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Photocatalytic activity of TiO₂ doped with Zn²⁺ and V⁵⁺ transition metal ions: Influence of crystallite size and dopant electronic configuration on photocatalytic activity

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ABSTRACT

Anatase TiO₂ was prepared by sol-gel method through the hydrolysis of titanium tetrachloride and doped with transition metal ions like V⁵⁺ and Zn²⁺. The photocatalysts were characterized by various analytical techniques. Powder X-ray diffraction studies revealed only anatase phase for the doped samples. The band gap absorption for the doped samples showed red shift to the visible region (~456 nm) as confirmed by UV-vis absorption spectroscopy and diffuse reflectance spectral studies. The surface area of the Zn²⁺ doped samples were higher than the V⁵⁺ doped samples as observed by BET surface area measurements due to their smaller crystallite size. Scanning electron microscopy showed almost similar morphology, while energy dispersive X-ray analysis confirmed the presence of dopant in the TiO₂ matrix. The photocatalytic activities of these catalysts were tested for the degradation of Congo Red under solar light. Although both the doped samples showed similar red shift in the band gap, Zn²⁺ (0.06 at.%) doped TiO₂ showed enhanced activity and its efficiency was five fold higher compared to Degussa P-25 TiO₂. This enhanced activity was attributed to smaller crystallite size and larger surface area. Further completely filled stable electronic configuration (d¹⁰) of Zn²⁺ shallowly traps the charge carriers and detraps the same to the surface adsorbed species thereby accelerating the interfacial charge transfer process.

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1. Introduction

The use of TiO₂ photocatalyst for the degradation of organic pollutant has been studied extensively [1-5]. However the large band gap of TiO₂ requires higher energy artificial UV light for activation. The effective way to shift the absorption band gap of TiO₂ to the visible region of the solar spectrum can be significantly achieved by transition metal ion doping. Although impressive publications are available in this regard [6-10], least attempt is made towards the correlating the nature of dopant, oxidation state, crystallite size and its electronic configuration with the photocatalytic activity. In this view, the present research work focuses on the doping of Zn^{2+} (ptype dopant) and V^{5+} (n-type dopant) which has completely filled and vacant 'd' orbitals respectively into the TiO₂ lattice. The photocatalytic activities were studied under natural solar light for the degradation of Congo Red (CR) as a probe reaction. The influence of crystallite size and the dopant electronic configuration on the photocatalytic activity is explored.

2. Materials and methods

TiCl₄ was supplied from Merck chemicals. Zinc oxalate (ZnC₂O₄), ammonium vanadate (NH₄VO₃), aqueous NH₃ concentrated H₂SO₄ and CR were of analytical grade. The molecular formula of CR is $C_{32}H_{22}N_6O_6S_2Na_2$ and molecular weight is 697 and has λ_{max} at 497 nm. The structure of the CR is shown in Fig. 1.

2.1. Catalyst preparation

Anatase TiO₂ is prepared by sol gel method through the hydrolysis of TiCl₄ [11]. 25 ml of diluted TiCl₄ with 1 ml of concentrated H₂SO₄ is taken in a beaker and it is further diluted to 1000 ml. The pH of the solution was maintained at 7–8 by adding liquor ammonia. The gel obtained was allowed to settle down. The precipitate is washed free of chloride and ammonium ions. The gelatinous precipitate is filtered and oven dried at 100 °C. The finely ground powder was then calcined at 550 °C for 4.5 h. A known concentration of the metal ion solution was added to the calculated amount of TiO₂ to get the dopant concentration in the range of 0.02–0.1 at.%. The mixture is grinded in a mortar and oven dried at 120 °C for 1 h. The process of grinding and heating is repeated for four times and the powder is finally calcined at 550 °C for 4.5 h.

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Fig. 1. Structure of CR dye.

2.2. Characterization of the catalyst

The phase structure of the catalyst were studied by powder Xray diffraction (PXRD) patterns using Philips PW/1050/70/76 X-ray diffractometer which was operated at 30 kV and 20 mA with Cu K α radiation with a nickel filter at a scan rate of 2° min⁻¹. The Diffuse Reflectance Spectra (DRS) of the samples in the wavelength range of 190–600 nm were obtained by UV-vis spectrophotometer Shimadzu-UV 3101 PC UV-vis-NIR using BaSO₄ as the standard reference. The band gaps were calculated by the Kubelka–Munk method. The Fourier transform infra-red (FTIR) spectral analysis was carried out using KBr discs in the range of 4000-400 cm⁻¹ using Nicollet IMPACT 400 D FTIR spectrometer. The specific surface area of the samples were determined by Digisorb 2006 surface area Nova Quanta Chrome corporation instrument multipoint BET adsorption system. Surface morphology was analyzed by SEM analysis using JSM 840 microscope operating at 25 kV on specimen upon which a thin layer of gold had been evaporated and an electron microprobe was used in the EDX mode to obtain quantitative information of the metal ions in the TiO₂ lattice.

2.3. Photocatalytic degradation procedure

The photocatalysis using solar light was performed between 11 a.m. and 2 p.m. during the summer season (May–June) in Bangalore, India. The solar intensity in this period was maximal. The latitude and longitude are 12.58N and 77.38E respectively. The average intensity of sunlight was around 1200 W cm^{-2} . The intensity of solar light was concentrated using a convex lens and the reaction mixture was exposed to this concentrated solar light. In a typical experiment, 250 ml of 10 ppm CR solution is taken along with 100 mg of the photocatalyst and stirred in dark for

Table 1

Detailed characterization of TiO ₂ ,	Zn ²⁺ -TiO ₂ and V ⁵⁺ -TiO ₂ s	samples
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Fig. 2. PXRD pattern of photocatalysts. (A) TiO_2, (B) Zn^{2+} (0.1 at.%)-TiO_2, (C) V^{5+} (0.1 at.%)-TiO_2.

15 min to ensure equilibrium adsorption of the CR on catalyst surface. The degradation was followed by UV-vis spectroscopy using Shimadzu UV-1700 pharmaspec UV-vis spectrophotometer.

3. Results and discussion

PXRD pattern of TiO₂, Zn^{2+} doped TiO₂ (Zn^{2+} -TiO₂) and V⁵⁺ doped TiO₂ (V⁵⁺-TiO₂) is shown in Fig. 2. Both the doped catalysts showed only anatase phase, which suggests that both the dopants stabilised anatase phase irresepective of their nature, oxidation state and electronic configuration. The crystallite sizes of the

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Photocatalyst ^a	2θ of crystal plane (101) of anatase	Lattice parameters (Å)	Unit cell volume (Å) ³	Crystallite size (nm)	Surface area (m²/g)	Band gap in eV	$\lambda^{b}(nm)$
TiO ₂ (0.00%)	25.32	<i>a</i> = <i>b</i> = 3.7828 <i>c</i> = 9.5023	135.97	26.2	18	3.2	380
Zn ²⁺ -TiO ₂ (0.02%)	25.34	<i>a</i> = <i>b</i> = 3.7832 <i>c</i> = 9.5123	136.14	23.6	24	2.9	420
Zn ²⁺ -TiO ₂ (0.06%)	25.37	<i>a</i> = <i>b</i> = 3.7824 <i>c</i> = 9.5311	136.35	19.8	30	2.7	456
Zn ²⁺ -TiO ₂ (0.10%)	25.39	<i>a</i> = <i>b</i> = 3.7820 <i>c</i> = 9.5009	135.75	15.6	31	2.8	442
V ⁵⁺ -TiO ₂ (0.02%)	25.34	<i>a</i> = <i>b</i> = 3.7822 <i>c</i> = 9.5113	136.05	26.2	23	2.9	424
V ⁵⁺ -TiO ₂ (0.06%)	25.36	<i>a</i> = <i>b</i> = 3.7824 <i>c</i> = 9.5136	136.10	26.3	26	2.7	456
V ⁵⁺ -TiO ₂ (0.10%)	25.33	<i>a</i> = <i>b</i> = 3.7825 <i>c</i> = 9.5152	136.13	28.4	24	2.8	446

^a The dopant concentration is in at.%.

^b Band gap absorption of the photocatalyst estimated by UV-vis absorption spectroscopy.

samples were calculated using Scherrer's equation

$$D = \frac{k\lambda}{\beta\cos\theta} \tag{1}$$

where *k* is the constant (shape factor, about 0.9), λ is the X-ray wavelength (0.15418 nm), β is the full width at half maximum (FWHM) of the diffraction line and θ is the diffraction angle. The values of β and θ are taken for crystal plane (101) of anatase phase.

It is well known that the anatase phase is thermodynamically stable for smaller crystallites [12]. The crystallite sizes for Zn^{2+} -TiO₂ samples decreased significantly with increase in dopant concentration, while V⁵⁺-TiO₂ samples showed almost similar crystallite size compared to undoped TiO₂ (Table 1). Hence it can be concluded that Zn^{2+} dopant stabilised the anatase phase mainly by decreasing the anatase grain growth. While the higher valent vanadium ion decreases the concentration of oxygen vacancies for charge compensation preventing the nucleation necessary for rutile growth [9,13]. It is reported that in the case of metal oxides there is critical value of dispersion capacity, at values lower than which the oxide might become highly dispersed on the support without the formation of a separate crystalline phase. Since, no characteristic peak corresponding to Zn and V species are present, it can be concluded that the metal ion doping is below the dispersion capacity.

The diffraction peaks of crystal planes (101), (200) and (004) in anatase phase are selected to determine lattice parameters of the samples. The changes in lattice parameter on doping were reflected in the slight elongation of 'c' axis while 'a' and 'b' remained almost constant. Since only 'c' parameter was changing marginally, it can be concluded that the dopants occupied bcc and fcc positions in the anatase structure [14].

It is well known that under the stress or strain effects, the diffraction peaks get displaced and broadened. In the observed powder patterns, the *d*-spacing for the most intense reflection (100%) of 2θ value 25.32 was observed to be 3.5096 for undoped TiO₂. On doping, this *d*-spacing shows a slight shift to 3.5148. Since the displacement was not quite pronounced, we assume that strain induced by the dopant is too minimal and our calculation of crystallite size by Scherrer's line broadening can be justified.

UV absorption spectral studies indicated the band gap extension of the doped samples to visible region (Fig. 3). This is due to the creation of impurity levels by the dopant within the band gap states of TiO₂. The mid band gap states calculated by Kubelka–Munk plot for both Zn^{2+} and V^{5+} doped sample was found to be 2.7 eV as shown in Fig. 4, due to the similarity in the electronegativity of both the dopants [15]. The extent of red shift increased when dopant concentration was increased from 0.02 to 0.06 at.% and a slight decrease was observed for higher dopant concentration (0.1 at.%). Energy dispersive X-ray analysis confirmed the presence of dopant in the samples (Table 2) while scanning electron microscopy showed almost similar morphology for all the doped samples (Fig. 5). BET analysis showed increase in the surface area for the doped samples compared to undoped sample due to introduction of additional nucleation sites by the dopant. The surface area for the Zn²⁺ doped samples was higher compared to V⁵⁺ doped samples due to their smaller crystallite size (Table 1).

3.1. FT-IR analysis

The following conclusions were drawn from FT-IR analysis:

- 1. TiO₂ shows strong absorption bands at 484 and 563 cm⁻¹ which can be assigned to Ti–O bond in the TiO₂ lattice [16].
- 2. The bands at 3432 cm⁻¹ and 1655 cm⁻¹ can be attributed to the vibrations of surface adsorbed water molecules and Ti–OH bonding [17].



Fig. 3. UV-vis absorption spectra of the various photocatalysts. (A) TiO₂, (B) V⁵⁺ (0.02 at.%)-TiO₂, (C) V⁵⁺ (0.06 at.%)-TiO₂, (D) Zn²⁺ (0.06 at.%)-TiO₂.



Fig. 4. Kubelka–Munk plot of *X* versus wavelength for V^{5+}/Zn^{2+} (0.06 at.%)-TiO₂; where $X = (1 - R_{\infty})^2/2R_{\infty}$. The inert plot shows the DRS of catalyst.

Table 2	
EDX data for undoped TiO_2 and M- TiO_2 (M = V ⁵⁺ and Zr	2+).

Dopant concentration	V ⁵⁺ -TiO ₂		Zn ²⁺ -TiO ₂	
	Ti	V	Ti	Zn
0.02 at.%	97.92	2.08	97.94	2.06
0.06 at.%	93.98	6.02	93.94	6.06
0.10 at.%	90.05	9.95	90.04	9.96





 V^{5+} (0.06 at. %)-TiO₂

Fig. 5. SEM images of the photocatalysts.

3. The higher charge of V⁵⁺ compared to Ti⁴⁺ additionally attracts one hydroxyl group for charge compensation which results in higher concentration of surface adsorbed hydroxyl groups and water molecules compared to undoped TiO₂ [18,19]. Zn²⁺ having lower charge than the host Ti⁴⁺ ion, reduction in the concentration of hydroxyl groups were observed (Fig. 6). However the bands were highly broadened for Zn²⁺ doped samples due to their smaller crystallite size compared to V⁵⁺ doped samples [17].

4. Photocatalytic activity

It is well known that the amount of adsorption of the substrate and the number of active sites on the catalyst surface is crucial for efficient degradation [20]. The percentage of CR adsorption on



Fig. 6. FT-IR spectra of photocatalysts. (A) V^{5+} (0.06 at.%)-TiO_2, (B) TiO_2, (C) Zn^{2+} (0.1 at.%)-TiO_2.

the catalyst surface was calculated by comparing its concentration before and after stirring, using the formula

$$\frac{C_0 - C}{C_0} \times 100 \tag{2}$$

where C_0 and C are the initial and residual concentration of CR respectively. The decreasing adsorption capacity of the catalysts is of the order: Zn^{2+} (0.1 at.%)-TiO₂ > Zn^{2+} (0.06 at.%)-TiO₂ > V^{5+} (0.06 at.%)-TiO₂ > Zn^{2+} (0.02 at.%)-TiO₂ > V^{5+} (0.02 at.%)-TiO₂ > V^{5+} (0.1 at.%)-TiO₂ > TiO₂. All the doped samples showed strong adsorption capacities compared to undoped TiO₂. The photocatalytic degradation reaction of CR was carried out under natural solar light. The doped catalysts showed better activity compared to undoped TiO₂ due to its absorption ability in the visible region. Zn^{2+} (0.06 at.%)-TiO₂ showed enhanced activity compared to all the other samples and its efficiency was almost five fold higher than the Degussa P-25 TiO₂ (Table 3). Though both the doped samples showed maximum red shift in the band gap to an extent of ~456 nm, the enhanced activity of Zn^{2+} -TiO₂ compared to V^{5+} -TiO₂ samples is mainly attributed to its smaller crystallite size and also to the stable electronic configuration of Zn^{2+} .

It is well known that the photogenerated hole reacts with surface water to produce hydroxyl radicals which is a powerful oxidant for the degradation of organic pollutants. This reaction competes with the electron-hole recombination reaction.

$$\mathrm{TiO}_2 + h\nu \to \mathrm{e}^- + \mathrm{h}^+ \tag{3}$$

$$h^+ + H_2 O_{ads} \rightarrow OH^{\bullet}_{ads}$$
 (4)

$$e^- + h^+ \rightarrow recombination$$
 (5)

Thus higher activity in the case of Zn^{2+} (0.06 at.%)-TiO₂ can be attributed to the generation of excess hydroxyl radicals due to the prolonged separation of charge carriers. At this optimum concentration (0.06 at.%) of Zn^{2+} , the surface barrier becomes higher and the space charge region gets extended, leading to the efficient sep-

Table 3

Percentage degradation, process efficiency (Φ) and rate constant (k) of all the catalysts under solar light for the degradation of CR dye.

Photocatalyst ^a	Degradation of azo chromophore (%)	Process efficiency ^b $(\Phi)(\times 10^{-7})$	First order rate constant 'k' ^c (×10 ⁻² min ⁻¹)
Degussa P-25	18	2.1	0.22
TiO ₂	06	0.7	0.06
Zn ²⁺ (0.02%)-TiO ₂	44	5.2	0.45
Zn ²⁺ (0.06%)-TiO ₂	100	11.8	3.70
Zn ²⁺ (0.1%)-TiO ₂	78	9.2	1.72
V ⁵⁺ (0.02%)-TiO ₂	38	4.5	0.54
V ⁵⁺ (0.06%)-TiO ₂	53	6.2	0.89
V ⁵⁺ (0.1%)-TiO ₂	44	5.2	0.69

^a The dopant concentration is in at.%.

^b Process efficiency (Φ) = ($C_0 - C$)/t/S. C_0 is the initial concentration of the substrate and C is the concentration at time't'. ($C_0 - C$) concentration degraded in mg/l or ppm. 'T is the irradiation intensity 1200 W. 'S' denotes the solution irradiated plane surface area (176 cm²) and 't' represents the irradiation time in minutes. It is expressed in ppm min⁻¹ W⁻¹ cm⁻².

^c First order rate constant is calculated from the plot of $-\log C/C_0$ versus time.

aration of electrons and holes. For higher dopant concentration (0.1 at.%), the space charge region becomes narrow and the penetration depth of light into TiO_2 lattice greatly exceeds the barrier width. Thus, the absorbed photon generates electron-hole pairs in the bulk of TiO_2 beyond the space charge region which is field free and hence readily undergoes recombination. Hence it can be concluded that inclusion of optimum concentration of dopant can restrain the recombination of charge carriers to a large extent.

4.1. Influence of electronic configuration of the dopant on photocatalytic activity

The dopant inside the TiO_2 matrix can serve as both electron and hole trap as shown below

$$M^{n+} + e^{-}_{CB} \to M^{(n-1)^{+}} \quad \text{electron trap}$$
(6)

$$M^{n+} + h_{VB}^+ \to M^{(n+1)^+} \quad \text{hole trap}$$
(7)

The energy level of $M^{n+}/M^{(n-1)^+}$ lies below the conduction band edge and the energy level of $M^{n+}/M^{(n+1)^+}$ lies above the valence band edge of TiO₂ [6].

Zinc in +2 oxidation state will have a stable full filled electronic configuration of $3d^{10}$. If Zn^{2+} is assumed to trap electron, its stable completely filled electronic configuration is disturbed. The trapped electron may thus be readily transferred to oxygen molecule to form a superoxide radical anion.

$$Zn^{2+} + e^- \to Zn^+ \tag{8}$$

$$Zn^+ + O_{2ads} \rightarrow Zn^{2+} + O_2 \cdot^- \tag{9}$$

Zn⁺ can also trap valence band hole to retain its stable electronic configuration.

$$Zn^+ + h^+ \to Zn^{2+} \tag{10}$$

Alternatively if Zn^{2+} ions are assumed behave as hole trap, its stable full filled electronic configuration is again disturbed. Thus the trapped hole can be easily transferred to hydroxyl anion adsorbed on the surface forming hydroxyl radical or it can also be transferred to adsorbed dye molecule to form a dye radical.

$$Zn^{2+} + h^+ \rightarrow Zn^{3+} \tag{11}$$

 $Zn^{3+} + OH^{-} \rightarrow Zn^{2+} + OH^{\bullet}$ (12)

$$Zn^{3+} + dye_{ads} \rightarrow Zn^{2+} + dye$$
 (13)

 Zn^{3+} can also trap conduction band electron and gets reduced to Zn^{2+} to retain its stable electronic configuration.

$$Zn^{3+} + e^- \rightarrow Zn^{2+} \tag{14}$$

V⁵⁺ can also shows similar mechanism

$$V^{5+} + e^- \to V^{4+}$$
 (15)

$$V^{4+} + O_{2ads} \to V^{5+} + O_2 \cdot^- \tag{16}$$

$$V^{5+} + h^+ \to V^{6+}$$
 (17)

$$V^{6+} + OH^- \rightarrow V^{5+} + OH^{\bullet} \tag{18}$$

$$V^{5+} + dye_{ads} \rightarrow V^{4+} + dye \tag{19}$$

Tian et al. have reported the similar trapping and detrapping mechanism for V⁴⁺ doped TiO₂ photocatalyst under UV/solar irradiation. The V⁴⁺ traps hole and electron and forms V⁵⁺ and V³⁺ respectively [21]. However, it is still vital that completely filled 'd' orbitals are more stable than the vacant 'd' orbitals. Hence Zn²⁺ doped samples may shallowly trap the charge carriers, while V⁵⁺ doped samples may deeply trap the charge carriers. The above processes accelerate the interfacial charge transfer process leading to the excess generation of free radicals (superoxide or hydroxyl radicals). Therefore the pre-requisite condition for an effective dopant lies not only in trapping the charge carriers to the surface of the catalyst.

4.2. Influence of crystallite size on photocatalytic activity

The shallow/deep trap of the photogenerated charge carriers by the photocatalyst samples critically depends on its crystallite size. For pure TiO₂, the charge carrier recombination process may be of two ways: (a) volume recombination (b) surface recombination. Volume recombination is a dominant process in large TiO₂ particles. Since V⁵⁺-TiO₂ samples possess large crystallite size, the average diffusion path length of charge carrier from the bulk to the surface becomes longer which results in deep trapping of charge carriers leading to its lower activity.

However Zn^{2+} -TiO₂ samples have smaller crystallite size compared to V⁵⁺-TiO₂ samples. The crystallite sizes of Zn^{2+} -TiO₂ samples are 26.2, 23.6, 19.8 and 15.6 nm for the dopant concentration of 0.0, 0.02, 0.06 and 0.1 at.% respectively. Reduction in crystallite size leads to larger surface area which increases the available surface active sites consequently resulting in higher photonic efficiency. Further the smaller crystallite size of the sample reduces the diffusion path length for the charge carriers from the site where they are photogenerated, to the surface site where they react. Reduction in this diffusion path length results in a reduced recombination rate of photogenerated charge carriers and hence results in greater efficiency. Hence higher photocatalytic activity is observed for Zn^{2+} (0.06 at.%)-TiO₂.

At higher dopant concentration (0.1 at.%), the crystallite size becomes small (15.6 nm) and most of the charge carriers are generated sufficiently close to the surface. As a result, the photogenerated charge carriers may quickly reach the surface resulting in faster recombination. This is also due to the excess trapping sites in the sample and lack of driving force to separate the charge carriers. Further interfacial charge transfer process will be outweighed by surface recombination rate for smaller crystallites. As Zn^{2+} can serve as both electron/hole trap, the possibility of trapping both the charge carriers will be high at higher dopant concentration and trapped charge carrier may recombine through quantum tunnelling [22]. Moreover, at a high dopant concentration, the charge carrier so that their mobility becomes extremely low and undergoes recombination before it can reach the surface. Therefore there is a need for optimal dopant concentration in the TiO_2 lattice to get effective crystallite size for higher photocatalytic efficiency. Beyond the optimum dopant concentration, the rate of recombination dominates in accordance with the equation:

$$K_{\rm RR}\alpha\,\exp\left(\frac{-2R}{a_0}\right)\tag{20}$$

where K_{RR} is the rate of recombination, *R* is the distance between the trap sites of photogenerated electron and hole pairs, a_0 is the hydrogenic radius of the wave function for the charge carrier [6]. From the above equation, it is clear that beyond optimum concentration of the dopant and with the reduction of distance between the trap sites, the rate of recombination increases exponentially leading to the decrease in photocatalytic activity which is in consistent with the experimental observations.

5. Conclusion

Anatase TiO₂ was prepared by sol-gel method and it was doped with transition metal ions Zn^{2+} and V^{5+} . The photocatalytic activities of these catalysts were tested under solar light for the degradation of CR as a probe molecule. Zn^{2+} (0.06 at.%)-TiO₂ catalyst showed enhanced activity probably due to the completely filled stable electronic configuration (d¹⁰) of Zn^{2+} which shallowly traps the charge carriers and detraps the same to surface adsorbed species thereby accelerating the interfacial charge transfer process. The small crystallite size of the sample reduces the diffusion path length for the charge carriers from the bulk to the surface which effectively prevents the recombination process.

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