

2008, *112*, 12092–12095 Published on Web 07/22/2008

Near-Monodisperse Ni–Cu Bimetallic Nanocrystals of Variable Composition: Controlled Synthesis and Catalytic Activity for H₂ Generation

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Received: July 01, 2008; Revised Manuscript Received: July 09, 2008

Near-monodisperse Ni_{1-x}Cu_x (x = 0.2-0.8) bimetallic nanocrystals were synthesized by a one-pot thermolysis approach in oleylamine/1-octadecene, using metal acetylacetonates as precursors. The nanocrystals form largearea 2D superlattices, and display a catalytic synergistic effect in the hydrolysis of NaBH₄ to generate H₂ at x = 0.5 in a strongly basic medium. The Ni_{0.5}Cu_{0.5} nanocrystals show the lowest activation energy, and also exhibit the highest H₂ generation rate at 298 K.

Bimetallic (e.g., Fe-Pt,^{1a} Au-Ni,^{1b} Co-Pd,^{1c} Pd-Ni,^{1d} Pd-Au,^{1e} Pt-Pd,^{1f,g} Pt-Au,^{1g} and Rh-Pt^{1h}) and trimetallic (e.g., Ru₅PtSn¹ⁱ) nanocrystals (NCs) with tunable chemical and physical properties have attracted extensive theoretical and practical interest.^{1,2} In particular, these NCs usually display composition-dependent surface structure and atomic segregation behavior, so they are important materials for developing new catalysts with enhanced activity and selectivity.1c-i,2 Solution based synthetic approaches such as nonhydrolytic reduction in hot surfactant solutions,1a,d dendrimer templating,1e,f micelle templating,1g and polyol reduction1h have been demonstrated to be versatile and robust methods to control the chemical composition, size, and shape of NCs. The selection of proper metal precursors and regulation of the bonding interactions between particle surfaces and capping molecules play crucial roles in obtaining highly monodisperse and compositionally homogeneous NCs.

Ni–Cu alloys are efficient catalysts for some important heterogeneous reactions such as methane decomposition,^{3a} citral and cinnamaldehyde hydrogenations,^{3b} and electrocatalytic oxidation of methanol.^{3c} So far, several limited physical and chemical methods have been developed to prepare Ni–Cu nanocrystalline particles.⁴ However, the synthesis of mono-disperse, faceted, and phase-pure Ni–Cu nanoscale alloys has still remained a great challenge.

In this Letter, we report the synthesis of near-monodisperse $Ni_{1-x}Cu_x$ (x = 0.2-0.8) NCs by a one-pot thermolysis approach in hot surfactant solutions. The NCs form large-area superlattices and show a catalytic synergistic effect in hydrolysis of NaBH₄ to generate H₂.

For a typical synthesis (Supporting Information Table S1), $0.1 \text{ mmol Ni}(acac)_2$ and $0.1 \text{ mmol of } Cu(acac)_2$ were pre-

dissolved in 1 mL of dry oleylamine (OM) at 85 °C in an oil bath. Dry oleylamine and 1-octadecene (ODE) in a given volume were put into a 50 mL three-necked flask at room temperature. The solvent was heated to 140 °C in an electromantle and evacuated at this temperature for 20 min to remove water and oxygen under magnetic stirring. The solvent was then heated to 230 °C at 10 °C min⁻¹. The predissolved metal precursors were injected into the heated solvent inside the flask with a plastic syringe in 20 s, and were allowed to further react for 10 min at this temperature under Ar. When the reaction was complete, an excess of absolute ethanol was added at room temperature to form a cloudy black suspension. This suspension was separated by centrifugation, and the NCs were collected.

Transmission electron microscopy (TEM, Philips FEI Tecnai 12, 100 kV) showed that $Ni_{1-x}Cu_x$ NCs formed ordered twodimensional (2D) hexagonal-close-packed (hcp) nanoarrays over large areas, demonstrating their low polydispersity and good surface capping by oleylamine (Figure 1 and Supporting Information Figure S1). The crystallite sizes of Ni, Ni_{0.8}Cu_{0.2}, $Ni_{0.6}Cu_{0.4}$, $Ni_{0.5}Cu_{0.5}$, $Ni_{0.4}Cu_{0.6}$, and $Ni_{0.2}Cu_{0.8}$ NCs are 22.3 \pm $3.4, 27.3 \pm 2.4, 23.1 \pm 2.1, 22.7 \pm 1.7, 17.8 \pm 3.4$, and 19.3 \pm 4.4 nm, respectively. High-resolution TEM (HRTEM, Philips CM200/FEG, 200 kV) revealed that the $Ni_{1-x}Cu_x$ NCs are of high crystallinity. The Ni_{0.5}Cu_{0.5} NCs are composed of singlecrystalline particles (Figure 2a), and multiple twined particles (MTPs) (Figure 2b), with exposed (111) faces. Energy dispersive X-ray (EDX) analysis of a single Ni_{0.5}Cu_{0.5} NC indicated that the NCs are 45 atom % Ni and 55 atom % Cu (Figure 2c), confirming the formation of bimetallic NCs with the nominal composition. The X-ray line scan profile along the line indicated in the ADF-STEM image further suggested that the bimetallic NCs are Cu rich in the core region and Ni rich in the surface region (Figure 2d). X-ray diffraction (XRD, Bruker D8 GADDS, $Co-K_{\alpha}$ radiation of $\lambda = 1.79$ Å) suggested that the as-obtained Ni NCs contain a mix of hexagonal (hcp) and face-centered cubic (fcc) phases (Figure 3). As x rose from 0 to 0.05 to 0.1,

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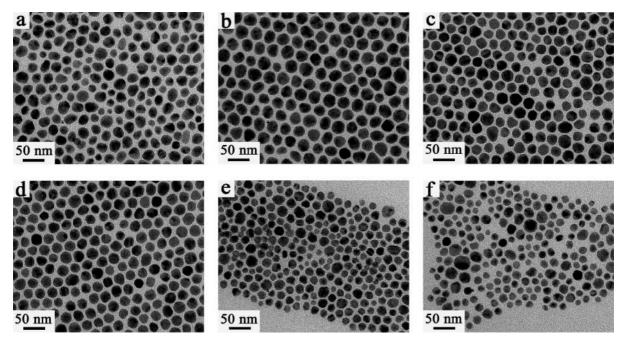


Figure 1. TEM images of Ni_{1-x}Cu_x NCs: (a) x = 0; (b) x = 0.2; (c) x = 0.4; (d) x = 0.5; (e) x = 0.6; (f) x = 0.8.

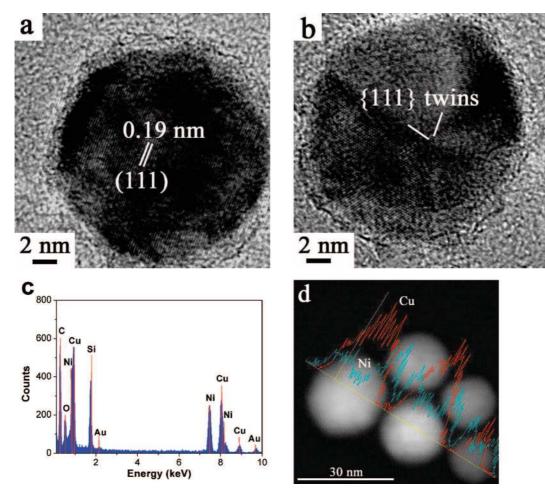


Figure 2. (a and b) HRTEM images of $Ni_{0.5}Cu_{0.5}$ NCs. (c) EDX spectrum of a single $Ni_{0.5}Cu_{0.5}$ NC on a Au grid. (d) ADF-STEM image and X-ray line scan profile along the line indicated in this image.

the amount of cubic phase in the NCs increased. For x = 0.2-0.8, the as-obtained Ni_{1-x}Cu_x NCs adopt an fcc structure (Figure 3), in agreement with previous reports.^{4a,5} The calculated lattice constants were a = 0.3607 nm for the Ni_{0.8}Cu_{0.2} NCs, a = 0.3609 nm for the Ni_{0.6}Cu_{0.4} NCs, a = 0.3613 nm for the

Ni_{0.5}Cu_{0.5} NCs (JCPDS: 71-7832), a = 0.3614 nm for the Ni_{0.4}Cu_{0.6} NCs, and a = 0.3616 nm for the Ni_{0.2}Cu_{0.8} NCs.

Optimal conditions for the formation of high-quality $Ni_{1-x}Cu_x$ NCs were explored through control experiments. Using 0.1 mmol of Ni(acac)₂ and 0.1 mmol of Cu(acac)₂ as precursors,

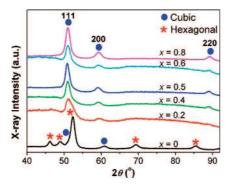


Figure 3. XRD patterns of $Ni_{1-x}Cu_x$ NCs.

the reaction at 220 °C in oleylamine produced poorly crystallized (Supporting Information Figure S2a) polydisperse (Supporting Information Figure S2b) Ni_{0.5}Cu_{0.5} NCs, while the reaction at 250 °C yielded phase-separated (Supporting Information Figure S2a) polydisperse NCs containing very big polygonal and rodlike particles (Supporting Information Figure S2d). Only at temperatures near 230 °C, phase-pure and highly crystalline (Supporting Information Figure S2a) near-monodisperse $Ni_{0.5}Cu_{0.5}$ NCs were formed (22.9 \pm 2.5 nm; Supporting Information Figure S2c). At a fixed temperature of 230 °C, the addition of different amounts of ODE into OM improved the monodispersity and permitted size control of the Ni_{0.5}Cu_{0.5} NCs. For example, 14.9 ± 1.5 nm (Supporting Information Figure S3a), 16.1 \pm 1.4 nm (Supporting Information Figure S3b), and 22.7 ± 1.7 nm (Figure 1d) Ni_{0.5}Cu_{0.5} NCs were obtained at ODE/OM = 0:20, 5:15, and 10:10 (in v/v), respectively. However, as the volume ratio of ODE/OM exceeds 10:10, the Ni_{0.5}Cu_{0.5} NCs grow much bigger and more polydisperse (Supporting Information Figure S3c,d). Therefore, high-quality $Ni_{1-x}Cu_x$ NCs were obtained by finely tuning the ODE/OM ratio at 230 °C (Supporting Information Table S1), due to the delicate control of the solvent's coordinating behavior.^{1a,6}

With the Langmuir–Blodgett (LB) technique,⁷ the Ni_{1-x}Cu_x NCs were deposited onto silicon wafers to form 2D model catalysts after the removal of excess oleylamine ligands by dissolution and precipitation treatment with hexanes and methanol. They were then tested for catalytic hydrolysis of NBH₄ to generate H₂ in a strongly basic medium (0.16 g of NaBH₄ in 5 mL of 15 wt % NaOH) at 298–308 K.⁸ The H₂ generation rates of all nanocatalysts monotonically increased from 298 to 308 K (Figure 4a), and the Ni_{1-x}Cu_x (x = 0.4-0.5) NCs were quite active for this reaction. At 298 K, the H₂ generation rates for the NCs are in the range 439–2127 mL min⁻¹ g⁻¹ (Figure 4b), and were comparable to those reported

for Ni-Co-B catalysts.^{8a} The apparent activation energy for hydrolysis by the Ni NCs was 62.3 kJ mol⁻¹, lower than that of bulk Ni (71 kJ mol⁻¹)^{8b} but close to that of Raney Ni (63 kJ mol⁻¹).^{8b} Activation energies for the Ni_{0.8}Cu_{0.2}, Ni_{0.6}Cu_{0.4}, Ni_{0.5}Cu_{0.5}, Ni_{0.4}Cu_{0.6}, and Ni_{0.2}Cu_{0.8} NCs are 86.6, 89.4, 52.1, 65.3, and 89.9 kJ mol⁻¹, respectively. Consequently, among the tested catalysts, Ni_{0.5}Cu_{0.5} NCs show the lowest activation energy, and also exhibit the highest H₂ generation rate at 298 K (Figure 4b), suggesting a catalytic synergistic effect in the hydrolysis of NaBH₄ at x = 0.5. This catalytic enhancement is supposed to be caused by the combined turnover from surface Ni sites and partially dissolved metal ions such as Ni²⁺ and Cu²⁺ as the NC surfaces were exposed to the strong basic solution.8 Furthermore, the observed activation energy dependence on the copper content in this work might arise from reaction kinetics changes for various compositions, predominately induced by the complicated surface reactions among surface metal atoms, residual capping species, and deionized OH⁻ ions in the presence of H₂ and NaBO₂.⁸

In conclusion, we demonstrated an efficient synthesis of highquality $Ni_{1-x}Cu_x$ (x = 0.2-0.8) bimetallic NCs by a one-pot thermolysis method in oleylamine/1-octadecene, using metal acetylacetonates as precursors. The NCs form large-area 2D superlattices, and display a catalytic synergistic effect in the hydrolysis of NaBH₄ to generate H₂ at x = 0.5 in a strongly basic medium. The Ni_{0.5}Cu_{0.5} NCs show the lowest activation energy, and also exhibit the highest H₂ generation rate at 298 K. The Ni–Cu bimetallic NCs may also be used as catalysts in other selective heterogeneous reactions, and the present synthesis has already been applied to many other bimetallic NCs (e.g., Ni–Pd, Ni–Rh, and Ni–Co).

Acknowledgment. This work was supported by the Director, Office of Science, Office of Advanced Scientific Computing Research, Office of Basic Energy Sciences, Materials Sciences and Engineering and Chemical Sciences, Geosciences, and Biosciences Divisions, of the U.S. Department of Energy under Contract No. DE-AC02-05CH11231. The authors thank Virginia Altoe at Molecular Foundry of the Lawrence Berkeley National Laboratory for X-ray line scan profile analysis. Y.Z. appreciates the financial aid of Huaxin Distinguished Scholar Award from Peking University Education Foundation of China.

Supporting Information Available: More TEM images, XRD results, and detailed experimentation. This material is available free of charge via the Internet at http://pubs.acs.org.

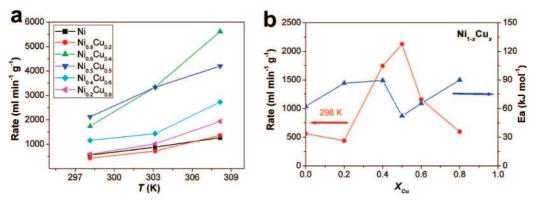


Figure 4. (a) H_2 generation rate as a function of absolute temperature (0.16 g of NaBH₄ in 5 mL of 15 wt % NaOH). (b) H_2 generation rate at 298 K and activation energy as a function of Cu molar fraction.

References and Notes

(a) Sun, S.; Murray, C. B.; Weller, D.; Folks, L.; Moser, A. Science
 2000, 287, 1989. (b) Salem, A. K; Searson, P. C.; Leong, K. W. Nat. Mater.
 2003, 2, 668. (c) Heemeier, M.; Carlsson, A. F.; Naschitzki, M.; Schmal, M.; Bäumer, M.; Freund, H.-J. Angew. Chem., Int. Ed. 2002, 41, 4073. (d)
 Son, S. U.; Jang, Y.; Park, J.; Na, H. B.; Park, H. M.; Yun, H. J.; Lee, J.;
 Hyeon, T. J. Am. Chem. Soc. 2004, 126, 5026. (e) Scott, R. W. J.; Wilson,
 O. M.; Oh, S.-K.; Kenik, E. A.; Crooks, R. M. J. Am. Chem. Soc. 2004, 126, 15583. (f) Ye, H.; Crooks, R. M. J. Am. Chem. Soc. 2004, 126, 15583. (f) Ye, H.; Crooks, R. M. J. Am. Chem. Soc. 2004, 126, 15583. (f) Ye, H.; Radmilovic, V.; Somorjai, G. A.; Yang, P. Nat. Mater. 2007, 6, 692. (h) Park, J. Y.; Zhang, Y.; Grass, M.; Zhang, T.;
 Somorjai, G. A. Nano Lett. 2008, 8, 673. (i) Hungria, A. B.; Raja, R.;
 Adams, R. D.; Captain, B.; Thomas, J. M.; Midgley, P. A.; Golovko, V.;
 Johnson, B. F. G. Angew. Chem., Int. Ed. 2006, 45, 4782.

(2) (a) Greeley, J.; Mavrikakis, M. *Nat. Mater.* **2004**, *3*, 810. (b) Fernández, J. L.; Walsh, D. A.; Bard, A. J. *J. Am. Chem. Soc.* **2005**, *127*, 357. (c) Ferrando, R.; Jellinek, J.; Johnston, R. L. *Chem. Rev.* **2008**, *108*, 845.

(3) (a) Reshetenko, T. V.; Avdeeva, L. B.; Ismagilov, Z. R.; Chuvilin, A. L.; Ushakov, V. A. *Appl. Catal.*, A **2003**, 247, 51. (b) Asedegbega-Nieto, E.; Bachiller-Baeza, B.; Guerrero-Ruíz, A.; Rodríguez-Ramos, I. *Appl.*

Catal., A **2006**, *300*, 120. (c) Jafarian, M.; Moghaddam, R. B.; Mahjani, M. G.; Gobal, F. J. Appl. Electrochem. **2006**, *36*, 913.

(4) (a) Van Ingen, R. P.; Fastenau, R. H. J.; Mittenmeijer, E. J. J. Appl. Phys. 1994, 76, 1871. (b) Pabi, S. K.; Joardar, J.; Manna, I.; Murty, B. S. Nanostruct. Mater. 1997, 9, 149. (c) Cattaruzza, E.; Battaglin, G.; Polloni, R.; Cesca, T.; Gonella, F.; Mattei, G.; Maurizio, C.; Mazzoldi, P.; D'Acapito, F.; Zontone, F.; Bertoncello, R. Nucl. Instrum. Methods Phys. Res., Sect. B 1999, 148, 1007. (d) Damle, C.; Sastry, M. J. Mater. Chem. 2002, 12, 1860.
(e) Foyet, A.; Hauser, A.; Schäfer, W. J. Solid State Electrochem. 2008, 12, 47. (f) Bonet, F.; Grugeon, S.; Dupont, L.; Urbina, R. H.; Guéry, C.; Tarascon, J. M. J. Solid State Chem. 2003, 172, 111.

(5) Massalki, T. B.; Okamoto, H.; Subramanian, P. R. *Binary Alloy Phase Diagrams*, 2nd ed.; ASM International: Metals Park, OH, 1990.
(6) Yin, Y.; Alivisatos, A. P. *Nature* 2005, *437*, 664.

(7) Zhang, Y.; Grass, M. E.; Habas, S. E.; Tao, F.; Zhang, T.; Yang, P.; Somorjai, G. A. J. Phys. Chem. C 2007, 111, 12243.

(8) (a) Ingersoll, J. C.; Mani, N.; Thenmozhiyal, J. C.; Muthaiah, A. J. Power Sources **2007**, 173, 450. (b) Kaufman, C. M.; Sen, B. J. Chem. Soc., Dalton Trans. **1985**, 2, 307.

JP805788X