

New Fe-based superconductors: properties relevant for applications

This article has been downloaded from IOPscience. Please scroll down to see the full text article.

2010 Supercond. Sci. Technol. 23 034003

(<http://iopscience.iop.org/0953-2048/23/3/034003>)

View [the table of contents for this issue](#), or go to the [journal homepage](#) for more

Download details:

IP Address: 130.39.183.12

The article was downloaded on 03/06/2010 at 18:37

Please note that [terms and conditions apply](#).

New Fe-based superconductors: properties relevant for applications

M Putti¹, I Pallecchi¹, E Bellingeri¹, M R Cimberle¹, M Tropeano¹, C Ferdeghini¹, A Palenzona¹, C Tarantini², A Yamamoto², J Jiang², J Jaroszynski², F Kametani², D Abraimov², A Polyanskii², J D Weiss², E E Hellstrom², A Gurevich², D C Larbalestier², R Jin³, B C Sales³, A S Sefat³, M A McGuire³, D Mandrus³, P Cheng⁴, Y Jia⁴, H H Wen⁴, S Lee⁵ and C B Eom⁵

¹ CNR-INFN-LAMIA and Università di Genova, via Dodecaneso 33, I-16146 Genoa, Italy

² Applied Superconductivity Center, National High Magnetic Field Laboratory, Florida State University, 2031 East Paul Dirac Drive, Tallahassee, FL 32310, USA

³ Materials Science and Technology Division, Oak Ridge National Laboratory, Oak Ridge, TN 37831, USA

⁴ Institute of Physics, Chinese Academy of Sciences, Beijing 100190, People's Republic of China

⁵ Department of Materials Science and Engineering, University of Wisconsin, Madison, WI 53706, USA

Received 25 September 2009, in final form 19 October 2009

Published 22 February 2010

Online at stacks.iop.org/SUST/23/034003

Abstract

Less than two years after the discovery of high temperature superconductivity in oxypnictide LaFeAs(O, F) several families of superconductors based on Fe layers (1111, 122, 11, 111) are available. They share several characteristics with cuprate superconductors that compromise easy applications, such as the layered structure, the small coherence length and unconventional pairing. On the other hand, the Fe-based superconductors have metallic parent compounds and their electronic anisotropy is generally smaller and does not strongly depend on the level of doping, and the supposed order parameter symmetry is s-wave, thus in principle not so detrimental to current transmission across grain boundaries. From the application point of view, the main efforts are still devoted to investigate the superconducting properties, to distinguish intrinsic from extrinsic behaviors and to compare the different families in order to identify which one is the fittest for the quest for better and more practical superconductors. The 1111 family shows the highest T_c , huge but also the most anisotropic upper critical field and in-field, fan-shaped resistive transitions reminiscent of those of cuprates. On the other hand, the 122 family is much less anisotropic with sharper resistive transitions as in low temperature superconductors, but with about half the T_c of the 1111 compounds. An overview of the main superconducting properties relevant to applications will be presented. Upper critical field, electronic anisotropy parameter, and intragranular and intergranular critical current density will be discussed and compared, where possible, across the Fe-based superconductor families.

(Some figures in this article are in colour only in the electronic version)

1. Introduction

In 2008 the Hosono group in the Tokyo Institute of Technology discovered superconductivity at 26 K in the oxypnictide LaFeAs(O, F) [1]. After only one month the critical temperature, T_c , doubled thanks to substitutions of the La by different rare earth (RE) elements (Sm, Ce, Nd, Pr and

Gd), yielding an increase up to 55 K with Sm [2]. The parent compounds exhibit antiferromagnetic ordering of the iron moments which is suppressed by doping in favor of superconductivity. The early awareness that magnetic order, even if in competition with superconductivity, is a key factor for determining superconductivity drove the discovery, within a short period, of new iron-based superconductor families

with different crystal structures such as (Ba, K)Fe₂As₂ [3], LiFeAs [4] and FeSe [5]. A large number of different compounds have now shown that superconductivity can be induced by carrier doping, both in the Fe–As layer and in the spacing layer, and by external as well as by internal pressure. For simplicity in the following we will refer to the different families as: 1111 (REFeAs(O, F)), 122 ((Ba, K)Fe₂As₂), 11(Fe(Se, Te)) and 111(LiFeAs).

These four families share several characteristics with the cuprate superconductors, such as layered structure, the presence of competing orders, low carrier density, small coherence length and unconventional pairing, all of which potentially hinder practical applications, especially due to their influence in exciting large thermal fluctuations and depressed grain boundary superconductivity. On the more positive side, however, the Fe-based superconductors have metallic parent compounds, their anisotropy is generally smaller and does not strongly depend on the level of doping, and their generally supposed order parameter symmetry is s-wave, which is in principle not so detrimental to current transport across grain boundaries.

As in the early days of cuprate superconductors, the main efforts are still devoted to distinguishing intrinsic from extrinsic behavior. The absence of significant transport currents in polycrystalline samples [6–8] has raised the question whether the low connectivity is an extrinsic effect due to low density, spurious phases, cracks or an intrinsic depression of the superconducting order parameter similar to that observed in cuprates for grain boundaries with larger than very small angles [9, 10].

The availability of different pnictide families allows us to compare them and so to identify trends that might provide a clue for understanding the nature of superconductivity in these compounds, as well as perhaps allowing us to focus on those matching the quest for better and more practical superconductors. The 1111 family, indeed, shows larger T_c , a huge but also anisotropic upper critical field and in-field, fan-shaped resistive transition reminiscent of those of cuprates [11, 12], while the 122 family is less anisotropic and exhibits narrow resistive transitions like those in low temperature superconductors [13, 14].

In the following an overview of the principal superconducting properties relevant to applications is presented. In section 2, the upper critical field, H_{c2} , the electronic anisotropy, the coherence lengths, the paramagnetic limit and the effect of thermal fluctuations are discussed and compared across the Fe-based superconductor families. In section 3, the critical current densities of single crystals, polycrystals and bicrystals are reviewed.

2. Upper critical fields

The huge upper critical field values of Fe-based superconductors require investigation in high magnetic field laboratories. Already the first magnetoresistance measurements of polycrystalline La-1111 up to 45 T [15] indicated a $\mu_0 H_{c2}$ value larger than 60 T which corresponds to a small coherence length of the order of a few nm. Moreover, $H_{c2}(T)$ was anomalous

Table 1. Significant superconducting state properties of pnictide single crystals.

	Nd-1111	Ba-122	Fe-11
$T_c(50\%R_n)$ (K)	47.4	22.0	14.5
$\mu_0(dH_{c2}^{\perp ab}/dT)$ (T K ⁻¹)	2.1	2.5	14
$\mu_0(dH_{c2}^{\parallel ab}/dT)$ (T K ⁻¹)	10.1	4.9	26
γ_H	5	1.9–1.5	1.9–1.1
ξ_{ab} (nm)	1.8	2.4	1.2
ξ_c (nm)	0.45	1.2	0.35
Ginzburg number Gi	8×10^{-3}	1.7×10^{-4}	1.3×10^{-3}

and exceeded the Werthamer–Helfand–Hohenberg (WHH) formula [16], similar to that observed in dirty MgB₂ [17, 18], suggesting that superconductivity in oxypnictides results from at least two bands. By replacing La with smaller rare earths like Nd and Sm, T_c and $H_{c2}(0)$ both increase [11]. Going from the lower (La-1111) to the higher T_c compounds (Nd-1111, Sm-1111), the in-field superconducting transitions become broader, approaching the broad magnetoresistive transitions of the cuprates for the highest T_c compounds. The H_{c2} slope at T_c increases with increasing T_c , reaching a slope of 9.3 T K⁻¹ in Sm-1111; even using WHH extrapolations which clearly underestimate many measurements, such dH_{c2}/dT values yield $\mu_0 H_{c2}(0) \approx 0.693 T_c \mu_0 |dH_{c2}/dT|_{T_c} \approx 400$ T, much larger than the paramagnetic limit.

The availability of single crystals, first of the 1111 compounds, allows the evaluation of H_{c2} parallel, $H_{c2}^{\parallel ab}$, and perpendicular, $H_{c2}^{\perp ab}$, to the ab plane [12]. The temperature dependence is very different in the two directions, strongly departing from the WHH behavior [16] mainly in the direction parallel to c . The anisotropy evaluated as $\gamma = \gamma_H = H_{c2}^{\parallel ab}/H_{c2}^{\perp ab}$ is also strongly temperature-dependent, reminiscent of the two-gap behavior seen in MgB₂ [17, 18]. However, a different situation is observed in the 122 family. (Ba, K)Fe₂As₂ single crystals exhibit nearly isotropic $\mu_0 H_{c2}$ with values of the order of 60 T at zero temperature and anisotropy going from 2, close to T_c , down to 1 at 5 K [13]. Similar results were reported in Ba(Fe,Co)₂As₂ [14].

These aspects have been investigated in high magnetic fields on three single crystals belonging to the different families of Fe-based superconductors. The main properties of these samples (NdFeAsO_{0.7}F_{0.3} (Nd-1111 in the following), Ba(Fe_{0.9}Co_{0.1})₂As₂ (Ba-122) and FeSe_{0.5}Te_{0.5} (Fe-11) [19–21] with critical temperatures of 47.4, 22.0 and 14.5 K, respectively, defined at 50% of the normal state resistivity) are summarized in table 1.

Magneto-transport measurements were performed in a 16 T Quantum Design PPMS and in high magnetic field in the 35 T resistive and 45 T hybrid magnets at the National High Magnetic Field Laboratory (NHMFL). Figure 1 shows the temperature dependence of magnetoresistance of the three single-crystal samples of Nd-1111, Ba-122 and Fe-11 in a magnetic field applied parallel to the c axis. For Nd-1111, the transitions broaden with increasing magnetic field, while for Ba-122 the breadth appears independent of field as in low temperature superconductors like Nb₃Sn [22]. For Fe-11 the situation is intermediate, even though it has a lower T_c , 14.5 K,

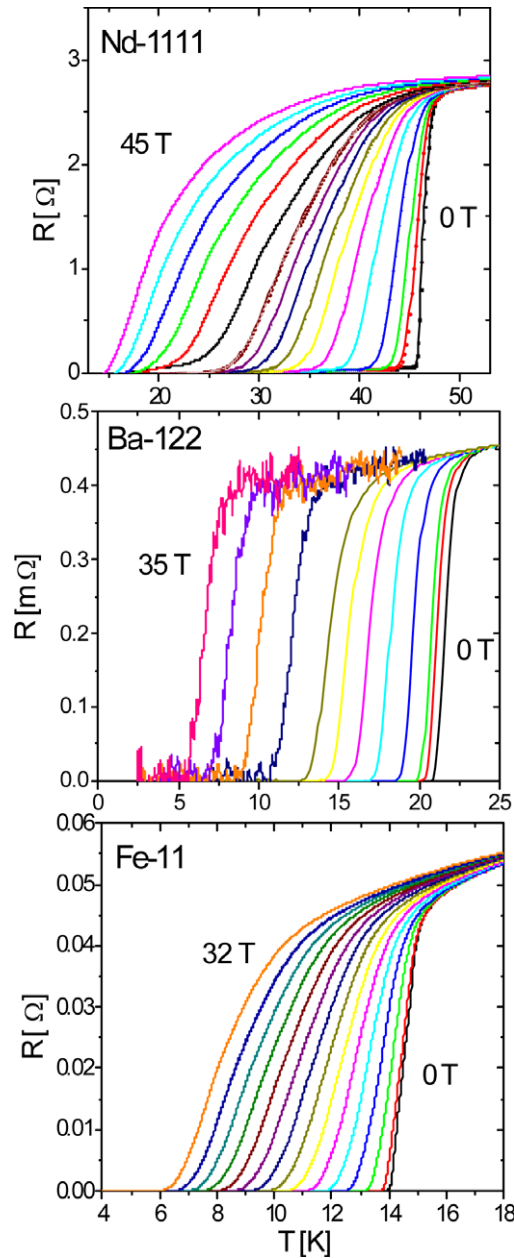


Figure 1. Magneto-transport measurements in high magnetic field applied perpendicularly to the samples for three different materials: NdFeAsO_{0.7}F_{0.3} (Nd-1111), Ba(Fe_{0.9}Co_{0.1})₂As₂ (Ba-122) and FeSe_{0.5}Te_{0.5} (Fe-11).

compared to 22 K for the 122 and 47 K for the 1111 single crystals.

Figure 2 shows $H_{c2}(T)$ in parallel and perpendicular field configurations determined with the 90% criterion. The three materials differ not only in T_c and absolute values of H_{c2} but importantly too in their temperature dependence of H_{c2} . Nd-1111 has a nearly linear behavior in both directions which partly differs from those observed in [12], where upturn and downward curvatures were observed in the perpendicular and parallel directions, respectively. Ba-122 shows an almost linear behavior in the perpendicular direction but exhibits a downward curvature in the parallel direction, in agreement with [13]. Finally Fe-11 exhibits clear downward curvatures

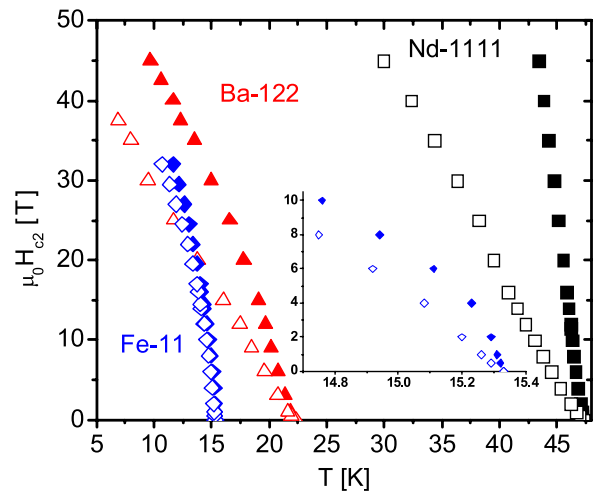


Figure 2. Temperature dependences of $H_{c2}^{\parallel ab}$ (filled symbols) and $H_{c2}^{\perp ab}$ (empty symbols) for the same samples of Figure 1: Nd-111 (squares), Ba-122 (triangles) and Fe-11 (diamonds). In the inset is shown a magnification of the region close to T_c for the Fe-11 sample.

in both directions; this behavior, rather unusual, was checked in several crystals and seems not related to inhomogeneities. The slope of H_{c2} close to T_c significantly varies in the different families. $\mu_0 dH_{c2}^{\perp ab}/dT$ near T_c varies from 2 T K⁻¹ in Nd-1111 to almost 14 T K⁻¹ in Fe-11 and $\mu_0 dH_{c2}^{\parallel ab}/dT$ from 5 T K⁻¹ in Ba-122 to the very high value of 25 T K⁻¹ in Fe-11. The H_{c2} anisotropy γ_H is particularly affected by the different temperature dependences in the two directions. While the anisotropy is almost constant and equal to 5 in Nd-1111, in the other two compounds it decreases with decreasing temperature. In Fe-11, for instance, the anisotropy close to T_c is about 2 but, due to the strong downward curvature of the parallel direction, γ_H drops rapidly with decreasing temperature and approaches 1 at the lowest measured temperature. This is better seen in the inset of figure 2 where a magnification of the region close to T_c for the Fe-11 sample is plotted.

2.1. Paramagnetic limit

The description of such upper critical field behavior is beyond the single-band, weak-coupling WHH model, where H_{c2} is limited by orbital pair breaking, γ_H is temperature-independent and dH_{c2}/dT is proportional to $T_c(1 + \lambda)/v_F l$, where λ is the electron–boson coupling constant, v_F is the Fermi velocity and l is the electron mean free path. On the other hand, the paramagnetic limit, where the superconductivity suppression is due to the alignment of the spins, is another pair-breaking mechanism to take into account [23]. In fact, the upper critical field of those materials strongly exceeds the BCS paramagnetic limit, H_p^{BCS} , which is given by $\mu_0 H_p^{\text{BCS}}(T) = 1.84 T_c$ (K) [24] as emphasized in figure 3, where the $\mu_0 H_{c2}/T_c$ as a function of T/T_c is reported. Those factors suggest a more complex scenario. First of all, to explain such a high H_{c2} , strong coupling has to be considered. The single-band Eliashberg theory allows the paramagnetic limit to be enhanced up to $\mu_0 H_p = 1.84 T_c(1 + \lambda)$, strongly exceeding the BCS limit [25]. The effect of the paramagnetic limit may explain the different

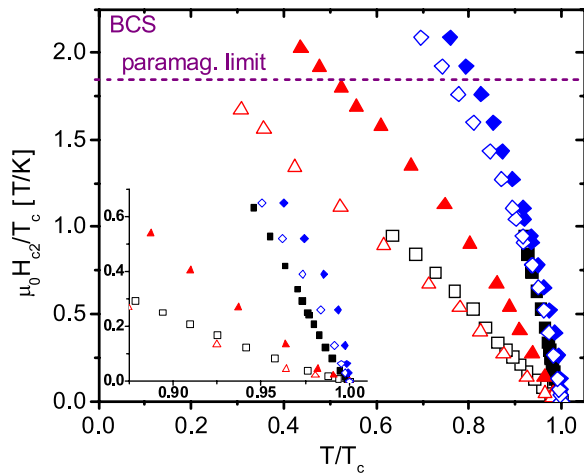


Figure 3. $\mu_0 H_{c2}/T_c$ as a function of T/T_c Nd-1111, Ba-1222 and Fe-11 for $H \parallel ab$ (filled symbols) and $H \perp ab$ (empty symbols): Nd-111 (squares), Ba-122 (triangles) and Fe-11 (diamonds). The broken line represent the BCS paramagnetic limit $\mu_0 H_p = 1.84 T_c (1 + \lambda)$. In the inset is shown a magnification of the region close to $T/T_c = 1$.

behavior of the samples considered here. As reported in figure 3, while for Ba-122 and Fe-11 $\mu_0 H_{c2}/T_c$ values exceed the BCS paramagnetic limit, for Nd-1111, given the higher T_c , $\mu_0 H_{c2}/T_c$ are well below the paramagnetic limit. Thus, for Nd-1111 $H_{c2}(T)$ in the experimentally accessible field range is not appreciably affected by paramagnetic suppression and shows a linear trend. In the case of Ba-122, suppression is mainly evident in the parallel direction where downward curvature is observed, while for Fe-11 both H_{c2} orientations show downward curvature, suggesting that both are affected by Pauli pair breaking. Because paramagnetic limitation is isotropic, a stronger effect is expected in the parallel direction of higher H_{c2} , which should induce an anisotropy which reduces with decreasing temperature, as observed in Ba-122 and Fe-11.

We conclude that paramagnetic effects play an important role in determining the temperature dependence of H_{c2} and its anisotropy in Fe-based superconductors. However, we cannot exclude that a temperature-dependent anisotropy may be partly induced by multiband effects, as suggested in [26] for $\text{Sr}(\text{CoFe})_2\text{As}_2$ epitaxial film.

2.2. Fluctuation effects

From the H_{c2} slope close to T_c we may evaluate the in-plane, ξ_a , and out-of-plane, ξ_c , coherence lengths from the Ginzburg–Landau expressions $\xi_a = [\phi_0 / (2\pi\mu_0(dH_{c2}^{\parallel c}/dT_c)T_c)]^{1/2}$ and $\xi_c = \xi_a/\gamma_H$, as reported in table 1. Interestingly, Fe-11 with the smallest T_c presents the smallest coherence length values. More generally, the values are small for all samples and ξ_c is comparable to the distance between the superconducting Fe layers, as for the CuO_2 layers in the cuprate superconductors. It was suggested that thermal fluctuations may cause the broadening of the in-field resistive transition observed in the 1111 family [11] and a two-dimensional fluctuation regime was indeed observed in magneto-conductance measurements of

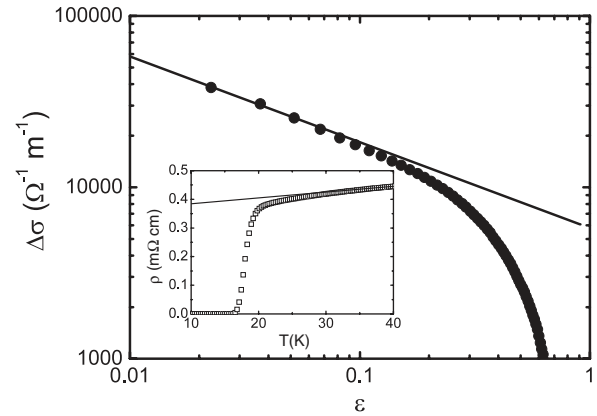


Figure 4. Fluctuation conductivity $\Delta\sigma$ as a function of $\varepsilon = \ln(T/T_c)$ in a log–log scale: filled symbols represent the experimental data and continuous line represents the 3D Aslamazov–Larkin behavior $\Delta\sigma = e^2/(32\hbar\xi_c\sqrt{\varepsilon})$. Inset: low temperature resistivity data (open square symbols) and linear extrapolation of normal state resistivity data (continuous line).

Sm-1111 compounds [27]. To understand whether fluctuation effects play a role also in the 122 and 11 families we evaluate the Ginzburg number Gi , which quantifies the temperature region GiT_c , where fluctuations are significant. It is expressed by [11] $Gi = (\pi\lambda_0^2 k_B T_c \mu_0 / 2\xi_c \Phi_0^2)^2$, where λ_0 is the London penetration depth, k_B is the Boltzmann constant and Φ_0 is the flux quantum. Ginzburg numbers, evaluated for the three compounds assuming for simplicity $\lambda_0 = 200$ nm for all the compounds [14, 28, 29], are reported in table 1. The Gi value obtained for Nd-1111 (8×10^{-3}) is the largest and is comparable with the value obtained for YBCO (10^{-2}). The smallest number is obtained for Ba-122 (1.7×10^{-4}), consistent with the narrow transitions in figure 1 and the low temperature superconductor-like behavior emphasized in [14]. The number we obtain for Fe-11 (1.3×10^{-3}), even if one order of magnitude lower than that obtained for high T_c superconductors, is four orders of magnitude larger than the value estimated for a low T_c superconductor with the same T_c such as V_3Si . This makes the 11 family unique in being a low T_c superconductor with an extremely short coherence length.

In order to detail the effect of fluctuations, resistivity measurements have been performed on an epitaxial film. The film, grown by pulsed-laser deposition by a target of nominal composition $\text{FeSe}_{0.5}\text{Te}_{0.5}$, has $T_c = 18$ K, larger than that of the target due to the strain developed during the growth [30]. The fluctuation conductivity $\Delta\sigma$ is evaluated as $\Delta\sigma = (\rho_n - \rho)/\rho\rho_n$, where ρ_n is the normal state resistivity. In the inset of figure 4, ρ and ρ_n as linearly extrapolated in the range above $2T_c \approx 40$ K are shown. In the main panel of the same figure, the fluctuation conductivity is plotted versus $\varepsilon = \ln(T/T_c)$. We identify the so-called Gaussian regime in the range $0.01 < \varepsilon < 0.1$, that is for temperatures from 0.4 to 2 K above T_c , between the critical regime very close to T_c and the high temperature regime of vanishing fluctuation conductivity. In this Gaussian regime, the behavior of $\Delta\sigma$ is well described by the 3D law $\Delta\sigma = e^2/(32\hbar\xi_c\sqrt{\varepsilon})$, where $\hbar = h/2\pi$ with $h = \text{Planck's constant}$ (continuous line in

figure 4). This 3D conclusion is consistent with the fact that $2\xi_c$ is of the order of the interplanar distance $s = 6.05 \text{ \AA}$. Indeed, the value of ξ_c obtained by fitting fluctuation conductivity data is of the order of 1 nm, in agreement within a factor of 2 with the value of 0.6 nm extracted from the critical field data of this same film.

3. Critical current behavior

Early studies of the critical current density (J_c) of 1111 polycrystalline samples emphasized the strong granularity of these compounds, which restricted global J_c values to very low values [6–8, 31]. A first optimistic claim came from Yamamoto *et al* [8] who found evidence for two distinct scales of current flow in polycrystalline Sm and Nd iron oxypnictides using magneto-optical imaging (MO) and study of the field dependence of the remanent magnetization. Such granular behavior has so far limited the properties of pnictide wires [32], even if wetting grain boundary phases and other extrinsic material inhomogeneities are one of the clear causes of this granularity. Even with substantial blocking by such grain boundary phases, the intergranular current densities appear to be more than one order of magnitude larger at 4 K than for early results on randomly oriented polycrystalline cuprates [33]. But certainly these early results make it clear that pnictides have different properties compared to randomly oriented MgB₂ polycrystals [34], where grain boundaries can also partially obstruct without evidence for intrinsic obstruction of current flow as in the cuprates or as now appears to be the case in the pnictides [35].

Before discussing the GB properties in section 3.2, we focus on bulk J_c properties mainly obtained from single crystals and discuss the operating flux pinning mechanisms and the anisotropy of J_c .

3.1. J_c in single crystals

As is usually the case, J_c for single crystals must be evaluated by magnetization measurement and use of the Bean model, a procedure almost always possible for field H parallel to the c axis but much less frequently possible for H parallel to the ab plane, where problems of the small size of crystals, significant anisotropy and difficulty of aligning crystals accurately with the field axis make extraction of J_c from the measured magnetic moment uncertain. For the 1111 class, Zhigadlo *et al* reported a high in-plane J_c of $\sim 2 \times 10^6 \text{ A cm}^{-2}$ at 5 K on an SmFeAsO_{1-x}F_x crystal. J_c is almost field-independent up to 7 T at 5 K [36]. Many single-crystal results were reported in the 122 system, since larger crystals can be easily grown. Yang *et al* reported significant fishtail peak effects and large current carrying capability up to $5 \times 10^6 \text{ A cm}^{-2}$ at 4.2 K in a K-doped Ba_{0.6}K_{0.4}Fe₂As₂ single crystal [37]. Yamamoto *et al* deduced $J_c \sim 4 \times 10^5 \text{ A cm}^{-2}$ at 4.2 K and also reported the fishtail peaks in their Co-doped Ba(Fe_{0.9}Co_{0.1})₂As₂ single crystals [14]. Prozorov *et al* showed J_c of $2.6 \times 10^5 \text{ A cm}^{-2}$ at 5 K for Ba(Fe_{0.93}Co_{0.07})₂As₂ single crystals and also showed fishtail peaks as well as very large magnetic relaxation rate, which were analyzed using collective pinning and creep

models [38]. As for the 11 system, Taen *et al* reported that J_c of tellurium-doped FeTe_{0.61}Se_{0.39} crystals with $T_c \sim 14 \text{ K}$ exceeded $1 \times 10^5 \text{ A cm}^{-2}$ at low temperatures [39].

All these results show that Fe-based superconductors exhibit rather high J_c values, independent of the field at low temperatures, similar to the behavior observed in YBCO [40]. Such results are all consistent with the nm-scale coherence lengths in table 1, the exceptionally high H_{c2} values and pinning associated with atomic-scale defects resulting from chemical doping. The common fishtail observation may indicate the presence of nanoscale phase separation into regions of weaker superconductivity that are proximity-coupled to the higher T_c matrix, perhaps an intrinsic effect or one caused by an inhomogeneous distribution of the Co or K doping agent. Irradiation with Au ions and neutrons has emphasized that pinning can be further increased by introducing defects without affecting T_c . Au ions produce columnar defects that increase the critical current density, but less than one order of magnitude at low field [41]. Similar results were obtained with neutron irradiation which produces a more isotropic defect structure [42]. In this respect too the pnictides appear quite similar to the cuprates.

Because large single crystals of 122 can be grown, it is possible to study their anisotropy magnetically. In a recent study, single crystals $0.16 \times 0.93 \times 1.3 \text{ mm}^3$ of Ba(Fe_{0.9}Co_{0.1})₂As₂ with a sharp T_c transition at 23 K were grown at the NHMFL using the FeAs flux method. Magnetic fields of up to 14 T were applied both parallel to the c axis and ab plane of the crystal in an Oxford vibrating sample magnetometer (VSM). Figures 5(a) and (b) show magnetic hysteresis loops at 4.2–20 K in both orientations. All loops show negligible background ferromagnetic moment, indicative of little free Fe, often present in such crystals. Loops for both configurations show large hysteresis and a slight fishtail in $M(H)$, consistent with strong pinning.

Extraction of the anisotropic J_c depends on assumptions about the anisotropic Bean model and current scale that we assume to be the full sample size. For $H \parallel c$, it is reasonable to assume that the currents flow in the ab plane and the Lorentz force drives vortices perpendicular to the ab planes. We call this current $J_{c,ab}$ ($H \parallel c$). In the case of $H \parallel a(b)$ the currents flow along $b(a)$ and c and the Lorentz force driving vortices along the c axis and $b(a)$, respectively. We call these currents $J_{c,ab}$ ($H \parallel ab$) and $J_{c,c}$ ($H \parallel ab$). Since the aspect ratio of our crystal ($b/c \sim 6$) is intermediate, both anisotropic currents contribute to the magnetic moment.

The field dependence of $J_{c,ab}$ for $H \parallel c$ was calculated on the basis of the Bean model and is shown in figure 6. The value of $J_{c,ab}$ is $5.3 \times 10^5 \text{ A cm}^{-2}$ at 4.2 K, which is similar to that reported previously [14, 38, 41]. The field dependence of J_c is rather mild, especially at low temperatures, consistent with the high H_{c2} . In order to obtain the anisotropy of J_c , we deduced J_c along the c axis ($J_{c,c}$) from hysteresis loops in fields parallel to the ab plane using the extended Bean model [43]. It was assumed that $J_{c,ab}$ under self-field does not change much, regardless of $H \parallel ab$ and $H \parallel c$. The estimated $J_{c,c}$ at 4.2 K under self-field is $\sim 1.3 \times 10^5 \text{ A cm}^{-2}$. The temperature dependence of $J_{c,ab}$ and $J_{c,c}$ are fitted well with an expression

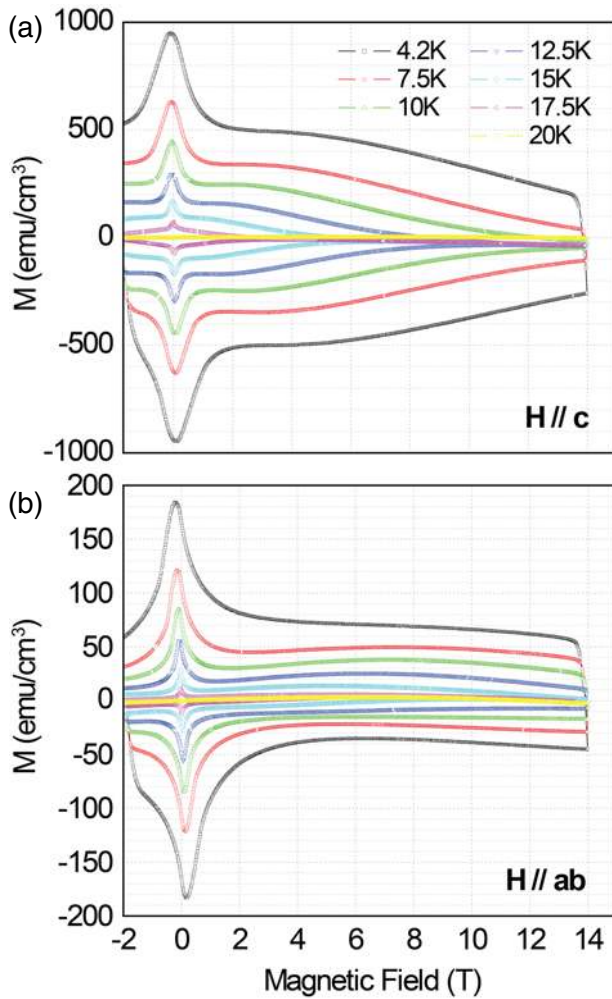


Figure 5. Magnetic hysteresis loops of the $\text{Ba}(\text{Fe}_{0.9}\text{Co}_{0.1})_2\text{As}_2$ crystal in field parallel to c axis (a) and ab plane (b) at 4.2, 7.5, 10, 12.5, 15, 17.5 and 20 K.

$J_c = J_c(0) \times (1 - T/T_c)^n$ with $J_{cab}(0) = 7.5 \times 10^5 \text{ A cm}^{-2}$, $n = 1.75$ for $J_{c,ab}$ and $J_{c,c}(0) = 2.0 \times 10^5 \text{ A cm}^{-2}$, $n = 2$ and shown in figure 7. In the inset of figure 7 the temperature-dependent anisotropy of J_c is plotted. The anisotropy $\gamma_J = J_{c,ab}/J_{c,c}$ is ~ 6 near T_c , decreases with decreasing T and reaches ~ 4 at low temperatures. The obtained value of γ_J is slightly larger compared to $\gamma_J \sim 2\text{--}3$ reported by Tanatar *et al* [44].

Analysis of the pinning force curve gives us insight into the underlying vortex pinning mechanisms. It is well known that the pinning force $F_p = J_c \times \mu_0 H$ of conventional metallic superconductors scales as $F_p \sim H_{c2}^m h^p (1-h)^q$, where $h = H/H_{\text{irr}}$ is the ratio between H and the irreversibility field H_{irr} . Here we show that the normalized pinning force $f_p = F_p/F_p^{\text{max}}$ as a function of reduced field h obtained from hysteresis loops in a field applied parallel to the c axis (figure 8 upper panel) and ab plane (figure 8 lower panel); H_{irr} has been estimated from a Kramer plot [45]. H_{irr} values evaluated in the two directions differ by a factor of 2, consistent with the H_{c2} anisotropy. As discussed above, $F_p(H \parallel c)$ and $F_p(H \parallel ab)$ are considered to be mainly determined by

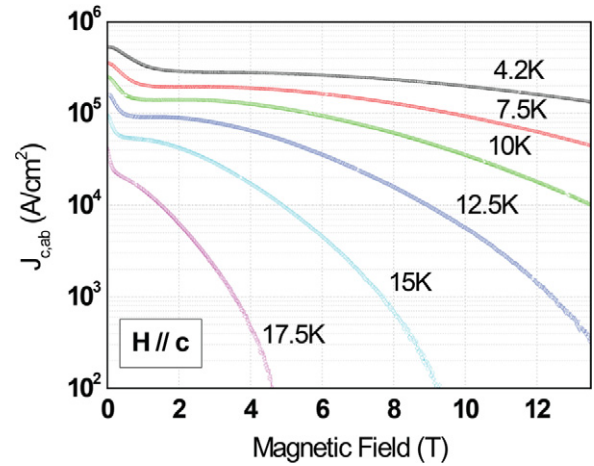


Figure 6. Critical current density for $H \parallel c$ ($J_{c,ab}$) as a function of magnetic field at 4.2–20 K for the $\text{Ba}(\text{Fe}_{0.9}\text{Co}_{0.1})_2\text{As}_2$ crystal.

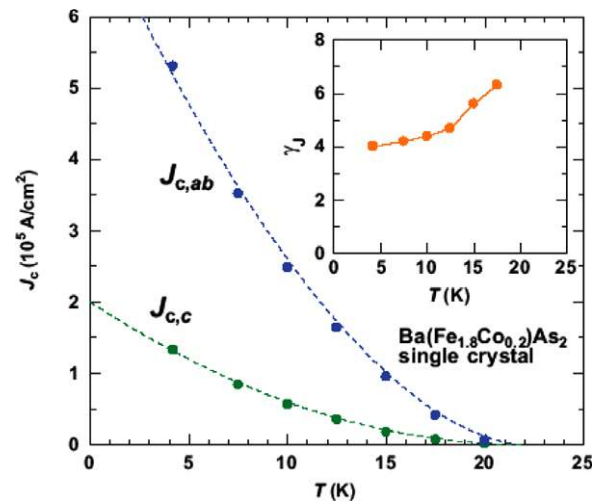


Figure 7. Temperature dependence of critical current density along the ab plane ($J_{c,ab}$) and c axis ($J_{c,c}$) for the $\text{Ba}(\text{Fe}_{0.9}\text{Co}_{0.1})_2\text{As}_2$ crystal. The dashed lines show fitting of the experimental data with $J_c = J_c(0) \times (1 - T/T_c)^n$ with $J_{cab}(0) = 7.5 \times 10^5 \text{ A cm}^{-2}$, $n = 1.75$ for $J_{c,ab}$ and $J_{c,c}(0) = 2.0 \times 10^5 \text{ A cm}^{-2}$, $n = 2$. Inset: temperature dependence of anisotropy of the critical current density $\gamma_J = J_{c,ab}/J_{c,c}$.

critical currents flowing in the ab plane and vortex motion along planes and across planes, respectively. Pinning force curves for both parallel and perpendicular field configurations scale well independently of temperature (from 4.2 to 17.5 K). This suggests that a single dominant vortex pinning mechanism works at all temperatures. The pinning force curves collapse according to the law $f_p \propto h^p (1-h)^q$ with $p = 1.1$ and $q = 3$ for $H \parallel c$ and $p = 1.25$ and $q = 2.25$ for $H \parallel ab$, respectively, as plotted by dashed lines in figure 8. The maxima of $f_p(h)$ curves occur at $h \sim 0.25$ for $H \parallel c$ and $h \sim 0.35$ for $H \parallel ab$. For a $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ single crystal the maximum is found at $h = 0.33$ [37]. This means that, once the H_{c2} anisotropy is taken into account by scaling the data with H_{irr} , a difference between $J_{c,ab}$ and $J_{c,c}$ still survives. The peak position may give clues to the pinning mechanism. As a first

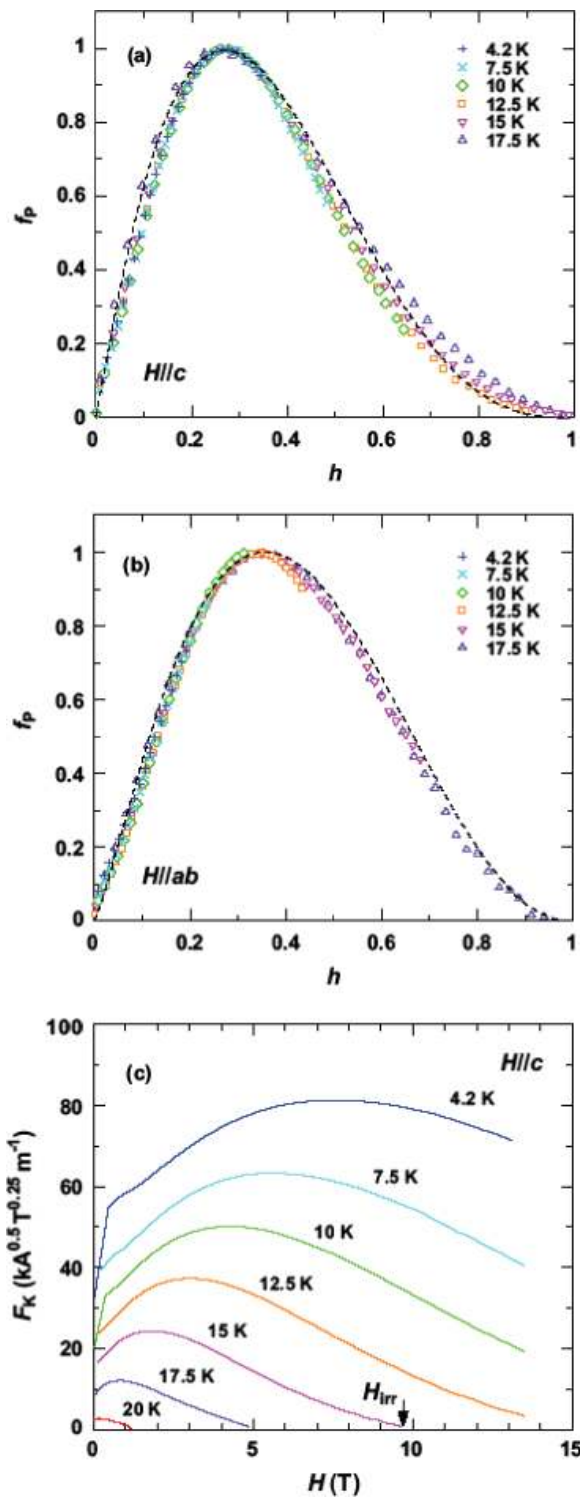


Figure 8. (a), (b) Normalized flux pinning force f_p as a function of reduced field $h = H/H_{irr}$ of the $\text{Ba}(\text{Fe}_{0.9}\text{Co}_{0.1})_2\text{As}_2$ crystal for field applied parallel to the c axis (a) and ab plane (b). (c) Kramer plots $F_k = J_c^{0.5} H^{0.25}$ for field applied parallel to the c axis. The arrow indicates irreversibility field H_{irr} estimated from extrapolation of F_k reaching zero.

approximation we can assume that the position of the peak shifts to higher h values with decreasing distance between pinning centers. We can assume that for $H \parallel c$ localized defects pin vortices, while for $H \parallel ab$ the modulation of the

order parameter along the c axis could play a role. This is compatible with strong bulk pinning suggested from scanning tunneling spectroscopy [46]. Recently, in single crystals of $\text{Ba}(\text{Fe}_{1-x}\text{Co}_x)_2\text{As}_2$ a combination of polarized-light imaging and magnetic measurements have show that the pinning is significantly enhanced by orthorhombic magnetic/structural domains [47]. This mechanism is supposed to be intrinsic to this phase and more significant in the slightly underdoped compositions.

3.2. Global J_c and grain boundary effects in polycrystalline materials

Practical use of the pnictides in large scale applications would be greatly enhanced if polycrystalline forms were not intrinsically electromagnetically granular, as is the case for the cuprates. That the pnictides are granular has been raised by multiple studies of polycrystals in bulk forms [6–8, 48], in wire forms [32] and also in thin film forms [49–51, 30]. But whereas it was relatively easy to get single-phase polycrystalline forms of the cuprates, it appears to be much harder in the case of the pnictides. We here briefly review recent studies of current transport in polycrystalline Sm- and Nd-1111 [8, 52, 53] that were made by high pressure synthesis at the Institute of Physics in Beijing (IOP-CAS). We have benchmarked these IOP-CAS samples of Ren *et al* against carefully made Sm-1111 samples made at the NHMFL and in INFM-LAMIA in Genoa, with and without the benefit of hot isostatic pressing. We find the intergranular connectivity of the IOP-CAS Sm-1111 sample to be the highest of all, even though there is clear evidence of significant wetting FeAs phase and unreacted RE_2O_3 , impurity phases found to be common to all. Thus we believe that these results have substantial general validity. Moreover, we find that polycrystalline samples of Co-doped 122 have 50–100 μm diameter grains, rather than the 5–10 μm diameter grains in the 1111 polycrystals. Magneto-optical images show essentially complete decoupling across the grain boundaries, but also substantial FeAs phase that wets the grain boundaries, as shown in figure 9. Only in the recent work of Lee *et al* [35] does the intrinsic behavior of second-phase free grain boundaries appear. Unfortunately it appears that symmetric [001] tilt grain boundaries grown epitaxially on SrTiO_3 exhibit substantial depression of J_c for misorientations of more than 3° .

The use of the low temperature laser scanning microscope (LTLMS) enables a direct spatial correlation between the position at which an electric field E occurs in the superconducting state and the microstructure with a precision of 1–2 μm . Figure 10 shows details of such correlations for two types of regions, type A and B, that show dissipative supercurrent flow only in self-or very weak fields up to ~ 0.1 T and region C, where flow remains dissipative even in 5 T after regions A and B have switched off. The SEM images of figure 10 show significant microstructural differences between regions A and B, and C. Precipitates of unreacted Sm_2O_3 are the most benign current-blocking defects because, although insulating, they have a small surface to volume ratio and mostly occur within Sm1111 grains. By contrast, the dark gray FeAs

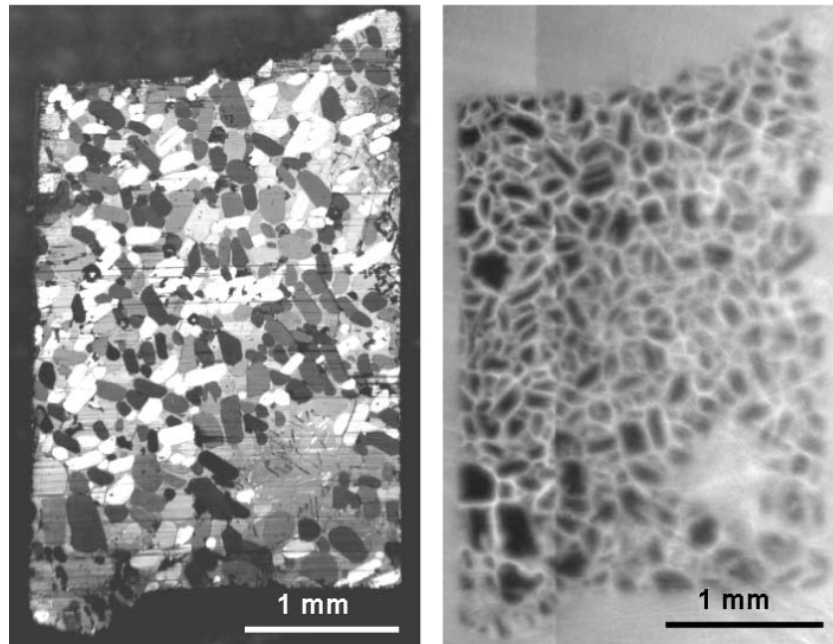


Figure 9. Optical image (left) of a polycrystal $\text{Ba}(\text{Fe}_{0.9}\text{Co}_{0.1})_2\text{As}_2$ bulk sample and the corresponding magneto-optical image (right) taken after zero-field cooling and applied field of 100 mT at 6.4 K.

phase wets many grain boundaries, thus interrupting grain-to-grain supercurrent paths, which are further degraded by extensive cracking, sometimes at grain boundaries (the black-appearing lines) and sometimes within grains.

At switch-off spot A of figure 10(a), a crack F on the upper side and the precipitate of Sm_2O_3 and a large FeAs phase G force current to cross grain boundary (H) containing a thin FeAs layer, producing the dissipation spot seen in the overlay image of figure 10(d). At switch-off, point B of figures 10(b) and (e), the current is channeled by cracks, Fe-As and Sm_2O_3 into a narrow passage crossing the FeAs regions too. In contrast, as shown in figures 10(c) and (f), spot C that remains dissipative even in 5 T field has its peak dissipation within a single grain at a constriction provided by two almost orthogonal sets of cracks which squeeze the current between the two diagonal cracks. The S-N-S (superconducting-normal-superconducting) nature of the connection across the metallic FeAs phase is strongly suggested by the very strong (tenfold) fall off of J_c even in 0.1 T field. Detailed analysis by MO imaging and remanent field analysis of subdivided samples had earlier shown that the intergranular current was both much smaller ($\sim 4000 \text{ A cm}^{-2}$ at 4 K) and had an SNS-like temperature dependence, while the intragranular current density significantly exceeded 10^6 A cm^{-2} (see figure 11). The reasonable conclusion to draw from these and many other studies of polycrystals is that granular behavior is quite evident but that one source of the granularity is uncontrolled second phase, particularly residual FeAs phase. Use of techniques such as remanent magnetization analysis, MO imaging and LTLMS imaging enable a quite detailed understanding of these effects [8]. The intragranular J_c values are largely consistent with the results obtained from single-crystal studies discussed above. All suggest that the pnictides are inherent nanomaterials

because of their short coherence lengths and thus produce high densities of pinning defects. We may conclude from the rapid fall off of J_c in figure 11(b), however, that many of these are point defects that are no longer effective at higher temperatures because vortices are easily thermally depinned.

3.3. J_c in thin films

Thin films have generally significantly lower J_c values than bulk single crystals and indeed have led to an independent conclusion that polycrystalline films exhibit electromagnetically granular behavior.

Thin films have not been easy to grow, especially of the 1111 compounds where doping is largely produced by F and O and where both are volatile and effectively uncontrolled in the final films. Study of La1111 by the Dresden group has concluded that the largely polycrystalline forms produced by *ex situ* growth produces an electromagnetically granular film, even though single-crystal LSAT substrates are used. In principle it should be possible to dope more easily in the Co-122 systems where the doping agent (Co) is not volatile and indeed this allows *in situ* growth and greater degrees of epitaxy [50, 54, 55]. However, growth of the 122 structure on LSAT does not seem to produce genuine epitaxy and J_c values are lower than those seen in bulk single crystals.

Epitaxial growth on STO immediately suggests that the classic bicrystal experiment [56] is possible and indeed this has now been reported by Lee *et al* [35]. The key result is that J_c is reduced by grain boundaries with 5° – 24° [001] tilt. Study of 3° , 6° , 9° and 24° bicrystals shows that there is a progressive reduction of J_c with increasing misorientation that is reminiscent of, but not as strong as in, cuprates, especially planar YBCO grain boundaries.

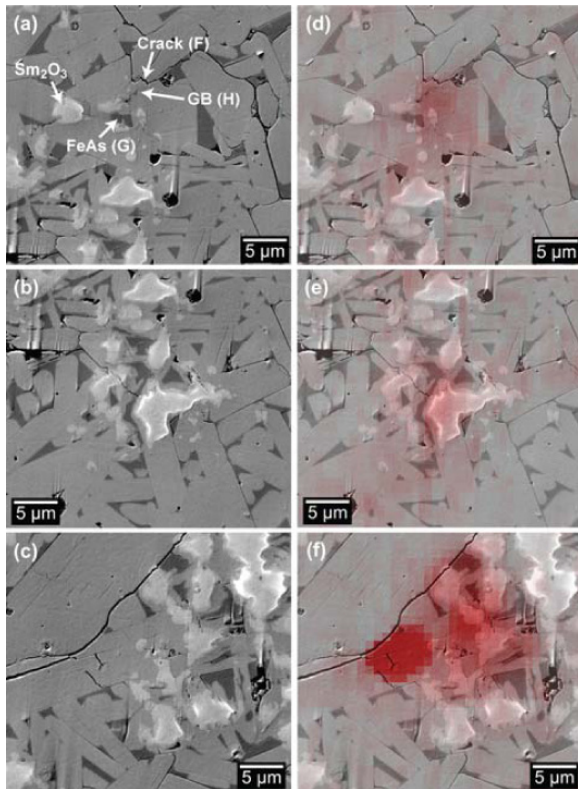


Figure 10. (a)–(c) High magnification SEM images of an Sm1111 polycrystal that was polished down to $20\ \mu\text{m}$ thickness and then examined in a low temperature laser scanning microscope in the superconducting state [53]. About 20 regions showing supercurrent dissipation were seen, of which about 3/4 switched off when more than $\sim 0.1\ \text{T}$ was applied. This image set shows microstructural details of typical switch-off spots A and B, and dissipation spot C that was still passing supercurrent at 5 T. There are second phases of Sm_2O_3 and FeAs with white and dark gray contrast in addition to platey Sm1111 grains. Cracks with dark line contrasts are also seen. (d)–(f) Dissipation spots at self-field superimposed on the SEM images of (a)–(c), respectively. Deeper red color represents the areas with stronger dissipation where higher supercurrent density was focused.

4. Summary

We have summarized recent studies of the pnictides from the viewpoint of potential applications. A key point is that they have properties intermediate between the LTS materials like Nb–Ti and Nb_3Sn and the cuprates like YBCO and Bi-2212 or Bi-2223. On their positive side is that they can have T_c up to 55 K and $H_{c2}(0)$ well over 100 T.

After a comparison among the families 122 comes out the most suitable for applications with rather high T_c , upper critical field, low anisotropy, reduced thermal fluctuations and intrinsic pinning mechanisms. In particular, the Co-doped 122 compound with T_c of $\sim 22\ \text{K}$ and $H_{c2}(0)$ of $> 50\ \text{T}$ has almost twice that of Nb_3Sn (30 T) with a T_c of 18 K. Although the Nb-base materials are isotropic, Co-122 is almost isotropic ($\gamma < 2$) too, making it potentially competitive as a low temperature superconductor. Even the highest T_c pnictides, Sm- and Nd-1111, have anisotropies much smaller than typical cuprates ($\gamma \sim 30$). However, a

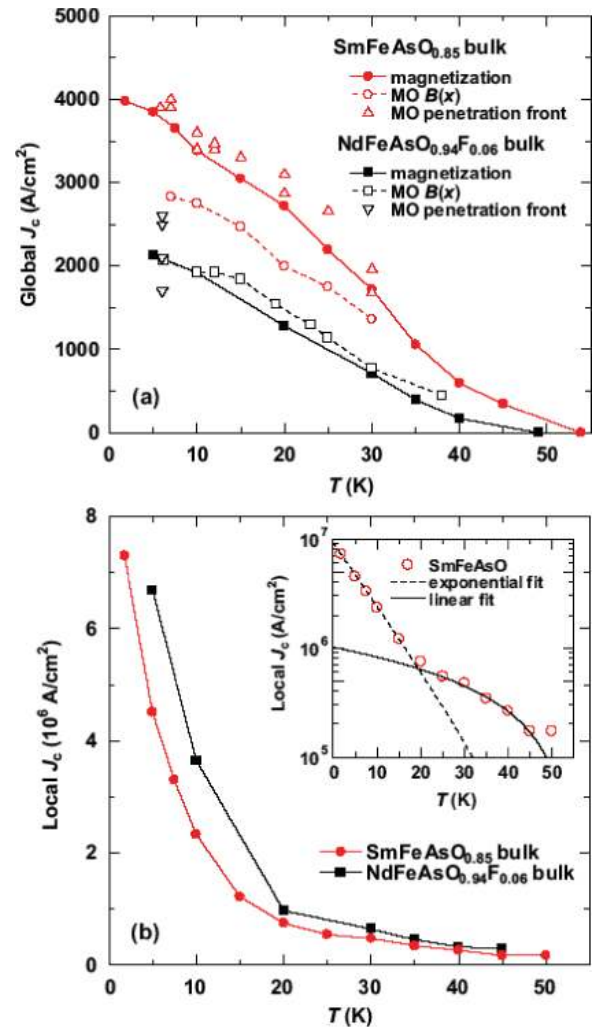


Figure 11. (a) Temperature dependence of global critical current density $J_c^{\text{global}}(T)$ for the polycrystalline $\text{SmFeAsO}_{0.85}$ and $\text{NdFeAsO}_{0.94}\text{F}_{0.06}$ bulk samples obtained from the remanent magnetization analysis (filled) and magneto-optical $B(x)$ flux profile analysis [8]. (b) Temperature dependence of critical current density of locally circulating current $J_c^{\text{local}}(T)$ for the polycrystalline $\text{SmFeAsO}_{0.85}$ and $\text{NdFeAsO}_{0.94}\text{F}_{0.06}$ bulk samples obtained from remanent magnetization analysis. Inset shows log-scale plots for the $\text{SmFeAsO}_{0.85}$ experimental data with an exponential and linear fitting.

typical YBCO sample has $\gamma \sim 5$, similar to the 1111. A clear drawback to present applications of the pnictides is their extrinsic and perhaps intrinsic granularity that significantly restrict the critical current density of polycrystalline forms. However, since only 18 months have passed since the first reports of T_c above 20 K in the pnictides, we should not expect that discoveries are yet over or that the final word on applications can yet be given.

Acknowledgments

A portion of this work was performed at the National High Magnetic Field Laboratory, which is supported by NSF Cooperative Agreement no. DMR-0654118, by the State of

Florida, and by the US Department of Energy. Explicit support for the pnictide work at the NHMFL comes from AFOSR under grant FA9550-06-1-0474. This work was also partially supported by the Italian Foreign Affairs Ministry (MAE)—General Direction for the Cultural Promotion. One of the authors (AY) is supported by a fellowship of the Japan Society for the Promotion of Science and (MP) by CNR under the project Short Term Mobility. The research at ORNL was sponsored by the Division of Materials Sciences and Engineering, Office of Basic Energy Sciences, US Department of Energy.

References

- [1] Kamihara Y, Watanabe T, Hirano M and Hosono H 2008 *J. Am. Chem. Soc.* **130** 3296
- [2] Ren Z A *et al* 2008 *Chin. Phys. Lett.* **25** 2215
- [3] Rotter M, Tegel M, Johrendt D, Schellenberg I, Hermes W and Pöttgen R 2008 *Phys. Rev. B* **78** 020503
- [4] Wang X C, Liu Q, Lv Y, Gao W, Yang L X, Yu R C, Li F Y and Jin C 2008 *Solid State Commun.* **148** 538
- [5] Hsu F C *et al* 2008 *Proc. Natl Acad. Sci. USA* **105** 14262
- [6] Yamamoto A *et al* 2008 *Appl. Phys. Lett.* **92** 252501
- [7] Senatore C, Wu G, Liu R H, Chen X H and Flukiger R 2008 *Phys. Rev. B* **78** 054514
- [8] Yamamoto A *et al* 2008 *Supercond. Sci. Technol.* **21** 095008
- [9] Larbalestier D C, Gurevich A, Feldmann D M and Polyanskii A 2001 *Nature* **414** 368
- [10] Hilgenkamp H and Mannhart J 2002 *Rev. Mod. Phys.* **74** 485
- [11] Jaroszynski J *et al* 2008 *Phys. Rev. B* **78** 064511
- [12] Jaroszynski J *et al* 2008 *Phys. Rev. B* **78** 174523
- [13] Yuan H Q, Singleton J, Balakirev F F, Baily S A, Chen G F, Luo J L and Wang N L 2008 *Nature* **457** 565–8
- [14] Yamamoto A *et al* 2009 *Appl. Phys. Lett.* **94** 062511
- [15] Hunte F, Jaroszynski J, Gurevich A, Larbalestier D C, Jin R, Sefat A S, McGuire M A, Sales B C, Christen D K and Mandrus D 2008 *Nature* **453** 903–5
- [16] Werthamer N R, Helfand E and Hohenberg P C 1966 *Phys. Rev.* **147** 295
- [17] Gurevich A 2003 *Phys. Rev. B* **67** 184515
- [18] Braccini V *et al* 2005 *Phys. Rev. B* **71** 012504
- [19] Jia Y, Cheng P, Fang L, Luo H, Yang H, Ren C, Shan L, Gu C and Wen H H 2008 *Appl. Phys. Lett.* **93** 032503
- [20] Sefat A S, Jin R, McGuire M A, Sales B C, Singh D J and Mandrus D 2008 *Phys. Rev. Lett.* **101** 117004
- [21] Sales B C, Sefat A S, McGuire M A, Jin R Y, Mandrus D and Mozharivskij Y 2009 *Phys. Rev. B* **79** 094521
- [22] Godeke A, Jewell M C, Fischer C M, Squitieri A A, Lee P J and Larbalestier D C 2005 *J. Appl. Phys.* **97** 093909
- [23] Fuchs G *et al* 2008 *Phys. Rev. Lett.* **101** 237003
- [24] Sarma G 1963 *J. Phys. Chem. Solids* **24** 1029
- [25] Schossmann M and Carbotte J P 1989 *Phys. Rev. B* **39** 4210
- [26] Baily S A, Kohama Y, Hiramatsu H, Maiorov B, Balakirev F F, Hirano M and Hosono H 2009 *Phys. Rev. Lett.* **102** 117004
- [27] Pallecchi I, Fanciulli C, Tropeano M, Palenzona A, Ferretti M, Malagoli A, Martinelli A, Sheikin I, Putti M and Ferdeghini C 2009 *Phys. Rev. B* **79** 104515
- [28] Drew A J *et al* 2008 *Phys. Rev. Lett.* **101** 097010
- [29] Gordon R T *et al* 2009 *Phys. Rev. Lett.* **102** 127004
- [30] Bellingeri E, Buzio R, Gerbi A, Marrè D, Congiu S, Cimberle M R, Tropeano M, Siri A S, Palenzona A and Ferdeghini C 2009 *Supercond. Sci. Technol.* **22** 105007
- [31] Prozorov R, Tillman M E, Mun E D and Canfield P C 2009 *New J. Phys.* **11** 035004
- [32] Gao Z, Wang L, Qi Y, Wang D, Zhang X, Ma Y, Yang H and Wen H 2008 *Supercond. Sci. Technol.* **21** 112001
- [33] Larbalestier D C *et al* 1987 *J. Appl. Phys.* **62** 3308
Seuntjens J M and Larbalestier D C 1990 *J. Appl. Phys.* **67** 2007
- [34] Larbalestier D C *et al* 2001 *Nature* **410** 186
- [35] Lee S *et al* 2009 arXiv:0907.3741
- [36] Zhigadlo N D, Katrych S, Bukowski Z, Weyeneth S, Puzniak R and Karpinski J 2008 *J. Phys.: Condens. Matter* **20** 342202
- [37] Yang H, Luo H, Wang Z and Wen H H 2008 *Appl. Phys. Lett.* **93** 142506
- [38] Prozorov R *et al* 2008 *Phys. Rev. B* **78** 224506
- [39] Taen T, Tsuchiya Y, Nakajima Y and Tamegai T 2009 *Phys. Rev. B* **80** 092502
- [40] Lan M D *et al* 1991 *Phys. Rev. B* **44** 233
Nguyen P P *et al* 1993 *Phys. Rev. B* **48** 1148
- [41] Nakajima Y, Tsuchiya Y, Taen T, Tamegai T, Okayasu S and Sasase M 2009 *Phys. Rev. B* **80** 012510
- [42] Eisterer M, Zehetmayer M, Weber H W, Jiang J, Weiss J D, Yamamoto A and Hellstrom E E 2009 *Supercond. Sci. Technol.* **22** 095011
- [43] Gyorgy E M *et al* 1989 *Appl. Phys. Lett.* **55** 283
- [44] Tanatar M A, Ni N, Martin C, Gordon R T, Kim H, Kogan V G, Samolyuk G D, Bud'ko S L, Canfield P C and Prozorov R 2009 *Phys. Rev. B* **79** 094507
- [45] Kramer E J 1973 *J. Appl. Phys.* **44** 1360
- [46] Yin Y, Zech M, Williams T L, Wang X F, Wu G, Chen X H and Hoffman J E 2009 *Phys. Rev. Lett.* **102** 097002
- [47] Prozorov R, Tanatar M A, Ni N, Kreyssig A, Nandi S, Bud'ko S L, Goldman A I and Canfield P C 2009 *Phys. Rev. B* **80** 174517
- [48] Polichetti M *et al* 2008 *Phys. Rev. B* **78** 224523
- [49] Backen E, Haindl S, Niemeier T, Huhne R, Freudenberg T, Werner J, Behr G, Schultz L and Holzapfel B 2008 *Supercond. Sci. Technol.* **21** 122001
- [50] Katase T, Hiramatsu H, Yanagi H, Kamiya T, Hirano M and Hosono H 2009 *Solid State Commun.* **149** 2121–4
- [51] Haindl S *et al* 2009 arXiv:0907.2271
- [52] Kametani F, Polyanskii A A, Yamamoto A, Jiang J, Hellstrom E E, Gurevich A and Larbalestier D C 2009 *Supercond. Sci. Technol.* **22** 015010
- [53] Kametani F *et al* 2009 *Appl. Phys. Lett.* **95** 142502
- [54] Hiramatsu H, Katase T, Kamiya T, Hirano M and Hosono H 2008 *Appl. Phys. Express* **1** 101702
- [55] Choi E M, Jung S G, Lee N H, Kwon Y S, Kang W N, Kim D H, Jung M H, Lee S I and Sun L 2009 *Appl. Phys. Lett.* **95** 062507
- [56] Dimos D, Chaudhari P, Mannhart J and Legoues F K 1988 *Phys. Rev. Lett.* **61** 219