



Mechanical properties of thin-film Parylene–metal–Parylene devices

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Structures and testing methods for measuring the adhesion strength, minimum bending diameter, and bending fatigue performance of thin-film polymer electronic architectures were developed and applied to Parylene–metal–Parylene systems with and without the moisture barrier Al₂O₃ [deposited using atomic layer deposition (ALD)]. Parylene–metal–Parylene interfaces had the strongest average peel test strength and Parylene–Parylene interfaces had the weakest peel test strength. Layers of ALD Al₂O₃ deposited within the device increased the average peel strength for Parylene–Parylene interfaces when combined with silane A-174, but did not increase that of the Parylene–metal–Parylene interface. Metal traces in the middle of 24 μm thick Parylene–metal–Parylene devices had a minimum bending diameter of ~130 μm before breaking and being measured as an open circuit. The addition of one layer of Al₂O₃ above the traces allowed them to be completely creased when bent away from the Al₂O₃ layer without producing an open circuit, but increased the minimum bending diameter to ~450 μm when bent towards the Al₂O₃. Although fatigue testing produced cracks in all devices after 100k bends, the insulation of the Parylene–metal–Parylene devices without Al₂O₃ performed well with electrochemical impedance spectroscopy showing only small decreases in impedance magnitude and small increases of impedance phase at low frequencies. However, devices with Al₂O₃ failed during EIS due to Al₂O₃ being deteriorated by water.

Keywords: Parylene, thin-film polymer, peel test, bend test, fatigue test, EIS, Al₂O₃

Introduction

Unlike traditional silicon-based electronics, thin-film polymeric devices are inherently flexible, which is important for devices targeting soft, moving tissues for various applications, such as monitoring electrical activity from muscles, measuring nerve impulses from the central or peripheral nervous system, and delivering therapeutic electric current, such as in pacemakers, deep brain stimulation, cochlear implants, and spinal cord stimulation. For such applications, flexibility is required to maintain good contact (important for sensing weak electrical signals and reducing the amount of current required for a therapeutic effect) and avoid damaging tissue through tethering effects.

In addition, thin-film polymeric devices use photolithography and other technologies borrowed from the semiconductor industry to pattern electrodes or other sensors or structures onto the surfaces of materials, such as Parylene (Johnson and Wise, 2012; Kuo et al., 2013; Zhao et al., 2013), polyimide (Kisban et al., 2009; Tolstosheeva et al., 2011; Vienti et al., 2010), or polydimethylsiloxane (Gray et al., 2004; Guo et al., 2014). Because thin-film polymeric devices are based on microfabrication technology used to make transistors or micro-sensors, these devices have micron-scale resolution or better and allow sensors, such as transistors (Nomura et al., 2004) or force sensors (Gutierrez et al., 2010), to

be integrated directly onto the flexible device. These devices are thin enough to be rolled up and delivered via catheter (Fonseca et al., 2006) or incorporated into the catheter itself (Li et al., 2008a). Thin-film polymeric devices for cochlear stimulation (Johnson and Wise, 2010), spinal cord stimulation (Rodger et al., 2007) as well as electrocardiography (ECG) (Viventi et al., 2010; Yu et al., 2012), electrocorticography (ECOG) (Toda et al., 2011), and intracortical neural recordings (Kim et al., 2013) have been demonstrated.

While thin-film polymeric devices are flexible, biocompatible, and allow for high resolution features, they are prone to delamination (Li et al., 2008b; Seymour et al., 2009) and carry concerns about their mechanical robustness (Schuettler et al., 2005). Mechanical tests for individual devices for specific applications have been performed, such as measuring the properties of a device while wrapping it around fixed-diameter structures (Chen et al., 2010; Kisban et al., 2009; Ochoa et al., 2013), fixing it to a moving tissue (Metallo et al., 2011; Viventi et al., 2010; Zhao et al., 2013), or allowing it to self-curl (Yamagiwa et al., 2013). In addition, tests have been performed to screen for methods to improve adhesion (Chang et al., 2011; Seymour et al., 2009). However, systematic studies that investigate in detail the mechanical performance and adhesion of multi-layer film stacks in generalized polymeric film devices are lacking in research literature.

Also, typical reports in literature present data from single devices (such as bend testing across cylinders of multiple diameters) from non-destructive experiments. Destructive and continuous peel/bend tests can provide specific limits to device performance. In this work, designs for polymeric thin-film devices and tests are presented as a way to determine the robustness of thin-film structures. In particular devices and tests to measure peel strength, minimum bending diameter, and bending fatigue characteristics of Parylene–metal–Parylene, structures are tested to determine their potential and limitations for use as thin-film devices. In addition, these devices and tests are used to measure the properties of Al₂O₃ [deposited using atomic layer deposition (ALD)], which – in combination with Parylene – has been shown to improve water-barrier properties of encapsulated silicon devices (Ghosh et al., 2005) and chosen in the present study to act as a water barrier and adhesion promoter for Parylene–metal–Parylene devices.

Materials and Methods

Table 1 summarizes the distinct test structure designs for measuring the peel strength, minimum bending diameter, and bending fatigue. Subsequent sections describe details of the fabrication, design, and testing of these devices.

Device Fabrication

Peel test, minimum bending diameter test, and bending fatigue test devices were all fabricated using similar fabrication methods (Figures S1A,B in Supplementary Material). Details of Parylene C-based thin-film devices can be found in previous publications (Kuo et al., 2013), therefore, only a brief description is given here. First, ~13 μm of Parylene C was deposited onto a prime silicon wafer using CVD (Specialty Coating Systems Inc., IN, USA). Next, standard photolithographic liftoff was used to pattern 200 nm of e-beam deposited Pt on top of the underlying Parylene C. Finally,

a second layer of Parylene was deposited to insulate the Pt patterns, and O₂ plasma was used to expose contact pads and cut out devices. ALD was used to deposit Al₂O₃ either before and after metal traces or only on top of metal traces (Figure S1B in Supplementary Material, details of Al₂O₃ given below in Section “Atomic Layer Deposition of Al₂O₃ Barrier”).

Atomic Layer Deposition of Al₂O₃ Barrier

Various processing techniques, such as thermoforming, cleaning, and chemical treatments, have been investigated to improve thin-film adhesion (Chang et al., 2011; Kim and Najafi, 2005). Also, barrier layers and adhesion promoters have been used to prevent failure of thin-film insulation (Xie et al., 2012; Xie et al., 2013). However, only static conditions were used to test for improved bond strength and insulation integrity. In the current study, ALD Al₂O₃ combined with the silane adhesion promoter A-174 was deposited between layers of Parylene to act as both a barrier layer and adhesion promoter. The effect of this layer, not only on adhesion promotion but also on trace performance in bending failure and bending fatigue, was measured.

Al₂O₃ has been shown to be an excellent barrier to water vapor (Ghosh et al., 2005) and for this reason was deposited with Parylene to improve the performance of implanted neural electrodes (Xie et al., 2013). Parylene serves as a barrier for bulk water, ions, and chemicals, while the Al₂O₃ serves as a barrier for water molecules that are able to penetrate the Parylene. Plasma-enhanced ALD is a process that sequentially deposits monolayers of molecules onto a substrate using self-limiting chemical reactions. ALD deposited Al₂O₃ is preferred over sputtered or plasma enhanced chemical vapor deposition (PECVD) because of argon sputtered Al₂O₃ can trap argon into the coating making it porous, and PECVD can lead to less dense films (Ghosh et al., 2005). ALD coatings are also highly conformal and pinhole free, thus avoiding gaps which can be major sources of corrosion in encapsulated electronics in contact with water (Vanhoestenbergh and Donaldson, 2013).

To incorporate Al₂O₃ into PMP devices, ALD was used to deposit 17–24 nm of [250 cycles on a Fiji 2 System (Ultratech, MA, USA)] either before and after metal deposition or only after metal deposition (Figure S1B in Supplementary Material). When Al₂O₃ was deposited before metal deposition, an adhesion layer of 20 nm layer of Ti was deposited over the Al₂O₃ to promote adhesion of Pt to Al₂O₃ and an adhesion layer of A-174 (Momentive, Waterford, NY, USA) was deposited over the top layer of Al₂O₃ to promote the adhesion of the top layer of Parylene C. In addition, because Al₂O₃ is unaffected by O₂ plasma, wafers were soaked for extra time (~6 min) in developer [AZ 340 (Microchemicals, Ulm, Germany):DI H₂O, 1:4 dilution]. These processes made devices with a variety of different combinations of Parylene C, Al₂O₃, and metal architectures (**Table 2**).

Peel Tests

Peel Test Device Design

Delamination is a common concern for thin-film polymeric devices, especially in aqueous solutions (Li et al., 2008b). Any interface in a thin-film polymer device that delaminates could potentially create paths to water intrusion and so determining the adhesion strength between layers is important. Thin films can bond to underlying

TABLE 1 | Summary of devices and tests used to measure peel strength, minimum bending diameter, and bending fatigue.

Test name and purpose	Description of device	Device layout
<p><i>Peel test strength</i> Measures the strength of adhesion between two thin film layers</p>	<p>Peel test devices contain a bonded region connected to two flaps that are clamped to a load cell, which measures the force required to separate the joined interfaces.</p>	
<p><i>Minimum bending diameter</i> Determines the minimum outer diameter a thin film device can be bent to before encapsulated traces break and result in open circuit</p>	<p>Minimum bending diameter devices consist of a rectangular cable with a single serpentine trace which transverses the length of the cable several times. The resistance of the trace is measured at the exposed ends while the device is crushed between two plates until an open circuit is detected.</p>	
<p><i>Bending fatigue failure</i> Measures the performance of thin film trace insulation after cyclic bending of thin film cables</p>	<p>Bending fatigue failure devices consist of a rectangular cable with multiple insulated traces with only one end exposed to make an electrical connection. The device is bent $\pm 90^\circ$ for a given number of cycles and then the electrochemical impedance between each trace and an external electrode as well as between adjacent electrodes is measured. This process is then repeated.</p>	

TABLE 2 | Different material combinations investigated for flexible Parylene devices.

Abbreviation	Interfacial layers (from bottom to top)
PP	Parylene/Parylene
PMP	Parylene/platinum/Parylene
PAP	Parylene/ Al_2O_3 /Parylene
PAMAP	Parylene/ Al_2O_3 /titanium/platinum/ Al_2O_3 /Parylene
PAAP	Parylene/ Al_2O_3 /Parylene
PMAP	Parylene/titanium/platinum/ Al_2O_3 /Parylene

Thickness of layers: Parylene C (top and bottom layers) 12–13 μm ; Al_2O_3 (top and bottom layers) 17–25 nm, Ti 20 nm; Pt 200 nm.

substrates physically by filling in pores or surface topography, through covalent bonds, or by van der Waals forces. In addition, interfaces that typically have poor adhesion, such as Parylene deposited on silicone, can be improved with adhesion promoters, such as the silane A-174, which bonds covalently both with Parylene free radicals during the deposition process and provide a strong

linkage with the underlying substrate (Charmet et al., 2013). Also, a combination of heat and compression can bond Parylene layers together by causing the polymer chains on the surface of the touching films to become entangled (Noha et al., 2004). Computational models suggest that metals, such as platinum and titanium can form stable and covalent bonds when deposited on Parylene (Vasenkov, 2011). Water intrusion (both from the edges of thin-film devices and through the polymer layers themselves) can weaken adhesion by penetrating between physical bonds or dissolving intermediate layers. Delamination occurs when upward forces perpendicular to the substrate are stronger than adhesion forces the top film.

Several methods have been used to measure the adhesion force of thin films. The “scotch test tape” qualitative tests the adhesion of a film deposited on a substrate by applying a piece of pressure sensitive tape to the film and pulling the tape off (Lim et al., 2006; Sridhar et al., 2009). If the top layer of the deposited film comes off (either in parts or in pieces), it is said to have “failed” the test.

The adhesion force of the film can also be determined using a load cell connected to the free end of a film which is then pulled

at a 90° angle to a fixed substrate (Chang et al., 2011) or 180° to a second attached flexible film, to measure force required to separate the film from a substrate. The peel strength is defined as the average load per unit width of bondline required to progressively separate the two materials.

When the film is pulled at a 180° angle, the test is called a “T-peel test” because of the shape formed by the two free ends pulled and the bonded perpendicular region forming base of the “T” (Table 1). T-peel tests are used for the simplicity of the test setup and ability to compare different forms of bonding or adhesives (Song and Yu, 2002); however, these tests are sensitive to the thickness of the substrates, which require energy to deform during the test and must be accounted for if direct measurement of cohesion strength is desired (Kim, 1988).

In the current experiment, T-peel tests were performed with specimens based on ASTM standard D1876-08, but modified to be used on thin-film devices. These tests allowed the peel strength of multiple interfaces to be measured, including PP, PMP, PMAP, PAP, PAMAP, and PAAP (Table 2).

Peel Test Device Fabrication

In addition to the fabrication steps described in Section “Device Fabrication” (Figure S1 in Supplementary Material), two flaps (4.3 mm × 14 mm) with 3 mm holes (for gripping) were created using a sacrificial photoresist that was later removed to enable them to be separated after the device was released (Figure S2 in Supplementary Material). The flaps were used to grip while pulling apart the bonded region (4.3 mm × 3 mm) of the device. Each Parylene layer of the peel test device was 13 μm thick.

Peel Test Setup and Characterization

Peel test devices were tested by attaching one flap to a stationary clamp and the other to a 50 g load cell (Omega, Stamford, CT, USA), which was attached to a motorized stage (Z812, Thorlabs, Newton, NJ, USA) that pulled the two flaps apart at a rate of 0.02 mm/s (Figure 1). A custom Labview program was used to control the stage and measure the voltage output from the load cell (Figure 1B). PMP devices were tested both dry and after a 24-h soak in 1× PBS solution at 37°C for 24 h. Scanning electron microscope (SEM) images were taken of peel test surfaces to examine morphology of adhesion failure.

Minimum Bending Diameter Tests

Minimum Bending Diameter Device Design

Implanted thin-film polymeric devices are exposed to mechanical forces, such as muscle contractions, acceleration, gravity, hydrostatic pressure, etc., that can damage the device. Muscles can contract to 50% of their resting length (Jenkins, 2001), which can cause devices near or on their surface to be flexed, bent, or pinched. When a material is subjected to bending, it experiences compression forces on the inside surface and tensile forces on the outside surface (Figure S3 in Supplementary Material). These stresses follow a linear relationship according to the following equation for bending stress (σ_b):

$$\sigma_b = \frac{My}{I} \quad (1)$$

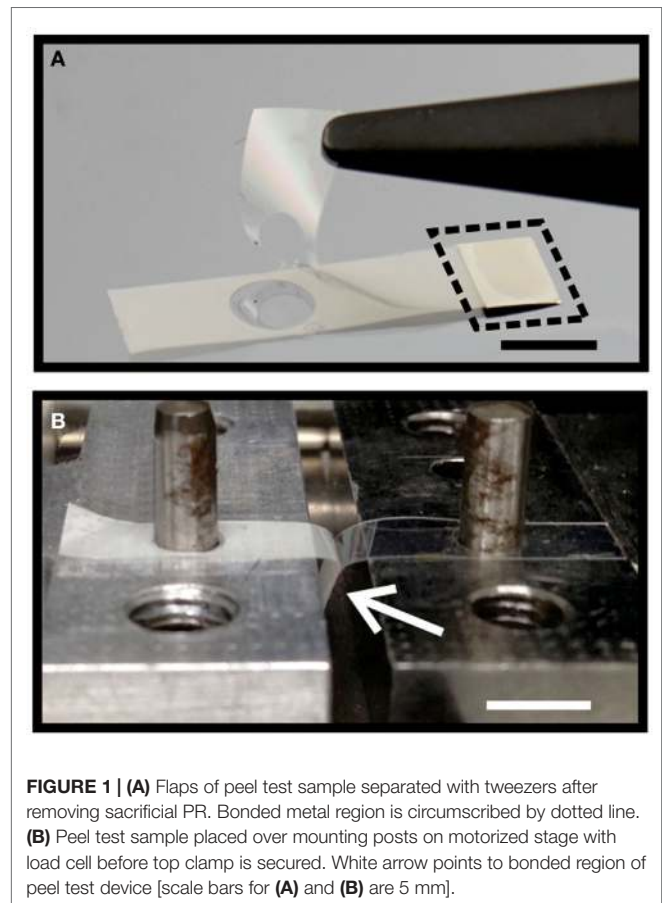


FIGURE 1 | (A) Flaps of peel test sample separated with tweezers after removing sacrificial PR. Bonded metal region is circumscribed by dotted line. **(B)** Peel test sample placed over mounting posts on motorized stage with load cell before top clamp is secured. White arrow points to bonded region of peel test device [scale bars for **(A)** and **(B)** are 5 mm].

where M is the bending moment, y is the vertical distance from the neutral axis, and I is the moment of inertia around the neutral axis. The neutral axis within the sample is where the transition from compressive to tensile forces occurs (Figure S3B in Supplementary Material). Because materials are generally fail due to cracks propagating perpendicular to tension forces, cracks in bent materials tend to propagate from the outside surface to the inside surface.

The effect of bending stress on thin-film devices can be assessed by wrapping films around cylinders of specific diameters ranging from hundreds of microns to several millimeters and then either measuring the resistance across the traces or the testing the performance (Ai et al., 2008; Chen et al., 2010; Kim et al., 2010; Lacour et al., 2010; Ochoa et al., 2013). Alternatively, films can be wrapped around spheres (Ochoa et al., 2013; Rodger et al., 2007) and inflatable balloons (Kim et al., 2011). Some devices are self-curling and were shown to maintain function while being bent 90° (Takeuchi et al., 2004) or down to a radius of 2 mm or less (Johnson and Wise, 2012; Yamagiwa et al., 2013).

A convenient and automated setup determining the minimum bend radius of a thin-film device called the “bending plate” method (Roll, 1976) involves bending a flexible film between two plates until failure (Sekitani et al., 2010). The bending plate method is a versatile technique that was developed to measure the tensile strength for optical fibers based on the distance between the plates

(*D*) the overall fiber thickness (*d*), the uncoated fiber radius (*r*), and the Young's modulus of the fiber (**Eq. 2**) (Matthewson et al., 1986), but has also been used to determine the smallest diameter that thin conductive films on flexible substrates can be bent to before registering an open circuit (Alzoubi et al., 2011; Leterrier et al., 2003) (i.e., the minimum bending diameter). The boundary condition of the bending plate method is that the radius of curvature is 0 where the plate is in contact with the cable, and the radius of curvature is maximum on the outside edge of the cable half way between the plates. The radius of curvature of the bend (*R*) is described by the equation: $\frac{d\theta}{ds} = \frac{1}{R}$ where θ is the

angle formed by a line tangent to the curve of the cable and a line at the midpoint at the bend parallel with the plates, and *s* is the length of the line from the intersection of the line tangent to the cable with the cable to the intersection of that same line to the midline. The moments about this point are given by the bending beam equation: $EI \frac{d\theta}{ds} = -Fx - M$ (Matthewson et al., 1986).

There are several advantages in using this type of bend test rather than a traditional tensile test, including eliminating the influence of the grips on the test and simple setup; however, the test is limited by difficulty in determining the exact position of 0 separation and only being able to test the portion of the sample between the plates.

$$\sigma_{\max} = 1.198E \frac{2r}{D-d} \quad (2)$$

To measure the minimum bending diameter of Parylene thin-film cables using this method, a thin-film flexible cable (with dimensions 48 mm × 4.3 mm × 0.026 mm) with a single 360 μm wide serpentine trace (longest straight dimension is 40 mm) was designed, whose resistance could be measured while compressing the cable between two acrylic plates (**Figure 2**). Failure was marked by the cable reading an open circuit. These devices were made using standard photolithographic processes described in Figure S1 in Supplementary Material.

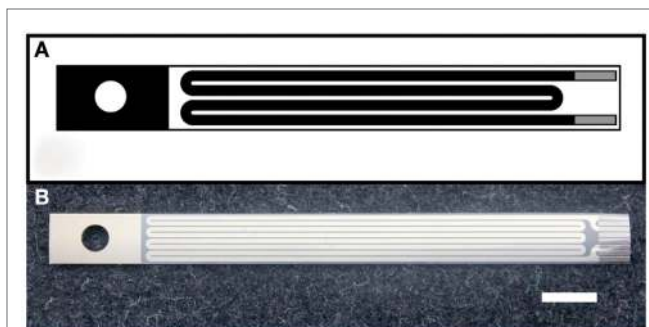


FIGURE 2 | (A) Cartoon and **(B)** photograph of minimum bending diameter test devices (scale bar is 5 mm). In cartoon **(A)**, the thickness of the traces has been increased and number of turns decreased for illustrative purposes. The metal and hole on the left side of device are for clamping the device to the testing fixture. The contact pads on the right side allow electrical resistance measurements.

Minimum Bending Diameter Test Setup

To measure the minimum bending diameter, a thin-film cable was between two acrylic blocks while resistance was monitored (Figure S4 in Supplementary Material). One of the acrylic blocks was attached to a 50-g load cell (Omega, Stamford, CT, USA) which, in turn, was connected to a motorized stage (Z812, Thorlabs, Newton, NJ, USA) for measuring axial force for the purpose of determining complete contact between the blocks. The devices were supported by a holder and connected to a precision multimeter (Keithley 2700, Keithley, Cleveland, OH, USA). Using a custom Labview program (National Instruments, Austin, TX, USA), the motorized stage was advanced at a rate of 0.02 mm/s, bringing the two acrylic blocks together and reducing the bend diameter. Failure was defined as an open circuit measurement. The minimum diameter was measured both with the devices folded away from the contact pads (“out”) and toward the contact pads (“in”) (Figure S4 in Supplementary Material).

The acrylic blocks were visually confirmed to make full contact when the load cell read 4.5 N. To determine the minimum diameter, therefore, the position of the stage where the load cell reached 4.5 N was defined as 0, and the difference of the stage position at that point and the position of the stage where the resistance of the minimum bend diameter test device directly before the trace was measured as an open circuit was the device's minimum diameter. The minimum bending curvature (radius) is half of the difference between the minimum bending diameter and the thickness of the thin-film device (26 μm).

To determine the failure diameter, the electrical resistance of a Pt trace across cable was monitored as a loop of cable was bent between two plates (Figure S4 in Supplementary Material).

Bending Fatigue

Bending Fatigue Device Design

When thin films are exposed to forces less than their tensile strength, even though they will not break on the first cycle, they can still fail due to fatigue caused by repeated motions present near parts of the body such as the heart, which contracts over 100k times/day, or the neck, which moves an average of 10 million times in each direction each year (Sterling et al., 2008). Fatigue occurs because dislocations and voids move and coalesce under bending stress. When a sufficient number of dislocations aggregate, crack propagation sites form that weaken the material. Fatigue in thin-film devices has been measured in test *in vitro* by continuously measuring resistance across polyimide (Kisban et al., 2009) or Parylene (Johnson, 2011) test cables bent to 90° angles for thousands or millions of cycles, and *in vivo* by measuring ECG signals for ~10,000 heart beats (Viventi et al., 2010).

To measure the effect of repeated deformation on thin-film Parylene devices, bend test samples (measuring 4.3 mm × 48 mm × 0.026 mm) with 40 mm long and 360 μm wide fully insulated traces (**Figure 3**) were designed and fabricated using standard lithographic techniques described in Figure S1 in Supplementary Material.

Electrochemical impedance spectroscopy (EIS) was used identify signs of failure in the device (e.g., delamination, water infiltration) by measuring the impedance between traces [the

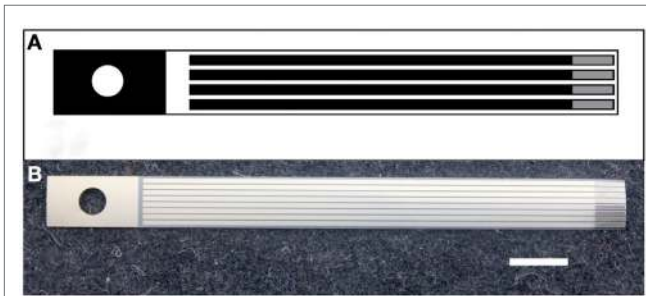


FIGURE 3 | (A) Cartoon and **(B)** photograph of bend test devices (scale bar is 5 mm) [in cartoon **(A)**, the thickness of the traces has been increased and number of traces decreased for illustrative purposes]. The metal and hole on the left side of device are for clamping the device to the testing fixture. The contact pads on the right side allow electrochemical impedance measurements with a potentiostat.

lateral impedance (Figure S5A in Supplementary Material)] and the impedance between a trace and an external counter electrode [the transverse impedance (Figure S5B in Supplementary Material)] while immersed in phosphate buffered saline (PBS).

Bending Fatigue Testing

To apply repeated bend cycles to the bending fatigue test devices, a custom cable flex tester (based on ASTM standard D4565-10 for testing cable insulation) was constructed consisting of a Sparkfun stepper motor (ROB-09238) (Sparkfun, Boulder, CO, USA) with an Arduino UNO (Arduino, Torino, Italy) controller and mounting fixtures (Figure S6 in Supplementary Material). In this setup, the top of the cable was attached to a rotating wheel and passed through two mandrels at the center of the wheel and two stationary rollers below the wheel. A weight [consisting of a section of flat flexible cable (FFC)] was attached to the bottom of the cable (below the stationary rollers) to keep the cable taut. The wheel was then rotated $\pm 90^\circ$. As the wheel turned the cable was bent around the center mandrels, while the cable below the mandrels remained free to move up and down through the stationary rollers. For a cable bent around a mandrel, the total stress in the cable (σ_{total}) can be approximated as the sum of the stress due to the tensile forces (σ_t) caused by the weight hanging from the cable) and the bending forces (σ_b). Because the radius of curvature can be assumed to be the cylindrical radius of mandrel σ_b can be approximated as:

$$\sigma_b = \frac{Et}{D+t} \quad (3)$$

where E is the elastic modulus of Parylene, t is the thickness of the cable, and D is the diameter of the mandrel. In a fatigue test, σ_{total} is much less than the tensile stress of the cable so that multiple cycles are required before signs of fatigue become apparent. In the current test setup, devices were subject to 10–100k bending cycles in which one cycle is defined by bending the between -90° and $+90^\circ$ around a mandrel (1.25 mm OD).

EIS was performed in $1\times$ PBS at 37°C with an amplitude of 10 mV_{rms} and a frequency range of 1–100,000 Hz. A platinum wire was used as a counter electrode and an Ag/AgCl (3 M NaCl) electrode was used as a reference (transverse impedance). The

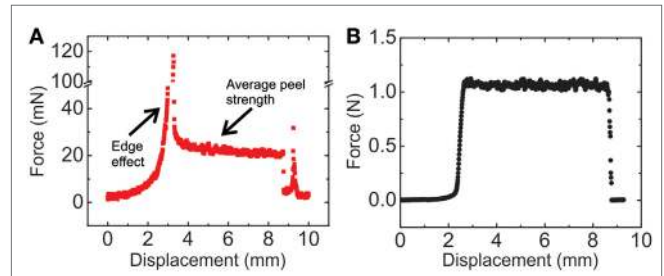


FIGURE 4 | Representative peel test raw data from (A) PP and (B) PMP test devices. The edge effect observed for the PP device was not observed for the PMP device, most likely due to the force magnitude of being dominated by the larger PMP average peel strength (~50 times larger). Note that the two plots are on different scales.

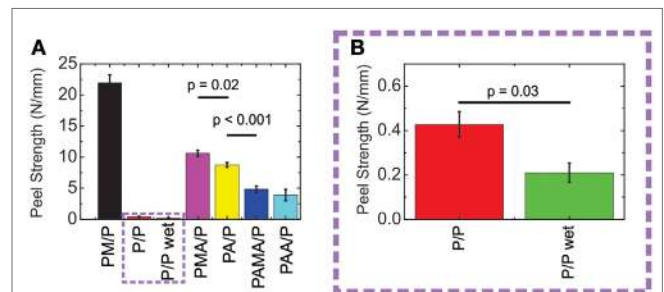


FIGURE 5 | (A) Combined average peel test data (mean \pm SE, $n = 3-8$) from various trials. Parylene/Parylene interfaces had the lowest peel strength and were regraphed separately **(B)** using a different scale. The slash indicates the interface at which delamination occurred.

impedance magnitude was normalized to absolute impedance of the magnitude at 0 bends and plotted.

In addition to EIS, devices were examined using optical and Nomarski differential interference coherence (DIC) microscopy as well as scanning electron microscopy (SEM), before and after bending.

The impedance was expected to behave according to the simplified Randle's model, with the values for the capacitive and resistive components changing as the insulation fails [Figure S7 in Supplementary Material (Loveday et al., 2004)]. Delamination of the Parylene film is expected to cause a drop in measured impedance starting at lower frequencies, which has been shown to be more sensitive to electrode failure (Tadmakov et al., 2015).

Results

Peel Testing

Peel testing resulted in characteristic T-peel curves with an increase in force once the samples become taut, and a relatively continuous force as the bonded area was separated by peeling (Figure 4). Overall, the average peel test strength of the PP devices was low with relatively large spikes observed in the force plots at the beginning and inconsistent spikes at the end of the peel tests (representative plot in Figure 4A).

Mean values of average peel strength are presented in Figures 5A,B indicated PMP devices, exhibited the highest

adhesion (21.9 ± 1.3 N/mm, mean \pm SE), followed by the PMAP, PAP, PAMAP, and PAAP devices (**Figure 5A**). The force required to pull apart PP and PP wet devices was the lowest (0.43 ± 0.06 and 0.21 ± 0.04 N/mm, respectively) (**Figure 5B**). PP samples soaked overnight at 37°C in 1 M PBS solution (PP wet) had significantly lower peel strength than devices that were not exposed to soaking (**Figure 5B**). In all cases, the top layer of Parylene was separated from the underlying layers.

Significant differences were found between PAP and PMAP and the PMAP and PAMAP devices. However, there is no significant difference between the PAMAP and PAAP devices. The cumulative mean average peel test bar graphs also show that samples with either one or two layers of Al_2O_3 showed much smaller variation in peel strength than samples without Al_2O_3 (**Figure 5A**).

SEM images were acquired for all conditions (**Figure 6**). The PMP surfaces (**Figure 6A**) contained cracked and delaminated Pt on the bottom layer of Parylene (left image) and plastically deformed Parylene on the top layer (right image). In contrast, the PP devices (**Figure 6B**) were relatively smooth and clean except for thin strands of Parylene on the bottom layer (left image). The devices with Al_2O_3 with or without a layer of Pt were similar (**Figures 6C–F**). The bottom layer of these interfaces (left images), which included the bottom layer of Parylene and one or two layers of Al_2O_3 , contained parallel ridges topped with plastically deformed flaps of Parylene. The top layers

(**Figures 6C–F**, right images), contained clusters of thin strings of Parylene.

Minimum Bending Diameter

A diameter of 0 mm was defined as the point in which the force measurement on the acrylic clamping blocks was 4.5 N and all data was normalized to this setting. Full contact of acrylic blocks at 4.5 N was confirmed using video images. The resistance of the traces as the bending diameter was reduced was graphed (**Figure 7**) and the minimum bending diameter just prior to the formation of an open circuit was recorded (**Figure 8**).

The minimum bending diameter of PMP devices was measured to be 138 ± 16 μm with no significant difference between bending the device toward the top layer of Parylene (in) or toward the bottom layer of Parylene (out). Contrarily, the PMAP device with one layer of Al_2O_3 underneath the top layer of Parylene failed much earlier (476 ± 60 μm) than the PMP devices when bent toward the Al_2O_3 layer, but did not break at all when bent away from the Al_2O_3 layer. The low yield of PAMAP devices was attributed to difficulty in establishing electrical contact and therefore, PAMAP devices were only tested under bending toward the top layer of Parylene.

SEM images were taken of both the top and bottom surfaces of devices that had been bent away from (**Figure 9**) and toward (**Figure 10**) the contact pads. Cracks that formed due to tension on the outside of the bend (**Figures 9A,C** and **10A,C**) appeared to be deeper with cleaner edges compared to cracks formed due to

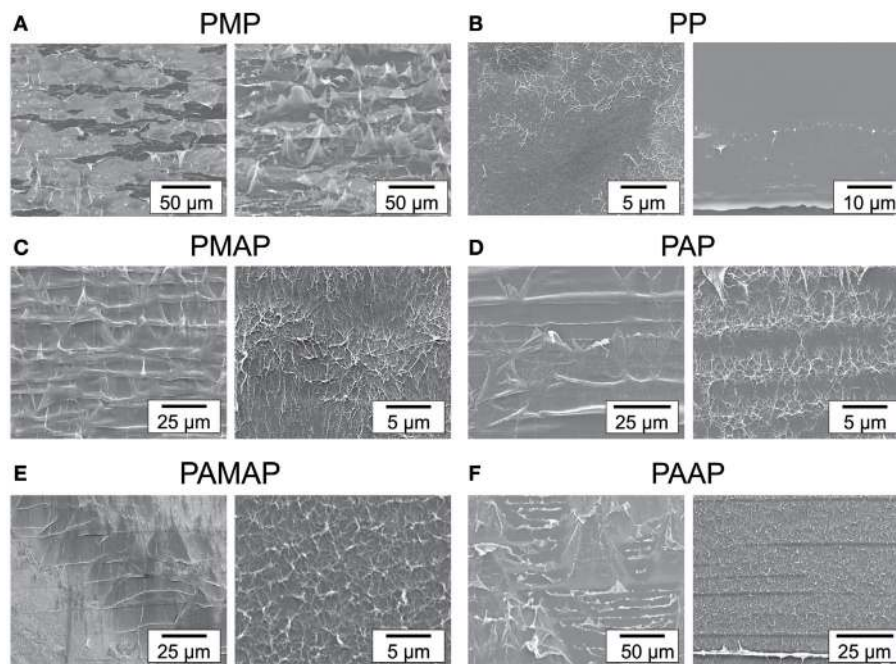
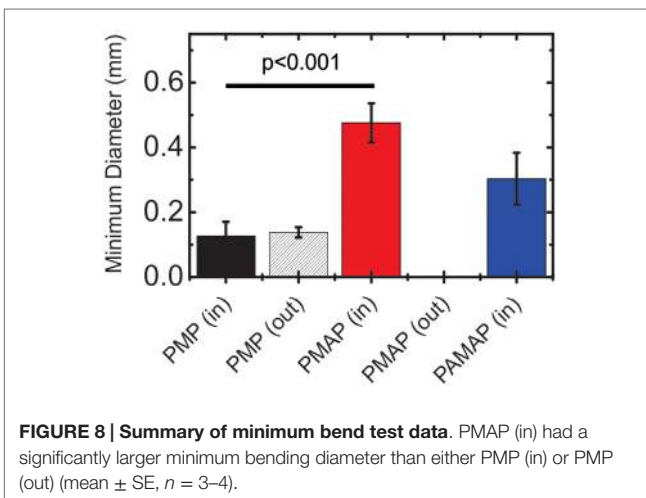
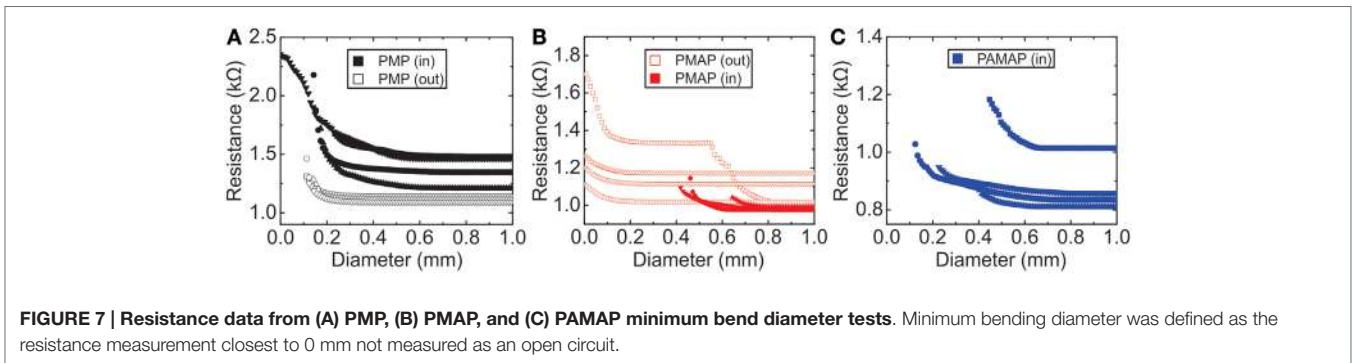


FIGURE 6 | SEM images of peel test surfaces. For each individual panel, the left image is the bottom (closest to the silicon wafer during processing) layer (consisting of Parylene, metal, and Al_2O_3) and the right image is the top layer of Parylene. The PMP surface (**A**) contained cracked and delaminated Pt on the bottom layer of Parylene (left image) and plastically deformed Parylene on the top layer (right image). PP devices (**B**) were relatively smooth and clean

except for thin strands of Parylene on the bottom layer (left image). Devices with Al_2O_3 with or without a layer of Pt (**C–F**) were similar in that the bottom layers (left images) contained parallel ridges topped with plastically deformed flaps of Parylene and the top layers (right images) contained clusters of thin strings of Parylene. Note that images are at different scales (surfaces were sputtered coated with Pt before imaging).



compression on the inside of the bent device (**Figures 9B,D** and **10B,D**), which are characterized by rope-like ridges of Parylene accompanied by plastically deformed shark-tooth shaped fringes.

Bending Fatigue

EIS of a PMP device released from a wafer that showed inconsistent adhesion during peel testing (data not shown) demonstrated the ability to reveal insulation damage without destructive testing (**Figure 11**). The magnitude of the transverse impedance gradually decreased at lower frequencies after even only 100 bend cycles, followed by a broad spectrum decrease after 5k bend cycles (**Figures 11A,B**). Similarly, the phase decreased at lower frequencies during the first 4k bend cycles, and then at higher frequencies after 5k bend cycles (**Figure 11C**). The lateral impedance proved to be more sensitive, showing a broad spectrum decrease in the magnitude of the impedance at 4k as well as 5k (**Figures 11D,E**). Before bend testing (0 cycles), the lateral impedance had a -90° phase at all frequencies, which is indicative of a fully insulated trace. However, the phase dropped at lower frequencies as soon as bend testing began, and there was a broad spectrum drop after 4k cycles (**Figure 11F**). Broad spectrum decreases in magnitude and phase suggested that the insulation has been damaged and there was a path for ions to move between traces.

The average transverse impedance from PMP devices (which peel testing showed good adhesion between the Parylene and

metal (**Figure 12**) showed a small drop in magnitude at lower frequencies, but remained relatively constant at higher frequencies (**Figure 12A**). However, as the number of bend cycles increased, the impedance at lower frequencies became lower. The phase of the PMP devices was initially -90° , but increased at lower frequencies after the first set of bends. At higher frequencies, the phase remained capacitive (around 90°) even when the devices were bent through 100k cycles (**Figure 12B**). In contrast to the PMP devices, the PMAP and PAMAP exhibited larger fluctuations in magnitude over most frequencies (**Figures 12C,E**) and compromised insulation prior to bending based on the phase data (**Figures 12D,F**).

The phase for PMP devices under lateral impedance testing started greater than -90° at lower frequencies (**Figure 13B**). However, similar to the transverse impedance, there was no broad spectrum shift of magnitude, but only a decrease at lower frequencies (**Figure 13A**). Similar to the transverse impedance, the lateral impedance of the devices with Al_2O_3 had large shifts in magnitude with large SEs (**Figures 13C,E**). In addition, the initial phase was greater than -90° even before the devices were bent (**Figures 13D,F**).

Cracks in the Parylene that started at edges (not shown) were visible by optical microscopy after 2k bend cycles (**Figure 14A**) and became pronounced after 5k bend cycles (**Figure 14B**). The cracks were either chevron-like, radiating from a central point such (top left traces in **Figure 14B**) or perpendicular to the traces (right trace in **Figure 14B**). Cracks were observed to cross over the metal traces and the Parylene region between traces after 100k bend cycles (Figure S8 in Supplementary Material).

Optical micrographs of PMAP and PAMAP devices showed noticeable water intrusion after 100 bend cycles (images were not taken after the initial pre-bend EIS and before first bend testing cycles) and continued to be present after subsequent bend tests (Figure S9 in Supplementary Material). In Figure S9 in Supplementary Material, Newton rings were observed in the Parylene next to the edge of the device, but there was also evidence of Newton rings near the bottom of the right trace.

Evidence of Al_2O_3 etching was evident in pock marks on the traces (Figure S9 in Supplementary Material) as well as discoloration and branched patterns that initiated from the outside traces (**Figure 15A**). The impedance of the traces measured at 1 kHz before bend testing commenced was much lower toward the edges of the device, but as the device was subjected to further bending

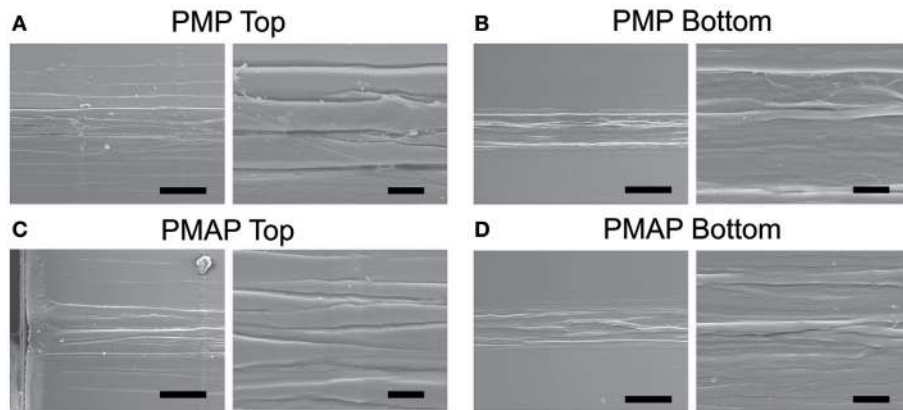


FIGURE 9 | SEM of cracks resulting from bending (A,B) PMP devices and (C,D) PMAP devices away from the contact pads (“out” orientation). The top layers of Parylene (A,C) were the outside of the bent device, and the bottom layer (C,D) were from the inside of the bent

device (images on the right side of each panel are zoomed in views of images on the left with scale bars of the left and right images being 50 and 10 μm , respectively. Parylene sputtered coated with Pt before imaging).

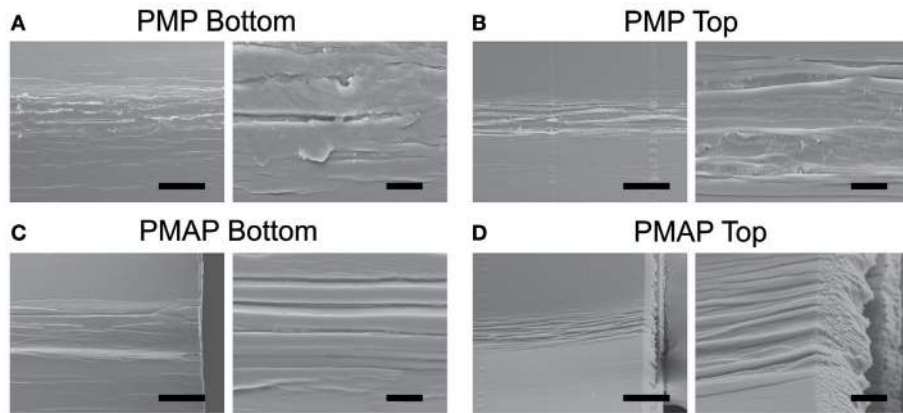


FIGURE 10 | SEM of cracks resulting from bending (A,B) PMP devices and (C,D) PMAP devices toward the contact pads (“in” orientation). The bottom layers of Parylene (A,C) were the outside of the bent device, and the top layer (C,D) were from the inside of the bent device (images on the right

side of each panel are zoomed in views of images on the left with scale bars of the left and right images being 50 and 10 μm , respectively. Parylene sputtered coated with Pt before imaging). In the PMAP sample, delamination at the bend was observed.

and EIS scans (each time an EIS was performed on all traces, the devices were immersed in PBS for ~ 45 min), the impedance was reduced across all traces and became more uniform (Figure 15B).

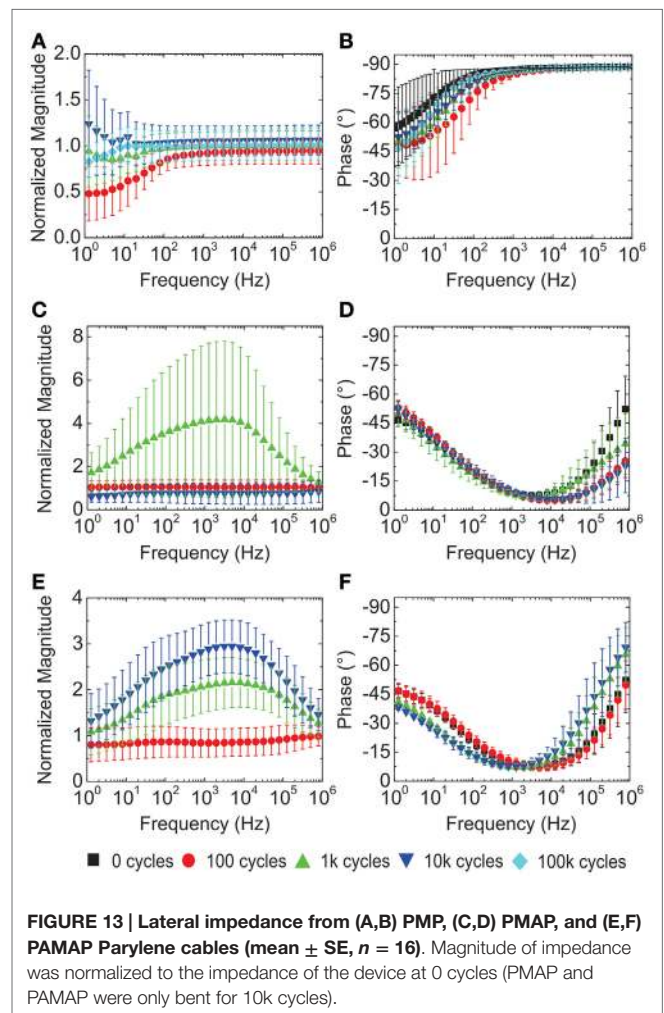
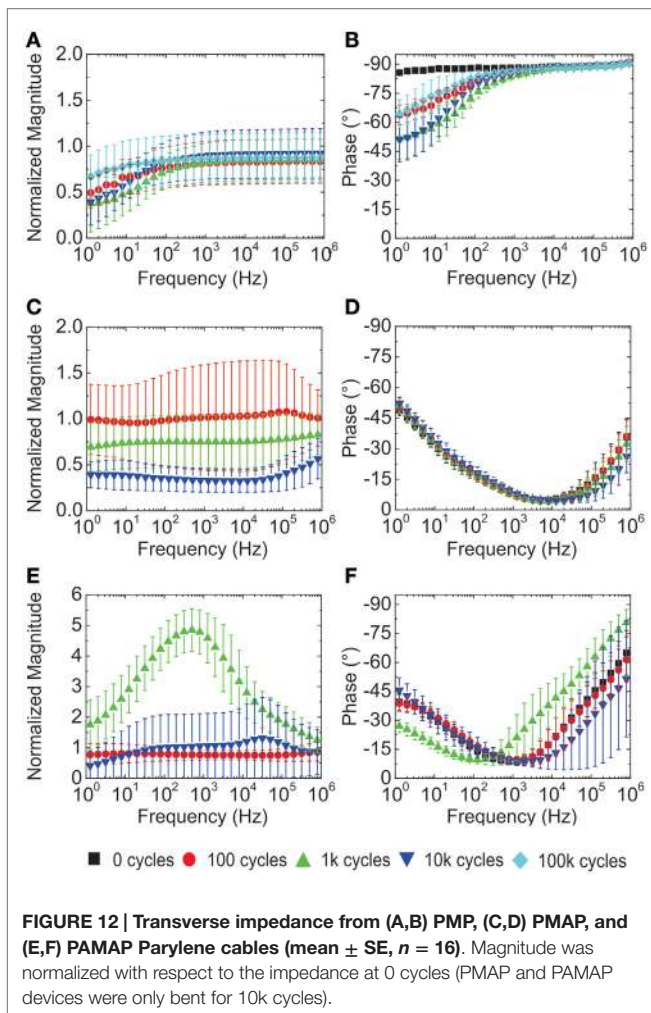
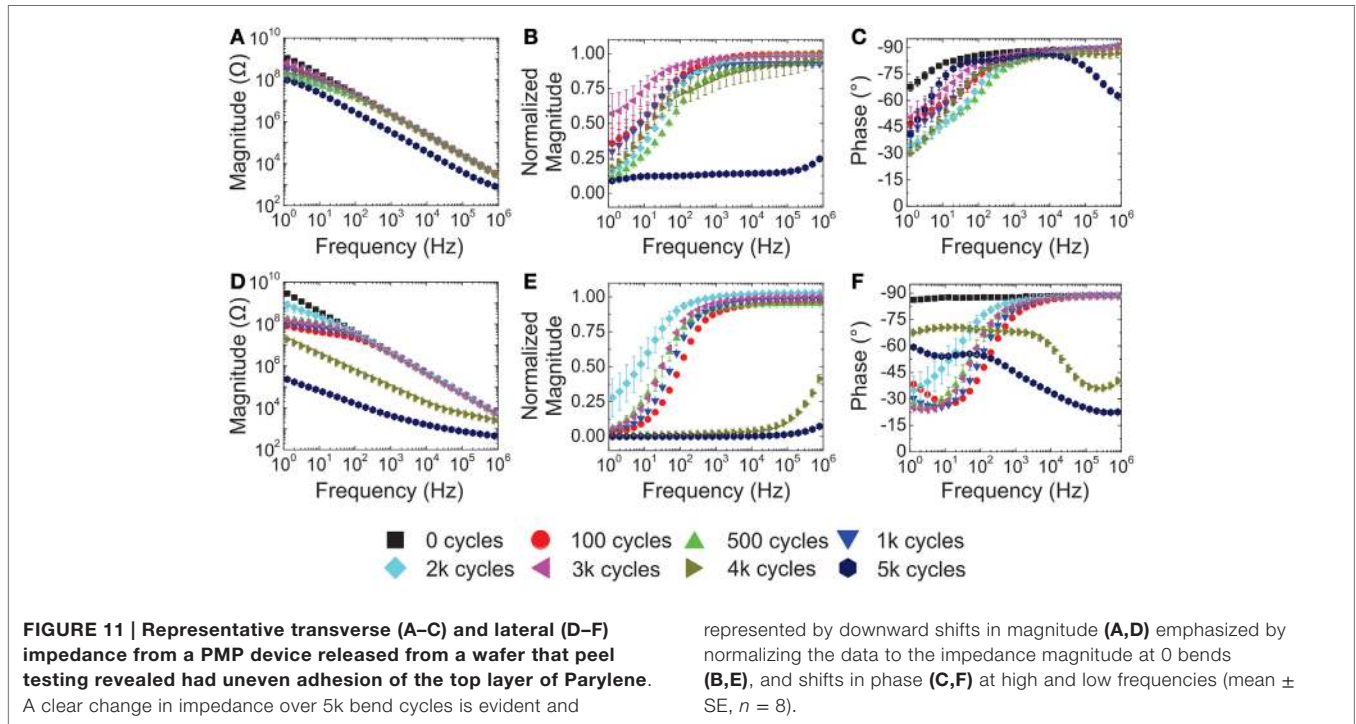
Discussion

Peel Testing

SEM images (not shown) revealed trace amounts of sacrificial photoresist that had not been fully removed from the Parylene flap regions, which may have caused the large spikes at the beginning of the PP peel tests (Figure 4A). The inconsistencies at the end of the test were mostly likely due to edge effects as the sample twisted when only a portion of the adhered regions were separated. This caused the samples to reorient so that the pulling force acted preferentially on the bonded surface area over the bondline. In contrast, the PMP interfaces were much stronger

than the PP interfaces and likely dominated any edge effects (Figure 4B).

Peel tests revealed that PMP interfaces had the strongest adhesion measuring nearly 20 times more than for PP interfaces. PP interfaces were further weakened by soaking for 24 h in 1 M PBS at 37°C (Figure 5). The reduced adhesion in soaked samples is likely due to water intrusion between the layers. The Al_2O_3 –Parylene interface was weaker than Pt–Parylene but stronger than PP. However, the adhesion between Al_2O_3 –Parylene interfaces for different processing conditions was inconsistent. The PAMAP and PAAP devices had lower strength than the devices with only one layer of Al_2O_3 . This is attributed to extra processing steps required for the extra layer of Al_2O_3 . These results imply that excellent adhesion is possible between platinum and Parylene, which has been suggested through the results of modeling (Vasenkov, 2011), and that platinum could potentially be used



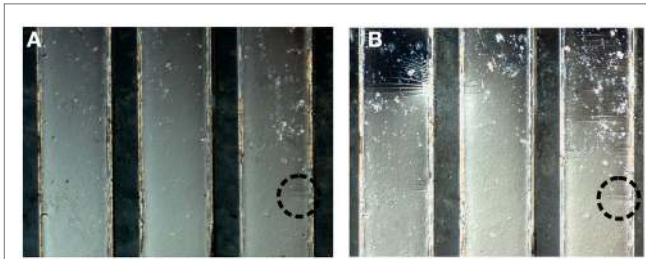


FIGURE 14 | Nomarski differential interference contrast images of fatigue cracks in the top Parylene layer covering three traces of a PAMAP devices after (A) 2k bend cycles and (B) 5k bend cycles. Metal regions are wide, light gray strips and Parylene appears as thin, dark gray strips. The majority of cracks appeared after 5k bend cycles; however, the dashed circle marks cracks present after 2k cycles (circle is 200 μm in diameter).

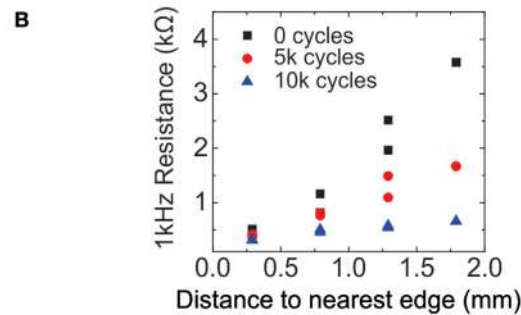


FIGURE 15 | (A) Micrograph of all eight traces of a PMAP device showing discoloration and branched patterns occurring on outside traces (scale bar is 500 μm). (B) The transverse impedance at 1 kHz for the traces shown in (A) versus their proximity to the edge. Before bend testing, traces closest to the edge had the lowest impedance, but this difference decreased as the device was exposed to more bend cycles.

as an adhesion promoter between Parylene layers. The higher PMAP and PAMAP peel strength compared to PP also suggested that Al_2O_3 and A-174 could promote adhesion between two layers of Parylene.

In all cases, the top layer of Parylene deposited on Parylene, metal, or Al_2O_3 , always separated from the underlying layer after peel testing. This indicated that the interface between materials (metal and Al_2O_3) deposited onto Parylene possessed greater adhesion than the interface formed when Parylene was deposited onto the same materials. Therefore, strategies to promote adhesion should focus on the substrate–Parylene interface.

Minimum Bending Diameter

The variation in minimum bending diameter between the PMP and the PMAP devices was likely due to the adhesion strength between the two layers. Because materials are stronger in compression than tension, cracks generally initiated on the outside edge of a bent cable and then propagated toward the inner edge. However, if there was poor adhesion between the Al_2O_3 and top layer of Parylene, then the crack could not easily propagate between the two layers of Parylene separated by Al_2O_3 . Conversely, it was observed that for PMAP devices oriented “in”, the metal trace is bonded to the outside layer of Parylene and cracked when the outside layer of Parylene cracks. When there was good adhesion on both sides, such as with the PMP devices, the traces lay along a neutral plane (Figure S3 in Supplementary Material), which reduced forces on the traces and resulted in breaking below the same minimum bending diameter, whether being bent “in” or “out”.

These results suggest that if it is desired to make devices with very tight diameters in one direction, it should be possible to deposit metal on only one side of a material, or make one layer of insulation thinner than the other and then fold the device toward the thinner layer of insulation. Depositing metal on only one side of the Parylene would also make the device prone to curling due to residual stresses in the various layers, which have been measured as 22 MPa (Dabral et al., 1992; Harder et al., 2002), 230 MPa (Branger et al., 1996), and 383 MPa (Tripp et al., 2006), for Parylene, platinum, and Al_2O_3 , respectively. Curling was not observed on our devices because the sandwich structure caused the residual stress in the two layers to cancel out (Branger et al., 1996). Also, the Pt and Al_2O_3 layers are thin compared to Parylene and the radius of curvature is proportional to the square of the thickness of the substrate over the thickness of the film. Accordingly, after the top layer of Parylene was removed from the PMP peel test samples, the side with the metal formed tight curls as would be predicted by the Stoney equation (Eq. 4) (Ohring, 2001) where σ_f and h_f are the internal stress and thickness of the film, E_s , h_s , and ν_s are the Young’s modulus, thickness, and the Poisson’s ratio of the substrate, and κ is the radius of curvature.

$$\sigma_f = \frac{E_s h_s^2 \kappa}{6 h_f (1 - \nu_s)} \quad (4)$$

The results of the minimum bending diameter tests also suggest that improving adhesion when the metal traces lie along the neutral plane should improve bidirectional bending of devices.

Bending Fatigue

The average transverse impedance from a PMP devices (which peel testing showed good adhesion between the Parylene and metal (Figure 5) dropped slightly in magnitude at lower frequencies which are more sensitive to device failure (Fernández et al., 2014), but remained relatively constant at higher frequencies (Figure 12A). The -90° phase of the PMP devices suggested that traces were fully insulated before bending, however, the insulation was in part compromised after the first set of bend

tests as indicated by the drop in phase at lower frequencies. Notably, however, the phase remained capacitive at higher frequencies even when the devices were bent through 100k cycles (**Figure 12B**) indicating that the insulation was at least partially intact.

In contrast to the PMP devices, the PMAP and PAMAP devices showed large variations in normalized magnitude, indicating large changes in the insulation. Increases in the magnitude (**Figure 12E**) could be due to cracks that broke the traces which would increase the impedance. The decrease in impedance back to the baseline after 10k bends could be due to re-established contact between broken traces, or further deterioration of the insulation through delamination. The small phase even at 0 bends (**Figures 12D,F**) indicates that there was an almost immediate intrusion of ions across the insulation to the traces in those devices. The large SE in the magnitude and phase of the PAMAP devices (**Figures 12C,E**) pointed to a wide variation in trace performance, which could be caused by flaws introduced during processing Al_2O_3 (addressed in the next section) or water intrusion.

The phase of the lateral impedance of the PMP devices was greater than -90° at lower frequencies before the devices were bent indicating that there were ionic paths between traces that could not be detected when measuring transverse impedance (**Figure 13B**). However, similar to the transverse impedance, the absence of a broad spectrum decrease in magnitude and the presence of an enduring capacitive phase (-90°) at higher frequencies implied that at least part of the Parylene insulation remained intact up to 100k bend cycles. Also similar to the transverse impedance, the phase of the lateral impedance of Al_2O_3 devices had small phase shifts, which suggested open ion paths between traces (**Figures 13C–F**).

Fatigue cracks in the Parylene were visible in the cables after 2k bends (**Figure 14A**). The number and size of the cracks increased with more bends (**Figure 14B**). The cracks appeared to originate in the Parylene layer and not the metal as the first appearance was on the device edges (in which there was no metal) and the cracks spanned across areas with and without metal (Figures S8A,B in Supplementary Material). In addition, while all devices showed extensive cracking, PMP devices did not exhibit impedance changes. Cracking in Parylene can also occur due to excessive heat (Hsu et al., 2007; Li et al., 2010), when flexed during accelerated lifetime testing (Minnikanti et al., 2013), and in long-term *in vivo* testing (Schmidt et al., 1988).

Failure of Al_2O_3 Devices

Optical micrographs of PMAP and PAMAP devices showed noticeable water intrusion after 100 bend cycles (images were not taken before 100 bend cycles) (Figure S9 in Supplementary Material). Water intrusion was more pronounced on Al_2O_3 devices and could be attributed to the reduced hydrophobicity of Al_2O_3 compared to Parylene and the removal of Al_2O_3 upon exposure to water by etching (Abdulagatov et al., 2011). Based on the impedance measurements, the water intrusion related etching of Al_2O_3 started on the device edges and propagated inward toward the center. These results indicate the open edges on the sides of

Parylene thin-film devices [Figure S1 in Supplementary Material (steps 5 and v)] allowed Al_2O_3 to be etched by the surrounding water. Assuming that Parylene films can sufficiently prevent water condensation, Al_2O_3 should be completely encapsulated in Parylene for devices intended for use in wet applications including implantation.

Conclusion

Thin-film flexible electronics have great potential as implantable devices. Many different thin-film devices have been demonstrated to be used to stimulate and record from many electrically active tissues such as the spinal cord (Rodger et al., 2007), heart (Viventi et al., 2010; Zhao et al., 2013), muscle (Spence et al., 2007), and cortex (Kim et al., 2013; Tolstosheeva et al., 2011). Although these studies have demonstrated the performance of the devices for short periods of time for their given application, there have not been comprehensive methods to test the mechanical properties of such thin-film architectures until failure. In this paper, simple devices and testing methods were presented to test the mechanical properties of thin-film architectures. In particular methods for measuring the peel strength, minimum bending diameter and bending fatigue failure were presented. These methods were then applied to a Parylene–metal–Parylene system with and without an Al_2O_3 layer, which others have used as a water barrier.

Peel tests showed that Al_2O_3 and A-174 can be used to increase the adhesion between two Parylene films, but that adhesion force is less than for Parylene–platinum interfaces. Parylene films deposited onto surfaces did not adhere as well as platinum and Al_2O_3 deposited onto Parylene. Minimum bend diameter tests showed that adding Al_2O_3 underneath the top Parylene allows the devices to be bent into tighter diameters when bent away from the Al_2O_3 , but increased the minimum bending diameter when bent toward the Al_2O_3 , likely due to poor adhesion. EIS performed on devices undergoing fatigue tests, showed that PMP can withstand up to 100k bends without significant evidence of insulation failure. When exposed to water, devices having Al_2O_3 failed earlier due to the etching of the film exposed at device boundaries.

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Supplementary Material

The Supplementary Material for this article can be found online at <http://journal.frontiersin.org/article/10.3389/fmech.2015.00010>

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Conflict of Interest Statement: The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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