurements of single-crystal neutron diffraction and magnetic susceptibility along different crystallographic directions have shown that the magnetic moments in $\text{Co}_4\text{Nb}_2\text{O}_9$ are confined in the trigonal basal plane instead of parallel to the c axis [12]. Later on, Deng *et al.* showed that the magnetic moments of Co atoms are noncollinearly arranged in the ab plane [23]. A recent neutron-diffraction study on $\text{Co}_4\text{Ta}_2\text{O}_9$ also evidenced noncollinearly canted spin arrangement of Co atoms in the ab plane [24].

In the present study, we not only determined the magnetic structure of FNO below T_N , but also uncovered a structural phase transition at $T_{\text{str}} < T_N$. It is noteworthy that such a temperature-driven structural phase transition observed in FNO has not been observed in related $A_4M_2O_9$ ($A = \text{Mn}, \text{Co}$) materials. The detailed structural parameters of FNO at different temperatures are given in Table S3 [20]. Such important information from this study enriches the magnetic and structural properties of the $A_4M_2O_9$ family. In contrast to the above-mentioned $A_4M_2O_9$ materials, the magnetic moments of both Fe1 and Fe2 atoms are found to be confined roughly in the ab plane and directed antiparallel with respect to its three nearest neighbors. Despite the distinct magnetic structures, the observation of dielectric anomaly around T_N in FNO suggests a strong ME coupling. The magnetic structure determined in this work can provide the basis for further studies of LME coupling in FNO.

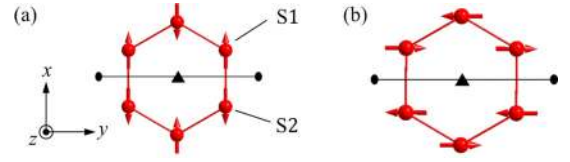


FIG. 5. Top view onto the two possible magnetic structures at 95 K in trigonal phase. Fe moments are directed along the (a) x axis and (b) y axis.

As mentioned above, our results reveal that Fe1 and Fe2 magnetic moments are collinearly arranged in the ab plane with two possibilities. We first define Cartesian coordinates, x, y, z , as shown in Fig. 5. For $T_{\text{str}} < T < T_N$, (1) if the magnetic structure is given by the basis vectors $1^*\psi_9 + 1/3^*\psi_{10}$, the magnetic moments of the Fe1 and Fe2 atoms are antiparallelly aligned along the $x//[2, 1, 0]$ direction, respectively, as shown in Fig. 5(a). Accordingly, the magnetic point group is $2/m'$. 2 is twofold rotation along the $y//[0, 1, 0]$ direction. (2) If the magnetic structure is given by the basis vectors $1^*\psi_{11} + 1/3^*\psi_{12}$, the magnetic moments of the Fe1 and Fe2 are antiparallelly aligned along the y direction, respectively, as shown in Fig. 5(b). Accordingly, the magnetic point group is $2'/m$. Below 85 K, the magnetic point group is $2/m'$ too. Both $2'/m$ and $2/m'$ are nonpolar and allow LME effect [25]. In particular, the linear magnetoelectric matrix form is

$$\begin{bmatrix} 0 & \alpha_{xy} & 0 \\ \alpha_{yx} & 0 & \alpha_{yz} \\ 0 & \alpha_{zy} & 0 \end{bmatrix} \quad (1)$$

for $2'/m$ and

$$\begin{bmatrix} \alpha_{xx} & 0 & \alpha_{xz} \\ 0 & \alpha_{yy} & 0 \\ \alpha_{zx} & 0 & \alpha_{zz} \end{bmatrix} \quad (2)$$

for $2/m'$, respectively, which are consistent with the reported LME behavior below 95 K [9].

In type-II multiferroic, the ME coupling below the magnetic-ordering temperature has been explained in terms of three well-known models, i.e., the spin-current mechanism for noncollinear magnetic structure [26], the magnetostriction model for collinear spin arrangements [27], and d - p hybridization mechanism between spin and its ligand atoms with spin-orbital interaction [28], in some low-symmetry crystals with noncollinear magnetic orders. In order to extract more information about the microscopic origin of ME coupling, we introduce a symmetry-based local ME tensor technique [29], which has successfully resolved the microscopic origin of polarization in multiferroic hexaferrites [30]. Let us consider that a spin pair S_1 and S_2 can produce a local dipole p as the quadratic functions in Einstein convention:

$$\begin{aligned} p &= p(S_1, S_2) + p(S_1) + p(S_2) \\ &= P_{12}^{\alpha\beta\gamma} S_{1\alpha} S_{2\beta} + \sum_{i=1,2} P_{ii}^{\alpha\beta\gamma} S_{i\alpha} S_{i\beta}, \end{aligned} \quad (3)$$

where α, β, γ run over all the Cartesian coordinates, x, y, z . All three well-known mechanisms mentioned above can be expressed as a special form of spin tensor $P_{ii}^{\alpha\beta\gamma}$ and two-spin tensor $P_{ij}^{\alpha\beta\gamma}$. In particular, the spin-current mechanism $p \propto e_{12} \times (S_1 \times S_2)$ leads to antisymmetric $P_{ij}^{\alpha\beta\gamma}$:

$$\begin{pmatrix} 0 & e_{12y}, -e_{12x}, 0 & e_{12z}, 0, -e_{12x} \\ -e_{12y}, e_{12x}, 0 & 0 & 0, e_{12z}, -e_{12y} \\ -e_{12z}, 0, e_{12x} & 0, -e_{12z}, e_{12y} & 0 \end{pmatrix}, \quad (4)$$

while exchange-striction mechanism $p \propto e_{12}(S_1 \bullet S_2)$ shows a $P_{ij}^{\alpha\beta\gamma}$ with diagonal term only:

$$\begin{pmatrix} e_{12x}, e_{12y}, e_{12z} & 0 & 0 \\ 0 & e_{12x}, e_{12y}, e_{12z} & 0 \\ 0 & 0 & e_{12x}, e_{12y}, e_{12z} \end{pmatrix}, \quad (5)$$

where $e_{12} = (e_{12x}, e_{12y}, e_{12z})$ is connecting vector of adjacent spin pair S_1 and S_2 .

In contrast, p - d hybridization, $p \propto e_i(e_i \bullet S_i)^2$, has a $P_{ii}^{\alpha\beta\gamma}$ tensor with the symmetric matrix form:

$$\begin{pmatrix} e_{ix}e_{ix}e_{ix}, e_{ix}e_{ix}e_{iy}, e_{ix}e_{ix}e_{iz} & e_{ix}e_{iy}e_{ix}, e_{ix}e_{iy}e_{iy}, e_{ix}e_{iy}e_{iz} & e_{ix}e_{iz}e_{ix}, e_{ix}e_{iz}e_{iy}, e_{ix}e_{iz}e_{iz} \\ e_{iy}e_{ix}e_{ix}, e_{iy}e_{ix}e_{iy}, e_{iy}e_{ix}e_{iz} & e_{iy}e_{iy}e_{ix}, e_{iy}e_{iy}e_{iy}, e_{iy}e_{iy}e_{iz} & e_{iy}e_{iz}e_{ix}, e_{iy}e_{iz}e_{iy}, e_{iy}e_{iz}e_{iz} \\ e_{iz}e_{ix}e_{ix}, e_{iz}e_{ix}e_{iy}, e_{iz}e_{ix}e_{iz} & e_{iz}e_{iy}e_{ix}, e_{iz}e_{iy}e_{iy}, e_{iz}e_{iy}e_{iz} & e_{iz}e_{iz}e_{ix}, e_{iz}e_{iz}e_{iy}, e_{iz}e_{iz}e_{iz} \end{pmatrix}, \quad (6)$$

where vector $e_i = (e_{ix}, e_{iy}, e_{iz})$ connects the transition metal i and its neighbor ligand atom. By introducing the crystal symmetry, the forms of the local ME tensors can be greatly simplified due to its polar tensor property [30]. For simplicity, we assume: (i) there are only three kinds of two-spin interactions that can generate local dipole, Fe1-Fe1, Fe2-Fe2, and Fe1-Fe2. As shown in Fig. 4(d), the Fe1 and Fe2 atoms are denoted as the red and purple symbols, respectively, and the Fe1-Fe1 and Fe2-Fe2 are bonds between the nearest-neighbor Fe atoms within the red and purple layers, having bond length of 3.043 and 3.412 Å, respectively. The Fe1-Fe2 bonds are along the c axis between two nearest-neighbor Fe1 and Fe2 atoms with a distance of 2.931Å. All three kinds of bonds between nearest-neighbor Fe atoms are considered for the local electric dipole; (ii) in any spin configuration, the spins in Fe1 and Fe2 layers are identical and respond in same manner to the external field H . As shown in Fig. 5, the spin at 1 site in Fe1 layer is S_1 and 2 site S_2 where $S_1 = -S_2$ only hold for $H = 0$. Then, the total polarization generated in Fe1 layer $p_{\text{Fe1-Fe1}}$ by Fe1-Fe1 two-spin interaction can be expressed as

$$p_{\text{Fe1-Fe1}} = \sum_{ij} P_{ij}^{\alpha\beta\gamma} S_{i\alpha} S_{j\beta} = \left(\sum_{ij} P_{ij}^{\alpha\beta\gamma} \right) S_{1\alpha} S_{2\beta}, \quad (7)$$

where ij runs all adjacent Fe1-Fe1 pairs and ensure that S_i is S_1 and S_j is S_2 . After considering all the 3_z , 2_y , and -1 symmetric operations, $\sum_{ij} P_{ij}^{\alpha\beta\gamma}$ can be simplified as

$$\sum_{ij} P_{ij}^{\alpha\beta\gamma} \propto \begin{pmatrix} 0 & 0 & f, 0, 0 \\ 0 & 0 & 0, f, 0 \\ -f, 0, 0 & 0, -f, 0 & 0 \end{pmatrix}. \quad (8)$$

For the Fe2-Fe2 and Fe1-Fe2 case, we derived the exactly same form in $\sum_{ij} P_{ij}^{\alpha\beta\gamma}$. Then Eq. (7) can only be expressed in a form similar to the spin-current mechanism:

$$p_{\text{Fe1-Fe1}} \propto (0, 0, f) \times (S_1 \times S_2), \quad (9)$$

where $(0, 0, f)$ is a vector along z . For $H = 0$, $H//x$, and $H//y$, $(S_1 \times S_2) = 0$ or $//z$ so that $p_{\text{Fe1-Fe1}} = 0$. For $H//z$, the $(S_1 \times S_2)//y$ and $//x$ for $2/m'$ and $2/m$ point group,

respectively. Then, $p_{\text{Fe1-Fe1}}$ should be along x and y direction, respectively, corresponding to nonzero α_{xz} and α_{yz} ME coefficients in $2/m'$ and $2/m$, respectively, in Eqs. (1) and (2). The same argument could be applicable to Fe2-Fe2 interactions. However, for Fe2-Fe1, the two spins are the same due to the ferromagnetic along the c axis, leading to zero net polarization. Therefore, we can draw the conclusion that only spin-current mechanism can contribute to α_{xz} or α_{yz} by Fe1-Fe1 and Fe2-Fe2 interactions. Exchange-striction and anisotropic-symmetric-exchange mechanisms [31] due to spin-orbital coupling can be completely ruled out in the linear ME effect from symmetry analysis [Eq. (9)].

As for the single-spin induced polarization p_{Fe} , it can be expressed as

$$p_{\text{Fe}} = \sum_i P_{ii}^{\alpha\beta\gamma} S_{1\alpha} S_{1\beta} + \sum_j P_{jj}^{\alpha\beta\gamma} S_{2\alpha} S_{2\beta}, \quad (10)$$

where S_1 spin resides on the i sites and S_2 spin on the j sites. After considering all the 3_z , 2_y , and -1 symmetric operations, $\sum_i P_{ii}^{\alpha\beta\gamma}$ and $\sum_j P_{jj}^{\alpha\beta\gamma}$ can be simplified as

$$\sum_i P_{ii}^{\alpha\beta\gamma} = A \begin{pmatrix} g, 0, h & 0, -g, 0 & k, 0, 0 \\ 0, -g, 0 & -g, 0, h & 0, k, 0 \\ k, 0, 0 & 0, k, 0 & 0, 0, l \end{pmatrix},$$

$$\sum_j P_{jj}^{\alpha\beta\gamma} = -A \begin{pmatrix} g, 0, h & 0, -g, 0 & k, 0, 0 \\ 0, -g, 0 & -g, 0, h & 0, k, 0 \\ k, 0, 0 & 0, k, 0 & 0, 0, l \end{pmatrix}, \quad (11)$$

where g, h, k, l , and A are constants. Comparing to Eq. (6) of the d - p hybridization mechanism, g, h, k, l can be nonzero. One could immediately find that $p_{\text{Fe}} = 0$ at $H = 0$. After applying $H//x, y$, or z , the electric dipole from two spin sites will no longer cancel each other out due to $S_1 + S_2 \neq 0$, leading to all allowed nonzero ME coefficients in Eqs. (1) and (2). Therefore, the spin-current mechanism is responsible for one of the ME coefficients, while the d - p hybridization mechanism can be the origin of every ME coefficient allowed in Eqs. (1) and (2) between T_{str} and T_N . Due to the lower crystal symmetry at $T < T_{\text{str}}$, the mathematical expressions for two-spin and single-spin ME tensor become very complex

and cannot be simplified further. It is possible that all three main mechanisms are the microscopic origins of linear ME effects in FNO.

In conclusion, we have investigated the crystal and magnetic structures of the recently discovered magnetoelectric materials $\text{Fe}_4\text{Nb}_2\text{O}_9$ by means of neutron-powder diffraction at low temperatures. We have successfully established the antiferromagnetic spin structure below $T_N \approx 94$ K, which is featured by antiparallel alignment of Fe moments with respect to its three nearest neighbors within the buckled honeycomb layers. Our results also reveal a structural transition from trigonal $P\text{-}3c1$ to monoclinic $C2/c$ upon cooling across $T_{\text{str}} \approx 80$ K, which should be responsible for the observed second dielectric anomaly. The magnetic structure is found to be preserved in the low-temperature monoclinic phase, and the Fe magnetic moment increases from $2.1(1)\mu_B$ at 95 K to $3.83(4)\mu_B$ at 10 K assuming an equal moment configuration. We have determined the magnetic point group and linear

magnetoelectric tensor at each temperature region, and proved that the microscopic origins of the ME effects between T_N and T_{str} are the spin-current and d - p hybridization mechanisms based on a symmetry-related tensor analysis.

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- [1] T. Kimura, T. Goto, H. Shintani, K. Ishizaka, T. Arima, and Y. Tokura, *Nature (London)* **426**, 55 (2003).
- [2] N. Hur, S. Park, P. A. Sharma, J. S. Ahn, S. Guha, and S.-W. Cheong, *Nature (London)* **429**, 392 (2004).
- [3] T. Kimura, Y. Sekio, H. Nakamura, T. Siegrist, and A. P. Ramirez, *Nat. Mater.* **7**, 291 (2008).
- [4] Y. Tokunaga, N. Furukawa, H. Sakai, Y. Taguchi, T. Arima, and Y. Tokura, *Nat. Mater.* **8**, 558 (2009).
- [5] D. N. Astrov, *Zh. Eksp. Teor. Fiz. (U.S.S.R.)* **40**, 1035 (1961) [*Sov. Phys. JETP* **13**, 729 (1961)].
- [6] A. Iyama, Y. Wakabayashi, N. Hanasaki, and T. Kimura, *Jpn. J. Appl. Phys.* **53**, 05FB02 (2014).
- [7] J. Hwang, E. S. Choi, H. D. Zhou, J. Lu, and P. Schlottmann, *Phys. Rev. B* **85**, 024415 (2012).
- [8] N. Mufti, G. R. Blake, M. Mostovoy, S. Riyadi, A. A. Nugroho, and T. T. M. Palstra, *Phys. Rev. B* **83**, 104416 (2011).
- [9] A. Maignan and C. Martin, *Phys. Rev. B* **97**, 161106(R) (2018).
- [10] S. N. Panja, L. Harnagea, J. Kumar, P. K. Mukharjee, R. Nath, A. K. Nigam, and S. Nair, *Phys. Rev. B* **98**, 024410 (2018).
- [11] Y. Fang, Y. Q. Song, W. P. Zhou, R. Zhao, R. J. Tang, H. Yang, L. Y. Lv, S. G. Yang, D. H. Wang, and Y. W. Du, *Sci. Rep.* **4**, 3860 (2014).
- [12] N. D. Khanh, N. Abe, H. Sagayama, A. Nakao, T. Hanashima, R. Kiyonagi, Y. Tokunaga, and T. Arima, *Phys. Rev. B* **93**, 075117 (2016).
- [13] Y. P. Lu, C. X. Ji, Y. L. Sun, Y. Fang, L. Zhang, Z. D. Han, B. Qian, X. F. Jiang, and W. P. Zhou, *J. Alloys Compd.* **679**, 213 (2016).
- [14] Y. M. Xie, C. S. Lin, H. Zhang, and W. D. Cheng, *AIP Adv.* **6**, 045006 (2016).
- [15] A. Maignan and C. Martin, *Phys. Rev. Mater.* **2**, 091401(R) (2018).
- [16] N. D. Khanh, N. Abe, S. Kimura, Y. Tokunaga, and T. Arima, *Phys. Rev. B* **96**, 094434 (2017).
- [17] B. B. Liu, Y. Fang, Z. D. Han, S. M. Yan, W. P. Zhou, B. Qian, D. H. Wang, and Y. W. Du, *Mater. Lett.* **164**, 425 (2016).
- [18] E. F. Bertaut, L. Corliss, F. Forrat, R. Aleonard, and R. Pauthenet, *J. Phys. Chem. Solids* **21**, 234 (1961).
- [19] A. S. Wills, *Physica B* **276–278**, 680 (2000).
- [20] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevB.100.094109> for the decomposition of the magnetic representation and the possible magnetic basis vectors for determining the magnetic structure at 95 and 10 K, as well as the refined structural parameters at different temperatures.
- [21] Y. Cao, M. Xiang, Z. Feng, B. Kang, J. Zhang, N. Guiblin, W. Ren, B. Dkhil, and S. Cao, *RSC Adv.* **7**, 13846 (2017).
- [22] Y. Cao, K. Xu, Y. Yang, W. Yang, Y. Zhang, Y. Kang, X. He, A. Zheng, M. Liu, S. Wei, Z. Li, and S. Cao, *J. Cryst. Growth* **492**, 35 (2018).
- [23] G. Deng, Y. Cao, W. Ren, S. Cao, A. J. Studer, N. Gauthier, M. Kenzelmann, G. Davidson, K. C. Rule, J. S. Gardner, P. Imperia, C. Ulrich, and G. J. McIntyre, *Phys. Rev. B* **97**, 085154 (2018).
- [24] S. Chaudhary, P. Srivastava, S. D. Kaushik, V. Siruguri, and S. Patnaik, *J. Magn. Magn. Mater.* **475**, 508 (2019).
- [25] R. E. Newnham, *Properties of Materials: Anisotropy, Symmetry, Structure* (Oxford University Press, New York, USA, 2005).
- [26] H. Katsura, N. Nagaosa, and A. V. Balatsky, *Phys. Rev. Lett.* **95**, 057205 (2005).
- [27] S. Greenwald and J. S. Smart, *Nature (London)* **166**, 523 (1950).
- [28] T.-H. Arima, *J. Phys. Soc. Jpn.* **76**, 073702 (2007).
- [29] H. J. Xiang, E. J. Kan, Y. Zhang, M.-H. Whangbo, and X. G. Gong, *Phys. Rev. Lett.* **107**, 157202 (2011).
- [30] Y. S. Chai, S. H. Chun, J. Z. Cong, and K. H. Kim, *Phys. Rev. B* **98**, 104416 (2018).
- [31] J. S. Feng and H. J. Xiang, *Phys. Rev. B* **93**, 174416 (2016).