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Lund Institute of Technology

PROPERTIES OF MATERIALS AT HIGH TEMPERATURES STEEL

Edited by

Y Anderberg

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RILEM-COMMITTEE 44-PHT

BEHAVIOUR OF STEEL AT HIGH

TEMPERATURES

by

Yngve Anderberg

February 1983

Members of the committee:

H.L. Malhotra (Chairman), United Kingdom

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- B. Alexander, United Kingdom
- D. Cluzel, France
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- J. Lindner, Poland
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- U. Schneider, Kassel, FGR

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FOREWORD

During mid-seventies international discussions indicated that research into fire and fire behaviour of structures was being pursued by a number of international organizations; a Commission of CIB (W14) was studying basic fire phenomena and had been asked by the liaison committee of the international engineering associations to provide a common approach for the design of structures against fire. Various materials committees were starting work on design codes and CIB/W14 set up a Code Advisory Panel to provide some general principles. A directorate of the EEC Commission started activity on the harmonization of fire test standards and fire safety codes. The fire test standards were prepared by the ISO Technical Committe TC 92. One obvious ommission in this system was the absence of agreed data on material properties which could be used in the structural design codes. An approach was made to the RILEM Permanent Committee, as the General Council was known at that time, to initiate a technical committee to deal with this topic. Approval was given in 1977 for the formation of Committee 44 - PHT and it met for the first time in April 1978. The active membership of the committee came from many countries as the list on the inside front page shows. Over a period of 5 years the committee held 10 meetings, including a special seminar in the middle of 1982 in the Hague.

When examining the range of materials to be studied, maximum emphasis was placed on the major materials of construction, i.e. concrete, masonry, steel and wood but others having an important role in the building construction were also included. However it was necessary to keep the work within manageable proportions and the maximum effort was devoted to the study of concrete and steel. The list of materials examined is as follows:

1997 - NAS

Concrete; dense and lightweight, Steel; structural, reinforcing and prestressing, Wood; soft and hardwoods, Masonry; brick and concrete blocks, Plactics; constructional, Gypsum; plaster and board.

The amount of data available on concrete and steel was large and it required much effort on the part of the specialist panels to consolidate the information. For steel this task was undertaken by Dr. Y. Anderberg who also edited the final document. The amount of data was so extensive that it became difficult for the RILEM secretariat with its limited resources to publish it. Lund Institute of Technology has kindly come to our assistance and agreed to publish the information so that it may become available to individuals and associations who are interested in this field. I wish to express my personal thanks and those of the RILEM Secretariat to Dr. Anderberg and the Division of Building Fire Safety and Technology for this most valuable help.

The document is unique in presenting information for the first time in this collective form and in a way that direct comparisons can be made. It has also highlighted areas where further research may be successfully carried out. Determination of the physical properties under transient heating conditions and their expression as a mathematical model will assist fire engineers in the design and analysis of structures exposed to fire conditions.

I appreciate the valuable help given by all committe members of 44 - PHT during our meeting in various host institutes and my only regret is that we were not able to complete the work on some of the other materials. Data on concrete is being published simultaneously. No doubt another RILEM technical committee will be able to take in on board.

H.L. Malhotra Chairman 44 - PHT

Content

Preface

Part I: Introduction

		Pa	age
1	Backgro	bund	1
2	Differe	ent methods of testing	2
2.1	Steady	state tests	3
	2.1.1	$\sigma - \varepsilon$ relationship, stress rate controlled	4
	2.1.2	$\sigma_{-\epsilon}$ relationship, strain rate controlled	4
	2.1.3	Creep	5
	2.1.4	Relaxation	5
2.2	Transient tests		5
	2.2.1	Failure temperature, total deformation	б
	2.2.2	Restraint forces	6

Part II: Steel properties

3	Particular properties	18
3.1	Tensile strength	18
3.2	Modulus of elasticity	40
3.3	Thermal strain	48
3.4	Creep strain	51
3,5	Stress-strain characteristics	57
	3.5.1 Steady state data	57
	3.5.2 Transient state data	77
3.6	Relaxation	93

		Page
3.7	Thermal properties	97
3.8	Residual properties	102
4	A review and comparative study of different properties	108
5	Analytical modelling	114
5.1	Models	114
5.2	Comparison test and calculation	116
5.3	How to compare results obtained at different	
	test procedures	127

Part III: Steel literature

6	References	0
6.1	Particular properties	I
	6.1.1 General properties	I
	6.1.2 Tensile strength	VII
	6.1.3 Modulus of elasticity	XI
	6.1.4 Thermal strain	XIII
	6.1.5 Creep strain	XIV
	6.1.6 Stress-strain characteristics	XVI
	6.1.6.1 Steady state tests	XVI
	6.1.6.2 Transient tests	XVIII
	6.1.7 Compressive strength	XIX
	6.1.8 Relaxation	XX
	6.1.9 Thermal properties	XXI
	6.1.10 Residual properties after cooling	XXIII
6.2	Structural behaviour	xxv
6.3	Analytical models	XXVI
	6.3.1 Material behaviour	XXVI
	6.3.2 Structural behaviour	XXVII
7	Symbols and abbreviations	XXIX

Preface

The objectives of the technical committee are to identigy those properties and characteristics, which are of inportance, survey the existing knowledge, evaluate the data and the relevant literature, suggest activity in areas not already covered and <u>provide facilities for</u> <u>the dissemination</u> of information in papers and by discussion.

- 1 -

In this report the main features of the behaviour of steel at high temperature are summarized and discussed. The report is divided into three parts:

Part I - INTRODUCTION Part II - STEEL PROPERTIES

Part III - STEEL LITERATURE

Part I: INTRODUCTION

1 Background

Many investigations on mechanical properties of steels at high temperatures have been reported during the last 30 years. However, the results are sometimes hard to interpret and to compare with other investigations due to the fact that

- test specimens are different (variation in chemical composition, shape and size)
- · descriptions are incomplete
- test procedures differ
- · test conditions are not comparable
- · unsatisfactory accuracy and reliability of testing equipment.

Therefore, it is useful to try to clarify and elucidate the main characteristics and features of steel behaviour and the different test methods used.

A careful distinction is made between stabilized temperature tests and transient tests, and also between hotrolled and cold-worked reinforcing steel and structural and prestressing steel. These steels behave in many ways differently. The most important properties are discussed and a comparative study is made of test results from different sources. The paper also briefly discusses analytical models of steel. In many papers one agrees to the fact that in a deformation process of steel the total strain can be separated into three components, i.e. thermal strain, instantaneous stress-related strain and creep strain. These strains are also discussed separately.

In the list of references the properties of steel are divided into a number of main headings following the same sequence as in chapter 3. Further, each property includes subheadings, which separate the three categories of steel, i.e. structural steel, reinforcing steel and prestressing steel. A few references to other materials such as cast iron are included. The reference list also contains sections on structural behaviour and analytical models of steel and steel structures.

2 Different methods of testing

There exist two main groups of tests, steady state tests and transient tests. Material properties measured are closely related to the test method used. It is therefore of great importance that the test conditions are well defined.

During a fire situation the material is normally subjected to transient processes with varying temperature and stress, and to understand this, transient tests are needed. Steady state tests cannot be used solely to explain the stress-strain characteristics of steel in all cases.

-- 2 --

Mechanical properties of steel can be established by following a number of different test procedures. The three main test parameters are the heating process, application and control of load, and control of strain. These can have constant values or be varied during testing, giving steady state or transient conditions depending on the heating procedure.

Six practical regimes which can be used for determining mechanical properties are illustrated in Fig. 1. Properties in these regimes are as follows:

steady state tests

- stress-strain relationship (stress rate controlled)
- stress-strain relationship (strain rate controlled)
- creep
- relaxation

transient tests

- · total deformation, failure temperature (stress control)
- total forces, restraint forces (strain control).

2.1 Steady state tests

Steady state tests are characterized by a heating period, $t_{\rm H}$, and a period of time, $t_{\rm S}$, during which the temperature of the specimen is stabilized before any load is applied, as shown in Fig. 2. The time $t_{\rm S}$ depends on the size of the specimen, but is usually not more than 0.5 h. The strain measured before the load is applied corresponds to the thermal expansion.

After time $t_{\rm H}$ + $t_{\rm S}$ the temperature is kept constant and four types of steady state tests are usually carried out as indicated above. Properties related to these tests are described below.

2.1.1 J-E relationship, stress rate controlled

- 4 -

After the stabilized period of heating, the specimen is subjected to a constant rate of loading until failure occurs (see Fig. 3a). The data obtained provide σ - ϵ relationships at different temperatures of the type shown in Fig. 3b and can be used to establish compressive or tensile strength, modulus of elasticity and ultimate strain at collapse. For prestressing steels, however, there exists no yielding plateau as indicated in Fig. 3b. The stress-strain relationship is often obtained at a high rate of loading, $\dot{\sigma}$, (loading time 1-2 minutes) in order to avoid the influence of creep, which is of importance above about 400° C for ordinary steel (for cold-worked steel above about 250° C). The influence of creep results in a displaced σ - ϵ curve and a lower ultimate (rupture) strength, which will be illustrated in chapter 4.

2.1.2 $\sigma - \epsilon$ relationship, strain rate controlled

 $\sigma - \epsilon$ relationships measured under strain rate control are closely related to the test described above. After the stabilized period of heating the specimen is loaded at a specified strain rate, $\dot{\epsilon}$, as indicated in Fig. 4a. This test yields $\sigma - \epsilon$ curves as illustrated in Fig. 4b. Depending on the strain rate, the influence of creep above a certain temperature level yields differing $\sigma - \epsilon$ curves. This will be illustrated in chapter 4.

In these kinds of tests the maximum stress is reached for a given strain and at higher strains the stress decreases somewhat with failure occurring at much greater strains than in the corresponding stress controlled $\sigma - \epsilon$ tests. The $\sigma - \epsilon$ curves obtained not only provide properties mentioned in 2.1.1, but also mechanical dissipation energy. The ultimate strain, however, is related to the maximum stress level and not to the failure state.

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2.1.3 Creep

When the desired temperature is attained in a creep test, the specimen is loaded and the load is then kept constant during the whole test period as shown in Fig. 5a. At the time $t_H + t_s$, when the load is applied, the immediate response results in an instantaneous elastic strain and thereafter a deformation as function of time, i.e. creep, takes place, which is illustrated at different temperatures in Fig. 5b. The test period of interest during a fire is usually from 2-4 hours. A typical creep curve is shown in Fig. 6 indicating that the creep process contains three phases, i.e. primary, secondary and a tertiary phase. Normally, only the initial creep, i.e. the primary and the secondary phase, are studied.

2.1.4 Relaxation

A relaxation test is closely related to a creep test, but when the specimen has been loaded the initial strain is kept constant and the decrease in stress is measured during the whole test period. Fig. 7a illustrates the whole strain history. The reduction of stress or relaxation as function of time, which is due to the creep, is shown in Fig. 7b. Normally, a period of 2-4 hours is of interest during fire situations.

2.2 Transient tests

Transient temperature tests or non-steady state tests are characterized by a varying temperature (often increasing linearly, see Fig. 8) and a simultaneous load. The load can be applied before heating or developed during heating by restraint against thermal expansion. These two types of transient tests are carried out with load and strain control, respectively. The second alternative appears only in compression. Properties related to these tests are described below.

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2.2.1 Failure temperature, total deformation

A load is applied to the specimen before heating; heating proceeds at a specified rate ($\mathring{T} = 5-50^{\circ}$ C/min) until failure occurs. Usually, the load is constant as shown in Fig. 9a. Total deformation as a function of time is recorded until the failure point, i.e. when the strain rate or total strain is approaching infinity. The temperature measured at that critical point is called the failure (critical) temperature. Typical results are illustrated in Fig. 9b for different load levels. The result is very much influenced by the rate of heating as creep cannot be avoided. The curves enable failure temperatures which provide strength/temperature relationships to be found in Fig. 9c. Similar relationships can be obtained from steady state σ - ε curves.

If one plots the total deformation minus the thermal strain, against the temperature, one obtains curves as illustrated in Fig. 9d. From these curves, $\sigma - \epsilon$ relationships can be constructed in which the creep strain corresponding to a specified heating rate is included, Fig. 9e. The heating rate has a great influence on these $\sigma - \epsilon$ curves, just as the strain rate has in steady state $\sigma - \epsilon$ tests. This is illustrated in chapter 3.5.2 and 4.

2.2.2 Restraint forces

If the deformation is kept constant during a transient heating process, restraint forces arise. The deformation can be zero or correspond to the initial strain response of load application before heating as indicated in Fig. 10a. In such a strain controlled transient test at temperatures above about 550-600[°]C, the total force (initial load + restraint force) is independent of the initial load, which can also be zero, see Fig. 10b. If the total load minus the initial load is considered, one will obtain differing curves depending on the varying influence of creep. Measurements of restraint forces can only be carried out on steel structures with no buckling influence, i.e. very short columns.

- 6 ---

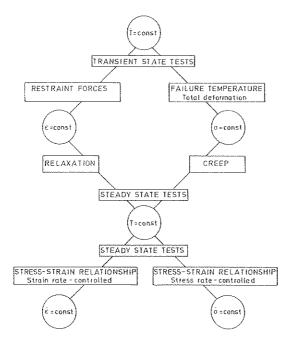
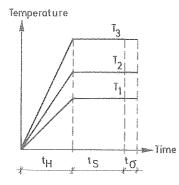


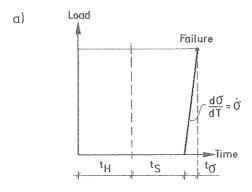
Fig 1 Different testing regimes for determining mechanical properties



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Fig 2 Typical steady state temperature curve $t_{\rm H}$ = heating period $t_{\rm S}$ = stabilizing period t_{σ} = loading period (1-3 minutes)

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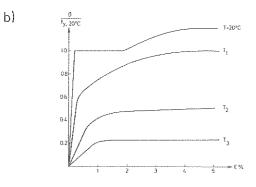
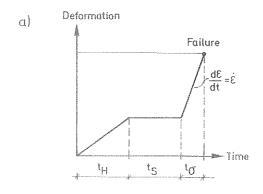
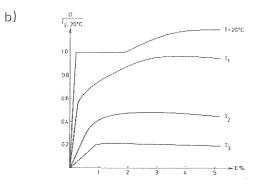
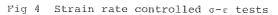


Fig 3 Stress rate controlled $\sigma - \epsilon$ tests

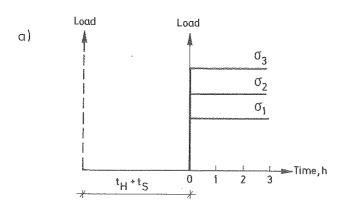
- a) Loading procedure b) Typical results







- a) Deformation procedureb) Typical results



- 11 -

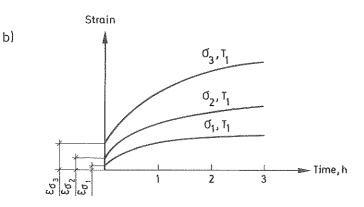


Fig 5 Creep tests

- a) Loading procedure b) Typical results ε_{σ} instantaneous stress-related strain

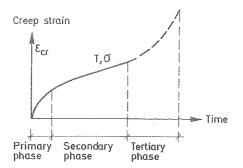
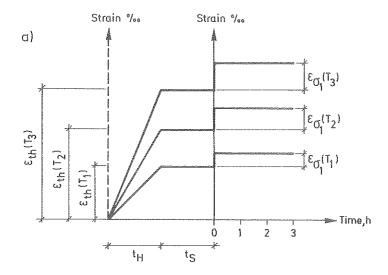


Fig 6 Typical creep curve

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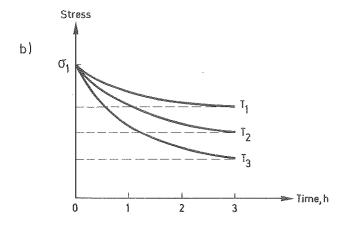
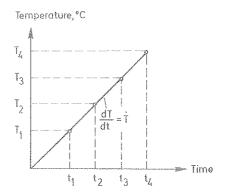


Fig 7 Relaxation tests

a) Strain history b) Typical results



- 14 -

Fig 8 Typical transient state temperature curve

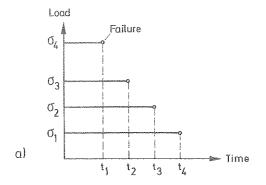
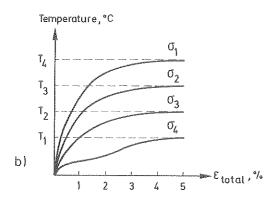


Fig 9 Transient tests with load control a) Load history

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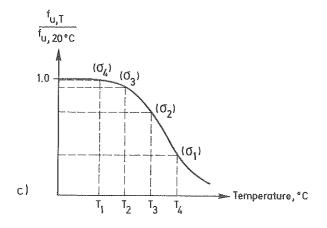
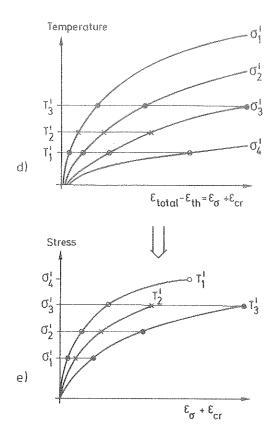


Fig 9 Transient tests with load control

- b) Typical deformation curves
- c) Typical strength-temperature relationships



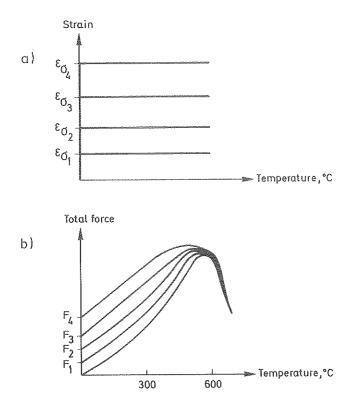
- 16 -

Fig 9 Transient tests with load control

- d) Typical $\epsilon\text{-}T$ curves at four different stress levels. $\epsilon = \epsilon_{\text{tot}} \epsilon_{\text{th}}$
- e) Calculated $\sigma-\varepsilon$ curves (including creep) at different temperatures

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- 17 -

Transient tests with strain control Fig 10

- a) Strain historyb) Typical curves of restraint forces

Part II: STEEL PROPERTIES

3 Particular properties

Properties of steel obtained or derived from different tests described above are collected in this chapter. The data are based on a literature survey and the source can always be found in the reference list in part III, chapter 6.

The presentation is based on relevant data as far as possible and the basic test conditions and the strength characteristics, the chemical composition (when available) and the reference are always given. This is a screening of individual reports on reinforcing, structural and prestressing steel, and further data may continuously appear in future.

3.1 Tensile strength

It is not only important to define the precise meaning of tensile strength, but also to define the type of test whether steady state or transient state. It can be the ultimate tensile strength, f_u (rupture strength), or the tensile strength related to a specified residual stress induced strain in the σ - τ curve as principally illustrated in Fig 11. The stress which produces for instance a residual stress, $\sigma_{0.2}$, although the 0.2% proof stress or the yield stress, $\sigma_{0.2}$, although the yielding plateau for hot-rolled steels disappears at temperatures above 300°C. When the strength is defined by the proof stress for a given residual, non-recoverable strain, for instance 0.2% and 0.5%, it will be denoted $f_{0.2}$ and f_0 5.

When the ultimate strength is measured in a transient test with constant load control it can either be related to the critical temperature when the strains reach infinite values or to a strain rate criteria, for instance $\dot{\epsilon} = 1.0 \cdot 10^{-4} \cdot \text{s}^{-1}$. If it is connected to a strain rate criteria this will be specially indicated when presenting data. The ultimate strength and proof stress are obtained from $\sigma - \epsilon$ curves at different temperature levels or from load controlled transient tests as described in chapter 2.2.1

- 18 -

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Depending on the test procedure the creep may influence the strength-temperature curve and, therefore, it must be considered when comparing such curves from different laboratories. The creep results in a decreased value of the tensile strength above about 250^{\circ}C and 400^{\circ}C for cold-worked and hotrolled steel, respectively.
```

Parameters of importance in an analysis of the tensile strength are

in steady state $\sigma - \epsilon$ tests

- rate of stress, σ, at stress rate control
- rate of strain, $\hat{\epsilon}$, at strain rate control

and in load controlled transient tests

- rate of temperature increase, T
- strain rate criteria.

Depending on the test characteristics the influence of creep is different.

A general conclusion is that the relative decrease in tensile strength of all hot-rolled and prestressing steels is almost the same, which means that the original strength has little influence on the strength-temperature curve. The 0.2% proof stress and the ultimate strength of such steels obtained in steady state and transient state tests have at $500-550^{\circ}C$ 50% of its original value left, and at $700^{\circ}C$ about 20%. For prestressing steel the 50 and 20% limits are reached at $350-400^{\circ}C$ and $500^{\circ}C$, respectively. Fig. 11 indicates that coldworked steel has a higher strength-temperature curve. Too little data are available to make a certain conclusion.

The influence of strain rate, $\dot{\epsilon} = 0.1 - 100 \cdot 10^{-3} \cdot \text{min}^{-1}$, is illustrated in Fig 8 and at 600°C the strength value varies from 0.65 to $0.3 \cdot f_{0.2,20} \circ_{\text{C}}$. In Fig 9 different rates of loading, 0.03-1.63 MPa/s, result in strength values 0.35-0.25 $\cdot f_{0.2,20} \circ_{\text{C}}$ at 600°C.

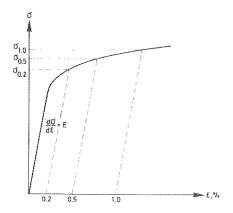
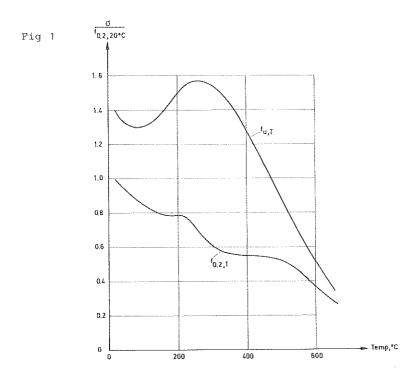


Fig 11 The principal for determining the proof stress, which is based on the residual stressinduced strain

3.1 TENSILE STRENGTH : ASTM A36, f_{0.2,20}°_C = 300 MPa a) Structural steel $f_{u,20}\circ_C = 440$ MPa Steady state : Strain rate control Test conditions $c = 72 - 102 \cdot 10^{-3} \cdot \text{min}^{-1}$ Specimen $: \phi = 5.2 \text{ mm}, \ell = 23 \text{ mm}$ C P S Mn 0.19 0.007 0.03 0.71 Chem. composition Si : 0.09 Remarks : Ultimate strength and 0.2% proof stress as function of temperature

Reference

: Harmathy & Stanzak (1970)

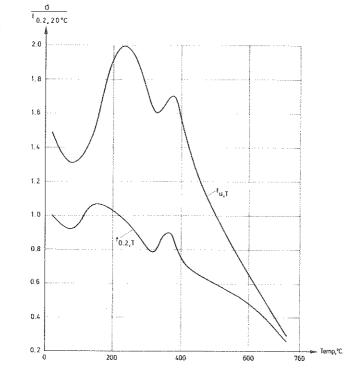


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a)	Structural steel	•	CSA G40.12, f _{0.2,20} ° _C = 340 MPa f _{u,20} ° _C = 520 MPa
	Test conditions	:	Steady state Strain rate control & = 51-76.10 ⁻³ .min ⁻¹
	Specimen	:	$\phi = 5.2 \text{ mm}, \ \ell = 23 \text{ mm}$
	Chem. composition	:	C P S Mn Si Cu 0.195 0.015 0.019 1.40 0.022 0.08 Ní Cr
	Remarks	:	0.03 0.01 Ultimate strength and 0.2% proof stress as function of temperature
	Reference	:	Harmathy & Stanzak (1970)

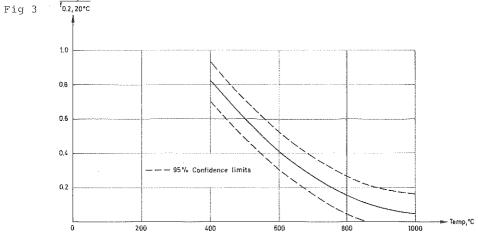




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a) Structural steel	: Different Kinds
Test conditions	: Steady state Strain rate control, $\dot{\epsilon} = 16.7 \cdot 10^{-3} \cdot \text{min}^{-3}$
Specimen	$: \phi = 6.3 \text{ mm}, \ell = 57 \text{ mm}$
Chem. composition	:
Remarks	: 0.2% proof stress for different kinds of steel as function of temperature
Reference	: Teesside Laboratories (1980)

- 23 -



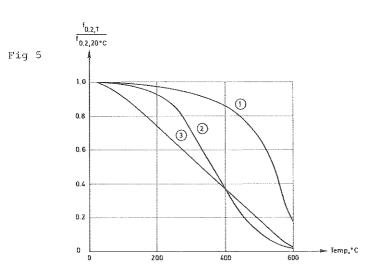
	3.1 TENSILE STRENGTH	- 24 -
		: Steel 37, f _{0.2,20} ° _C = 250 MPa
	Test conditions	: Steady state Strain rate control, $\dot{\epsilon} = 2 \cdot 10^{-3} \cdot \text{min}^{-1}$
	Specimen	$: \phi = 7.2 \text{ mm}, \ \epsilon = 71 \text{ mm}$
	Chem. composition	: C P S Mn Si Cr 0.27 0.033 0.041 0.65 0.128 0.16 ^{Ni} 0.086
	Remarks	0.086 : Proof stress related to different stress induced strains
	Reference	: Skinner, D.H. (May 1972)
Fig 4	σ 10.2, 20°C	
	2.0	
	1.8	
	1.5	
	1.4	
	1.2	
	1.0	
	0.8	<u> </u>
	0.6	
	0,4	

0.2 L

∞ Temp,*C

3.1 TENSILE STRENGTH a) Structural steel : (1) St 60/90, $f_{0.2,20}o_{C} = 590$ MPa : (2) St 145/165, $f_{0.2,20}^{\circ}c = 1520 \text{ MPa}$ c) Prestressing steel (3) St 160...180, $f_{0.2,20}o_C = 1570$ MPa Test conditions : Steady state Stress rate control, o is unknown (1) Hot-rolled, $\emptyset = 26 \text{ mm}$ (2) Cold-drawn, $\emptyset = 5.2 \text{ mm}$ (3) Cold-drawn, $\emptyset = 5 \text{ mm}$ Specimen : Chem. composition : Remarks : 0.2% proof stress as function of temperature Reference : Dannenberg et al (1959)

- 25 --

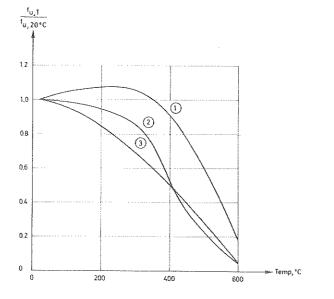


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3.1 TENSILE STRENGTH

Structural steel Prestressing steel	: ① St 60/90, f _{u,20} ° _C = 1050 MPa : ② St 145/165, f _{u,20} ° _C = 1650 MPa ③ St 160180, f _{u,20} ° _C = 1740 MPa
Test conditions	: Steady state Stress rate control, $\overset{\bullet}{\sigma}$ is unknown
Specimen	: (1) Hot-rolled, $\emptyset = 26 \text{ mm}$ (2) Cold-drawn, $\emptyset = 5.2 \text{ mm}$ (3) Cold-drawn, $\emptyset = 5 \text{ mm}$
Chem. composition	:
Remarks	: Ultimate strength (rupture strength) as function of temperature
Reference	: Dannenberg et al (1959)

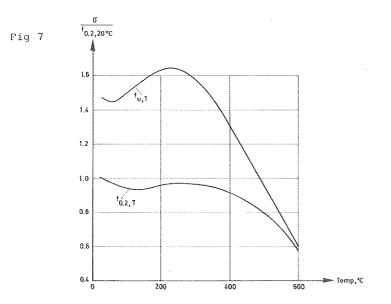




- 26 -

3.1 TENSILE STRENGTH	-
a) Structural steel	: Plate steel A36, $f_{0.2,20}O_C = 300$ MPa Low carbon $f_{u,20}O_C = 440$ MPa
Test conditions	: Steady state
Specimen	:
Chem. composition	
Remarks	: Ultimate strength and 0.2% proof stress as function of temperature
Reference	: Klippstein (1979)

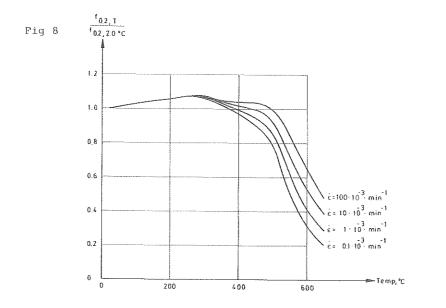
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3.1 TENSILE STRENGTH

: Steel 37, f_{0.2,20}°_C = 250 MPa a) Structural steel $f_{1,20}o_{C} = 496 \text{ MPa}$ Test conditions : Steady state Strain rate control, $e = 0.1-100 \cdot 10^{-3} \cdot \text{min}^{-1}$ Specimen $: \emptyset = 7.2 \text{ mm}, \ell = 71 \text{ mm}$ С Р S Si Mn Cr : 0.27 0.033 0.041 0.65 0.128 0.16 Chem. composition Ni. 0.086 Remarks : The influence of strain rate $\dot{\epsilon}$ on 0.2% proof stress Reference : Skinner (1972)

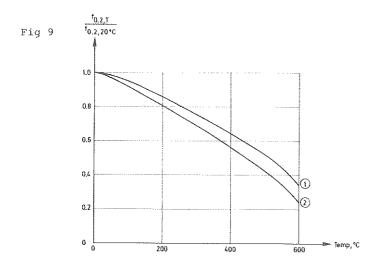


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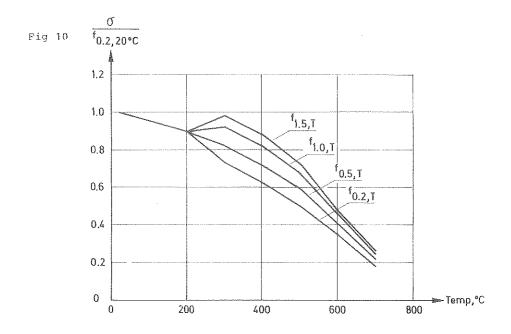
3.1 TENSILE STRENGTH	
a) Structural steel	: SIS 14 1411, f _{0.2,20} ° _C = 340 MPa
Test conditions	: Steady state Stress rate control ① o = 1.63 MPa/s ② o = 0.03 MPa/s
Specimen	: $l = 95$ mm, b = 15 mm, t = 3 mm
Chem. composition	C P S Mn Si Cr : 0.27 0.033 0.041 0.65 0.128 0.16 Ni 0.086
Remarks	: The influence of rate of loading on 0.2% proof stress
Reference	: Thor, J. (1972)

- 29 -



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3.1 TENSILE STRENGTH
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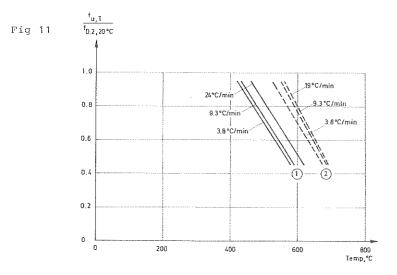
b) Reinforcing steel	: Different kinds
Test conditions	: Steady state Stress rate control, σ = 3.5 MPa/s
Specimen	: $\phi = 8-10$ mm, $t = 100$ mm Hot-rolled
Chem. composition	:
Remarks	: Proof stress related to different stress induced strains
Reference	: Anderberg (1978)



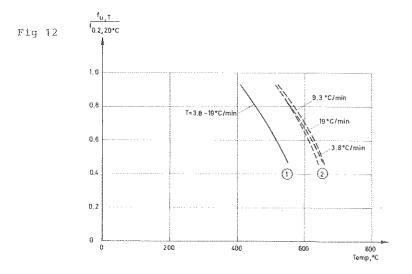
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3.1 TENSILE STRENGTH	
b) Reinforcing steel	 (1) BSt 42/50 RU, f_{0.2}, 20^oC = 430 MPa (2) BSt 42/50 RK, f_{0.2}, 20^oC = 430 MPa
Test conditions	: Transient state with load control Rate of heating, \dot{T} = 3.8 - 24 ^o C/min
Specimen	: () Hot-rolled, $\emptyset = 16 \text{ mm}$, $\ell = 10 \times \emptyset$ (2) Cold-worked, $\emptyset = 18 \text{ mm}$, $\ell = 10 \times \emptyset$
Chem. composition	:
Remarks	: Ultimate strength (rupture strength) as function of temperature
Reference	: Ruge & Winkelmann (1978-80)

- 31 -

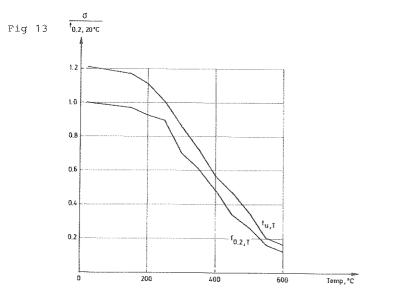


b) Reinforcing steel : (1) BSt 42/50 RU,
$$f_{0.2,20}\circ_{C} = 430$$
 MPa
(2) BSt 42/50 RK, $f_{0.2,20}\circ_{C} = 430$ MPa
Test conditions : Transient state with load control
Rate of heating, $T = 3.8 - 19^{\circ}C/min$
Failure temperature criteria, $\hat{\epsilon} =$
 $= 1.0 \cdot 10^{-4} \cdot s^{-1}$
Specimen : (1) Hot-rolled, $\phi = 16$ mm, $\ell = 10 \times \phi$
(2) Cold-worked, $\phi = 18$ mm, $\ell = 10 \times \phi$
.
Chem. composition :
Remarks : Ultimate strength as function of tempera-
ture, related to the strain rate
 $\hat{\epsilon} = 1.0 \cdot 10^{-4} \cdot s^{-1}$
Reference : Ruge & Winkelmann (1978-80)



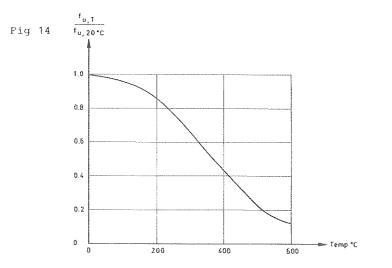
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3.1 TENSILE STRENGTH c) Prestressing steel : $f_{0,2,20}\circ_{\rm C} = 1559$ MPa $f_{u,20}^{\circ}C = 1848$ MPa Test conditions : Steady state Strain rate control, ¿ is unknown $t_{g} = 10 \text{ min}$ Specimen : Ø = 4.5 mm Chem. composition : Remarks : Ultimate strength and 0.2% proof stress as function of temperature Reference : Voves, B. (1978)

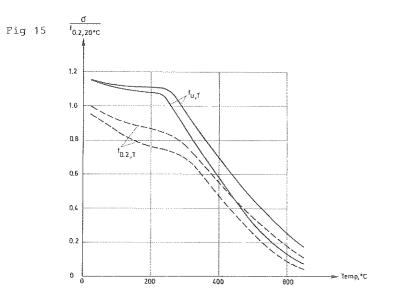


3.1 TENSILE STRENGTH

c)	Prestressing steel	: 87 E 35, f _{u,20} ° _C = 1700 MPa Fagersta steel, f _{u,20} ° _C = 1900 MPa	
	Test conditions	: Steady state Strain rate control, å = 75.10 ⁻³ .min ⁻¹	
	Specimen	: $\phi = 4.3 \text{ mm}, \ \ell = 200 \text{ mm}$	
	Chem. composition	:	
	Remarks	: Ultimate strength as function of te perature	≥m
	Reference	: Anderberg (1983)	



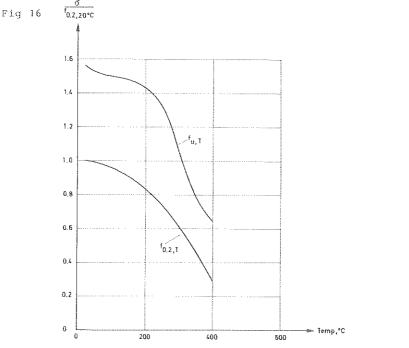
3.1 TENSILE STRENGTH	
c) Prestressing steel	: ASTM A 421, f _{0.2,20} ° _C = 1470 MPa f _{u,20} ° _C = 1720 MPa
Test conditions	Steady state : Strain rate control $c = 19-29 \cdot 10^{-3} \cdot \text{min}^{-1}$
Specimen	: $\phi = 4.4 \text{ mm}, \ \ell = 23 \text{ mm}$
Chem. composition	C P S Mn Si : 0.749 0.012 0.031 0.78 0.187
Remarks	: Ultimate strength and 0.2% proof stress as function of temperature
Reference	: Harmathy & Stanzak (1970)



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3.1 TENSILE STRENGTH c) Prestressing steel : $f_{0.2,20}\circ_{C} = 1070$ MPa $f_{u,20}\circ_{C} = 1630$ MPa Test conditions : Steady state Specimen : C P S Mn Si Chem. composition : 0.78 0.024 0.022 0.63 0.21 Remarks : Ultimate strength and 0.2% proof stress as function of temperature Reference : Cahill (1965) _____

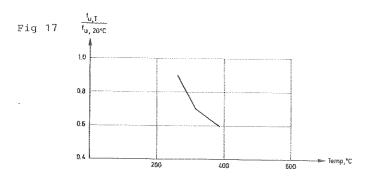
- 36 -



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3.1 TENSILE STRENGTHc) Prestressing steel:f_{0.2,20}\circ_{C} = 1559 MPa<br/>f_{u,20}\circ_{C} = 1848 MPaTest conditions: Transient state with load control<br/>Rate of heating, T = 12^{\circ}C/minSpecimen:\emptyset = 4.5 \text{ mm}Chem. composition:Remarks: Ultimate strength (rupture strength)<br/>as function of temperature
```

: Voves, B. (1978)

- 37 -



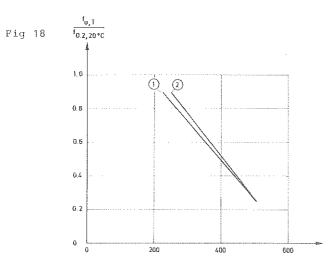
Reference

3.1 TENSILE STRENGTH

c)	Prestressing steel	: St 1570/1770, 1 f _{0.2,20} ° _C = 1570 MPa 2 f _{u,20} ° _C = 1660 MPa
	Test conditions	: Transient state with load control Rate of heating, $\hat{T} = 4^{\circ}C/min$ Failure temperature criteria, $\hat{\epsilon} = 1.0 \cdot 10^{-4} \cdot s^{-1}$
	Specimen	: (1) Cold-worked, ϕ = 7.5 mm (2) " , ϕ = 5.0 mm ℓ 10 x ϕ
	Chem. composition	: Ultimate strength as function of temperature, related to the strain rate $\epsilon = 1.0 \cdot 10^{-4} \cdot s^{-1}$
	Remarks	:

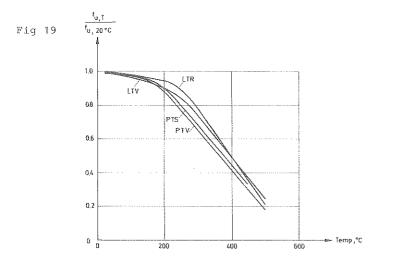
- 38 --

Reference : Ruge & Winkelmann (1978-80)



3.1 TENSILE STRENGTH	
a) Prestressing steel	: PTV f _{u,20} ° _C = 1750 MPa LTP " = 1550 MPa LTR " = 1550 MPa PTS " = 1700 MPa
Test conditions	: Transient state with load control Rate of heating, T is unknown
Specimen	:
Chem. composition	:
Remarks	: Ultimate strength (rupture strength) as function of temperature
Reference	: Baus, Brenneisen & Langueville (1968)

- 39 -



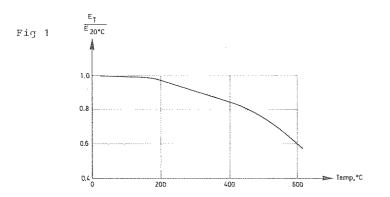
3.2 Modulus of elasticity

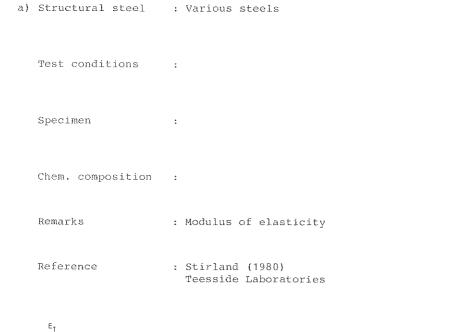
The variation of the modulus of elasticity with time, $\rm E_{T}$, can be directly obtained from $\sigma-\epsilon$ relationships as the initial inclination of the curve. If the $\sigma-\epsilon$ relationships include creep strains the value of $\rm E_{T}$ will be somewhat decreased. The test conditions are therefore of importance, when comparing results from different sources. If $\rm E_{T}$ is determined in a dynamic test the result can also differ. If $\rm E_{T}$ is obtained from constructed $\sigma-\epsilon$ curves which are related to transient tests (see chapter 3.5.2) the influence of creep cannot be avoided.

The modulus of elasticity decreases with temperature and at $500-550^{\circ}$ C the value is about $0.5 \cdot E_{20} \circ_{C}$. The original strength characteristics seems to have little influence on E_{T} . Essential results are summarized in Fig 1-7.

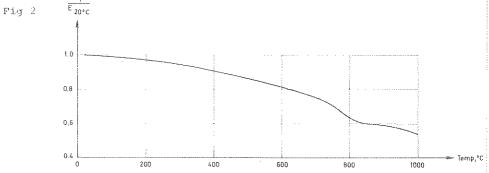
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a) Structural steel : Plate steel A36, $f_{0.2,20}o_C = 300$ MPa Low carbon $f_{u,20}o_C = 440$ MPa Test conditions : Steady state Specimen : Chem. composition : Remarks : Modulus of elasticity Reference : Klippstein (1979)



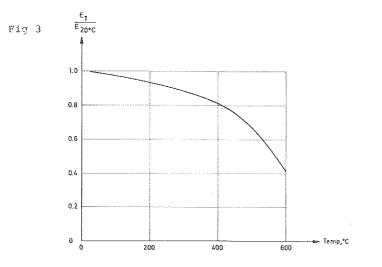


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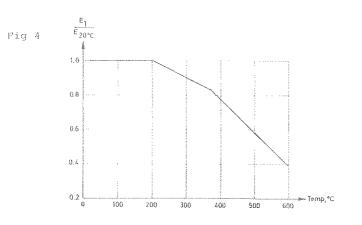
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3.2 MODULUS OF ELASTICITY
a) Structural steel : Carbon steels
Test conditions : Steady state
Specimen :
Chem. composition :
Remarks : Modulus of elasticity (mean curve from
three sources: Garofalo (1950), Lea et
al (1914), Versé (1935)
Reference : Harmathy (1967)
```

- 43 -



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a) Structural steel :
Test conditions : Steady state
Stress rate control, o = 1.63 MPa/s
Specimen :
Chem. composition :
Remarks : Modulus of elasticity
Reference : Thor, J. (1972)
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      3.2 MODULUS OF ELASTICITY

      a) Structural steel
      : Fe E 24, f_{0.2,20}^{\circ}c = 270 MPa

      Test conditions
      : Transient state<br/>Constant load control<br/>Rate of heating, T = 10°C/min

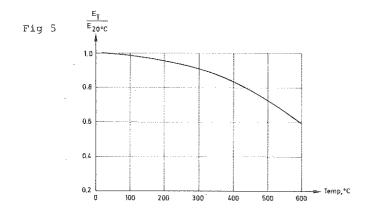
      Specimen
      :

      Chem. composition
      :

      Remarks
      : Modulus of elasticity

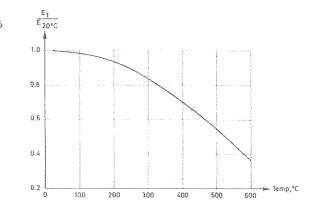
      Reference
      : Copier, W.J. (1972)
```

- 45 -



3.2 MODULUS OF ELASTICITY

b)	Reinforcing steel	: Different kinds, Ks 40, Ks 60
	Test conditions	: Steady state Stress rate control, σ = 3.5 MPa/s
	Specimen	: Hot-rolled, \emptyset 8-10, ℓ = 100 mm
	Chem. composition	•
	Remarks	: Modulus of elasticity (mean curve)
	Reference	: Anderberg (1978)

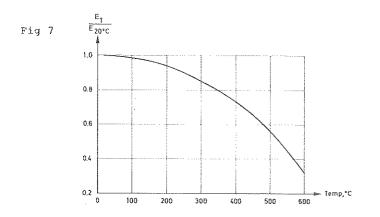


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Fig 6

3.2 MODULUS OF ELASTICITY

c) Prestressing steel : St 1570/1770, $E_{20}O_C = 2.14 \cdot 10^5$ MPa Test conditions : Steady state Rate of stress, \dot{o} is unknown Specimen : Cold-worked, $\phi = 7.5$ mm Chem. composition : Remarks : Modulus of elasticity Reference : Ruge & Winkelmann (1978-80)

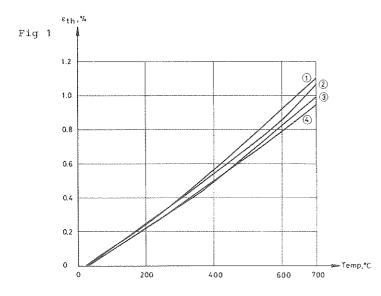


3.3 Thermal strain

The thermal strain or thermal expansion is measured on unloaded specimens in a transient test. Investigations published in literature indicate small deviations for structural and prestressing steels. Type of steel and strength characteristics seem to have no significant influence. In Fig 1 the thermal strain for structural steel is taken from four different sources. The curves are relatively close together. For reinforcing steels there is only one source of information available which indicates lower thermal strain than other steels as shown in Fig 2. In this figure the results from a prestressing steel are also indicated.

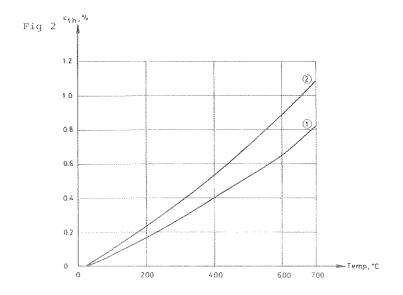
3.3 THERMAL STRAIN

Fig 1 Thermal strain (expansion) for structural steel as function of temperature () St 37-2, Ruge & Winkelmann (1978-80) (2) Steel 37, Skinner (1972) (3) A 36, Harmathy (1967) (4) Stirland (1980)



3.3 THERMAL STRAIN

- Fig 2 Thermal strain (expansion) for (1) reinforcing and (2) prestressing steel as function of temperature
 - () Anderberg (1978)
 - ② Ruge & Winkelmann (1978-80)



3.4 Creep strain

Creep behaviour is unique for every type of steel and a common description is hard to find. This is due to the fact that the chemical composition and the degree of processing strongly influences this. The creep tendency seems not to be related to the 0.2% proof stress or other strength characteristics at room temperature. Therefore the absolute value of stress is used when describing creep analytically. Analytical creep models will be presented in chapter 5. The creep starts to be of importance at about 250° C for cold-worked steel and at about 400° C for hot-rolled steel.

The creep strain can only be directly measured in steady state tests and if the stress is kept constant it can be separated into two phases as illustrated in Fig 6. However, the creep from steady state tests can be used in order to predict the creep process in transient tests, which is illustrated in chapter 5.

Measurements on creep at high temperatures are very complicated and the accuracy and reliability of the testing equipment are of decisive importance for the results. In Fig 5 creep results are shown from two identical tests on specimens from the same heat. The creep strain after 1.5 h indicates a difference of about 20%. If the specimens are not from the same heat deviation may rise to 40%. The scatter in the results are not only due to the testing equipment but also to the difference in shape and processing of the specimen.

Some results on creep published in literature are illustrated in Fig 1-5. More results for analytical studies are given in chapter 5.

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3.4 CREEP STRAIN
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a) Structural steel : Al 149, $f_{0.2,20}\circ_C = 250$ MPa X^{-60} , $f_{0.2,20}\circ_C = 450$ MPa Test conditions : Steady state Constant load control Specimen :

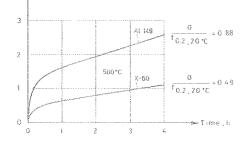
Chem. composition :

Remarks : Creep at 500⁰C for different load levels

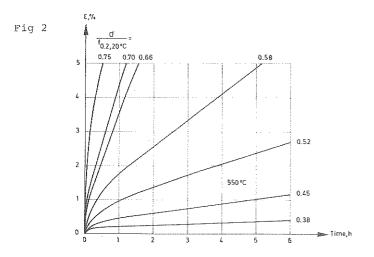
Reference : Knight et al (1971)



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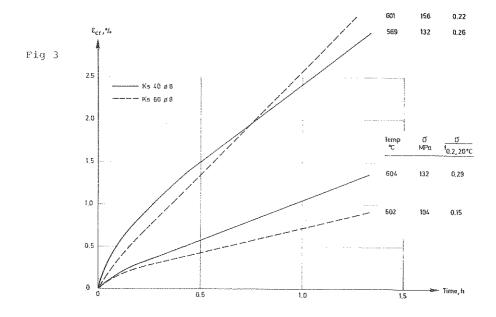


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a)	Structural steel	: Steel 37, $f_{0.2,20}\circ_C = 250 \text{ MPa}$ $f_{u,20}\circ_C = 496 \text{ MPa}$
	Test conditions	: Steady state Constant load control
	Specimen	$: \phi = 7.2 \text{ mm}, \ \varrho = 71 \text{ mm}$
	Chem. composition	C P S Mn Si Cr : 0.27 0.033 0.041 0.65 0.128 0.16 Ni 0.086
	Remarks	: Creep at 550 ⁰ C for different load levels
	Reference	: Skinner (1972)



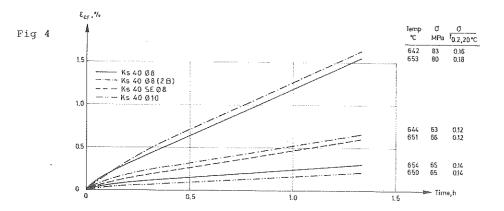
3.4 CREEP STRAIN

b}	Reinforcing steel	:	Ks 40, f _{0.2,20} ° _C = 456 MPa Ks 60, f _{0.2,20} ° _C = 710 MPa
	Test conditions	;	Steady state Constant load control
	Specimen	:	$\emptyset = 8 \text{ mm}, \& = 100 \text{ mm}$ Hot-rolled
	Chem. composition	:	C P S Mn Si 0.41 0.029 0.039 0.66 0.27
	Remarks	:	Creep at about 600 ⁰ C for different load levels
	Reference	:	Anderberg (1978)



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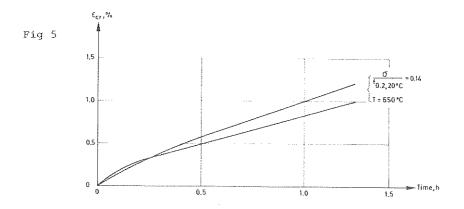
3.4 CREEP STRAIN b) Reinforcing steel : Ks 40, different kinds Test conditions : Steady state Constant load control Specimen : Ø = 8mm, l = 100 mm Hot-rolled Chem. composition : Remarks : Creep at about 650^OC for different hot-rolled steels Reference : Anderberg (1978)



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3.4 CREEP STRAIN
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b)	Reinforcing steel	2	Ks 40, f _{0.2,20} ° _C = 483 MPa
	Test conditions	0	Steady state Constant load control
	Specimen	e .	$\emptyset = 10 \text{ mm}, \ \epsilon = 100 \text{ mm}$ Hot-rolled
	Chem. composition	:	C P S Mn Si 0.31 0.024 0.021 0.75 0.27
	Remarks	:	Deviation in results in two creep tests with specimens from the same heat



3.5 Stress-strain characteristics

The data presented are divided into two categories, namely steady state and transient state properties. A survey of the most important properties related to the test procedure is given.

3.5.1 Steady state data

 σ - ϵ relationships obtained under strain rate or stress rate control are illustrated in eighteen (18) diagrams for the three types of steel: structural, reinforcing and prestressing steel. All curves are drawn dimensionless and are related to $f_{0,2,20}\sigma_{C}$.

The stress-strain relationship is often obtained at a high rate of stress or a high rate of strain (test duration period less than 1-2 min for structural and reinforcing steel and less than 1/2 min for prestressing steel) in order to avoid the influence of creep which is of importance above about 250 and 400°C for hot-rolled and cold-worked steels, respectively. At these higher temperatures a slow stress rate or strain rate (test duration period is more than about 1 h for structural and reinforcing steel and more than about 5 min for prestressing steel) considerably influence the amount of creep observed. This results in a displaced $\sigma - \varepsilon$ curve and a lower ultimate strength.

It can be generally stated for hot-rolled steel that the ultimate strength (maximum point in the σ - ϵ curve) at 500 and 600^OC decreases to about 0.8 and 0.5 of f_{0.2,20}°_C, respectively, if the test duration period is short (high rate of stress or strain). For prestressing steel the comparative values are about 0.45 and 0.25 of f_{0.2,20}°_C.

Some comparisons have also shown that steels that are not cold-worked have very similar dimensionless $\sigma-\epsilon$ relationships at elevated temperatures.

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In the presented test results the strain rate, $\dot{\epsilon}$, has been varied from 2 to $200 \cdot 10^{-3} \cdot \text{min}^{-1}$ for structural steel and from 25 to $700 \cdot 10^{-3} \cdot \text{min}^{-1}$ for prestressing steel. This means a variation of the test duration period from about 1 h to 1 min and from about 6 min to 20 s, respectively. The influence of $\dot{\epsilon}$ can be separately studied in Figs 7 and 17.

The stress rate, $\mathring{\sigma}$, has been varied from 0.03 to 3.5 MPa/s, which means a variation of the test duration period from about 2.5 h to 15 s. The influence of $\mathring{\sigma}$ is illustrated in Figs 10, 11 and 13.

It can be generally noticed that the maximum decrease of ultimate strength due to creep effects amounts to about 15% of $f_{0.2,20}$ °C at 550-600°C except for prestressing steel.

3.5.1 STEADY STATE DATA

a) Structural steel : CSA G40.12, $f_{0.2,20}o_{C} = 340$ MPa

Test conditions : Strain rate control $\dot{\epsilon} = 51-76 \cdot 10^{-3} \cdot \text{min}^{-1}$

Specimen : $\phi = 5.2 \text{ mm}, t = 23 \text{ mm}$

Chem. composition	:			S 0.019	Mn 1.40	Si 0.022	Cu 0.08
		N 0.03	Cr 0.01				
Remarks :		σ-ε re	lations	hips			

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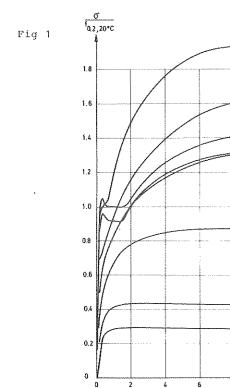
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Reference

: Harmathy & Stanzak (1970)

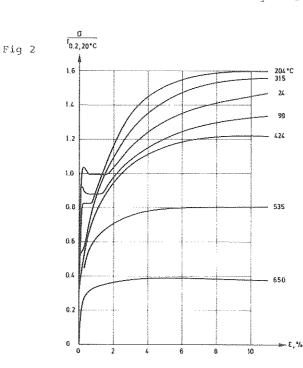


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3.	5.1 STEADY STATE DAY	ΓA	
a)	Structural steel	:	ASTM A36, $f_{0,2,20}^{o}c = 300 \text{ MPa}$
	Test conditions	:	Strain rate control $\dot{\epsilon} = 72 - 102 \cdot 10^{-3} \cdot \min^{-1}$
	Specimen	4	$\phi = 5.2 \text{ mm}, \ \epsilon = 23 \text{ mm}$
	Chem. composition	:	C P S Mn Si 0.19 0.007 0.03 0.71 0.09
	Remarks	:	σ-ε relationships

- 60 -

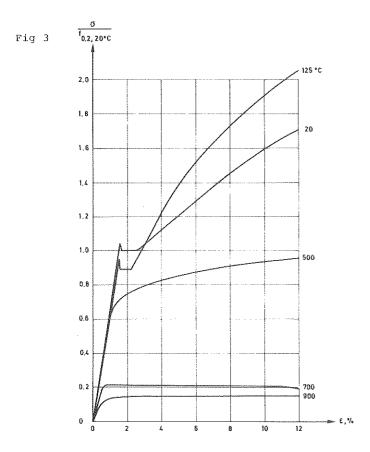
Reference	:	Harmathy	&	Stanzak	(1970)
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3.5.1 STEADY STATE DATA

a) Structural steel : Grade 43A, $f_{0.2,20}o_C = 260 \text{ MPa}$ Test Conditions : Strain rate control, $\epsilon = 17 \cdot 10^{-3} \cdot \text{min}^{-1}$ Specimen : $\emptyset = 6.3 \text{ mm}, \ \ell = 57 \text{ mm}$ Chem. composition : Remarks : $\sigma - \epsilon$ relationships Reference : Teesside Laboratories (1980)

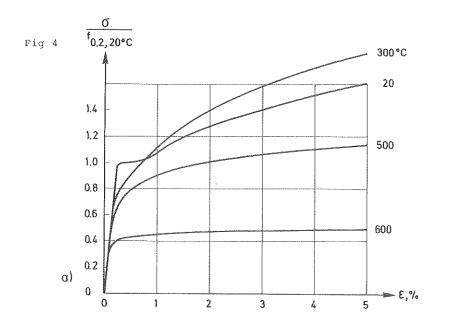


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3.5.1 STEADY STATE DATA
a) Structural steel : Steel 37, f_{0.2,20}o_C = 253 MPa
                               : Strain rate control
    Test conditions
                                 \dot{\epsilon} = 2 \cdot 10^{-3} \cdot \text{min}^{-1}
    Specimen
                               : \emptyset = 7.2 \text{ mm}, \ell = 71 \text{ mm}
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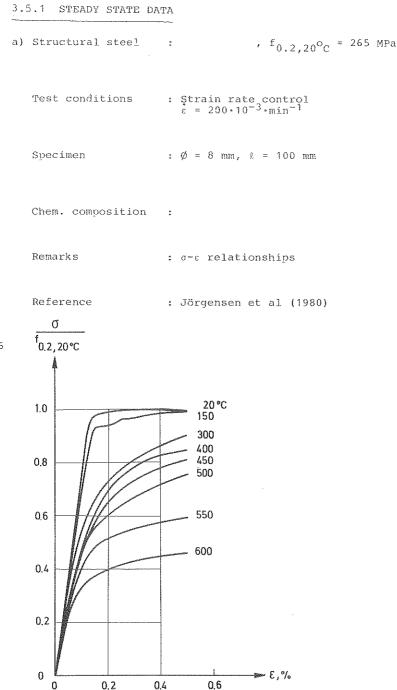
Chem. composition	:	0.27	Р 0.033	S 0.041	Mn 0.65	Si 0.128	Cr 0.16
		N1 0.086					
Remarks		σ-ε relationships					

Reference : Skinner (1972)



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3.5.1 STEADY STATE DATA
        a) Structural steel
                                   :
                                                       f_{0.2,20}^{\circ}c = 265 \text{ MPa}
            Test conditions
                                      : Strain rate control c = 5 \cdot 10^{-3} \cdot \text{min}
            Specimen
                                     : Ø = 8 mm, l = 100 mm
            Chem. composition :
            Remarks
                                     : σ-ε relationships
            Reference
                                     : Jörgensen et al (1980)
               Q
            <sup>1</sup>0.2, 20°C
Fig 5
                                            150 °C
                                            20
300
            1.0
                                            400
                                            450
            8.0
                                            500
            0.6
                                            550
            0,4
                                            600
            0.2
            0
                                                       - €,%
               0
                         0.2
                                   0,4
                                              0,6
```

- 63 -

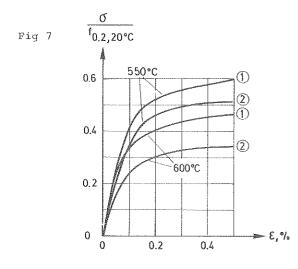


- 64 -

Fig 6



a) Structural steel : $f_{0.2,20^{\circ}C} = 265 \text{ MPa}$ Test conditions : Strain rate control (1) $\dot{\varepsilon} = 200 \cdot 10^{-3} \cdot \min^{-1}$ (2) $\dot{\varepsilon} = 5 \cdot 10^{-3} \cdot \min^{-1}$ Specimen : $\phi = 8 \text{ mm}, \ \ell = 100 \text{ mm}$ Chem. composition : Remarks : Influence of strain rate on $\sigma - \varepsilon$ relationship att 550 and $600^{\circ}C$ Reference : Jörgensen et al (1980)



방문 이 전에 관

3.5.1 STEADY STATE DATA

```
3.5.1 STEADY STATE DATA

a) Structural steel : SIS 141411, f_{0.2,20}o_C = 340 MPa

Test conditions : Stress rate control, \mathring{\sigma} = 1.63 MPa/s

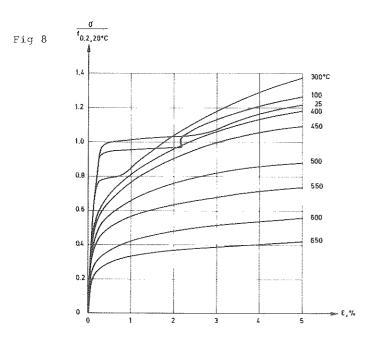
Specimen : \& = 95 mm, b = 15 mm and t = 3 mm

Chem. composition :

Remarks : \sigma - \varepsilon relationships
```

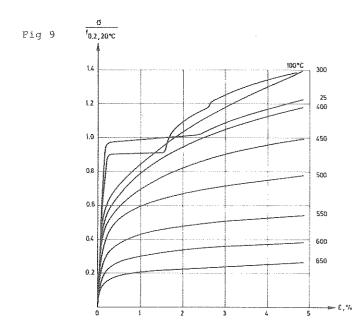
- 66 -

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Reference : Thor, J. (1972)
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-67 -
3.5.1 STEADY STATE DATA
a) Structural steel : SIS 141411, f_{0.2,20}o_C = 340 MPa
Test conditions : Stress rate control, \dot{o} = 0.03 MPa/s
Specimen : \ell = 95 mm, b = 15 mm and t = 3 mm
Chem. composition :
Remarks : \sigma - \epsilon relationships
Reference : Thor, J. (1972)
```

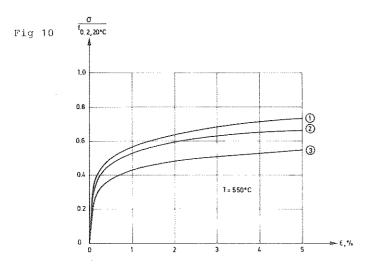


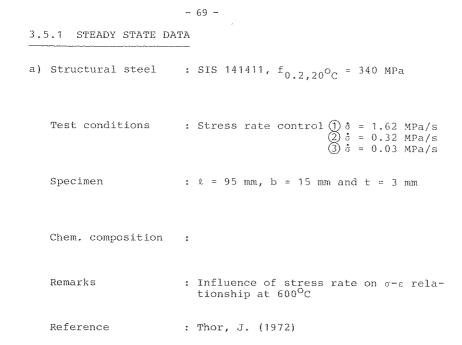
```
3.5.1STEADY STATE DATAa) Structural steel: SIS 141411, f_{0.2,20}o_C = 340 MPaTest conditions: Stress rate control (1) \dot{\sigma} = 1.61 MPa/s<br/>(2) \dot{\sigma} = 0.32 MPa/s<br/>(3) \dot{\sigma} = 0.03 MPa/s<br/>(3) \dot{\sigma} = 0.03 MPa/s<br/>SpecimenSpecimen: \ell = 95 mm, b = 15 mm and t = 3 mmChem. composition:: Influence of stress rate on \sigma - \epsilon relationship at 550^{\circ}c
```

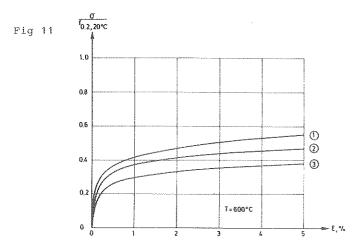
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- 68 -

Reference	:	Thor,	J.	(1972
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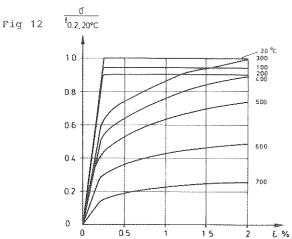




b)	Reinforcing steel	:	Different kinds
	Test conditions	:	Stress rate control, d=3.5 MPa/s
	Specimen	:	$\phi = 8-10 \text{ mm}, \ell = 100 \text{ mm}$ Hot-rolled
	Chem. composition	:	
	Remarks	:	σ - ϵ relationships

- 70 -

Reference : Anderberg (1978)



 3.5.1 STEADY STATE DATA

 b) Reinforcing steel
 : Different kinds

 Test conditions
 : Stress rate control $\sigma = 3.5$ and 0.07 MPa/s

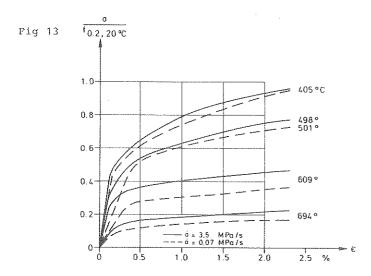
 Specimen
 : $\phi = 8-10$ mm, t = 100 mm Hot-rolled

 Chem. composition
 :

 Remarks
 : Influence of stress rate on $\sigma - \varepsilon$ relationship at different temperatures

 Reference
 : Anderberg (1978)

- 71 -



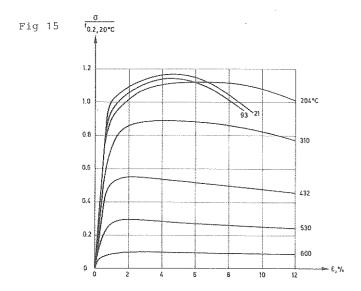
c) Prestressing steel : ASTM A421, $f_{0.2,20}o_C$ = 1470 MPa $f_{u,20}o_{c} = 1720$ MPa Test conditions : Strain rate control $\dot{\epsilon} = 19 - 29 \cdot 10^{-3} \cdot \text{min}^{-1}$ Specimen $: \phi = 4.4 \text{ mm}, \ell = 23 \text{ mm}$ Chem. composition : C P S Mn Si 0.749 0.012 0.031 0.78 0.187 Remarks : σ-ε relationships Reference : Harmathy & Stanzak (1970)

Ø Fig 14 10.2.20°C 1.2 204 °C 1.0 93 21 0.8 315 0.6 429 0,4 0.2 540 654 ٥ а 2 4 6 8 10 12

3.5.1 STEADY STATE DATA

c)	Prestressing steel	:	ASTM A421, $f_{0.2,20^{\circ}C} = 1470 \text{ MPa}$
			f _{u,20} o _C = 1720 MPa
	Test conditions	:	Strain rate control $\dot{\epsilon} = 56-98 \cdot 10^{-3} \cdot \text{min}^{-1}$
	Specimen	:	Ø = 4.4 mm, & = 23 mm
	Chem. composition	:	C P S Mn Si 0.794 0.012 0.031 0.78 0.187
	Remarks	:	σ~ε relationships
	Reference	:	Harmathy & Stanzak (1970)

- 73 -



경험 것은 물리고 있다.

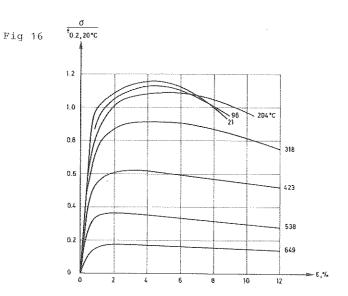
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3.5.1 STEADY STATE DATA

c) Prestressing steel : ASTM A421, $f_{0.2,20}\circ_{C} = 1470$ MPa $f_{u,20}\circ_{C} = 1720$ MPa Test conditions : Strain rate control $\dot{\epsilon} = 370-700 \cdot 10^{-3} \cdot \text{min}^{-1}$ Specimen : $\phi = 4.4$ mm, $\ell = 23$ mm Chem. composition : C = P = S = Mn = Si = 0.794 = 0.012 = 0.031 = 0.78 = 0.187Remarks : $\sigma - \epsilon$ relationships

Reference

: Harmathy & Stanzak (1970)

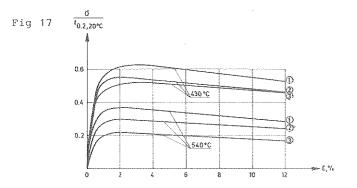


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3.5.1 STEADY STATE DATA

c) Prestressing steel	: ASTM A421, f _{0.2,20} ° _C = 1470 MPa f _{u,20} ° _C = 1720 MPa
Test conditions	: Strain rate control (1) $\dot{\epsilon} = 550 \cdot 10^{-3} \cdot \min^{-1}$ (2) $\dot{\epsilon} = 80 \cdot 10^{-3} \cdot \min^{-1}$ (3) $\dot{\epsilon} = 25 \cdot 10^{-3} \cdot \min^{-1}$
Specimen	$\phi = 4.4 \text{ mm}, 2 = 23 \text{ mm}$
Chem. composition	C P S Mn Si : 0.749 0.012 0.031 0.78 0.187
Remarks	: Influence of strain rate on $\sigma^{-\epsilon}$ relationship at 430 and 540 $^{\rm O}C$
Reference	: Harmathy & Stanzak (1970)



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3.5.1 STEADY STATE DATA

c) Prestressing steel : , f_{0.2,20}\circ_C = 1559 MPa

f_{u,20}\circ_C = 1848 MPa

Test conditions : Steady state

Strain rate control, \dot{\epsilon} is unknown

t_s = 10 min

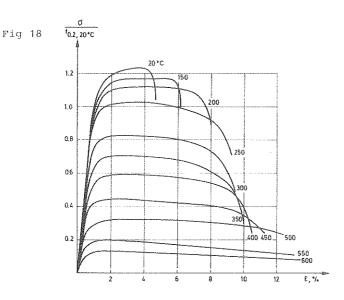
Specimen : \phi = 4.5 mm

Chem. composition :

Remarks : \sigma \sim \epsilon relationships

Reference : Voves, B. (1978)
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- 76 -



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3.5.2 Transient state data

In this chapter two kinds of data will be presented namely deformation as function of temperature and constructed $\sigma - \varepsilon$ relationships as described in section 2.2.1. The deformation can be total or total deformation minus initial elastic strain and/or thermal strain. Results are presented in fifteen (15) diagrams with constructed $\sigma - \varepsilon$ curves in Figs 6, 7, 8 and 15.

The deformation-temperature curves are very much influenced by the rate of heating, \mathring{T} , and in the presented results it has been varied from 1.7 to 50° C/min. For protected steel the rate of heating is in most cases between 5 and 10° C/min.

Some of the results indicate an instability phase of the steel which suddenly results in increased deformation and give a more or less marked plateau in the diagram, see Figs 1, 3, 9, 12, 13 and 14. This deformation instability occurs at the load level $0.6-0.9 \cdot f_{0.2,20} \circ_{C}$. How the rate of heating influences this behaviour is not clear.

Due to the rate of heating, the time for the creep to develop is different and this influences the total deformation. As a consequence this also makes the critical temperature dependent on the value of $\hat{\Phi}$. By studying the results presented in the diagrams, the following decrease in failure temperature is obtained:

Source	T, ^O C/min	Decrease, ^O C
Skinner (1972)	10-1.7	35
Copier (1972)	50-5	40
Ruge & Winkelmann (1978-80)	10-4	20

It may be noted that there are great differences in the deformation curves taken from different sources.

Finally, it can be noticed that in $\sigma - \varepsilon$ curves constructed from transient data a great deal of creep is included. The results for structural steel given in Figs 6 and 7 are in a relatively good agreement.

3.5.2 TRANSIENT STATE DATA : FeE 24, $f_{0.2,20}o_C = 270$ MPa a) Structural steel : Constant load control Rate of heating, $\mathring{T} = 10^{\circ}C/min$ Test conditions Specimen : Chem. composition : Remarks : Total deformation minus thermal strain as function of temperature at different load levels Reference : Copier, W.J. (1972) Temp, °C σ 10.2,20°C = 0.2 07 03 600 - 0,6 0.7 0.8 500 400 300 200 100

- 78 -

0

0

Fig 1

0.5

1,0

1.5

2.0

2.5

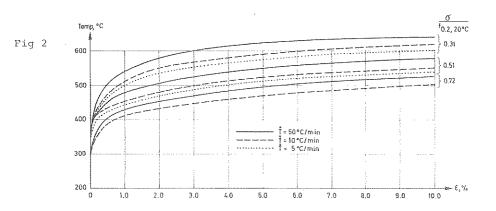
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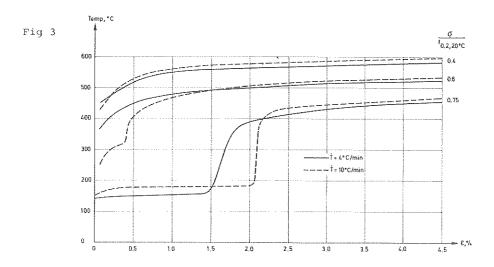
3.5.2 TRANSIENT STATE DATA

a)	Structural steel	: Fe E24, $f_{0.2,20}o_{C} = 270 \text{ MPa}$
	Test conditions	: Constant load control Rate of heating, Ť = 5,10,50 ⁰ C/min
	Specimen	:
	Chem. composition	:
	Remarks	: Influence of rate of heating on total deformation minus thermal and initial elastic strain as function of tempera- ture at different load levels
	Reference	: Copier, W.J. (1972)



```
3.5.2 TRANSIENT STATE DATA
                         : St 52, f_{0.2,20} = 480 MPa
a) Structural steel
   Test conditions
                         : Constant load control
                           Rate of heating, \dot{T} = 4 and 10^{\circ}C/min
   Specimen
                         : b = 10 \text{ mm}, 2 = 100 \text{ mm}
   Chem. composition
                       :
   Remarks
                         : Total deformation minus thermal strain
                           and initial elastic strain as function
                           of temperature at different load levels
   Reference
                         : Ruge & Winkelmann (1978-80)
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- 80 -

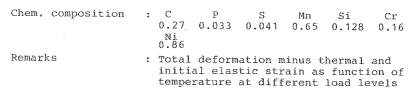


3.5.2 TRANSIENT STATE DATA

a) Structural steel : Steel 37, f_{0.2,20}°_C = 250 MPa $f_{u,20} \circ_{C} = 496 \text{ MPa}$

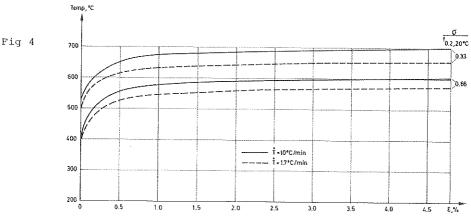
: Constant load control Rate of heating, T = 1.7 and 10° C/min Test conditions

Specimen $: \emptyset = 7.2 \text{ mm}, \ell = 71 \text{ mm}$



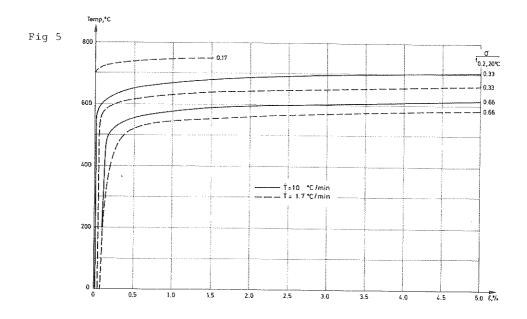
Reference

: Skinner (1972)

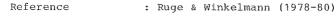


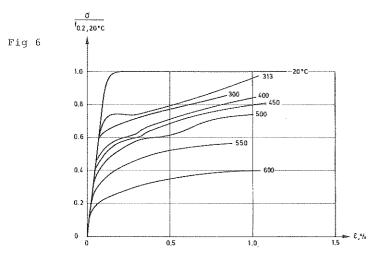
a) Structural steel	: Steel 37, $f_{0.2,20}C = 250 \text{ MPa}$ $f_{u,20}C = 496 \text{ MPa}$
Test conditions	: Constant load control Rate of heating, $\dot{T} = 1.7$ and 10° C/min
Specimen	$: \phi = 7.2 \text{ mm}, \ \ell = 71 \text{ mm}$
Chem. composition	C P S Mn Si Cr : 0.27 0.033 0.041 0.65 0.128 0.16 Ni 0.086
Remarks	: Influence of rate of heating on total deformation minus thermal strain as function of temperature at different
Reference	load levels : Skinner (1972)

- 82 -



3.5.2 TRANSIENT STATE DATA a) Structural steel : St 37-2, $f_{0.2,20}\circ_C = 255$ MPa Test conditions : Constant load control Rate of heating, $T = 10^{\circ}C/min$ Specimen : b = 10 mm, $\ell = 100$ mm Chem. composition : Remarks : Constructed $\sigma - \epsilon$ relationships



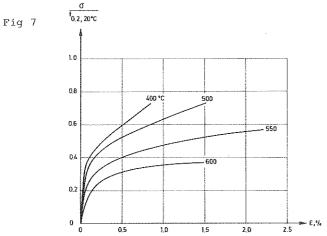


- 83 -

3.5.2 TRANSIENT STATE DATA a) Structural steel : FeE 24, $f_{0.2,20}$ c = 270 MPa : Constant load control Rate of heating, \mathring{T} = $10^{\rm O} C/min$ Test conditions Specimen : Chem. composition : Remarks : Constructed $\sigma \text{-}\epsilon$ relationships

Reference

: Copier, W.J. (1972)

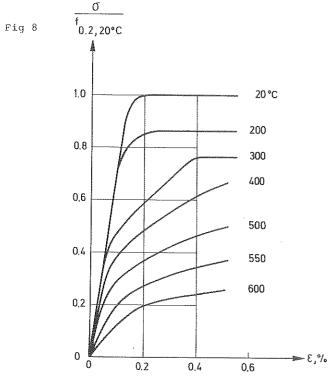


3.5.2 TRANSIENT STATE DATA

a) Structural steel : Fe E24, $f_{0.2,20}$ = 270 MPa Test conditions : Constant load control Rate of heating, $\dot{T} = 10^{\circ}C/min$ Specimen : Chem. composition : Remarks : Constructed one relationships for ECCS

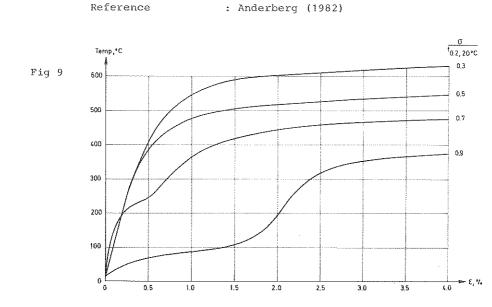
- 85 -

Reference : Witteveen, J. et al (1977)



3.	5.2 TRANSIENT STATE	DATA
b}	Reinforcing steel	: Ks 60, f _{0.2,20} ° _C = 710 MPa
	Test conditions	: Constant load control Rate of heating, $\dot{T} = 10^{\circ}C/min$
	Specimen	: $\phi = 8 \text{ mm}$, $\ell = 100 \text{ mm}$
	Chem. composition	: C P S Mn Si 0.31 0.040 0.041 1.11 0.033
	Remarks	: Total deformation minus initial elastic strain as function of temperature at different load levels
	Doference	. Indent-our (1000)

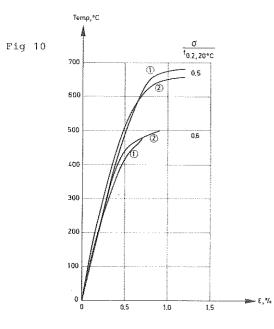
- 86 -

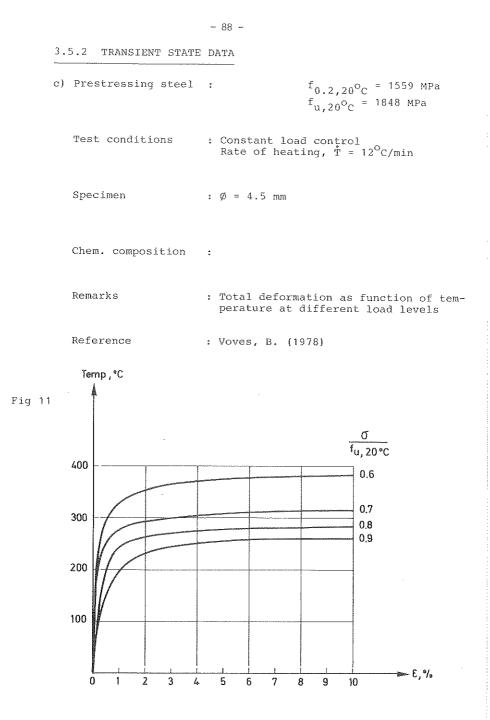


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a)	Reinforcing steel	<pre>: ① Hot-rolled, f_{0.2,20}°_C = 210-670 MPa ② Torsteel, f_{0.2,20}°_C = 430-460 MPa</pre>
	Test conditions	: Constant load control Rate of heating, $\mathring{T} = 10^{\circ}C/min$
	Specimen	: Ø = 10 mm, & = 400 mm (1) Hot-rolled (2) Cold-worked
	Chem. composition	:
	Remarks	: Total deformation minus initial elastic strain at different load levels as func- tion of temperature
	Reference	: Soretz (1967)

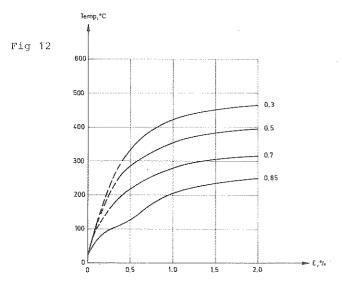
- 87 -





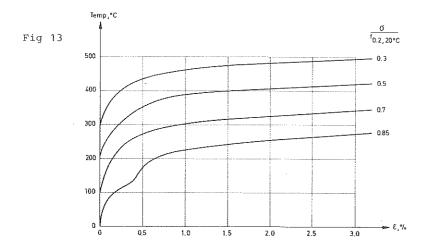
c) Prestressing steel : St 1570/1770, $f_{0.2,20}o_{C} = 1570$ MPa Test conditions : Constant load control Rate of heating, $T = 4o_{C/min}$ Specimen : Cold-drawn, $\emptyset = 7.5$ mm, $\ell = 10 \times \emptyset$ Chem. composition : Remarks : Total deformation minus initial elastic strain as function of temperature at different load levels Reference : Ruge & Winkelmann (1978-80)

- 89 -



c) Prestressing steel	: St 1570/1770, f _{0.2,20} ° _C = 1570 MPa
Test conditions	: Constant load control Rate of heating, Ť = 4 ⁰ C/min
Specimen	: Cold-worked, ϕ = 7.5 mm, ℓ = 10 x ϕ
Chem. composition	:
Remarks	: Total deformation minus thermal strain and initial elastic strain as function of temperature at different load levels
Reference	: Ruge & Winkelmann (1978-80)

- 90 -



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 3.5.2 TRANSIENT STATE DATA

 c) Prestressing steel : St 1570/1770, $f_{0.2,20}o_C = 1660$ MPa

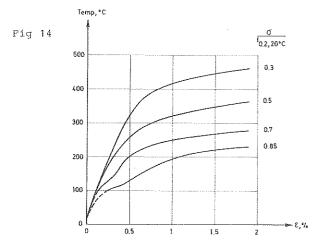
 Test conditions : Constant load control Rate of heating, $\dot{T} = 4^{o}C/min$

 Specimen : Cold-worked, $\phi = 5$ mm, $\ell = 10 \times \phi$

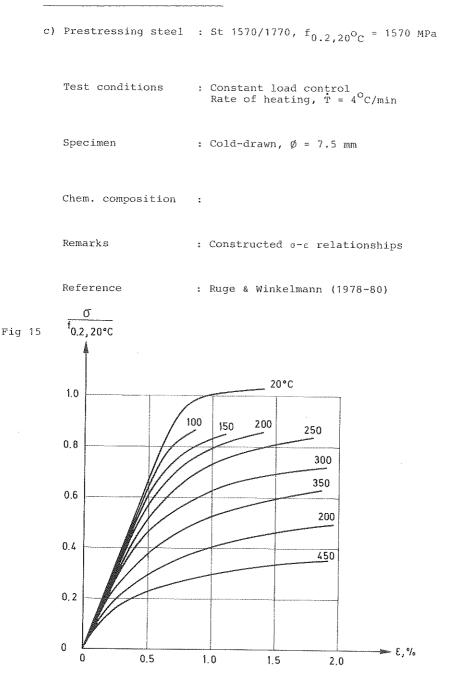
 Chem. composition :

 Remarks : Total deformation minus initial elastic strain as function of temperature at different load levels

 Reference : Ruge & Winkelmann (1978-80)



- 91 -



3.6 Relaxation

A relaxation test is in most cases carried out at constant strain and temperature and the stress decrease is studied as a function of time. However, in a fire-exposed prestressed concrete structure the temperature is varying and the loss in prestress is due to substantial relaxation. As the temperature increases the relaxation process develops more rapidly, which makes the situation more complicated. The relaxation in prestressed concrete structures can only be studied analytically, which is only done fragmentarily so far.

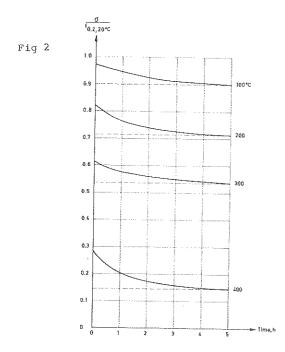
Dimensionless relaxation curves for reinforcing and prestressing steel are illustrated in Figs. 1-3 from Anderberg (1983), Cahill (1965) and Voves (1978). The results deviate very much from each other, the reason of which is hard to find when only three sources are to be found.

- 94 -

c) Prestressing steel : $f_{0.2,20}\circ_{C} = 1070 \text{ MPa}$ $f_{u,20}\circ_{C} = 1630 \text{ MPa}$ Test conditions : Steady state An initial strain related to 0.2% proof stress at each test temperature was kept constant during the test Specimen : C P S Mn Si Chem. composition : 0.78 0.024 0.022 0.63 0.21 Remarks : Relaxation at different temperatures

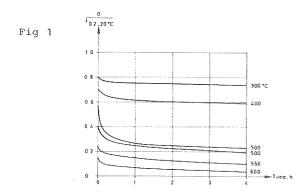
Reference

: Cahill (1965)

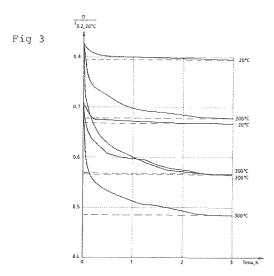


b)	Reinforcing steel	å	^f 0.2,20 ⁰ C = 710 MPa f _{u,20} 0 _C = 880 MPa
	Test conditions	1	An initial strain related to diffe- rent load levels was kept constant at different temperatures
	Specimen	**	
	Chem. composition	ţ	C P S Mn Si 0.31 0.040 0.33 1.11 0.041
	Remarks	:	Relaxation at different temperatures
	Reference	:	Anderberg (1983)

- 95 -



c) Prestressing steel , f_{0.2,20}°c = 1559 MPa . $f_{u,20}o_{c} = 1848$ MPa Test conditions : Steady state An initial strain related to different load levels was kept constant at different temperatures Specimen : Ø = 4.5 mm Chem. composition : Remarks : Relaxation at different temperatures Reference ': Voves, B. (1978)



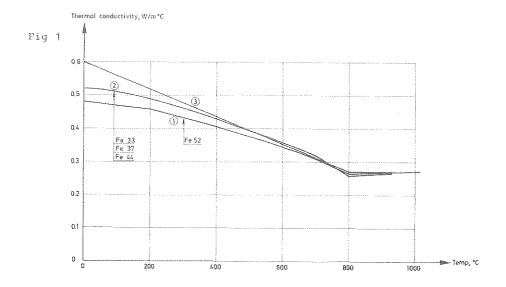
3.6 RELAXATION

3.7 Thermal properties

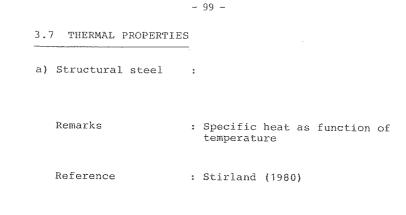
Information about thermal properties of steel is in literature very limited. Four references are sorted out representing the thermal conductivity, specific heat and specific volumetric enthalpy as function of temperature for structural steel, see Figs 1-4. The thermal conductivity curves taken from three different investigations seem to be in good agreement with each other.

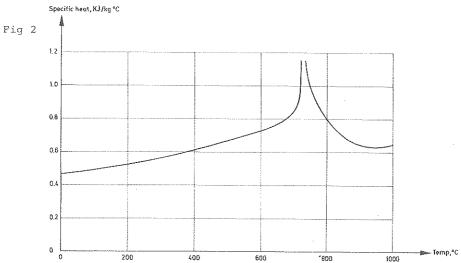
a) Structural steel	: (1) Fe 52 (2) Fe 33, Fe 37, Fe 44
Remarks	: Thermal conductivity as function of temperature
Reference	: (1) and (2) Stirland (1980) (3) Alpsten (1967), Wickström (1979)

~ 98 ~



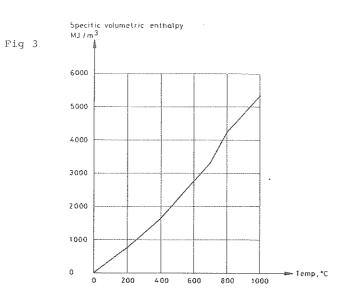
3.7 THERMAL PROPERTIES





3.7 THERMAL PROPERTIES

- a) Structural steel :
 - Remarks : Specific volumetric enthalpy as function of temperature
 - Reference : Alpsten (1967), Wickström (1979)



3.7 THERMAL PROPERTIES

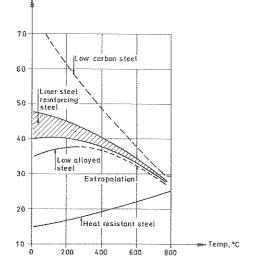
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a,b,c) Different steels :
```

Remarks

- : Thermal conductivity as function of temperature
- Reference : Schneider & Diederichs (1981)



Thermal conductivity,W/m°C



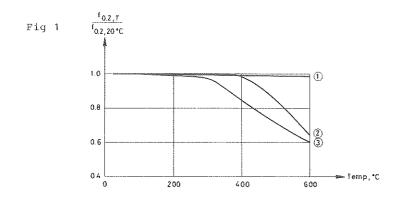
3.8 Residual properties

Residual properties as 0.2% proof tensile stress and ultimate tensile strength are illustrated in Figs 1-5. The residual 0.2% proof stress is only meaningful to be measured when the specimen is unloaded during heating and cooling. Hot-rolled steel is not influenced by temperature as concerns neither 0.2% proof stress nor tensile strength. Cold-drawn steel may have some decrease in strength above $500^{\circ}C$ (see Fig 4, specimen is loaded during heating), which, however, only amounts to 10% of the original strength. Prestressing steel does not recover from temperatures above about $250^{\circ}C$ and exposed to $600^{\circ}C$ the residual strength reaches about 0.6 $f_{0.2,20^{\circ}C}$.

3.8 RESIDUAL PROPERTIES

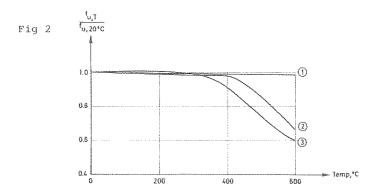
: ① St 60/90, f_{0.2,20}°_C = 590 MPa a) Structural steel c) Prestressing steel : (2) St 145/165, $f_{0.2,20}^{\circ}c$ = 1520 MPa (3) St 160...180, $f_{0.2,20}o_{\rm C} = 1570$ MPa Test conditions : No loading during heating or cooling : (1) Hot-rolled, $\phi = 26 \text{ mm}$ Specimen (2) Cold-drawn, $\phi = 5.2 \text{ mm}$ (3) Cold-drawn, $\phi = 5 \text{mm}$ Chem. composition : Remarks : Residual 0.2% proof tensile stress Reference : Dannenberg et al (1959)

- 103 -



3.8 RESIDUAL PROPERTIES

: (1) St 60/90, f_{u,20}0_C = 590 MPa a) Structural steel ② St 145/165, f_{u,20}0_C = 1650 MPa c) Prestressing steel (3) St 160...180, $f_{u,20}o_{c} = 1740 \text{ MPa}$ Test conditions : No loading during heating or cooling : () Hot-rolled, $\phi = 26 \text{ mm}$ Specimen (2) Cold-drawn, $\phi = 5.2 \text{ mm}$ (3) Cold-drawn, $\phi = 5 \text{ mm}$ Chem. composition : Remarks : Residual ultimate tensile strength Reference : Dannenberg et al (1959)

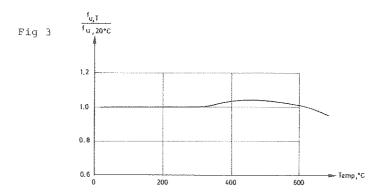


- 104 -

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3.8 RESIDUAL PROPERTIES
a) Structural steel : BSt 37-2, f_{0.2,20}^{\circ} = 255 MPa
   Test conditions : Rate of heating, \hat{T} = 10^{\circ}C/min
   Specimen
                        : b = 10 \text{ mm}, \ell = 100 \text{ mm}
   Chem. composition :
   Remarks
                       : Residual ultimate tensile strength
   Reference
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: Ruge & Winkelmann (1978-80)

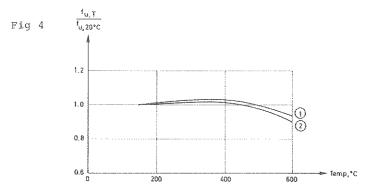
- 105 -



3.8 RESIDUAL PROPERTIES

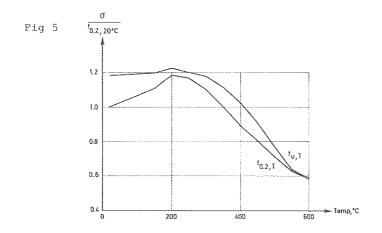
b) Reinforcing steel : BSt 42/50 RK (1) $f_{0.2,20}\circ_{\text{C}} = 430 \text{ MPa}$ (2) $f_{0.2,20}\circ_{\text{C}} = 470 \text{ MPa}$ Test conditions : Rate of heating, $\mathring{\text{T}} = 10^{\circ}\text{C/min}$ Specimen : $\emptyset = 18 \text{ mm}, \ \ell = 10 \text{ x} \ \emptyset$ $\emptyset = 28 \text{ mm}, \ \ell = 10 \text{ x} \ \emptyset$ Cold-drawn Chem. composition : Remarks : Residual ultimate tensile strength Reference : Ruge & Winkelmann (1978-80)

- 106 -



3.8 RESIDUAL PROPERTIES

C)	Prestressing steel	:	, f _{0.2,20} ° _C = 1559 MPa , f _{u,20} ° _C = 1848 MPa
	Test conditions	:	Heated to a specified temperature level with a stabilizing period of 10 minutes before a cooling down to room temperature
	Specimen	:	Ø = 4.5 mm
	Chem. composition	:	
	Remarks	:	Residual ultimate tensile strength and 0.2% proof stress
	Reference	:	Voves, B. (1978)



4 A review and comparative study of different properties

When the results are comparable (i.e. test procedure is similar), it can be shown that the relative decrease in ultimate tensile strength with increasing temperature in most cases is almost the same for all steels, prestressing steels and others separated. Som deviations may occur for cold-worked steel. For hot-rolled steel the decrease in 0.2% proof stress has the same tendency.

When comparing the ultimate strength curves obtained from a steady state test with stress rate control and from a transient state test, a difference can be observed as illustrated for hot-rolled steels in Fig 12. The transient test results are here about 10% lower at temperatures above 450°C. If, however, the stress rate had been chosen much lower, the curves should be much closer to each other.

 $\sigma - \epsilon$ relationships on stress rate controlled steady state conditions for Swedish hot-rolled steel and different structural steels are compared in Figs 13-15. The $\sigma - \epsilon$ curves shown in Fig 13 are in good agreement due to very similar test conditions. The agreement is not that good in Figs 14 and 15, which may be explained by the fact that the exact test conditions are not known for Reichel's results.

A similar comparison is illustrated for two kinds of structural steel under strain rate controlled steady state conditions. The $\sigma-\epsilon$ curves are in good concordance with each other.

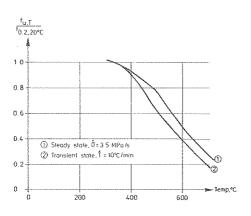
In transient tests the rate of heating influences the deformation-temperature curve. Comparing results when $\dot{T} = 10^{\circ}$ C/min for reinforcing and structural steels, curves as illustrated in Fig 17 are obtained. The curves are quite close to each other. The instability phase at about 275°C, however, are more marked in Copier's results.

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From transient test results $\sigma - \epsilon$ curves can be evaluated. Such curves on structural steel from two different sources are compared in Fig 18. The curves from Ruge & Winkelmann are all above those from Copier.

In Fig 19 $\sigma-\epsilon$ curves from steady state and transient state tests on prestressing steel are compared. The curves on steady state conditions are lying above the corresponding constructed curves. If, however, the stress rate or strain rate in the steady state test is chosen lower, the curves are approaching each other.

Theoretically, it is possible to evaluate what stress rate or strain rate in a steady state test that shall be coupled with a certain rate of temperature increase in a transient test in order to get the same $\sigma - \varepsilon$ curve or ultimate strength. Analytical modelling will be discussed and illustrated in chapter 5.



- 110 -

Fig 12 Ultimate strength as function of temperature for hotrolled steels in (1) a steady state test and in (2) a transient state test. Anderberg (1982)

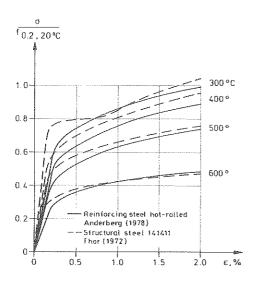


Fig 13 $\sigma-\epsilon$ curves from steady state tests under stress rate control for reinforcing and structural steel. $\dot{\sigma}$ = 3.5 and 1.6 MPa/s, respectively

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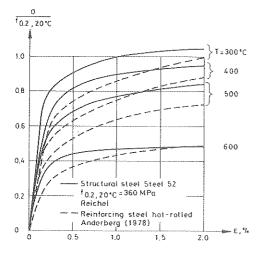


Fig 14 $\sigma-\epsilon$ curves from steady state tests under stress rate control for structural and reinforcing steel

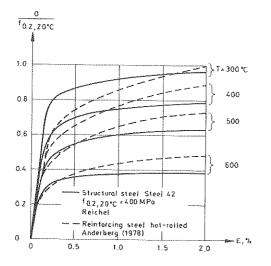


Fig 15 $\sigma{-}\epsilon$ curves from steady state tests under stress rate control for structural and reinforcing steel

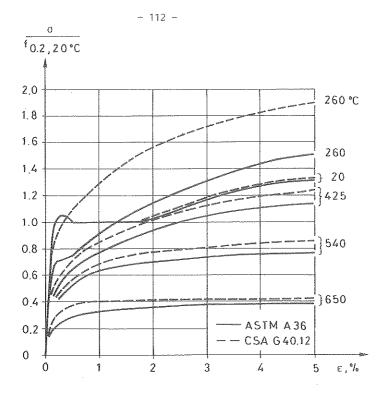


Fig 16 $\sigma - \varepsilon$ curves from steady state tests under strain rate control for different structural steels. $\dot{\varepsilon} = 50 - 100 \cdot 10^{-3} \cdot \text{min}^{-1}$. Harmathy & Stanzak (1970)

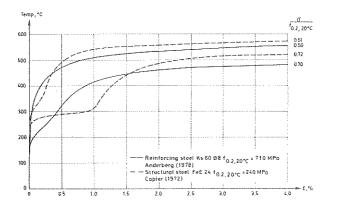


Fig 17 Total deformation minus initial elastic and thermal strain as function of temperature for reinforcing and structural steel. $\dot{T} = 10^{\circ}C/min$

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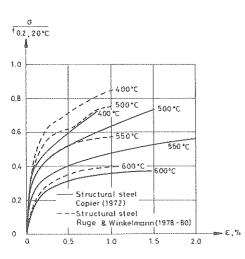


Fig 18 Calculated $\sigma-\epsilon$ curves from transient tests for structural steel. $\dot{T} = 10^{\circ}C/min.$

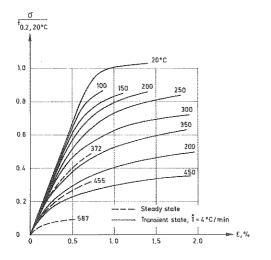


Fig 19 A comparison of $\sigma-\epsilon$ curves for prestressing steel St 1570/1770 obtained from steady state and transient state tests

5 Analytical modelling

5.1 Models

It is generally agreed that the deformation process of steel at elevated temperatures can be described by the three strain components defined below, namely thermal strain, instantaneous, stress-related strain and creep strain. Thus

- 114 -

$$\varepsilon_{\text{total}} = \varepsilon_{\text{t}} = \varepsilon_{\text{th}}(\mathbf{T}) + \varepsilon_{\sigma}(\mathbf{T},\sigma) + \varepsilon_{\text{cr}}(\mathbf{t},\mathbf{T},\sigma)$$
(1)

where

$e_{th}^{(T)}$	= thermal strain
ε _σ (Τ,σ)	= instantaneous, stress-related strain based on
	σ-ε relationship
ε _{cr} (t,T,σ)	= creep strain or time dependent strain

The strains are found separately in steady state tests and several analytical expressions are presented in literature. Behaviour models based on steady state data only partially predict the transient behaviour of any given fire process, load and strain history.

An analytical description of the $\sigma - \varepsilon$ curve as a function of temperature can be made in different ways as illustrated in Figs 20 and 21. In the first case the curve is approximated by two straight lines and in the second case by an elliptic branch placed between straight lines. The curve can also be approximated by other analytical expressions.

Models of creep are in most cases based on a concept put forward by Dorn (1954), in which the effect of variable temperatures is considered. The extension of the model to be applicable to variable stress can, for instance, be based on the strain hardening rule.

The creep strain is assumed to be dependent on the magnitude of stress and on the temperature-compensated time evaluated

from the expression

$$\Theta = \int_{\Omega}^{t} e^{-\frac{\Delta H}{RT}} dt \quad (h)$$
 (2)

where

AH = activation energy of creep, J/mol
R = gas constant, J/mol·K
t = time

The relation between creep strain, $\varepsilon_{\rm Cr}$, and temperature-compensated time, 0, at different stress levels is shown principally in Fig 22. The change from the curved branch to the straight line (primary and secondary phase) is denoted by $\theta_{\rm O}$ and the intersection between the straight line and the creep axis is called $\varepsilon_{\rm Cr,O}$. The slope of the straight line is called Z. Harmathy (1967) has derived an analytical expression between $\varepsilon_{\rm Cr}$ and the parameters 0, Z and $\varepsilon_{\rm Cr,O}$ as follows:

$$\epsilon_{\rm cr} = (\epsilon_{\rm cr,o}/\ln 2) \operatorname{arcosh} (2^{ZO/\epsilon_{\rm cr,o}})$$
 (3)

A modified expression is also made by Plem (1975) where the primary phase is defined by a parabolic equation and the secondary phase by a linear slope. The transfer occurs at time θ_{o} . The mathematical formula is

$$\varepsilon_{\rm cr} = \varepsilon_{\rm cr,o} \left(2 \cdot \sqrt{\frac{z}{\varepsilon_{\rm cr,o}}}\right) \quad \text{when} \quad 0 \le 0 \le 0_{\rm o}$$

$$\varepsilon_{\rm cr} = \varepsilon_{\rm cr,o} \left(1 + \frac{z}{\varepsilon_{\rm cr,o}}\right) \quad \text{when} \quad 0 \ge 0_{\rm o}$$

$$(4)$$

where

$$\theta_{\rm O} = \frac{\varepsilon_{\rm Cr,O}}{Z} \tag{5}$$

A simplified way is to approximate the creep curve by two straight branches with slopes $\rm Z_{p}$ and $\rm Z_{g}$ (Anderberg 1978)

as illustrated in Fig 23. The creep strain is an explicit function of time, temperature and stress, where

$$\varepsilon_{cr}(t, T, \sigma) = t \cdot Z(T, \sigma)$$

$$\varepsilon_{cr} = Z_{p}(T, \sigma) \quad \text{if } 0 \le t \le t_{t}$$

$$\varepsilon_{cr} = Z_{s}(T, \sigma) \quad \text{if } t > t_{t}$$

$$(7)$$

Nomograms for the rate of creep in primary and secondary phase for reinforcing steel Ks 40 \emptyset 8 were published by Anderberg (1978). These nomograms including the determination of the transition time, t_t, are given in Fig 24. These diagrams facilitate an estimation of the magnitude of creep.

The Dorn-Harmathy creep theory has not only been used by Harmathy but also by Thor (1972). In Figs 25-27 the validity of the creep model can be studied. The agreement is satisfactory. In Anderberg (1978) the modified version of the Dorn-Harmathy theory has been used and the concordance between test and calculation is shown in Figs 28-30.

Creep parameters for different kind of steels and used in calculations are collected in Table 5.1. The following equations are governing Z and $\epsilon_{\rm cree}$:

$e_{\rm cr,o} = A \cdot \sigma^{\rm B}$	(8)
$z = \begin{cases} C \cdot \sigma^{D} & \text{if } \sigma \leq \text{SIG 1} \\ H \cdot e^{F \cdot \sigma} & \text{if } \sigma > \text{SIG 1} \end{cases}$	(9)

The coefficients in these equations are given in Table 5.1 for steels accounted for in three sources, 1. Thor (1972), 2. Harmathy & Stanzak (1970) and 3. Anderberg (1978).

5.2 Comparison test and calculation

Using the complete behaviour model as expressed in Eq (1), any test can be simulated. The total deformation as a function of

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temperature measured in a transient test, $\dot{T} = 10^{\circ}C/min$, is calculated in Anderberg (1983) and the comparison is illustrated in Fig 31. The calculation is very close to the measured curves. $\sigma - \varepsilon$ curves from steady state and transient state tests at different temperatures are compared to each other in Fig 32. Agreement between calculations and measurements is also close in this case. The curve of the steady state conditions contains no creep due to the high stress rate. In the curve of the transient state conditions the influence of creep, however, is of importance. The difference between the curves at $\varepsilon > 2\%$ and the decrease in ultimate strength at 500 and 600°C amount to about 15% of $f_{0.2,20}\circ_{\rm C}$.

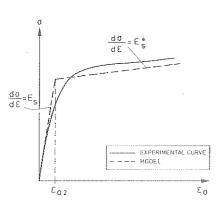
 σ - ϵ curves from steady state tests (strain controlled) on prestressing steel are compared to calculated data in Fig 33. The agreement is very good up to the maximum stress. However, the negative slope is not obtained in the calculation which may be due to a contraction of the steel area during the experiment. The reduced cross-section causes a decrease in total load but the stress may be constant?

It is also possible to simulate a relaxation test. A comparison is made between measured curves (Cahill 1965) and curves predicted on the basis of creep data according to A421-65 (Harmathy 1967). The result is illustrated in Fig 34. The American prestressing steel seems to be a little more sensitive to relaxation above 300° C, but at 400° C the curves are close to each other.

The predictions illustrated above indicate that any test process can be calculated satisfactorily, when the model is based on steady state data.

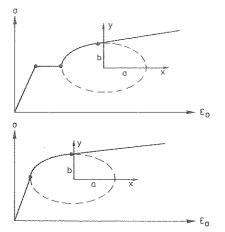
Table 5.1

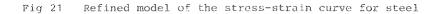
Creep parameters for different kind of steels (1) Thor (1972), (2) Harmathy-Stanzak (1970) and (3) Anderberg (1978)



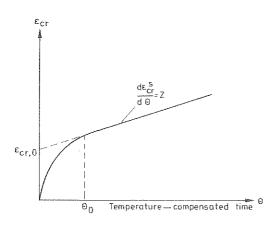
- 119 -

Fig 20 Simplified model of the stress-strain curve for steel





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- 120 -

Fig 22 Principal creep curve for steel according to Dorn's theory

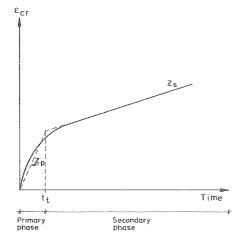


Fig 23 Measured creep curve approximated by two straight branches with slopes $\rm Z_p$ and $\rm Z_s$

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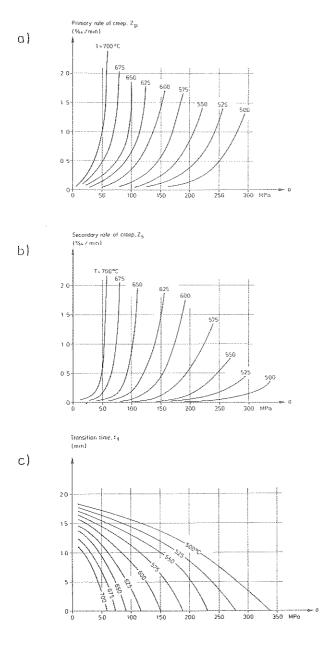


Fig 24 Nomogram for rate of creep

- a) Primary phaseb) Secondary phase
- c) Transition between primary and secondary phase

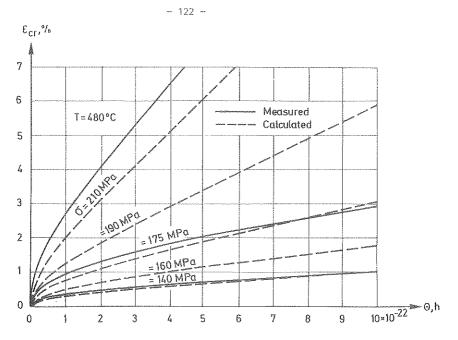


Fig 25 Measured and predicted creep (Dorn-Harmathy theory) at different stress levels. Structural steel ASTM A36, $f_{0.2,20^{\circ}C}$ = 300 MPa, Harmathy (1967)

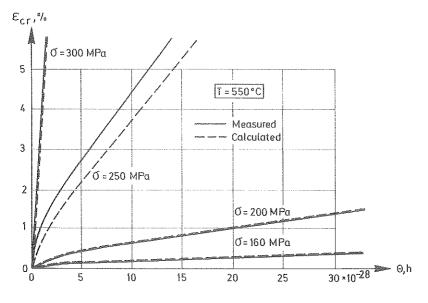


Fig 26 Measured and predicted creep (Dorn-Harmathy theory) at different stress levels. Structural steel SIS 142172, f_{0.2,20}°_C = 340 MPa, Thor (1972)

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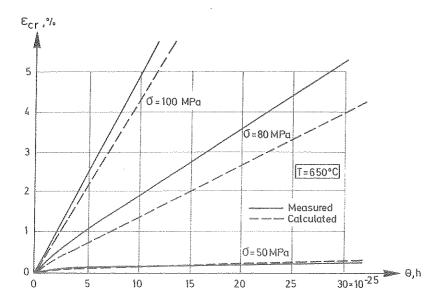


Fig 27 Measured and predicted creep (Dorn-Harmathy theory) at different stress levels. Structural steel SIS 142172, $f_{0.2,20}$ $_{\rm C}$ = 340 MPa, Thor (1972)

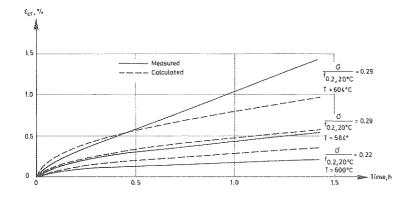


Fig 28 Measured and predicted creep (modified Dorn-Harmathy theory) at different stress levels. Reinforcing steel Ks 40 Ø8, f_{0.2,20}°C = 456 MPa, Anderberg (1978)

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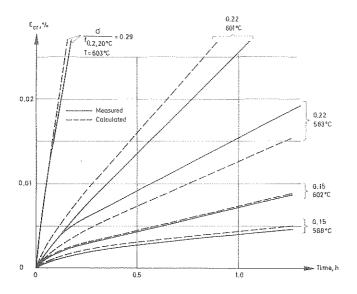


Fig 29 Measured and predicted creep (modified Dorn-Harmathy
theory) at different stress levels. Reinforcing steel
Ks 60 Ø8, f_{0.2,20}°_C = 710 MPa, Anderberg (1978)

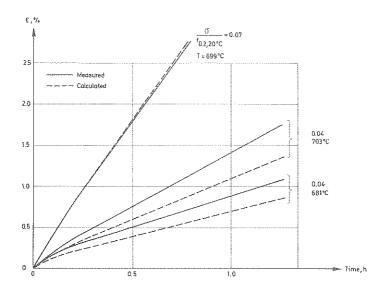


Fig 30 Measured and predicted creep (modified Dorn-Harmathy theory) at different stress levels. Reinforcing steel Ks 60 \emptyset 8, f_{0.2,20}°_C = 710 MPa, Anderberg (1978)

- 125 -

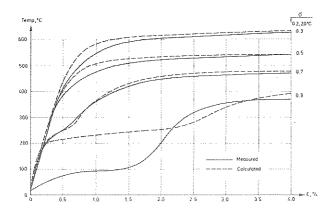


Fig 31 Measured and predicted total deformation as a function of temperature at different load levels in a transient process, $\tilde{T} = 10^{\circ}C/min$. Reinforcing steel Ks 60 Ø8, $f_{0.2,20^{\circ}C} = 710$ MPa, Anderberg (1978)

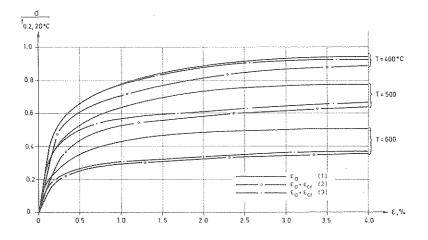


Fig 32 Measured and predicted $\sigma-\epsilon$ curves from steady state and transient state tests.

- 1. Steady state test, stress-controlled, $\sigma{=}3.5~\text{MPa/s}$
- 2. Transient state test, constructed curve, $\mathring{\mathtt{T}}\text{=}10^{\text{O}}\text{C/min}$
- 3. Transient process, curve derived analytically, $\mathring{T}=10\,^{O}C/min$

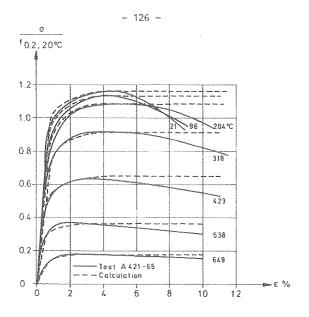
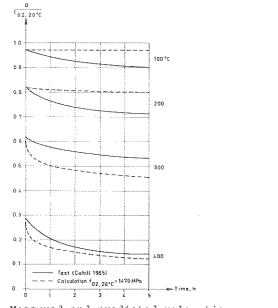
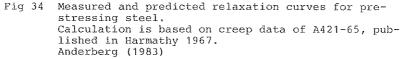


Fig 33 Measured and predicted $\sigma - \varepsilon$ curves from strain controlled steady state tests on prestressing steel ($\dot{\varepsilon} = 700 \cdot 10^{-3} \cdot \min^{-1}$). Anderberg (1983)





5.3 How to compare results obtained at different test procedures

- 127 -

Analytical modelling opens up the possiblity of relating steady state and transient state tests. For instance, the strain or stress rate can be determined in steady state tests to give the same ultimate strength as occurs in a transient test for a given rate of heating.

The predicted influence of stress and strain rates on the stressstrain relationship under steady state conditions can be studied in Figs 35-37. From such curves the ultimate strength as a function of temperature can be evaluated (Figs 38-41). In Fig 40 ab, the influence of stress and strain rates on the ultimate strength under steady state conditions is illustrated. The influence of rate of temperature under transient state conditions is given in Fig 40 c. The ultimate strength is defined by the stress at which the strain is 4%, the thermal strain excluded.

In the analytical modelling (Anderberg 1983) the strain and stress rate in a steady state procedure to give the same ultimate strength as under transient state conditions is found, if the rate of heating is chosen to 10° C/min. The result is shown in Fig 40 d for a specific reinforcing steel (Ks 60, $f_{0.2,20^{\circ}C}$ = 710 MPa). The strain and stress rates obtained by computation are thus

 $\dot{\epsilon}$ = 20% /min and $\dot{\sigma}$ = 0.20 MPa/s = 12 MPa/min

These values presented here vary somewhat depending on the type of steel, which is shown in Table 5.3.1 for

a) structural steel 1411, $f_{0.2,20}o_{C} = 340 \text{ MPa}$

b) hot-rolled reinforcing steel Ks 40, $f_{0.2,20}$ = 456 MPa c) hot-rolled reinforcing steel Ks 60, $f_{0.2,20}$ = 710 MPa d) prestressing steel ASTM A 421-65, $f_{0.2,20}$ = 1470 Mpa

Table	5.3.1	(Anderberg	1983)
-------	-------	------------	-------

Type of steel	°C/min	g MPa/s	é %0/min
Steel 1411	10	0.20	10
Ks 40	10	0.20	10
Ks 60	10	0.20	20
ASTM A 421-65	10	0.50	10

For the additional steels in Table 5.3.1 the results of computation are given in Figs 38-39, 41.

If the strength-temperature curve, based on the transient state condition where for instance $\dot{T} = 10^{\circ}$ C/min is accepted as a reference curve, test results can be directly interpreted and, thus, comparable. If the rates $\dot{\epsilon}$ and $\dot{\sigma}$ are below the values mentioned above, the measured ultimate strength should be corrected upwards. If one has the opposite situation the strength values are corrected downwards.

The rate of stress or strain in a steady state test and the rate of temperature in a transient test are governing the development of creep which causes the change in the ultimate strength. The influence of creep for three different procedures is also fully illustrated in Figs 35-41. It is also important to observe that there is a great difference between the time history of creep development in a stress and strain controlled steady state test. This is principally illustrated in Fig 42, where curve 1 represents a stress-strain curve at a very high stress or strain rate and curve 2 and 2' represent a curve at a slow stress and strain rate, respectively. The points A and A' indicate the time at which creep starts to be significant and point B is the assumed failure point ($\varepsilon = 4$ %). In the stress controlled test the time to reach point A is about 50% of the total testing period but in the strain controlled test the corresponding time is only about 15%. If a test is carried out at $600\,^{\rm O}C$ at the stress rate $\overset{*}{\sigma}$ = 12 MPa/min and at the strain rate $\dot{\epsilon}$ = 20‰/min the testing period amounts to 22 and 2 min, respectively. Although these differences exist, the creep influence on the ultimate strength is the same. In the corresponding transient test this period is about 58 min. This means that the testing period is not a critical parameter in the analytical study by rather the time period of the test when creep develops.

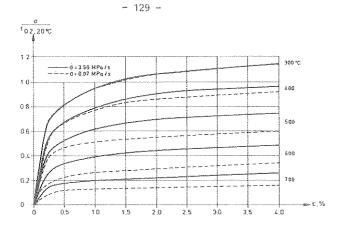


Fig 35 Predicted $\sigma - \epsilon$ curves at different stress rates for reinforcing steel (Ks 60, $f_{0.2,20}$ °C = 710 MPa). Anderberg (1983)

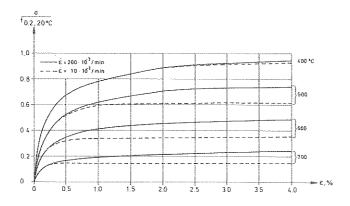


Fig 36 Predicted $\sigma-\epsilon$ curves att different strain rates for reinforcing steel (Ks 60, $f_{0.2,20}o_{\rm C}$ = 710 MPa). Anderberg (1983)

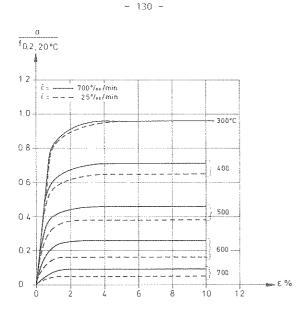


Fig 37 Predicted $\sigma-\epsilon$ curves at different strain rates for prestressing steel (A 421-65, $f_{0.2,20}\circ_{C}$ = 1470 MPa). Anderberg (1983)

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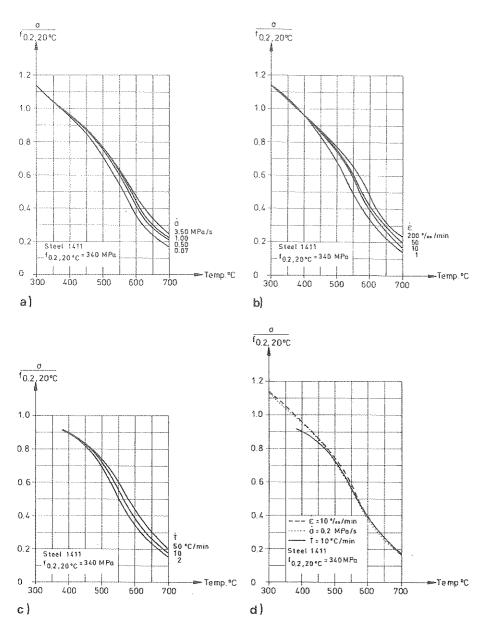
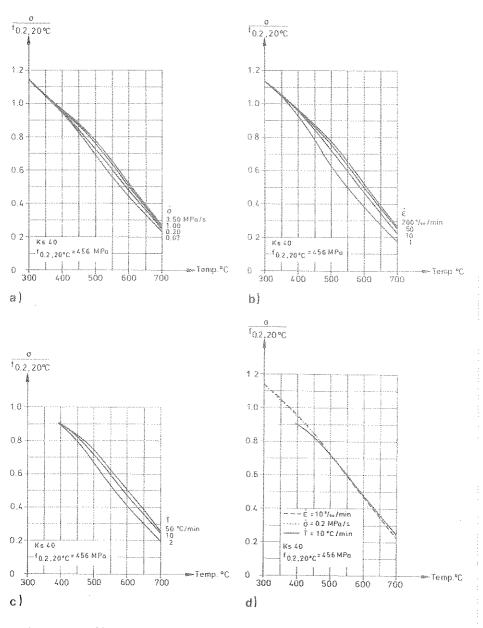


Fig 38 Predicted ultimate strength versus temperature for structural steel.

- a) Steady state, stress rate controlled
- b) Steady state, strain rate controlled
- c) Transient state
- d) Comparison between steady state and transient state

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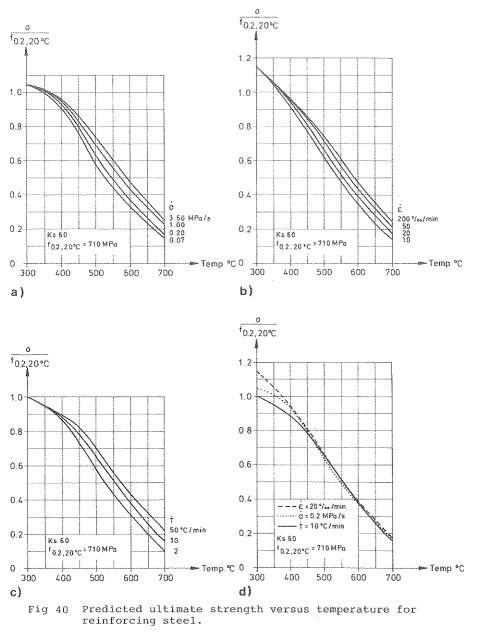
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Predicted ultimate strength versus temperature for Fig 39 reinforcing steel.

- a) Steady state, stress rate controlled
- b) Steady state, strain rate controlled
- c) Transient stated) Comparison between steady state and transient state

- 133 --



- a) Steady state, stress controlled
- b) Steady state, strain controlled
- c) Transient state
- d) Comparison between steady state and transient state

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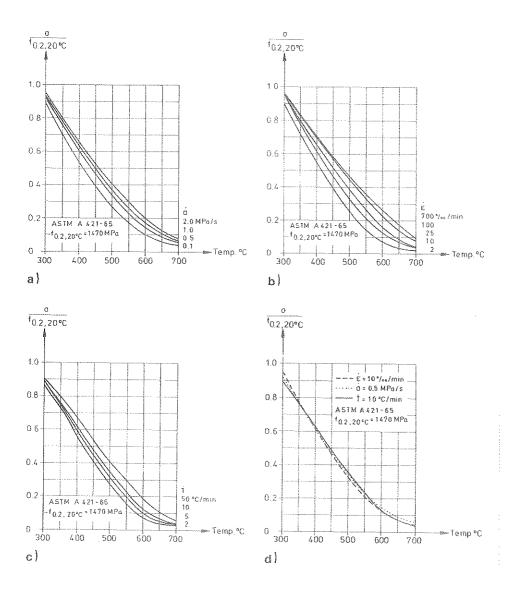
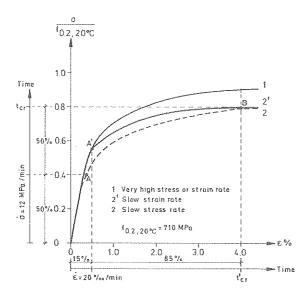


Fig 41 Predicted ultimate strength versus temperature for prestressing steel.

- a) Steady state, stress rate controlled
- b) Steady state, strain rate controlled
- c) Transient state
- d) Comparison between steady state and transient state

1929 - P. 19



- 135 -

Fig 42 Principal σ - ϵ curves in stress and strain rate controlled steady state tests with time scales indicated (t'_cr and t_cr represent the testing period).

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Part III: STEEL LITERATURE

6 References
6.1 Particular properties
6.1.1 General properties
6.1.2 Tensile strength
6.1.3 Modulus of elasticity
6.1.4 Thermal strain
6.1.5 Creep strain
6.1.6 Stress-strain characteristics
6.1 Steady state tests
6.2 Transient tests
6.1.7 Compressive strength
6.1.8 Relaxation
6.1.9 Thermal properties
6.1.10 Residual properties after cooling

6.2 Structural behaviour

6.3 Analytical models

6.3.1 Material

6.3.2 Structures

Subheadings

- a) Structural steel
- b) Reinforcing steel
- c) Prestressing steel
- d) Cast iron
- e) Insulation material for steel protection

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- 6.1.4. Thermal strain
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- XXVIII -

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headings

7 Symbols and abbreviations c_p specific heat kJ/kg^OC I specific volumetric enthalpy MJ/m³ λ thermal conductivity $W/m^{O}C$ $^{\rm E}20^{\rm O}C$ $\,$ modulus of elasticity at room temperature -"- temperature T Em $^{\rm f}0.2,20^{\rm O}C$ ~ 0.2 % proof stress at room temperature -"- temperature T f0.2,T $^{\rm f}\text{u},20^{\circ}\text{C}$ — ultimate strength at room temperature -"- temperature T fur t time (h, s) T temperature ^OC T rate of heating $^{O}C/min$ e strain ϵ rate of strain s⁻¹ ε_{th} thermal strain ε_{α} stress-related strain e_{ct} creep strain stress MPa σ a rate of stress MPa/s