

University of Groningen

Phase diagram of the ferroelectric relaxor $(1-x)\text{PbMg}_{1/3}\text{Nb}_{2/3}\text{O}_3-x\text{PbTiO}_3$

Noheda, B.; Cox, D.E.; Shirane, G.; Gao, J.; Ye, Z.-G.

Published in:
Physical Review B

DOI:
[10.1103/PhysRevB.66.054104](https://doi.org/10.1103/PhysRevB.66.054104)

IMPORTANT NOTE: You are advised to consult the publisher's version (publisher's PDF) if you wish to cite from it. Please check the document version below.

Document Version
Publisher's PDF, also known as Version of record

Publication date:
2002

[Link to publication in University of Groningen/UMCG research database](#)

Citation for published version (APA):

Noheda, B., Cox, D. E., Shirane, G., Gao, J., & Ye, Z-G. (2002). Phase diagram of the ferroelectric relaxor $(1-x)\text{PbMg}_{1/3}\text{Nb}_{2/3}\text{O}_3-x\text{PbTiO}_3$. *Physical Review B*, 66(5). <https://doi.org/10.1103/PhysRevB.66.054104>

Copyright

Other than for strictly personal use, it is not permitted to download or to forward/distribute the text or part of it without the consent of the author(s) and/or copyright holder(s), unless the work is under an open content license (like Creative Commons).

The publication may also be distributed here under the terms of Article 25fa of the Dutch Copyright Act, indicated by the "Taverne" license. More information can be found on the University of Groningen website: <https://www.rug.nl/library/open-access/self-archiving-pure/taverne-amendment>.

Take-down policy

If you believe that this document breaches copyright please contact us providing details, and we will remove access to the work immediately and investigate your claim.

Downloaded from the University of Groningen/UMCG research database (Pure): <http://www.rug.nl/research/portal>. For technical reasons the number of authors shown on this cover page is limited to 10 maximum.

Phase diagram of the ferroelectric relaxor $(1-x)\text{PbMg}_{1/3}\text{Nb}_{2/3}\text{O}_3-x\text{PbTiO}_3$

B. Noheda,* D. E. Cox, and G. Shirane

Department of Physics, Brookhaven National Laboratory, Upton, New York 11973

J. Gao and Z.-G. Ye

Department of Chemistry, Simon Fraser University, Burnaby, BC, Canada V5A 1S6

(Received 22 March 2002; published 2 August 2002)

Synchrotron x-ray powder diffraction measurements have been performed on unpoled ceramic samples of $(1-x)\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3-x\text{PbTiO}_3$ (PMN-xPT) with $30\% \leq x \leq 39\%$ as a function of temperature around the morphotropic phase boundary, which is the line separating the rhombohedral and tetragonal phases in the phase diagram. The experiments have revealed very interesting features previously unknown in this or related systems. The sharp and well-defined diffraction profiles observed at high and intermediate temperatures in the cubic and tetragonal phases, respectively, are in contrast to the broad features encountered at low temperatures. These peculiar characteristics, which are associated with the monoclinic phase of M_C -type previously reported by Kiat *et al.* [Phys. Rev. B **65**, 064106 (2000)] and Singh and Pandey [J. Phys. Condens Matter **13**, L931 (2001)], can only be interpreted as multiple coexisting structures with M_C as the major component. An analysis of the diffraction profiles has allowed us to properly characterize the PMN-xPT phase diagram and to determine the stability region of the monoclinic phase, which extends from $x=31\%$ to $x=37\%$ at 20 K. The complex landscape of observed phases points to an energy balance between the different PMN-xPT phases which is intrinsically much more delicate than that of related systems such as $\text{PbZr}_{1-x}\text{Ti}_x\text{O}_3$ or $(1-x)\text{Pb}(\text{Zn}_{1/3}\text{Nb}_{1/3})\text{O}_3-x\text{PbTiO}_3$. These observations are in good accord with an optical study of $x=33\%$ by Xu *et al.* [Phys. Rev. B **64**, 020102 (2001)], who observed monoclinic domains with several different polar directions coexisting with rhombohedral domains, in the same single crystal.

DOI: 10.1103/PhysRevB.66.054104

PACS number(s): 77.84.Dy, 61.10.-i

I. INTRODUCTION

Solid solutions of the relaxor-ferroelectrics $(1-x)\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3-x\text{PbTiO}_3$ and $(1-x)\text{Pb}(\text{Zn}_{1/3}\text{Nb}_{2/3})\text{O}_3-x\text{PbTiO}_3$, known as PMN-xPT and PZN-xPT, respectively, are under active consideration for a new generation of electromechanical devices. When properly oriented, they have piezoelectric coefficients which are the highest yet reported, with electromechanical deformations one order-of-magnitude larger than those of conventional high piezoelectric PbZrO_3 - PbTiO_3 (PZT) ceramics.^{1,2} The exceptional electromechanical properties of these lead oxide-based solid solutions have long been known to be related to the nearly vertical phase boundary between the rhombohedral and the tetragonal phases, the so-called morphotropic phase boundary (MPB), which is a common feature of the PbMeO_3 - PbTiO_3 systems.³⁻⁵

However, the origin of this unusual behavior remained unknown until recently, when a monoclinic phase was observed around the MPB of PZT, in between the rhombohedral and tetragonal phases.^{6,7} The monoclinic phase in PZT, with space group Cm , has the unique axis b_m along the [110] direction, and a unit cell that is doubled with respect to the primitive cubic one and rotated 45° about the c axis with respect to it. We designate this monoclinic phase as M_A -type, following the notation of Vanderbilt and Cohen, who in a recent theoretical study derived for the first time a region of stability for ferroelectric monoclinic perovskites by an extension of the classic Devonshire-Landau expansion of the free energy to eighth-order.⁸ Diffraction experiments with an electric field applied *in situ*⁹ together with first-principles

calculations¹⁰ on PZT ceramics have demonstrated a direct link between the M_A phase and the high electromechanical deformations in PZT due to rotation of the polarization between the [111] and [001] directions. This fact explains some other studies in which an enhancement of the piezoelectric properties of [001]-oriented rhombohedral PZT was reported.^{11,12}

First-principles calculations¹³ have also shown that the anomalously high strain values observed in rhombohedral PZN-xPT crystals under a [001]-electric field¹ can be attributed to polarization rotation from the rhombohedral [111] to the tetragonal [001] polar axis. Diffraction experiments with an electric field applied *in situ*^{14,15} have permitted the direct observation of such polarization rotation paths, and have shown that for rhombohedral PZN-8PT, which has a composition very close to the MPB, a *different* monoclinic phase (M_C in the notation of Ref. 8) is induced when an electric field is applied along the [001] direction. The M_C phase has the Pm space group, with the unique axis b_m oriented along the pseudocubic [010] direction. With increasing electric field, the M_C phase becomes less distorted and approaches tetragonal symmetry but when the field is decreased to zero, an orthorhombic (O) phase is reached, similar to that of BaTiO_3 , which can be viewed as a M_C phase with $a_m = c_m$.¹⁴ The appearance of this irreversibly induced O phase in PZN-8PT suggested that an orthorhombic phase would probably exist close by in this region of the phase diagram. Shortly afterwards, x-ray powder diffraction experiments^{16,17} did in fact reveal the existence of such an O phase in a narrow composition region ($8\% < x < 11\%$) between the rhombohedral (R) and tetragonal (T) phases in the PZN-xPT

phase diagram. The O phase can be regarded as being very close in energy to the M_C phase for the following two reasons; first, a very small electric field is enough to induce the O - M_C transformation¹⁴ and, second, one of the four PZN-9%PT samples studied by Uesu *et al.*¹⁸ showed M_C symmetry rather than O .

The MPB phases of the piezoelectric system PMN-xPT^{5,19} have not been extensively studied until the last two years, during which a series of papers on this material have appeared. PMN-33%PT single crystals poled along the [011] direction appear to be orthorhombic,²⁰ while x-ray diffraction experiments have shown that a PMN-35%PT crystal poled along the [001] directions has M_A -type monoclinic symmetry, similar to that of PZT.²¹ The latter work also reports that unpoled PMN-35%PT single crystals are purely rhombohedral, in contrast to previous optical-microscopy studies of crystals of the same nominal composition which reported that they display a complicated pattern of rhombohedral and tetragonal domains.²² Very recently, fascinating domain patterns, arising from different monoclinic phases with different polarization directions, have been observed by Xu *et al.* in a PMN-33%PT crystal.²³ In addition, both neutron²⁴ and x-ray²⁵ powder diffraction measurements have revealed the existence of a monoclinic phase of M_C -type, in PMN-35%PT at low temperatures, and in PMN-34%PT at room temperature, respectively. All these results indicate that the behavior of PMN-xPT around the MPB is very complicated and requires more systematic studies to be properly understood.

In the present work we report a high-resolution synchrotron x-ray powder diffraction study on PMN-xPT ceramic samples of several compositions as a function of temperature around the MPB. An intermediate monoclinic phase of M_C type has been observed in agreement with previous reports.^{24,25} However, in contrast to the latter results, it has been found that the monoclinic phase does not exist as a single phase in any of the samples studied. The present study has allowed us to determine the region of stability of the intermediate phase and to construct a new phase diagram for PMN-xPT.

II. EXPERIMENT

Samples of nominal composition $(1-x)\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3-x\text{PbTiO}_3$ with $x=0.30, 0.31, 0.33, 0.35, 0.37,$ and 0.39 (hereafter designated xPT, with x in %) were synthesized using the two-step columbite precursor technique.²⁶ The starting reagents were oxide powders with purities better than 99.9%. In the first step, a mixture of MgO and Nb_2O_5 containing a 15.5 wt % excess of MgO over the 1:1 stoichiometric proportions was thoroughly ground in ethanol and cold-pressed. The excess amount of MgO was added in order to compensate for subsequent weight losses at high temperatures as determined from a previous DTA/TG analysis. The pellet was dried at 80 °C and calcined at 1100 °C for 12 h to form pure MgNb_2O_6 of columbite structure. In the second step, this precursor powder was mixed with PbO and TiO_2 in the appropriate proportions and a 2 wt % excess of PbO added to compensate for the evaporation

losses during the calcining and sintering processes. Each composition was thoroughly ground and calcined at 900 °C for 4 h to form the perovskite phase. The calcined powders were reground with the addition of a few drops of polyvinyl alcohol (PVA) and cold-pressed into pellets about 3 mm thick and 15 mm in diameter, which were first heated to 650 °C in an open Pt crucible for 1 h to drive off the PVA, and then sintered at 1200 °C for 4 h in an alumina crucible to form highly densified ceramics. The light yellow surfaces of the ceramic pellets were polished with fine diamond paste and ultrasonically cleaned. X-ray diffraction patterns obtained on a laboratory diffractometer with $\text{Cu } K\alpha$ radiation confirmed the formation of perovskite-type phases with no evidence of any impurities.

Synchrotron x-ray powder diffraction measurements were performed at beamline X7A at the Brookhaven National Synchrotron Light Source on several different occasions. In each case a double-crystal channel-cut Si (111) monochromator was used in combination with a Ge (220) analyzer and a scintillation detector. The wavelength was set to ≈ 0.7 Å and calibrated with a Si reference standard. With this configuration, the angular resolution is better than 0.01° . In most cases the data were collected directly from the ceramic pellets in Bragg-Brentano geometry by carrying out step-scans at 0.005 or 0.01° intervals over selected angular regions while the sample was rocked over a few degrees to improve powder averaging. This type of scan provides good counting rates but can result in incorrect intensity ratios if significant preferred orientation is present in the pellets. However, the peak positions and hence the derived unit cell parameters are not affected by this problem. Temperature dependence measurements, between 20 and 500 K, were performed with the samples loaded in a closed-cycle cryostat. A further set of measurements was made at room temperature on capillary samples which were rotated at about 1 Hz during data collection. Such data are free from preferred-orientation effects, but suffer from the disadvantage that the counting rates are quite slow. For this purpose, a small piece of each pellet was carefully crushed and sieved, and the resulting 325–400 mesh fraction (about 38–44 μm) loaded into a thin glass-walled capillary of nominal diameter 0.2 mm.¹⁶ In most cases scans were carried out over narrow angular regions centered about the six pseudocubic reflections (100), (110), (111), (200), (220), and (222), from which it is possible to determine unambiguously the crystal symmetry within the limits of the instrumental resolution, together with the corresponding lattice parameters and any zero offset if present. For the data analysis, the individual reflection profiles were fitted to a pseudo-Voigt function appropriately corrected for asymmetry²⁷ with intensity, peak position, peak-width [Full width at half maximum (FWHM)] and mixing parameter as variables. When appropriate, the peak widths and mixing parameters were constrained to be the same for all the peaks in a clump. Additional peaks were added based on the difference patterns, the relative intensities, and the goodness-of-fit residuals. Compared to the standard Rietveld-type technique of profile-matching, in which the full diffraction pattern is fitted without a structural model, the advantage of this procedure is that it does not require any assumptions about the

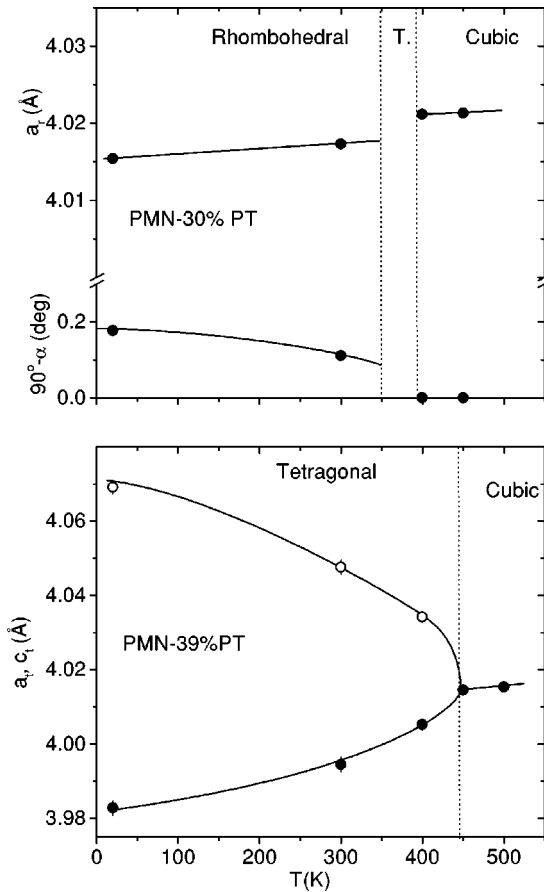


FIG. 1. Lattice parameters as a function of temperature for the PMN-30PT (top) and PMN-39PT (bottom) compositions. Dotted lines represent the respective transition temperatures as defined in the text. Except when indicated, the error bars are smaller than or comparable to the symbols.

phases to be included, how the FWHM and mixing parameters vary as a function of 2θ , and how to allow for anisotropic broadening, when neighboring peaks have significantly different widths. We have found that this is a fairly common effect in PbO-based perovskite systems, and is likely to occur when one or more of the lattice parameters is more sensitive to compositional fluctuations than the others. If overlooked, anisotropic broadening can be mistakenly interpreted as a lower-symmetry structure in a profile-matching analysis.

Some information about the microstructure and compositional fluctuations in the samples can be obtained from Williamson-Hall plots²⁸ of the peak widths, which can be used to provide estimates of the mean coherence length, L , and fluctuations in the d -spacings, $\Delta d/d$. The observed values of $\Delta d/d$ in the cubic phase are consistent with long-range compositional fluctuations in the samples of typically $\pm 1\%$ (FWHM), based on the lattice parameters reported by Noblanc *et al.*¹⁹ for compositions in the range 25–40PT.

III. TEMPERATURE DEPENDENCE

The temperature evolution of the lattice parameters for 30PT and 39PT is shown in Fig. 1. These two compositions

are located to the left and right sides, respectively, of the MPB in the PMN-xPT phase diagram, and the Curie temperatures, T_C , are shown as vertical dotted lines. The determination of T_C , which is usually made from dielectric measurements, is difficult in this case due to the broadness of the dielectric constant peak. In this paper the T_C values have been adapted from those of Noblanc *et al.*¹⁹ as the average of the two temperatures, T_m (maximum) and T_d (depoling), obtained from dielectric measurements on poled samples, which produce better-defined dielectric anomalies. Below their respective T_C 's of about 400 and 450 K, 39PT is tetragonal down to 20 K, while 30PT is first tetragonal, becoming rhombohedral at about 350 K,¹⁹ and remaining so down to 20 K.

The diffraction profiles around the pseudo-cubic (111), (200), and (220) diffraction peaks from 30PT and 39PT at 300 K are shown in Figs. 2(a) and 2(d), respectively, in which the least-squares fits to the data points are shown as solid lines and the peak positions by vertical arrows. In each case, the observed diffraction peaks show no evidence of phase coexistence; however, an analysis of the peak broadening reveals that the two compositions behave quite differently. Figure 3 shows the respective Williamson-Hall plots,²⁸ which have slopes of $2\Delta d/d$ and intercepts of λ/L , where L is the coherence length. While the 39PT composition shows isotropic broadening with $\Delta d/d=9.5\times 10^{-4}$ and a coherence length $L=0.5\ \mu\text{m}$, the 30PT composition shows highly anisotropic broadening and much smaller L values of about $0.2\ \mu\text{m}$, as shown in Fig. 3 (top). In the 30PT composition, the broadening is very small for the (hhh) peaks, with $\Delta d/d=2.6\times 10^{-4}$, and most pronounced for the ($h00$) peaks, with $\Delta d/d$ an order-of-magnitude larger. This situation is very similar to that observed in the rhombohedral phase of PZT, in which a large degree of intrinsic disorder²⁹ is known to exist due to the existence of local displacements of monoclinic type even though the long-range structure is rhombohedral.³⁰

Intermediate compositions gave considerably more complicated diffraction patterns, as can be seen in Figs. 2(b) and 2(c) for the 31PT and 33PT samples at 300 K. From a detailed analysis of the peak positions and intensities of the fitted profiles, it was possible to identify a monoclinic phase of M_C -type in the 31PT, 33PT, 35PT, and 37PT compositions, similar to that reported by Kiat *et al.*²⁴ and Singh *et al.*²⁵ For example, in the case of 31PT it can be seen in Fig. 2(b) that the profiles of the pseudo-cubic (200) and (220) reflections are distinctly different from those of 30PT shown in Fig. 2(a). Various possibilities were considered to explain the observed peak positions and intensities of 31PT; in particular the lower-symmetry phases previously found in this and related systems, i.e., orthorhombic O , monoclinic M_A , and monoclinic M_C . Among these, both O and M_C are found to account quite well for most of the observed features, while M_A can be ruled out altogether. However, there are several discrepancies in the intensity ratios which can only be accounted for by the presence of a minority rhombohedral phase, with an estimated volume fraction of about 30%.

As mentioned earlier, O and M_C are closely related and

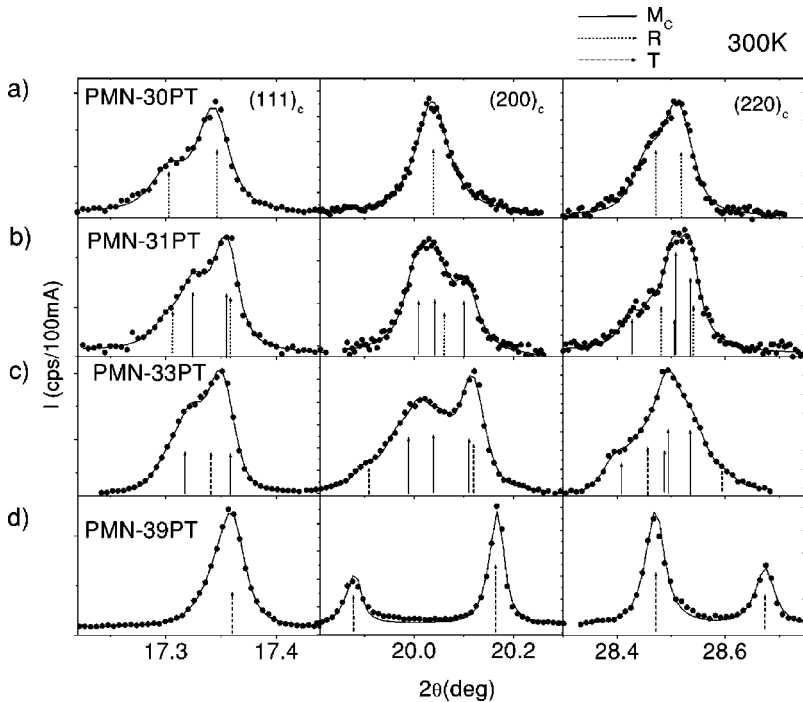


FIG. 2. Selected regions of the diffraction patterns at 300 K for (a) PMN-30PT, (b) PMN-31PT, (c) PMN-33PT, and (d) PMN-39PT, obtained from capillary samples. The solid lines are the least-squares fits to the data points and the arrows indicate the peak positions obtained from the fits.

very difficult to distinguish from each other, since the O phase simply represents the limiting case of the M_C phase when $a_m = c_m$. In a diffraction pattern, the difference between the two is most obvious for the pseudo-cubic (200) reflection, which would split into three peaks with roughly equal intensities for M_C , but only two peaks with a 2:1 intensity ratio for O . In the case of 31PT, it is not immediately apparent whether the broad lower-angle peak is com-

posed of one or two peaks; however, we were able to obtain a significantly better fit with two peaks, which together with the rest of the monoclinic reflections, enabled the lattice parameters of the M_C phase to be determined with reasonable accuracy. The corresponding peak positions are indicated by solid vertical arrows in Fig. 2(b).

The monoclinic lattice parameters of 31PT are plotted as a function of temperature in Fig. 4. At 20 K, a_m and c_m are fairly well-differentiated, but they approach each other with increasing temperature and can no longer be resolved at 350 K, as indicated by the error bars in Fig. 4. At the same time b_m increases and the monoclinic angle β , approaches 90° . Between 350 and 375 K, the monoclinic phase transforms to a tetragonal one, with lattice parameters a_t and c_t . It is noteworthy that the variation between b_m and a_t is continuous at the phase transition, in striking similarity to the behavior observed for the orthorhombic phase in PZN-xPT.^{16,17} Finally, between 400 and 425 K, the sample transforms into the cubic phase, consistent with the PMN-xPT phase diagram in Ref. 19.

At 300 and 20 K, the minority rhombohedral phase has the lattice parameters listed in Tables I and II. The corresponding peak positions at 300 K are indicated by the dotted arrows in Fig. 2(b). The volume fraction of the rhombohedral phase remains constant at roughly 30% up to 350 K.

The same regions of the powder diffractogram are plotted in Fig. 2(c) for the 33PT composition at 300 K. As in the case of 31PT, identification of the fitted peaks was based upon consideration of the R , T , M_A , M_C , and O phases, both alone and in various combinations. Once again, the M_C phase was found to account satisfactorily for most of the peaks. Compared to the M_C phase of 31PT [Fig. 2(b)] there is a larger splitting of the three peaks in the pseudocubic (200) profile, corresponding to a monoclinic distortion with a greater difference between a_m and c_m , and hence more dis-

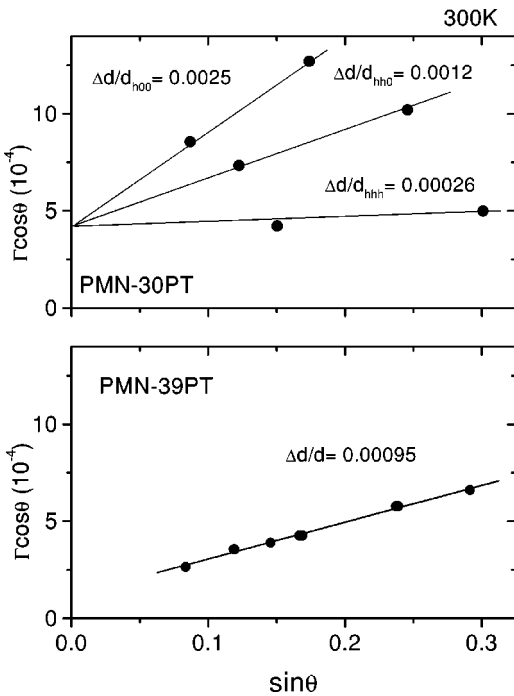


FIG. 3. Williamson-Hall plots for PMN-30PT (top) and PMN-39PT (bottom) at 300 K illustrating the highly anisotropic peak broadening in PMN-30PT as compared to 39PT.

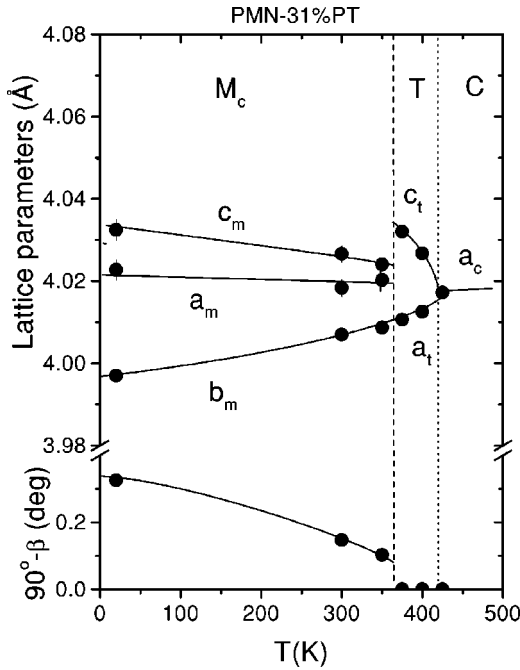


FIG. 4. Temperature evolution of the lattice parameters of PMN-31PT. The dashed line indicates the M_C - T transition temperature. The dotted line represent the T_C derived from the results of Noblanc *et al.* (Ref. 19), as described in the text. Except when indicated, the error bars are smaller than or comparable to the symbols.

tinct from the limiting orthorhombic phase. However, as before, there were several discrepancies in the intensity ratios and clear evidence of a low-angle shoulder on the $(h00)$ profiles which could only be explained by the presence of a minority phase, but in this case tetragonal instead of rhombohedral, with a volume fraction of about 25%. The calculated peak positions for the M_C and T phases are indicated in Fig. 2(c) by solid and dashed arrows, respectively.

The temperature evolution of the monoclinic lattice parameters is plotted in Fig. 5, and is very similar to that of 31PT except for the larger difference between the a_m and c_m lattice parameters. Between 300 and 325 K, the monoclinic phase transforms into a tetragonal one, which then trans-

TABLE I. Lattice parameters at 300 K for the PMN-xPT compositions studied. The symmetries of the different phases (S) and their volume fractions (f) are indicated in the second and third columns, respectively.

x (%)	S	f (%)	a (Å)	b (Å)	c (Å)	$\alpha(=\gamma)$ (°)	β (°)
30	R	100	4.017	4.017	4.017	89.89	89.89
31	R	30	4.017	4.017	4.017	89.89	89.89
31	M	70	4.018	4.007	4.026	90	90.15
33	M	75	4.019	4.006	4.032	90	90.19
33	T	25	4.005	4.005	4.046	90	90
35	$M+O$	35	4.018	4.000	4.035	90	90.12
35	T	65	4.000	4.000	4.044	90	90
37	T	80	3.998	3.998	4.049	90	90
39	T	100	3.994	3.994	4.047	90	90

forms to the cubic phase between 400 and 425 K, in agreement with the previously reported PMN-xPT phase diagram.¹⁹ The volume fraction of the minority T phase is also found to be constant at about 25% over the entire temperature range in which it coexists with the M_C phase. The a_t lattice parameters at 300 and 20 K are not shown in Fig. 5, but are very similar to b_m at 300 and 20 K, as inferred from the overlap of the tetragonal (200) and monoclinic (020) peaks on the high-angle side of the (200) pseudocubic reflection [see Fig. 2(c)], and thus follow a continuous trend with a_t in the purely tetragonal phase at higher temperatures. c_t follows a similar continuous trend at low temperatures with c_t at higher temperatures. The tetragonal lattice parameters for 33PT at 300 and 20 K are listed in Tables I and II, respectively.

The behavior observed in 35PT is more complicated. The diffraction profiles around the pseudocubic (200), (220), and (222) reflections are plotted in Fig. 6 for the three different symmetry regions observed as a function of temperature: the cubic phase at 450 K, the tetragonal phase at 400 K, and the low temperature region at 20 K. The strikingly broad features observed at low temperatures are in sharp contrast to the narrow and well-resolved peaks observed in the tetragonal and cubic phases. The high-angle peak in the (200) profile, which corresponds to the smallest lattice parameter, b_m , shifts toward higher angles and remains relatively sharp. In contrast, the intensity in the lower-angle region becomes very broad and the fitting procedure becomes accordingly more difficult. However, it is quite clear that this broadening cannot be explained simply by the existence of a second tetragonal phase. Fortunately, the $(hh0)$ and (hhh) reflections do not show such pronounced broadening and can be fitted satisfactorily. The inset to Fig. 6(b) shows the experimental data points around the pseudocubic (220) reflection at 20 K together with the solid line representing a six-peak fit which is in excellent agreement with the observed profile. Most of the peak positions are consistent with the coexistence of M_C and T phases, and the corresponding lattice parameters at 300 and 20 K are listed in Tables I and II, respectively. However, there remain some significant discrepancies in the intensity ratios and also one unexplained peak in the $(hh0)$ profiles. It turns out that these features can be accounted for very well by the presence of about 30% of a third orthorhombic (O) phase with b_m and β equal to those of M_C phase, and with $a_m = c_m$ equal to the mean value of the a_m and c_m values for M_C (see Table II). At 300 K, the O and M_C lattice parameters cannot be resolved.

The 37PT composition was found to be a two-phase mixture of M_C and T at low temperatures, in the approximate ratio 55:45 at both 20 and 200 K, and predominantly tetragonal at room temperature. The evolution of the lattice parameters is plotted in Fig. 7 showing for clarity only those phases with a volume fraction in excess of 40%. The volume fraction of the T phase increases from about 45% at 200 K, to 80% at 300 K, and is essentially 100% at 325 K. Thus, as in the case of 35PT, there is a temperature interval in which the relative fractions of M_C and T phases are found to vary, which has been indicated as a shaded region in Fig. 7. The lattice parameters for the T phase at 300 K and for the T and

TABLE II. Lattice parameters at 20 K for the PMN-xPT compositions studied. The symmetries of the different phases (S) and their volume fractions (f) are indicated in the second and third columns, respectively.

x (%)	S	f (%)	a (Å)	b (Å)	c (Å)	$\alpha(=\gamma)$ (°)	β (°)
30	R	100	4.015	4.015	4.015	89.82	89.82
31	R	30	4.020	4.020	4.020	89.85	89.85
31	M	70	4.023	3.997	4.032	90	90.32
33	M	75	4.018	3.995	4.039	90	90.32
33	T	25	3.994	3.994	4.062	90	90
35	M	40	4.011	3.990	4.049	90	90.34
35	O	35	4.030	3.990	4.030	90	90.34
35	T	25	3.988	3.988	4.067	90	90
37	M	55	4.015	3.985	4.039	90	90.28
37	T	45	3.989	3.989	4.062	90	90
39	T	100	3.994	3.994	4.047	90	90

M_C phases at 20 K, are also shown in Tables I and II, respectively. Because of peak overlap, the lattice parameters of the minority monoclinic phase at 300 K could not be determined. The vertical dotted line denotes the temperature of the T - C phase transition obtained, as previously noted, from the mean values of the two temperatures reported in Ref. 19.

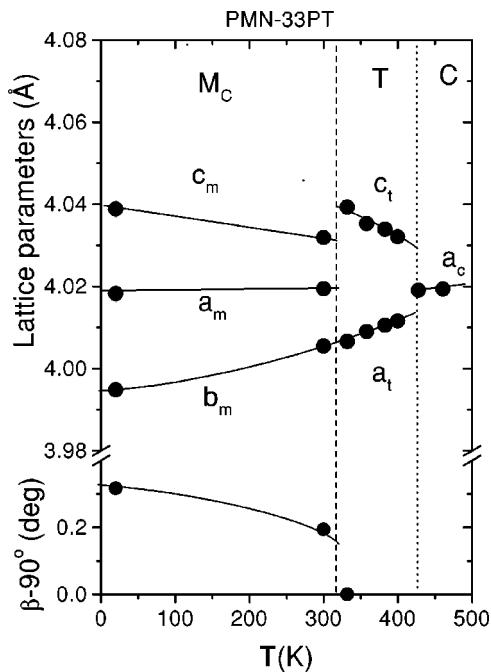


FIG. 5. Temperature evolution of the lattice parameters of PMN-33PT. The lattice parameters of the minority tetragonal phase between 20 and 300 K (see Tables I and II) have been omitted for the sake of clarity. The dashed line indicates the M_C - T transition temperature. The dotted line represent the T_C derived from the results of Noblanc *et al.* (Ref. 19), as described in the text. Except when indicated, the error bars are smaller than or comparable to the symbols.

IV. COMPOSITION DEPENDENCE

Figures 8 and 9 show the evolution of the monoclinic phase between the rhombohedral and tetragonal regions as a function of composition at 300 and 20 K, respectively. As in previous figures, only the values of the majority phases have been plotted for clarity, except for 37PT at 20 K, where the volume fractions of the two phases are roughly the same. The lattice parameters reported in Ref. 25 for PMN-34%PT are also plotted in Fig. 8, and are seen to be in good agreement with the present data. The region of stability of the M_C phase lies approximately between 31PT and 35PT at 300 K, as reported by Singh *et al.*,²⁵ and it is found to extend slightly further at 20 K, to 37PT. The trends in the lattice parameters show similar features at both temperatures: c_m increases and b_m decreases with increasing Ti content, while a_m and (β -90°) remain approximately constant at 300 K, and decrease only slightly at 20 K. Although there are relatively large errors in the determination of the lattice parameters, we can nevertheless conclude that the transition between the rhombohedral and monoclinic states is rather abrupt, as expected on symmetry grounds.³¹ More unexpectedly, the transition between the monoclinic and tetragonal phases also seems to be fairly abrupt, implying that there is no continuous rotation of the polarization all the way to the tetragonal state with increasing Ti content.

Based on the above-presented analysis, we have constructed a new phase diagram for PMN-xPT in the vicinity of the MPB, as shown in Fig. 10, in which the stability region of the M_C phase has been shaded. The solid line indicating the transition to the cubic phase has been adapted from Ref. 19 as the average of the two temperatures, T_m (maximum) and T_d (depoling), reported by Noblanc *et al.*,¹⁹ as previously mentioned. These transition temperatures were found to be in better agreement with our data (solid-symbols) than those reported in Ref. 5. To take into account the observed phase coexistence, a dotted vertical line and horizontal dotted arrows have been included in the phase diagram, to indicate the rhombohedral ($x \leq 32\%$) or tetragonal ($x \geq 32\%$) symmetry of the secondary phase.

V. DISCUSSION

It is interesting to note that the x-ray profiles shown in Fig. 2(c) for 33PT are clearly similar to those in Fig. 1 of Ref. 25 for 34PT. In particular, there appears to be definite evidence of a low-angle shoulder in the pseudocubic (200) profile, and a high-angle shoulder in the (220) profile, both of which would imply the presence of a minority tetragonal phase in the latter sample.

Recent work by Topolov³² shows that stress fields may play a very important role at the MPB. It therefore seems plausible that local internal strain due to lattice mismatching between coexisting phases is the cause of the observed broadening of the pseudocubic (h00) profile in the M_C phase. This type of strain would produce a distribution of a_m and c_m lattice parameters that could explain the striking profiles seen for the 35PT sample at 20 K and could include the limiting case of an orthorhombic phase for which $a_m = c_m$. The fact

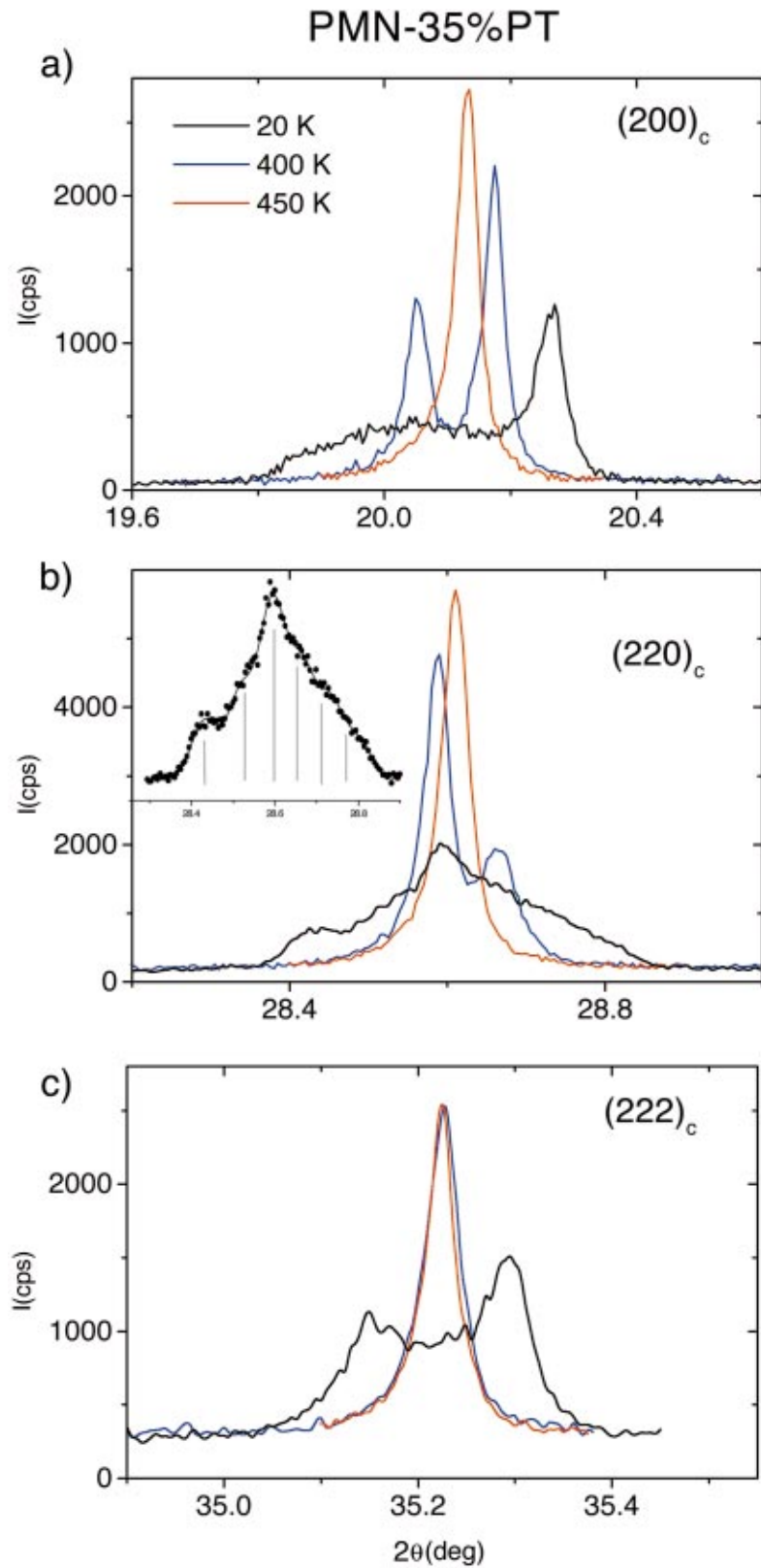


FIG. 6. (Color) Selected regions of the diffraction pattern of PMN-35PT in the cubic (450 K), tetragonal (400 K), and monoclinic (20 K) phases. The inset shows the data points at 20 K around (220), together with the fit (solid line). The peak positions obtained from the fit are indicated by the vertical lines.

that such a distribution is less evident in the other profiles [see Figs. 6(b) and 6(c)] is because the corresponding peak splittings are defined mainly by the monoclinic angle and the average values of a_m and c_m for $(hh0)$, and exclusively by those values for (hhh) . It is also apparent that the existence

of long-range compositional fluctuations about $\pm 1\%$, indicated by Williamson–Hall plots for the cubic phases, as earlier described, can only partially explain the observed distribution of monoclinic lattice parameters.

Furthermore, in the context of the three-phase coexistence

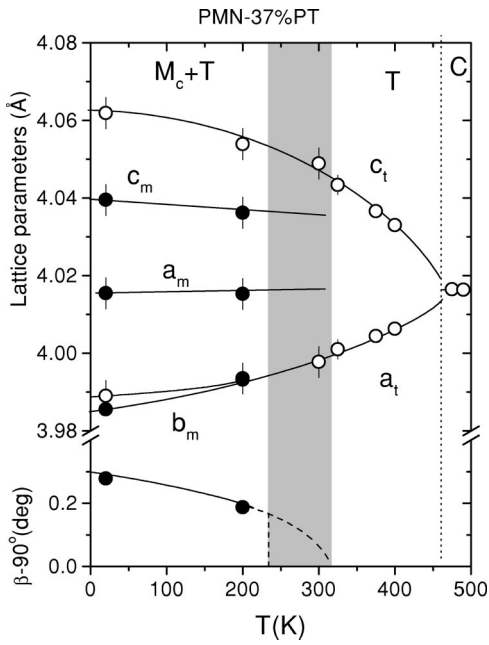


FIG. 7. Temperature evolution of the monoclinic and tetragonal lattice parameters of PMN-37PT. The temperature interval in which the volume fraction of the monoclinic (tetragonal) phase gradually decreases (increases) is shown by the shaded region. The dotted line represents the T_C derived from the results of Noblanc *et al.* (Ref. 19), as described in the text. Except when indicated, the error bars are smaller than or comparable to the symbols.

observed in 35PT at low temperatures, it is interesting to note that a degenerate point at which these three phases, T , O , and M_C coexist has been predicted by Vanderbilt and Cohen⁸ in an eighth-order expansion of the free energy. Although this degeneracy is actually an artifact of the model, and could be removed by considering higher-order terms in the Devonshire expansion, it is reasonable to suppose that these three phases are extremely close in energy and can in fact coexist in real systems due to internal strain, thermal fluctuations, etc. The phase diagram reported in Ref. 8 may explain why the MPB of the PMN-xPT system (around the T - M_C - O “degenerate” point) is considerably more complicated than that of PZT⁷ (around the T - M_A - R tricritical point) or that of PZN-xPT¹⁷ (around the R - O - T tricritical point). Burton *et al.*³³ have reported a slightly lower disorder energy in PMN than in PZN. Moreover, this disorder can be expected to increase at the MPB of the PMN-xPT system due to the larger Ti content, and could also contribute to the exceptional anharmonicity of the free-energy surfaces,⁸ a feature believed to be related to the nature of the Pb-O bond in these lead oxide solid solutions.³⁴

There is the question of whether the observed phase coexistence, $R+M_C$, $T+M_C$, or even $T+M_C+O$ is an intrinsic feature of the PMN-xPT system. The previously noted degree of compositional fluctuation might by itself explain the coexistence of M_C and R phases in 31PT, but not the coexistence of M_C and T over a much wider range. According to the eighth-order Devonshire expansion of the free energy,⁸ a phase transition between R and M_C cannot occur directly and is only possible *via* an O phase.³⁵ In this work,

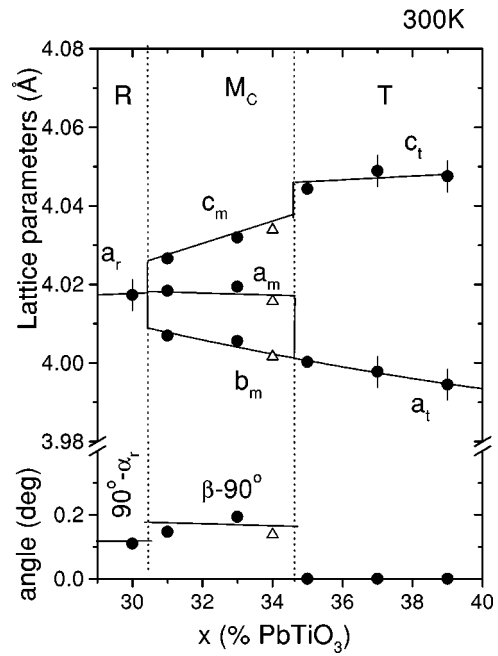


FIG. 8. Composition dependence of the lattice parameters of the majority phases in PMN-xPT around the MPB at 300 K. Except when indicated, the error bars are smaller than or comparable to the symbols. The open triangles represent the data from Ref. 25.

we have observed that 31PT is very close to orthorhombic, and it seems entirely plausible that compositions might be found between 30PT and 31PT which are truly orthorhombic, in which case the transition between R and M_C will indeed occur *via* the O phase.

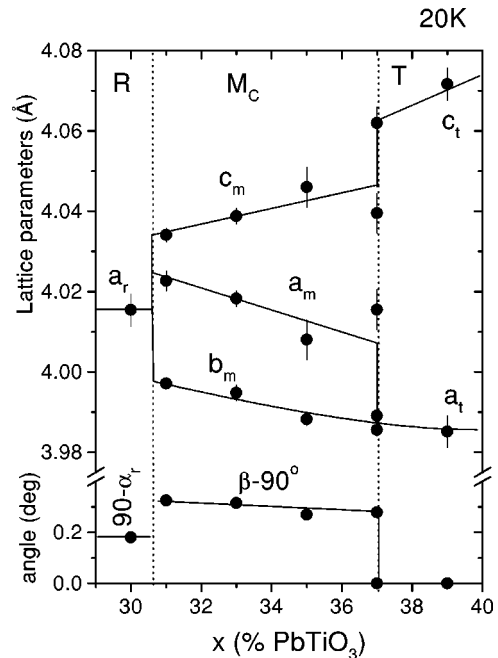


FIG. 9. Composition dependence of the lattice parameters of the majority phases in PMN-xPT around the MPB at 20 K. Except when indicated, the error bars are smaller than or comparable to the symbols.

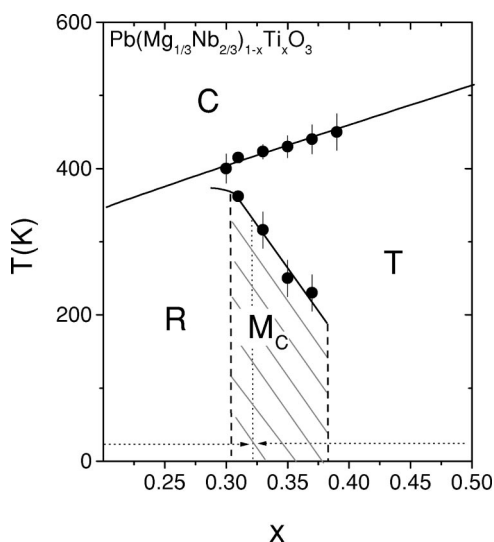


FIG. 10. Modified phase diagram of PMN-xPT around the MPB. The solid line indicating the transition to the cubic phase is the average of the two temperatures reported by Noblanc *et al.* (Ref. 19) from dielectric measurements. The symbols separating the M_C and T phases represent the temperatures at which the M_C - T phase transition begins to take place.

Furthermore, according to the eighth-order theory,⁸ the M_C - T phase transition should be second-order, and no phase coexistence region should be observed. However, the O - T phase transition is first-order, and therefore the presence of an O phase could explain the wide region of M_C + T phase coexistence observed. In fact, the existence of a minority third O phase, similar to that found in 35PT cannot necessarily be ruled out in 31PT, 33PT, and 37PT, although in these cases we are unable to resolve it from the monoclinic phase. On the other hand, it is also possible that the continuous character of the M_C - T transition may be changed by considering higher-orders in the free energy expansion,⁸ since the Landau conditions for second-order transitions that are fulfilled in this case are necessary but not sufficient.³¹ This would be more in accordance with the fairly abrupt changes

in the lattice parameters, in particular the order parameter (β - 90°), observed at the transition as a function of composition, and the M_C - T phase coexistence could be naturally explained.

To conclude, a new phase diagram for PMN-xPT has been proposed which delimits the the composition range of the M_C phase near the MPB. The existence of a secondary phase, either tetragonal or rhombohedral, has been found in all cases ($31\% \leq x \leq 37\%$). Moreover, the presence of a third minority orthorhombic phase, observed for 35PT, cannot be ruled out in the other monoclinic compositions, and could explain the observed phase coexistence of M_C and T . This complex landscape of phase mixtures, which is in agreement with the observations of Xu *et al.* for 33PT single crystals,²³ cannot be explained simply by long-range compositional fluctuations in the sample and is believed to be an intrinsic feature of the PMN-xPT system. The evolution of the monoclinic lattice parameters as a function of composition shows that the M_C phase evolves from the O limit and approaches the T limit, indicating that the polarization rotates in the monoclinic (010) plane. However, the rotation appears to be discontinuous at both limits. The present results show that the MPB of the PMN-xPT system is characterized by multi-phase components and complex phase behavior. We hope that our observations will stimulate further experimental and theoretical work needed to clarify more precisely the nature of the MPB and its relationship to the electromechanical properties.

ACKNOWLEDGMENTS

We would like to thank A. Bokov and M. Dong for their help in sample preparation and B. Burton for useful discussions. Research was carried out at the National Synchrotron Light Source, Brookhaven National Laboratory, which is supported by the U.S. Department of Energy, Division of Material Science and Division of Chemical Science. Financial support from the U.S. Department of Energy under Contract No. DE-AC02-98CH10886 and U.S. Office of Naval Research Grant No. N00014-99-1-0738 is also gratefully acknowledged.

*Present address: Condensed Matter Physics, Vrije Universiteit, De Boelelaan 1081 HV, Amsterdam, The Netherlands. Electronic address: noheda@nat.vu.nl

¹S.-E. Park and T.R. Shrout, *J. Appl. Phys.* **82**, 1804 (1997).

²D. Viehland, A. Amin, and J.F. Li, *Appl. Phys. Lett.* **79**, 1006 (2001).

³B. Jaffe, W. R. Cook, and H. Jaffe, *Piezoelectric Ceramics* (Academic, London, 1971).

⁴J. Kuwata, K. Uchino, and S. Nomura, *Ferroelectrics* **37**, 579 (1981).

⁵T. Shrout, Z.P. Chang, N. Kim, and S. Markgraf, *Ferroelectr. Lett. Sect.* **12**, 63 (1990).

⁶B. Noheda, J.A. Gonzalo, L.E. Cross, R. Guo, S.-E. Park, D.E. Cox, and G. Shirane, *Phys. Rev. B* **61**, 8687 (2000); B. Noheda, D.E. Cox, G. Shirane, J.A. Gonzalo, L.E. Cross, and S.-E. Park, *Appl. Phys. Lett.* **74**, 2059 (1999).

⁷B. Noheda, D.E. Cox, G. Shirane, R. Guo, B. Jones, and L.E. Cross, *Phys. Rev. B* **63**, 014103 (2001).

⁸D. Vanderbilt and M.H. Cohen, *Phys. Rev. B* **63**, 094108 (2001).

⁹R. Guo, L.E. Cross, S.-E. Park, B. Noheda, D.E. Cox, and G. Shirane, *Phys. Rev. Lett.* **84**, 5423 (2000).

¹⁰L. Bellaiche, A. Garcia, and D. Vanderbilt, *Phys. Rev. Lett.* **84**, 5427 (2000).

¹¹X.-H. Du, J. Zheng, U. Belengu, and K. Uchino, *Appl. Phys. Lett.* **72**, 2421 (1998).

¹²D.V. Taylor and D. Damjanovic, *Appl. Phys. Lett.* **76**, 1615 (2000).

¹³H. Fu and R.E. Cohen, *Nature (London)* **403**, 281 (2000).

¹⁴B. Noheda, D.E. Cox, G. Shirane, S.-E. Park, L.E. Cross, and Z. Zhong, *Phys. Rev. Lett.* **86**, 3891 (2001).

¹⁵K. Ohwada, K. Hirota, P. Rehrig, P.M. Gehring, B. Noheda, Y. Fujii, S.-E. Park, and G. Shirane, *J. Phys. Soc. Jpn.* **70**, 2778 (2001).

¹⁶D.E. Cox, B. Noheda, G. Shirane, Y. Uesu, K. Fujishiro, and Y. Yamada, *Appl. Phys. Lett.* **79**, 400 (2001).

¹⁷D. La-Orauttapong, B. Noheda, Z.-G. Ye, P.M. Gehring, J. Tou-

- louse, D.E. Cox, and G. Shirane, *Phys. Rev. B* **65**, 144101 (2002).
- ¹⁸Y. Uesu, M. Matsuda, Y. Yamada, K. Fujishiro, D.E. Cox, B. Noheda, and G. Shirane, *J. Phys. Soc. Jpn.* **71**, 960 (2002).
- ¹⁹O. Noblanc, P. Gaucher, and G. Calvarin, *J. Appl. Phys.* **79**, 4291 (1996).
- ²⁰Y. Lu, D.-Y. Jeong, Z.-Y. Cheng, Q.M. Zang, H.-S. Luo, Z.-W. Yin, and D. Viehland, *Appl. Phys. Lett.* **78**, 3109 (2001).
- ²¹Z.-G. Ye, B. Noheda, M. Dong, D.E. Cox, and G. Shirane, *Phys. Rev. B* **64**, 184114 (2001).
- ²²Z.-G. Ye and M. Dong, *J. Appl. Phys.* **87**, 2312 (2000).
- ²³G. Xu, H. Luo, H. Xu, and Z. Yin, *Phys. Rev. B* **64**, 020102 (2001).
- ²⁴J.-M. Kiat, Y. Uesu, B. Dkhil, M. Matsuda, C. Malibert, and G. Calvarin, *Phys. Rev. B* **65**, 064106 (2002).
- ²⁵A.K. Singh and D. Pandey, *J. Phys.: Condens. Matter* **13**, L931 (2001).
- ²⁶S.L. Swartz and T.R. Shrout, *Mater. Res. Bull.* **17**, 1245 (1982).
- ²⁷L.W. Finger, D.E. Cox, and A.P. Jephcoat, *J. Appl. Crystallogr.* **27**, 892 (1994).
- ²⁸G.K. Williamson and W.H. Hall, *Acta Metall.* **1**, 22 (1953).
- ²⁹A.M. Glazer, S.A. Mabud, and R. Clarke, *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.* **34**, 1060 (1978).
- ³⁰D.L. Corker, A.M. Glazer, R.W. Whatmore, A. Stallard, and F. Fauth, *J. Phys.: Condens. Matter* **1**, 6251 (1998).
- ³¹L. D. Landau and E. M. Lifshitz, *Statistical Physics I* (Pergamon, New York, 1980).
- ³²V.Y. Topolov, *Phys. Rev. B* **65**, 094207 (2002).
- ³³B.P. Burton and E. Cockayne, *Phys. Rev. B* **60**, R12 542 (1999).
- ³⁴J.-M. Kiat, G. Baldinozzi, M. Dunlop, C. Malibert, B. Bkhil, C. Menoret, O. Masson, and M-T. Fernandez-Diaz, *J. Phys.: Condens. Matter* **12**, 8411 (2000).
- ³⁵This would also be possible via a M_A phase. A triclinic phase in between the M_A and M_C phase would be likely to exist in this case (Refs. 8 and 10).