

## On the Accuracy of Total-IBA

C.Jeynes<sup>1\*</sup>, V.V.Palitsin<sup>1</sup>, M.Kokkoris<sup>3</sup>, A.Hamilton,<sup>2</sup> G.W.Grime<sup>1</sup>

1. University of Surrey Ion Beam Centre, Guildford, England

2. Civil and Environmental Engineering, University of Strathclyde, Glasgow, Scotland

3. Department of Physics, National Technical University of Athens, Greece

### Abstract

“Total-IBA” implies the synergistic use of multiple IBA techniques. It has been claimed that Total-IBA inherits the accuracy of the most accurate IBA technique used. A specific example is now given of this where (*in vacuo*) EBS/PIXE of a glass sample uniform in depth is validated against absolutely calibrated EPMA of the same sample. The EPMA results had a mass closure gap of  $2.0 \pm 0.6$  wt%; the full PIXE analysis determined the composition of this missing 2 wt%. The PIXE calibration was against a single certified glass sample, with uncertainties per line  $\sim 10$ %. Benchmarking also demonstrates  $\sim 10\%$  underestimation of the Si scattering cross-section at proton energies  $\sim 3$  MeV. But the Total-IBA determination of the silica content had a low standard uncertainty of about 2%. This is due to the strong constraints of both the chemical prior and also the mass closure properties of the EBS. Irradiation-induced sodium migration in this soda-lime glass is explored.

**Keywords:** Ion Beam Analysis, particle-induced X-ray emission, elastic backscattering, RBS, SEM-WDX, XRF, Na migration, precision, benchmark, Rosslyn Chapel

### Introduction

Ion Beam Analysis (IBA) typically uses MeV light ion beams to probe the elemental composition of the near-surfaces of materials.<sup>1</sup> Such energetic ion beams can excite the *atoms* of the target, yielding the characteristic X-rays also excited by primary beams of electrons or photons. Photon excitation is termed X-ray fluorescence (XRF). Here we directly compare particle-induced X-ray emission (PIXE) and the electron-induced X-ray emission generated in an SEM (a scanning electron microscope), analysed in this case by a wavelength-dispersive spectrometer (WDX: PIXE usually utilises an energy-dispersive spectrometer – EDX). SEM-WDX is also often referred to as electron-probe microanalysis (EPMA) when the instrument includes multiple WDX spectrometers.

But MeV ion beams can also excite the atomic *nuclei*, either elastically giving so-called “elastic backscattering spectrometry” (EBS), or inelastically giving “nuclear reaction analysis” (NRA) which does not concern us here. A variety of NRA is particle-induced gamma-emission (PIGE): gamma spectroscopy has been central to the determination of nuclear energy levels, and remains important for IBA (especially for <sup>19</sup>F and <sup>23</sup>Na).

“Rutherford backscattering spectrometry” (RBS) is a special case of EBS where the scattering cross-section is approximated analytically by considering point charges in a screened Coulomb field. This approximation is very good for 1.5 MeV <sup>4</sup>He beams (for example) because the screening correction is quite accurately known for such beams.<sup>2</sup> The intrinsic accuracy of RBS has been exploited to demonstrate it as a primary reference method for non-destructively determining *quantity of material* in thin films at unsurpassed accuracy.<sup>3</sup> RBS has already been used in the determination of Ga photo-ionisation cross-sections at high traceable accuracy by Unterumsberger *et al.* (2018)<sup>4</sup> who also determine other XRF **Fundamental Parameters**, saying :-

XRF quantification can be absolute through proper modelling of the physics since X-ray ionization and absorption processes are well-understood: this method requires detailed knowledge of the Fundamental Parameters and has been in use since the 1980s for all the XRF methods, including EPMA and PIXE.

1 **Total-IBA**<sup>5</sup> is the synergistic use of multiple IBA methods, most usually EBS and PIXE since EBS is  
2 very sensitive to the depth of the atom in the sample but rather insensitive to atomic mass and PIXE is  
3 very sensitive to atomic number but rather insensitive to depth (although differential PIXE methods  
4 can be used to obtain useful depth information, especially for samples too thick for EBS: for a recent  
5 discussion see for example Šmit & Holc<sup>6</sup>; such methods both have relatively poor depth resolution,  
6 and also are time consuming since multiple measurements must be made at different geometries or  
7 beam energies).

8 The power of Total-IBA is that complex samples may be analysed where the PIXE and EBS signals are  
9 individually intractable (the samples cannot be solved using either PIXE or EBS alone): at present there  
10 is only one code (the DataFurnace code<sup>7</sup>) capable of handling such cases efficiently (the MultiSIMNRA  
11 code<sup>8</sup> can do Total-IBA but cannot at present handle PIXE data). It seems evident that accuracy should  
12 be conserved in a synergistic analysis, but this has not until now been demonstrated unequivocally for  
13 IBA. This work is a first attempt to assign good estimates of uncertainty to such an analysis.

14 The present work simply obtains the composition of a homogeneous glass by (*in vacuo*) Total-IBA, and  
15 is part of a wider study for conservation purposes of the historic glass from the Rosslyn Chapel.<sup>9</sup> This  
16 wider study used an external beam analysis of as-received (inhomogeneous) glass samples which will be  
17 reported elsewhere. Such an analysis is now standard where PIXE and PIGE are used (but not EBS; see  
18 for example Calligaro 2008<sup>10</sup>, and Hunault *et al.* 2017<sup>11</sup>). Analytically, the sample structure is trivially  
19 simple (uniform) but the glass is chemically complex, having 16 independently measurable elements,  
20 which can nevertheless be determined without prior assumptions by IBA (see *Analytical Protocol*).

21 After describing the experimental and analytical *Methods*, including a detailed discussion of the particle  
22 scattering cross-section data used, and after a digression on the measurement of sodium in glasses  
23 despite its well-known *Mobility* under irradiation, we give a detailed account of the *Calibration* of a  
24 Total-IBA system together with a discussion of the analytical *Protocol* for this case. Calibration is  
25 central to all accurate analysis and is usually taken for granted in “mature” techniques like IBA. But in  
26 fact there are no standard (published) protocols for accurate IBA (except the protocol local to Surrey  
27 established under ISO 17025 for the measurement of Quantity of Material by RBS [ref.3]), in contrast  
28 to the detailed standards now available for modern workhorse analytical methods (for example, those  
29 for DESI-MS<sup>12</sup>, SIMS<sup>13</sup> and XPS<sup>14</sup> established under the aegis of VAMAS). The *Results* are presented  
30 in some detail, since for this very simple sample it is only the details that are interesting. In particular  
31 we present new *Benchmarking* results that significantly affect the interpretation of the EBS. We can  
32 then properly present the estimates of *Uncertainty*, and comment on the properties of *Mass Closure*  
33 that yield greatly reduced uncertainty. The *Discussion* leads to extended *Conclusions* which expand  
34 significantly on the summary in the Abstract.

## 43 *Experimental and Analytical Details*

44 The sample analysed was a piece of glass about 3 mm thick from the Rosslyn Chapel. It was cross-  
45 sectioned at Glasgow University (embedded in a circular resin block and polished) and analysed by  
46 **EPMA** using a JEOL JXA-8530F and its proprietary software<sup>15</sup>. The sample was about 3×5 mm<sup>2</sup>, and  
47 was analysed using 20 keV in 10 spots distributed along the sample centre line, with an electron current  
48 of 20 nA (¼ mA/cm<sup>2</sup> using a spot size defocussed to 100 microns) verified to avoid Na migration within  
49 the 60 s measurement time. The O and S signals were not measured, and the WDX detectors were  
50 calibrated against Albite (Na<sub>2</sub>O), Wollastonite (SiO<sub>2</sub> & CaO), Corundum (Al<sub>2</sub>O<sub>3</sub>), Orthoclase (K<sub>2</sub>O),  
51 and Fe (Fe<sub>2</sub>O<sub>3</sub>). 60 s peak measurements were made, with two corresponding 30 s background  
52 measurements each side of the peak.

53 Quantification was carried out using the instrument’s built-in software, including the ZAF correction  
54 algorithm (that is, the correction for atomic number, self-absorption, self-fluorescence). The ZAF  
55

correction was obtained explicitly per line for each EPMA analysis: the “F” (fluorescence) correction is negligible in the Rosslyn glass analysis for all lines except Si (¼ %) and Na (<½ %). The PIXE code GUPIX<sup>16</sup> (which does not implement the general case of layered samples) of course implements the fluorescence correction; but the Total-IBA code DataFurnace<sup>17</sup> does not implement it, since it is non-trivial to do it correctly for the general case and anyway it is usually very small (as in this case).

Total-IBA with a proton beam of various energies around 3 MeV was done at Surrey University on the cross-sectioned sample using a standard microbeam vacuum chamber<sup>18</sup> but a redesigned beamline. The beamline components are mounted on a continuous length of a high stiffness aluminium extrusion supported on synthetic granite blocks. The object apertures are formed by Electrical Discharge Machining cut-outs in the polished edges of 150 µm high purity tantalum foils. The object apertures are rectangular with aspect ratio 4:1 matching the XY demagnification factors of the focussing system: sizes range from (2×½) mm down to (80×20) µm. Circular collimator apertures were made in tantalum foil by precision micromachining and the edges cleaned using a high current FIB (focussed ion beam) technique. Motorised ladders accept 10 apertures for easy size selection. Thermal loads are reduced by water-cooled motorised beam dump slits before the apertures. A Wien filter<sup>19</sup> is installed in the object space of the beamline to reduce the aperture’s edge-scattering halo.

A new nuclear microprobe system uses a spaced triplet configuration of Oxford Microbeams OM-50 quadrupoles in which the spacing of the first two lenses can be easily adjusted. This was designed using WinTRAX simulations<sup>20</sup> and enables routine submicron spatial resolution for 2.5 MeV protons and beam currents ~1 nA. The separation of the first two quadrupoles in the Oxford triplet results in increased demagnification and optimum acceptance. After an extensive assessment a single spaced Oxford triplet configuration was chosen, that is, the first and the second lenses are placed apart with spacing of one quadrupole length with calculated XY demagnifications of 87×25.

A PIXE detector (in the IBM geometry) and a particle detector (in the Cornell geometry) are both at backward angles to the beam of respectively 135°, 155°; with solid angles respectively 18, 41.4 msr. The PIXE detector is a Si(Li), with a 12.5 µm Duraberyllium entrance window and a (nominally) 125 µm pure Be filter to stop scattered protons. The beam is focussed to ~2 µm diameter and scanned over a defined area (here usually ½ mm square). A lead-glass certified standard (BCR126A<sup>21</sup>) was used to obtain the X-ray detector effective solid angle (from the Pb L $\alpha$  line) and the filter thickness (from the Si K $\alpha$  line).

OMDAQ		Sample	energy keV	time s	scan (µm)		charge nC	current (pA)		flux pA/µm <sup>2</sup>
#	date				X	Y		average	measured	
989009	14 Dec	BCR glass	2734	226	250	250	62	274	300	0.004
989010	14 Dec	BCR glass	3066	416	75	75	52	125	150	0.022
989012	14 Dec	Rosslyn glass	2827	1808	75	75	1696	938	800	0.167
989013	14 Dec	Rosslyn glass	2827	1810	5	5	1419	784		31.359
989014	14 Dec	Rosslyn glass	2827	1784	2	2	1271	712		178.111
989020	15 Dec	Rosslyn glass	3080	7211	500	2	115	16		0.016

**Table 1: Measurement conditions**

Data collected (in vacuum) 14, 15 December 2018. Nominal beam incidence normal on samples. 2 detectors (PIXE & EBS) respectively with solid angles 18, 41.4 msr, and at angles to the beam 135°, 155°.

The accelerator is a 2 MV HVEE Tandatron, with generating voltmeter (GVM) energy stabilisation; a GVM factor calibrating the nominal terminal voltage of **1.00011(4)** is here measured directly using the very well-established 2663±7 keV resonance in the <sup>16</sup>O(p,p)<sup>16</sup>O cross-section function (with the method of Colaux *et al.*<sup>22</sup> see **Figure 3b** below). This resonance energy is accurately known, with an uncertainty given in the compilations.<sup>23</sup> It has long been of great interest<sup>24</sup> and was accurately determined over 50 years ago.<sup>25</sup>

OMDAQ-3<sup>26</sup> (16, 16X; version 3.2.3.884, 20<sup>th</sup> Nov 2018) was used both to collect the data and to interpret the datasets. DataFurnace<sup>27</sup> (NDFv10.0 rev.28/2/2019 and WiNDFv9.3.98) was used to fit EBS/PIXE data. GUPIX [ref.16] was used to extract the characteristic X-ray peak areas (from the PIXE spectra) required as input to DataFurnace. GUPIX corrects these peak areas for pileup and line overlaps (including escape peaks). The use of NDF for PIXE was described in Pascual-Izarra *et al.*<sup>28</sup> PIGE is supported by NDF but is not needed here.<sup>29</sup> SRIM 2003 stopping powers<sup>30</sup> were used (later versions of SRIM have not changed the values noticeably). Pulse pileup was calculated for the particle detector using the algorithm of Wielopolsky & Gardner.<sup>31</sup>

**Table 2: Proton scattering cross-section functions used**

Nucleus	Rutherford <sup>a</sup>	SigmaCalc <sup>b</sup>		Measurements	
	4% limit, keV	Energy, keV	Date	Energy, keV	Reference
<b>He</b>	None	500 – 6000	6/11/2015		
<b><sup>6</sup>Li</b>	600			1250 – 5530	Fasoli, 1964 <sup>32</sup>
<b><sup>7</sup>Li</b>	407			1340 – 2750 2480 – 12000	Malmberg, 1956 <sup>33</sup> Gleyvod, 1965 <sup>34</sup>
<b>C</b>	<360	360 – 7000	7/1/2015		
<b>O</b>	<100	100 – 4100	12/8/2001		
<b>Na</b>	573	570 – 1480	12/11/2013	2212 – 5200 1480 – 3500	Caciolli, 2008 <sup>35</sup> Vanhoy, 1987 <sup>36</sup>
<b>Mg</b>	700	700 – 2700	28/5/2013	2710 – 4200 <sup>c</sup>	Valter, 1963 <sup>37</sup>
<b>Al</b>	936	900 – 1820	12/11/2013 <sup>d</sup>	780 – 3030 2430 – 5070	Chiari, 2001 <sup>38</sup> Ziketić, 2007 <sup>39</sup>
<b><sup>28</sup>Si</b>	1272	1000 – 3500	15/2/2017		
<b>S</b>	1800	1400 – 4000	7/11/2015		
<b>K</b>	<1731	1700 – 2500	12/11/2013	1730 – 3230 3000 – 5000	de Meijer, 1970 <sup>40</sup> Kokkoris, 2009 <sup>41</sup>
<b>Ca</b>	2159	1700 – 2900	6/6/2013	2350 – 4830 <sup>e</sup>	Koltay, 1975 <sup>42</sup>
<b>Fe</b>	2050	2010-3300	13/5/2010		

**Notes:**

(a) Up to the given energy the cross-sections are within 4 % of Rutherford. See Fig.9 of Jeynes & Colaun, *Analyst* 2016 [ref.1]. Note that comparable data for 1 % variation are not available.

(b) *Evaluated* cross-sections are most reliable where they are available. See references in the text.

(c) Wang<sup>43</sup> is at 150° where Valter is at 140°, but Valter matches SigmaCalc well & includes the resonance at 3065 keV

(d) The evaluation reveals many very fine (unobservable) resonances, so Chiari is used instead up to 3030 keV

(e) Koltay data does not cover 2.9-3.5 MeV, where the cross-section is assumed constant (at ~1.2\*Rutherford)

*Robustified*<sup>44</sup> chi-squared (“*pseudo- $\chi^2$* ”) fitting is used to fit PIXE line areas self-consistently with EBS spectra. This is needed because a) the chi-squared function is mathematically rather poorly behaved for poor fits, and b) PIXE data is introduced into NDF as line areas, which have a different statistical behaviour from spectral data and for which the chi-squared minimisation procedure is not strictly valid. In this work extensive use of “logical elements” (SiO<sub>2</sub> and other oxides) was made to impose chemical constraints<sup>45</sup> on the fitting.

This work analyses 6 datasets (##09, 10, 12, 13, 14, 20) whose analytical conditions are given in Table 1. The particle detector is large (100 mm<sup>2</sup>) and subtends a large angle at the sample (about 14°); the PIXE detector is somewhat smaller (80 mm<sup>2</sup>), but is moveable and subtends about 25° when fully

1 inserted. However, the effective takeoff angle from the sample (which for these large detectors is not  
2 the same as the average takeoff angle) is assessed experimentally. We do not do the large angle  
3 correction: in principle the reaction cross-sections should be integrated over all takeoff angles to  
4 obtain the correct effective cross-section, but we simply take an average that fits the dataset. The  
5 inverse situation (where the irradiation area is not small) has been treated for PIXE/XRF (for the  
6 APXS instrument of the Mars Rover) by Campbell *et al.*, 2019<sup>46</sup>.

## 7 Scattering Cross-Sections

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9 In the absence of evidence to the contrary, theoretical screened<sup>47</sup> RBS cross-sections were used in  
10 this work for Pb, Ba, As, Mn. *Evaluated* EBS scattering cross-sections are used where they are  
11 available: these fit all available experimental data to a nuclear model. The **SigmaCalc**<sup>48</sup> (v.2.0<sup>49</sup>)  
12 code is publicly available, and can generate cross-section functions for any given scattering angle.  
13 Some of the important reactions have been published separately: these include the C<sup>50,51</sup>, O<sup>52</sup>, Mg<sup>53</sup>,  
14 Al<sup>54</sup>, and Si<sup>55</sup> reactions (for proton scattering, see **Table 2**); and the C<sup>56</sup>, O [ref.22], and Si<sup>57</sup> reactions  
15 (for alpha scattering). Where evaluated data are not available, measured data are taken from the  
16 collection in *IBANDL*<sup>58,59</sup>. Of course, 3 MeV proton scattering on He, Li, Na, Mg, K, Ca, Fe are all  
17 also strongly non-Rutherford, and appropriate cross-sections are used (see **Table 2**).  
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20  
21 RBS is known to be a poor approximation for Fe (which has some two dozen strong and sharp  
22 resonances for this energy, see *SigmaCalc* and Lindstrom *et al*, 1971<sup>60</sup>) and hence presumably Mn as  
23 well (there are measured nuclear reactions at 2 MeV, see *IBANDL* and Kenny *et al*, 1980<sup>61</sup>). EBS  
24 cross-sections are used for Fe but not Mn, since the Fe signal is noticeable in EBS but the Mn signal  
25 is not (in any case there are no EBS data in *IBANDL* for Mn).  
26

27  
28 There are also no EBS cross-section data in *IBANDL* for As, Ba or Pb, but Zn has an EBS resonance  
29 just below 3 MeV (Bogdanović *et al*, 1995<sup>62</sup>) and both isotopes of Ag have substantial gamma yields  
30 below 2.5 MeV and measurable yields down to 1.5 MeV and below (Deconninck & Demortier,  
31 1975<sup>63</sup>). The same authors also measured substantial gamma yields for Au at 2.5 MeV. At present  
32 the RBS/EBS boundary for a proton beam is poorly defined for Z > 26 (see Fig.9 of the 2016 Review  
33 [ref.1]). Nevertheless, we assume RBS for the heavier elements (Mn, As, Sr, Ba, Pb), that is, we  
34 assume that Rutherford cross-sections are good approximations.  
35  
36

## 37 Precision/Accuracy

38  
39 This work is concerned with a careful determination of uncertainty associated with Total-IBA: for  
40 convenience, considering the confusion surrounding terms relating to uncertainty, we will give  
41 definitions of “uncertainty” and related terms following the VIM<sup>64</sup>. Correct methods of treating  
42 “uncertainty” is explained authoritatively by the GUM<sup>65</sup>.  
43

44 It is essential to start by noting that the VIM (§2.9) insists that to be a valid *measurement*, its  
45 uncertainty must also be reasonably estimated; expanding on this by saying that an “*Uncertainty*  
46 *Budget*” (§2.33) is a “*statement of: a measurement uncertainty, of the components of that*  
47 *measurement uncertainty, and of their calculation and combination*” which “*should include the*  
48 *measurement model, and the measurement uncertainties associated with the quantities in the*  
49 *measurement model, covariances, type of applied probability density functions, degrees of freedom,*  
50 *type of evaluation of measurement uncertainty, and any coverage factor*”. The underlined  
51 expressions are relevant here and are discussed in detail by both the VIM and the GUM. They are  
52 very general statements applicable to all types of measurement: our case is very simple numerically  
53 so that (for example) we will not need to consider “covariances” explicitly, and we will always be  
54 able to express *standard uncertainties* simply as the standard deviation of the expected distribution of  
55 *measurements*.  
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We also note here that the GUM and the VIM explicitly avoid any consideration of the “true value of a measurement” on the grounds that this is unknowable in principle. The use of “error” (instead of “uncertainty”) is also deprecated (except for convenience in “error bars”), unless a mistake is to be indicated.

The **accuracy** of a measurement is defined by the VIM (§2.13) as the “closeness of agreement between a measured quantity value and a true quantity value of a measurand”, commenting that it is “NOTE 3: sometimes understood as closeness of agreement between measured quantity values that are being attributed to the measurand”; although insisting that, strictly speaking, “accuracy” is a qualitative judgment, not a quantitative measure: “NOTE 1: The concept ‘measurement accuracy’ is not a quantity and is not given a numerical quantity value”.

Here we treat the **accuracy** of a measurement as being given by the **standard combined uncertainty** (VIM §2.31) of the measurement as evaluated by the **uncertainty budget** (that is, considering *all* known sources of uncertainty). The proper use of the Uncertainty Budget in IBA was powerfully argued for nearly a generation ago by the (then) IRMM group at Geel.<sup>66</sup> In the case of the EPMA measurements, their **accuracy** is assessed from their **reproducibility** (VIM §2.25). Strictly, the VIM regards this measure as a sort of *precision*, but an evaluation of the reliability of the EPMA measurements gives us confidence that in this case it is a reasonable proxy for accuracy.

The **precision** of a measurement is defined by the VIM (§2.15) as the “closeness of agreement between indications or measured quantity values obtained by replicate measurements on the same or similar objects under specified conditions”. It is essentially a measure of the **repeatability** (VIM §2.20) of the measurements, that is, not considering some possible sources of uncertainty. Note that the accuracy cannot be better (smaller) than the precision, but usually the precision is smaller than the accuracy (unless counting statistics dominate the uncertainty, in which case the precision is essentially the same as the accuracy). The term “**standard uncertainty**” usually refers to the *precision*, that is, the standard deviation of the observed distribution of measurements in a particular case.

Uncertainties may be expressed as *absolute* or *relative*: for example in Table 4 the Na<sub>2</sub>O content of the glass is given as 12.43±0.21 wt% (absolute) or 12.43 wt% ±1.7% (relative).

The GUM says (§6.1.2) “Although [the standard combined uncertainty] can be universally used to express the uncertainty of a measurement result, in some ... applications ... it is often necessary to give a measure of uncertainty that defines an interval about the measurement result that may be expected to encompass a large fraction of the distribution of values that could reasonably be attributed to the measurand.” Hence the idea of “**expanded uncertainty**” (GUM §6.2), where the “**coverage factor k**” (GUM §6.3) specifies the level of confidence in the specified uncertainty. In our simple cases, where the distribution of measurements is expected to be essentially normal, we are used to specifying a 95% confidence interval as “2 *sigma*” and a 99% confidence interval as “3 *sigma*”. In the more general terminology of the GUM these would be expressed as “k=2” or “k=3”. In GUM terms, “k=1” would refer simply to the *standard uncertainty*. But *k* is used only in the context of an *expanded uncertainty*.

## **Sodium Migration during Analysis**

In the wider study for which this study is preliminary, glass weathering effects such as alkali metal leaching are expected to be prominent, and will affect the Na profile near the surface. Therefore, to measure such profiles it is necessary to establish the analytical validity of Na measurements for these samples, since it is well known that Na is mobile under a charged particle beam. Although IBA is not deliberately destructive, of course energetic beams can damage or modify samples in various ways,<sup>67</sup> of which this is only one.

Walker & Howitt<sup>68</sup> find that Na-depletion is a function of electron current but not of electron energy (they used 10, 15, 20 keV on the SEM), observing Na depletion to 70% of the initial value after {100, 150, 250, 500} secs for electron current densities of {4.8, 1.2, 0.7, 0.3} mA/cm<sup>2</sup>.

For the present work the Na content in the glass is determined by EPMA since there is effectively no signal for Na in PIXE (although in the wider study Na will be measured by H-PIXE, He-PIXE and PIGE).

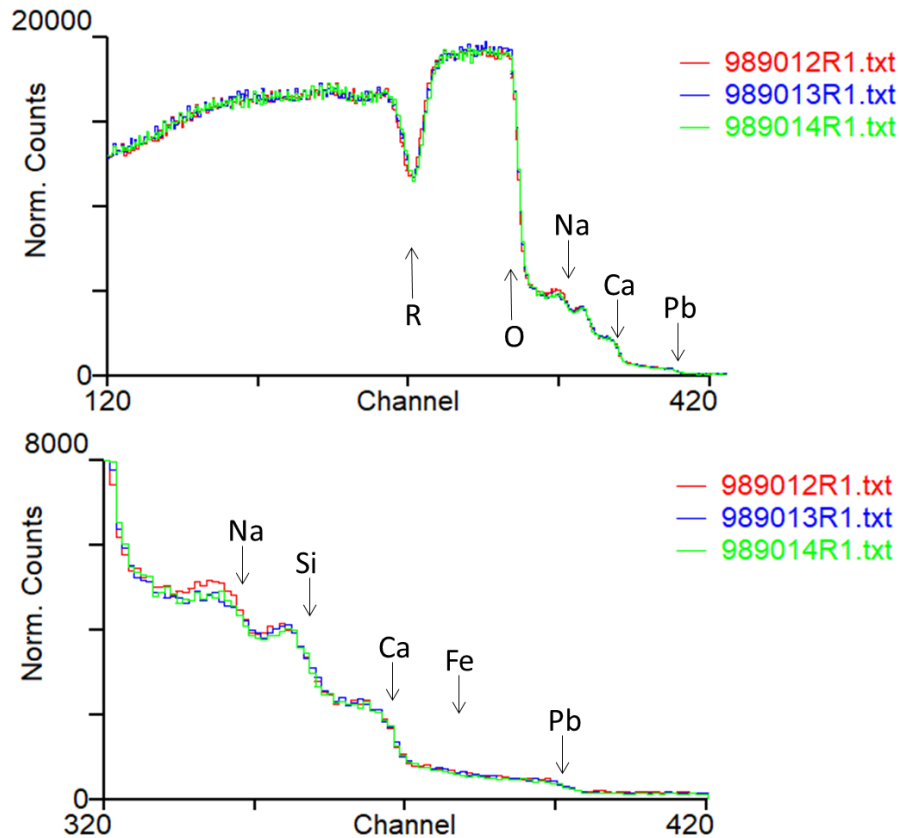


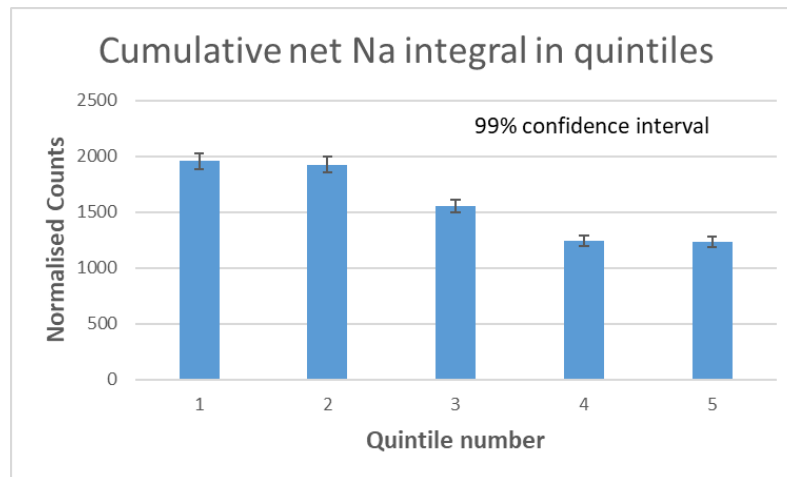
Figure 1: Normalised EBS spectra of Rosslyn glass with increasing current density

OMDAQ run numbers 989012, 13, 14. 2827 keV. The 2663 keV resonance in the O signal is marked “R”. Elemental edges are marked *Above*: full spectrum; *Below*: detail. The signal above the Pb edge is pileup. The current densities for runs [##12](#), [13](#), [14](#) are [respectively](#) about [0.02](#), [3](#), [18](#) mA/cm<sup>2</sup> (see [Table 1](#)).

We have confirmed Walker & Howitt’s result using beams of energies {10, 20, 30 keV}, beam current densities ranging from 0.3 mA/cm<sup>2</sup> to over 300 mA/cm<sup>2</sup>, and spot sizes from focussed (2 μm) to 100 μm. Na *instability* in this glass under a 20 keV electron beam was measured by us over 60 secs at beam current densities 2 mA/cm<sup>2</sup> (40 nA with 50 μm beam diameter). The present results were obtained with lower current densities of 0.25 mA/cm<sup>2</sup> (20 nA with defocussed beam 100 μm diameter), for which stability to the electron beam is obtained over time scales long enough to make valid measurements.

However, electrons and ions are not the same: Battaglin *et al.*<sup>69</sup> investigated Na mobility during Ar<sup>+</sup> implantation and found that Na-depletion was simply a function of the energy deposited in the volume. But the nuclear energy loss behaviour of heavy ions is entirely different from that of protons: the effect of 600 keV proton irradiation was also explored in a separate [broad-beam](#) study by Battaglin *et al.*<sup>70</sup> who found a “residence time” effect, although at 140 μA/cm<sup>2</sup> the residence time (see their Fig.2) is near zero. But they also say that the residence time effect depends on beam *current* more than the current *density*, [citing measurements with a spot size a factor 2 smaller](#); they use a uniformly

irradiated 4 mm<sup>2</sup> area (this is apparently for both spot sizes). Ours is a focussed proton beam orders of magnitude smaller (with an average current density an order of magnitude smaller), so that we may expect the residence times to become significant.



**Figure 2: Na mobility in Rosslyn Glass under a proton beam**

Background corrected normalised counts from #989013 are shown, with error bars from counting statistics

On the other hand, March & Rauch<sup>71</sup> investigate Na depletion in leached soda-lime glasses using the 591 keV resonance of the <sup>23</sup>Na(p,α)<sup>20</sup>Ne reaction (see Carnera *et al.*, 1977<sup>72</sup>: note that Battaglin *et al.* use the same NRA reaction; Carnera *et al.*'s cross-section data are not included in *IBANDL*) with a proton beam of 7-10 nA scanned over 3×3 mm<sup>2</sup> and a total dose of about 5.10<sup>13</sup> protons (about 10 μC, or about 15 minutes irradiation). They believe that Na mobility under these conditions “should be small”.

This conclusion is supported by measurements of Mosbah & Duraud<sup>73</sup>, who investigate two synthetic glasses (one of which is similar both to ours and also to the “Brill-B” glass reported by Vicenzi *et al.*<sup>74</sup>) using a 2 MeV proton beam with high fluxes from 2 to 23 mA/cm<sup>2</sup> to damage the glasses. Our fluxes are orders of magnitude smaller (see Fig.1). This Brill-B glass has been used as a standard for “decades” (T.Calligaro, private communication) without seeing any Na loss.

Finally, we mention in passing Melcher *et al.* (2010)<sup>75</sup> for a recent account of glass weathering as revealed by SIMS, AFM and SEM; although it does not directly address the problem of Na mobility under an analytical beam they clearly believe that they have made valid measurements.

**Figure 1** shows that Na mobility effects can be observed in EBS measurements of this glass. There is a noticeable Na depletion for measurements at high flux density. For enhanced sensitivity, these EBS measurements are made at an energy where the (non-Rutherford) scattering cross-section ratio for Na/Si is near a maximum. However, the thickness of the modified layer cannot be determined from these data, except to say that it must be more than about ½ μm.

**Figure 2** shows quintiles from the list mode file #989013, which was collected over 30 minutes with a high beam current of about 300 pA/μm<sup>2</sup>; that is, segments integrated over 6 minutes. It is very clear that there is systematic variation between the start and end of data collection: not only does the Na signal decrease significantly but the Si signal also increases slightly (as it must due to mass closure). In confirmation of this, the variation of the integrals is much more than expected simply from counting statistics for the Na signal, and also slightly more than expected for the Si signal.

Figure 2 also shows that there appears to be a ten minute “incubation time” before the Na becomes mobile, even at the high average beam current used for #989013 (which is some 200 times greater than for #989012). The fact that the first two quintiles of the #989013 spectrum are indistinguishable

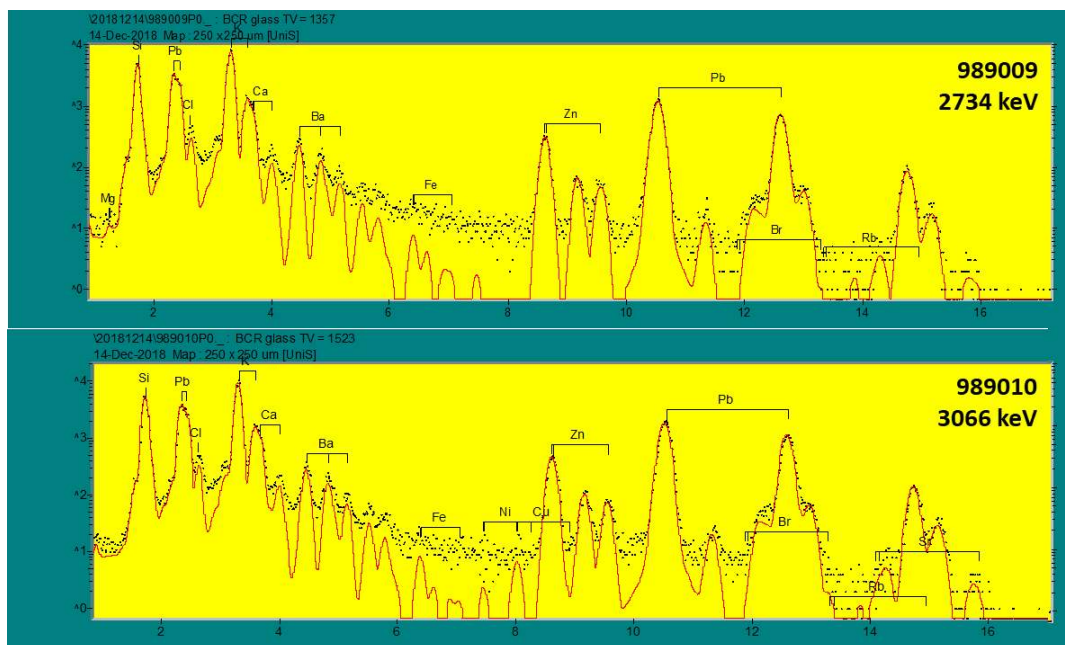


either from each other or from the #989012 spectrum is evidence that Na migration is not a process that starts very rapidly, unlike H loss for example.

We conclude that we can measure reliably (that is, without Na migration) for a suitably low flux proton beam. In any case this present work is not sensitive to the Na content, which we do not here try to measure by IBA.

## Calibration

The calibration of the EPMA is automated under proprietary software, and details are not normally available to the analyst. But the calibration is absolute per line, so that a recursive analysis first obtains the interaction volume from the thick target composition deduced from *assuming* mass closure and full oxidation; and then effectively gives the measured elements as an areal density converted to a mass fraction per measured element from the interaction volume, *without* normalising the results to 100 % mass closure. Therefore the analysis is valid provided that mass closure is approximately obtained. It is worth noting that XRF (but not EPMA or PIXE) can determine the average Z of the material from the primary photon scattering signals, allowing the hydration state of the sample also to be inferred.<sup>76</sup> This is *Discussed* below.



**Figure 3a: PIXE spectra from BCR glass with GUPIX fits**

The ordinate has a logarithmic scale ranging from  $10^0$ - $10^4$ . The GUPIX fit does not include the background

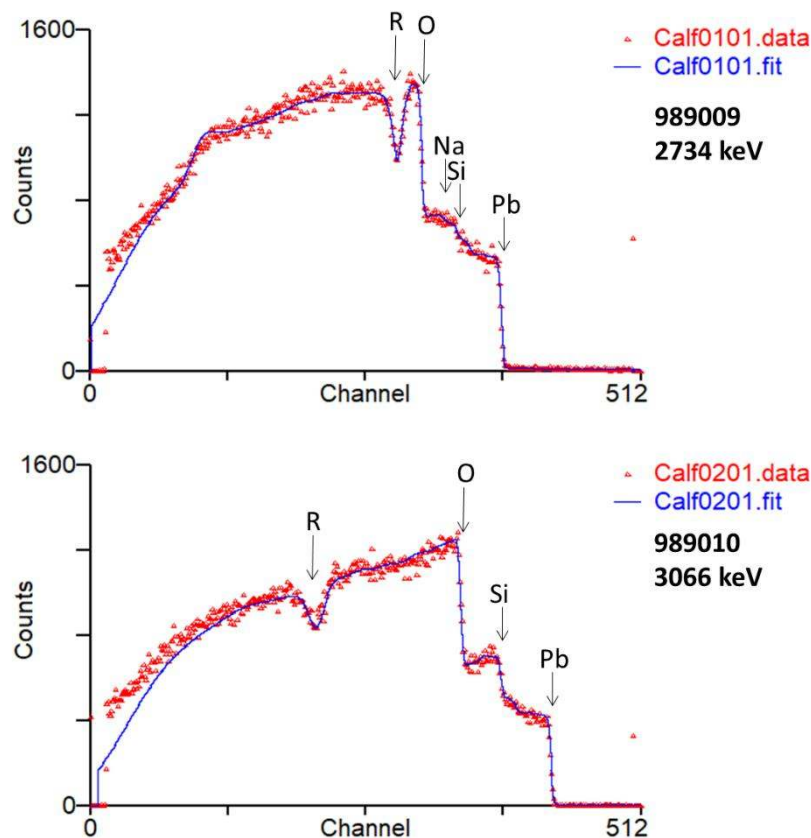
IBA experimental parameters were determined using the BCR126A [ref.21] certified standard (a lead glass with elemental content about 60 at% O, 26 at% Si, 6 at% K, 3 at% Pb & Na and 0.9 at% Li); these fitted parameters include sample tilt and the effective detector solid angles, the filter thickness and efficiency curve for the PIXE detector, and the accelerator energy and EBS spectrometer calibrations. This method of energy calibration using nuclear resonances in EBS [ref.22] is sensitive to small sample-charging effects: much larger effects can exist (without electrostatic instability) which are observable directly with RBS.<sup>77</sup> The angles of the detectors to the beam are determined separately, but confirmed by this calibration.

**Figure 3a** shows the PIXE data with the GUPIX fit, where GUPIX is used as a convenient way of identifying characteristic lines as well as obtaining valid line areas even where there are line overlaps (it also fits pileup and escape peaks). The Cl signal is assumed to be from surface contamination and is

ignored in the NDF simulation. GUPIX allows the detector response function to be individually fitted per detector,<sup>78</sup> but here it is defaulted (which has only a marginal effect on the extracted line areas).

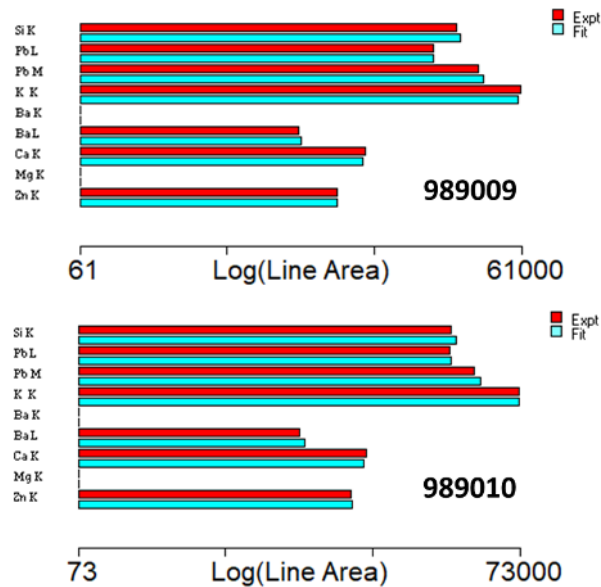
**Figure 3b** shows the EBS data with the NDF fit. From the Pb and O edges of these spectra the spectrometer gain may be obtained, and the incident beam energy (and hence the GVM factor) is obtained from the position of the 2663 keV resonance in the O signal, which for the 3066 keV measurement is very sensitive to pathlength thus giving a sample tilt of 5° away from the particle detector.

Note that the parameters are fitted but the spectrum is *simulated* from the certified glass composition, using appropriate scattering cross-section functions as given in **Table 2**. The mismatch in the EBS spectrum for the Si signal due to the underestimated Si scattering cross-section (discussed below) is less noticeable for this EBS spectrum because of the high background of the Pb signal. The charge·solid-angle product for the particle detector is given by the total yield of the spectra. The gain is assumed linear (that is, ignoring the particle detector pulse height defect,<sup>79</sup> since spectra collected with proton beams of similar energies have negligible non-linear effects).



**Figure 3b: EBS spectra from BCR glass with NDF simulations**

Sample tilted 5° away from particle detector. Elemental edges and the 2663 keV resonance in the O signal (“R”) are labelled



**Figure 3c: PIXE data and NDF simulation of BCR glass**

charge·solid-angle product from Pb L line and detector filter thickness (151  $\mu\text{m}$  Be) from Si K line. Sample tilted 5° towards X-ray detector. Numerical data in Table 3.

Line	#909009: 2734 keV			#909010: 3066 keV			Hfactor
	Data	Sim	data/sim	Data	Sim	data/sim	
Si K	22015	23392.1	0.94	25113	27134.1	0.93	0.9
Pb L	15445	15513.2	1.00	24411	24890.4	0.98	1.0
Pb M	31099	33616.8	0.93	35585	40021.5	0.89	1.2
K K	60455	58651.7	1.03	71844	72996.3	0.98	1.2
Ba L	1878	1945.6	0.97	2355	2568.9	0.92	1.0
Ca K	5356	5089.8	1.05	6630	6421.5	1.03	1.2
Zn K	3414	3454.1	0.99	5218	5384.5	0.97	1.2

**Table 3: PIXE yield (counts per line) for BCR glass**

These data shown are graphically in **Figure 3c**. Line areas simulated by NDF from certified glass composition. Sample tilted 5° towards X-ray detector: gives correct results for Ca and K lines.

**Table 3** shows the numerical values obtained (using the given “H-factor”). Here the high energy major line (Pb L $\alpha$ ) determines the charge·solid-angle product; the low energy major line (Si K) largely determines the effective detector filter thickness of 151  $\mu\text{m}$  Be (which includes the nominal 12.5  $\mu\text{m}$  Duraberyllium detector window), which is about 10  $\mu\text{m}$  (of Be equivalent) thicker than expected probably mostly due to icing. The “H factor” simply multiplies the simulated line counts by the given value.

Given this calibration, **Figure 3c** shows the NDF simulation of the PIXE line areas for the BCR glass standard. Note that this H-factor calibration for PIXE only has a precision of 10% deriving as it does from a single calibration sample.

The calibration curve (labelled “H-factor” in **Table 3**) is well-behaved, being uniformly 1.2 for the K (and M) lines (but reduced to 0.9 for the lowest energy), and 1.0 for the L lines. Notice that the change of energy between #989009 and #989010 has a noticeable effect on the spectra, which are simulated faithfully (usually better than 2%). K and Ca are sensitive to pathlength to the detector, and the 5° tilt inferred from the EBS spectrum is confirmed independently by the PIXE spectra.

## Analytical Protocol

1 Having used the certified standard (the BCR126A lead-glass sample) to calibrate the spectrometry  
2 system, we use the calibrated system to measure the unknown sample (see **Figures 4, 5**). Note that  
3 the beam is assumed normal to this sample: the tilt angle depends on the sample mounting and is not  
4 necessarily the same as for the calibration sample.  
5

6 In the general case of an inhomogeneous sample, the PIXE data can only be interpreted using the  
7 depth profile obtained from the EBS spectra. Even though in this present simpler case the sample is  
8 homogeneous with depth, we still use a general method; partly (i) because in the simple case  
9 important aspects of the accuracy of the general case become tractable and can be evaluated in detail,  
10 partly (ii) because the PIXE (**Figure 4**) here has no sensitivity to Na (a matrix element: it also has an  
11 unreliable signal for Al, a minor element undetectable in EBS), and partly (iii) because it is easy to  
12 implement chemical priors in DataFurnace.  
13  
14

15 The EBS spectrum (**Figure 5a**) is analysed as a glass, with the constituents grouped to match the major  
16 information in the spectrum, and treated as oxides: that is, applying *prior chemical constraints* to the  
17 data (see the examples and discussion in Jeynes *et al*, 2000 [ref.45], and §8.3 of the 2003 *Topical*  
18 *Review*<sup>80</sup>). In Figure 5a there are four steps in the spectrum: at the O, Si, Ca and Pb edges. Therefore  
19 three “logical elements” (**LE**, see Barradas & Jeynes, 2008 [ref.17]) are used, where “?” indicates that  
20 the LE composition may vary during the fit, and where in all cases the (unnormalised) molar  
21 composition is indicated:  
22  
23

24 LE1: SiO<sub>2</sub> 747 Al<sub>2</sub>O<sub>3</sub> 6.8 Na<sub>2</sub>O 128  
25 LE2: CaO 962 K<sub>2</sub>O ?=22 SO<sub>2</sub> ?=44  
26 LE3: Fe<sub>2</sub>O<sub>3</sub> 660 TiO<sub>2</sub> ?=33 MnO ?=79 CuO ?=2 ZnO ?=7 PbO ?=62 BaO ?=73 AsO ?=80 SrO ?=6.5  
27

28 In this case the Si:Al:Na ratios are obtained from EPMA: this is another prior chemical constraint for  
29 the IBA. The Al is undetectable in the EBS spectrum being far below the detection limit: it is also  
30 unmeasurable in the PIXE since the Al signal background is both high and not well-modelled. The Na  
31 is measurable in EBS but the sensitivity is relatively poor because of the high background signal. There  
32 is no signal in this case for Na in the PIXE data. Had EPMA not been available to determine the  
33 Si:Al:Na ratios we would have used EBS, in which case LE1 would have been split into two logical  
34 elements (ignoring the Al): SiO<sub>2</sub> and Na<sub>2</sub>O. The uncertainties would then also have increased.  
35  
36

37 LE1 is a fixed mixture of the oxides of Si, Al, Na (using the prior information from EPMA), but LE2 is  
38 a variable mixture of the oxides of Ca, K, S, and LE3 is a (variable) mixture of 9 oxides. The  
39 DataFurnace code fits the spectrum first by obtaining “best” ratios of logical elements 1,2,3 (given their  
40 starting composition), then adjusts the composition of the variable LEs, and then iterates. The fitting  
41 can be either a local minimisation (using a standard grid search) or a global minimisation (using  
42 simulated annealing: see Barradas *et al*, 1997 [ref.27]). The objective function being minimised is in  
43 this case a pseudo- $\chi^2$  function obtained from both the EBS spectrum and the PIXE line areas.  
44  
45

46 It is very easy and quick using the DataFurnace *simulator* to obtain approximate values for the ratios  
47 of all the logical elements, since it simultaneously calculates both the EBS spectrum and the PIXE  
48 line areas, with both displayed together. But obtaining a high quality *fit* is nearly impossible with  
49 manual methods because all the four main EBS signals are quite strong functions of all the others.  
50  
51

52 In this case the PIXE determines the compositions of the two variable LEs almost independently of  
53 the EBS, but the Si:Ca ratio is well-determined both by EBS and by PIXE (EBS and PIXE are here  
54 independent measurements which will agree only if the calibrations are correct). In this case, where  
55 the sample is homogeneous with depth, the PIXE is independent of the EBS (otherwise it would not  
56 be). The EBS always gives information about the surface composition nearly independent of PIXE  
57  
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63  
64  
65

since the step height is well defined (although the interpretation of the step background may depend quite strongly on the PIXE, as here).

In any case, the step height for the EBS O signal is always given by the cation content, that is, this signal in the EBS spectrum is not independently determined even though it is the dominant signal. This is one major difference between RBS and EBS: in RBS the O signal is usually very small (because the RBS cross-section goes with  $Z^2$ ), even where the O content dominates. But for EBS the backscattering cross-section for O is greatly enhanced (by a factor  $\sim 8$ , see **Table 8a** below) so that the O signal is large even for a lead-glass (see **Figure 3b**). This means that where samples have no elements lighter than O, the EBS spectra give 100% mass closure: for such samples this is a powerful extra constraint on the solution.

It is essential to recognise that, given the charge·solid-angle product (a well defined experimental value for the spectrum) and the scattering cross-section functions (well defined constants of nature), the number of counts in the EBS spectrum is a strong function of the average atomic number ( $Z$ ) of the sample (well-determined where the sample is homogeneous). Note that the scattered photon signal in XRF is also a function of  $Z$  ([ref.76]).

## Results

**Table 4** shows the results of the EPMA analysis, where the (average) wt% results in the Table have a mass closure of  $98.0 \pm 0.6\%$  (hence the “Missing” component of 2 wt%), and where the given uncertainties are the relative standard deviations of the 10 repeat measurements (of course the software assumes standard glass oxidation). Note that the “Missing” uncertainty is dominated by the uncertainty of the main lattice-former element (Si).

The missing elements are supplied by the PIXE analysis (see below) from which we obtain the average atomic weight (43.35 amu) of the component missing in EPMA and hence the mol% column in **Table 4**. The mol% uncertainty is derived from the measured wt% uncertainty. The present PIXE analysis is insensitive to Na and Al, so we fit the IBA data using Na/Si and Al/Si ratios derived from EPMA. Table 4 also gives calibration data for the six lines measured.

Rosslyn glass sample					Calibration from standards		
Line	wt%	±	mol%	±	Line	nA	cps
Na2O	12.43	1.7%	12.79	1.7%	Na	20.0	3518
SiO2	70.38	0.4%	74.70	0.5%	Si	20.0	35178
K2O	0.32	2.6%	0.22	1.7%	K	20.0	12364
CaO	12.69	0.7%	9.62	0.5%	Ca	20.0	40721
Al2O3	0.66	1.4%	0.68	1.5%	Al	10.0	37476
Fe2O3	1.50	0.6%	1.00	0.4%	Fe	20.0	42179
Missing	2.03	0.6%	1.00	0.3%			
sum	100.00		100.00				

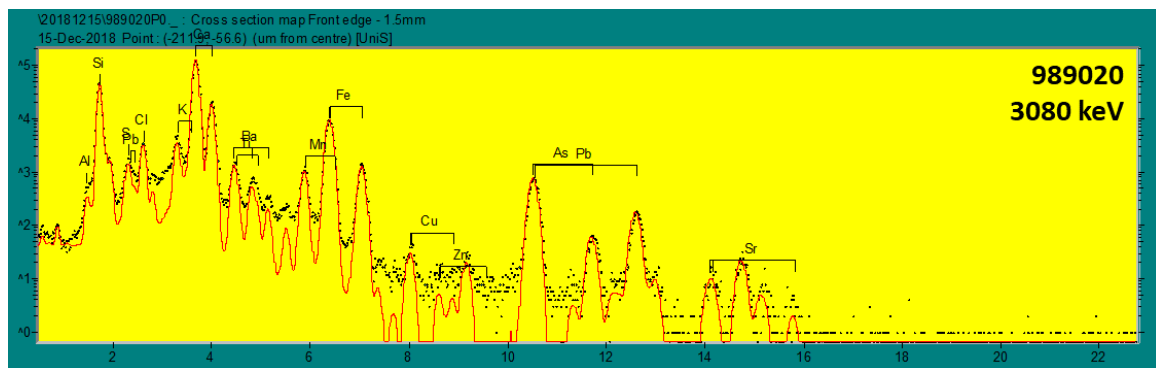
**Table 4: EPMA analysis of Rosslyn glass**

Relative standard uncertainties are given, except for the “Missing” column where absolute standard uncertainties are given

**Figure 4** shows a PIXE spectrum (OMDAQ #989020) from the Rosslyn glass cross-section sample measured by EPMA. Of course, far more elements are visible in PIXE than were measured by EPMA. But there is no PIXE signal for Na, and the signal for Al is unreliable in the presence of the poorly fitted low energy tailing. Therefore we impose the Na:Si and Al:Si ratios from EPMA in the Total-IBA fitting, and otherwise use the PIXE calibration from the BCR lead-glass standard sample. The results for this are shown numerically in **Table 5**. There is an EBS signal for Na, but its



sensitivity is limited (see **Table 7a** below). However, the EPMA is relatively sensitive, as well as having good accuracy since the EPMA instrumental factor is directly calibrated for Na.



**Figure 4: PIXE of Rosslyn glass with GUPIX fit**

Centre of cross-section, 500×500 μm<sup>2</sup> map, 2 hr run time

	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	Na <sub>2</sub> O	CaO	K <sub>2</sub> O	SO <sub>2</sub>	Fe <sub>2</sub> O <sub>3</sub>	Totals
<b>IBA</b>	76.3%	0.69%	13.1%	8.20%	0.179%	0.318%	0.80%	<b>99.3%</b>
<b>±</b>	1.4%			0.43%	0.010%	0.023%	0.05%	
<b>EPMA</b>	74.7%	0.68%	12.8%	9.62%	0.216%		1.001%	<b>99.0%</b>
<b>±</b>	0.4%	0.02%	0.3%	0.05%	0.004%		0.004%	
<b>Δ</b>	2%			-15%	-17%		-20%	
	TiO <sub>2</sub>	MnO <sub>2</sub>	CuO	ZnO <sub>2</sub>	PbO	BaO	AsO	SrO
<b>IBA</b>	0.0455%	0.0998%	0.0025%	0.0007%	0.0656%	0.1003%	0.0842%	0.0068%
<b>±</b>	0.0040%	0.0061%	0.0003%	0.0003%	0.0045%	0.0062%	0.0055%	0.0010%

**Table 5: Composition (mol%) of Rosslyn glass cross-section by IBA and EPMA**

EPMA results from **Table 4** are included. The “±” rows give the combined absolute standard uncertainties from counting statistics (Total-IBA) and reproducibility (EPMA). The “Δ” row gives the relative difference between IBA and EPMA. “Totals” exclude SO<sub>2</sub> and trace elements. Note that the ratio of Ca, K and Fe to Si is obtained by Total-IBA independent of EPMA.

**Table 5** shows the final IBA result with EPMA for comparison. That the Ca and K are lower than expected may indicate a sample tilt, but this spectrum is less sensitive than the lead-glass spectra to tilt effects since there is far less Pb and hence far less absorption by the Pb M line. That the Fe content is low compared to EPMA is a consequence of the high uncertainty of the (effectively) uncalibrated PIXE signal.

The “Totals” inset shows the mol% sum of the six elements measured by EPMA: note that the IBA and EPMA are indistinguishable given the EPMA uncertainty (see **Table 4**). Note also that this consistency is not a consequence of the analysis since the Si:Ca:Pb ratios are determined by EBS *independently* of EPMA results.

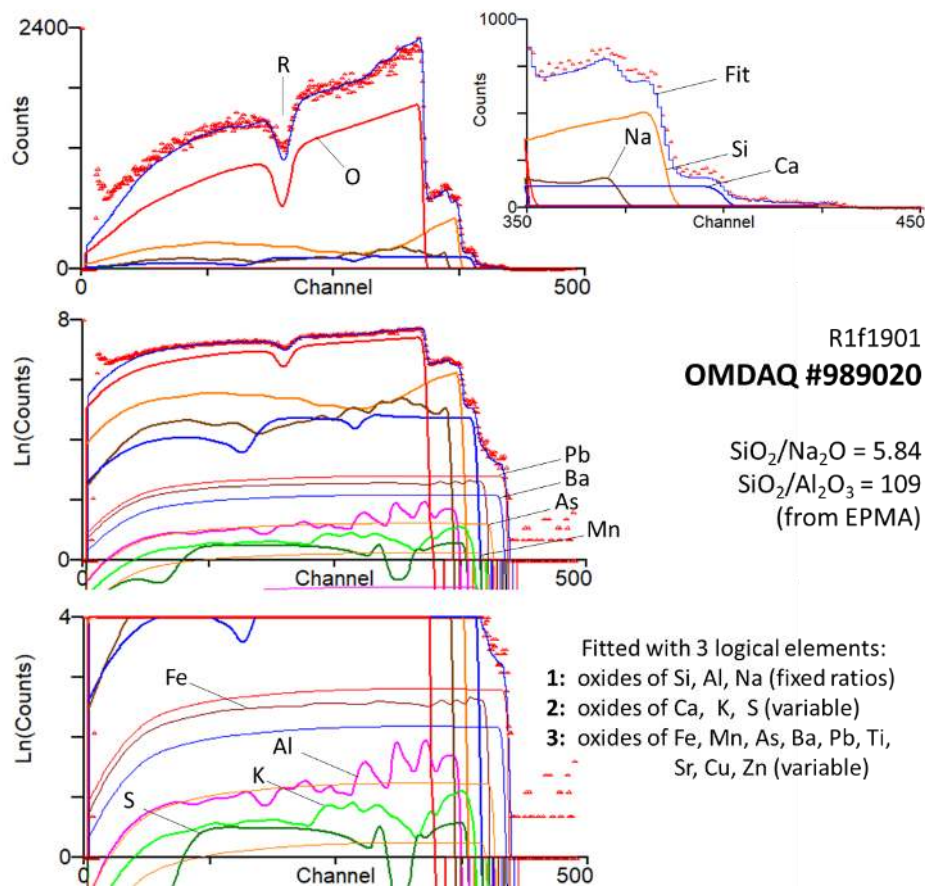
**Table 6** summarises the final measured elemental glass composition in both mol% and wt% for convenience.

The PIXE in **Figures 4, 5b** shows expected (calculated) but unobserved signals for Ba K (32 keV) and Sr L (1.8 keV). There is no sensitivity for either since the Ba K is above the measured energy range and the Sr L is overlapped by the strong Si signal (it is between the Si K $\alpha$  and K $\beta$  peaks). The GUPIX fitting also grossly overestimates the Al signal, due to the very poor fitting of the low energy detector tailing effects. This signal is adjusted to fit the calculation.

**Figure 5a** gives more detail of the fitting of the EBS spectrum showing all the minor elements contributing to the signal.

### **Benchmarking EBS scattering cross-sections**

Following the work of Paneta *et al.*<sup>81</sup> (stimulated by an IAEA *Coordinated Research Project* which reported formally in 2015 [ref.58], but see also Abriola *et al.*, 2011 [ref.59]), evaluated EBS scattering cross-sections have systematically been compared against benchmarking data (Gurbich, 2016 [ref.49]) and the central importance of benchmarking underlined. The issue is how to assess the uncertainty of an EBS cross-section function. This was previously (2013) addressed directly by Gai & Gurbich<sup>82</sup> who used the standard method of deriving the covariance matrix: however, they also showed that there existed quite large systematic errors that are unquantifiable in detail, commenting that in such cases these standard statistical methods are not strictly valid. They also commented that Mayer's earlier (2012<sup>83</sup>) approach to assessing EBS scattering cross-section uncertainties cannot be correct in principle, although pragmatically it seems very helpful (provided a nuclear model is available to allow direct comparability of data for different scattering angles).



**Figure 5a: Total-IBA of Rosslyn glass (particle spectrum)**

**Top (linear scale):** EBS partial spectra are shown for O, Na, Si, Ca (respectively purple, blue, orange, green).

**Top (inset):** detail of the EBS signal above the O edge. **Centre & bottom (natural log scale):** EBS of minor elements Al, K, S (purple, green, dark green); RBS of minor elements Pb, Ba, As, Mn (red, blue, orange, magenta)

Benchmark measurements are quite complicated, and so far they have not been used to underpin formal estimates of measurement uncertainties. They are usually used informally to give qualitative support for particular cross-sections, which is how they are presented on the SigmaCalc site (see Gurbich, 2016 [ref.49]): given a well-known sample, the particle scattering spectrum obtained

should be the same as the spectrum simulated for those conditions, provided correct cross-sections are used. Pragmatically, the spectrum is calculated and if the match is reasonable then so are the cross-sections. Kokkoris *et al.* (2017)<sup>84</sup> have helpfully quantified this judgment for 3 MeV protons on Si and O at various scattering angles: we will use these results in *Uncertainties*.

**Figure 5a** can be regarded as a benchmark for the sum of all the scattering cross-section functions used since the sample is a) uniform and b) has a composition well-determined (prior to IBA) by EPMA. The point here is that the full interpretation of the EBS spectrum relies on *mass closure* since the energy loss of the material is sensitive to the average Z of the material, in just the same way (but more directly) that the photon scattering in XRF is sensitive to average Z as pointed out in 2016 by Perrett *et al.* [ref.76].

This means that the spectral misfit (see again Figure 5a), which is a function of how much in error the scattering cross-sections actually are, does not in this case determine the uncertainty in the average Z. For in the present case, the spectrum is dominated by the O yield, but the O content is constrained by the assumption of full oxidation. That is, it is given unambiguously by the cation content of the glass. Therefore, the composition is known with a significantly lower uncertainty than would be assessed simply from the EBS misfit. The EBS spectra tell us, independently, *both* the composition *and* the error in the sum of the scattering cross-section functions (and also the charge·solid-angle product for this spectrum: although this is not independent of the *average* scattering cross-section it is nearly independent of the *shape* of the cross-section function). If we wished, DataFurnace could quantify the uncertainties in these parameters using the numerical methods (Markov chain Monte Carlo, MCMC) central to the code.<sup>85</sup> In the present case this would be somewhat misleading since we would obtain only the analysis *precision* which would look very high, being from a single dataset. We will estimate instead the analysis *accuracy* using more transparent and general methods.

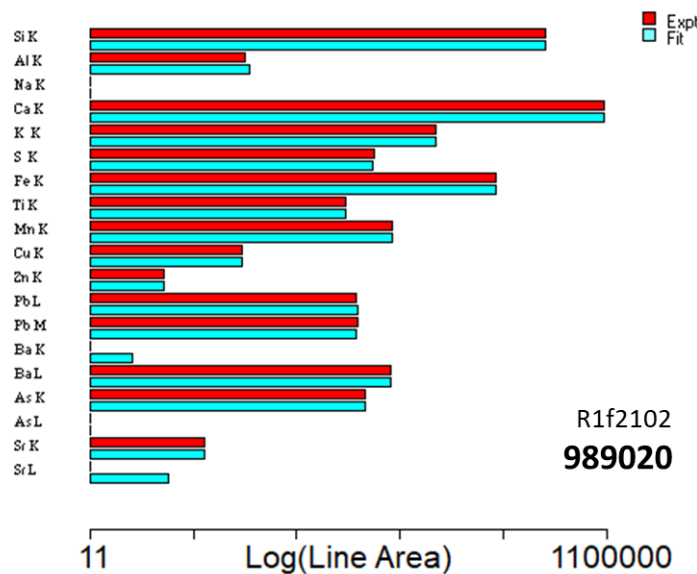


Figure 5b: Total-IBA of Rossllyn glass (PIXE data and fit)

	<b>Si</b>	<b>Al</b>	<b>Na</b>	<b>Ca</b>	<b>K</b>	<b>S</b>	<b>Fe</b>	<b>O</b>
<b>mol%</b>	25.44	0.278	8.722	4.099	0.1194	0.1061	0.3195	60.74
<b>wt%</b>	33.96	0.356	9.530	7.807	0.2219	0.1616	0.8479	46.18
	<b>Ti</b>	<b>Mn</b>	<b>Cu</b>	<b>Zn</b>	<b>Pb</b>	<b>Ba</b>	<b>As</b>	<b>Sr</b>
<b>mol%</b>	0.0152	0.0333	0.0013	0.0002	0.0328	0.0502	0.0421	0.0034
<b>wt%</b>	0.0345	0.0822	0.0033	0.0007	0.3228	0.3274	0.1500	0.0141

Table 6: Elemental bulk composition of Rossllyn glass

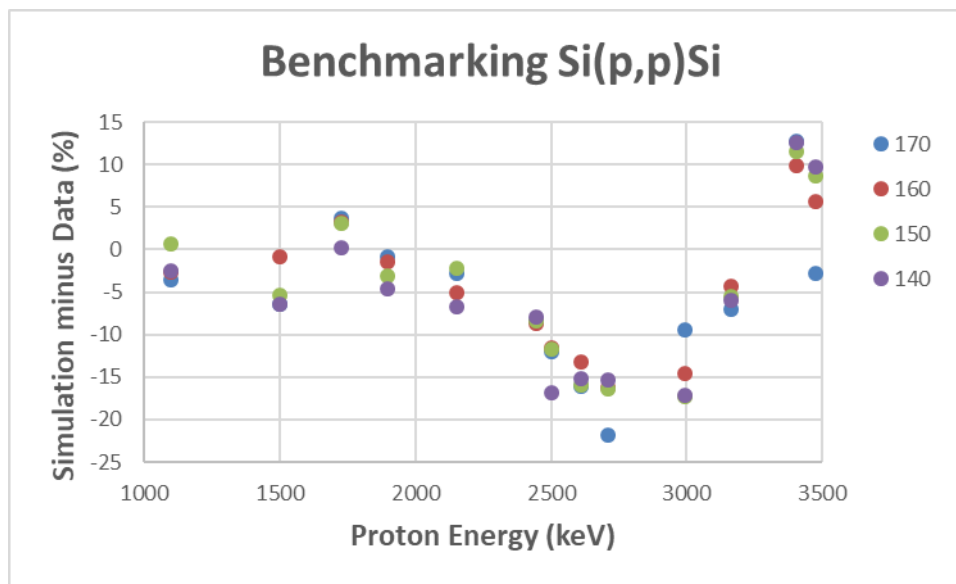
The most important cross-section in the present work on glass is that for  $^{16}\text{O}(\text{p,p})^{16}\text{O}$  since about 60% of the glass atoms are O. A nuclear model for this reaction was published in 1997 [ref.52] and the model parameters updated in 2001 (**Table 2**). Inspection of the 1997 paper shows that the model uncertainty at 3 MeV appears to be low (probably <2%), and inspection of the benchmark measurements (Fig.11 of Kokkoris *et al.*, 2017 [ref.84]) shows only ~5 % deviation at ~3 MeV.

However, the next most abundant element is Si, for which the cross-section data are much more uncertain. **Figure 6** shows benchmark data for this reaction reworked from observations on a pure Si test sample made previously by Kokkoris *et al* (2017 [ref.84]).

A nuclear model for  $^{28}\text{Si}(\text{p,p})^{28}\text{Si}$  was published in 1998 [ref.55]. However, the benchmarks of Kokkoris *et al* (2017 [ref.84]) using these cross-sections show 22% deviations for Si near 3 MeV (see their Table 3 and Figs.5c, 6b where simulations are *below* the data). The nuclear model parameters for the Si reaction were updated in 2017 (**Table 2**), and the benchmarks were recalculated (**Figure 6**) with significantly lower deviations. However, at 3080 keV there remain deviations ~10%.

## Uncertainties

**Tables 7a, 7b** give the measurement uncertainty from the counting statistics for the EBS and PIXE data respectively, together with the comparable reproducibility values for EPMA. We shall assume that the EPMA values represent the combined uncertainty fairly well, since the accuracy of this method for appropriate samples is recognised. But although a metrologically valid uncertainty budget has been published previously for RBS,<sup>86</sup> and an uncertainty budget has recently been presented (for the first time) for measurements of certain XRF parameters [ref.4], no uncertainty budget has yet been presented either for EBS or for PIXE.



**Figure 6: Comparing EBS data from pure Si with simulated spectra**

$^{nat}\text{Si}(\text{p,p})^{nat}\text{Si}$  scattering cross-sections from evaluation (SigmaCalc updated Feb. 2017). Various backscattering angles (140°, 150°, 160°, 170°) are used. Compare Fig.6 of Kokkoris *et al.*, 2017 [ref.84]

For PIXE (and the other XRF techniques) the fundamental difficulty is in assessing the combined uncertainties of the multiple cross-section databases (ionisation, fluorescence yield, absorption) together with issues related to the excitation volume (for electron excitation) or energy loss cross-sections (for ion excitation). A “bottom-up” analysis seems intractable, but “top-down” results are now well accepted. Indeed, it has been known for thirty years that XRF techniques (that is: XRF,

EPMA and PIXE) are capable of high accuracy (1%) analysis of flat homogenous thick target samples, provided good calibration standards are available. With proper calibrations of these methods, sub-1% accuracy has been repeatedly demonstrated for homogeneous samples (Hutton & Elliott, 1980<sup>87</sup>, Younis *et al.*, 2017,<sup>88</sup> Campbell *et al.*, 2019 [ref.46]).

EPMA accuracy has been discussed at length quite recently (2002) by Peter Statham<sup>89</sup>, who wrote the basic papers in the 1980s (see discussion in Bailey *et al.*, 2009 [ref.97]). This accuracy is absolute since the calibrations are general, determining the instrumental response (a spectrometer constant) as a function of X-ray energy. This is confirmed by the recent work on data collected by the Mars Rover: extra-terrestrial samples are intrinsically unknown (that is, comparable samples cannot be identified, strictly speaking) but the authors demonstrate that the analysis accuracy is extraordinarily high.<sup>90,91</sup> Moreover, the apparent systematic calibration difference between K, L, and M lines (apparently necessitating separate response curves, seen here in **Table 3**) have been shown recently to be an artefact of the databases.<sup>92</sup>

In the present work, the EPMA is assumed to have this accuracy, but the PIXE is essentially uncalibrated (since **Table 3** only specifies the calibration factor at about 10%). We wish to know in what way the PIXE accuracy is determined by the EBS (and *vice versa*).

For EBS the difficulty is assessing the uncertainty of the scattering cross-section function, which has not yet been done satisfactorily (*pace* Mayer [ref.83] and Gai & Gurbich [ref.82], see *Benchmarking* above). For RBS spectra, in the case where the composition of a uniform sample is required, most of the contribution to the uncertainty budget usually comes from the counting statistics, first demonstrated two decades ago in the case where the In content of InGaAs thin films was determined at 1% with a fully traceable estimate of the uncertainty.<sup>93</sup> Where the information is in the spectral step heights, the stopping power uncertainties cancel out almost entirely. And it is because the scattering cross-sections are known *analytically* that RBS is so accurate.

	EBS				EPMA
	RoI	integral	background	precision	<u>accuracy</u>
<b>Pb</b>	412-430	226	163	8.7%	
<b>Fe</b>					0.4%
<b>Ca</b>	390-400	1106	455	3.6%	0.5%
<b>Si</b>	375-383	4369	1506	1.8%	0.5%
<b>Na</b>	360-373	1888	8700	5.4%	1.7%
<b>O</b>	335-343	14833	5762	1.0%	

**Table 7a: EBS (and comparative EPMA) uncertainties for Rosslyn glass analysis**

The integral (in counts) for the given elemental signal is given for the appropriate region of interest (RoI: see **Figure 5a**); the background signal in this RoI is also given, together with the relative standard uncertainty from the counting statistics (including the contribution from O, see text). The relative standard uncertainty of the corresponding EPMA reproducibility is also given: see **Table 4**.

How can the composition uncertainty traceable to the EBS cross-section uncertainty in **Figure 5a** be characterised? Clearly the details of the spectrum depend heavily on the details of quite a large set of scattering cross-sections, each of which has a complicated functional shape with many parameters. Can any reasonable uncertainty be estimated? As for the XRF techniques, this problem is intractable to a “bottom-up” approach.

The sensitivity of the cross-section to scattering angle – at 150°: 0.6%/° for O(p,p) and 2%/° for Si(p,p) – is given by the nuclear model (see **Table 2** for references and updates). **Tables 8** indicate the sensitivity to scattering angle of the interpretation of the *steps* in the EBS spectrum at the elemental edges (see **Figure 5a**). These steps directly give the composition of homogeneous samples,



provided that the cross-sections are known. The cross-sections for various nuclei are given relative to Rutherford as a function of angle in **Table 8a**, and the consequent interpretation of a step height in the EBS spectrum as an elemental ratio is given in **Table 8b**.

LineName	Data	Fit	Uncertainty			EPMA %
	counts	counts	GUPIX %	Misfit %	Total %	Uncertainty
Si K	285330	285107	1.0	0.1%	2.1	0.5
Ca K	1041700	1041650	0.2	0.0%	3.7	0.5
K K	25064	25057.6	1.6	0.0%	4.0	1.7
S K	6234	6149.1	4.8	1.4%	6.1	
Fe K	94466	94469.5	0.4	0.0%	4.1	0.4
Ti K	3355	3348.8	6.7	0.2%	7.9	
Mn K	9526	9527.8	1.7	0.0%	4.5	
Cu K	328	328	10.0	0.0%	10.8	
Zn K	58	57.9	38.3	0.2%	38.5	
Pb L	4296	4368.8	3.6	1.7%	5.5	
Pb M	4430	4169.2	8.5	6.3%	9.5	
Ba L	9082	9119.8	1.9	0.4%	4.5	
As K	5130	5123.3	2.9	0.1%	5.0	
Sr K	142	142.3	14.3	0.2%	14.9	
Al K		388.5	11.9			1.5
Na K		1.2				1.7

**Table 7b: PIXE (and comparative EPMA) uncertainties for Rosslyn glass analysis**

For OMDAQ dataset #989020. The integral (in counts) for the given X-ray line is derived from GUPIX and given to NDF (“Data”); NDF “Fits” the data minimising the total misfit. The standard uncertainty per line from the counting statistics (including line overlaps) is obtained from GUPIX, with the “Misfit” (between Data and Fit) also given. The Total uncertainty is the quadrature sum of the GUPIX uncertainty and the appropriate EBS signal ratio uncertainty (from **Table 7a**: see text). The relative standard uncertainty of the corresponding EPMA reproducibility is copied from Table 7a.

	Beam	Scattering Angle					
	MeV	150°	155°	160°	165°	170°	175°
<b>16O</b>	3.080	7.669	7.901	8.085	8.223	8.318	8.373
<b>natSi</b>	3.080	1.869	2.205	2.544	2.852	3.099	3.259
<b>40Ca</b>	2.920	1.249	1.280	1.307	1.329	1.346	1.356
<b>natFe</b>	3.081	0.992	0.998	1.004	1.008	1.011	1.013
<b>208Pb</b>		1.000	1.000	1.000	1.000	1.000	1.000

**Table 8a: EBS cross-sections from proton beam relative to Rutherford from SigmaCalc 2.0**

EBS Step			Elemental Ratio							
Element	counts	±	Rutherford	150°	155°	160°	165°	170°	175°	measured
<b>O/Si</b>	3.39	2.1%	10.40	2.53	2.90	3.27	3.61	3.87	4.05	<b>2.39</b>
<b>O/Ca</b>	16.4	3.7%	102.4	16.7	16.6	16.6	16.6	16.6	16.6	<b>14.8</b>
<b>O/Pb</b>	138	8.8%	14532	1895	1839	1797	1767	1747	1736	<b>1853</b>

**Table 8b: Spectral steps converted to elemental ratios for Rosslyn glass analysis**

For OMDAQ dataset #989020. Ratio of step heights (in counts, with counting statistics uncertainty from **Table 7a**) for data converted to elemental ratios from **Table 8a** and compared with measured values (at 3080 keV, from **Table 6**). Note that the O/Si ratio is very sensitive to the scattering angle. The ratio for RBS at 150° is also given.

Standard Uncertainty		Comment
Counting Statistics	1.8%	From Table 7a
Angular (2° error)	3.6%	From Table 8b
Energy	0.3%	From 2663 keV resonance
Random Uncertainty	4.1%	
Scattering cross-section	10%	From Figure 6
Combined Uncertainty	10.8%	

Table 9: Simplified Uncertainty Budget for Si content by 3.08 MeV EBS

	mol%	Precision			Accuracy		Comment
		EBS	PIXE	Total-IBA	EPMA	Total-IBA	
O	60.7371	1.0%		1.0%			Chemical prior: assume oxides
Si	25.4415	1.8%	1.0%	0.9%	0.5%	<b>2.0%</b>	accuracy from EBS mass closure
Na	8.7218	5.4%		5.4%	1.7%	<b>2.6%</b>	accuracy from Si, fixed by EPMA
Ca	4.0988	3.6%	0.2%	0.2%	0.5%	<b>3.3%</b>	accuracy from Ca/Si
Fe	0.3195		8.8%	8.8%	0.4%	<b>10.6%</b>	accuracy from Pb
Al	0.2775				1.5%	<b>2.5%</b>	accuracy from Si, fixed by EPMA
K	0.1194		4.0%	4.0%	1.7%	<b>5.2%</b>	accuracy from Ca
S	0.1061		6.1%	6.1%		<b>6.9%</b>	accuracy from Ca
Ba	0.0502		9.0%	9.0%		<b>10.8%</b>	accuracy from Pb
As	0.0421		9.2%	9.2%		<b>11.0%</b>	accuracy from Pb
Mn	0.0333		8.9%	8.9%		<b>10.8%</b>	accuracy from Pb
Pb	0.0328	8.7%	7.5%	5.7%		<b>6.0%</b>	PIXE: L+M lines; accuracy from Pb/Si
Ti	0.0152		11.0%	11.0%		<b>12.6%</b>	accuracy from Pb
Sr	0.0034		16.8%	16.8%		<b>17.8%</b>	accuracy from Pb
Cu	0.0013		13.3%	13.3%		<b>14.6%</b>	accuracy from Pb
Zn	0.0002		39.3%	39.3%		<b>39.7%</b>	accuracy from Pb
sum	100.0000						

Table 10: Accuracy of Total-IBA (see text)

Mol% is from Table 6, EBS & PIXE uncertainties are from counting statistics (and misfits for PIXE; Tables 7a, 7b), EPMA uncertainty is from reproducibility (Table 7a). The Total-IBA Accuracy column is derived in various ways indicated by the Comments column: see text. Standard uncertainties are used throughout, and combined uncertainties are obtained simply by adding in quadrature.

However, although the O/Si ratio is sensitive to the scattering angle at about 2%/° at 150° as already mentioned (see Table 8b) there is a separate systematic and large uncertainty in the Si cross-section itself at this energy.

Table 9 summarises the contributions to the uncertainty of determining the Si signal height: this has a reproducibility dominated experimentally by the scattering angle uncertainty, but an absolute accuracy dominated by the uncertainty of the scattering cross-section itself at that energy. We use the benchmark calculations (Figure 6) to indicate a reasonable value of this latter dominating uncertainty. The uncertainty in the beam energy in this case is insignificant.

For this fit of the data, the PIXE data are fitted precisely (see Figure 5b). This means that the PIXE and EBS are consistent with each other considering the uncertainties, but that the fitting algorithm is able to give priority to the PIXE data since most of the elements visible in PIXE are either not well-determined or are entirely undetermined by EBS. However, the fact that the PIXE Si signal fits accurately means that the 151 µm Be filter thickness (determined from the certified standard, Figure 3c) is determined correctly. Another way of putting this is to say that the calibration is confirmed.

On the other hand, the PIXE calibration for Ca and K is clearly sensitive to geometry: the well-fitted Ca and K content in the BCR glass (see the H-factors in **Table 3**) is obtained using a sample tilt consistent with the EBS signal. The poorly fitted composition of the Rosslyn glass (see Ca and K in **Table 5**) may be an indication of a geometry error, although with less Pb content the sensitivity to this effect is smaller than for the BCR glass (the absorption edge of Pb M is at 2.6 keV and there is much more Pb in the lead-glass).

The Pb M calibration appears to vary by 20% between the BCR and Rosslyn glasses but this variation is consistent with the uncertainty of about 10% for the ratio of Pb L to Pb M (**Table 7b**).

**Table 10** summarises the overall uncertainty of the analysis, giving an explicit account of how the accuracy is inherited. Uncertainties are combined in a simple-minded way since it is not obvious how to do this rigorously. We have here attempted to include neither energy nor geometry errors, nor the systematic calibration errors for PIXE, nor the errors due to incorrect scattering cross-sections for EBS. Moreover, Papp & Maxwell (2017) show that counting statistics in PIXE are not Poisson, so that the usual  $1/\sqrt{N}$  measure may seriously underestimate the uncertainty.<sup>94</sup>

The “Accuracy” column in **Table 10** does not include the systematic (calibration) uncertainties of PIXE, and is determined variously as indicated in the “Comments” column :-

1. The Si accuracy is determined independently (see **Table 5**) and is discussed extensively above in the context of the EBS Uncertainty Budget (**Table 9**). Note that Table 9 shows that the random uncertainty (4 %) is larger, and the systematic uncertainty (10 %) is much larger, than the 2 % relative of Table 5 precisely because the chemical prior (that is, full oxidation of the cations) together with mass closure greatly reduce real uncertainties. This is the baseline uncertainty of the whole measurement.
2. For Na the Si accuracy is combined with the EPMA uncertainty: there is no measurable PIXE Na signal, and although there is an EBS Na signal this adds no information to the EPMA.
3. For Al the Si accuracy is combined with the EPMA uncertainty: there is no measurable Al signal for either EBS or PIXE.
4. For Ca the Si accuracy is combined with both EBS and PIXE uncertainties.
5. For Pb the Si accuracy is combined with both EBS and PIXE uncertainties, where the PIXE includes both the L and the M lines.
6. Uncertainties for the elements in LE2 (K, S) are combined from their PIXE uncertainty and the combined uncertainty for Ca.
7. Uncertainties for the elements in LE3 (Fe, Ti, Mn, Cu, Zn, Ba, As, Sr) are combined from their PIXE uncertainty and the combined uncertainty for Pb.

Considering Pb as an example (#5 in the list above) it is clear that where the accuracy from EBS cannot be better than 8.7% the Total-IBA accuracy inherits the accuracy of the PIXE (6%) where this in turn (together with the whole analysis) rests on the accuracy of the the determination of Si content from EBS (#1 in the list above), which depends on the mass closure criterion (see below). Inheritance is Discussed at length below.

In this case, considering **Figure 5a**, the O signal appears too high near the O edge, and the cation signal appears systematically too low, as does the signal deeper than the resonance (“R”). Clearly, the charge·solid-angle product has been adjusted slightly to give an optimal  $\chi^2$  fit over the whole spectrum. This may indicate that the  $^{16}\text{O}(p,p)^{16}\text{O}$  scattering cross-sections are too high near 3 MeV. However, the benchmarking (**Figure 6**) makes it clear that the Si(p,p)Si scattering cross-sections are too low, which is probably sufficient to account for the discrepancy between IBA and EPMA in both the Si and the Ca (and hence K) signals (see **Table 5**).

## Mass Closure

**Table 9** shows the 4.1 % EBS combined uncertainty of the Si signal, not including the uncertainty introduced by that of the EBS scattering cross-section for Si. The benchmark measurements shown in **Figure 6** indicate that these cross-sections are underestimated by about 10% at this energy. The Si content of the glass (as silica) is given by **Table 5**. In this section we show that the 2 % difference between IBA and EPMA shown for Si in Table 5 is much smaller than expected from a simple consideration of these uncertainties, and is a consequence of the mass closure constraint imposed by EBS together with the chemical prior of full oxidation.

It is clear that the underestimation of the EBS cross-section for Si has caused the Si content to be overestimated. But because this necessarily results in a consequent overestimation of the O content (since the Si only occurs as the oxide) both the Si and the O signals are too high. This is partially compensated by an arbitrary reduction in the assumed charge·solid-angle product (which is treated as a fitting parameter in this analysis). This is also partially compensated by a reduction in the Ca content: the relative concentrations of Si and Ca in the glass are given by **Table 6** and the relative sensitivity of EBS to Si and Ca at this energy can be determined from **Table 8b**. The spectral data for the Si/Ca ratio from EBS are given in **Table 7a**.

Indeed, **Table 5** shows that IBA biases the Si content high (because of the underestimated Si scattering cross-section) and therefore the Ca & K content low (taking the EPMA composition as “true”). This is also partially because the PIXE data strongly constrains the Si/Ca ratio completely independently of EBS (the spectral data for the Si/Ca ratio from PIXE are given in **Table 7b**). There may also be a further effect of an uncorrected sample tilt.

If there was no chemical prior for the glass composition, that is, if full oxidation was not assumed, then the O content would be an independent variable and in principle the total (combined) oxidation state of the sample could be measured independently. In practice, the derived value for the O content would be too sensitive to errors in the scattering cross-sections used to be useless for any but rather crude distinctions between oxidation states. Crucially, the derived cation content would also be very sensitive to errors in the scattering cross-sections.

Summarising, the mass closure properties of Total-IBA derive essentially from the EBS measurement and result in a significant improvement in absolute accuracy: they can be listed for clarity :-

1. The EBS signal is sensitive to the *average Z* of the sample. This has long been recognised (see §5.3 in Jeynes *et al.*, 2003 [ref.80]). Surprisingly, it is also true for XRF (not PIXE) spectra: the Guelph group has shown that the scatter peaks carry such information, which has been used to infer the existence of water on Mars (Perrett *et al.*, 2016 [ref.76])
2. Mass closure means that all major and significant minor elements have measurable signal: this is specifically true for soda-lime glasses measured by EBS (as here: glasses and minerals with significant Li, B, F would need NRA, usually PIGE, for mass closure). H can be detected with ERD, but the presence of H can usually be ignored since the energy loss due to H is usually negligible.
3. Mass closure very strongly constrains the stoichiometry. This was already demonstrated implicitly in the accurate measurement by RBS of the In content of InGaAs samples [ref.93].
- 3.4. If the charge·solid-angle product and stopping cross-sections are known, then mass closure very strongly constrains the scattering cross-sections. In this *in vacuo* case (and in external beam systems such as the AGLAE facility<sup>95</sup> in Paris) the solid-angle is an instrumental constant, and the charge can be effectively measured. And at present the scattering cross-sections for Si at this proton energy are known to be significantly underestimated.

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4.5. The known oxidation state of glasses is a strong chemical prior for the EBS, providing extra information that (with mass closure) allows the charge·solid-angle product to be inferred. It also (independently) allows some limited correction of the scattering cross-sections. The value of the chemical prior was first cogently argued by Butler (1990), demonstrated for optical coatings by Jeynes *et al.*, 2000 [ref.45], and discussed at length in both §8.5 of the *Topical Review* of 2003 [ref.80] and also §§3.9, 4.2 of the *Tutorial Review* of 2016 [ref.1]).

5.6. PIXE does not have mass closure, but it does give an independent measure of elemental ratios. The systematic uncertainties for EBS (relating to scattering cross-section uncertainties) are unrelated to the systematic uncertainties for PIXE (involving the Fundamental Parameters databases which are mostly common between PIXE and XRF: for recent work on these see Unterumsberger *et al.*, 2018 [ref.4]).

## Discussion

In this work the samples are homogeneous in depth, which is the case GUPIX is designed to handle. However, NDF is used instead because NDF allows simultaneous (synergistic) treatment of the particle spectrum, and also because NDF supports general layered samples (needed in the wider study). The PIXE module of NDF uses LibCPIXE<sup>96</sup> which is an independent code almost entirely equivalent to GUPIX (as shown by Bailey *et al.*, 2009<sup>97</sup>)

PIXE and EPMA ought to be consistent with each other, since they differ only in the excitation mechanism. (Actually, the databases used are independent, as are the analytical codes.) The EPMA calibration method here is more detailed and more accurate: the PIXE calibration here uses only a single reference sample, and essentially fits only the filter thickness and average efficiency of the X-ray detector to two main lines, where the other lines give a rough calibration (~10%) of the energy dependence of the efficiency.

The PIXE detector calibration from the BCR glass is well-behaved at <10% precision. This calibration carries over successfully to the Rosslyn glass sample, where the main composition determined by Total-IBA is consistent (within the uncertainties) with the EPMA values (see **Table 5**).

In this case the pivot elements (for which precise signals are available both in EBS and in PIXE) are Si, Ca, Pb. The values for these elements of the standard precision are respectively {2%, 4%, 9%} for EBS (**Table 7a**) and {2%, 4%, 6%} for PIXE (**Table 7b**). The present fitting code appears to give greater weight to the PIXE which is fitted perfectly, but that may largely be because the various X-ray lines do not interact strongly (unlike the EBS signals). Even in this simplified case with a sample homogeneous with depth, the two datasets (PIXE, EBS) interact in complicated ways so that the covariance table is not simple to construct.

What has become very clear is that both the beam energy and the geometry must be closely controlled for reasonable accuracy. We can demonstrate that large mismatches for the Ca and K PIXE signals in **Table 3** compared to EPMA can arise from quite small (5°) sample misorientations, consistent with the 5° tilt value obtained from the behaviour of the 2663 keV resonance in the EBS O signal (see **Figure 3b**).

But the largest contribution to uncertainty in EBS is due to the uncertain scattering cross-sections: the effect of these is the mismatch clearly visible above the O edge in the inset to **Figure 5a**. The benchmarking (see **Figure 6**) shows that the cross-sections for Si are low for this beam energy: therefore the Si content is forced high and consequently (because of mass closure) the Ca content is forced low (see the comparison with EPMA in **Table 5**). These data confirm the results of the benchmarks: it is a benchmark for the benchmarks.



1 There are a number of intricate issues that remain unresolved. The  $\chi^2$  fitting (even robustified) is  
2 suboptimal for PIXE, but it is not yet clear how to proceed correctly. The interplay between PIXE  
3 and EBS (that is, which is given priority in the fitting) is uncontrolled, and at present is accidental in  
4 practice, depending on the numerical details. In principle this should be under the control of the  
5 analyst. The present algorithm does facilitate some control: in particular in the way the logical  
6 elements are set up (see *Analytical Protocol*), but more facilities are needed for more complex  
7 datasets.

## 8 **Conclusions**

9 *Conclusions regarding the analysis method:*

- 10 1. These measurements are of an homogeneous glass, which can be analysed unambiguously  
11 (but not necessarily completely) by any of EPMA, PIXE, EBS independently. In this case  
12 EPMA was of only 6 major and minor elements where 8 are present (S, O are not measured);  
13 PIXE is also sensitive to the 8 trace elements present (but major & minor elements Al, Na, O  
14 were not measured); and EBS is sensitive to the major elements (O, Na, Si, Ca) and also the  
15 sum of these trace elements: {Pb, Fe, Ba}.
- 16 2. A weathered glass has a modified surface layer: the depth profile cannot be obtained by  
17 EPMA, or by PIXE alone. But the Total-IBA measurements reported here for homogeneous  
18 glass (for which they are not necessary) generalise successfully to these other cases (for  
19 which they are indispensable).
- 20 3. Glasses cannot usually be analysed successfully by EBS which does not have sufficient mass  
21 resolution to account adequately for the many minor and trace elements. But EBS is an  
22 energy loss method and therefore sensitive to the average Z of the sample. Consequently,  
23 Total-IBA (where the minor and trace elements are determined by PIXE) has powerful mass  
24 closure properties, enabling a significantly higher accuracy than one might expect from a  
25 simple (but incorrect) assumption of independence of all the contributions to uncertainty.
- 26 4. In such complex analyses the beam energy and the sample geometry must be tightly  
27 controlled. Moreover, in EBS it is also necessary to use good scattering cross-sections even  
28 for minor elements.
- 29 5. Benchmarking shows that EBS scattering cross-sections for Si(p,p)Si near 3 MeV are about  
30 10% too low. The present results are effectively a further and unequivocal benchmark for the  
31 previous benchmarking: they are actually quantifiable because the glass composition is  
32 known independently from EPMA. That the sample is not pure silicon is not necessarily a  
33 disadvantage since the more complex spectrum is able to fix most of the experimental  
34 parameters that contribute to the uncertainty in the simpler case.
- 35 6. It may be possible to use such glass samples to directly measure EBS cross-sections by the  
36 thick target inversion method, using the Bayesian inference facilities of DataFurnace as  
37 pioneered by Barradas *et al.* 2008.<sup>98</sup>
- 38 7. Sodium mobility under the beam does not preclude valid analysis in this case.

39 *Conclusions regarding the analysis uncertainties:*

- 40 1. The present EPMA measurements obtain the mass closure with 0.6% accuracy, independently  
41 confirmed by IBA demonstrating internal consistency of the EPMA/IBA analysis. Therefore  
42 the EPMA results are regarded here as “true”, enabling the accuracy of the Total-IBA to be  
43 fairly assessed.
- 44 2. Synergistic Total-IBA measurements give a Si content with a deviation of only 2% from the  
45 EPMA result. This is in the context of intrinsic (rather uncertain) accuracies of ~10% for  
46 both PIXE (calibration uncertainties) and EBS (cross-section uncertainties). This accuracy  
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derives from the mass closure properties due to all the major elements being visible in the EBS spectra.

- Measurements of individual elements show that accuracy is inherited from the most accurate technique, with the trace elements having PIXE accuracy and the major elements having EBS accuracy. The whole analysis is more accurate than expected from a simple uncertainty budget that neglects the powerful mass closure properties of EBS.

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**AUTHOR DECLARATION**

We wish to confirm that there are no known conflicts of interest associated with this publication and there has been no significant financial support for this work that could have influenced its outcome.

We confirm that the manuscript has been read and approved by all named authors and that there are no other persons who satisfied the criteria for authorship but are not listed.

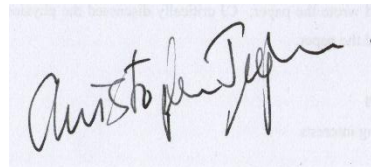
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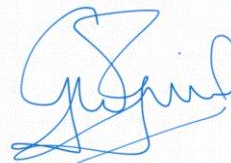
M.Kokkoris, 4<sup>th</sup> June 2019



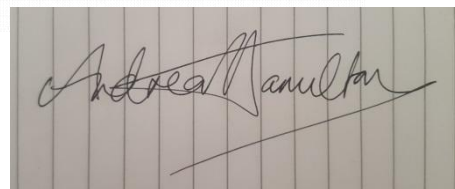
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**Declaration of interests**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

## Author Contribution Statement

CJ      Conceptualization; Data curation; Formal analysis; Funding acquisition; Methodology; Project administration; Resources; Software; Supervision; Validation; Visualization; Writing - original draft; Writing - review & editing.

VVP    Investigation; Resources; Software; Writing - review & editing.

MK      Data curation; Formal analysis; Investigation; Methodology; Resources; Validation; Visualization; Writing - review & editing.

AH      Conceptualization; Data curation; Formal analysis; Investigation; Methodology; Resources; Writing - review & editing.

GWG    Data curation; Formal analysis; Investigation; Resources; Software;