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Magnetization-control and transfer of spin-polarized Cooper pairs into a half-metal manganite

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12 The pairing state and critical temperature (T_c) of a thin s-wave superconductor (S) on two or more 13 ferromagnets (F) are controllable through the magnetization-alignment of the F layers. 14 Magnetization misalignment can lead to spin-polarized triplet pair creation, and since such triplets 15 are compatible with spin-polarized materials they are able to pass deeply into the F layers and so, 16 cause a decrease in T_c . Various experiments on $S/F_1/F_2$ "triplet spin-valves" have been performed 17 with the most pronounced suppression of T_c reported in devices containing the half-metal 18 ferromagnet (HMF) CrO₂ (F₂) albeit using out-of-plane magnetic fields to tune magnetic non-19 collinearity [Singh et al., Phys. Rev. X 5, 021019 (2015)]. Routine transfer of spin-polarized triplets to 20 HMFs is a major goal for superconducting spintronics so as to maximize triplet-state spin-21 polarization. However, CrO₂ is chemically unstable and out-of-plane fields are undesirable for 22 superconductivity. Here, we demonstrate low field (3.3 mT) magnetization-tuneable pair conversion 23 and transfer of spin-polarized triplet pairs to the chemically stable mixed valence manganite 24 La_{2/3}Ca_{1/3}MnO₃ in a pseudo spin-valve device using in-plane magnetic fields. The results match 25 26 27 microscopic theory and offer full control over the pairing state.

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I. **INTRODUCTION**

31 Superconducting spintronics represents a new paradigm for information processing involving the 32 coexistence of spin-polarization and superconducting phase coherence [1–3]. Conventional s-wave 33 superconductivity involves the condensation of spin-singlet electron pairs with antiparallel spins. 34 Although singlet pairs are energetically unstable in a ferromagnet, they are able to penetrate a 35 transition metal ferromagnet (F) at a superconductor/ferromagnet (S/F) interface over distances of a 36 few nanometers [4–10], but without transferring a net spin. Furthermore, singlet pairs are blocked at 37 a S interface with a half metallic ferromagnet (HMF) as there are no available states for one of the 38 two spins of a pair to enter since the Fermi energy for the minority-spin electrons falls within a gap.

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40 Electrons pairs in the *p*-wave superconducting compound Sr₂RuO₄ [11] have parallel spins and so 41 such spin-triplet pairs carry a net spin in addition to charge. However, the extreme sensitivity of p-42 wave superconductivity to structural and electronic disorder, creates major obstacles to the 43 development of *p*-wave devices [12]. Spin-triplet pairs with parallel spins, but *s*-wave symmetry may 44 form at magnetically inhomogeneous s-wave S/F interfaces [1–3]. Since such pairs are compatible 45 with fully spin-polarized materials, their routine creation and transfer to HMFs would open up 46 exciting opportunities for applications in superconducting spintronics where high spin-polarization 47 and long spin-flip scatter lengths are desirable.

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49 Spin-polarized triplet pairs form via spin mixing and spin-rotation processes at S/F interfaces [13]. At 50 homogeneously magnetized S/F interfaces or within magnetically collinear $S/F_1/F_2$ spin-valves, spin-51 singlet pairs experience a spatially constant exchange field that acts differentially on the antiparallel 52 spins of a pair, causing transformation to a spin-zero triplet state (spin-mixed state). A rotation of the 53 magnetization at a S/F interface or within a $S/F_1/F_2$ spin-valve has the effect of transforming spin-54 zero triplets to pairs with a parallel projection of spin (spin-rotation). For $S/F_1/F_2$ spin-valves where S 55 and F_1 ("spin-mixer" layer) are thinner than the spin-singlet coherence length (40 nm in Nb [14] and 1 56 nm in Co, Fe and Ni [15, 16]), spin-polarized triplet pair creation leads to an effective leakage of 57 superconductivity from S into F_2 and a reduction of the critical temperature (T_c). "Triplet spin-valves" 58 (TSVs) are therefore sensitive devices for investigating singlet-to-triplet pair conversion [17–20].

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60 Experiments over the past few years have mainly focused on magnetization-control of triplet pair 61 creation in *S/F/S* Josephson devices and TSVs. In *S/F/S* devices various symmetric spin-mixer layers 62 have been added to the *S/F* interfaces, including rare earth magnetic spirals [21, 22], 63 antiferromagnets [23], Heusler alloys [24], and transition metal ferromagnets [25–29]. Similarly in 64 *S/F*₁/*F*₂ TSVs, *F*_{1,2} metals [30–33] or *F* metals (*F*₁) in combination with the HMF CrO₂ (F₂) [34] have 65 been successfully demonstrated. See also related works on *F/S/F* spin-valves [35–37] and 66 spectroscopy experiments on various S/F systems experiments [38–48].

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68 The most pronounced suppressions of T_c was reported in a MoGe/Ni/Cu/CrO₂TSV in which out-of-69 plane magnetic fields created a misalignment between the magnetizations of Ni and CrO₂ [34]; the 70 largest suppression of T_c was close to -800 mK with a constant out-of-plane magnetic field of 2 T. This 71 pioneering work extended previous experiments that demonstrated Josephson coupling across CrO₂ 72 [49] (see also [27,35]) in devices that did not contain intentional spin-mixer layers at the S/HMF 73 interfaces. However, CrO_2 is chemically unstable and so there is a need to identify alternative HMFs 74 in which thin films can be grown and combined with various S/F structures with enhanced chemical 75 stability.

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77 Mixed valance manganites ($La_{1-x}Ae_xMnO_3$, where Ae is an alkaline earth) such as $La_{1-x}Sr_xMnO_3$ (LSMO) 78 and $La_{2/3}Ca_{1/3}MnO_3$ (LCMO) are highly attractive alternatives to CrO_2 since they are chemically stable 79 and their relatively narrow spin up and spin down conduction bands are completely separated leading 80 to HMF behaviour at low temperatures[50, 51]. In this Article, we report TSV with 81 Nb/Cu/Py/Au/LCMO layers in which a non-monotonic dependence of T_c on the relative magnetization 82 angle (θ) between Py(NiFe) and LCMO is observed, thus demonstrating pair conversion and transfer 83 of spin-polarized triplets to LCMO. Recently, we detected Josephson coupling across thin (< 30 nm) 84 layers of LCMO [52], but without intentional spin-mixers at the S/LCMO interfaces. Related 85 experiments that probe spectroscopic signatures triplet pairing in S/LCMO structures have also been 86 reported [42–44, 53], but again without intentional spin-mixer layers. The motivation of the work 87 reported here was to investigate magnetization-control of triplet pair creation and transfer to LCMO, 88 which is fundamental to the development triplet superconductivity based on mixed valance 89 manganites. Furthermore, we wanted to demonstrate triplet pair creation in TSVs with small in-plane 90 magnetic fields to avoid complications due voritices that will be present in TSV that require large out-91 of-plane magnetic fields

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II. EXPERIMENT

95 We prepared Nb(25nm)/Cu(5nm)/Py(3.5nm)/Au(5nm)/LCMO(120nm) TSVs in several stages. Epitaxial 96 (002) LCMO was grown from a stoichiometric target by pulse laser deposition (PLD) (KrF laser, 97 wavelength λ = 248 nm) on 5 mm x 5mm single crystal SrTiO₃ (001) at a growth temperature of 800 °C 98 in flowing N₂O at 130 mTorr with a pulse fluence of 1.5 J/cm² for 15 minutes and repetition rate of 2 99 Hz, then 30 minutes at 3 Hz. The films were annealed *in situ* at the same temperature in oxygen (46 100 kPa) for 8 hours and cooled to room temperature at a rate of 10 °C/min. High resolution X-ray 101 diffraction (Fig. 1S) confirmed single (002) orientation of LCMO with rocking curves on the (002), 102 (004), (006) and (008) Bragg peaks showing full width at half maximum values of 0.12°, 0.18°, 0.209° 103 and 0.227°, respectively. The c-axis lattice parameter was determined to be 7.670±0.002 Å, consistent 104 with powder diffraction simulations [54]. Au was deposited on LCMO at room temperature using a 105 fluence of 2.5 J/cm² for 3 minutes at 5 Hz in 30 mTorr of Ar (Au was chosen due to its oxidation 106 resistance and limited solubility with Ni). Au/LCMO bilayers were then transferred in air to an 107 ultrahigh vacuum sputtering system with a base pressure of 3 x 10⁻⁹ mBar and Nb/Cu/Py trilayers 108 were deposited on Au/LCMO in Ar at 1.5 Pa while rotating below stationary magnetrons. The surface 109 of Au was cleaned in situ by Ar ion plasma etching (-0.6 kV extraction energy and 1 kV ion energy) and 110 different etching times in the 0-5 minute range were investigated. During the sputter process, 111 samples experienced a constant in plane magnetic field of approximately 50 mT.

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113 Control samples of Au(5nm)/LCMO(120nm) and Nb(25nm)/Cu(5nm)/Py(3.5nm)/Au(5nm) were 114 prepared on 5 mm x 5 mm area STO (001) and single crystal silicon substrates, respectively, to 115 characterize the isolated magnetic properties of LCMO and Py. Magnetization M versus applied field 116 H is shown in Fig. 1(a,b) at 10 K. The M(H) of LCMO shows an easy-plane behaviour with an in-plane 117 saturation field (H_s) of 50 mT and coercivity (H_c) of 20 mT. In the Supplemental Materials [55] we also 118 show that (Fig. 4S) the LCMO is magnetically isotropic in-plane at 10 K. In comparison, the Py shows 119 some in plane anisotropy with an easy axis (EA, defined as 90°) parallel to the field direction during 120 growth and H_c of 1.8 mT and a harder axis (HA, defined as 0°) at a right angle to the EA with H_c = 1.1 121 mT. The volume saturation magnetizations of LCMO and Py were 470±15 emu/cm³ and 650±25 122 emu/cm³ respectively, which are similar to values reported elsewhere [For LCMO see [56] and [57] for 123 Py].

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Figure 1(c) shows M(H) of the TSV at 10 K where M is dominated by the 120-nm-thick LCMO layer and so, for comparison easy-axis M(H) loop is plotted for the Nb/Cu/Py/Au control (reproduced from Fig. 1(b)). The M(H) loops show that the TSV magnetization state is parallel (P) beyond ±30 mT, and a reversal field of -1.8 mT switches the Py moment to achieve an antiparallel (AP) state.

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Resistance vs temperature R(T) measurements of the TSVs were performed using a four-point current-bias technique on unpatterned samples in a pulse-tube measurement system. The T_c was defined as the temperature corresponding to 50% of the normal state resistance. We note that care was taken to ensure that the bias-current (10 μ A) had no effect on R(T) through the superconducting transition and therefore that the T_c was current-bias independent (meaning the bias-current is not large enough for vortex-induced voltages to dominate the transport signal). In all cases, R(T) did not show anomalies (e.g. steps) through the superconducting transition.

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138 The effect of in-plane magnetization configuration on T_c was investigated by measuring R(T) though 139 the superconducting transition as a function of the relative magnetization angle (θ) between LCMO 140 and Py. The $T_{\rm c}(\theta)$ measurement routine is illustrated in Fig. 2(a) and described here: (1) at 10 K an 141 external field of 100 mT was applied along the HA of Py to magnetize LCMO and Py (along 0°); (2) a 142 magnetic field of -3.3 mT (<H_c of LCMO) was then applied along the HA of Py to reverse the Py 143 moment and obtain the AP-state (along 180°) and from R(T) in cooling and warming $T_{c}(180°)$ was 144 obtained; (3) the sample warmed to 10 K and rotated in-plane to an angle θ in a constant field of 145 amplitude 3.3 mT and from R(T) in cooling and warming $T_{c}(\theta)$ was obtained. Stage (3) was repeated 146 at 20° increments to obtain $T_{\rm C}(\theta)$ between 0° - 180°. We note that a field of -3.3 mT was large enough 147 to fully magnetize Py in all in-plane field directions without altering the remnant state of LCMO.

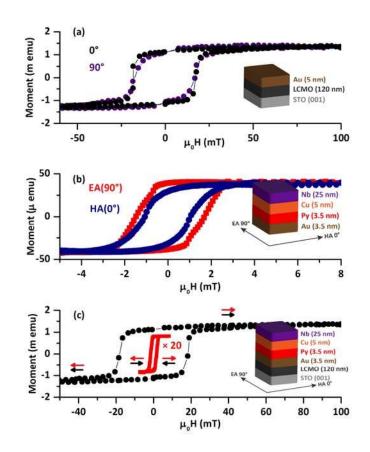


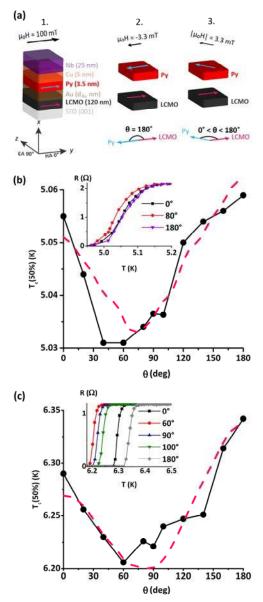
Fig. 1. (a) M(H) loops of LCMO for orthogonal in-plane fields at 10 K. (b) M(H) of Py with the field parallel to the easy axis (EA) and hard axis (HA). (c) M(H) loop of a complete TSV which is dominated by the magnetization from the 120-nm-thick LCMO and hence the Py loop (EA) reproduced from (b) is shown for comparison.

III. RESULTS & DISCUSSION

Figure 2(b) shows $T_{c}(\theta)$ for a TSV in which the Au layer has not been etched. Comparing P- and AP-states, we see a standard (albeit small) singlet spin-valve effect with $T_{\rm c}(AP)$ - $T_{\rm c}(P)$ close to 10 mK. For angles in the 0° < θ < 180° range, $T_{c}(\theta)$ decreases to a local minima of 5.32 K, close to θ = 60° giving a maximum T_c suppression (defined as $\Delta T_c(\theta) = T_c(AP) - T_c(\theta)$) of -28 mK, which is smaller than the average superconducting transition width. To check that $T_{\rm c}(\theta)$ cannot be attributed to potential effects arising from field non-uniformity on T_c as the TSV is rotated in-plane during measurements of R(T) (e.g. if the sample is not mounted perfectly parallel to the applied field), we investigated $T_{c}(\theta)$ of the Nb(25nm)/Cu(5nm)/Py(3.5nm)/Au(5nm) control sample with the field applied in-plane and tilted out-of-plane by 10° (see Fig. 2S). A maximum $\Delta T_{c}(\theta)$ of 10 mK (matching the temperature stability of our system) was observed with no dependence of T_c on θ , meaning that the functional form of $\Delta T_c \theta$) in Fig. 2(b) is related to the relative magnetizations of Py and LCMO and not field non-uniformity.

197 The small maximum value of $\Delta T_{c}(\theta)$ (-28 mK) seen in Fig. 2(b) indicates low interfacial transparency at 198 the Py/Au or Au/LCMO interfaces although we note that R(T) does not show anomalous features in 199 the superconducting transition, suggesting a homogeneous interfacial resistance (heterogeneous 200 transparency would result in currents paths changing direction through the superconducting 201 transition so as to preferentially flow in superconducting regions). To improve the Py/Au interface,

we Ar-ion etched the Au in situ prior to the sputter-deposition of Nb/Cu/Py and investigated $\Delta T_{c}(\theta)$ on etching time (the Au etch rate is 0.75 ± 0.04 nm/min). The largest $\Delta T_c(\theta)$ of -140 mK (Fig. 2(c)) was achieved for an etch time of 2 minutes with no observable dependence of T_c on θ for an etch time of 8 minutes. These data indicate that increasing the etch time has the effect of improving the interface transparency between Py and Au with ΔT_{c} (θ) increasing by 110 mK. Simultaneous, etching had the effect of enhancing the singlet spin-valve effect with $T_{\rm c}(AP)$ - $T_{\rm c}(P)$ increasing from 10 mK (without etching) to 40 mK after 2 minutes of etching (Fig. 3). Over etching the Au, however, risks introducing roughness and ferromagnetic coupling between Py and LCMO and so a decrease in $\Delta T_{c}(\theta)$ beyond a certain etch time is expected (as seen for an etch time of 8 minutes). We note that, we also investigated using Cu as an alternative to Au at the LCMO interface, but only a singlet spin-valve effect was observed ($T_c(AP) > T_c(P)$); see Supplemental Materials for further details.



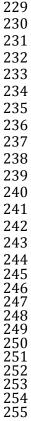


Fig. 2. (a) Measurement sequence to measure T_c as a function of θ . The blue and pink arrows show the likely magnetization configuration of Py and LCMO. (b) and (c) show example data of $T_c(50\%)$ vs θ for Nb(25nm)/Cu(5nm)/Py(3.5nm)/Au(d_{Au})/LCMO(120nm) TSVs without etching of Au ((b); d_{Au}=5nm) and following two minutes of etching ((c); d_{Au}=3.75nm). The dashed pink lines show the simulated values of $T_c(50\%)$. The insets shows selected R(T) transitions for various magnetization angles (labelled).

256 To compare our results to theory, we calculated $\Delta T_{c}(\theta)$ of the Nb/Cu/Py/Au/LCMO TSVs using a fully 257 microscopic procedure, based on numerical solutions to the self-consistent Bogoliubov-de Gennes 258 (BdG) equations, as extensively discussed in [18, 19, 58]. Each layer is assumed to be infinite in the y-259 z plane (see Fig. 2(a)). The four interfaces between Nb and LCMO will have differing transparencies 260 and to account for spin-independent scattering at these interfaces, we include repulsive delta 261 function potentials $H_i\delta(\mathbf{x}-\mathbf{x}_i)$ at each interface position x_i (where i = 1-4 refers to the interface number: 262 i = 1 corresponds to the Nb/Cu interface while i = 4 the Cu/LCMO interface). The scattering strength 263 is parameterized in dimensionless units by the quantity $H_{\rm Bir}$ written as $H_{\rm Bi}=mH_{\rm i}/k_{\rm F}$, where $k_{\rm F}$ is the 264 Fermi wavevector, and m is the effective mass. Thus, increasing H_{Bi} decreases the interface transparency^{18, 19}. To effectively characterize the TSV and maintain a tractable parameter space, it is 265 266 necessary to keep the scattering strength combinations as simple as possible. We found good 267 correlation with experiment when setting $H_{B1} = H_{B3} = 0.2$ for the Au/LCMO and Cu/Py interfaces 268 respectively. For the unetched TSV in (b), we assume a lower transparency at the Py/Au interface 269 with H_{B2} = 1.2, while the Nb/Cu interface is represented with H_{B4} = 0.14. Using these optimised 270 parameters, the model is able to capture the experimental $T_{\rm c}(\theta)$ behavior seen in Fig. 4 where the 271 local minima in T_c theoretically relates to the transfer of spin-polarized triplet pairs to LCMO (see also 272 Supplementary Material).

274 It is interesting to note that the experimental and theoretical minimum in $T_c(\theta)$ are shifted from the 275 orthogonal magnetic configuration (θ = 90°). Properly accounting for proximity effects can alter the 276 traditional simple view of the triplet spin valve, whereby the equal spin triplet components undergo 277 a maximum at 90 degrees (leading to a corresponding dip in T_c). By including interface scattering, the 278 quasiparticle amplitudes can undergo phase shifts that push the minimum in T_c away from 90°. The 279 same effect also arises in the ballistic regime [18] from the superposition of quasiparticle interactions 280 with the interfaces and outer system walls that causes equal spin triplet pair amplitudes to be largest 281 at relative magnetization angles away from 90°. See also [58] where similar effects are found in the 282 diffusive regime.

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Measured T_c(AP)-T_c(P)

Measured [T_c(AP)-T_c(j)]_{max}

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· Theory [T_(AP)-T_(0)]

- 🛆 · Theory T, (AP)-T, (P)



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Fig. 3. Theory and experimental ΔT_{C} vs etch Au etch time and Au layer thickness.

ż

(nm)

Nb (25 nm)

u (5 nm)

Py (3.5 nm) Au (da., nm)

à

LCMO (120 nm) STO (001)

Etch time (minutes)

3

Δ

5

100

50

0 21₆ (mk) 20-

-100

-150

0

In Fig. 3, we have compared the experimental and calculated dependence of the maximum value of $\Delta T_{c}(\theta)$ as a function of etching time. To focus on the effect of etching time on the Py/Au interface, we fix all interface scattering parameters, except H_{B2} (relating to the Py/Au interface) which is allowed to vary in such a way that is consistent with the measured etch rate. Namely, we set $H_{B1} = H_{B3} = 0.4$, H_{B4} = 0.14, and $0.7 \le H_{B2} \le 1.2$. After a certain time, continued etching is assumed to have no further effect on the interface scattering parameter H_{B2} . The thickness of the Au, however, decreases (0.75)

307 nm/min) with etching. For each datum point, we self consistently calculate $T_c(\theta)$ and extract ΔT_c and 308 $T_c(AP)-T_c(P)$. This results in good agreement with the experimental findings. In particular, the spin 309 valve effect is enhanced for an etching time of 2 minutes whereby an increased singlet-to-triplet pair 310 conversion takes place. Since the normal metal layers tend to host spin-polarised triplet pairs, 311 reducing their thickness can also result in a limited T_c reduction that signifies the emergence of spin 312 polarized triplet pairs.

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IV. SUMMARY

317 We have demonstrated triplet pair creation through magnetization control in Nb/Cu/Py/Cu/LCMO 318 TSVs using in-plane magnetic field as small as 3.3 mT. Efficient pair conversion and spin-polarized 319 triplet pair transfer to LCMO is achieved for relative magnetization angles between 60° to 90° with a 320 maximum $\Delta T_{c}(\theta)$ close to -150 mK through band matching optimization at the Au/LCMO interface. 321 Although $\Delta T_{c}(\theta)$ is smaller than observed for TSVs containing CrO₂ which achieved -800 mK [34]) in an 322 out-of-plane magnetic fields of 2 T, our results agree well with a fully microscopic self-consistent 323 model and demonstrate that the fully spin-polarized and chemically stable mixed valance manganites 324 are highly attractive for superconducting spintronics.

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