Observation of the anomalous Hall effect in a collinear antiferromagnet

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Time-reversal symmetry breaking is the basic physics concept underpinning many magnetic topological phenomena such as the anomalous Hall effect (AHE) and its quantized variant¹⁻⁵. The AHE has been primarily accompanied by a ferromagnetic dipole moment, which hinders the topological quantum states and limits data density in memory devices^{1,6-8}, or by a delicate noncollinear magnetic order⁹⁻¹⁴ with strong spin decoherence, both limiting their applicability. A potential breakthrough is the recent theoretical prediction of the AHE arising from collinear antiferromagnetism in an anisotropic crystal environment¹⁵. This new mechanism does not require magnetic dipolar or noncollinear fields. However, it has not been experimentally observed to date. Here we demonstrate this unconventional mechanism by measuring the AHE in an epilayer of a rutile collinear antiferromagnet RuO₂. The observed anomalous Hall conductivity is large, exceeding 300 S·cm⁻¹, and is in agreement with the Berry phase topological transport contribution¹⁵⁻¹⁷. Our results open a new unexplored chapter of timereversal symmetry breaking phenomena in the abundant class of collinear antiferromagnetic materials.

Anomalous Hall effect (AHE) is a text-book time-reversal symmetry breaking (TRSB) phenomenon, which remains at the forefront of the research of topological magnetism and dissipationless nanoelectronics^{1,3,4}. The AHE current is transverse to the applied electric field and is odd under time-reversal. The phenomenon is microscopically understood as a consequence of a spin-splitting of electronic states by the magnetic ordering, which is linked to the electrons' momenta by the spin-orbit coupling^{1,18}. An intrinsic, scattering-independent contribution to AHE is of topological origin, given by the momentum-space Berry curvature of the electron Bloch states^{1,16,17}.

The bulk of the AHE literature is focused on TRSB by a ferromagnetic order, illustrated in Fig. 1a by the Fe crystal^{1,7,8}. In collinear ferromagnets like Fe, a majority of electrons occupy one spin state, resulting in a macroscopic dipole moment aligned with the majority spin axis. In the past two decades, AHE was demonstrated also as a consequence of TRSB by a noncollinear magnetic order. The studies included ferromagnets^{9,12}, as well as antiferromagnets^{11,13,14,19-22}. For example, in the Mn₃X family of noncollinear antiferromagnets (X = Ir, Sn, Pt)^{13,14,20}, illustrated in Fig. 1b, the magnetic dipole is weak, while strong TRSB effects and AHE were associated with the presence of a magnetic octupole²³. However, both dipolar and noncollinear magnetic orders represent an obstacle for realizing quantized variants of the anomalous Hall systems, and for their practical utility in nanoelectronic devices. On one hand, the dipolar fields disturb the topological quantum states, or limit the data density in magnetic memories. The limitations imposed by the magnetic dipole can be remedied in the (nearly) compensated noncollinear antiferromagnets. On the other hand, this comes at the cost of strongly decohered spin quantum numbers of the electrons. Moreover, the noncollinear states are delicate due to frustrated magnetic interactions in the lattice, the presence of competing magnetic phases, and the sensitivity to variations in structure parameters, such as the film thickness²⁴ or disorder^{11,13}.

To date, an anomalous Hall current in collinear antiferromagnets has been experimentally identified only as a consequence of canting of the magnetic moments by an applied magnetic field, or due to a field-induced spin-flip transition into a ferromagnetic state^{6,25,26}. In both cases, AHE was ascribed to the magnetic dipole in the material, generated by the applied magnetic field²⁶. In the absence of an external magnetic field, collinear antiferromagnets were commonly perceived to exclude AHE. This is indeed the case when the opposite magnetic sublattices are connected by

time-reversal combined with a spatial symmetry operation (*e.g.*, inversion or half-unit cell translation)^{11,19,27,28}. The resulting effective time-reversal symmetry, T_{AF} , prohibits the AHE.

Recently, this general perception has been broken by Smejkal *et al.*¹⁵, who predicted large AHE arising from a previously overlooked TRSB mechanism, generated by a collinear antiferromagnetic order in an anisotropic crystal environment. The mechanism applies to a broad materials class of collinear antiferromagnetic crystals allowing, by symmetry, also for a ferromagnetic dipole. Since the dipole moment is weak, earlier studies assumed that the response of these so called weak ferromagnets was close to that of the perfectly compensated T_{AF} -symmetric antiferromagnets²⁹. However, taking collinear antiferromagnet RuO₂ as an example, Smejkal *et al.*¹⁵ have theoretically demonstrated that the weak ferromagnetism is due to relativistic spinorbit fields and represents only a perturbative correction^{29,30}. In contrast, there is an additional strong TRSB mechanism arising from the collinear magnetic-exchange fields and the anisotropic crystal structure, which produces large AHE even when the ferromagnetic dipole moment vanishes^{15,31,32}.

Before turning to our experimental observation of large AHE in the RuO₂ antiferromagnet, we present density functional theory (DFT) calculations addressing the relevant experimentally geometry. Specifically, we show that for the Néel vector aligned with the [110] crystal axis, the collinear antiferromagnet has a large Berry curvature and corresponding intrinsic AHE. In the first step, we demonstrate strong unconventional TRSB on non-relativistic spin-split bands of RuO₂^{15,33}. In the second step we show how this TRSB is translated into the momentum-dependent Berry curvature when turning on the relativistic spin-orbit coupling.

DFT calculations in Fig. 1c illustrate that TRSB seen in the local magnetization densities of RuO₂ corresponds to the symmetry lowering of magnetic Ru sites by the presence of the non-magnetic oxygen atoms. Here, the T_{AF} -symmetry connecting the antiferromagnetic sublattices is broken by the oxygen octahedra and generates the low-symmetry magnetization isosurfaces, shown in Fig. 1c. These anisotropic real-space magnetization densities, dominated by d_{xz} and d_{yz} orbitals of Ru, are directly linked in the momentum-space to a strong spin-splitting of non-relativistic energy bands, reaching 1 eV around the Fermi level, as shown in Fig. 1d,f and Supplementary Fig. 1 (ref. ^{15,33-35}).

The oxygen octahedra formed around the two Ru atoms in the unit cell are rotated with respect to each other by 90° and this symmetry is related to the perfect spin-compensation, *i.e.*, the lack of a magnetic dipole in the (non-relativistic) spin-split bands. Instead of a dipole, the RuO₂ spin-split Fermi surfaces¹⁵, shown in Fig. 1d, have the characteristic form of a magnetic toroidal quadrupole³⁶. This is in contrast to a conventional dipolar spin-splitting, which is illustrated in Fig.1e on a hypothetical ferromagnetic phase of RuO₂. In addition, the spin-component along the Néel vector is a good quantum number in the RuO₂ band structure, *i.e.*, commutes with the Hamiltonian, in the absence of the relativistic spin-orbit coupling. This is highlighted in Fig.1d,f where the bands are labeled by the up and down-spin quantum numbers. The collinear antiferromagnetic TRSB in RuO₂, therefore, shares the vanishing dipole magnetic moment with the non-collinear antiferromagnetic TRSB, while keeping the favorable spin-conserving nature of ferromagnetic TRSB.

When we connect the spin and momentum sectors by adding spin-orbit coupling in our DFT calculations, the RuO₂ antiferromagnet can exhibit an anomalous Hall vector σ_{AHE} : $j_{AHE} = \sigma_{AHE} \times E$, where j_{AHE} is the anomalous Hall current, E is the applied voltage, and $\sigma_{AHE} = (\sigma_{yz}, \sigma_{zx}, \sigma_{xy})$. The presence and orientation of σ_{AHE} is sensitive to the orientation of the antiferromagnetic Néel vector $N = (M_A - M_B)/2$ (ref. ¹⁵). In Fig. 1g we show our DFT calculation of the crystal momentum-resolved Berry curvature¹⁵ for the antiferromagnetic Néel vector oriented along the [110] direction. We observe hotspots where the Berry curvature is larger in magnitude than in ferromagnetic Fe²⁶, or in noncollinear antiferromagnets¹⁴. By integrating the Berry curvature, we obtain the intrinsic anomalous Hall conductivity which shows large values up to 1000 S/cm, depending on the position of the Fermi level (see Supplementary Fig. 2,3). This indicates the possibility of unexpectedly large Berry curvature effects, and particularly anomalous Hall conductivity, in RuO₂ with the Néel vector aligned along the [110] crystal direction¹⁵.

Rutile RuO₂ is a good metal with a room-temperature resistivity of only $\rho \sim 35 \ \mu\Omega \cdot \text{cm}$ (ref. ³⁷). Recent neutron scattering³⁸ and resonant X-ray scattering³⁹ studies demonstrated the presence of collinear antiferromagnetism in both its thin films and bulk crystals above room temperature, where the spin axis is slightly tilted off the *c*-axis. It has been previously shown⁴⁰ that thin-film growth of rutile RuO₂ on oxide single-crystal MgO and SrTiO₃ substrates by pulsed laser deposition yields highly-quality (110)-oriented and (100)-oriented epitaxial films, respectively, which facilitates the investigation of the AHE in RuO₂ thin films along different crystallographic orientations.

To explore the AHE experimentally, we fabricated epitaxial thin films of RuO₂. To optimize the electrical conductivity, systematic growth of RuO₂ onto MgO single-crystal substrates was performed for a wide range of oxygen partial pressure between 10^{-2} and 10^{-6} Torr and a wide growth temperature range between 500 and 800°C. The thickness of the RuO₂ films was kept at 27 nm. The optimal growth conditions are 550°C and 10^{-3} Torr oxygen pressure, for which the lowest room-temperature resistivity of $\rho \sim 64 \ \mu\Omega \cdot cm$ was achieved (Supplementary Fig. 4). Although it is higher than the bulk resistivity, the value for rutile RuO₂ films is one fifth of the resistivity of a commonly used perovskite metallic oxide SrRuO₃ (ref. ⁴¹). Generally, RuO₂ thin films grown below 700°C are all epitaxial and exhibit metallic transport behavior (Supplementary Fig. 4).

Figure 2a shows an X-ray diffraction pattern of an optimized 27-nm-thick RuO₂/MgO heterostructure and reveals the (110) orientation of epitaxial RuO₂, which is consistent with the previous growth⁴⁰. The quality of our films is also evidenced by the cross-section transmission electron microscopy image in Fig. 2b. Magnetization measurements indicate a weak net moment for such RuO₂ thin films reaching 0.2 μ_B per unit cell at 50 T, as shown in Fig. 2c (see Methods).

To demonstrate the presence of the antiferromagnetic order in our RuO₂ films, we performed magnetization measurements of a Co₉₀Fe₁₀/RuO₂ exchange bias stack. The soft magnetic layer of Co₉₀Fe₁₀ with a thickness of 5 nm and a 2-nm-thick Pt cap to prevent Co₉₀Fe₁₀ from oxidation were deposited on top of the RuO₂ (Fig. 2d-f) at room temperature in a magnetic field of ~20 mT, yielding a clear exchange bias at 50 K (Fig. 2f). This agrees with the reported antiferromagnetic order of rutile RuO₂ films revealed by resonant X-ray scattering studies³⁹. The interfacial exchange coupling between RuO₂ and Co₉₀Fe₁₀ causes a large enhancement of the coercivity field of Co₉₀Fe₁₀ (Fig. 2d). The blocking temperature of ~200 K (Fig. 2f) could be increased towards the Néel temperature of RuO₂ by post-annealing in a magnetic field⁴².

As shown below, the AHE is suppressed in otherwise similar RuO_2 films deposited onto $SrTiO_3$. These films, therefore, serve as crucial reference systems in our experimental AHE study. The RuO_2 films deposited onto $SrTiO_3$ at the same optimized conditions as for the RuO_2/MgO growth are highly ordered as well, but (100)-oriented (Supplementary Fig. 5). Magnetization measurements display again a weak net moment reaching ~0.2 μ_B per unit cell at 50 T (Supplementary Fig. 6).

In Fig. 3 we compare transport measurements on (110)-oriented RuO₂/MgO and on (100)-oriented RuO₂/SrTiO₃. The temperature-dependent longitudinal resistivity is nearly identical in the two film-orientations (Fig. 3a). The two orientations have also the same Néel transition temperature which is close to 350 K, as inferred from the transport anomaly⁴³ highlighted in the inset of Fig. 3a by the peak in $d\rho/dT$. (We note that the Néel temperature above 300 K in our films is consistent with earlier reports^{38,39}). Figure 3b shows the longitudinal magnetoresistance measured with out-of-plane fields up to 50 T. The observed form corresponds to the ordinary positive magnetoresistance due to the Lorentz force, typical in metals. Again, the plots are nearly identical in the RuO₂/MgO and RuO₂/SrTiO₃ films over the entire range of applied fields.

In in the inset of Fig. 3c, we plot Hall measurements in a magnetic field applied along the out-ofplane direction of the thin films at 340 K. As for the longitudinal magnetoresistance, these hightemperature linear Hall signals in the two film-orientations are identical and can be ascribed to the ordinary Lorentz force contribution. However, at lower temperatures, a clear anomaly appears in the Hall signal of (110)-oriented RuO₂/MgO, while the Hall data remain linear in (100)-oriented RuO₂/SrTiO₃, as shown for 10 K in the main plot of Fig. 3c. This is highlighted in Fig. 3d by subtracting the linear fit to the total Hall data between -40 and -50 T. The nonlinear Hall contribution is seen in the (110)-oriented RuO₂/MgO film (while it is absent in RuO₂/SrTiO₃) over an entire range of temperatures from 10 to 300 K (Supplementary Fig. 7 & 8).

We show below that the nonlinear Hall signal is consistent with TRSB and the AHE described in theoretical Fig. 1. Specifically, we present the following set of arguments: (i) The AHE signal has the theoretically expected magnitude, sign, and temperature dependence. (ii) The (110)-oriented RuO₂/MgO and (100)-oriented RuO₂/SrTiO₃ films have identical ordinary longitudinal and Hall signals which confirms that the additional contribution seen only in the (110)-oriented film is of the anomalous origin, *i.e.*, unrelated to the ordinary Lorentz force Hall effect. (iii) The observed anisotropy of AHE is again consistent with the theoretical phenomenology. This includes the vanishing anomalous Hall signal for the Néel vector aligned with the [001] easy-axis and the corresponding absence of a finite and hysteretic anomalous Hall signal at zero magnetic field. (iv)

Related to the previous point, the strong magnetic field required for observing AHE is only needed for reorienting the Néel vector away from the easy-axis. It is not required, and up to the measured 50 T it is not contributing significantly, to the TRSB mechanism of AHE. In addition, also the spin-flip transition is excluded since the hypothetical ferromagnetic state (Fig.1b) has a much higher energy than the antiferromagnetic ground state (by 56 meV per unit cell), and has a net magnetization of ~0.4 μ_B per unit cell which we do not observe in our experiment in Fig. 2c. We now discuss these points in detail.

Considering the extracted anomalous Hall resistivity ρ_{AHE} and the longitudinal (zero-field) resistivity ρ (Fig. 3a) of the RuO₂/MgO film at different temperatures, the anomalous Hall conductivity estimated from $\sigma_{AHE} \approx -\rho_{AHE}/\rho^2$ is plotted as a function of temperature in Fig. 3e. At low temperatures, σ_{AHE} reaches ~330 S·cm⁻¹. This is over three times that of the non-collinear antiferromagnet Mn₃Sn¹³ (see Supplementary Tab. 1 for a detailed comparison) and even on the same order as the anomalous Hall conductivity of Fe thin films⁴⁴. Above 50 K, σ_{AHE} falls off rapidly with increasing temperature, to approximately 3 S·cm⁻¹ at room temperature. We comment on the temperature dependence below in the theory discussion.

The nonlinear Hall signal in RuO₂/MgO could in principle occur due to the ordinary contribution from the orbital Lorentz force combined with a multi-band transport, or multiple types of carriers or anisotropic Fermi surfaces. However, the measured ordinary longitudinal magnetoresistance shows no significant anisotropy between RuO₂/MgO and RuO₂/SrTiO₃ films (Fig. 3b). Hence, the ordinary Lorentz force contributions to transport are the same in the two film orientations, implying that the observed anomalous nonlinear Hall contribution in RuO₂/MgO cannot be ascribed to the ordinary Hall effect. Consistent with this picture, the total measured Hall slopes for the two RuO₂ film orientations approach each other at 340 K where the anomalous Hall contribution in (110)-oriented RuO₂/MgO approaches zero. We can therefore ascribe the high-temperature linear Hall data above 340 K in both films to the ordinary Hall effect. Since the ordinary longitudinal magnetoresistance shows no anisotropy between the two film orientations down to 10 K, we can also expect that their ordinary Hall contributions are similar down to low temperatures. The corresponding carrier densities at different temperatures, assuming a single-band approximation, are given in Supplementary Tab. 2.

By subtracting the 10 K linear Hall data measured in (100)-oriented $RuO_2/SrTiO_3$ from the total 10 K Hall signal in (110)-oriented RuO_2/MgO , we obtain an anomalous Hall conductivity estimate which is about a factor of 2 larger than the 10 K value plotted in Fig. 3e. The comparison of the two estimates suggests that the 10 K anomalous Hall component is not fully saturated at 50 T.

Finally, we found that (110)-oriented RuO₂ films can be also synthesized by using (011)-oriented PMN-PT (0.7PbMg_{1/3}Nb_{2/3}O₃--0.3PbTiO₃) single-crystal substrates. However, these films are less ordered as revealed by X-ray diffraction and transmission electron microscopy characterizations (Supplementary Fig. 9). Although a deviation from the linear ordinary Hall effect can be clearly seen at low temperatures (Supplementary Fig. 10), the effect is much smaller than that of highly-ordered RuO₂/MgO films, as shown in Fig. 3e. This suggests that the crystallinity is a key and therefore an effective tuning factor in determining AHE in this collinear antiferromagnet.

We show next, by systematic DFT calculations (see Methods), that our experimental observations are consistent with the theoretically predicted AHE in RuO_2 (ref. ¹⁵). Our calculations indicate that the experimentally observed anomalous Hall signal originates from the collinear antiferromagnetism¹⁵ and the external magnetic field is required only for the reorientation of the magnetic moments to the directions with a large anomalous Hall conductivity, *i.e.*, off the [001] easy-axis. Note that an analogous magnetic reorientation was previously shown in a similar collinear antiferromagnet CoF_2^{45} . The small net magnetic moment represents only a minor correction to the anomalous Hall conductivity and the anomalous Hall conductivity is primarily independent of this weak magnetization, as shown earlier by Smejkal et al.¹⁵. In Fig. 4a we show DFT energy bands for the Néel vector along the [110] axis, which were used for the Berry curvature and anomalous Hall conductivity calculations. We have further verified by DFT calculations (see Methods) that the large experimental magnetic fields have only a small effect on these energy bands, as we shown in Supplementary Fig. 2. The magnetic fields have also a small effect on the net magnetization. For the Néel vector along the [110] axis and Hubbard U = 1.6 eV, we obtain a net magnetization of 0.08 μ_B at zero field, and 0.09 μ_B at 35 T. (We note that the majority of the net moment (~80%) comes from the orbital part.) Furthermore, we show in Fig. 4b the dependence of the net moment and Néel vector (sublattice moment) on Hubbard U. We see that the values of the total net moment ~0.1 μ_B/Ru correspond well to the experiment value ~0.2 μ_B per unit cell shown in Fig. 2c.

In Fig. 4c we plot the dependence of the calculated anomalous Hall conductivity on the Fermi level, where zero corresponds to a neutrality point in the employed DFT scheme (see Methods). When shifting the Fermi level to ~0.1 eV, we obtain σ_{AHE}^{DFT} ~300 S/cm, which corresponds in both the magnitude and sign to the experiment at low temperatures. The small Fermi level shift to match the experiment can be ascribed to disorder effects but is also within a numerical scatter seen between different DFT implementations. The position of energy bands around the Fermi level depends on the correlation strength and approximations used in the DFT calculations. We evaluate the Hall conductivity from a Wannier Hamiltonian constructed from pseudopotential calculations with Hubbard U included via the Dudarev scheme (see Methods). However, the Fermi level can shift by $\sim 0.1 \text{ eV}$ towards the bands corresponding to the peak in AHE when we use the full potential calculation within the fully localized limit, as we show in Supplementary Fig. 2. In Supplementary Fig. 3 we show the corresponding out-of-plane and in-plane magnetic anisotropy energies (MAE). We see that the large values of the anomalous Hall conductivity correspond to an out-of-plane MAE $K_z \approx 0.1 - 5$ meV. These values are consistent with the antiferromagnetic Néel vector along the [001] axis and a vanishing anomalous Hall signal¹⁵ when there is no applied magnetic field.

Figure 4d demonstrates that to achieve a large anomalous Hall conductivity it is sufficient to tilt the Néel vector out of the [001] direction (not necessarily to fully reorient it into the (001)-plane). In a simple uniaxial antiferromagnet with zero net moment in the ground state, the typical scale of the reorienting magnetic field is given by the spin-flop field, $\sqrt{2K_zJ}$, where K_z and J are the anisotropy and exchange fields, respectively. The exchange field can be estimated from the Néel temperature $J \sim \frac{3}{2} k_B T_N \sim 350$ T for $T_N \sim 350$ K (see Fig. 3a). The calculated anisotropy energies shown in Supplementary Fig. 3 are on the order of ~0.25 meV ($K_z \sim 4$ T), giving $\sqrt{2K_zJ} \sim 50$ T. In the experiment, we see a sizable AHE signal already below 50 T for the (110)-oriented RuO₂/MgO film. We ascribe this to the weak equilibrium moment of RuO₂ which from the calculations is parallel to the Néel vector when the Néel vector is reoriented along the [110] axis, in effect having a ferrimagnetic order. The presence of the weak moment and the corresponding additional Zeeman energy gain can facilitate the Néel vector reorientation by fields along the [110] axis of magnitudes below the spin-flop scale, similarly as in CoF2⁴⁵. We also point out that the weak net moment in RuO₂ allows for selecting one of the two domains with opposite Néel vector along the [110] axis. For a given sign of the magnetic field applied along the [110] axis, one domain will have a lower energy while the opposite domain will prevail when reversing the sign of the magnetic field¹⁵.

While the measured absence of AHE in (100)-oriented RuO₂/SrTiO₃ allowed us to use this film as a reference when experimentally assigning the non-linear Hall signal in the (110)-oriented RuO₂/MgO film to the AHE, we have not identified the primary reason for the suppressed AHE in RuO₂/SrTiO₃. The differences between RuO₂/SrTiO₃ and RuO₂/MgO films, that could potentially explain the unequal AHE response, include the in-plane MAE (see Supplementary Fig. 3) and the corresponding anisotropy in the magnetic-field reorientation behavior between the two films, anisotropy of AHE itself (see Supplementary Fig. 3), or the presence of opposite crystal-chirality grains leading to a cancellations in the net AHE signal.

The steep drop of the anomalous Hall conductivity with increased temperature in RuO₂/MgO can be explained by the decreased magnitude of the Néel vector. In Fig. 4b,e we demonstrate the correlation between the anomalous Hall conductivity and the magnitude of the Néel vector which we control by tuning Hubbard U for the antiferromagnetic vector along the [110] axis. We observe that for small and large Hubbard U, the Hall conductivities are negligible. This is due to the vanishing antiferromagnetic moment for small U and opening of an insulating bandgap for large U, respectively. Values of $U \approx 1.6-2.0$ eV correspond to the antiferromagnetic Néel vector magnitude consistent with previous DFT studies^{15,33,38,39}. Decreasing U simulates the quenching of the antiferromagnetic moment with increasing temperature. Figure 4e shows that the anomalous Hall conductivity steeply decreases with decreasing U, which is consistent with the experimentally observed drop in the Hall conductivity at high temperatures.

In conclusion, by combining epitaxial growth of RuO₂ thin films, pulsed high-field Hall measurements and DFT calculations, we observe the AHE associated with collinear antiferromagnetic TRSB. The experimental identification of the AHE in a collinear antiferromagnet removes the, often limiting, requirements of a net moment or a complex magnetic structure for observing this prominent topological phenomenon. Simultaneously, RuO₂ is only one example of a broad family of materials allowing for the collinear antiferromagnetic TRSB mechanism and strong spin splitting^{15,33-35}. Our work thus opens new directions in frontier materials research of magnetic topological insulators, axion insulators, or quantum AHE systems

in the abundant class of collinear antiferromagnets, many of which have high ordering temperatures and are composed of common elements.

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Materials and Methods

Growth: RuO₂ thin films were first grown onto (001)-oriented MgO single-crystal substrates at different growth oxygen pressures ranging from 10^{-2} to 10^{-6} Torr and different growth temperatures between 500 and 800°C by pulsed laser deposition with a based pressure of 1.5×10^{-8} Torr. The target-substrate distance was 60 mm. The laser fluence was ~1.6 J/cm² and the repetition rate was kept at 10 Hz during deposition. The ramp rate was 20 °C/min for heating and 10 °C/min for cooling. The growth of RuO₂ thin films onto (001)-oriented SrTiO₃ and (011)-oriented 0.7PbMg_{1/3}Nb_{2/3}O₃-0.3PbTiO₃ (PMN-PT) single-crystal substrates was performed at 550 °C and 10^{-3} Torr with the same laser fluence of ~1.6 J/cm² and repetition rate of 10 Hz. Room-temperature growth of the ferromagnetic Co₉₀Fe₁₀ and the capping Pt thin films was carried out by a d.c. sputtering system with a base pressure of 7.5×10^{-9} Torr. For the Co₉₀Fe₁₀ deposition, the d.c. sputtering power was 90 W and the Ar pressure was 3 mTorr. The growth rate was ~0.11 Å/s. For Pt sputtering, the power was by 30 W and the Ar pressure was 3 mTorr. The growth rate was determined to be 0.5 Å/s. No chemical treatment was taken for substrates before deposition.

X-ray diffraction: X-ray spectra were collected in an XRD-SmartLab diffractometer. The Cu-*Ka* X-ray wavelength 1.541882 Å.

Transmission electron microscopy: The focused ion beam technique was used to fabricate crosssection samples. Afterwards, the transmission electron microscopy characterization was conducted in a FEI Talos F200X setup under 200 kV.

Electrical measurements: Electrical contacts were fabricated by wire bonding through Al wires with a diameter of 25 μ m. The linear four-probe geometry with a space of 1 mm was utilized for resistivity measurements of RuO₂ thin films, which were carried out in a Quantum Design physical property measurement system with a measuring current of 1 mA. The typical two-probe resistance for two 1-mm-far bonds was ~30 Ω and the absolute four-probe resistance was ~10 Ω for optimized RuO₂ thin films.

Pulsed high-field Hall measurements: The conventional Hall geometry was established by Cu wires at Wuhan National High Magnetic Field Center, Huazhong University of Science and Technology, Wuhan, China. The magnetic field was applied along the out-of-plane direction of thin film samples. The amplitude of the a.c. current was 3 mA and the frequency was 100 kHz. The Hall voltage collection was performed by a National Instruments PXIe 5105 oscilloscope at a sampling frequency of 4 MHz.

Pulsed high-field magnetic moment measurements: Pulsed-field magnetic moment signals were collected by a pickup coil coaxial with the pulsed magnetic field and calibrated by low-field magnetic moment obtained by a Quantum Design superconducting quantum interference device. To obtain the high-field magnetic moment of the RuO₂ thin film, both RuO₂/MgO (SrTiO₃) heterostructures and MgO (SrTiO₃) substrates of the same dimensions were measured, and the difference in magnetic moments of the heterostructure and the substrate was subsequently extracted to represent the thin film signal.

Density functional theory calculations: We performed the density functional theory (DFT) calculations employing the projector augmented plane wave method⁴⁶ implemented in VASP code and we used the spherically symmetric Dudarev DFT+U (ref. ⁴⁷). We set the energy cut-off of the plane-wave basis to 520 eV, the PBE exchange-correlation functional⁴⁸, and the crystal momentum grid $16 \times 16 \times 24$. We used DFT relaxed lattice parameters a = b = 4.5337 Å,

c = 3.124 Å and we set the antiferromagnetic moments along the [110] axis. We constructed the maximally localized Wannier functions in the Wannier90 code⁴⁹ and we calculated the intrinsic Hall conductivity by employing the Berry curvature formula¹⁵⁻¹⁷. We used the fine-mesh of $321 \times 321 \times 321$ Brillouin zone sampling points. The influence of the magnetic field via the Zeeman effect and electronic correlation method on the electronic structure was studied by calculations in full potential ELK⁵⁰ code with the same parameters as in VASP.

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Data availability: The data and simulations codes that support the findings of this study are available from the corresponding author upon request.

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Author contributions

Z.F. & Xiaorong Z. performed sample growth, electrical, structural and magnetic measurements with assistance from L.W., Z.Z., H.G., X.W., H.Y., P.Q., Xin.Z., H.W., H.C., Z.X. & C.J. Theoretical calculations and analysis were performed by L.S., R.G.H., J.S. & T.J. The manuscript

was written by Z.L., Z.F. Xiaorong Z., M.C., L.S., J.S. & T.J. All authors commented on manuscript. This project was conceived and led by Z.L.

Competing interests

The authors declare no competing financial interests.

Additional information

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Figure 1

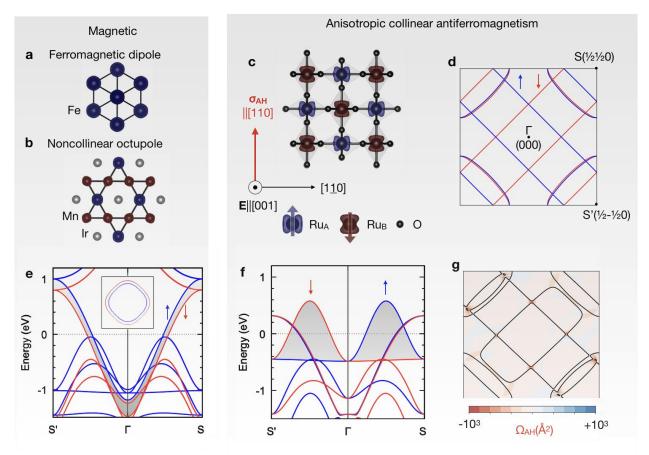


Figure 1 | Anisotropic collinear antiferromagnetism and anomalous Hall Berry curvature in RuO₂ **from first principles. a & b,** Broadly explored time-reversal symmetry breaking mechanism with ferromagnetic dipole (example of Fe magnetization density) and noncollinear octupole (example of IrMn₃ magnetization densities). **c,** The side view of crystal structure of RuO₂ with the two Ru antiferromagnetic sublattices and shown O octahedra. Collinear antiferromagnetic RuO₂ with moment along [110] crystal axis exhibits anomalous Hall vector **σ** along [110] and an anomalous Hall voltage can be measured along [110] for electric field **E** applied along [001]. In panels **d-f**, spin-orbit coupling is set to zero. **d**, Calculated spin-split Fermi surfaces shaped as magnetic toroidal quadrupole. **e,** Energy bands of RuO₂ in an artificial ferromagnetic state exhibits conventional ferromagnetic spin splitting (gray shading and Fermi surface shown in the inset). **f,** Strong momentum dependent spin splitting (gray shading). **g,** Strong crystal momentum resolved Berry curvature in RuO₂ calculated with spin-orbit coupling and moments along [110] corresponding to intrinsic anomalous Hall conductivity vector along [110]. The black contours mark Fermi surface. We set in all first principle calculations U = 1.6 eV and Fermi level to zero.



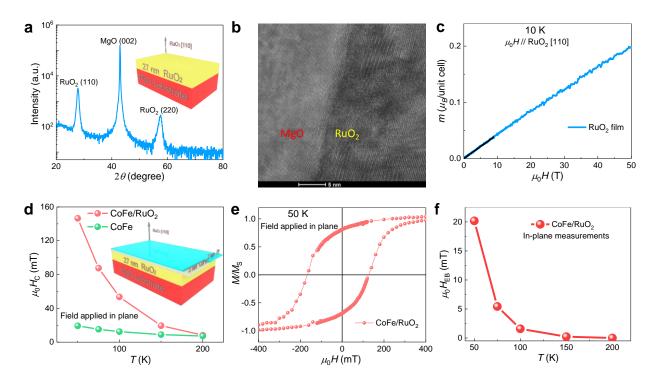


Figure 2 | Antiferromagnetic order in a RuO₂ thin film. a, X-ray diffraction spectrum of an optimized RuO₂/MgO film, indicating a highly ordered (110) orientation of the RuO₂ film. Inset: Schematic of the RuO₂/MgO heterostructure. **b**, Cross-section transmission electron microscopy image of an optimized RuO₂/MgO. **c**, Out-of-plane magnetic moment versus magnetic field of the RuO₂/MgO film at 10 K. The black line represents the static-field data measured up to 9 T. **d**, Comparison of the in-plane coercivity field of Co₉₀Fe₁₀ (CoFe) films in a Pt/CoFe/RuO₂/MgO heterostructure and a Pt/CoFe/MgO heterostructure. Inset: Schematic of a RuO₂/MgO heterostructure capped by a 5-nm-thick CoFe layer and a 2-nm-thick Pt top layer. **e**, Normalized in-plane magnetization of the Pt/CoFe/RuO₂/MgO heterostructure versus temperature.



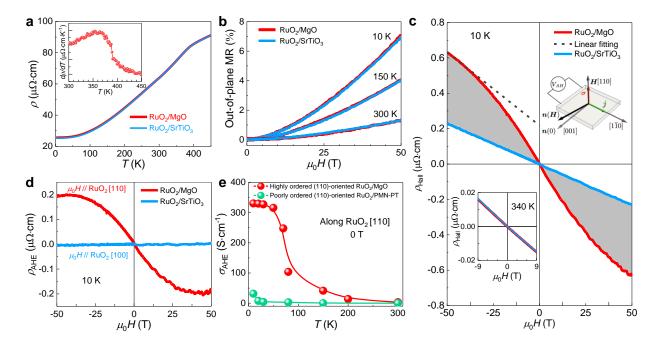


Figure 3 | **Hall effect along the out-of-plane direction of RuO₂ thin films. a**, Resistivity versus temperature for (110)-oriented RuO₂/MgO and (100)-oriented RuO₂/SrTiO₃. The residual resistivity ratio defined as $\rho(300K)/\rho(2K)$ is 2.49 and 2.52 for RuO₂/MgO and RuO₂/SrTiO₃, respectively. Inset: Temperature derivative of the resistivity highlighting the Néel temperature transport anomaly. b, Out-of-skplane longitudinal magnetoresistance up to 50 T in the two films at different temperatures. **c**, Hall signal of the two films up to 50 T measured at 10 K (for other temperatures see Supplementary Fig. 7 & 8). The grey shaded region corresponds to the nonlinear anomalous Hall resistivity. In the top inset we sketch the experimental setup (*j* and *H* refer to the applied electric current and magnetic field, respectively. *n* and *σ* are the Néel and the Hall vector, respectively. *V*_{AH} is the measured Hall voltage.). Hall signals measured at 340 K are shown in the bottom inset. **d**, Anomalous Hall resistivity ρ AHE obtained by subtracting from the total Hall signal a linear Hall component corresponding to the linear fit between -40 and -50 T (see Supplementary Fig. 7). **e**, The anomalous Hall conductivity σ AHE as a function of temperature for high crystal quality RuO₂/MgO and for lower quality RuO₂/PMN-PT (see also Supplementary Fig. 9 & 10).



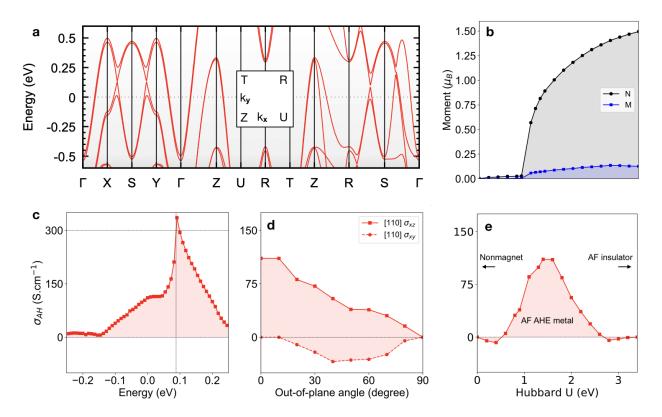
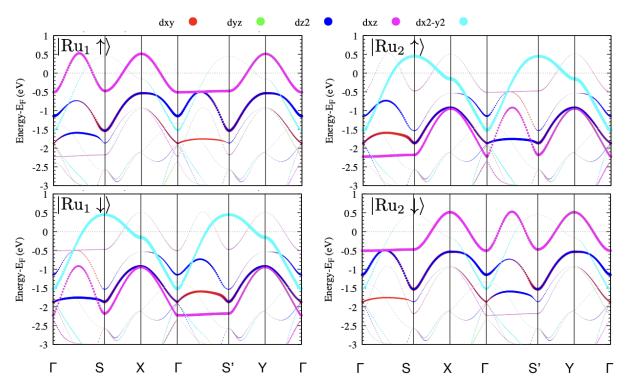
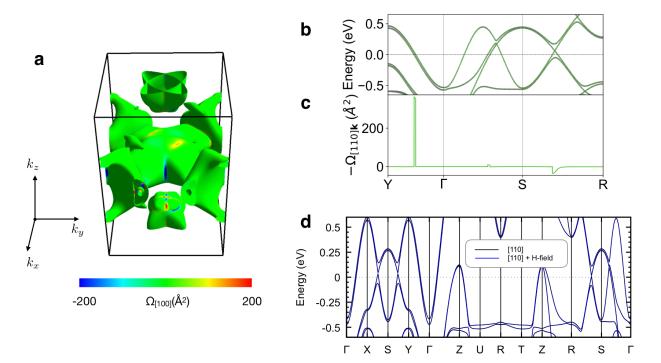


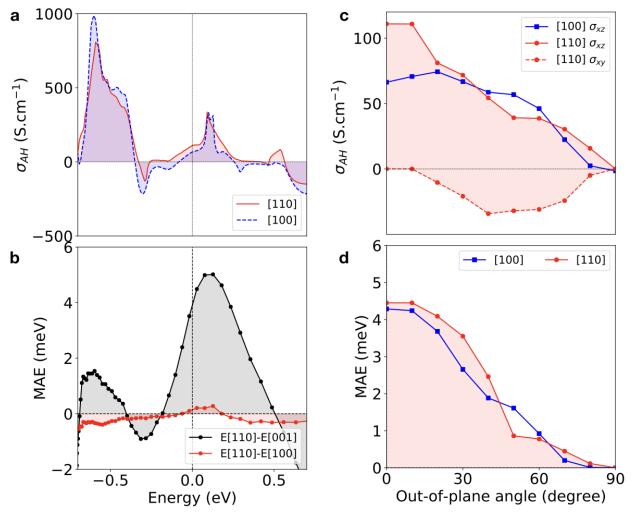
Figure 4 | Anomalous Hall conductivity of RuO₂ calculated from first principles. a, Energy bands of RuO₂ calculated with moments along the [110] direction and spin-orbit interaction (the negligible influence of the magnetic field is shown in Supplementary Fig. 2). In the inset we mark the high symmetry points in the Brillouin zone in the momentum plane $k_z = \pi$. **b**, The magnitude of the antiferromagnetic Néel vector and net moment as a function of the electronic correlation parameter, Hubbard *U. c*, Anomalous Hall conductivity versus energy calculated from Wannier Hamiltonian. **d**, Anomalous Hall conductivity components for the antiferromagnetic vector rotation from the [110] to [001] crystal axes. **e**, Anomalous Hall conductivity dependence on the electronic correlation parameter. Where not specified, Hubbard U = 1.6 eV, Néel vector is along the [110] crystal axis and Fermi energy at charge neutrality corresponds to zero.



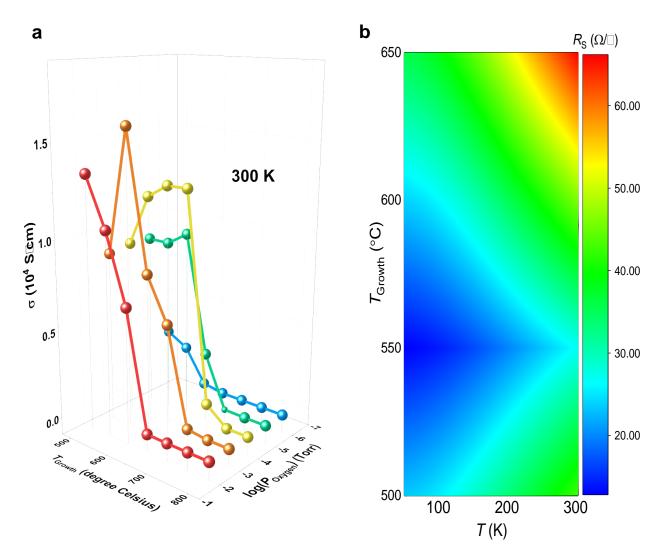
Supplementary Figure 1 | Orbital (Ru *d*-states) and spin resolved energy bands in antiferromagnetic state calculated without spin-orbit interaction in VASP.



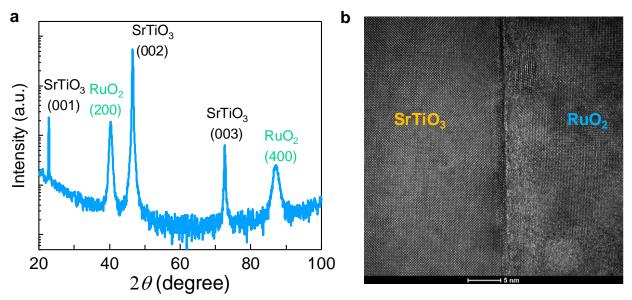
Supplementary Figure 2 | Detailed Berry curvature calculations and energy bands in Zeeman magnetic field. a, Berry curvature component - Ω [100] calculated in Wannier90 code^{S1} on Fermi surface. b, Energy bands and c, Berry curvature - Ω [110] calculated from Wannier Hamiltonian on high symmetry lines in Wannier90 code. All plots use Hubbard U = 1.6 eV and the antiferromagnetic vector along the RuO₂[110] crystal axis. Large Berry curvature arises from spin-orbit coupling gapped nodal feature along the ΓX (ΓY) line (blue color in A). These lines are spin degenerate without spin-orbit interaction. Detailed band structures of RuO₂ without spin-orbit coupling can be seen in the Supplementary Fig. S6 of our previous theoretical work^{S2}. d, Energy bands along high symmetry lines calculated for the Néel vector along [110] (black line) and with addition of Zeeman external magnetic field H ~35 T. The panel was calculated in ELK code.



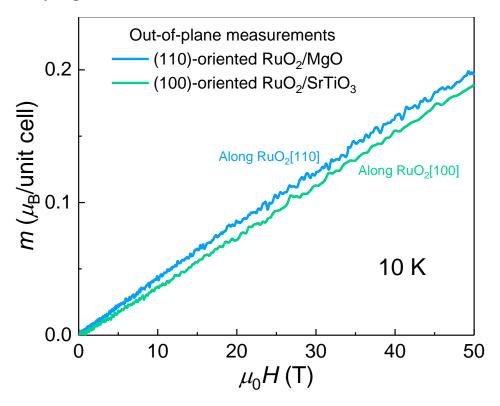
Supplementary Figure 3 | Detailed anomalous Hall conductivity and magnetic anisotropy energy calculations from first principles. a, Anomalous Hall conductivity vs energy calculated for the antiferromagnetic vector along [110] and [100] crystal axis in broader energy window. **b**, Corresponding out-of-plane (E[N||110]-E[N||001]) and in-plane (E[N||110]-E[N||100]) magnetic anisotropy energy (MAE) vs Fermi energy. Calculated dependence of the **c**, anomalous Hall conductivity components and **d**, MAE on the antiferromagnetic vector rotation from [100], to [001] and [110] to [001] crystal axis, respectively.



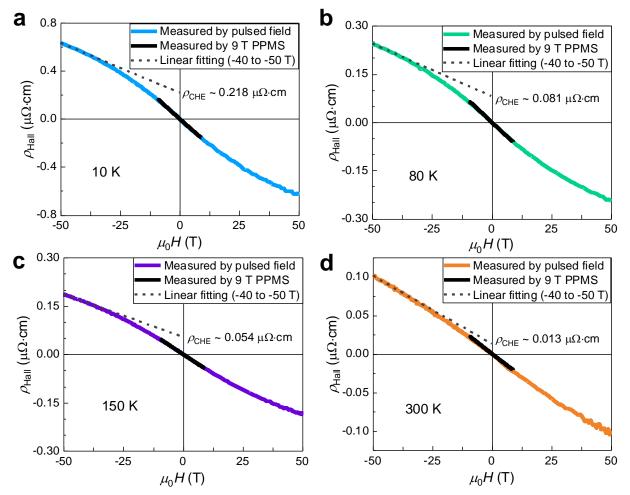
Supplementary Figure 4 | Conductivity optimization and characterization of RuO_2 thin films on MgO (ref. ^{S3}). a. Phase diagram of electrical conductivity of RuO_2 thin films fabricated by pulsed laser deposition at different oxygen pressures and deposition temperatures. b. Temperature-dependent sheet resistance contour mapping of the RuO_2 films fabricated in the growth temperature range between 500 and 650°C.



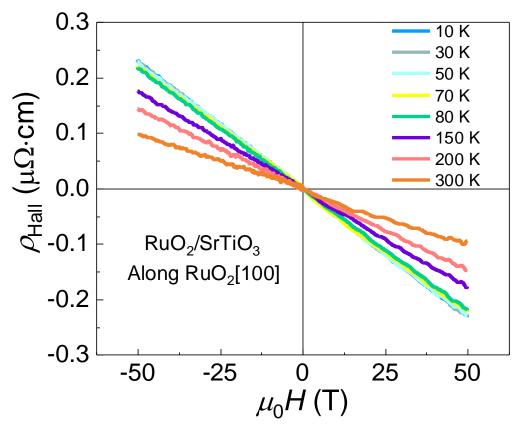
Supplementary Figure 5 | Structural characterization of a RuO₂/SrTiO₃ heterostructure fabricated at 550°C and 10⁻³ Torr oxygen partial pressure. **a.** X-ray diffraction pattern. **b.** Cross-section transmission electron microscopy image.



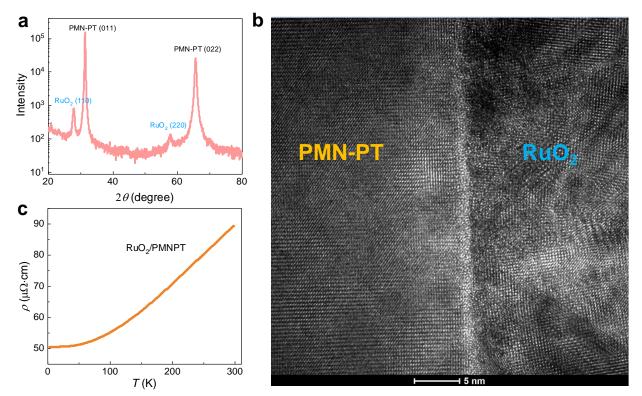
Supplementary Figure 6 | Out-of-plane magnetic moments of RuO₂ thin films obtained by pulsed-field measurements at 10 K.



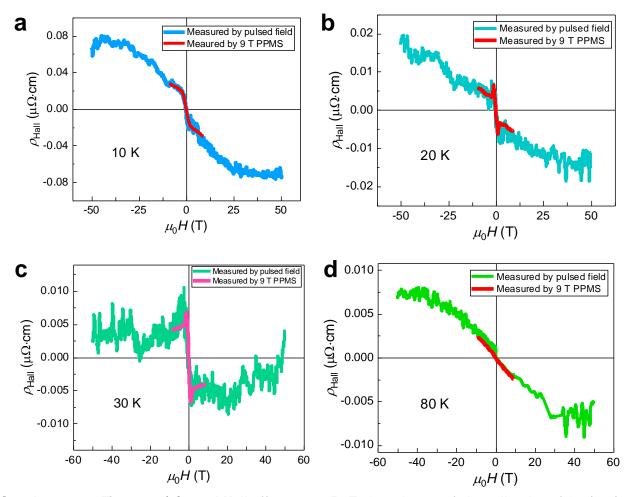
Supplementary Figure 7 | Total Hall signal for the RuO₂/MgO film obtained by pulsed-field and static-field measurements at different temperatures. Dashed lines represent the linear fitting curves of the total Hall data between -40 and -50 T and the extrapolated zero-field Hall resistivity values are used to represent the zero-field crystal Hall resistivities.



Supplementary Figure 8 | Total Hall signal of the (100)-oriented RuO₂/SrTiO₃ film measured at various temperatures.



Supplementary Figure 9 | RuO₂ film grown on a (011)-oriented PMN-PT (0.7PbMg_{1/3}Nb_{2/3}O₃– 0.3PbTiO₃) single-crystal substrate deposited at 550°C and 10⁻³ Torr oxygen partial pressure. a. Xray diffraction spectrum of the RuO₂/PMN-PT heterostructure, indicating the (110) orientation of the RuO₂ film^{S3}. b. Cross-section transmission electron microscopy image of the RuO₂/PMN-PT heterostructure. **c.** Temperature-dependent resistivity of the RuO₂/PMN-PT heterostructure.



Supplementary Figure 10 | Crystal Hall effect up to 50 T along the out-of-plane direction of the (110)oriented RuO₂/PMN-PT heterostructure obtained by pulsed-field and 9 T-PPMS measurements at different temperatures.

Hall effect	Antiferromagnet	Longitudinal resistivity ρ _{xx} (μΩ·cm)	Anomalous Hall resistivity ρ _{ΑΗΕ} (μΩ·cm)	Anomalous Hall conductivity σ _{AHE} (S·cm ⁻¹)
Topological	Mn_5Si_3 (Ref. ^{S4})	240	0.001-0.04	0.02-1
Anomalous	Mn₃Sn (Ref. ^{s₅})	200	4	100
Anomalous	RuO ₂ (Present work)	26	0.21	330

Supplementary Table 1 | Comparison of anomalous Hall effects in antiferromagnets.

Temperature (K)	ρ _{xx} under zero field (μΩ·cm)	ρ _{АНЕ} extrapolated to zero field (μΩ·cm)	σ _{AHE} for zero field (S⋅cm⁻¹)	Single-band carrier density estimation based on the -40~ 50 T linear slope (holes/cm ³)	Single-band mobility estimation based on the - 40~-50 T linear slope (cm ² ·V ⁻¹ ·s ⁻¹)
10	25.67	0.218	330.83	0.75×10 ²³	3.24
80	27.94	0.081	103.76	1.92×10 ²³	1.16
150	36.19	0.054	41.23	2.35×10 ²³	0.73
300	63.74	0.013	3.20	3.53×10 ²³	0.28

Supplementary Table 2 | Physical parameters for the RuO₂/MgO heterostructure.

Supplementary References

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S4. Sürgers, C., Winkel, G. P. & Löhneysen, H. v. Large topological Hall effect in the non-collinear phase of an antiferromagnet. *Nat. Commun.* **5**, 3400 (2014).

S5. Nakatsuji, S., Kiyohara, N. & Higo, T. Large anomalous Hall effect in a non-collinear antiferromagnet at room temperature. *Nature* **527**, 212–215 (2015).