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# Novel Pt electrocatalysts: multifunctional composite supports for enhanced corrosion resistance and improved CO tolerance

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#### **Abstract**

The electrochemical peculiarities of novel 20 wt.% Pt electrocatalysts supported on Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C composite materials in low-potential CO oxidation reaction (LPCOR) were investigated. The oxidation of CO on the Mo-containing Pt-based catalyst commences at exceptionally low potential values (ca. 100 mV). The results suggest that only CO adsorbed on specific Pt sites, where Pt and Mo atoms are in atomic closeness, can be oxidised below 400 mV potential. When the weakly bounded CO is oxidized, some hydrogen adsorption can take place on the released surface, although this amount is much smaller than in the case of a CO-free Pt surface. The Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst loses its activity in LPCOR when Mo becomes oxidized (above ca. 400 mV). Accordingly, presence of Mo species in lower oxidation state than 6+ is supposed to have crucial role in CO oxidation. Nevertheless, rereduction of oxidized Mo species formed above 400 mV is strongly hindered when adsorbed CO species are still present. Note that CO<sub>ads</sub> species can be completely removed only above 550 mV. Oxidized Mo species can be re-reduced and the activity in the LPCOR can be restored if the platinum surface is CO-free. Clear correlation between the so-called "prepeak", the molybdenum redox phenomenon and the CO tolerance of the 20 wt.% Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C system was established. Better performance of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C electrocatalyst compared to commercially available reference Pt/C and state-of-art COtolerant PtRu/C (Quintech) catalysts was also demonstrated.

**Keywords** Conducting Ti-based mixed oxides, TiMoO<sub>x</sub>, Composite materials, Pt electrocatalysts, CO-tolerance

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#### 1 Introduction

Considerable interest has been devoted in recent years to develop suitable electrode materials for the possible application in Polymer Electrolyte Membrane Fuel Cells (PEMFC). For fuel cells, state-of-art systems contain large quantities of platinum supported on carbon black. However, active carbon tends to corrode under the fast load change conditions [1]. It is therefore important to explore alternative materials that can provide improved stability across the anticipated potential/pH window compared to carbon [2].

Moreover, Pt/C electrocatalysts suffer from other drawback regarding their poor tolerance to CO poisoning, as a small amount of CO is inevitably present in fuels obtained by reforming of hydrocarbons. This problem has led to an intensive effort to develop so-called CO-tolerant anode catalysts. Accordingly, current research on fuel cell catalysts focuses on developing durable, CO-tolerant and efficient electrocatalysts, which are the most compelling practical needs for real-life applications.

Materials composed of Pt nanoparticles (NPs), TiO<sub>2</sub> and carbon materials are identified as efficient electrocatalysts since they combine the high conductivity of carbon and corrosion resistance of the oxide with the synergistic effect between metal oxides and Pt [3]. Highly graphitic carbons provide a good backbone for retaining the original nanoscale and uniform TiO<sub>2</sub> coatings (< 10 nm) without sintering during heat treatment [4]. The strong interaction between the TiO<sub>2</sub> coating on carbon and the Pt results in drastically increased electrochemically active Pt surface area and inhibits the agglomeration and corrosion of the metal NPs [5]. Composite support materials, such as TiO<sub>2</sub>-C also show improved stability in accelerated aging tests [6-8].

Doping of titania with n-type dopants such as transition metals (e.g., W, Nb and Ta [9-12]) results in materials with high electrochemical stability and unique CO-tolerance, along with an increase of the electric conductivity. However, conductivities of these materials remain still substantially lower than that of carbon. In Ref. [13] the Ti<sub>0.7</sub>W<sub>0.3</sub>O<sub>2</sub> mixed oxide was mixed with carbon powder in an attempt to increase its conductivity required for electrocatalyst applications.

Due to their enhanced CO tolerance tungsten- and molybdenum-oxide supported Pt electrocatalysts have been considered as potential candidates at the PEMFC anode [14-17]. A very high electrocatalytic activity observed at MoO<sub>x</sub>-Pt catalyst for the hydrogen oxidation reaction (HOR, anode reaction) in the presence of 100 ppm CO was attributed to chemical surface reaction of adsorbed CO with Mo surface oxides [18,19].

However, their application is limited by their instability due to leaching of the non-noble component at high potentials in both alkaline and acidic conditions [20]. The latter is particularly important, as PEMFC works under acidic conditions. The leaching can not only decrease catalyst activity and CO tolerance but may also result in reduced electrolyte performance due to encapsulation of the leached components in the membrane.

Despite to the beneficial co-catalytic effect of molybdenum in electrooxidation of CO and alcohols, its behaviour is very complex since it may be present in several oxidation states and might form distinct soluble species and various oxides [21]. In the literature the information about the stability of Mo-containing catalysts is still controversial. While some studies report the occurrence of the dissolution of non-alloyed molybdenum after only a few voltammetric cycles [22,23], other investigations have shown the presence of stable molybdenum species even after several cycles at potentials between 0.0 and 1.2 V [24]. Upon using the Pt<sub>4</sub>Mo/C catalysts Papakonstantinou et al. [25] proposed that CO oxidation process was accompanied with oxidation of the  $Mo^{x+}$  (x= 4 or 5) species into  $MoO_3$ , which dissolved and then, upon negative current application, reactivated into catalytically active sites via electrodeposition and subsequent reduction. However, using Pt<sub>x</sub>Mo<sub>y</sub>/C catalysts [26] it has been stated that the intermetallic bonding stabilizes Mo on the surface under potentials of practical interest for use in fuel cell anodes. Thus, it has been suggested [26] that at ca. 0.4 V and higher there is oxidation of Mo to the (6+) state, which in the pure metal results in dissolution, but in the alloy this oxidation state is stabilized in the surface by the intermetallic bonding. The effectiveness of the MoO<sub>x</sub>@Pt core-shell and well-defined Pt<sub>0.8</sub>Mo<sub>0.2</sub> alloy electrocatalysts to improve anode performance with higher CO tolerance and durability was demonstrated [27,28]. It has been shown that leaching may be mitigated with a well-ordered alloy or a core-shell catalyst architecture with a seemingly pure Pt surface.

It has been demonstrated that [29,30] strong metal support interaction between Pt and MoO<sub>3</sub> results in significant improvement of the stability of the electrocatalysts. Recently, the combination of the highly stable TiO<sub>2</sub> with MoO<sub>x</sub>, was reported to have high electronic conductivity and relative stability in acid solutions. Accordingly, the multifunctional Ti<sub>0.7</sub>Mo<sub>0.3</sub>O<sub>2</sub> support material was successfully applied for Pt-based catalysts for the oxygen reduction reaction (ORR, cathode reaction) [31,32]. Recently the use of Mo-doped titanium suboxide (Ti<sub>3</sub>O<sub>5</sub>-Mo) has been demonstrated as an alternative carbon-free catalyst support for the ORR in low-temperature fuel cells [33]. The Pt/Ti<sub>3</sub>O<sub>5</sub>-Mo catalyst exhibits excellent electrocatalytic activity and stability toward the ORR. A deep accelerated potential cycling

between 0 and 1.2 V vs RHE up to 5000 cycles demonstrated the remarkable stability of this catalyst.

Furthermore, in acidic solutions hydrogen molybdenum bronze of type  $H_xMoO_3$  could also be formed by the "spillover" of hydrogen dissociated on Pt metal sites towards Mo, that may have a contribution to the enhanced catalytic activity in CO and methanol electrooxidation [34-36]. The intercalation of hydrogen into the  $MoO_3$  lattice is accompanied by reduction of the metal ions and formation of  $H_xMoO_3$  with Mo in mixed valent state (5+/6+) [34]. Moreover, strong metal support interaction required between Pt and  $MoO_3$  for the formation of  $H_xMoO_3$  leads to significant enhancement in electrocatalytic activity and stability [34].

In our recent studies [37-39] stabilization of tungsten was achieved by incorporation into the rutile-TiO<sub>2</sub> lattice, while the conductivity was ensured by a mixed oxide-activated carbon composite. A new synthesis method optimized for preparation of  $Ti_{0.7}W_{0.3}O_2$ -C composites with the  $Ti_{0.7}W_{0.3}O_2$ /C mass ratio of 50:50 and almost exclusive tungsten incorporation into the rutile lattice was developed. Enhanced CO tolerance of novel 40 wt.% Pt/ $Ti_{0.7}W_{0.3}O_2$ -C electrocatalysts compared to the commercial Pt/C or state-of-art CO tolerant PtRu/C (Quintech, Pt= 20 wt.%, Ru= 10 wt.%) catalysts were also evidenced [37]. In fact, performance even exceeding that of the PtRu/C benchmark was observed.

The electrochemical activity and CO-tolerance was confirmed by measurements in a single fuel-cell test device as well, where it was shown that 100 ppm CO concentration in the H<sub>2</sub> fuel caused 140 mV voltage loss at 1 A cm<sup>-2</sup> current density upon using the 40 wt.% Pt/Ti<sub>0.7</sub>M<sub>0.3</sub>O<sub>2</sub>-C (M= W, Mo) catalysts, while in the case of the Pt/C and PtRu/C the loss was 360 and 240 mV, respectively, confirming the advantageous properties of the new type of composite supported electrocatalysts [40].

The described results demonstrate that the idea of combining the high surface area and good conductivity of active carbon with the excellent stability of  $TiO_2$  and with the beneficial co-catalytic properties of M dopant oxides (M= W, Mo) is indeed a feasible way to develop a new generation of Pt-based electrocatalysts for PEM fuel cells. Moreover, leaching of the non-noble metal modifiers can be effectively suppressed by exclusive incorporation of the dopant into the  $TiO_2$ -rutile lattice.

Preparation of 20 wt.% Pt/Ti<sub>(1-x)</sub>Mo<sub>x</sub>O<sub>2</sub>-C (x= 0.2 - 0.4) composite supported catalysts with the  $Ti_{(1-x)}Mo_xO_2/C$  mass ratio 75:25 were described in our recent paper [41]. It has been demonstrated that different Mo species have critical importance on the electrocatalytic performance both in hydrogen oxidation and CO tolerance. The importance of Mo ions

incorporated into the rutile lattice was underlined. Non-incorporated Mo species are not stable upon applying a wide cyclic polarization window and as a consequence they are removed gradually, resulting in unstable electrocatalytic performance. The Ti/Mo atomic ratio of 80/20 has been suggested as an optimal composition having the largest ratio of incorporated/non-incorporated Mo species.

In the present work our aim is to give a more elaborate insight into the behaviour of the  $Ti_{(1-x)}Mo_xO_2$ -C composite supported Pt catalysts. The primary emphasis is on investigating the electrochemical peculiarities of the system in order to establish a connection between the so-called "pre-peak", the molybdenum redox phenomenon and the CO tolerance of the  $Ti_{0.6}Mo_{0.4}O_2$ -C composite supported Pt catalyst in comparison with commercially available Pt/C and CO-tolerant PtRu/C catalysts.

#### 2 Experimental

# 2.1 Preparation of Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C support materials and Pt catalysts

Titanium-isopropoxide (Ti(O-i-Pr)<sub>4</sub>, Aldrich, 97%) and ammonium heptamolybdate tetrahydrate ((NH<sub>4</sub>)<sub>6</sub>Mo<sub>7</sub>O<sub>24</sub> × 4H<sub>2</sub>O, Merck, 99%) were used as Ti and Mo precursor compounds.

In order to prepare the composite support containing mixed oxide with Ti/Mo atomic ratio of 60/40 (Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>) and activated carbon in a mass ratio of 75:25 between the oxide and the carbon, first transparent acidic TiO<sub>2</sub> colloidal solution was made by adding concentrated HNO<sub>3</sub> to a vigorously stirred mixture of Ti(O-*i-Pr*)<sub>4</sub> and distilled water. Active carbon (CABOT, Black Pearls 2000, specific surface area:  $1475 \text{ m}^2/\text{g}$ ) was added after 4 h stirring at room temperature (RT). In order to facilitate the formation of rutile nuclei, the mixture was aged at RT for 4 days. After the aging, molybdenum precursor compound ((NH<sub>4</sub>)<sub>6</sub>Mo<sub>7</sub>O<sub>24</sub> × 4H<sub>2</sub>O) was added and the solution was evaporated at 80°C. The powder was dried at 80°C overnight. Detailed description of the synthesis procedure can be found in our previous studies [40,41]. The obtained powder has been submitted to high-temperature treatment (600°C, 8h, Ar flow).

The support material was loaded with 20 wt.% Pt via the previously used modified NaBH<sub>4</sub>-assisted ethylene-glycol reduction-precipitation method [40,41]. The materials were washed with water by centrifugation and dried at 80°C overnight.

## 2.2 Electrochemical characterization

Electrochemical measurements were carried out at RT in a standard three-electrode electrochemical cell. The supporting electrolyte was  $0.5 \text{ M H}_2\text{SO}_4$ , the counter electrode was a platinum wire, the reference electrode was a hydrogen electrode immersed in the same electrolyte as the working electrode and all potentials are given on the RHE scale. The working electrode for both static and rotating disc electrode (RDE) measurement was prepared by dipping a drop of catalyst ink on a freshly polished glassy carbon electrode (d = 0.3 cm, geometric surface area  $A = 0.0707 \text{ cm}^2$ ), and air-drying at RT for 30 min. Pt loading of the electrode was  $10 \text{ µg cm}^{-2}$ . For the catalyst ink 5 mg catalyst sample was suspended in 4 ml  $\text{H}_2\text{O} + 1 \text{ ml}$  isopropanol + 20 µl Nafion solution (D520 Nafion Dispersion - Alcohol based 1000 EW at 5 wt%, DuPont<sup>TM</sup> Nafion<sup>®</sup>).

Before the measurements the electrolyte was purged with Ar for at least 20 minutes then 10 cyclic voltammogramms (CVs) were recorded with 100 mV s<sup>-1</sup> sweep rate between 50 and 1000 mV for conditioning the catalysts.

For the electrochemical characterization of the catalyst samples two methods were applied: Ar purged and H<sub>2</sub> purged CO<sub>ads</sub> stripping voltammetry.

In a case of Ar purged CO<sub>ads</sub> stripping measurements after conditioning the electrode, the potential was set to 50 mV and CO gas was fed into the electrolyte solution for 20 minutes, then the liquid was purged with Ar for another 20 minutes. Ar purged CO<sub>ads</sub>-stripping voltammetry measurements were done using three different potential programs. For the whole series of measurements the potential programs used can be seen in the insert of corresponding figures.

The effect of CO tolerant behaviour on hydrogen oxidation reaction was also investigated by RDE method. A modified  $H_2$  purged  $CO_{ads}$  stripping voltammetry coupled with HOR on RDE was used in this study.

First CO gas was bubbled into the electrolyte solution for 30 minutes then  $H_2$  for 30 minutes while maintaining the potential at 50 mV. At the end of the  $H_2$  purging the working electrode was polarized to 250 mV and held for 3 minutes at this potential. Then it was polarized to 0 mV, and 3 CVs were recorded between 0 mV and 250 mV with 10 mV s<sup>-1</sup> sweep rate while the disc electrode was rotated at 625 rpm and the  $H_2$  gas was bubbled to the solution. This process was repeated with the difference in the anodic potential limits: 350 mV, 450 mV, 550 mV and 1000 mV.

For comparison as reference commercially available 20 wt.% Pt/C (Quintech, C-20-Pt, on Vulcan; denoted hereafter as Pt/C) and PtRu/C, considered as the state-of-art CO-tolerant

electrocatalyst (Quintech, C-20-/10-Pt/Ru, Pt= 20 wt%, Ru= 10 wt% on Vulcan; denoted hereafter as PtRu/C) were also studied by the same method as described above.

### 2.3 Physical characterization

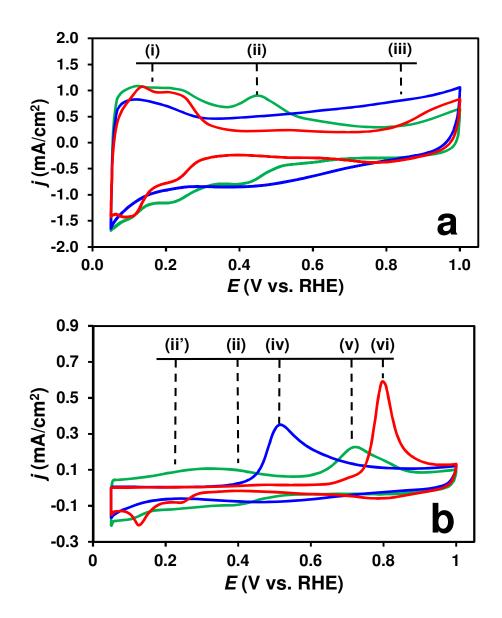
X-ray photoelectron spectroscopy (XPS) measurements were carried out using an EA125 electron spectrometer manufactured by OMICRON Nanotechnology GmbH (Germany). The photoelectrons were excited by unmonochromatized MgKα (1253.6 eV) radiation. Spectra were recorded in the Constant Analyser Energy mode of the energy analyser with 30 eV pass energy resulting in a spectral resolution around 1 eV. The powder catalyst samples were suspended in isopropanol then some of this suspension was dried to a standard OMICRON sample plate. Binding energies were referenced to the main component of the C 1s spectrum of the support (graphite at 284.4 eV binding energy). Data were processed using the CasaXPS software package [42].

#### 3 Results and Discussion

Fig. 1 compares the CVs and the CO<sub>ads</sub> stripping voltammograms recorded on the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C, PtRu/C and Pt/C catalysts. The base voltammograms for PtRu/C and Pt/C catalysts (Fig. 1a) show the characteristic hydrogen (i) and oxygen (iii) adsorption/desorption regions [43] and shall not be discussed here. A typical CV of Pt with the classical features of the underpotentially deposited hydrogen adsorption/desorption between 50 mV and 350 mV (i) along with a redox peak between 380 and 530 mV (ii) was observed on the Mo-containing composite supported sample presented in Fig. 1a. According to the literature [34,44] the appearance of strong redox peaks in the voltammograms clearly confirms that there is an active interface between the Pt NPs and Mo-containing support.

Comparison of the CO<sub>ads</sub> stripping behaviour of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C electrocatalyst to the reference commercial Pt/C and PtRu/C catalysts (see Fig. 1b) demonstrates increased CO tolerance of the composite supported catalyst. As can be seen in Fig. 1b, on the Pt/C catalysts the main CO oxidation peak is located at ca. 795 mV (vi), while the corresponding peak in the case of the CO-tolerant PtRu/C catalysts is around 505 mV (iv). The CO<sub>ads</sub> stripping voltammograms for PtRu/C and Pt/C catalysts (Fig. 1b) resemble those commonly observed in the literature [43] with the onset potential for CO oxidation (E<sub>CO, onset</sub>) of about 255 and 625 mV, respectively. As shown in Fig. 1b the CO<sub>ads</sub> stripping voltammogram for the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst displays two clear oxidation features: the main anodic peak at 705 mV (v), and a "pre-peak" consisting of two small overlapping peaks centered at ca. 215 (ii')

and 400 mV (ii), which may be attributed to partial oxidation of weakly adsorbed CO (ii') and oxidation of Mo surface species (ii), respectively; details on their position (Fig. 4a) as well as their relationship to the CO tolerant behaviour will be discussed below. According to the Arpurged voltammograms, the  $E_{CO, onset}$  on the composite supported catalyst is not higher than 50-100 mV. It is necessary to mention, that the early onset of CO oxidation as a "pre-peak" is typical for Pt-based catalysts containing Ti, Ta, W, and Mo suboxides [33,45].



**Fig. 1.** Cyclic (a) and Ar-purged  $CO_{ads}$  stripping (b) voltammograms of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C (green line), PtRu/C (blue line) and Pt/C (red line) catalysts. Recorded in 0.5 M H<sub>2</sub>SO<sub>4</sub>, T = 25 °C, at sweep rate 100 mV s<sup>-1</sup> (a) and 10 mV s<sup>-1</sup> (b). Marks of the anodic peaks: (i) - hydrogen desorption; (ii') - oxidation of weakly bonded CO; (ii) - oxidation of Mo surface

species; (iii) - oxygen adsorption; (iv), (v) and (vi) - oxidation of strongly bonded CO on PtRu/C, Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C and Pt/C, respectively.

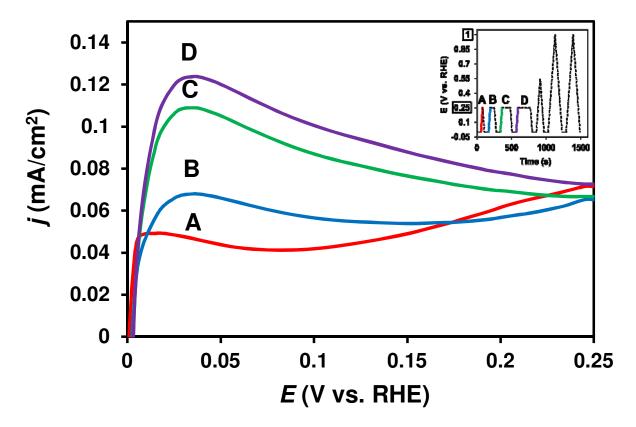
It has been demonstrated by Differential Electrochemical Mass Spectrometry (DEMS) that both peaks observed in the CO<sub>ads</sub> stripping voltammograms ("pre-peak" and the main peak) are related to the CO<sub>2</sub> production [44,46]. At the same time, DEMS results presented for Pt-electrodes modified with Mo reveal that CO<sub>2</sub> formation starts around 0.15 V overlapping with the onset of Mo oxidation [46]. Although the mechanism by which Mo promotes the oxidation of the adsorbed CO in the Pt surface is not yet known, it is generally considered to be of a bifunctional nature in conjunction with electronic effect (influence of the catalyst support on the electronic state of active phase) [47]. In agreement with the bifunctional mechanism, the co-catalytic activity is supposed to be due to the good redox properties of the oxides involved.

Nevertheless, identifying the co-catalytically active molybdenum ionic state is a challenging task. Usually, molybdenum oxides do not exist in a single phase but in mixed-valence oxides ( $MoO_x$ ) [35]. The specific assignment of the Mo oxidation peak is not consensual, but the comparison with Pourbaix diagram for Mo indicates that this redox behaviour is due to the  $Mo^{4+} \leftrightarrow Mo^{6+}$  couple [26]. However, according to *in situ* X-ray absorption spectroscopy (XAS) studies of the Mo K edge of PtMo/C catalysts it has been demonstrated by Mukerjee *et al.* [24] that at potentials as low as 0.0 V Mo is present in  $Mo^{5+}$  oxidation state as some sort of a hydrated oxide close to  $MoO(OH)_2$ . Confirming the electrochemical results direct spectroscopic evidence of change in the oxidation state  $Mo^{5+}/Mo^{6+}$  between 0.0 and 0.54 V was provided [24,47]. The  $MoO(OH)_2$  species [24-26,48] or in general the hydrogen Mo bronzes  $H_xMoO_3$  [49,50] are considered as the reactive moiety with adsorbed CO. It has been proposed that CO species, which are adsorbed on Pt atoms adjacent to Mo atoms, react easily with the OH species of the  $MoO(OH)_x$  toward the formation of  $CO_2$  at potentials lower than 0.25 V, cleaning this way the Pt active site from CO poisoning species [25,35,51]:

$$(MoO_x)-OH_{ads} + Pt-(CO)_{ads} \rightarrow CO_2 + Pt + MoO_x + H^+ + e^-$$
 (1)

Subsequently, the empty space created on the  $Mo^{x+}$  species facilitates the heterolytic dissociative adsorption of water and the regeneration of  $MoO(OH)_x$  [25]. Thus, the unique properties of Mo-containing catalysts at low potentials (< 0.2 V) depend on the oxidation state

of the Mo surface species and specifically on the reactive OH species of  $MoO(OH)_x$  and their ability to spontaneously dissociate  $H_2O$ . Papakonstantinou *et al.* [25] proposed that at potentials higher than 0.2 V the average oxidation state of  $Mo^{x+}$  is larger than 5+, resulting in the deactivation of the Mo species due to the formation of  $MoO_3$ . However, the reduction upon negative current application results in the regeneration of  $(H_2MoO_3)_{ads}$  species on the Pt surface. The electrochemical experiments presented below were designed to give further insight into this question.



**Fig. 2.** Argon purged  $CO_{ads}$  stripping voltammograms obtained on the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst applying the potential program represented by the insert (the colour of the insert solid lines corresponds to the colour of voltammograms shown on the figure). Electrolyte: 0.5 M H<sub>2</sub>SO<sub>4</sub>; scan rate: 10 mV s<sup>-1</sup>; T = 25 °C.

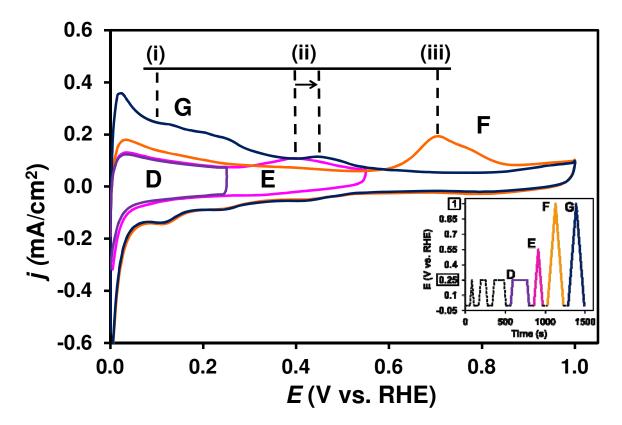
The fact that CO oxidation indeed occurs below 250 mV potential on the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst has been verified using electrochemical measurements. A set of Arpurged CO<sub>ads</sub> stripping experiments can be seen in Fig. 2. In this experiment at the end of the Ar purging the working electrode was set to 0 mV and held at this potential for 1 minute. Since at Pt surfaces hydrogen adsorption and evolution take place at 0 mV potential, this treatment results in significant oxidation current during the anodic scan. In this case all Pt

sites not blocked by the pre-adsorbed CO are involved in the oxidation of hydrogen. Following this hydrogen exposure a linear potential sweep was measured between 0 and 250 mV. This potential program (i.e. 1 min at 0 mV and linear sweep) was repeated three times with the difference that before going back to 0 mV the potential was kept at 250 mV for 1, 2 and 3 minutes, respectively. The potential program of this series of measurements can be seen in the insert of Fig. 2.

As emerges from curve *A* in Fig. 2 there is no significant oxidation current during the first sweep from 0 mV to 250 mV. Since adsorbed CO blocks the platinum catalytic sites both adsorption and evolution of hydrogen are hindered. However, from the beginning of the second and during the subsequent cycles an anodic current peak can be observed in this potential region. As emerged from Fig. 2, the longer the holding time at 250 mV is, the higher this anodic peak becomes, which can be correlated to the oxidation of the hydrogen adsorbed on accessible Pt sites. In other words [37,43,52] such experiments reveal the potential at which vacancies are formed in the CO<sub>ads</sub> monolayer. Once CO desorbs from a fraction of the catalytic sites, Pt sites become available for the dissociation / electrooxidation of H<sub>2</sub> molecules resulting in a steep increase of the current up to the diffusion-limited plateau. These results indicate that oxidation of CO commences at as low as 250 mV potential and hydrogen adsorption and evolution can take place on the gradually released active sites.

Results of the continuation of the previous measurement are presented in Fig. 3. After the measurements presented in Fig. 2 the anodic limit of the potential scan was increased to 550 mV (above the so called pre-peak potential (ii)) and finally to 1000 mV in order to oxidize the whole amount of adsorbed CO (iii) which remained at the surface during the previous scans (the corresponding potential program can be seen in the insert of Fig. 3).

Fig. 3 shows the influence of the step-by-step increase of the polarization limits on the shape of the voltammograms and on the presence or absence of the pre-oxidation peak. It can be seen that increasing the anodic limit of the polarization to more positive values resulted in an increase of the hydrogen oxidation currents (i).

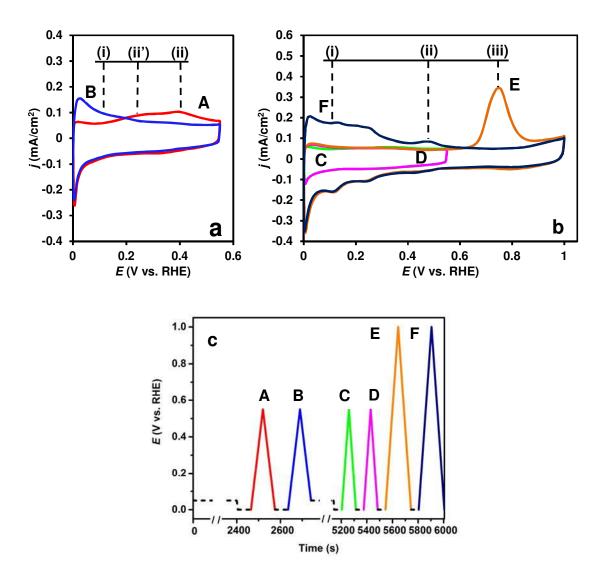


**Fig. 3.** The CO<sub>ads</sub> stripping voltammograms of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst obtained after Ar purging using different potential limits. The insert shows the potential program (the colour of the insert lines indicating the potential cycling conditions applied during the measurements corresponds to the colour of voltammograms shown on the figure). Obtained in 0.5 M H<sub>2</sub>SO<sub>4</sub> at 10 mV s<sup>-1</sup>, T = 25 °C. Marks of the anodic peaks: oxidation of (i) - hydrogen, (ii) - Mo surface species, and (iii) - strongly bonded CO.

Comparing the curves D and E in Fig. 3 it can be seen that after holding the potential at 250 mV for 3 minutes there is no further change in the shape of the voltammetric curves in the hydrogen oxidation region (i) (up to 250 mV curve D practically coincides with curve E). From this result it can be concluded that during this time the whole amount of weakly adsorbed CO, which can be oxidized below 250 mV, has been eliminated from the catalyst surface. After increasing the potential limit of the polarization up to 550 mV a characteristic "pre-peak" (ii) was observed in the first cycle around 400 mV (see curve E in Fig. 3). As it was mentioned above, the presence of the pre-oxidation peak can be associated with two processes: partial oxidation of weakly adsorbed CO and oxidation of Mo surface species, which correspond to the peaks (ii') and (ii) in Fig. 1b, respectively.

This pre-oxidation peak (ii) completely disappears in the next cycle (see curve F in Fig. 3), however it has been re-emerged when the electrode was polarized to 1000 mV (see

curve *G* in Fig. 3), where the whole amount of adsorbed CO was oxidized and Pt surface has been completely recovered, resulted in pronounced increase of the hydrogen oxidation peak (i).



**Fig. 4.** The  $CO_{ads}$  stripping voltammograms of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst obtained after Ar purging using different potential limits (a, b); (c) shows the potential program (the colour of the insert lines indicating the potential cycling conditions applied during the measurements corresponds to the colour of voltammograms shown on the figure). For further details see the text. Obtained in 0.5 M H<sub>2</sub>SO<sub>4</sub> at 10 mV s<sup>-1</sup>, T = 25 °C. Marks of the anodic peaks: oxidation of (i) - hydrogen, (ii') - weakly bonded CO, (ii) - Mo surface species, and (iii) - strongly bonded CO.

The following experiment gives further insight into the nature of the pre-peak and its relation to the CO oxidation process on Mo-containing catalysts. Saturation of the catalyst

surface with  $CO_{ads}$  and purging the CO from the solution were carried out the same way as in the preceding experiments; then two cyclic voltammograms were recorded after holding the potential of the electrode at 0 mV for 1 min (curves A and B in Fig. 4a).

The shape of the first voltammogram (curve *A* in Fig 4a) recorded between 0-550 mV is similar to that in Fig. 1b. Comparison with curve *E* in Fig. 3 reveals that in addition to the feature around 400 mV (ii), in this case a clear peak (or hump) appears below 300 mV potential, which can be assigned to the electrooxidation of weakly bounded CO (ii') (these species were removed during the steady polarization at 250 mV in the experiment represented by Figs. 2-3). Between 0 and 550 mV the anodic part of the second cycle (curve *B*, Fig. 4a) is identical with curve *F* in Fig. 3; there is only characteristic current signal (i) of the oxidation of the small amount of hydrogen (which could be evolved or adsorbed on the active sites released from the weakly bounded CO).

After recording these two voltammograms presented in Fig 4a the CO-saturation step was repeated; first CO and then Ar was bubbled into the solution for 20-20 min, while holding the potential at 50 mV (the applied potential cycling conditions are shown on Fig. 4c). In spite of the exactly same procedure, the first and the second voltammograms (curves C and D, in Fig 4b) obtained after the second CO saturation significantly differ from curves C and C in Fig 4a. All characteristic peaks of the voltammograms represented by curves C and C in Amaly, the pre-oxidation peak of the weakly bounded CO (ii') and the hydrogen oxidation peak in the second cycle (i) are missing from curves C and C in Fig 4b. It means that the catalytic surface lost its capability to oxidise CO at low potential values and the active sites are blocked by adsorbed CO. At the same time the Mo redox peak pair around 400 mV (ii) also disappeared, which indicate that CO tolerance can be in close connection with these features.

However, in this case the loss of activity is not an irreversible process. After oxidation of the strongly adsorbed CO (iii) with a simple potential scan to 1000 mV (see curve E in Fig 4b) the hydrogen oxidation peak (i) and the Mo redox peak pair (ii) could be recovered (curve E in Fig 4b), as well as the low potential CO oxidising capability of the surface.

The co-catalytic effect of Mo on the oxidation of adsorbed CO at low potentials can be explained by an oxygen spillover effect (see Eq. 1) requiring OH groups from Mo species with oxidation state lower than 6+. This process is only working for CO in the weakly adsorbed state, while at high potentials (above 550 mV) the oxides of Pt begin to play a principal role in the electrooxidation of strongly bonded CO.

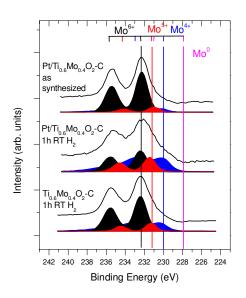
Moreover, it has been observed that, when the catalysts are submitted to potentials above 400 mV, the Mo species become oxidized and simultaneously their activity is lost. These results suggest that this extensive oxidation hinders the oxygen spillover capability of the Mo species in close vicinity to Pt particles, leading to the loss of the CO oxidation activity at low potentials. In order to restore this activity, reduction of the mentioned Mo species is needed, which requires an abundant source of hydrogen in the form of hydrogen spillover [53] and/or charge transfer from Pt particles released from CO [54]. It is supposed that the hydrogen replacing the weakly adsorbed CO (evidenced by the hydrogen oxidation peak (i) in Fig. 4a (curve *B*)) is either not reactive enough or not abundant enough to restore the reduced state of Mo species.

The existence of the hydrogen spillover process needed for reactivation of the catalyst is indicated by our XPS measurements. In Fig. 5 Mo 3d core level spectra are shown for a Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst in the as received state (after the final annealing of the synthesis, the sample was stored in air for a while) and after exposure to 300 mbar H<sub>2</sub> in the preparation chamber of the electron spectrometer for 1 h at room temperature. These spectra are compared to that obtained for a Pt-free Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C support after a hydrogen exposure with the same conditions as in the case of the Pt-containing electrocatalyst.

As it was described in our earlier works [40,41], air exposure of the composite support results in oxidation of surface Mo species to the Mo<sup>6+</sup> oxidation state, which is reflected by the strong spin-orbit doublet around 232.4 and 235.5 eV [55]. In the spectrum of the as received Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst weak contributions from Mo<sup>5+</sup> and Mo<sup>4+</sup> arise as a doublet at 231.0 and 234.1 eV (Mo<sup>5+</sup>, [55]) and as a broad band with leading peak at 230.0 eV (Mo<sup>4+</sup>, [56,57]), respectively. In this work, the Mo 3d core level spectrum from Mo<sup>4+</sup> species was modelled by a complex line shape consisting of a combination of a narrower and a broad spin-orbit splitted doublet representing differently screened final states, with the most intense contribution between 229-230 eV [56,57]. In spite of the air exposure of the as received samples, the Pt content of the composite supported electrocatalyst is predominantly metallic (data not shown).

The room temperature hydrogen exposure almost completely reduced the Pt. Moreover, in Fig. 5 considerable reduction of the Mo content of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C sample is evident: the intensity from both the Mo<sup>5+</sup> and Mo<sup>4+</sup> contributions significantly increased at the expense of the Mo<sup>6+</sup> signal (all Mo 3d spectra were satisfactorily fitted with the mentioned Mo<sup>6+</sup>, Mo<sup>5+</sup> and Mo<sup>4+</sup> contributions). On the contrary, only a marginal Mo reduction occurs in the Pt-free composite support material. As hydrogen is dissociatively adsorbed on Pt, our

observation demonstrates that the Pt particles act as sources for reactive hydrogen species, which easily reduce even the heavily oxidized Mo species in the vicinity of the Pt. In the lack of Pt, such spillover induced reduction is absent. It may be worth to note that the Ti content of the composite support cannot be reduced this way: the Ti 2p spectra of either the Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C support or the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C electrocatalyst are essentially identical both in the as received (air exposed) or reduced (H<sub>2</sub> exposed) state.



**Fig. 5.** Effect of 300 mbar H<sub>2</sub> exposure for 1 h at room temperature on the Mo 3d core level spectra of Pt-loaded and Pt-free Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C composites compared to that of a Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst in the "as received" state. Hydrogen treatments were carried out in the preparation chamber of the electron spectrometer, thus air exposure was eliminated during the experiments.

The above electrochemical observations confirm that the key feature of the CO tolerance of the Mo-containing Pt-based electrocatalysts is the presence of partially oxidized Mo species, while the XPS data proved that these species have to be located in atomic closeness to Pt. These species provide oxidizing agents for oxidation of CO in a process described in Eq. 1 above. They are relatively stable under the conditions of the electrochemical experiments, unless the applied potential exceeds 350 mV. Under higher potential, they are oxidized into stable, non-reactive Mo species (presumably into Mo<sup>6+</sup>), which cannot be reduced by a cathodic scan unless the surface is polarized to potentials high enough for oxidation of the strongly bound CO species. As it was mentioned above, it can be

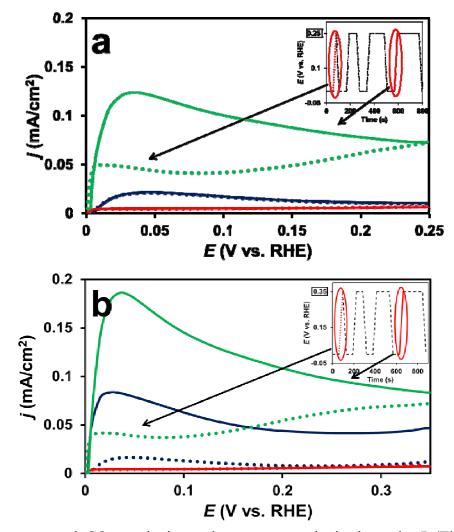
assumed that hydrogen replacing the weakly adsorbed CO is not suitable to accomplish this regenerative process. Another possibility may be that some CO oxidation products block the inactivated site and prevent its reduction, until they are removed together with the strongly bound CO. The extensive literature on CO electrooxidation on Pt suggests that carboxyl groups may be candidates for this role: both *in situ* Fourier-transform infrared spectroscopy (FTIR) [58,59] and electrochemical quartz crystal microbalance (EQCM) [60,61] experiments as well as *ab initio* calculations [62] indicate their formation and demonstrate their stability up to at least 0.55 V. Carbonate species arising from surface reaction of CO<sub>2</sub> with water adsorbed on Pt (where H<sub>2</sub>O acts as a source of oxygen for the oxidation reaction [58,63]) are reported to be observable above 0.8 V. However, it has been mentioned, that the concentration of CO<sub>3</sub><sup>2-</sup> or HCO<sub>3</sub><sup>-</sup> is very low in the strong acid electrolyte. Carbonyl species like the formyl group are less likely candidates for this site blocking role, as such groups are usually observed upon electrooxidation of methanol or formic acid in acid solutions [64]. Nevertheless, further discussion of the possible intermediates formed during the CO electrooxidation is out of the scope of this work and will not be further considered.

Qualitatively very similar findings to those described above were earlier reported by Papakonstantinou *et al.* [25] on carbon supported Pt<sub>4</sub>Mo alloy catalysts. They argued that since the presence of CO on the surface blocks the adsorption Pt sites, the reduction of the Mo<sup>x+</sup> species on the Pt surface are prevented. As a consequence, in the presence of CO, negative current application does not result in the reactivation of the catalytically active sites. Moreover, much more pronounced Mo redox process involving protons and electrons can be observed for the Mo sites located in a close contact with Pt [18,26,53,65,66]. Thus we believe that the hindered reduction of highly oxidized Mo species in the presence of strongly bound CO is a general feature of the Mo-Pt catalyst system, regardless if Mo is present in PtMo alloy form or Mo oxide is used as a constituent of the support.

In order to demonstrate the exceptional low potential, where the CO oxidation commences, experiments presented in Fig. 2 were repeated extending the investigation to commercial PtRu/C and Pt/C catalyst samples. Fig. 6a shows the results of this experiment with 250 mV upper potential limit, while in Fig. 6b this potential limit was increased to 350 mV. For the sake of visibility only the first and the fourth voltammograms are depicted.

Upon using of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst (see Fig. 6a) in the first anodic scan no oxidation current can be observed, but during the fourth anodic scan immediately from the beginning high peak of oxidation current was detected. In contrast, for the commercial PtRu/C and Pt/C catalysts the current density values are the same in the first and the fourth scans. For

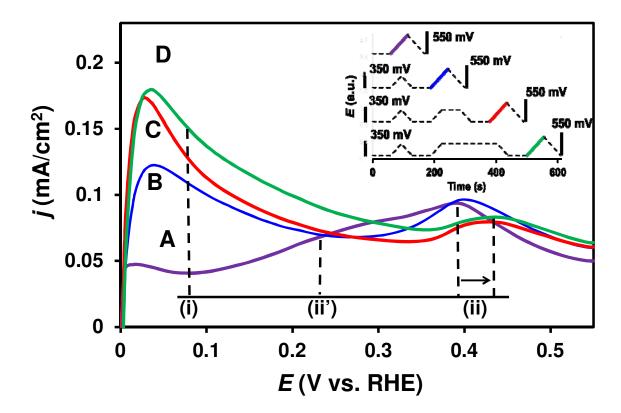
the Mo-containing catalyst the oxidation current of the *in-situ*-developed-H<sub>2</sub> is higher in the case of the experiment limited to 350 mV due to the larger CO free platinum surface caused by the higher potential value of the sweep, and a small activity already appears for the PtRu/C catalyst as well, see on Fig. 6b. However in case of Pt/C no similar effect could be observed.



**Fig. 6.** Argon purged  $CO_{ads}$  stripping voltammograms obtained on the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C (green line), PtRu/C (blue line) and Pt/C (red line) catalysts using different potential limits (A: 250 mV; B: 350 mV) of the hydrogen potential range. The insert shows the potential program (the arrow indicates the potential cycling conditions applied during the measurements). Voltammograms were recorded after a potential stop at 0 V for 1 min (dotted lines) and after sequential potential cycling steps according to the insert (solid lines). Obtained in 0.5 M  $H_2SO_4$  at 10 mV s<sup>-1</sup>, T = 25 °C.

Complex character of the pre-peak in the case of the  $Pt/Ti_{0.6}Mo_{0.4}O_2$ -C catalyst is demonstrated by Fig. 7. Anodic scans of four different modified  $CO_{ads}$  stripping voltammetry

measurements are represented between 0 mV and 550 mV. The purpose of these measurements was to provide information about the correlation between the CO oxidation pre-peak and the hydrogen oxidation activity. In this experiment each measurement was carried out on freshly prepared electrode, thus each was preceded by the CO saturation and Ar purging.



**Fig. 7.** Argon purged  $CO_{ads}$  stripping voltammograms obtained on the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst applying the potential program represented by the insert (the colour of the insert solid lines corresponds to the colour of voltammograms shown on the figure). Electrolyte: 0.5 M H<sub>2</sub>SO<sub>4</sub>; scan rate: 10 mV s<sup>-1</sup>; T = 25 °C. Marks of the peaks: oxidation of (i) - hydrogen, (ii') - weakly bonded CO, and (ii) - Mo surface species.

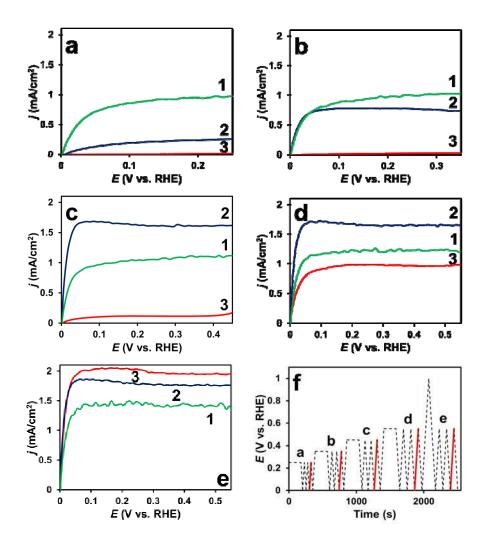
At the first measurement (after the gas bubbling process) the working electrode was set to 0 mV for 1 minute then polarized to 550 mV and back to 0 mV (curve *A* in Fig. 7). In the second the working electrode was set to 0 mV for 1 minute then polarized to 350 mV and immediately back to 0 mV. It was held again for 1 minute at 0 mV and then polarized to 550 mV and back to 0 mV (curve *B* in Fig. 7) as in the previous case. In the subsequent experiments the procedure was the same with the difference that the working electrode was

held at 350 mV for 1 or 3 minutes (curves *C* and *D* in Fig. 7). The potential program is shown on Fig. 7 as an insert.

At the first anodic scan (curve *A* in Fig. 7) we can see a sweep which has not been preceded by any other scans therefore there is no hydrogen oxidation current (i) and the so-called pre-peak consisting of oxidation of both weakly bonded CO (ii') and Mo surface species (ii) appears. The weakly bonded CO oxidation peak (ii') observed on curve *A* is larger than the following ones and starts at around 100 mV. The following scans were all recorded after some polarization of the electrode between 0 mV and 350 mV (see the applied potential program on the figure). It can be seen that the increase of the prolongation of the treatment of the catalyst before the recorded scan results in an increase of the oxidation current of the *in situ* developed hydrogen (i), disappearance of the CO oxidation related contributions (ii') and some shift of the Mo oxidation peak (ii) toward more positive potentials. In Ref. [54] it has been demonstrated that the Mo redox couple is affected by the presence or absence of CO adsorbed on Pt. After CO electrooxidation similar shift of the Mo oxidation peak toward more positive potentials was observed.

The results of the modified  $H_2$ -purged  $CO_{ads}$  stripping voltammetry measurement followed by a HOR measurement on RDE provide further details about the CO tolerance and catalytic activity of the compared catalysts (Fig. 8). Both the Mo-containing composite supported and the commercially available Pt/C and PtRu/C catalysts were tested this way.

We can see that there is no catalytic activity on both the Pt/C and the PtRu/C catalysts in HOR if the previously CO covered electrode was polarized only to 250 mV in comparison with the Mo-containing composite supported catalyst which shows significant activity (see Fig. 8a). On Fig. 8b with 350 mV upper potential limit for polarization it seems that the Pt/C is not active, the activity of PtRu/C catalyst is increased but it is still below that of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst. On Fig. 8c and 8d the PtRu/C catalyst shows more activity than the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C and the Pt/C shows improvement too. On Fig. 8e, which shows curves recorded after removing CO from the whole active surface by polarizing the working electrode to 1000 mV and thus oxidizing the CO, it seems that the most active catalyst is the Pt/C, the second one is the PtRu/C, and the third one is the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C.



**Fig. 8.** Linear potential sweeps (a-e) of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C (1 green line), PtRu/C (2 blue line) and Pt/C (3 red line) catalyst samples obtained in  $H_2$  saturated 0.5 M  $H_2$ SO<sub>4</sub> at a rotating disc electrode (at 625 revolutions/min). Before the measurements the electrodes were polarized at 0 mV for 30 minutes in CO-saturated electrolyte (CO adsorption) and exposed to the potential program depicted in figure (f) in  $H_2$ -saturated electrolyte. Sweep rate: 10 mV s<sup>-1</sup>.

# **4 Conclusions**

The peculiarities of novel 20 wt.% Pt electrocatalyst supported on Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C composite material in low-potential CO oxidation reaction were investigated. From the results of the electrochemical measurements it can be concluded that:

(i) oxidation of CO takes place at exceptionally low potential values at the  $Pt/Ti_{0.6}Mo_{0.4}O_2$ -C catalyst (the onset potential  $E_{CO,onset}$  was shifted to less positive potentials by 530 and 150 mV comparing to the Pt/C and PtRu/C, respectively);

- (ii) the so-called "pre-peak" on the voltammograms obtained during the CO<sub>ads</sub> stripping measurements, consist of at least two oxidation reaction: partial oxidation of weakly adsorbed CO and oxidation of Mo surface species;
- (iii) when the weakly bounded CO is oxidized, some hydrogen adsorption can take place on the released surface, although this amount is much smaller than in the case of a CO-free Pt surface;
- (iv)the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C catalyst loses its activity in the low-potential CO oxidation reaction when Mo becomes oxidized (above ca. 400 mV);
- (v) in the presence of CO-free platinum surface oxidized Mo can be re-reduced and the activity in the low-potential CO oxidation reaction can be restored; nevertheless, re-reduction of oxidized species formed above 400 mV is strongly hindered when adsorbed CO species are still present. Note that CO<sub>ads</sub> species can be completely removed only above 550 mV.

These results suggest that only CO adsorbed on specific Pt sites (weakly bounded CO), where Pt and Mo atoms are in atomic closeness, can be oxidised below 400 mV potential. Clear correlation between the so-called "pre-peak", the molybdenum redox phenomenon and the CO tolerance of the 20 wt% Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C system was established. Accordingly, presence of Mo species in lower oxidation state than 6+ is supposed to have crucial role in CO oxidation. Better performance of the Pt/Ti<sub>0.6</sub>Mo<sub>0.4</sub>O<sub>2</sub>-C electrocatalysts compared to the reference Pt/C and state-of-art CO-tolerant PtRu/C catalysts was also demonstrated.

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