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Magnetic coupling between A' and B sites in the A-site-ordered perovskite BiCu₃Mn₄O₁₂

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The ionic states of Cu and Mn ions and the magnetic structure in an *A*-site-ordered perovskite BiCu₃Mn₄O₁₂ were investigated by powder neutron diffraction and soft x-ray absorption and magnetic circular dichroism spectroscopy experiments. A substitution by Mn³⁺ was found at the square-planar *A'* Cu²⁺ site, leading to a composition Bi(Cu²⁺_{0.8}Mn³⁺_{0.2})₃Mn^{3.6+}₄O₁₂. This compound is a ferrimagnet with a collinear spin configuration below T_C =350 K, and the magnetic structure is stabilized by a strong ferromagnetic coupling between the *A'* and *B* site Mn ions and an antiferromagnetic coupling between Cu and Mn ions, leading to a near zero net moment at the *A'* site.

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I. INTRODUCTION

Perovskite oxides containing transition metal (TM) ions at the A site of the ABO_3 structure are quite rare. This is mainly because TM ions are too small to be stabilized at the usually 12-fold A site. However, particular types of distortions in the BO₆ octahedron network produce quasisquare-planar coordination at some part of the A sites that can incorporate Jahn-Teller (JT) active TM ions.¹ An example of such an A-site-ordered perovskite is $AA'_{3}B_{4}O_{12}$, in which 3/4 of the A sites in an \hat{ABO}_3 perovskite (denoted as the A' site) are occupied by JT ions such as Cu^{2+} and Mn^{3+} , with a $2a_0$ $\times 2a_0 \times 2a_0$ cubic unit cell of Im-3 symmetry (a₀: lattice constant of the perovskite prototype), as illustrated in Fig. 1. The presence of the TM ions at the A' sites is expected to induce A' - A' and A' - B interactions as well as the B - B interaction seen in many perovskite oxides, which give rise to novel physical properties in the $AA'_{3}B_{4}O_{12}$ compounds.

Compounds with this characteristic structure show wide variety of properties. Cu 3d electrons in the insulating $CaCu_3B_4O_{12}$ (B=Ti,Ge,Sn) are localized as A'-site S=1/2 spins. CaCu₃Ti₄O₁₂ exhibits temperature-independent colossal dielectric constant² and the B=Ge, Sn materials are very rare ferromagnetic cuprates.³ For compounds with Ru or V at the B site, in contrast, Cu 3d electrons are itinerant and the materials show metallic behavior.^{4,5} Metallic CaCu₃Ru₄O₁₂ is reported to exhibit a heavy-fermionlike behavior without f electrons.⁴ CaCu₃Mn₄O₁₂ is a ferrimagnet with a Curie temperature above room temperature, and its ferrimagnetism is explained by a ferrimagnetic ordering of the A'-site Cu and B-site Mn spins.⁶ Conduction carriers doped by substituting La^{3+} or Bi^{3+} for Ca^{2+} induce metallic conductivity and a large magnetoresistance effect at very low magnetic fields.^{7,8} A first-principles band-structure calculation revealed that the substitution caused mixed valence states in Mn in $LaCu_3Mn_4O_{12}$ and $BiCu_3Mn_4O_{12}$, leading to half metallic states.

In this paper, we have examined the ionic states of Cu and Mn ions in $BiCu_3Mn_4O_{12}$ and their magnetic interaction.

Soft x-ray absorption and magnetic circular dichroism spectroscopy (XAS-MCD) gave a direct evidence of antiferromagnetic coupling between the A'-site Cu and B-site Mn spins. We will also discuss the magnetic structure of the compound from the analysis of neutron powder-diffraction (NPD) data.

II. EXPERIMENTS

A polycrystalline sample of BiCu₃Mn₄O₁₂ was prepared using a high-pressure technique, as described in Ref. 8. The obtained sample was almost single phase but it included a small amount of CuO impurity. As shown in Fig. 2, a large magnetization of the sample at low temperature was confirmed by a superconducting quantum interference device magnetometer (Quantum Design, magnetic property measurement system) measurement. The magnetic transition temperature T_C determined from the magnetic susceptibility data in a magnetic field of 0.1 T was 350 K. Assuming the stoichiometric composition, saturation magnetization of 10.6 $\mu_B/f.u.$ was obtained in a magnetic field of 5 T at 5 K, in good agreement with the previous report.⁸

The XAS-MCD measurements at the Cu $L_{2,3}$ and Mn $L_{2,3}$ edges were carried out at 9 K, by a total electron yield



FIG. 1. (Color online) Crystal structure of the A-site-ordered perovskite $AA'_{3}B_{4}O_{12}$.



FIG. 2. (Color online) Isothermal magnetization of BiCu_{2.4}Mn_{4.6}O₁₂ at 5 K and the temperature dependence of the dc magnetic susceptibility in a magnetic field of 0.1 T (inset), both assuming stoichiometric composition. The saturation magnetization was corrected to be 12.1 $\mu_{\rm B}$ /f.u. considering the off-stoichiometry, as described in the text.

method at the soft x-ray beamline BL25SU of SPring-8 in Japan. The energy resolution $E/\Delta E$ was greater than 5000 during the measurements. The incident photon energy was calibrated by measuring the energies of the Ti $L_{2,3}$ edges of TiO₂ and the Ni $L_{2,3}$ edges of NiO. Powder sample was pasted uniformly on a sample holder by using a carbon tape, and a magnetic field of 2 T was applied during the measurement. The MCD was obtained by altering the direction of the magnetic field with respect to the spin of the circularly polarized soft x-ray. The MCD spectrum is defined as I_+-I_- , where I_+ and I_- represent the absorption intensity with the direction of the magnetization being parallel and antiparallel to the photon spin, respectively.

NPD patterns were collected in the temperature range from 5 to 400 K on the high-resolution D2B diffractometer of the Institute Laue-Langevin in Grenoble. About 1 g of powder sample was used and good quality patterns were collected with the high-flux mode. The counting time was 6 h for the 400 and 5 K patterns, and 4 h for the rest. A wavelength of 1.594 Å was used and the data were analyzed by the Rietveld method, using the program GSAS.⁹

III. RESULTS AND DISCUSSION

The NPD patterns above $T_{\rm C}$ were well reproduced with an A-site-ordered structure model with a space group Im-3. Here Bi atoms were placed at 2a (0,0,0) positions, Cu at 6b(0, 1/2, 1/2)(A' site), Mn at 8c (1/4, 1/4, 1/4)(B site), and O at 24g (0, y, z). The result of the Rietveld refinement at 400 K is shown in Fig. 3 and the refined parameters are listed in Table I. The fit improved considerably with a reduction in $R_{\rm wp}$ from 6.25% to 4.96%, by partially incorporating Mn at the A' site. Incorporating Cu at the B site in the refinement did not give significant improvement. We note that the coherent neutron-scattering length of Mn is negative (-3.73 fm) while that of Cu is positive (7.69 fm) so that neutron diffraction is very sensitive for distinguishing Mn and Cu. Oxygen vacancies were not detected in the refinement of the oxygen site occupancy. Thus, the



FIG. 3. (Color online) Observed (+), calculated (full line), and difference (bottom) NPD Rietveld profiles for $BiCu_{2,4}Mn_{4,6}O_{12}$ at (a) 400 K and (b) 5 K. The bottom row of tick marks corresponds to the small CuO impurity in both patterns and the middle row at 5 K corresponds to the magnetic structure.

obtained chemical formula was Bi(Cu_{0.8}Mn_{0.2})₃Mn₄O₁₂ (BiCu_{2.4}Mn_{4.6}O₁₂). The additional Mn³⁺ included in this compound is also a JT ion, and is known to be incorporated into the A' site, as in NaMn₃Mn₄O₁₂ or LaMn₃Cr₄O₁₂.^{10,11} Such off-stoichiometry is also reported in CaCu₃Mn₄O₁₂.¹² The off-stoichiometry is also consistent with the observations of CuO (5.7 wt %) and Bi₂(CO₃)O₂ impurities⁸ detected in the present neutron and x-ray (data is not shown) diffraction data. Assuming JT active Cu²⁺ and Mn³⁺ at the square-planar coordinated A' site, the refined chemical composition gives a nominal valence of 3.6+ for Mn at the B site The oxidation states of the cations in (Mn_R) . BiCu_{2.4}Mn_{4.6}O₁₂ were also assessed by means of Brown's bond valence sum (BVS) calculation,¹³ using the observed metal-oxygen distances at room temperature. No structural transition was observed at magnetic $T_{\rm C}$ and the cubic Im-3 crystal structure was kept intact below $T_{\rm C}$. The refined structural parameters at 300 K are listed in Table I, and selected bond distances and bond angles are listed in Table II. The BVS calculated from the bond distances were 2.22 for $Cu_{A'}$ and 2.76 for $Mn_{A'}$ (both based on the average $Cu_{A'}/Mn_{A'}$ -O distances), 3.61 for Mn_B, and 3.12 for Bi. The results are fully with the consistent nominal composition $Bi^{3+}(Cu^{2+}_{0.8}Mn^{3+}_{0.2})_3Mn^{3.6+}_{4}O_{12}$. (The small deviation of the BVS for $Cu_{A'}/Mn_{A'}$ from 2+/3+ should come from the usage of the average atomic distances.)

The ionic states of the A'-site Cu ions and the B-site Mn ions and their magnetic coupling were also examined in detail with XAS-MCD measurements. The Cu and

		400 K ^b	300 K	5 K
a (Å)		7.33130(8)	7.32352(8)	7.31188(7)
V (Å ³)		394.04(1)	392.79(1)	390.92(1)
Bi	$U_{\rm iso}$ (Å ²)	0.020(1)	0.018(1)	0.0109(9)
Cu/Mn	Cu occupancy	0.800(4)	0.8	0.8
	Mn occupancy	0.200(4)	0.2	0.2
	$U_{\rm iso}$ (Å ²)	0.0089(6)	0.0077(5)	0.0045(4)
	Magnetic moment (μ_B)	0	0	0.24(4)
Mn	$U_{\rm iso}$ (Å ²)	0.0098(5)	0.0096(6)	0.0068(5)
	Magnetic moment (μ_B)	0	1.60(9)	2.86(5)
0	у	0.3046(2)	0.3046(2)	0.3044(2)
	Z.	0.1810(2)	0.1807(2)	0.1808(2)
	$U_{\rm iso}$ (Å ²)	0.0103(2)	0.0092(2)	0.0059(2)
$R_{\rm wp}$ (%)		4.96	4.89	5.55
$R_{\rm p}$ (%)		3.45	3.69	3.80
$\dot{\chi^2}$		4.65	3.77	5.33

TABLE I. Representative refined structural par	arameters of BiCu ₂ Mr	146012 at 400.	. 300. and 5 K. ^a
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^aCubic space group *Im*-3 (No, 204) was adopted for all the crystal structural refinements, where the atomic positions were Bi 2*a* (0,0,0)(*A* site), Cu/Mn 6*b* $(0,\frac{1}{2},\frac{1}{2})(A' \text{ site})$, Mn 8*c* $(\frac{1}{4},\frac{1}{4},\frac{1}{4})(B \text{ site})$, and O 24*g* (0, y, z).

^bThe total occupancy of the A' site was constrained to unity in the refinement at 400 K.

Mn $L_{2,3}$ -edge XAS-MCD spectra for BiCu₃Mn₄O₁₂ at 9 K are shown in Fig. 4. The observed spectra correspond to $2p \rightarrow 3d$ excitations of Cu and Mn. Both Cu L_2 - and L_3 -edge structures in the XAS spectra are single peaks, resembling those of Cu^{2+} with d^9 electronic configuration and squareplanar coordination, in CaCu₃Ti₄O₁₂.¹⁴ This suggests that Cu in $BiCu_3Mn_4O_{12}$ is divalent. The MCD intensities for the Cu L_2 and L_3 edges are evident, confirming that Cu²⁺ spins at the A' sites are magnetically ordered. On the other hand, the Mn L_2 - and L_3 -edge structures do not resemble those of pure Mn³⁺ or Mn⁴⁺, which is consistent with the averaged valence of 3.6+ for Mn calculated from the composition. Mn spins are also magnetically ordered from the MCD spectrum. An important observation here is that the signs of the MCD intensities of Cu L edges are opposite to those of the corresponding Mn L edges. This clearly demonstrates that the Cu and Mn spins are coupled antiferromagnetically. The total magnetic moment of Cu estimated from the XAS-MCD sum rules^{15–17} was 1.16 $\mu_{\rm B}$, which is in a good agreement with the value expected for divalent Cu. Here contribution from the magnetic dipole operator $\langle T_z \rangle$ was neglected and the number of the Cu 3d electrons was assumed as 9. The magnetic moment of the Mn could not be evaluated from the XAS-MCD spectra because the L_2 - and L_3 -edge signals over-

The magnetic structure below $T_{\rm C}$ was determined from the Rietveld analyses of the NPD patterns. The patterns from 5 to 400 K are shown in Fig. 5. Large increases in intensity were found in some reflections such as 020 and 022 below $T_{\rm C}$, corresponding to the evolution of the ordered magnetic moment. The absence of any additional reflection peaks to the *Im*-3 symmetry indicates ferromagnetic alignment at both Cu and Mn sites. Considering the antiferromagnetic coupling of the Cu and Mn spins revealed by the XAS-MCD measurements, the magnetic structure can be described as a ferrimagnetic order of Cu and Mn spins. Magnetic moments at both A' site $(M_{A'})$ and B sites (M_B) were refined for the patterns below 320 K. The result of the refinement of the pattern at 5 K is shown in Fig. 3 and summarized in Table I. Since the absolute orientation of the magnetic moments in a cubic structure cannot be determined by the NPD technique, both $M_{A'}$ and M_B were set to the [001] direction in the refinements. $M_{A'}$ was found to be very small and could only be successfully refined at 5 and 100 K although there is no

TABLE II. Selected bond distances, bond angles, and BVS of $BiCu_{2,4}Mn_{4,6}O_{12}$ at 400, 300 and 5 K. a,b

	400 K	300 K	5 K
d(Bi-O) (Å) [×12]	2.598(2)	2.593(2)	2.589(2)
$d(\operatorname{Cu}_{A'}/\operatorname{Mn}_{A'}-\operatorname{O})$ (Å) [×4]	1.953(1)	1.949(1)	1.948(1)
$d(\operatorname{Cu}_{A'}/\operatorname{Mn}_{A'}-\operatorname{O})$ (Å) [×4]	2.743(2)	2.742(1)	2.737(2)
$d(\operatorname{Cu}_{A'}/\operatorname{Mn}_{A'}-\operatorname{O})$ (Å) [×4]	3.233(1)	3.232(1)	3.225(1)
$d(Mn_B-O)$ (Å) [×6]	1.9429(3)	1.9415(4)	1.9380(3)
Cu-O-Cu (deg)	101.32(6)	101.28(6)	101.25(6)
Mn-O-Mn (deg)	141.24(6)	141.13(6)	141.20(6)
Cu-O-Mn (deg)	109.14(3)	109.19(3)	109.15(3)
BVS(Bi)	3.07	3.12	3.15
$BVS(Cu_{A'})$	2.19	2.22	2.22
$BVS(Mn_{A'})$	2.73	2.76	2.77
$BVS(Mn_B)$	3.59	3.61	3.64

^aBVS of Cu/Mn at the A' site $BVS(Cu_{A'})/BVS(Mn_{A'})$ are both based on the average A'-O distances.

^bThe BVS parameters for trivalent/tetravalent Mn were used for $Mn_{A'}/Mn_B$.



FIG. 4. (Color online) (a) Cu $L_{2,3}$ -edges and (b) Mn $L_{2,3}$ -edges XAS-MCD spectra for BiCu_{2,4}Mn_{4.6}O₁₂ taken at 9 K. Solid (I_+) and dashed (I_-) lines show the XAS spectra measured with external field (2 T) applied parallel and antiparallel to the light propagation axis. The bottom line represents the MCD intensity (I_+ - I_-).

magnetization evidence for a second magnetic transition below $T_{\rm C}$. The refined magnetic moments for the two sites at 5 K were found to be parallel with values, $M_{A'}$ =0.24(4) $\mu_{\rm B}$ and M_B =2.86(5) $\mu_{\rm B}$, giving a total magnetic moment of $M_{\rm tot}$ =12.2(3) $\mu_{\rm B}/f.u$. The refined total magnetic moment seems to be considerably large compared to the observed magnetization of 10.6 $\mu_{\rm B}/f.u$. in the magnetic measurement. However, if we consider the off-stoichiometric composition and the amounts of impurities as we discussed before, the saturation magnetization is corrected to be 12.1 $\mu_{\rm B}/f.u.$,¹⁸ which is quite close to the refined value. Figure 6 shows the temperature dependence of the refined magnetic moments $M_{A'}$, M_B , and $M_{\rm tot}$. The thermal evolution of $M_{\rm tot}$ agrees well with the M/H behavior shown in Fig. 2.

The small positive value for $M_{A'}$ appears anomalous considering the antiferromagnetic coupling between the A'-site Cu spins and the *B*-site Mn spins evidenced by MCD. This is most likely due to the presence of Mn³⁺ ions at the A' site. With the off-stoichiometric model Bi(Cu²⁺_{0.8}Mn³⁺_{0.2})₃Mn^{3.6+}₄O₁₂ obtained above, the 80% Cu²⁺ (S=1/2) and 20% Mn³⁺ (S=2) spins at the A' sites each produce an ideal moment contribution of 0.8 $\mu_{\rm B}$ per site. The observed average A'-site moment of ~+0.2 $\mu_{\rm B}$ demonstrates an antiferromagnetic coupling of Cu²⁺ and Mn³⁺ moments at the A' sites, resulting in an almost complete cancellation of their moments. The lack of pure magnetic reflections indicates ferromagnetic alignment at each sublattices. Thus the magnetic structure can be described as



FIG. 5. (Color online) Thermal evolution of the low-angle region of the neutron powder-diffraction pattern of $BiCu_{2.4}Mn_{4.6}O_{12}$. Inset shows the magnetic structure below $T_{\rm C}$.

illustrated in the inset of Fig. 5, where the A'-site Cu²⁺ spins couple antiferromagnetically with both the A'-site Mn³⁺ and the *B*-site Mn^{3,6+} spins. The saturation magnetization expected for Bi(Cu²⁺_{0.8}Mn³⁺_{0.2})₃Mn^{3,6+}₄O₁₂ with such magnetic structure is 13.6 $\mu_{\rm B}$ /f.u., close to the measured one.

Since Mn_B in BiCu_{2.4} $Mn_{4.6}O_{12}$ have mixed 3+/4+ oxidation states, a strong ferromagnetic double exchange interaction is expected as in other AMnO₃ perovskites with mixed valence $Mn^{3+/4+}$, which should be responsible for the metallic conduction below $T_{\rm C}$, as well as the large magnetoresistance. The ferromagnetic coupling between the A'- and B-site Mn spins suggests that some A'-B double exchange interactions also occur. The Cu²⁺ spins are localized and do not participate in the double exchange, and so couple through antiferromagnetic superexchange to their Mn_{R} neighbors. Direct Cu²⁺-Cu²⁺ interaction in the A-site-ordered perovskites are very weak as the CuO₄ square-planar units are separated from each other.3 Hence the magnetism of BiCu_{2.4}Mn_{4.6}O₁₂ is dominated by the strong ferromagnetic Mn-Mn and antiferromagnetic Cu-Mn interactions. The mixed 3+/4+ oxidation state in Mn_B and the ferrimagnetic coupling between the A'-site Cu and B-site Mn produce a half metallic electronic structure, where only up-spin bands cross the Fermi level below $T_{\rm C}$. The spin-polarized conduction carriers are responsible for the large magnetoresistance in BiCu_{2.4}Mn_{4.6}O₁₂.



FIG. 6. (Color online) Temperature variation in the ordered magnetic moments for A' site $(M_{A'})$ and B site (M_B) , and the total magnetic moment per formula unit (M_{tot}) .

IV. CONCLUSIONS

The cation off-stoichiometry and magnetic order within the A-site-ordered double perovskite BiCu₃Mn₄O₁₂ have been established using NPD and MCD spectroscopy. Offstoichiometry originated from the substitution of Mn³⁺ for Cu²⁺ at the A' sites, leading to the chemical composition of Bi(Cu²⁺_{0.8}Mn³⁺_{0.2})₃Mn^{3.6+}₄O₁₂, as confirmed by the XAS spectroscopy and BVS calculations. A ferrimagnetic ordering between the A'-site Cu²⁺ and B-site Mn^{3+/4+} spins was obtained from the MCD and NPD experiments with the Mn^{3+/4+} and Cu²⁺ spins ordering simultaneously at T_C. However, the antiparallel alignment of Mn³⁺ and Cu²⁺ moments at the A' sites results in a very small net moment. The magnetism of the compound is dominated by the strong ferromagnetic Mn-Mn and antiferromagnetic Cu-Mn interactions.

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- 18 The stoichiometric starting composition produces 87.1 wt % BiCu_{2.4}Mn_{4.6}O₁₂, 9.0 wt % CuO, and 3.9 wt % Bi₂(CO₃)O₂. The saturated magnetization of BiCu_{2.4}Mn_{4.6}O₁₂ (molar weight=806.20 g/mol) is corrected with the assumption of those amounts of impurities.