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# Non equilibrium anisotropic excitons in atomically thin $\text{ReS}_2$

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We present a systematic investigation of the electronic properties of bulk and few layer ReS<sub>2</sub> van der Waals crystals using low temperature optical spectroscopy. Weak photoluminescence emission is observed from two non-degenerate band edge excitonic transitions separated by  $\sim 20$  meV. The comparable emission intensity of both excitonic transitions is incompatible with a fully thermalized (Boltzmann) distribution of excitons, indicating the hot nature of the emission. While DFT calculations predict bilayer ReS<sub>2</sub> to have a direct fundamental band gap, our optical data suggests that the fundamental gap is indirect in all cases.

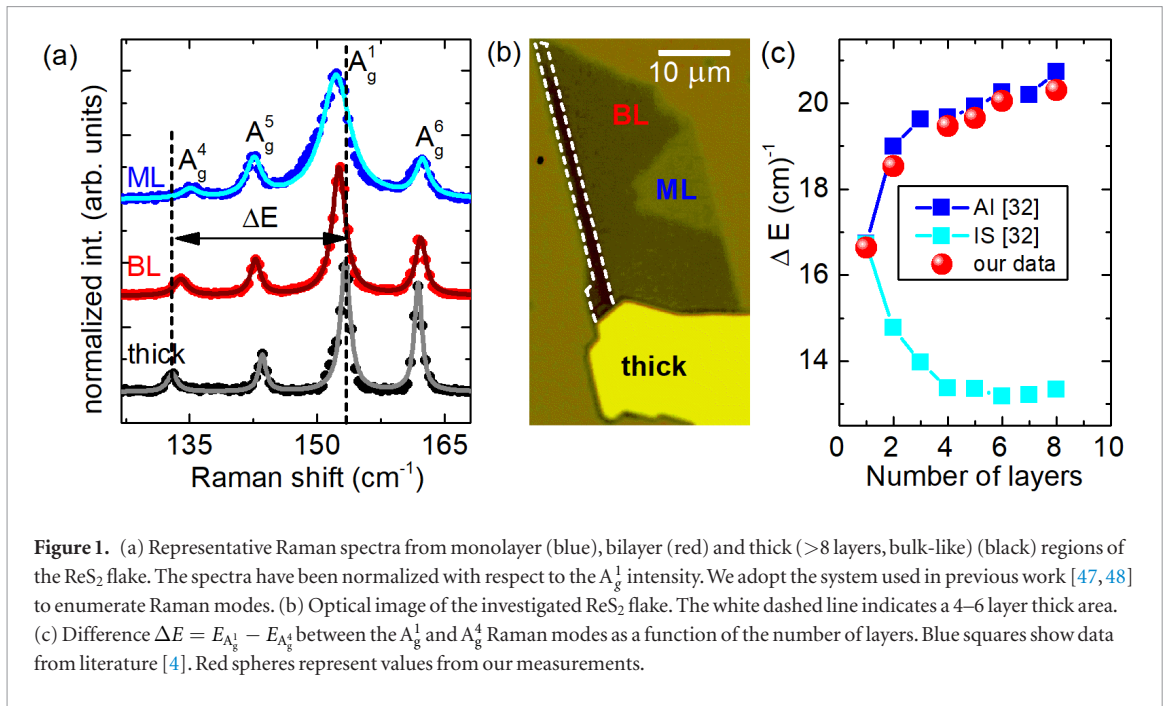
Emerging transition metal dichalcogenides (TMDs) such as MoS<sub>2</sub>, MoSe<sub>2</sub>, MoTe<sub>2</sub>, WS<sub>2</sub> and WSe<sub>2</sub> are attracting great attention due to their remarkable electronic properties. In particular, the energy and the character of the band gap can be easily tuned by varying the number of atomic layers in the crystal [1–5]. The two dimensional confinement and reduced dielectric screening in the single and few layer limit result in significantly enhanced exciton and trion binding energies [6–9], while the lack of inversion symmetry in the monolayer leads to valley-selective optical selection rules [10–13].

Recently, layered semiconductors with in-plane anisotropy such as black phosphorus [14, 15], and rhenium dichalcogenides (ReX<sub>2</sub>, where X stands for Se or S atoms) [16] have joined the family of intensively investigated van der Waals crystals. The sizable in-plane crystal asymmetry results in anisotropic optical [3, 5, 17–25], and electrical [17, 23, 26–28] properties, which can be employed in field effect transistors [29–32], polarization sensitive photodetectors [33], and new plasmonic devices [34]. The major advantage of Re dichalcogenides over black phosphorus is their

stability under ambient conditions [35], making them potentially interesting for applications.

Unlike the more extensively studied Mo and W based dichalcogenides, Re based TMDs crystallize in the distorted 1T' structure (schematically shown in figure 1(a)) of lower triclinic symmetry [4, 36–38]. In rhenium dichalcogenides, two non-degenerate direct excitons couple to light, as observed in the reflectivity contrast and photoluminescence (PL) spectra [3, 5, 39]. The strong linear polarization of the excitonic transitions provides a new degree of freedom to control the optical response [40, 41] of this material. Despite the flurry of recent investigations of the ReS<sub>2</sub> and ReSe<sub>2</sub> [16], knowledge about their fundamental electronic properties is extremely limited. For example, the nature of the fundamental band gap remains controversial.

Existing band structure calculations provide no consensus concerning the nature of the fundamental band gap [4, 5, 38, 42–44]. Studies regarding the absorption edge indicate that both materials have an indirect band gap in the bulk form [18–21, 45]. This assignment is supported by recent angle-resolved



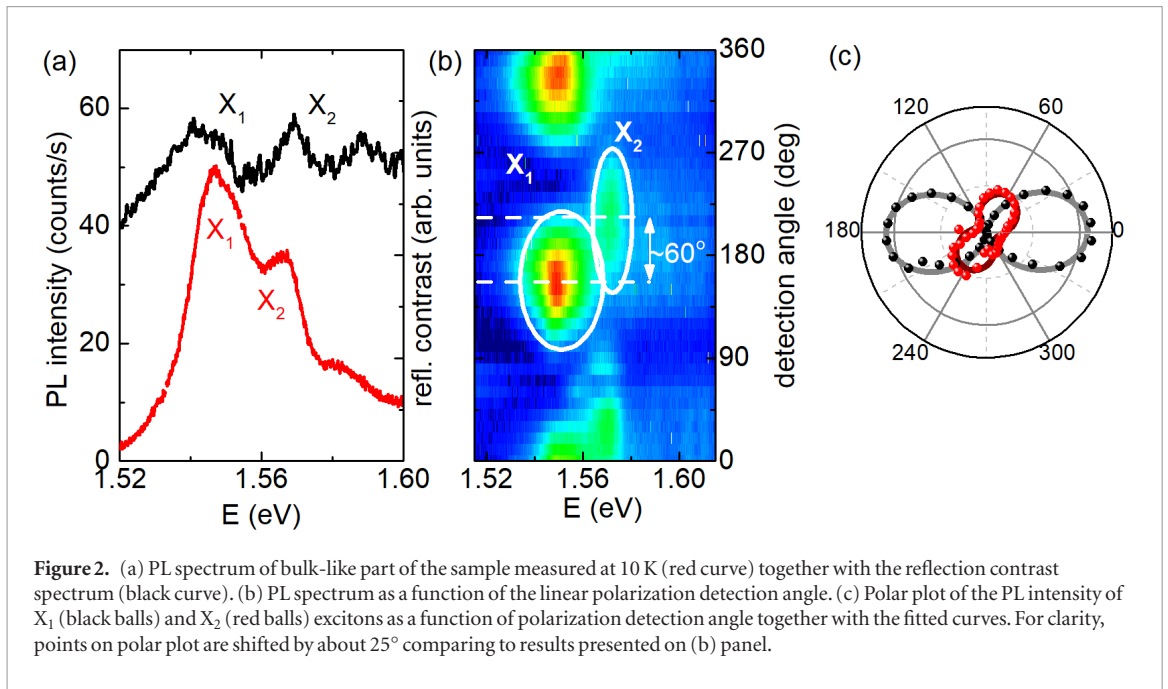
photoemission spectroscopy (ARPES) [44] and photoemission electron microscopy (PEEM) [43] for the few- and monolayer crystals. However, the observed PL [3, 5, 46] might suggest a direct nature of the band gap in this semiconductor. Intriguingly, a comparable emission intensity is observed from both excitonic states at cryogenic temperatures, despite the fact that they are separated by a few tens of meV [5]. This clearly indicates a strong departure from a thermalized (Boltzmann) exciton distribution, which is expected if the direct excitons are the lowest states.

We have performed a systematic low temperature optical investigation of bulk and few layer ReS<sub>2</sub> combining Raman, PL and reflectivity contrast measurements. Our experimental results support the indirect character of the band gap from bulk down to single layer ReS<sub>2</sub> crystals. The PL has features of hot emission. Their observation, even in the presence of an indirect band gap, is related to the relatively short radiative life time of direct excitons in ReS<sub>2</sub>. Although the indirect nature of the gap is generally supported by density functional theory (DFT) band structure calculations, intriguingly, our DFT calculations predict a direct gap for bilayer ReS<sub>2</sub>, while experimentally we detect only weak hot emission, characteristic for the indirect gap.

Mono- and few-layer ReS<sub>2</sub> was obtained from bulk crystals (Hq-graphene) by scotch tape exfoliation onto a surface of degenerately-doped Si covered with a 270 nm thick layer of SiO<sub>2</sub>. A flake containing monolayer, bilayer and a thick region (>8 layers) was selected for the investigation. The flake thickness was confirmed by optical contrast calibrated with atomic force microscopy on flakes of different thicknesses (see supporting information ([stacks.iop.org/TDM/6/015012/mmedia](http://stacks.iop.org/TDM/6/015012/mmedia))). A microscope image of the flake is shown in figure 1(b).

Raman measurements were performed in back scattering geometry under ambient conditions, at room temperature, and using a 532 nm laser. The laser light was focused on a spot of approximately 1 μm in diameter using a 100× objective (numerical aperture, NA, 0.9). The spectral resolution of the setup was 0.5 cm<sup>-1</sup>. The PL and reflectivity contrast measurements were performed in a standard μPL setup in back-scattering geometry, with a long working distance 50× objective having NA = 0.55. The sample was excited by CW 532 nm laser for PL or with white light from a tungsten-halogen lamp for reflectivity contrast measurements. The sample was mounted in a helium flow cryostat.

Few-layer ReS<sub>2</sub> can be found in two different polytypes depending on the stacking order referred as isotropic or anisotropic [4, 49]. They differ in the relative orientation of the Re–Re bonds in neighboring layers. The number of layers and the stacking order can be determined using Raman spectroscopy [4, 48]. Figure 1(a) shows typical Raman spectra measured for different thicknesses of the crystal. The colour coding of the spectra corresponds to that of the labels in the optical image in figure 1(b). The pronounced A<sub>g</sub><sup>1</sup> mode at around 155 cm<sup>-1</sup>, related to the in-plane motion of Re and S atoms, together with lower intensity A<sub>g</sub><sup>4</sup>, A<sub>g</sub><sup>5</sup> and A<sub>g</sub><sup>6</sup> are observed in all spectra. The four peaks were fitted with Lorentzian curves to precisely determine the characteristic vibration energy. With decreasing number of layers, the A<sub>g</sub><sup>1</sup> mode softens while the A<sub>g</sub><sup>4</sup> mode hardens. This is characteristic for anisotropic stacking order [4]. Isotropic stacking order shows the opposite trend, i.e. the distance between A<sub>g</sub><sup>1</sup> and A<sub>g</sub><sup>4</sup> mode decreases with increasing thickness. Therefore, the energy separation between A<sub>g</sub><sup>1</sup> and A<sub>g</sub><sup>4</sup> provides a



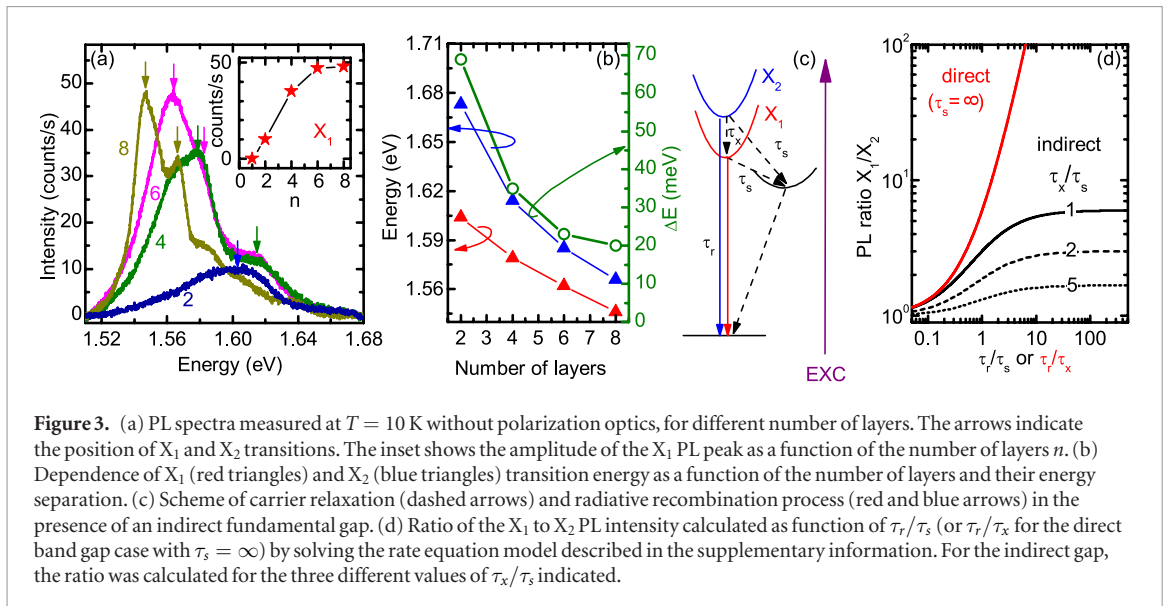
reliable signature of the number of layers. By comparing our data with the previous work (see figure 1(c)), we identify areas of monolayer and bilayer thicknesses, together with thicker regions at different positions on the flake.

A typical PL spectrum collected from a thick part of the crystal is presented in figure 2(a). The two emission peaks at 1.546 eV and 1.566 eV are non-degenerate direct excitonic transitions, previously observed in reflectivity contrast and PL [3, 39, 45]. The free excitonic nature of these PL peaks is further supported by the reflectivity contrast, also plotted in figure 2(a). In the reflectivity contrast, both excitonic features are Stokes-shifted by  $\approx 10$  meV with respect to the PL peaks. The full polarization dependence of the PL spectrum of the bulk-like sample is presented in figure 2(b). The emission intensity of both excitons strongly depends on the detection polarization angle, as demonstrated by the polar plot of figure 2(c). These data are plotted in a non-polar plot in the supplementary information. We also show in the supplementary information the PL intensity of low energy exciton as a function of detection angle. These measurements demonstrate that the orientation of regions of different thicknesses within a same flake is the same. The variation of the PL intensity with the detection angle shows a characteristic figure of eight dependence in the polar plot. This behavior is well described using  $I_0 + I_x \cos^2(\theta + \theta_0)$ , from which we extract the angle between the maximum PL intensity for each exciton. In agreement with previous reports [3, 40], the principal polarization axes of the excitons are rotated by  $\approx 60^\circ$  with respect to each other.

Now we discuss the evolution of the PL spectrum as a function of the number of layers, in light of the theoretical predictions about the nature of the bandgap. In the literature reports, one can find statements that

the band gap nature is direct for all thicknesses [38] as well as the prediction that the fundamental band gap is direct only for the bilayer [43]. The latter is in agreement with our DFT calculations presented in supplementary information and absorption measurements for the bulk crystals [18–21, 45]. However, none of these predictions is fully supported by the PL behavior, as we show in the next paragraphs.

The evolution of the PL spectrum with the flake thickness is presented in figure 3(a). As the number of layers decreases, there is a systematic increase of the emission energy, related to the opening of the band gap at the  $\Gamma$  point (see DFT calculations in supplementary information). The dependence of  $X_1$  and  $X_2$  transition energies versus number of layers is summarized in figure 3(b). In our sample, six and four layer thick  $\text{ReS}_2$  corresponds to a narrow stripe (see figure 1(b)). The PL spectrum measured in these regions receives some contribution from the bilayer emission, visible around 1.62 eV. The twin-peak PL structure can be consistently observed in all places on our sample with thickness  $\geq 4$  layers (see figure 3(a)). The PL from monolayer area of the flake was too weak to be observed, consistent with the predicted indirect nature of the gap. For bilayer, the PL spectrum is dominated by a weak, single PL peak attributed to the  $X_1$  exciton, with a weak, high energy shoulder corresponding to the  $X_2$  emission. Given that the  $X_2$  of the bilayer is weak in the unpolarized PL spectrum, we determine its transition energy based on the polarization resolved PL measurements shown in the supplementary information. The evolution of the amplitude of the  $X_1$  PL peak with the number of layers is shown in the inset of figure 3(a). The PL intensity from the bilayer clearly matches the trend, with no sign of enhanced emission, which is expected to be orders of magnitude stronger if the fundamental gap were direct (as observed in  $\text{MoS}_2$  [1]).



The observation of both  $X_1$  and  $X_2$  emissions is at first sight surprising, since they are separated by at least 20 meV, which is more than an order of magnitude larger than  $k_B T$  at 10 K ( $\sim 0.8$  meV). If the exciton population follows a Boltzmann distribution, the intensity ratio between two peaks should be vanishingly small ( $\sim 10^{-9}$ ), i.e. the higher energy  $X_2$  emission should not be observed. The observed PL peaks must arise from hot PL involving emission from excitons that are not fully thermalized. For this to occur, the radiative lifetime must be sufficiently short, compared to the non-radiative relaxation time. In addition, if there is an effective way to depopulate both  $X_1$  and  $X_2$ , the intensity of both will be reduced, while the ratio of the intensities will be pushed towards unity. The presence of the indirect band gap below the direct band gap transitions provides such an effective depopulation path. The scheme of the carrier kinetics with the band structure and relaxation paths are indicated schematically in figure 3(c). Here,  $\tau_r$  is the radiative lifetime of the direct excitons,  $\tau_s$  is the lifetime for scattering to the indirect dark exciton state, and  $\tau_x$  is the lifetime for scattering from  $X_2$  to  $X_1$ . The radiative recombination time of the direct exciton transition is not known for  $\text{ReS}_2$ . We expect, however, that it lies in the picoseconds time scale, as for other TMDs [50–52]. We estimate that the PL intensity here is  $\approx 2$  orders of magnitude weaker than in direct band gap TMDs. Assuming  $\tau_r \approx 1$  ps, this gives a reasonable ball park figure for the non radiative lifetime  $\tau_s \approx 10$  fs. We have solved the rate equation model to see what conditions should be fulfilled to obtain the observed ratio of  $X_1$  and  $X_2$  PL intensity for the direct and indirect case, which is plotted as a function of  $\tau_r/\tau_s$  in figure 3(d) (see supplementary information for more details). If the fundamental gap is direct, the ratio of the emission intensities diverges, as expected for  $\tau_r/\tau_s > 1$ . When the fundamental gap is indirect, for  $\tau_r/\tau_s > 1$  the ratio reproduces nicely the experimental observations, saturating between  $\approx 1$ –6 depending the

value of  $\tau_x/\tau_s$  used. For the direct gap, reproducing the  $X_1$  to  $X_2$  intensity ratio observed experimentally would require  $\tau_r/\tau_x \leq 1$ , which is not expected for semiconductors and incompatible with weak PL. We therefore conclude that the observation of hot PL emission is a smoking gun signature of the indirect gap. Since both excitonic transitions are observed for all thicknesses, we deduce that in all cases the gap is indirect.

It is interesting to note that *ab initio* calculations at different levels of theory successfully predict the evolution of the band structure for TMDs, such as  $\text{MoS}_2$  [1, 53–56]. However, DFT and GW calculations do not seem to capture with sufficient accuracy the changes in the electronic structure of bilayer  $\text{ReS}_2$ . Both methods predict that the bilayer is a direct gap semiconductor [43]. In  $\text{MoS}_2$ , the indirect band gap rapidly increases with the reduction of the layers number, because of significant out of plane component of the orbitals, which compose the indirect band edges [53]. In the monolayer limit, this results in an indirect band gap that is much larger (hundreds of meV at the K points) than the direct one, because the latter is contributed mainly by orbitals confined in the transition metal plane, which are rather insensitive to the surroundings. In contrast, the dependence of the band structure of  $\text{ReS}_2$  on the number of layers is not so pronounced, which is attributed to significantly weaker interlayer coupling [38]. Therefore, the difference between direct and indirect energy transition falls within the limit of calculation accuracy for the bilayer, which explains the inconsistency with experimental results. Additionally, the inclusion of Coulomb interaction might lead to an improved accuracy of the prediction of the nature of the fundamental band gap for all thicknesses.

To conclude, we have presented a combined investigation of low temperature PL and reflectivity contrast of  $\text{ReS}_2$  of different thicknesses with anisotropic stacking order. The weak PL observed from two direct excitons ( $X_1$  and  $X_2$ ) separated by  $\approx 20$  meV indicates the

hot nature of the PL, together with the presence of a lower lying indirect band gap. This conclusion is supported by the solution of a rate equation model. The weak intensity of the PL, observed for both excitons, is consistently observed down to bilayer thickness, which points to an indirect fundamental band gap of ReS<sub>2</sub>, independent of the number of layers. For bilayer ReS<sub>2</sub>, this contradicts the predictions of our DFT calculations, as well as the GW calculations and PEEM investigation found in the literature [43]. However, the difference between optical and fundamental gap is small, as the DFT and GLLB-SC calculations do not include the exciton binding energy.

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## References

- [1] Mak K F, Lee C, Hone J, Shan J and Heinz T F 2010 *Phys. Rev. Lett.* **105** 136805
- [2] Splendiani A, Sun L, Zhang Y, Li T, Kim J, Chim C Y, Galli G and Wang F 2010 *Nano Lett.* **10** 1271
- [3] Aslan O B, Chenet D A, van der Zande A M, Hone J C and Heinz T F 2016 *ACS Photon.* **3** 96
- [4] Qiao X F, Wu J B, Zhou L, Qiao J, Shi W, Chen T, Zhang X, Zhang J, Ji W and Tan P H 2016 *Nanoscale* **8** 8324
- [5] Arora A et al 2017 *Nano Lett.* **17** 3202
- [6] Mak K F, He K, Lee C, Lee G H, Hone J, Heinz T F and Shan J 2013 *Nat. Mater.* **12** 207
- [7] Ross J S et al 2013 *Nat. Commun.* **4** 1474
- [8] Chernikov A, Berkelbach T C, Hill H M, Rigosi A, Li Y, Aslan O B, Reichman D R, Hybertsen M S and Heinz T F 2014 *Phys. Rev. Lett.* **113** 076802
- [9] He K, Kumar N, Zhao L, Wang Z, Mak K F, Zhao H and Shan J 2014 *Phys. Rev. Lett.* **113** 026803
- [10] Xiao D, Liu G B, Feng W, Xu X and Yao W 2012 *Phys. Rev. Lett.* **108** 196802
- [11] Mak K F, He K, Shan J and Heinz T F 2012 *Nat. Nanotechnol.* **7** 494
- [12] Zeng H, Dai J, Yao W, Xiao D and Cui X 2012 *Nat. Nanotechnol.* **7** 490
- [13] Xu X, Yao W, Xiao D and Heinz T F 2014 *Nat. Phys.* **10** 343
- [14] Xia F, Wang H and Jia Y 2014 *Nat. Commun.* **5** 4458
- [15] Carvalho A, Wang M, Zhu X, Rodin A S, Su H and Neto A H C 2016 *Nat. Rev. Mater.* **1** 16061
- [16] Hafeez M, Gan L, Bhatti A S and Zhai T 2017 *Mater. Chem. Frontiers* **1** 1917
- [17] Ho C, Hsieh M, Wu C, Huang Y and Tiong K 2007 *J. Alloys Compd.* **442** 245
- [18] Ho C H, Huang Y S, Tiong K K and Liao P C 1998 *Phys. Rev. B* **58** 16130
- [19] Friemelt K, LuxSteiner M and Bucher E 1993 *J. Appl. Phys.* **74** 5266
- [20] Ho C H, Liao P C, Huang Y S, Yang T R and Tiong K K 1997 *J. Appl. Phys.* **81** 6380
- [21] Liang C, Chan Y, Tiong K, Huang Y, Chen Y, Dumcenco D and Ho C 2009 *J. Alloys Compd.* **480** 94
- [22] Wolverson D, Crampin S, Kazemi A S, Ilie A and Bending S J 2014 *ACS Nano* **8** 11154
- [23] Cui Q, He J, Bellus M Z, Mirzokarimov M, Hofmann T, Chiu H Y, Antonik M, He D, Wang Y and Zhao H 2015 *Small* **11** 5565
- [24] Cui Y, Lu F and Liu X 2017 *Sci. Rep.* **7** 40080
- [25] Chenet D A, Aslan O B, Huang P Y, Fan C, van der Zande A M, Heinz T F and Hone J C 2015 *Nano Lett.* **15** 5667
- [26] Ho C H, Huang Y S, Tiong K K and Liao P C 1999 *J. Phys.: Condens. Matter* **11** 5367
- [27] Lin Y C, Komsa H P, Yeh C H, Bjrkman T, Liang Z Y, Ho C H, Huang Y S, Chiu P W, Krashennnikov A V and Suenaga K 2015 *ACS Nano* **9** 11249
- [28] Tiong K, Ho C and Huang Y 1999 *Solid State Commun.* **111** 635
- [29] Yang S, Tongay S, Li Y, Yue Q, Xia J B, Li S S, Li J and Wei S H 2014 *Nanoscale* **6** 7226
- [30] Corbet C M, McClellan C, Rai A, Sonde S S, Tutuc E and Banerjee S K 2014 *ACS Nano* **9** 363
- [31] Corbet C M, Sonde S S, Tutuc E and Banerjee S K 2016 *Appl. Phys. Lett.* **108** 162104
- [32] Ovchinnikov D, Gargiulo F, Allain A, Pasquier D J, Dumcenco D, Ho C H, Yazyev O V and Kis A 2016 *Nat. Commun.* **7** 12391
- [33] Zhang E et al 2016 *ACS Nano* **10** 8067
- [34] Low T, Roldán R, Wang H, Xia F, Avouris P, Moreno L M and Guinea F 2014 *Phys. Rev. Lett.* **113** 106802
- [35] Favron A, Gaufrès E, Fossard F, Phaneuf-LHeureux A L, Tang N Y, Lévesque P L, Loiseau A, Leonelli R, Francoeur S and Martel R 2015 *Nat. Mater.* **14** 826
- [36] Murray H, Kelty S, Chianelli R and Day C 1994 *Inorg. Chem.* **33** 4418
- [37] Kelty S P, Ruppert A F, Chianelli R R, Ren J and Whangbo M H 1994 *J. Am. Chem. Soc.* **116** 7857
- [38] Tongay S et al 2014 *Nat. Commun.* **5** 3252
- [39] Huang Y S, Ho C H, Liao P C and Tiong K K 1997 *J. Alloys Compd.* **262** 92
- [40] Sim S, Lee D, Noh M, Cha S, Soh C H, Sung J H, Jo M H and Choi H 2016 *Nat. Commun.* **7** 13569
- [41] Sim S, Lee D, Trifonov A V, Kim T, Cha S, Sung J H, Cho S, Shim W, Jo M H and Choi H 2018 *Nat. Commun.* **9** 351
- [42] Zhong H X, Gao S, Shi J J and Yang L 2015 *Phys. Rev. B* **92** 115438

- [43] Gehlmann M *et al* 2017 *Nano Lett.* **17** 5187
- [44] Webb J L, Hart L S, Wolverson D, Chen C, Avila J and Asensio M C 2017 *Phys. Rev. B* **96** 115205
- [45] Zelewski S J and Kudrawiec R 2017 *Sci. Rep.* **7** 15365
- [46] Gutiérrez-Lezama I, Reddy B A, Ubrig N and Morpurgo A F 2016 *2D Mater.* **3** 045016
- [47] Feng Y *et al* 2015 *Phys. Rev. B* **92** 054110
- [48] McCreary A, Simpson J R, Wang Y, Rhodes D, Fujisawa K, Balicas L, Dubey M, Crespi V H, Terrones M and Hight Walker A R 2017 *Nano Lett.* **17** 5897
- [49] Hart L, Dale S, Hoye S, Webb J L and Wolverson D 2016 *Nano Lett.* **16** 1381
- [50] Godde T *et al* 2016 *Phys. Rev. B* **94** 165301
- [51] Robert C *et al* 2016 *Phys. Rev. B* **93** 205423
- [52] Robert C, Amand T, Cadiz F, Lagarde D, Courtade E, Manca M, Taniguchi T, Watanabe K, Urbaszek B and Marie X 2017 *Phys. Rev. B* **96** 155423
- [53] Padilha J, Peelaers H, Janotti A and Van de Walle C 2014 *Phys. Rev. B* **90** 205420
- [54] Molina-Sánchez A, Sangalli D, Hummer K, Marini A and Wirtz L 2013 *Phys. Rev. B* **88** 045412
- [55] Ramasubramaniam A 2012 *Phys. Rev. B* **86** 115409
- [56] Qiu D Y, Felipe H and Louie S G 2013 *Phys. Rev. Lett.* **111** 216805