



# Methacrylate copolymers and their composites with nano-CdS: synthesis, characterization, thermal behavior, and antimicrobial properties

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## Abstract

Homo- and copolymers of 2-(*N*-phthalimido)ethyl methacrylate (NPEMA) and *p*-chlorophenyl methacrylate (PCPMA) were prepared in *N,N*-dimethyl formamide (DMF) solution at 70 °C using 2,2-azo-bisisobutyronitrile (AIBN) as initiator. The nano-CdS-doped polymer composite of NPEMA and PCPMA was prepared via in situ technique. The homo- and copolymers of NPEMA and PCPMA were characterized using FT-IR spectroscopy and gel permeation chromatography (GPC). The polymer nano composites were characterized using FT-IR spectroscopy, X-ray diffraction, and transmission electron microscopy. The reactivity ratios ( $r_1$  and  $r_2$ ) were obtained from the various linear graphical methods. The values of  $r_1$  (NPEMA) = 0.55 and  $r_2$  (PCPMA) = 1.30 were found from the same graphical methods. The copolymer microstructures were found from the mean sequence length, run number, and dyad fraction. Thermal behavior of polymers and polymer nano composites under nitrogen atmosphere was studied. The activation energies of neat polymers were varied in the range of 56–85 kJ/mol, while 28–56 kJ/mol energies were found for nano-CdS-doped polymer composites. The thermodynamic parameters of thermal degradation were also obtained. Kinetic and thermodynamic parameters were confirming the stability of the neat polymers than polymer nano composites. The polymers were assessed on different microorganisms for obtaining the antimicrobial properties. Overall, the polymers permit 10–52, 20–58, and 18–56% growth of bacteria, fungi, and yeast, respectively.

**Keywords** Copolymerization · Polymer nano composites · Reactivity ratio · Thermal properties · Antimicrobial properties

## Introduction

Copolymerization technique is an adaptable technique to prepare copolymers from two or more different types of monomers. These copolymers have different properties than their individual homopolymers. The acrylates or methacrylates are normally polymerized through various techniques. These techniques are free radical solution polymerization [1], bulk polymerization [2], emulsion polymerization

[3], atom transfer free radical polymerization (ATRP) [4], reversible addition–fragmentation chain transfer free radical polymerization (RAFT) [5], redox polymerization [6], photo polymerization [7] living polymerization [8], etc. It is required to choose the polymerization process to obtain copolymers having desirable chemical, optical, mechanical, and electrical properties. Free radical solution polymerization is a simple, versatile, and effective technique to prepare the acrylate or methacrylate homo- and copolymers. The functionalized copolymers of acrylates were applied in various fields such as biomedical [9], textile [10], coating [11], adhesions [12], food packaging [13], and many more [14, 15]. Knowing the copolymer microstructure is useful to predict the properties of the copolymers. The microstructure of copolymers is easily predicted from the values of reactivity ratios of two monomers and the ratios of these values are possible to find from the various linear and non-linear methods [16–18].

Polymers are the most preferable host materials to prepare the inorganic nano particles (Nps)-doped composites.

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These classes of materials have very good applicable properties (i.e., electrical, mechanical, and optical). Materials having both inorganic and organic characteristics were well researched [19–23]. Recently, many techniques [24–28] were developed to prepare the new potential materials that were used in various fields [29–33] such as biological labeling, light-emitting diodes, transistors, solar cells, organic-based electronics, sensors, optical switching, etc.

The numbers of research reports were published on the polymer composites with nano-metal oxides [34, 35], metal sulfides [36, 37], and many more [38–41]. Among them, semiconducting polymer nano composites of metal sulfides are very useful for their optoelectronic properties. A semiconducting polymer composite of CdS Nps is this class of material. It has broad range of applications in the various fields. The various polymers are used as host materials to prepare the CdS/polymer nano composites. The different CdS-Nps-doped polymer composites are prepared via different techniques. CdS/polystyrene nano-composite was prepared by a chemical route using an ex situ technique [42]. Oxidation polymerization technique was used to prepare CdS/polyaniline nano-composite [43]. CO<sup>+2</sup>-doped CdS/polyvinyl pyrrolidone composite was synthesized by chemical precipitation method [44]. Spin-coating technique was used to prepare thin films of CdS/poly[2-methoxy-5-(2'-ethyl-hexyloxy)-1,4-phenylene vinylene] (MEH-PPV) nano-composite [45]. Several other CdS/polymer nano composites such as CdS/polyamidoamine dendrimer [46], CdS/poly[4-(*n*-acryloyloxyalkoxy)benzoic acid] [47], and many more [48, 49] were also investigated. Although the properties and synthesis of CdS/polymer nano composites are well investigated, yet the effect of CdS Nps on the degradation of polymer in inert atmosphere (i.e., N<sub>2</sub> gas) is not investigated.

The synthesis of macromolecules having antimicrobial properties is useful in various areas like hospitals, dental equipment, water purification systems, food storage, and food packaging units. Polymeric materials having antimicrobial properties are prepared mainly by: (1) polymerization of antimicrobial monomers which have characteristic functional groups responsible for antimicrobial properties and (2) chemical modification of synthetic polymers by antimicrobial agents or biocides. In this article, we have synthesized chlorine containing copolymers and assessed various microorganisms to confirm the copolymer's antimicrobial potency.

The present paper covers the synthesis, characterization, thermal behavior, and antimicrobial screening of homo- and copolymers of NPEMA with PCPMA. Polymer composites with nano-Cds were prepared and characterized by spectroscopic tools. The main objective of this paper is to study the effect of CdS Nps on thermal degradation of polymer matrix in nano composites.

## Experimental

### Materials

Ethanol amine, phthalic anhydride, triethyl amine (TEA), *N,N*-dimethyl formamide (DMF), methanol, cadmium nitrate, sodium sulfide, and 2,2-azo-bis-isobutyronitrile (AIBN) were purchased from the Loba chem Pvt. Ltd. (India). The nutrient broth, potato dextrose broth, and tryptone glucose yeast extract were purchased from the Himedia Laboratories Pvt. Ltd. The pre-grown slants of various microorganisms (i.e., bacteria, yeast, and fungi) were obtained from the Microbial-Type Culture Collection and Gene Bank (MTCC), CSIR-Institute of Microbial Technology, Chandigarh, India. All chemicals are analytical grade and are used as received.

### Synthesis

#### Synthesis of starting materials and monomers

The synthesis of methacryloyl chloride (MAC), *N*-(2-hydroxy ethyl) phthalimide (NHEP), and 2-(*N*-phthalimido) ethyl methacrylate (NPEMA) was prepared, as reported in Ref. [50]. The *p*-chlorophenyl methacrylate (PCPMA) was prepared by earlier reported synthesis [51].

#### Copolymerization of NPEMA with PCPMA

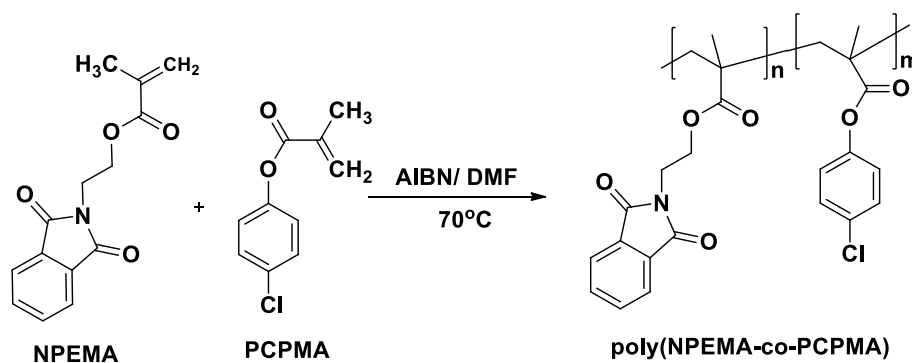
The free-radical polymerization technique [50] was employed to synthesize copolymers of NPEMA with PCPMA having different feed compositions. *N,N*-Dimethyl formamide (DMF) was used as a solvent and AIBN as a free-radical initiator. The feed composition of monomers is given in Table 1. Appropriate quantities of monomers, DMF, and AIBN (0.5% w/w based on total monomers 1 and 2) were added to a flask fitted with reflux condenser. The reaction mixture was heated to 70 °C for 5 h with stirring. It was kept for cooling at room temperature, and then, the resulting polymer solution was slowly poured in a large volume of methanol with stirring. The polymer is followed out. It was filtered and then washed with methanol. Solid polymers were purified by repeated precipitation from DMF solution using methanol. Synthesis of copolymer is shown in Scheme 1.

#### Preparation of CdS-Nps-doped polymer composites

Polymer (0.5 g) dissolved in 50-ml DMF with stirring in a 250-ml conical flask, and then, the flask was kept for sonication. Under sonication, cadmium nitrate solution (0.1 M) in

**Table 1** Monomer feed ratio of NPEMA with PCPMA

Sample code	Monomer feed composition						Composition of NPEMA in copolymer	% Yield
	$M_1$ (NPEMA)			$M_2$ (PCPMA)				
	mol	g	wt%	mol	g	wt%		
B1	1.0	259	100	–	–	–	100	80
B2	0.8	207.2	80	0.2	39.3	20	81	75
B3	0.6	155.4	60	0.4	78.6	40	61	80
B4	0.5	129.5	50	0.5	98.3	50	51	78
B5	0.4	103.6	40	0.6	117.9	60	40	65
B6	0.2	51.8	20	0.8	157.2	80	20	63
B7	–	–	–	1	196.5	100	–	78

**Scheme 1** Copolymerization (B2–B6) of NPEMA with PCPMA

DMF was added slowly to polymer solution. Then, 0.1-M sodium sulfide (0.11 M) solution prepared in methanol was added dropwise to the solution containing cadmium nitrate and polymer solution under sonication. After complete addition, the solution becomes yellowish and cloudy. The flask is putted for 4 h at room temperature to settle the yellow color polymer composites of nano-CdS. The obtained polymer nano composites were separated by ultracentrifugation. The nano composites were washed several times with methanol to remove unreacted or excess reagents. The polymer nano-composite is dried into a vacuum desiccator.

### Characterization

The  $^1\text{H-NMR}$  resonances were recorded with a Bruker 400 MHz FT NMR spectrophotometer. The IR spectra of solid samples in KBr pellets were obtained from Nicolet 6700 FT-IR spectrophotometer. Copolymer compositions and reactivity ratios were determined by spectroscopic data from UV–visible–NIR Shimadzu-3600 spectrophotometer. The phase and crystallographic structures of CdS in polymer nano composites of NPEMA/PCPMA were identified by X-ray diffraction (XRD, Bruker D2 Phaser 3600 X-ray Diffractometer having Cu- $\alpha$  radiation source,  $\lambda=0.154$  nm). The morphology of polymer nano composites of CdS was derived using transmission electron microscope

(Tecnai 20, Philips, Holland, with W-emitter and LaB6 as electron source and accelerating voltage up to 200 kV). Thermal analysis was performed at 10 °C/min heating rate on Mettler-Toledo thermogravimetric analyzer for TGA/DSC-1.

### Antimicrobial screening

The quantitative method [50] was used to assess antimicrobial properties. The homo- and copolymers of NPEMA/PCPMA tested against different microorganisms which are commonly employed for biocompatibility test. These microorganisms are bacteria (*Escherichia coli*, *Bacillus subtilis*, and *Staphylococcus citreus*), fungi (*Sporotrichum pulverulentum*, *Aspergillus niger*, and *Trichoderma lignorum*), and yeast (*Candida utilis*, *Pichia stipitis*, and *Saccharomyces cerevisiae*). The antimicrobial activity of poly(NPEMA) was already discussed [50]. All microorganism cultures were prepared from their respected slants. All microorganism cultures were sub-cultured periodically and maintained in their respected media. The bacteria, yeast, and fungi cultures were kept overnight at 37, 30, and 27 °C, respectively. In the present study, the bacterial culture's %OD is set at 0.1 in 600 nm which corresponds to  $10^8$  CFU/ml. The yeast culture's %OD was set at 0.8 in 600 nm which corresponds to  $2 \times 10^6$  CFU/ml. The fungi culture was prepared



by inoculation with 0.7-mm plug (spore) containing actively growing fungi's slants. Each test was carried out three times and the mean results are shown in a bar graph.

## Results and discussion

### $^1\text{H-NMR}$ spectroscopy

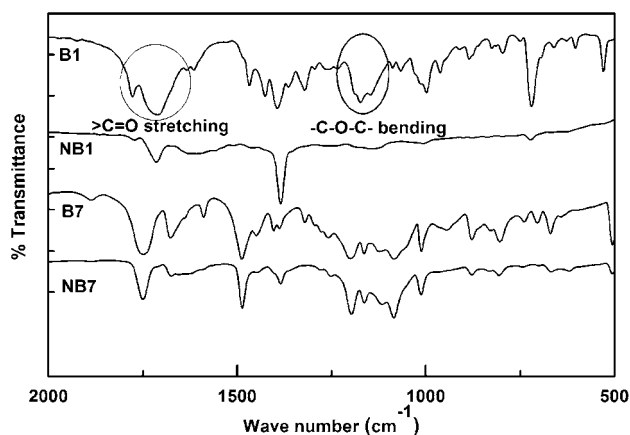
The monomers NPEMA and PCPMA were characterized from the proton NMR spectroscopy. The proton NMR spectra of NPEMA monomer have been discussed in earlier report [50].

*NPEMA*  $^1\text{H-NMR}(\text{CDCl}_3)$   $\delta$  (ppm) = 7.5–7.8 (4H, m, Ar-CH), 5.9 (1H, d, vinylic-H), 5.5 (1H, d, vinylic-H), 4.3 (2H, t,  $-\text{CH}_2-\text{O}$ ), 3.9 (2H, t,  $-\text{CH}_2-\text{N}$ ), and 1.9 (3H, s,  $\text{CH}_3$ ).

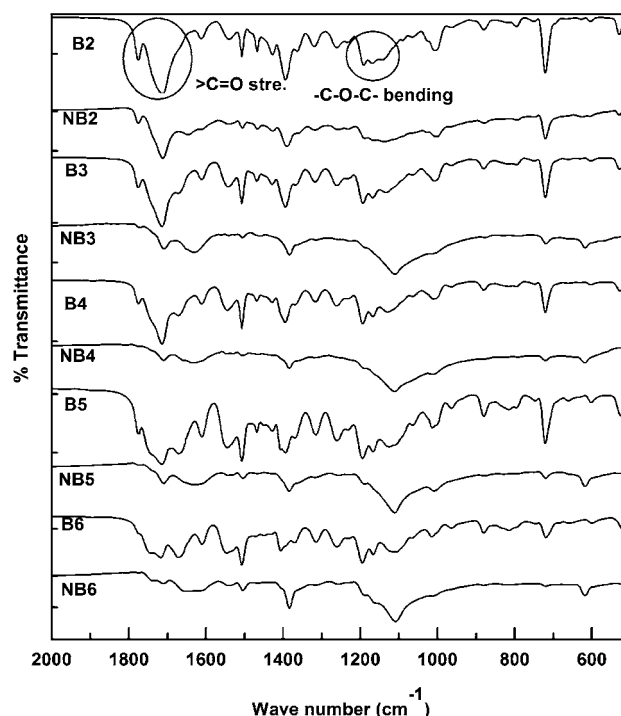
*PCPMA*  $^1\text{H-NMR}(\text{CDCl}_3)$   $\delta$  (ppm) = 6.9–7.3 (3H, m, Ar-CH), 6.4 (1H, d, vinylic-H), 5.7(1H, d, vinylic-H), 2.3 (3H, s, Ar- $\text{CH}_3$ ), and 2.1 (3H, s,  $\text{O}=\text{C}-\text{C}-\text{CH}_3$ ).

### FT-IR spectroscopy

The combined FT-IR spectra ( $2000\text{--}500\text{ cm}^{-1}$ ) of homo- and copolymers of NPEMA with PCPMA and their polymer composites with nano-CdS are shown in Figs. 1 and 2, respectively. From the spectra of neat polymers, the bands observed at  $\sim 1750\text{ cm}^{-1}$  are assigned for the stretching vibration of  $-\text{C}=\text{O}$  in ester group. Another absorbance observed at  $\sim 1710\text{ cm}^{-1}$  corresponds to  $-\text{C}=\text{O}$  stretching in phthalimide [50]. The bands between  $1600$  and  $1400\text{ cm}^{-1}$  are assigned to the aromatic ring breathing vibration. The same band is also observed in all copolymers IR spectra. Medium bands  $\sim 1465$  and  $\sim 1387\text{ cm}^{-1}$  correspond to asymmetrical and symmetrical bending vibration of methyl group, respectively. The bands at  $1173$ ,  $1146$ , and  $1048\text{ cm}^{-1}$  are seen in



**Fig. 1** FT-IR spectra of poly(NPEMA) (B1); poly(PCPMA) (B7) and nano-CdS-doped polymer composite of poly(NPEMA) (NB1); poly(PCPMA) (NB7)



**Fig. 2** FT-IR spectra of poly(NPEMA-co-PCPMA) (B2–B6) and nano-CdS-doped polymer composite of poly(NPEMA-co-PCPMA) (NB2–NB6)

the homopolymers of NPEMA, whereas the bands at  $1258$ ,  $1200$ ,  $1122$ ,  $1084$ , and  $1012\text{ cm}^{-1}$  are seen in the homopolymers of PCPMA which are attributed to the  $\text{C}-\text{C}(=\text{O})-\text{O}$  and  $\text{O}-\text{C}-\text{C}$ -bending vibration in ester. The bands at  $1258$ ,  $1199$ ,  $1160$ ,  $1122$ ,  $1080$ , and  $1012\text{ cm}^{-1}$  are seen in copolymers of NPEMA and PCPMA which are assigned for the bending vibration of  $\text{C}-\text{C}(=\text{O})-\text{O}$  and  $\text{O}-\text{C}-\text{C}$  in ester. The sharp band at  $720\text{ cm}^{-1}$  is observed in homopolymers of NPEMA. The same absorption band is also observed in the spectra of copolymers and this band's intensity is decreased with decrease in the NPEMA content in the copolymers. The band  $\sim 680\text{ cm}^{-1}$  is assigned for the  $\text{C}-\text{Cl}$ -stretching vibration in homopolymers of PCPMA. The same band is also observed in spectra of copolymers.

In the case of nano-CdS-doped polymer composites, the intensity of the bands is decreased. The intensity of bands corresponding to  $-\text{C}=\text{O}$  ( $\sim 1750$  and  $\sim 1710\text{ cm}^{-1}$ ),  $\text{C}-\text{C}(=\text{O})-\text{O}$  ( $1210\text{--}1164\text{ cm}^{-1}$ ), and  $\text{O}-\text{C}-\text{C}$  ( $1164\text{--}1000\text{ cm}^{-1}$ ) is effectively decreased and becomes broader. This suggests that the attraction between CdS Nps and polymer matrix occurring on these sites which depict the presence of CdS Nps in polymer matrix affects the vibration of  $\text{C}-\text{O}$ . As NPEMA content decreased in copolymer feed, the interaction between polymer matrix and inorganic matrix also decreased. This was due to the less sites that were available for interactions in the poly(PCPMA) than



poly(NPEMA). The band at  $1387\text{ cm}^{-1}$  which corresponds to the symmetric vibration of methyl group is not affected by the presence of CdS Nps. This reveals that no interaction is possible between the methyl and CdS Nps. A weak Van der Waals interaction between polymer matrix and nano-CdS may be envisaged.

### Average molecular weights

GPC instrument equipped with Waters 1525 binary pump, manual injector and connected to Styragel HR 4 column and R.I. detector (waters 2414) is employed to record GPC curves. Tetrahydrofuran (THF) at 1.0 ml/min flow rate is used as a mobile phase throughout the analysis. All the measurements are carried out at 30 °C temperature. About 5–10 mg of each polymer sample is dissolved in 10-ml THF and the resulting solutions are filtered. 20- $\mu\text{l}$  sample solution is injected for the GPC analysis. The average molecular weights, namely,  $\overline{M}_n$ ,  $\overline{M}_w$ ,  $\overline{M}_z$ ,  $\overline{M}_z + 1$ , and polydispersity, are shown in Table 2. The GPC data for poly(NPEMA-co-PCPMA) provided the values of  $\overline{M}_n$ ,  $\overline{M}_w$ ,  $\overline{M}_z$ , and  $\overline{M}_z + 1$  which vary from 20,109 to 22,381, 31,518 to 34,804, 47,784 to 51,883, and 65,954 to 71,200, respectively, for copolymers with different feed ratio of the monomers; polydispersity index lies between 1.54 and 1.60 (Table 2). The results reveal that molecular weights decrease and polydispersity index changes randomly as the NPEMA content decreases in the copolymers. The values of  $\overline{M}_n$ ,  $\overline{M}_w$ ,  $\overline{M}_z$ , and  $\overline{M}_z + 1$  are, respectively, 26,052, 43,620, 67,474, 91, and 839 for poly(PCPMA) and the polydispersity index is 1.67. The polydispersity of homopolymers and copolymers was nearly 1.5, indicating termination of growing chain by radical combination (Table 2).

### X-ray diffraction study and transmission electron microscopy

Powder XRD pattern of nano-CdS-doped poly(NPEMA-co-PCPMA) (0.5/0.5 mol ratio) is shown in Fig. 3. The three

**Table 2** Average molecular weights of homo- (B1 and B7) and copolymers (B2–B6) of NPEMA with PCPMA

Sample no.	Average molecular weights (in Dalton)				Polydispersity ( $\bar{D}$ )
	$\overline{M}_n$	$\overline{M}_w$	$\overline{M}_z$	$\overline{M}_z + 1$	
B1	18310	26613	37812	50279	1.45
B2	20109	31518	47784	65954	1.57
B3	21080	33782	51641	71153	1.60
B4	21225	33511	50566	69333	1.58
B5	22127	34044	50094	67778	1.54
B6	22381	34804	51883	71200	1.56
B7	26052	43620	67474	91839	1.67

**Table 3** Reactivity ratios obtained for NPEMA and PCPMA monomers using different linear methods

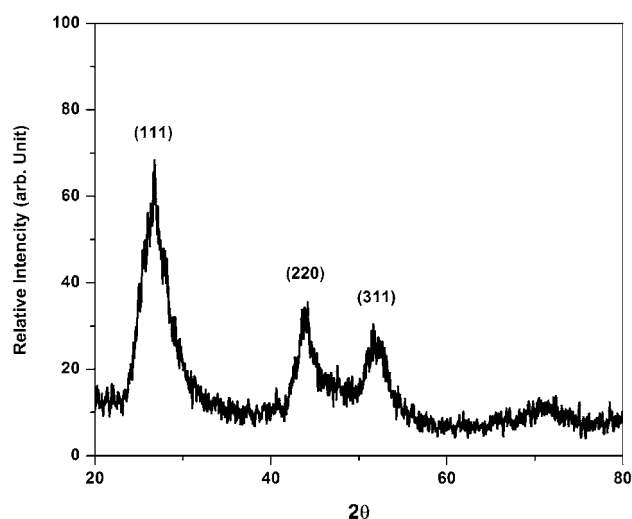
Method	Reactivity ratio				
	$r_1$	$r_2$	$r_1 \times r_2$	$1/r_1$	$1/r_2$
F–R	0.53	1.29	0.68	1.89	0.78
Invert. F–R	0.62	1.37	0.85	1.61	0.73
K–T	0.53	1.27	0.67	1.89	0.79
Ext. K–T	0.5	1.28	0.64	2.00	0.78
Average	0.55	1.30	0.71	1.83	0.77

peaks are observed in each composite which corresponds to the three reflection planes. These reflection planes [i.e., (111), (220), and (311)] confirm that the CdS Nps are cubic and match with JCPDE number 089-0440. The size of CdS nano particles is calculated using Debye–Scherrer formula [52] for (111) reflection plane of the XRD pattern. The size of CdS Nps varied between 4.11 and 5.32 nm.

Figure 4 shows the TEM images of the nano-CdS-doped poly(NPEMA-PCPMA) (0.5/0.5) (NB4) composite. The images clearly show the formation of CdS nano composites with a spherical shape and homogeneous distribution in polymer matrix.

### Monomer reactivity ratio

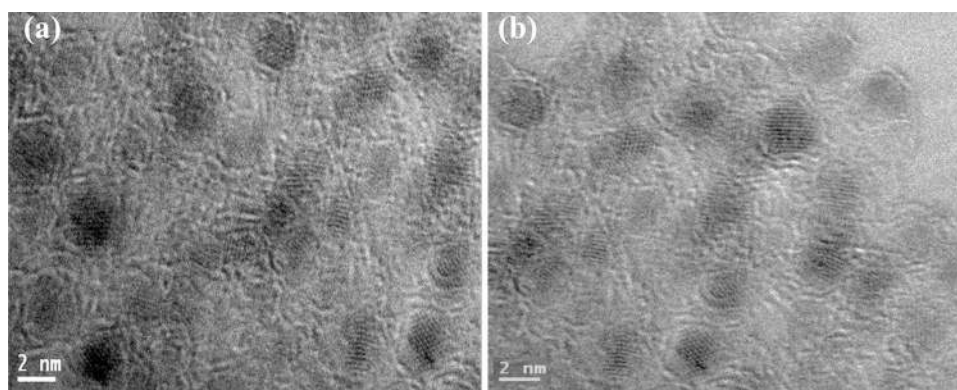
The copolymer composition depends on the monomer feed composition and on the relative monomer reactivity. The UV spectroscopy was employed to find the concentration of NPEMA monomer in copolymer. The procedure to find concentration of NPEMA in copolymers using UV spectroscopy was already discussed [50]. The monomer reactivity ratios



**Fig. 3** XRD pattern of nano-CdS-doped poly(NPEMA-co-PCPMA) composite (mole ratio: 0.5/0.5) (NB4)



**Fig. 4** TEM images of nano-CdS-doped poly(NPEMA-co-PCPMA) composite (mole ratio: 0.5/0.5) (NB4)



were determined using four linear methods Fineman–Rose (F–R) [53], inverted Fineman–Rose (Invert. F–R), Kelan–Tudos (K–T) [54], and extended Kelen–Tudos (Ext. K–T) [55]. The values of  $r_1$  (NPEMA),  $r_2$  (PCPMA),  $1/r_1$ , and  $1/r_2$  and product of  $r_1$  and  $r_2$  are tabulated in Table 3.

It was seen that the value of  $r_1$  is less than  $r_2$ ; this suggests that the reactivity of NPEMA monomer is less than that of PCPMA. This also proves that the PCPMA units were found more in the copolymer content.  $r_1 < r_2$  suggest that the NPEMA favors the cross propagation, whereas PCPMA favors the homopropagation. As  $1/r_1 > 1/r_2$ , it was concluded that there were more growing radicals with NPEMA ends than with PCPMA ends due to  $r_1 < r_2$ , which identified which monomer amongst the two was more reactive.

## Copolymer microstructure

### Mean sequence length

The mean sequence lengths  $\mu_{\text{NPEMA}}$  and  $\mu_{\text{PCPMA}}$  are calculated using the following equation [56]:

$$\mu_{\text{NPEMA}} = 1 + r_1 \left( \frac{[M_1]}{[M_2]} \right) \quad (1)$$

$$\mu_{\text{PCPMA}} = 1 + r_2 \left( \frac{[M_2]}{[M_1]} \right), \quad (2)$$

where  $[M_1]$  and  $[M_2]$  are concentrations of NPEMA and PCPMA, respectively. The ratio of the mean sequence length distribution  $\mu_{\text{NPEMA}}/\mu_{\text{PCPMA}}$ , which theoretically corresponds to the ratio  $[M_1]/[M_2]$ , is tabulated in Table 4. For example, at  $[M_1] = 20.0\%$  and  $[M_2] = 80.0\%$ , each copolymer segment with  $M_2$  units was approximately six times longer than its adjoining segment with  $M_1$  units. The sequence may be expressed as NPPPPPN, where  $P$  stands for PCPMA and  $N$  is for NPEMA. The number of NPEMA units in copolymer increases with increasing concentration of NPEMA in the feed. The results of mean sequence length and values of  $1/r_1$  and  $1/r_2$  compare each other very well;  $1/r_1$  is greater than  $1/r_2$ , and as expected in copolymers, the homopropagation of PCPMA decreased with decreasing PCPMA in monomer feed, while cross propagation of monomeric units increased with increasing concentration of NPEMA in monomer feed.

### Run number and dyad fraction

From the reactivity ratios of monomers, the run number,  $R_N$ , was determined following Harwood and Reichy [57] as the average number of monomer alternations per 100 monomeric units in a copolymer chain:

$$R_N = \frac{200}{2 + (r_1 \times X) + \left( \frac{r_2}{X} \right)}. \quad (3)$$

**Table 4** Mean sequence length of copolymers of NPEMA with PCPMA

Sample no.	Monomer feed		$\mu_1$	$\mu_2$	$\mu_1:\mu_2$	$\mu_1/\mu_2$	Distribution
	$M_1$	$M_2$					
B2	0.8	0.2	3.2	1.3	3:1	2.41	NNNPNNN
B3	0.6	0.4	1.8	1.8	2:2	0.98	NNPPNN
B4	0.5	0.5	1.6	2.3	2:2	0.67	NNPPNN
B5	0.4	0.6	1.4	2.9	1:3	0.46	NPPPN
B6	0.2	0.8	1.1	6.2	1:6	0.18	NPPPPPN

$\mu_1 = \text{NPEMA (N)}$ ;  $\mu_2 = \text{PCPMA (P)}$

**Table 5** TGA data of homo- (B1 and B7) and copolymers (B2–B6) of NPEMA with PCPMA

Sample no.	% Weight loss at various temperature (°C)					Decomposition temperature range (°C)		$T_{\max}^a$		$T_{50}^b$	IPDT <sup>c</sup>
	200	300	400	500	600	Step-I	Step-II	Step-I	Step-II		
								Step-I	Step-II		
B1	2	22	65	100	100	172–318	318–430	281	405	377	351
B2	4	35	85	98	99	170–314	314–446	257	394	357	340
B3	3	33	79	97	97	173–314	314–449	260	394	364	333
B4	4	38	83	98	98	165–316	316–443	261	395	351	334
B5	3	35	74	98	98	170–330	330–457	269	400	365	341
B6	3	30	74	96	96	203–333	333–451	275	407	372	347
B7	2	16	37	95	95	227–397	397–530	349	472	448	437

<sup>a</sup>Temperature for maximum rate of decomposition

<sup>b</sup>Temperature for 50% weight loss

<sup>c</sup>Integral procedural decomposition temperature by Doyle's method

The  $R_N$  value provides a view of sequence distributions in copolymer chains, and consequently, it is possible to assess the relationship between physical properties of copolymers and their compositions. The value of  $R_N$  can be calculated from reactivity ratios. The variation of  $R_N$  with  $X$  ( $= [M_1]/[M_2]$ ) is shown in Fig. 5a.

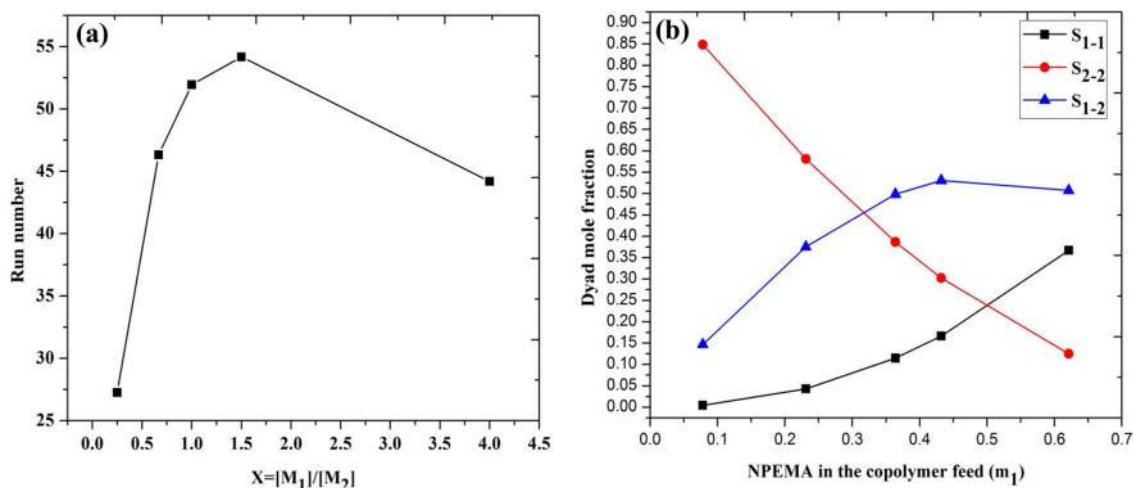
The maximum value of  $R_N$  is 54.2 at  $X=1.5$  for the poly(NPEMA-co-PCPMA) system. To gain further information about the copolymer structure, the formation probabilities of dyad fractions as a function of the molar fraction of monomeric units in the copolymer can be calculated from the monomer feed compositions and reactivity ratios [58]:

$$S_{1-1} = m_1 - \left\{ 2m_1m_2 / \left( 1 + \left[ (2m_1 - 1)^2 + 4r_1r_2m_1m_2 \right]^{1/2} \right) \right\} \quad (4)$$

$$S_{2-2} = m_2 - \left\{ 2m_1m_2 / \left( 1 + \left[ (2m_1 - 1)^2 + 4r_1r_2m_1m_2 \right]^{1/2} \right) \right\} \quad (5)$$

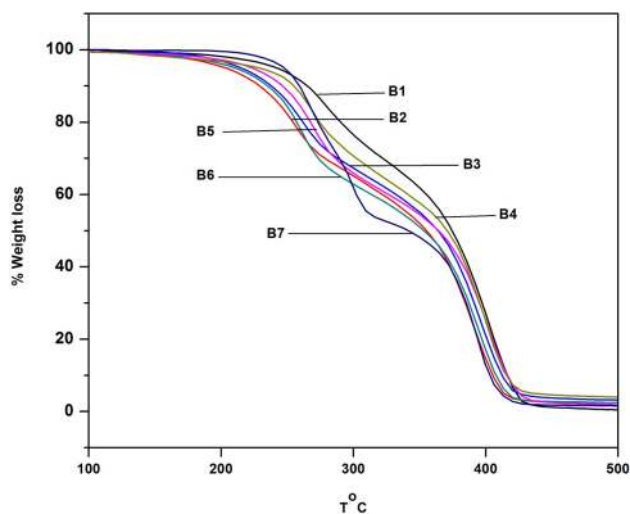
$$S_{1-2} = 4m_1m_2 / \left( 1 + \left[ (2m_1 - 1)^2 + 4r_1r_2m_1m_2 \right]^{1/2} \right). \quad (6)$$

The variation of the dyad fractions with the NPEMA mole fraction in the copolymers is depicted in Fig. 5b. Figure 5b shows  $S_{1-1}$  is less than  $S_{2-2}$ , suggesting that NPEMA has negligible attraction with NPEMA monomer, and so, NPEMA homopolymerization is not favorable. The value of  $S_{1-2}$  implied that homopolymerization of PCPMA is quite dominant. The consequences of these calculations are the same as those obtained from reactivity ratios.

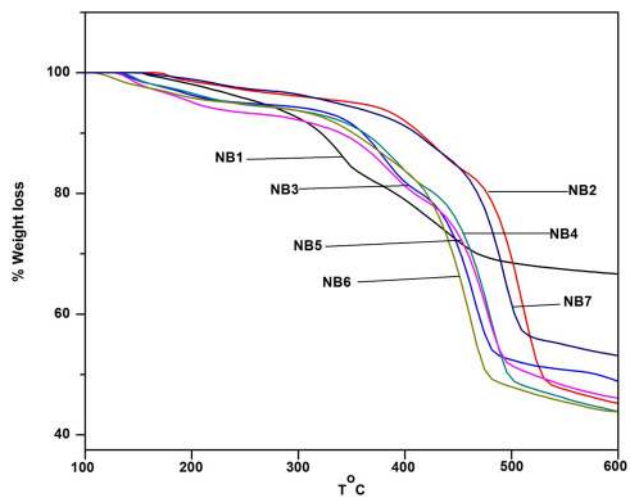


**Fig. 5** **a** Run number for NPEMA and PCPMA monomer pair; **b** dyad monomer sequence fractions vs. the NPEMA mole fractions for the copolymers NPEMA with PCPMA





**Fig. 6** TGA traces of neat homo- (B1 and B7) and copolymers (B2–B6) of NPEMA/PCPMA



**Fig. 7** TGA traces of nano-CdS-doped homo- (NB1 and NB7) and copolymers (NB2–NB6) of NPEMA/PCPMA

## Thermal analysis

### Thermal stability of polymer and their composites with nano-CdS

Thermogravimetric analysis (TGA) technique was used to investigate the thermal stability of polymers and their composite with nano-CdS. TGA traces of homo- (B1 and B7) and copolymers of NPEMA with PCPMA (B2–B6) and its nano composites (NB1–NB7) are shown in Figs. 6 and 7, respectively. The % weight loss at various temperatures, decomposition range,  $T_{\max}$ ,  $T_{50}$ , and integral decomposition temperature (IPDT) for homo- and copolymers of NPEMA with PCPMA are tabulated in Table 5. The TGA traces show

that the decomposition of polymers occurs in two steps. The first decomposition step was in the range of 165–397 °C, whereas the range between 314 and 530 °C was observed for second decomposition step. The first step seen at lower temperature range corresponds to the smaller polymeric chain degradation. The second step observed at higher range corresponds to the scission of main polymeric chain. Poly(PCPMA) has higher thermal decomposition steps than other polymers which confirms that the thermal degradation of poly(PCPMA) was slower. It was also notable (from Table 5) that the % weight loss of poly(PCPMA) at 700 °C was slightly lower than the poly(NPEMA), whereas the % weight loss of copolymers (B2–B6) was relatively higher than each other. The IPDT of homopolymers of NPEMA and PCPMA is, respectively, 351 and 467 °C. The IPDT of copolymers of NPEMA with PCPMA varied between 333 and 347 °C. It was concluded that with the increase of PCPMA feed in copolymers, the thermal stability of copolymers (B2–B6) was also increased. Overall, the neat polymers and polymer nano composites show the two-step thermal degradation.

### Kinetic and thermodynamic parameters of the thermal degradation

The stability of polymers and their composites with nano-CdS was also confirmed from the kinetic (i.e.,  $E_a$ ) and thermodynamic (i.e.,  $\Delta H$ ,  $\Delta S$ , and  $\Delta G$ ) parameters of the thermal degradation of respective polymers and polymer nano composites. The kinetic parameters of thermal degradation were obtained using Briodo (BR) [59] and Coats–Redfern (CR) [60] methods. The activation energy (by BR) was evaluated from the slope of the  $\ln [1/y]$  vs.  $1/T$  plot, whereas activation energy (by CR) was obtained from the slope of  $\ln [-\ln (1-y)/T_2]$  vs.  $1/T$  plot. In these entire calculations,  $y = (W_t - W_\alpha)/(W_0 - W_\alpha)$ , where  $W_t$  is the weight at temperature,  $W_0$  is initial weight, and  $W_\alpha$  is the weight at the end of pyrolysis.

Activation energies (in kJ/mol) for homopolymers of NPEMA and PCPMA for both degradation steps obtained from Briodo method were, respectively (56 and 83 kJ/mol) and (80 and 145 kJ/mol); the same from Coats–Redfern were (56 and 80 kJ/mol) and (85 and 150 kJ/mol). The energy of activation in copolymers B2 (80:20; NPEMA:PCPMA) obtained from Briodo was (50 and 69 kJ/mol) and Coats–Redfern provided the same values. The energies of activation for other copolymers (B3–B7) with different monomer feed are shown in Table 6. It was noticeable that the activation energy of the first degradation step was lower than that of the second step. This concludes that less energy is required for degradation of shorter polymeric chains, whereas higher activation energy corresponds to the higher energy needed for degradation of higher polymeric chains.



**Table 6** Kinetic and thermal parameters of degradation for homo- (B1 and B7) and copolymers (B2–B6) of NPEMA with PCPMA

Kinetic/thermal parameters	Method	Steps	Polymer numbers						
			B1	B2	B3	B4	B5	B6	B7
$E^*$ (kJ/mol)	BR	I	56	50	62	53	54	56	80
		II	83	69	69	90	72	80	145
	CR	I	56	50	62	53	54	56	85
		II	80	69	69	90	72	80	150
$\Delta S^*$ (J/mol K) <sup>a</sup>	BR	I	–249	–242	–263	–246	–248	–251	–285
		II	–272	–254	–253	–284	–257	–261	–342
	CR	I	–190	–197	–187	–198	–201	–199	–193
		II	–203	–195	–221	–205	–214	–195	–197
$\Delta H^*$ (kJ/mol) <sup>b</sup>	BR	I	52	46	58	48	50	52	75
		II	77	63	63	85	66	74	139
	CR	I	52	46	58	48	50	52	40
		II	75	63	63	85	66	74	144
$\Delta G^*$ (kJ/mol) <sup>c</sup>	BR	I	190	174	198	180	184	187	239
		II	261	232	232	274	239	255	394
	CR	I	157	150	158	154	159	159	151
		II	213	193	210	221	210	205	290

<sup>a, b, c</sup> These three parameters are thermodynamic parameters. Those are calculated from the values of activation energies ( $E^*$ ) and pre-exponential factor ( $A$ ). So, these parameters are not belongs to Broido or Coat-redferns method

The activation energy of poly(PCPMA) was higher than that of poly(NPEMA). The activation energies of copolymers (B2–B6) randomly increased or decreased with increase in PCPMA content in copolymer feed.

Thermodynamic parameters (i.e.,  $\Delta S$ ,  $\Delta H$ , and  $\Delta G$ ) of thermal degradation were also useful to confirm the thermal stability of polymers. The value of enthalpy change ( $\Delta H$ ) was positive, which indicated the endothermic nature of thermal degradation of polymers. Lower the value of  $\Delta H$ , lower the difference between potential energy barrier of reagents and activated complex. This suggests that the formation of activated complex was easily favored, and hence, degradation process was faster [61, 62]. How close the system to its equilibrium could define from the value of entropy change ( $\Delta S$ ). Lower the value of  $\Delta S$  suggests that the formation of activated complex was slow and vice versa [61, 62]. The negative value of  $\Delta S$  and positive value of  $\Delta G$  for a decomposition process indicate that steps are non-spontaneous. As the values of  $\Delta G$  increase, the process of formation of activated complex was slow which means that thermal degradation process was slower and vice versa.

The values of thermodynamic parameters for neat polymers indicate that poly(NPEMA) (B1) was less stable than the poly(PCPMA) (B7) and poly(NPEMA-co-PCPMA) (B2–B6). It was notified from Table 6, and the first decomposition step has the lowest values of  $E_a$ ,  $\Delta H$ , and  $\Delta G$  than the second decomposition step. This reveals that the first decomposition step was faster than the second

decomposition step. This was further supported from the higher value of  $\Delta S$  for first decomposition step.

The obtained values of kinetic and thermodynamic parameters for polymer composites with nano-CdS are tabulated in Table 7. The values of the activation energy obtained from both Broido and Coats–Redfern were found to be same. The values of activation energies for both steps of degradation for polymer composites of poly(NPEMA) and poly(PCPMA) doped with nano-CdS were 28 and 30 and 27 and 83 kJ/mol, respectively. The values of activation energies for both degradation steps of copolymer composites of poly(NPEMA-co-PCPMA) with nano-CdS varied between 20–34 and 30–78 kJ/mol, respectively. The values of activation energy were found relatively lower for polymer nano composites (NB1–NB7) than the neat polymers (B1–B7). It was also noticed that the values of  $\Delta H$  and  $\Delta G$  for polymer composites were found to be relatively lower than their corresponding neat polymers. In case of  $\Delta S$ , the values were found higher than the neat polymers. Overall, the result reveals that the decomposition of polymer matrix in polymer nano composites was easily favorable than in neat polymers. It was also concluded that the neat polymers were relatively more stable than their respective polymer nano composites. In case of polymer nano composites, the inorganic nano particles can affect the structure of polymers. These result in higher or lower thermal stability of polymer nano composites. Few research papers were found in this area (Table 8). The neat polyamide 6, polyethylene oxide (PEO)/polyvinyl alcohol (PVA), isotactic polypropylene



**Table 7** Kinetic and thermodynamic parameters of nano-CdS-doped homo- (NB1 and NB7) and copolymers (NB2–NB6) of NPEMA and PCPMA

Kinetic/thermal parameters	Method	Steps	Polymer numbers						
			NB1	NB2	NB3	NB4	NB5	NB6	NB7
$E^*$ (kJ/mol)	BR	I	28	27	34	26	20	23	27
		II	30	78	66	74	62	78	83
	CR	I	28	27	34	26	20	23	27
		II	30	78	66	74	62	78	83
$\Delta S^*$ (J/mol K) <sup>a</sup>	BR	I	–203	–203	–213	–200	–193	–196	–202
		II	–202	–271	–255	–265	–247	–269	–277
	CR	I	–182	–183	–184	–182	–182	–183	–181
		II	–183	–186	–188	–186	–186	–184	–184
$\Delta H^*$ (kJ/mol) <sup>b</sup>	BR	I	23	23	29	21	16	18	23
		II	24	73	61	68	56	73	77
	CR	I	23	23	29	21	16	18	23
		II	24	73	61	68	56	73	77
$\Delta G^*$ (kJ/mol) <sup>c</sup>	BR	I	139	135	145	132	123	125	138
		II	158	246	222	237	215	247	256
	CR	I	127	124	130	121	117	120	123
		II	146	191	182	188	177	191	195

<sup>a, b, c</sup> These three parameters are thermodynamic parameters. Those are calculated from the values of activation energies ( $E^*$ ) and pre-exponential factor ( $A$ ). So, these parameters are not belongs to Broido or Coat-redferns method

(iPP), and poly(butylene succinate-co-adipate) (PBSA) have 175, 101, 220.57, and 159.9 kJ/mol of activation energies, respectively. The activation energies of their nano composites, respectively, with clay, ZnO, palladium (Pd), and organically modified synthetic fluorine mica were higher than neat polymers. These suggest that the stronger interaction between polymer matrix and inorganic nano particles than the polymer nano composites has higher value of activation energy than the corresponding neat polymers [63, 66–68]. The activation energies of neat poly(3-hydroxy butyrate) (PHB), polyvinyl ester and nylon 6 were, respectively, 136.8, 186.3, and 276 kJ/mol, while the activation energies of their nano-composite, respectively, with  $Ag_2S$ , *O*-montmorillonite (*O*-MMT), and Glass fibers were found lower than neat polymers. This is due to the weak interaction between polymer matrix and inorganic nano particles [64, 65]. The lower value of kinetic and thermodynamic parameters suggests the catalytic effect arises in polymer nano composites.

### Antimicrobial screening

The functional polymers having chlorine and fluorine groups were mostly used for antimicrobial properties. The effectiveness of homo- and copolymers of NPEMA and PCPMA to hinder the growth of microorganisms is shown in Fig. 8. It appears that compared to poly(NPEMA) and poly(NPEMA-co-PCPMA), the poly(PCPMA) more effectively hinders the microorganism's growth. The copolymers (Fig. 8a) allowed growth of 30–20, 32–25, and 50–38% for *E. coli*, *B. subtilis*,

and *S. citreus*, respectively. The poly(NPEMA) tolerated, respectively, 35, 42, and 52% growth for *E. coli*, *B. subtilis*, and *S. citreus*; poly(PCPMA) allowed, respectively, 10, 18, and 20% growth for the species indicated above.

The effect of homo- and copolymers of NPEMA and PCPMA on the growth of different fungi is demonstrated in Fig. 8b. It is seen that the hindrance to growth by poly(NPEMA) was less than their corresponding copolymers with PCPMA. The effect of poly(NPEMA) on the fungi has already been indicated in the preceding section. Growths of *A. niger*, *S. pulverulentum*, and *T. lignorum*, respectively, were 30, 20, and 34% in poly(PCPMA). The copolymers allowed 52–28, 48–28, and 58–40% growth of *A. niger*, *S. pulverulentum*, and *T. lignorum*, respectively.

The poly(NPEMA-co-PCPMA) and poly(PCPMA) were more effective than the poly(NPEMA) to reduce the growth of yeast (Fig. 8c). The poly(NPEMA) permitted 40, 50, and 58% growth of *S. cerevisiae*, *C. utilis* and *P. stipitis*, respectively; poly(PCPMA) allowed, respectively, 18, 32, and 28% growth of the same. The copolymers tolerated 38–22, 50–38, and 56–40% growth of *S. cerevisiae*, *C. utilis*, and *P. stipitis*, respectively.

In general, poly(PCPMA) hindered the growth of all the microorganisms effectively, while the poly(NPEMA) was less effective than poly(PCPMA) and their copolymers. This may be traced to the presence of chlorine in poly(PCPMA) and poly(NPEMA-co-PCPMA). It was observed that increasing the concentration of PCPMA in copolymers resulted in increase of antimicrobial properties



**Table 8** Activation energies for other polymer nano composites

No.	Polymer/polymer nano-composite	Method name	Activation energy $E_a$ (KJ/mol)	References
1.	Polyamide 6	Kissinger	175	[63]
	Polyamide 6/clay		199	
2.	Poly(3-hydroxy butyrate) (PHB)	Kissinger	136.8	[64]
	0.05% Ag <sub>2</sub> S/PHB <sub>2000</sub>		110.2	
	0.12% Ag <sub>2</sub> S/PHB <sub>1000</sub>		95.1	
	0.5% Ag <sub>2</sub> S/PHB <sub>250</sub>		91.8	
	1.28% Ag <sub>2</sub> S/PHB <sub>100</sub>		94.5	
	2.6% Ag <sub>2</sub> S/PHB <sub>50</sub>		99.1	
3.	Polyvinyl ester	Coats–Redfern	186.3	[65]
	Polyvinyl ester/1% O-montmorillonite (O-MMT)		173.4	
	Polyvinyl ester/5% O-MMT		154.9	
4.	Polyvinyl ester	Briodo	190.2	[65]
	Polyvinyl ester/1% O-montmorillonite (O-MMT)		185.2	
	Polyvinyl ester/5% O-MMT		166.7	
5.	Polyethylene oxide (PEO)/polyvinyl alcohol (PVA)	Coats–Redfern	101	[66]
	PEO-PVA/1% ZnO		122	
	PEO-PVA/5% ZnO		130	
5.	Polyethylene oxide (PEO)/polyvinyl alcohol (PVA)	Briodo	159	[66]
	PEO-PVA/1% ZnO		164	
	PEO-PVA/5% ZnO		169	
6.	Isotactic polypropylene (iPP)	Flynn-Wall	220.57	[67]
	iPP/0.27% Pd		227.85	
7.	Nylon 6	Coats–Redfern	264	[40]
	Nylon 6/Glass fiber		181	
	Nylon 6/3 5% crysnano nano-clay		219	
	Nylon 6/5% crysnano nano-clay		247	
8.	Nylon 6	Briodo	276	[40]
	Nylon 6/glass fiber		193	
	Nylon 6/3 5% crysnano nano-clay		231	
	Nylon 6/5% crysnano nano-clay		259	
9.	Poly(butylene succinate-co-adipate) (PBSA)	Kissinger	159.9	[68]
	PBSA/OSFM (organically modified synthetic fluorine mica)		168.9	

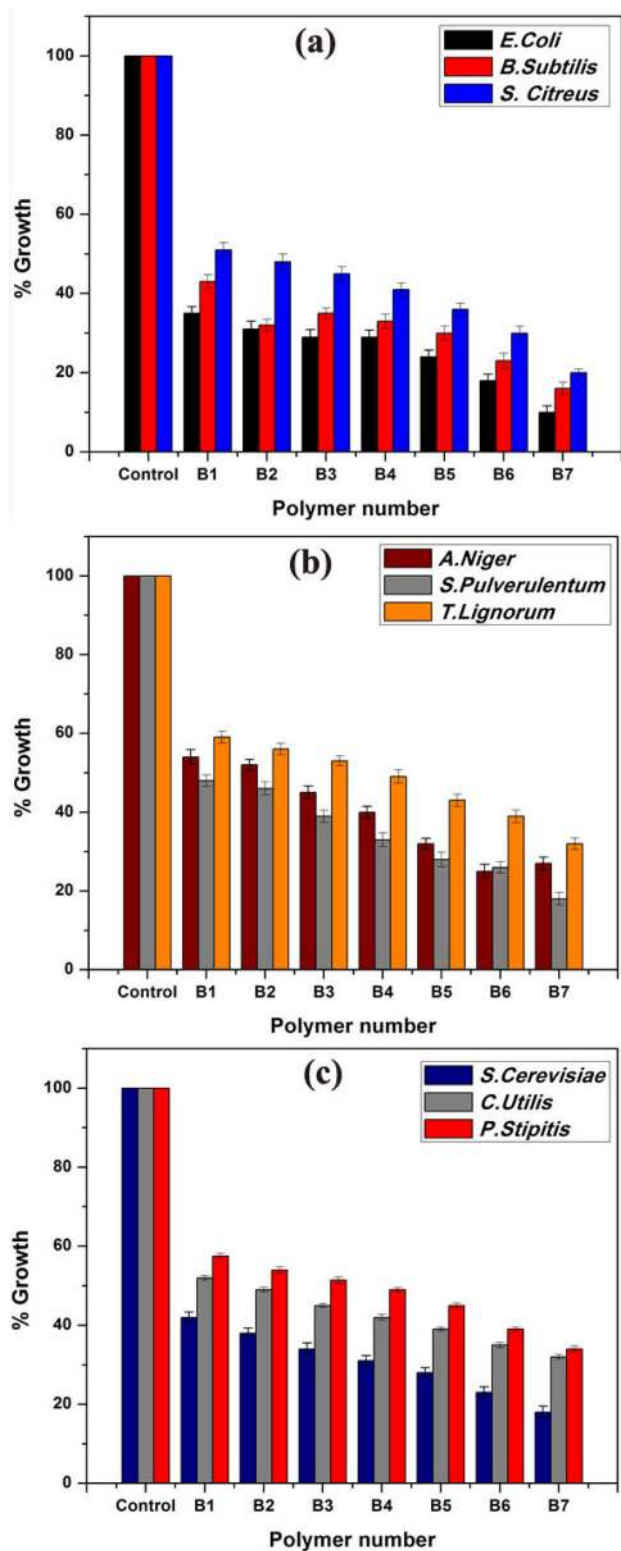
and vice versa. It was further seen that the homo- and copolymers of PCPMA effectively hindered the growth of bacteria, while the growth of yeast and fungi was moderately hindered.

Overall, the percentage of NPMA in the copolymers increased, the effectiveness of the copolymers to inhibit the growth of microorganisms decreased, while increase in the percentage of monomers which have active groups (i.e., –Cl) will increase the ability of polymers to inhibit the growth of microorganisms. This investigation indicates that monomers containing chlorine can be used to prepare polymers which can be used as general antimicrobial agents in water purification systems and food packaging materials.

## Conclusion

The homo- and copolymers (with different monomer feeds) of NPMA and PCPMA were prepared via free radical solution polymerization technique. The relative intensity of different bands (i.e., C=O group, –CH<sub>2</sub> rocking vibration) in FT-IR spectra of copolymers was increased or decreased. This reveals that the composition of monomers in copolymers was well matched. The polydispersity of homo- and copolymers varied between 1.45 and 1.67. Nano-CdS-doped polymer composite was prepared via in situ technique. The nano-CdS-doped polymer composite is characterized by XRD and FT-IR. The XRD pattern confirms the cubic structure of CdS nano particles with three reflection planes (111), (222), and (311). The FT-IR spectra of polymer nano





**Fig. 8** Antimicrobial properties of homo- and copolymers of NP/PCPMA—**a** bacteria; **b** fungi, and **c** yeast

composites confirm the weak interaction between inorganic nano particles and polymer matrix. The TGA analysis of polymers confirms that the poly(PCPMA) was more stable than poly(NPEMA-*co*-PCPMA) and poly(NPEMA). The values of activation energy,  $\Delta S$ ,  $\Delta H$ , and  $\Delta G$ , confirm the stability of poly(PCPMA). The values of activation energy for polymer nano composites were found lower than the neat polymers. This confirms the lower thermal stability of polymer nano composites than the neat polymers. The lower thermal stability of polymer nano composites also confirms from the values of  $\Delta S$ ,  $\Delta H$ , and  $\Delta G$ . It is possible that CdS nano particles act as a catalyst during the degradation of polymer nano composites.

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