

CrossMark  
click for updatesCite this: *RSC Adv.*, 2016, 6, 83589

# Recent advances in the synthesis and application of photocatalytic metal–metal oxide core–shell nanoparticles for environmental remediation and their recycling process

Kunal Mondal<sup>\*a</sup> and Ashutosh Sharma<sup>\*b</sup>

Core–shell nanoparticles are a special category of materials with nanostructures that have obtained more attention in recent times owing to their fascinating properties and extensive choice of applications in catalysis, photocatalysis, materials chemistry, biology, drug delivery, sensors and other electronic device applications. By wisely modifying the cores along with the shells material and morphology, a variety of core–shell nanostructures can be created with tunable properties that can perform vital roles in several catalytic and photocatalytic developments and promise sustainable solutions to current environmental remediation problems. Recently, the development of core–shell nanoparticles with Au, Ag, Pt, Pd, Zn, Ni etc. metals as a core and ZnO, TiO<sub>2</sub>, SiO<sub>2</sub>, Cu<sub>2</sub>O, Fe<sub>2</sub>O<sub>3</sub> and SnO<sub>2</sub> etc. metal oxide semiconductors as a shell has engrossed huge research interest in catalysis, photocatalysis, solar photovoltaics and so on owing to their tunable nanoscale functionalities in the core and shell particles. This review covers the core–shell nanomaterials, mainly, metal–metal oxide core–shell nanostructures with an overview of the recent advances in their synthesis, and applications. It includes a critical review of application of these materials towards photocatalytic wastewater purification and bacterial disinfections under UV and visible light irradiation. It also includes a brief discussion on mechanisms of heterogeneous and homogeneous photocatalysis and the effect of different morphologies of photocatalysts in view of photocatalysis for wastewater treatment. Finally, recycling, reuse and actual photocatalysis in a photocatalytic reactor of core–shell nanostructures have also been discussed which is very important for a sustainable wastewater treatment. The future prospects of such core–shell nanoparticles for cancer treatment by hyperthermia, drug delivery and several other interesting applications have been covered with a distinct emphasis on their structural depending properties.

Received 15th July 2016  
Accepted 25th August 2016

DOI: 10.1039/c6ra18102c

www.rsc.org/advances

## 1. Introduction

Core–shell nanostructured materials have developed significant research interest for the last couple of decades owing to their impending uses in several fields such as heterogeneous photocatalysis, energy storage, biosensing, gas sensing, biomedical and electronic and bioelectronics applications, and many others.<sup>1–12</sup> Interestingly, a recent investigation shows that around one thousand reports have been published per year on core/shell nanoparticles mentioning their synthesis and applications.<sup>13</sup> The core–shell nanostructured materials can be fabricated with diverse shapes and sizes of varying core as well as shell widths by means of different structural morphologies. These nanoparticles could be of any shape like centric, eccentric, spherical,

star-shaped, fibrous or tubular and so on.<sup>14</sup> The physicochemical properties of the core–shell nanostructures may be modified simply by adjusting their size and shape focusing on their particular applications.<sup>14–16</sup> Different core–shell nanostructures have many distinct characteristics which have been used in various fields of biomedical and nanomedicine, like drug delivery,<sup>17</sup> bioimaging,<sup>18</sup> cancer treatment,<sup>19</sup> so on and so forth.<sup>20</sup> Recently, core–shell nanoparticles are engineered with solid core and porous shell such as porous metal–silica, metal–zeolite *etc.*, have been progressively used for extremely capable separation with reasonably low back pressure and quick flow rate.<sup>21</sup> The macroscopic solid core and shell offers low working back pressure whereas the nanoporous shell and small solid core can deliver very higher surface area for the separation to happen. When the surface morphology of the nanoparticles is altered by means of coating with a thin film of other materials, adding various chemical functional groups or even attaching other molecules they display better material characteristics than the uncoated nonfunctionalized nanoparticles. Interestingly, there

<sup>a</sup>Department of Chemical and Biomolecular Engineering, North Carolina State University, 911 Partners Way, Raleigh, North Carolina 27695, USA. E-mail: kmondal@ncsu.edu

<sup>b</sup>Department of Chemical Engineering, Indian Institute of Technology Kanpur, Kanpur-208016, Uttar Pradesh, India. E-mail: ashutos@iitk.ac.in

are various kinds of core–shell nanostructures are possible, such as metal-core and semiconductor shell, metal-core and nonmetal shell, metal-core and different metal shell, semiconductor-core and metal shell, semiconductor core and dissimilar semiconductor shell, metal-core and polymer shell, nonmetal-core and nonmetal shell, semiconductor-core and polymeric shell, polymeric-core and nonmetallic shell and core and shell of two dissimilar polymers. Intended for these types of nanostructures, the material of the core and the shell may be interchanged. There is a schematic illustration of a metal-core and semiconductor-shell nanoparticles which is shown in Fig. 1.

The design and production of innovative core–shell nanostructured nanomaterials with metals, polymers, semiconductors, alloys, graphene, carbon nanotubes, and quantum dots *etc.* that own extraordinary functionalities are one of the key challenges in core–shell nanoparticle research.<sup>22–28</sup> In a typical core–shell morphology, the core and the shell can be formed by different materials or the identical materials but with dissimilar structures.<sup>1,29–32</sup> Fig. 2 shows the schematic depiction of different kinds of core/shell nanoparticle morphologies where the core and the shell are conveyed in different colors. It was observed that sometimes the core may be a single sphere (Fig. 2a) or aggregation of many continuous shells (Fig. 2b). It is also possible that the shell is continuous and incorporated several small spheres within it (Fig. 2c). Complex core–shell assemblies may also be made where the shell may be made of either many small identical spherical particles or different spheres shown in Fig. 2d and e, respectively. The shell structure can be a porous layer (Fig. 2f) when core is nonporous solid or both the core and the shell can be porous (Fig. 2g). It is also found that the core is in the form of multiple small sphere which are covered by a continuous single shell (Fig. 2h) or a shell layer with several spherical particles (Fig. 2i). Interestingly, the core–shell morphology can be made in the form of solid fibers as shown in Fig. 2j. Also, the core–shell structure can form yolk–shell structure where there is a movable core situated inside a hollow shell (Fig. 2k).<sup>33</sup> The selection of morphology of core–shell particles depends on specific application, for example, the core–shell particles employed in chromatography are generally made of the similar material, commonly SiO<sub>2</sub>, however, with a porous shell and solid core. The size of the core

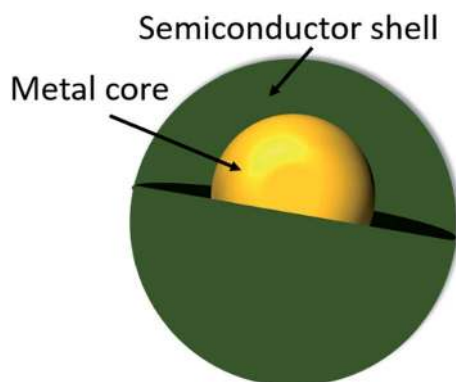


Fig. 1 An illustration of a spherical metal–semiconductor core–shell nanostructure.

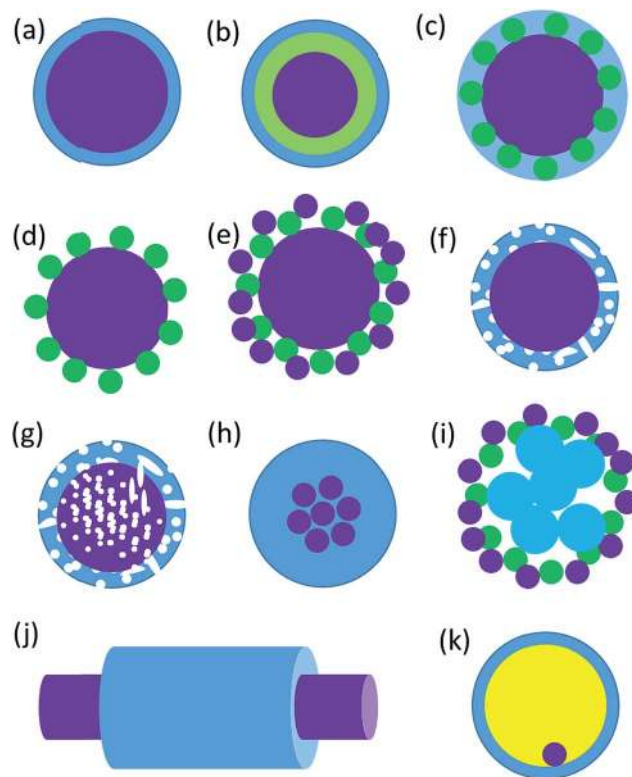


Fig. 2 Different core–shell nanostructures: (a) core is a single sphere, (b) core with multiple concentric shells, (c) shell incorporated with smaller spheres, (d) shell in the form of identical smaller spheres, and (e) shell in the form of different spheres, (f) solid core and porous shell, (g) core–shell both porous, (h) multiple spherical cores, (i) multiple cores covered by shell formed by several small spheres, (j) core–shell fiber, and (k) yolk–shell morphology shows a movable core within a hollow shell.

particle, the thickness of the shell material and the porosity in the core or shell are adjusted to garb different kinds of applications. In case of photocatalytic application, the porous semiconducting shell and conducting core particles are preferred.

In this context, the metal and metal oxide semiconductor, polymer core–shell have engrossed substantial curiosity as a class of innovative nanomaterials with interesting uses in numerous electronics containing organic photovoltaics solar cells, sensors, organic light-emitting diodes, and field effect transistors *etc.* owing to their benefits, such as cheap price and stress-free device fabrication capabilities.<sup>13,34–37</sup>

The core–shell nanoparticles are designed by numerous fabrication methods like sol–gel process, hydrothermal production, emulsion polymerization technique, micro-emulsion polymerization, solvothermal synthesis, and chemical vapor deposition method *etc.*<sup>38–44</sup> These particles are typically fabricated by a dual-step or many-step synthesis route. Firstly, the core is made and then shell is formed on the core particle through various techniques subjected to the variety of core–shell nanostructured materials and their surface morphologies.<sup>45</sup> Interestingly, the core–shell nanostructured nanoparticles have been mostly explored, as compared to core–shell microspheres.

Recently, copious research consideration has been dedicated to metal–semiconductor core–shell particles created on silver, gold, platinum, and palladium *etc.*, core and titanium dioxide, silica, zirconia, zinc oxide *etc.*, shell since their nanoscale properties which distinctly vary from their bulk counterparts.<sup>2,46–48</sup> These nanoparticles of metal–semiconductor core–shell morphologies exhibit size-reliant quantum-size effects<sup>49</sup> and can be utilized for a variety of applications requiring advanced functionality such as sensors, energy storage, electronics, optoelectronics, catalysis and photocatalysis.<sup>50</sup> The catalytic and plasmonic characteristics of the novel metals and the photocatalytic properties of the semiconductors open up a new area of application for these metal–semiconductor core–shell nanoparticles as a promising photocatalyst material.<sup>51–53</sup> The metal–semiconductor core–shell nanostructured photocatalysts can modify greatly semiconductor energy band diagram, lower the band gap energy and causing a slow electron–hole pair recombination rate with a fast electron transfer ability.<sup>54</sup> Thus, these core–shell photocatalysts with metal–semiconductor nanoparticles can perform efficient visible light photocatalyst and find interesting applications in the field of air and wastewater purification, bacterial disinfections and so on. Therefore, the motivation in the exploration of core–shell nanoparticles is to associate the chosen belongings of dissimilar materials and nanostructures in order to provide synergistic result, to stabilize the active nanoparticles, or to deliver biocompatibility into the functional matrix and explore various applications accordingly.

In this review we have discussed the challenges and recent advancement in the synthesis of metal–semiconductor core–shell nanostructures and their application prospects utilizing their interesting structures and catalytic and electronic properties. Also the photocatalytic reaction mechanisms for the core–shell nanostructures and their potential applications for environmental remediation, in particular towards waste treatment by organic pollutant removal and bacterial disinfection from wastewater, air purification by UV and visible light mediated photocatalysis.

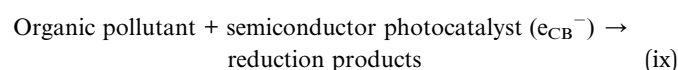
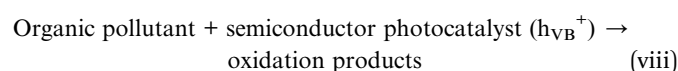
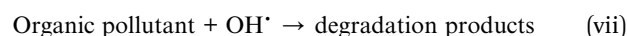
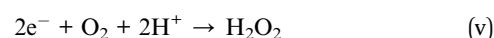
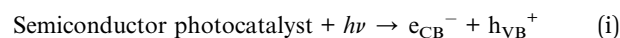
## 2. Mechanisms of photocatalysis

The energy difference between the valence band (VB) and the conduction band (CB) of the semiconductor is acknowledged as the 'Band Gap'. When TiO<sub>2</sub>, ZnO, ZrO<sub>2</sub>, and CdSe *etc.*, photocatalyst semiconductor absorbs ultraviolet energy from sunlight or any irradiated artificial light source (fluorescent lamps, light emitting diodes *etc.*), it yields an electron–hole pair only if the energy is more than the bandgap energy of the material. The conduction band electron ( $e_{CB}^-$ ) of the photocatalyst gets excited because of the exposure of the illuminated light energy. This phase is said as the photocatalyst's 'photo-excitation' state. The photo generated electron is negative and the hole is positively charged. The positively charged hole of the photocatalysts in valence band ( $h_{VB}^+$ ) splits the water molecule into hydrogen gas and high energetic hydroxyl free radical. The electron counters with oxygen and produces super oxide anion. This cycle lasts until the light energy is accessible.

The photocatalysis process and the related chemical reactions are elucidated as follows with the help of eqn (i)–(ix). The reaction mechanism for ZnO and TiO<sub>2</sub> photocatalysts are explained by Sharma's group.<sup>55,56</sup> By way of the semiconductor photocatalyst exposed with illuminated light, the valence band electron goes to the conduction band creating a positively charged hole in the valence band. The photocatalysts would be further efficient if these  $h_{VB}^+$  will spend longer time prior to recombine. The  $e_{CB}^-$  electrons counter with oxygen and  $H^+$  of the aqueous medium and produce H<sub>2</sub>O<sub>2</sub> which further originated into OH<sup>-</sup> ion and OH<sup>•</sup> free radical. Likewise,  $h_{VB}^+$  respond with H<sub>2</sub>O to contribute OH<sup>•</sup> free radical. Creation of these highly energetic reactive species supports to reduce the recombination rate of electron–hole pair and provides additional time for dealings with the organic pollutant molecules. Fig. 3 shows idea of the photocatalysis thorough a pictorial illustration.

The mechanisms of two other situations (a) the CB of photocatalyst is less negative than the O<sub>2</sub>/O<sub>2</sub><sup>-</sup> redox potential and (b) the VB of photocatalyst is less positive than the H<sub>2</sub>O/OH<sup>-</sup> redox potential. To achieve overall wastewater treatment, the energy necessities indicate that the bottom of the CB must be located at a more negative potential than the reduction potential of H<sup>+</sup>/H<sub>2</sub> (0 V vs. NHE at pH 0), whereas the top of the VB essentially be located more positively than the oxidation potential of H<sub>2</sub>O/O<sub>2</sub> (1.23 V vs. NHE). Conferring to this theoretical value, the photocatalysis can only be directed if the photon energy is greater than 1.23 eV. This amount energy is equal to the energy of a photon (wavelength of ~1010 nm), signifying that visible light holds enough energy for the degradation of wastewater in aqueous medium.

The photocatalytic oxidation of organic compounds under UV light can be presented as follows:<sup>56</sup>



The recombination of the excited electron–hole pair produces excess energy from the excitation of the electron which releases in the form of heat. As the recombination process is unwanted and causes to an incompetent photocatalysis, therefore the

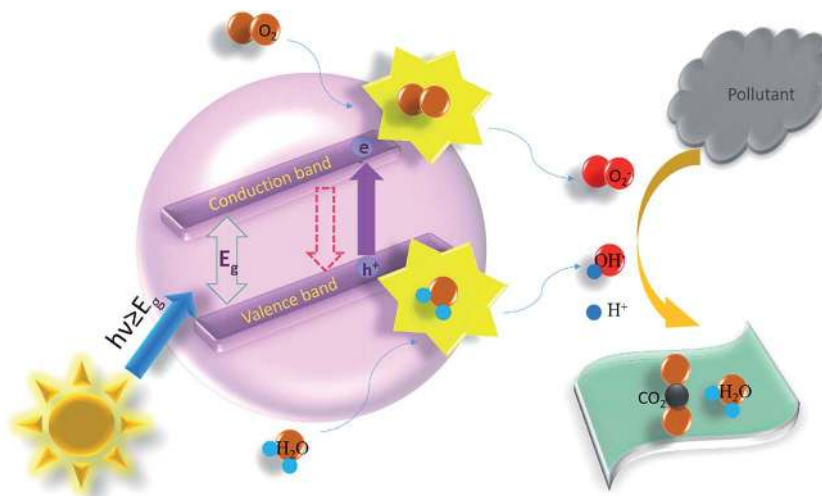


Fig. 3 The idea of degradation of pollutant by heterogeneous photocatalysis. Here, the redox potentials ( $E^0$ ) of some active radicals, such as  $H^+$ / $H_2$ ,  $O_2/O_2^-$  and  $H_2O/OH^-$  should be noted as 0 V,  $-0.33$  V and  $-0.42$  V, respectively.<sup>57</sup>

crucial aim of the photocatalysis is to achieve a catalytic reaction among the excited  $e_{CB}^-$  electrons and oxidant to harvest a reduced product. Furthermore, a reaction among the generated with a reductant is a must desirable to yield an oxidized product. It is important to mention that the reduction and oxidation reactions occur at the surface of photocatalysts since this reaction creates positive holes and electrons. In the oxidative reaction, the positive  $h_{VB}^+$  counter with the moisture content on the catalyst surface and yield a hydroxyl free radical. In addition, if more oxygen is supplied from outside to the photocatalytic reactor, this will perform as electron acceptors and can boost the degradation of pollutant by further delaying the recombination rate.<sup>55</sup> Also, the total time for the complete pollutant degradation increases with the reduction in photocatalyst loading. Fig. 4 depicts a typical photocatalytic degradation of a PAH pollutant naphthalene in wastewater where it

describes the gradual degradation of the pollutant by measuring its absorption with time.

## 2.1 Heterogeneous photocatalysis

Heterogeneous photocatalysis is a well-developed branch of chemical catalysis appreciated for wastewater treatment and air purification. In case of heterogeneous catalysis, the phase of the catalyst is different from the reactants. Heterogeneous photocatalysis contains a large variety of reactions, such as dehydrogenation, minor or total oxidations, hydrogen transfer, water detoxification, metal deposition, deuterium-alkane isotopic exchange, gaseous and aqueous pollutant removal, *etc.* The utmost common examples of heterogeneous photocatalyst are the transition metal oxides and semiconductors, which have exceptional features.

## 2.2 Homogeneous photocatalysis

In a homogeneous photocatalytic reaction, the reactants and the photocatalysts are exist in the similar phase. The ozonolysis and photo-Fenton systems are the most commonly employed homogeneous photocatalysts. Fenton revealed that a combination of hydrogen peroxide and Fe(II) in acidic environment have very dominant oxidizing properties.<sup>58</sup> However, the exact mechanism of this reaction, now identified as the Fenton reaction, is still the topic of some debate.<sup>59,60</sup> It is usually presumed to be an imperative chemical source of hydroxyl radicals. The classical mechanism can be demonstrated as a simple redox reaction in which Fe(II) is being oxidized to Fe(III) and  $H_2O_2$  and further reduced to hydroxide ion and the hydroxyl radical.

Here in this catalysis the reactive species  $\cdot OH$  is employed for diverse commitments. The hydroxyl radical production mechanism of by ozone can occur in the following paths.<sup>61</sup>

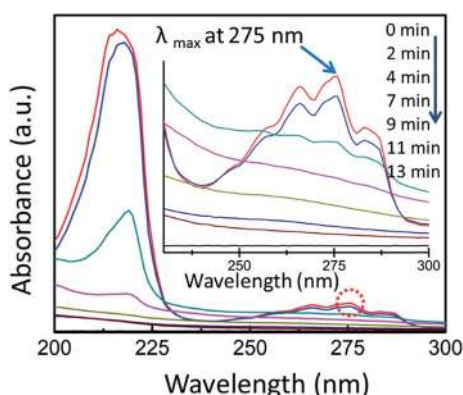
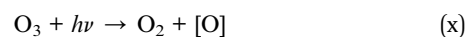
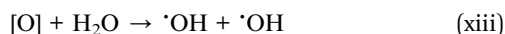
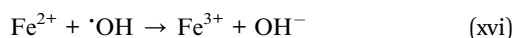
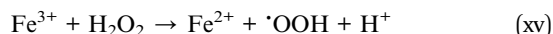
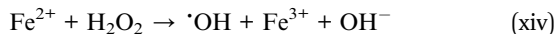


Fig. 4 Monitoring of a typical photocatalysis by UV-vis absorption spectrophotometer. Here, a typical aqueous naphthalene solution was mixed with  $TiO_2$ -NF and data collected at different time interval. The inset image shows magnified  $\lambda_{max}$  (monitored at 275 nm wavelength). Reprinted with permission from ref. 56. Copyright (2014) American Chemical Society.<sup>56</sup>

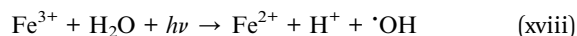
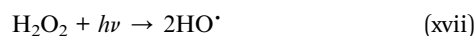




In the same way, the Fenton system yields hydroxyl radicals by the following way:<sup>62</sup>



In a photo-Fenton kind reaction processes, additional sources of  $\cdot\text{OH}$  generated over photocatalysis of hydrogen peroxide, and reduction of  $\text{Fe}^{3+}$  ions in presence of light energy:<sup>63</sup>



The productivity of the photon-Fenton kind routes is governed to some operational parameters such as pH of the solution, concentration of  $\text{H}_2\text{O}_2$ , and intensity of UV light. The key benefit of this reaction is the capability of expending solar energy spectrum up to 470 nm, and can escaping the use of high priced UV and electrical energy sources. The photon-Fenton processes have been recognized much competent than the other photocatalytic reactions. In spite of that, the main drawbacks of these process are the low operating pH standards which are necessary for the reactions, since iron precipitates at higher pH.

### 3. Effect of photocatalyst morphology on photocatalysis

The photocatalytic oxidation process be influenced by creation and recombination of electron-hole pair within the photocatalyst. The oxygen molecules absorbed on the surface of the semiconductor photocatalyst performs as an electron prisoner and can govern the photo induced electron-hole pair recombination process. The nanostructured semiconductor owns superior photocatalytic competence as compared to the bulk photocatalysts.<sup>64,65</sup> The causes behind this can be explained by quantum size effect and higher specific surface area.

The size of the catalyst particles when turn out to be less than a certain precarious limit (*i.e.*, size decreases to the nanometer size range) this causes quantum size effects owing to the confine movement of electrons. The quantum size effect is one of the utmost direct effects and this can leads to the conduction band and valence band of the semiconductor catalyst to change into discretized energy states. This discretization process depends on the size of the material structure, which indicates that either the potential of valence band alters more positively, or the conduction band potential becomes further negative.

Thus, the red-ox potential of the electrons and holes is augmented, and thereby the oxidation reactivity of nano-structured photocatalysts is improved.<sup>66,67</sup>

The other important factor is the higher specific surface area of the catalysts. The adsorption capacity of the semiconductor photocatalysts to organic waste materials increased if there are additional atoms be present onto the surface. The photocatalytic activity of the catalyst is related to the time period drained by electron-hole pairs to get in contract to the particles surface. When the sizes of the particle are in nano-regime the diameter of those catalysts becomes very minute, then the movement of the charge carriers from inside to the surface becomes very straightforward, and so benefits to red-ox reaction. Greater the surface area to volume ratio, lesser the diameter of the particle and the shorter time may be taken up by electron-hole pairs diffusing to the surface from inside of the catalyst. This can deliver reduced probability for the recombination of electron and hole. Consequently, the more photocatalytic action can be attained. Therefore, the nanosized semiconductors have superior photocatalytic proficiency than the bulk.<sup>68,69</sup>

The catalytic activity greatly influenced by porous nature and porosity distribution of the photocatalyst. The optimum porosity distribution in a micro (pore size < 2 nm)/mesoporous (pore size ~ 2–50 nm) catalyst is essential for efficient catalysis. Auyeung *et al.*<sup>70</sup> demonstrated the method for controlling structure and porosity in catalytic nanoparticles. In case of heterogeneous catalyst with an enforced nanotexture, such as zeolite, mesoporous sieve *etc.* the distribution of macropores and large mesopores helps if efficient transportation reactants to overcome diffusion limitations during catalytic reaction. The pore size and porosity distribution are of great significance to the practical use of sensibly designed nanoporous catalysts, as the absence of an elegant porous interconnected network assembly at greater length scales might unfavorably affect the overall photocatalytic performance. Interestingly, the presence of big pores in channels ease access to the active catalytic sites, however, it diminishes the quantity of active catalysts. Thus, there is a clear need of an optimally distributed porosity. In this context, Coppens group demonstrated that the catalysts usually get benefit from a hierarchically porous network with an extensive pore-size distribution.<sup>71</sup>

#### 3.1 Nanostructured metal-oxide semiconductor photocatalysts

A nanostructured semiconductor photocatalyst is a class of semiconducting nanomaterials that produces catalytic activity by means of light energy. Semiconductor metal-oxides nano-structured photocatalysts have attained excessive technological significance in electronics and environmental remediation. The encouraging electronic structural arrangement, superior light absorption abilities, and favorable charge transport properties of the metal-oxide nanomaterials has made conceivable its use as photocatalyst.<sup>55</sup>

The noteworthy features of the photocatalysts are the appropriate band gap, preferred morphology, high surface area

to volume ratio, chemical and thermal stability and most important the reusability.<sup>72–75</sup> Titanium dioxide, zinc oxide, zirconium dioxide, cadmium selenide are the typical photocatalyst materials. Other metal-oxides like oxides of vanadium, tin, cerium and chromium, possessing these features show comparable photocatalysis and are able to oxidize organic pollutant molecules to degrade completely or else convert them to less hazardous derivatives.

There are many nanomorphologies of photocatalysts such as nanoparticles, thin films, nanospheres, nanofibers, nanotubes, nanoribbons and core-shell nanostructured photocatalysts described in the literature for their exciting photocatalytic activities. Among those nanostructures core-shell nanostructured photocatalysts are the most efficient one.

There are lot of semiconductor metal-oxides nanostructures plentifully available in nature such as ZnO, SnO<sub>2</sub>, TiO<sub>2</sub>, and CeO<sub>2</sub>, BiVO<sub>4</sub>, Bi<sub>2</sub>WO<sub>6</sub>, InTaO<sub>4</sub>, Zn<sub>1.7</sub>GeN<sub>1.8</sub>O, ZnAl<sub>2</sub>O<sub>4</sub>, ZnGaNO and so on which have also been extensively used as photocatalysis, predominantly as heterogeneous photocatalysis application since quite a few decades. Metal-oxides nanostructured semiconductors are very useful in chemical catalysis due to their excellent chemical stability in a diverse environmental conditions, biocompatibility, and competence to create electron-hole pair when irradiated with necessary expanse of photon energy owing to their nanoscopic properties.

### 3.2 Nanotube

There is an ongoing research exertion to exploit the distinctive structures of these nanoscale photocatalysts by altering parameters such as shape, size, and arrangement of patterns. One-dimensional (1D) nanostructured morphologies such as nanotubes, nanorods, and nanofibers, have turn out to be a hot research topic in the area of photocatalysis due to their exceptional physicochemical belongings and anisotropic configurations.<sup>77–79</sup> Efforts have been made on various processes parameters to obtained different shaped nanostructures and among these, tubular and rod shaped structures are much explored. The photocatalyst with nanotubes have structurally stable to several hundred degree Celsius temperature. The nano interface, wall thickness, and tube-to-tube interaction points seem to have important roles in defining the photocatalytic performance.<sup>80</sup> In the recent past, Kang *et al.*<sup>81</sup> have synthesized nanotube nanostructures of titanium dioxide photocatalyst and demonstrated towards the photocatalytic reduction of methylene blue. They have observed that within the slight temperature range, the titania nanotube arrays showed a binary combination of brookite and anatase crystalline phases which are highly effective for photolysis. Grimes's group has fabricated tapered, conical-shaped TiO<sub>2</sub> nanotubes expending anodic oxidation process.<sup>80</sup> Interestingly, Ratanatawanate *et al.*<sup>82</sup> have been synthesized S-nitrosocysteine ornamented PbS quantum dots/TiO<sub>2</sub> nanotubes for better creation of singlet O<sub>2</sub> where they have used the nanomorphology as a nitric oxide releasing vehicle to promote photocatalysis. In a recent study Gupta and his co-workers have been developed extreme-dense ZnO nano-forest comprising of vertically grown high aspect ratio nanotube/nanorod like nanostructures

and demonstrated their prospective as photocatalysts.<sup>76</sup> Interestingly, Fig. 5 shows the typical FESEM images of 1D zinc oxide nanotube/nanorod nanomorphologies pinned in nanocarpet which was reported by Gupta *et al.*<sup>76</sup>

### 3.3 Nanofibers

Nanofibre photocatalyst is one of the advanced expertise and broadly used due to their large surface area to volume ratio and has tunable nanopore morphology. It provides a structural flexibility and superior functionality of its surface and furthermore the mechanical properties of nanofibre such as the tensile strength is surprising.<sup>83,84</sup> Amongst the enormous number of fabrication processes aimed to achieve 1D nanofiber nanostructures, the electrospinning method is a most opportune and scalable technique.<sup>85,86</sup> Nanofibers attained by the employing electrospinning technique have copious notable characteristics, for instance, a high surface area, tunable huge porosity, and exceptional substrates for the gathering of secondary nanostructured morphologies.<sup>87</sup> Currently, nanofibers photocatalysts have involved much consideration owing to their striking features of 1D nanostructure for uses in photocatalysis.<sup>88–93</sup> This has also been conveyed that electrospun nanofibrous photocatalysts were advantageous to the vectorial transportation of photocreated charge carriers through their grain boundaries, causing an enriched separation of electron-hole pairs compared to other nanomorphologies.<sup>94,95</sup> Therefore, the design and architecture of nanofibrous nanostructured photocatalysts response in the visible-light irradiation are still of major research emphasis. Recently, Singh *et al.*<sup>55</sup> have been demonstrated an innovative approach for the free-standing mesoporous ZnO nanofibrous mats fabrication of by employing electrospinning technique. The FESEM images in Fig. 6 shows the morphology of the calcined (Fig. 6a and b) ZnO nanofiber and free-standing photocatalyst mat (Fig. 6c) with porous crystallinity (Fig. 6d) fabricated by employing electrospinning method. They have shown that the optimized ~60 nm diameter nanofiber photocatalysts are extremely effective in the photocatalysis of the polycyclic aromatic hydrocarbon (PAH) dyes, such as anthracene and naphthalene. In the same context, Sharma's group has been fabricated electrospun partially aligned free-standing mesoporous pure anatase TiO<sub>2</sub> nanofibers' mats for aqueous photocatalysis application.<sup>56</sup> They have demonstrated that the nanofiber photocatalyst mats are kinetically very proficient in whole photocatalysis of aqueous naphthalene solution, which is assisted by the hierarchical porosity and extraordinary surface to volume ratio of the fibrous photocatalysts. The uniform fiber morphology of the electrospun TiO<sub>2</sub> nanofibers (Fig. 7a and b) along with visible mesopores (Fig. 7c) and crystalline nature (Fig. 7d) is shown here in Fig. 7. It is also interesting that the photocatalyst can be engineered easily by doping with desired metal, carbon and other nanocomposites into the nanofibrous structures during electrospinning.<sup>56</sup>

### 3.4 Nanoribbons

In recent times, Feng *et al.* have fabricated various ultrathin nanoribbons/nano sheets of Fe<sub>2</sub>V<sub>4</sub>O<sub>13</sub>,<sup>96</sup> Zn<sub>2</sub>GeO<sub>4</sub>,<sup>97</sup> (shown in

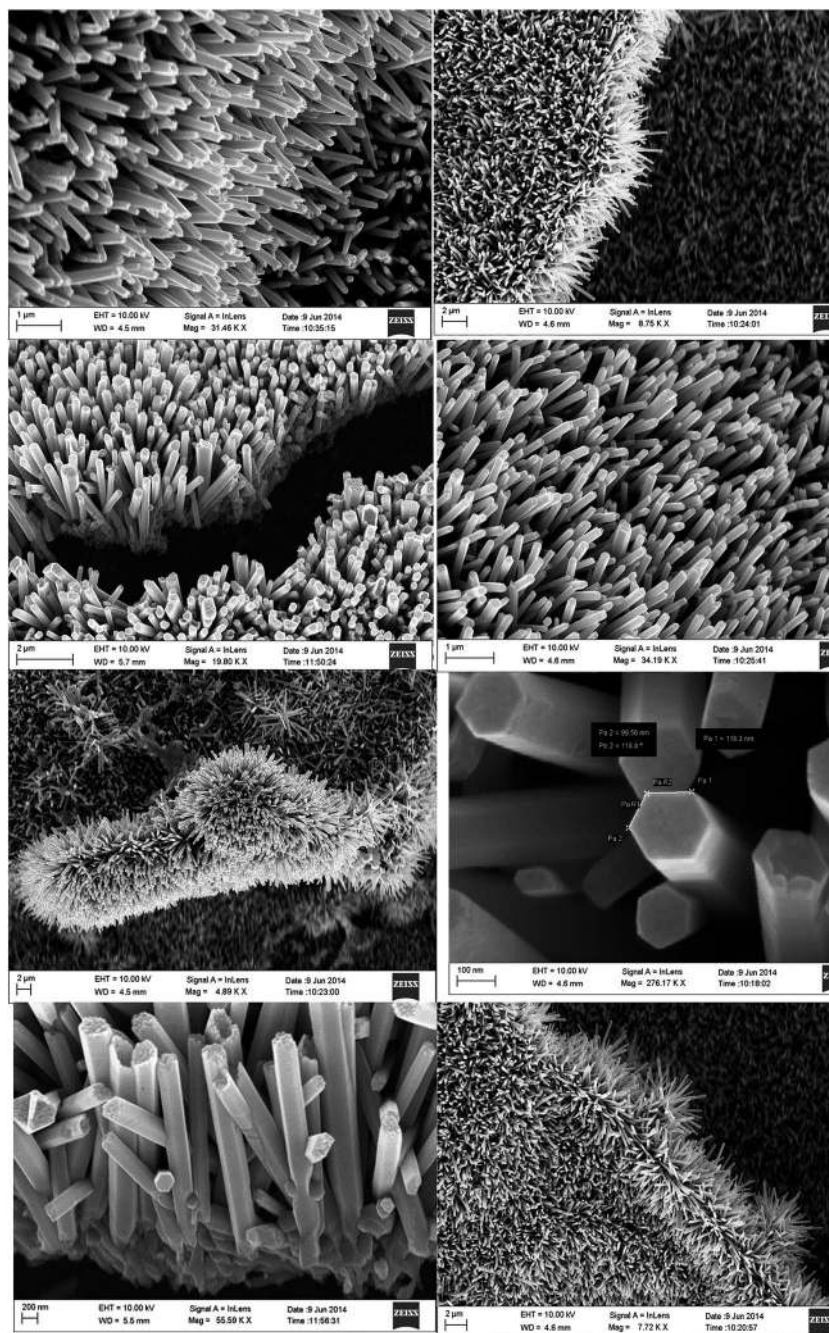


Fig. 5 FESEM micrographs of ZnO nanoforest which were imaged at different zones and magnifications using normal/angled sample stage of the electron microscope. Reproduced with permission from ref. 76. Copyright 2015, Royal Society of Chemistry.<sup>76</sup>

Fig. 8 with their FESEM, TEM, and HRTEM images) and nano-sheets of  $\text{Bi}_2\text{WO}_6$ ,<sup>98</sup>  $\text{TiO}_2/\text{graphene}$ ,<sup>99</sup> and  $\text{WO}_3$ ,<sup>100</sup> and demonstrated them as very efficient and highly selective photocatalysts.

The unusual high photocatalytic activity of the ultrathin nanostructure can be arises because of the thickness scale down to some tens of nanometer which offers a huge specific surface area and nanoscale geometry permits fast mobility of charge carriers from the inside onto the surface to contribute in the photocatalysis in compare with their bulky counter parts. Interestingly, a micron thick, ultra-thin  $\text{Na}_2\text{V}_6\text{O}_{16} \cdot \text{H}_2\text{O}$  nanoribbon has been synthesized by the same group using hydrothermal

technique and the nanoribbons was confirmed to significantly endorse the photocatalytic activity under exposure of visible light.<sup>101</sup> In similarly context, Zhang's group<sup>102</sup> has prepared well-dispersed boron-doped graphene nanoribbons and Shao's and his coworkers<sup>103</sup> have developed single-crystalline  $\alpha\text{-MoO}_3$  nanoribbons for efficient photocatalytic applications.

### 3.5 Other nanostructures

There are many other different nanomorphologies have been explored towards photocatalysis such as nanospheres, thin



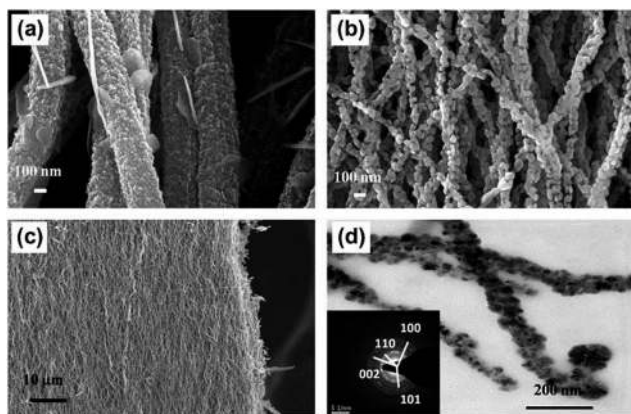


Fig. 6 FESEM micrograph of electrospun ZnO nanofibers formed at two different calcination temperatures; 450 °C (a), 650 °C (b); (c) shows low magnified view of the ZnO nanofiber mat after and TEM image of ZnO nanofibers (d). The selected area electron diffraction (SAED) pattern shows the crystal planes (inset). Reprinted from ref. 55, copyright (2013), with permission from Elsevier.<sup>55</sup>

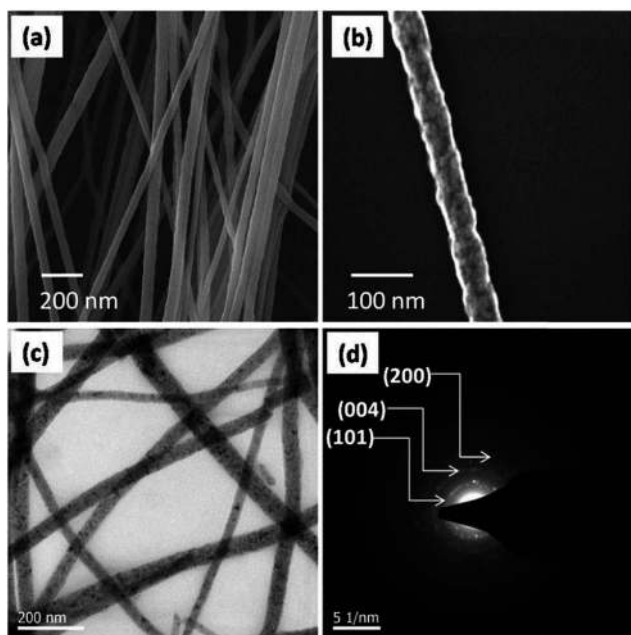


Fig. 7 FESEM image of electrospun TiO<sub>2</sub> nanofiber mat produced at 500 °C (a); (b) shows the higher magnified view of a single fiber; TEM micrograph of TiO<sub>2</sub>-nanofibers (c); and the SAED pattern presentation the crystal planes of anatase TiO<sub>2</sub> (d). Reprinted with permission from ref. 56. Copyright (2014) American Chemical Society.<sup>56</sup>

films, nanocone, nanoribbons *etc.* For example, Zhang *et al.*<sup>104</sup> have been developed a method to produced porous TiO<sub>2</sub> hollow nanospheres.

The photocatalytic activities of the nanocatalysts were assessed by the photodegradation of methyl orange dye under UV irradiation. Their studies indicates that the TiO<sub>2</sub> hollow nanospheres displayed outstanding photocatalysis, improved than that of commercial Degussa P25 in the existence of Cr(VI) owing to the elevated specific surface area. Recently, Bi and his

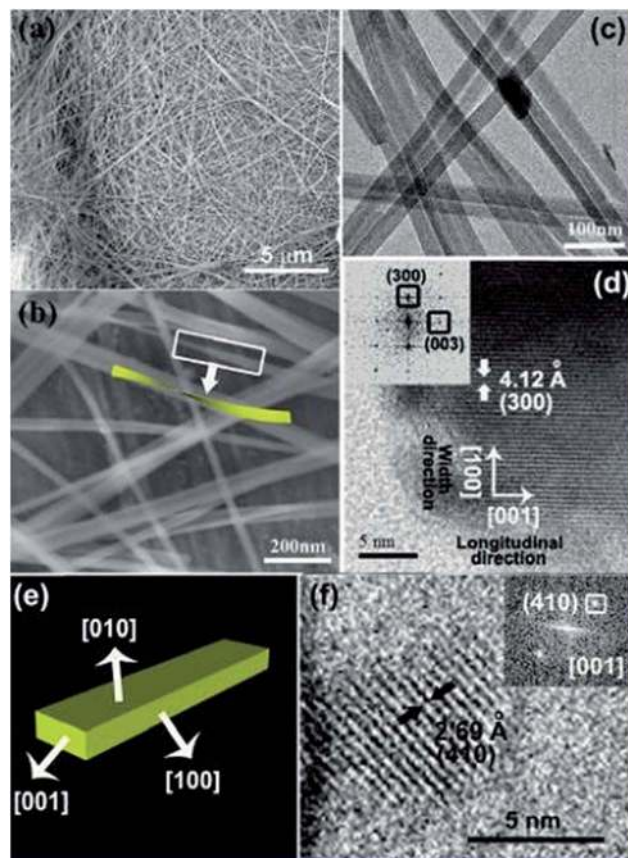


Fig. 8 Zn<sub>2</sub>GeO<sub>4</sub> nanoribbon nanostructures are described by (a and b) FESEM, (c) TEM, and (d) HRTEM images. Structural model of a nanoribbon shown in (e). (f) HRTEM micrograph imaged along [001] plane. The inset of (d) displays the FFT pattern achieved from the HRTEM micrograph. Reprinted with permission from ref. 97. Copyright (2010) American Chemical Society.<sup>97</sup>

coworkers fabricated Bi and BiAg alloy nanospheres *via* a simplistic hydrothermal route and demonstrated photocatalytic activity towards H<sub>2</sub> generation.<sup>105</sup> Zhou's group has reported a synthesis protocol for a C@Ag/TiO<sub>2</sub> composite photocatalyst by combining a hydrothermal and sol-gel method and claimed that the photocatalytic activity considerably boosted under visible light.<sup>106</sup> In the context of composite nanostructured photocatalyst, Mondal *et al.*<sup>107</sup> have been fabricated macroporous TiO<sub>2</sub>/polyacrylonitrile and TiO<sub>2</sub>/carbon hybrid films using spin coating technique. They have demonstrated that porous carbon film acts as a catalytic support material for the TiO<sub>2</sub> nanoparticles with an average size 40 nm. The resultant TiO<sub>2</sub>/carbon hybrid porous films reveal outstanding photocatalysis as displayed for the degradation of aqueous rhodamine B dye.

#### 4. Metal–semiconductor nanostructured photocatalysts

An indispensable requirement for the photocatalyst is its robustness to the reaction occurring at the solid/liquid interface, which can involve its physicochemical belongings.



However, it has demonstrated hard to find an ideal photocatalyst, which encounters all the necessities like physico-chemical stability, resistance to corrosion, harvesting visible light and appropriate band gaps that would enhance photocatalytic response as a viable alternative. Providentially, recent research in the field of nanoscience and nanotechnology has improved the alteration of obtainable photocatalysts and the finding and improvement of new materials.<sup>108,109</sup> The fast growing number of scientific publications establishes vibrant bibliographical confirmation for the implication of this hot subject matter. Several papers considered the influence of diverse nanostructures of nanomaterials on their performance as photocatalysts, as their efficiency of energy conversion is mostly determined by nanoscopic properties.

In this context, metal-oxide semiconductors are of excessive scientific significance in electronics and environmental remediation as they have competence to create charge carriers when irradiated with requisite expanse of energy. The encouraging organization of electronic arrangement, efficient light absorption capabilities, and interesting charge transport features of maximum of the metal-oxides has ended promising its use as photocatalyst. But, there are some drawbacks in semiconductor photocatalysts, such as, fast recombination of photogenerated charge carriers, wide bandgap which makes the possibility to engineer new composite photocatalysts.

The most important objective of designing composite photocatalyst is to modify the photoelectrochemical possessions of the semiconductor nanomaterials. A sequence of metal–semiconductor and semiconductor–semiconductor composite nanoparticles have been demonstrated to ease charge refinement in the metal-oxide semiconductor nanostructures.<sup>110–113</sup> The noble metal can perform as a sink for photogenerated charge carriers and thus helps interfacial charge-transfer procedures in these composite photocatalysts.<sup>114</sup> The schematic describing the electron transfer in noble metal–semiconductor photocatalytic system is shown in Fig. 9. In a latest report, research effort was ended to include noble metal ions to encompass the photoresponse of titania keen on the visible light radiation.<sup>115</sup> A straight correspondence between the photocatalysis for the production of ammonia from azide ions and the work function of the metal has been prepared for metallized TiO<sub>2</sub> systems has been described by Nosaka *et al.*<sup>116</sup> Also, doping of TiO<sub>2</sub> nanomaterials with transition metal ions has been explored by a number of researchers to increase the photocatalytic properties of metal-oxide semiconductor photocatalysts.<sup>117,118</sup> An infrequent photoelectrochemical consequence has also been described for Ni/TiO<sub>2</sub> photocatalyst in the form of films.<sup>119</sup> These fascinating features of metal–semiconductor photocatalytic systems encouraged many research groups to probe the impact of metal nanoparticle on the photocatalytic properties of metal-oxide semiconductors.<sup>120</sup>

#### 4.1 Metal–metal-oxide semiconductor heterostructure photocatalysts

Heterogeneous nanoparticles comprising two or more dissimilar functional units in epitaxial conjugation are of immense

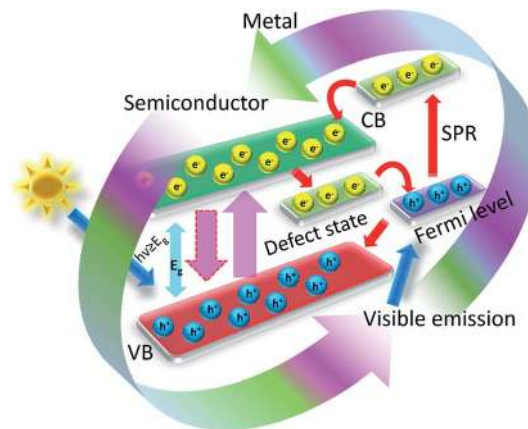


Fig. 9 Scheme diagram of the electron transfer mechanism of the metal–metal-oxide nanostructures.

research concern since they possess exclusive physicochemical, magnetic, electronic, optical, catalytic and photocatalytic belongings.<sup>121–123</sup> These properties typically consequence from their nanoscopic junction effects existing in the structure. It is well-known that for a heterojunction between a heterostructured metal and metal oxide nanoparticle, the material characteristics at both ends adjacent to the interface are altered from their bulk counterpart. This could be instigated by various influences, comprising surface reconstruction nearby the junction, lattice mismatch persuaded crystal strain, and electron interaction/transfer through the interface. However, modifying material belongings by heterogeneous conjugation to a large degree depends on the epitaxial link between the two unlike materials.<sup>124,125</sup> A typical metal–metal-oxide semiconductor heterostructure of Au–Fe<sub>3</sub>O<sub>4</sub>, Ag–Fe<sub>3</sub>O<sub>4</sub>, Pt–Fe<sub>3</sub>O<sub>4</sub> and AuAg–Fe<sub>3</sub>O<sub>4</sub> nanostructures are shown in Fig. 10.<sup>126</sup>

The metal–metal-oxide semiconductor nanostructured photocatalysts can be differentiate mostly in to two classes. Firstly, the materials is photoenergetic and, upon excitation, the excited charge carriers are shifted to further share of the material, which thereby prompts the catalytic method. The plasmonic nanomaterials such as gold, silver, platinum the grouping of elevated bandgap semiconductors like TiO<sub>2</sub> fall in this group, where upon photoexcitation the plasmon electrons are transported from gold to the TiO<sub>2</sub> semiconductor for starting the catalytic activity.<sup>127–132</sup> These are identified as plasmonic photocatalysts which have been extensively studied and described in the literature. A probable electron removal from the surface plasmonic state of gold to the semiconductor material has been illustrated in Fig. 11 while connecting the recent literature.<sup>133–135</sup> The second further option is that both the semiconductor and metal are photosensitized and captivate the solar light. The material systems where the plasmonic material gold is combined with small bandgap semiconductors (for example Au–CdS, Au–PbS, Au–CdSe, *etc.*),<sup>136–138</sup> counted in this class.<sup>136</sup> Interestingly, it is perceived that there is a chance of the electron transfer from both ends, *i.e.*, either metal to semiconductor or the reverse subjected to the bandgap arrangement along with the excitation basis.

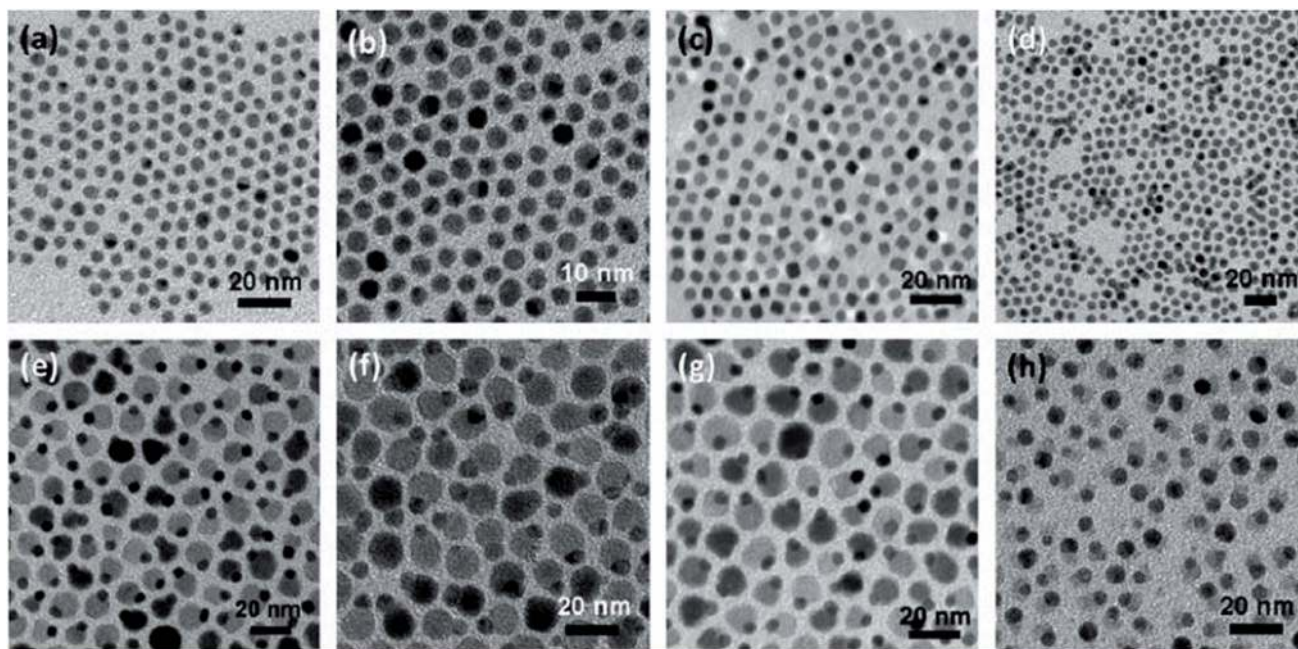


Fig. 10 TEM micrographs showing different noble metal nanoparticles: 5 nm Au (a), 5 nm Ag (b), 5 nm Pt (c), 6 nm AuAg (d); noble metal–metal oxide heterostructures of 5–12 nm Au–Fe<sub>3</sub>O<sub>4</sub> (e), 5–12 nm Ag–Fe<sub>3</sub>O<sub>4</sub> (f), 5–12 nm Pt–Fe<sub>3</sub>O<sub>4</sub> (g), and 6–10 nm AuAg–Fe<sub>3</sub>O<sub>4</sub> (h) nanoparticles. Reprinted with permission from ref. 126. Copyright (2010) American Chemical Society.<sup>126</sup>

In certain occasion, if only the metal is excited, then electron handover can trail alike to the scheme shown in Fig. 11a, providing the surface plasmonic state rests above the conduction band (CB) of semiconductor. In the same way, with selective excitation of the semiconductor can ease the electron movement from CB to the electronic band of gold and a schematic presentation of such electronic movement has been shown in Fig. 11b. But then, the circumstance turn out to be further complex once both semiconductors and metal are photoexcited simultaneously (shown in Fig. 11c). Despite the fact utmost of the information anticipated the photoexcited electron assignments from the agitated state of semiconductor to plasmonic metal, but few latest research on gold–cadmium sulfide metal–semiconductor exhibited that, the electron assignment

in together traditions are probable, yet more probable is the handover from CdS toward Au. The coupling of the exciton is also possible between semiconductor and plasmon of metal.<sup>138–140</sup> Conversely, it is demonstrated that the occasion when both the semiconductor and metal are photoexcited, it turns out to be more encouraging, as they together can captivate solar radiation and produce more photoexcited charge carriers than the rest two kinds. On the other hand, even though most of the cases it is considered for metal–semiconductor heterostructures, but besides gold, further metals are also been coupled with many semiconductors and can be employed in efficient photocatalysis. For example plasmonic platinum has been considered even further effective in few circumstances.<sup>141,142</sup> But mostly the metallic Au is used for metal–

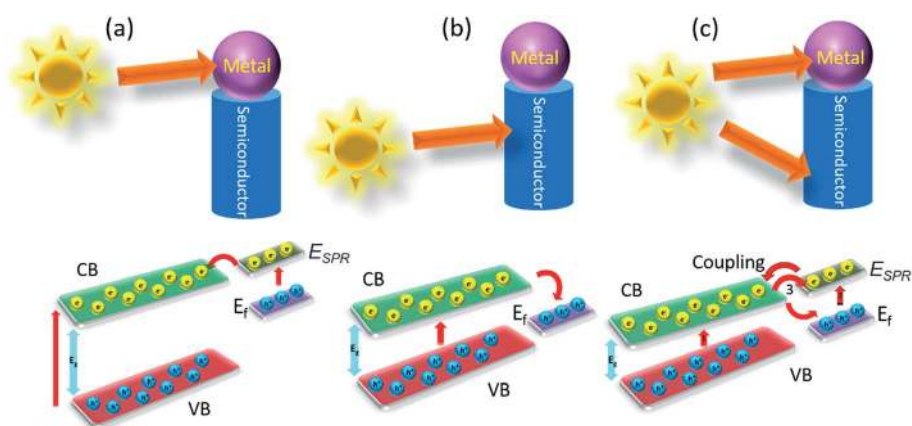


Fig. 11 Schematic representation of the electron transfer process in a metal–semiconductor heterostructure.

semiconductor photocatalyst systems as this one has surface plasmon resonance (SPR) which falls in the strong vicinity of the solar spectrum.

#### 4.2 Metal–metal oxide semiconductor core–shell nanostructured photocatalysts

In recent past, there have been made noteworthy advances in the designing of core–shell metal–semiconductor photocatalysts.<sup>143–147</sup> Until now, the struggles to employ these core–shell nanostructures as photocatalyst materials in the photo-energy renovation systems such as hydrogen generation, water splitting, photoelectrochemical cells, photocatalysis of wastewater *etc.* are restricted.

In this context, it is imperative to interpret the effect of the metal core on the photocatalytic belongings of exterior metal oxide shell (shown in HRTEM micrograph of an individual Au–Cu<sub>2</sub>O core–shell nanostructure along with its SAED patterns in Fig. 12a and b respectively). As a distinctive class of innovative core–shell photocatalysts, metal-core and semiconductor-shell nanocomposites are capable of doing noteworthy advantages as a candidate in heterogeneous photocatalysts. Firstly, encapsulating metal nanoparticles inside a metal-oxide semiconductor outer shell can significantly improve their stability in contrast to agglomeration, and help to escape unwanted photo corrosion or suspension in the course of the process of catalytic reactions in real uses.<sup>148,149</sup> Secondly, as per the core material being a noble metal, its small Fermi energy level can function as a pool of photo generated electrons and extend the lifespan of photogenerated charge carriers. Thus this phenomenon perhaps increases the total photocatalytic performance.<sup>4,120,150,151</sup> These features have been well described by Kamat's and his coworkers, which can be realistically explained by a semiconductor–metal nanocomposite system as shown in Fig. 13.<sup>127,143,152</sup> Furthermore, the core–shell nanostructural design offers a 3D close interaction between the semiconductor shell and metal core which takes most advantage of the metal-support interfacial interaction. In this manner it eases the interfacial charge transfer procedure.<sup>143</sup> Lastly, the metal–metal-oxide core–shell nanostructured configuration delivers a homogeneous photocatalytic environment for reactions to continue.<sup>153,154</sup> These exceptional

advantages essentially accompanying with the metal-core and metal-oxide semiconductor-shell nanocomposite photocatalysts propose that they can be employed as a innovative approach for light harvesting in heterogeneous photocatalysis.

#### 4.3 Synthesis of metal core–metal oxide shell photocatalysts

The designing of metal core–metal oxide shell nanostructures needs full control over some processes parameters, including nucleation and growth in the semiconductor shell nanostructure surrounding the seeds of metal-core nanoparticles. These routes are controlled by means of various experimental factors, for example, temperature, time, solution concentration, mismatch in lattice parameters, presence of surfactant, interfacial energy, and so on.

Usually, the mismatch in lattice parameters and shortage of chemical interaction between metal–metal oxides interfaces frequently consequences in a huge interfacial energy generation between the metal oxide and a metal. Unfortunately, if there exists a lattice incompatibility between two constituent materials then that would be very problematic to create their core–shell formation. Yet, suitable ligand or surfactant might be helpful to adjust the interfacial energy between these two constituent materials. The nanoparticle assembly with the noble metal core and SiO<sub>2</sub> shell is quite recognized, however, this technique has not been widely applied to other oxide shells, mostly for the reason of agglomeration in the solution.<sup>156</sup> Furthermore, syntheses of various metal oxides, excluding silicon dioxide, need the usage of metal precursor salt, which could lead to aggregation of the nanoparticle seeds. Although, if the consumption of metal salt as a precursor is inevitable, the introduction of proper surfactant, ligand *etc.* can be an alternative favorable solution to synthesize core–shell nanomorphology. Therefore, the choice of a friendly ligand or surfactant remains interesting, and thus there are very few literature exist reporting the synthesis of core–shell nanoparticles with metal oxides, even if their analogous pure oxides are available commonly. This is a probable the cause why the core–shell nanoparticles with metal and metal oxides is case

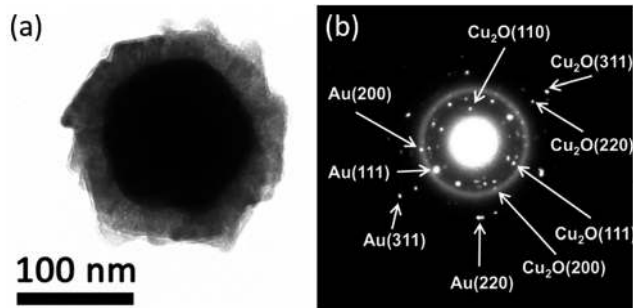


Fig. 12 (A) TEM micrograph of an individual Au–Cu<sub>2</sub>O core–shell nanoparticle with a compact metal oxide shell and (B) SAED pattern showing individual crystal planes. Reprinted with permission from ref. 155. Copyright (2012) American Chemical Society.<sup>155</sup>

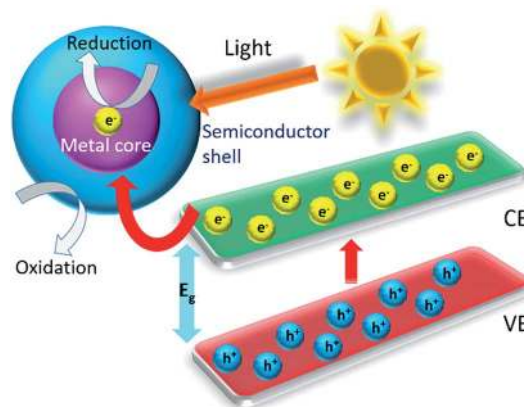


Fig. 13 Scheme of a light-induced charge separation mechanism in a typical metal–metal oxide semiconductor core–shell in a photocatalytic reaction.



specific. The fabrication of  $\text{TiO}_2$  and  $\text{SnO}_2$  core-shell nanoparticles is comparatively much easier as both of them have cubic crystal structure comparable to silver, gold, platinum *etc.* noble metals, and as a result, there exists a good match in lattice parameters for core-shell construction in this combination. However, it is very difficult to create a ZnO nanostructure surrounding a noble metal to design core-shell structure since ZnO is hexagonal and differently grows along *c*-axis. In consequence, the formation of core-shell structure by metal-ZnO combination was challenging owing to their lattice mismatch and very unfortunately neither a surfactant nor a ligand can help to overcome the encapsulation problem during synthesis.<sup>157</sup> As a solution to this problem, recently, Zhao *et al.*<sup>158</sup> have fabricated Ag-ZnO core shell nanoparticles by laser ablation technique nanoparticles and proposed that synthesized nanoparticles may have outstanding application towards micro-nano optoelectronic device fabrication. They have synthesized for the first time a silver-ZnO core-shell by a 248 nm KrF excimer pulsed laser ablation in a liquid solution. It was observed that Ag-ZnO core-shell nanostructures show strong surface plasma resonance absorption and that can be altered by tuning the thickness of the ZnO shell. Also, Sun *et al.*<sup>159</sup> have successfully produced metal-ZnO core-shell structure with distinct morphology and obtained a comprehensive growth mechanism (Fig. 14a-f show TEM images and photographs of metal-ZnO core-shell nanostructures that were produced from different noble metal used as cores). They have established a general scheme for coating ZnO on various core metals, oxides, polymer nanoparticles, graphene oxide, and carbon nanotube seeds. They have also shown that this technique works well and can be further extended to include many other semiconductors like  $\text{Fe}_3\text{O}_4$ ,  $\text{MnO}_2$ ,  $\text{Mn}_3\text{O}_4$ ,  $\text{MnO}$ ,  $\text{Co}_2\text{O}_3$ ,  $\text{TiO}_2$ ,  $\text{Eu}_2\text{O}_3$ ,  $\text{Gd}_2\text{O}_3$ ,  $\text{ZnS}$ ,  $\text{Tb}_2\text{O}_3$ ,  $\text{CdS}$  and  $\beta\text{-Ni(OH)}_2$ , as the shell materials (shown in Fig. 15a-i). It was observed that polyvinylpyrrolidone (PVP) and 4-mercaptobenzoic acid played an imperative role in the Au-ZnO synthesis. They have also revealed that 4-mercaptobenzoic acid can substantially decrease

the Au-ZnO interfacial energy. Although, ZnO does not cooperate well with the hydrophobic ligands attached to the surface on the gold in the absence of PVP polymer demonstrating that the amphiphilic nature of PVP is crucial in the ZnO encapsulation on the metallic core with hydrophobic ligands.

Interesting core-shell structure with Ag metal and  $\text{WO}_3$  metal-oxide nanostructures were synthesized by a hydrothermal method for effective localized surface plasmon propagation by Liu's group.<sup>160,161</sup> Size and shaped dependent silver nanoparticles with diameter in the 25–60 nm range have been prepared successfully. First, Ag nanoparticles were dispersed into a  $\text{Na}_2\text{WO}_4$  solution and then after adding nitric acid  $\text{Ag}_x\text{-H}_2\text{WO}_4$  was precipitated in the solution. A high temperature calcination process then employed and removes water from the precipitate to form  $\text{Ag}_x\text{-WO}_3$  core-shell nanostructure where the shell thickness was  $\sim 60$  nm in a  $\sim 200$  nm overall particle size.

In a recent study, Lu *et al.*<sup>162</sup> have been demonstrated a robust room-temperature wet chemistry methodology to fabricate hollow Au-Cu<sub>2</sub>O core-shell nanostructures where they can control the regulation and augmentation of plasmonic properties of cores (hollow gold nanoparticles) *via* tuning the nanosized dielectric Cu<sub>2</sub>O shells. It was evident that, the epitaxial growth of Cu<sub>2</sub>O on the Au nanoparticle surface happens because of the good affinity of Cu<sub>2</sub>O to gold surface and their comparable crystal symmetry. Fig. 16a-d show typical TEM micrographs of hollow core-shell nanomorphology of Au-Cu<sub>2</sub>O particles with increasing shell thicknesses and Fig. 16e describes the size distribution.

There are very few information on the realization of core-shell nanoparticles using  $\text{Co}_3\text{O}_4$  as shell material. In this context, Hu *et al.*<sup>164</sup> the synthesized Au- $\text{Co}_3\text{O}_4$  core-shell nanocubes *via* solvothermal process, however, there is no strong indication of core-shell nature in these nanostructures. Xu's group has been reported a simplistic synthesis method to attain a Pd-CeO<sub>2</sub> hollow core-shell nanostructures made up of Pd nanoparticle cores coated with CeO<sub>2</sub> hollow shells.<sup>165</sup> Kim

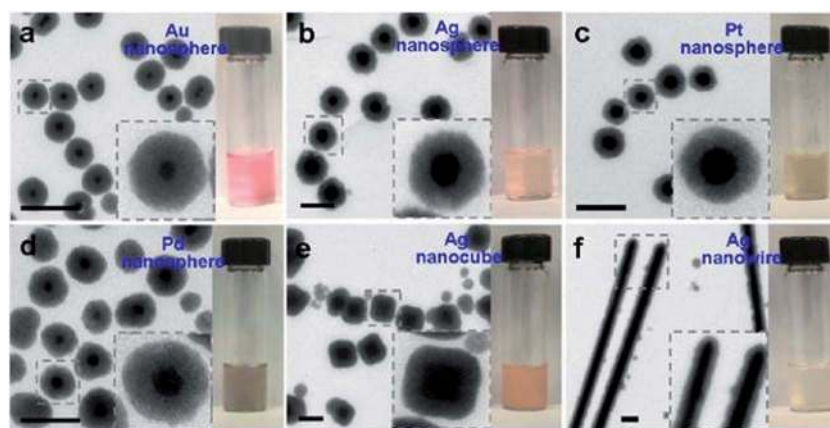


Fig. 14 TEM micrographs and digital photographs of metal-ZnO nanoparticles that were made from different noble metal cores: citrate-stabilized nanoparticles, including nanospheres of Au (a); Ag (b); and Pt (c). PVP-stabilized nanoparticles, including nanospheres of Pd (d); Ag (e); and Ag nanowires (f) (diameter = 120 nm, length = 3–5  $\mu\text{m}$ ). Insets display magnified views of typical nanomorphologies. All scale bar: 200 nm. Reprinted with permission from ref. 159]. Copyright (2013) American Chemical Society.<sup>159</sup>

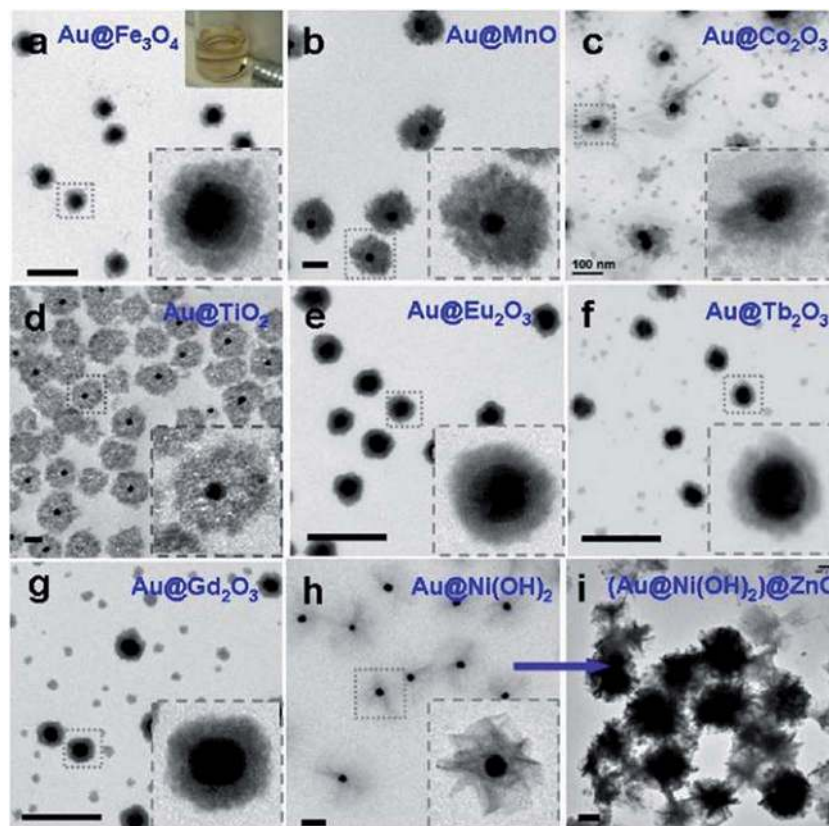


Fig. 15 TEM micrographs of Au-oxide nanoparticles (diameter of Au = 40 nm) with different types of metal oxide shell materials: Au-Fe<sub>3</sub>O<sub>4</sub> (a), Au-MnO (b), Au-Co<sub>2</sub>O<sub>3</sub> (c), Au-TiO<sub>2</sub> (d), Au-Eu<sub>2</sub>O<sub>3</sub> (e), Au-Tb<sub>2</sub>O<sub>3</sub> (f), Au-Gd<sub>2</sub>O<sub>3</sub> (g), Au-Ni(OH)<sub>2</sub> (h), and (Au-Ni(OH)<sub>2</sub>)-ZnO (i). Insets describe magnified views of typical nanoparticles. All scale bar: 200 nm. Reprinted with permission from ref. 159. Copyright (2013) American Chemical Society.<sup>159</sup>

and other co-workers have fabricated Au-Co<sub>3</sub>O<sub>4</sub> core-shell nanowires.<sup>166</sup> Recently, Yan's group reported the preparation of monodisperse Au-Co<sub>3</sub>O<sub>4</sub> core-shell nanostructures, where final structures produced from the oxidation of monodisperse Au-Co core-shell nanocrystals and explored their catalytic activity for oxygen evolution application.<sup>167</sup> Zhang *et al.*<sup>163</sup> reported metal-Al<sub>2</sub>O<sub>3</sub> yolk-shell nanostructured particles with improved thermal stability where metal nanoparticle is confined inside alumina nanoshell. They have employed a one-pot hydrothermal reaction to prepare a core-shell structure of metallic core surrounded by carbon shell and that was then used as seeds for the subsequent surface growth of Al<sub>2</sub>O<sub>3</sub>. Fig. 17 shows the TEM images of Au/carbon/Al<sub>2</sub>O<sub>3</sub> nanostructures where in Fig. 17a and b, confirms that the diameter of carbon shell rise to 200 and 300 nm when extending the hydrothermal reaction. It is worth mentioning that, varying the parameters like aluminum salt concentration and the reaction time, the Al<sub>2</sub>O<sub>3</sub> nanoshell thickness could be tuned (shown in Fig. 17c and d). In the same context, Huo's group has demonstrated the simplistic approach for encapsulation of prefabricated nanoparticles into carboxylic acid mediated metal-organic framework using metal-metal oxide core-shell nanoparticles as the self-template.<sup>168</sup> The yolk-shell surface morphology of the reaction product was characterized by TEM and STEM which are shown in Fig. 18a-c. They have shown that the core-shell nanoparticle-metal oxide self-

template could be converted yolk-shell metal-metal oxide heterostructures with preferred encapsulating functional materials and metal oxide shells. The thickness of the shell, encapsulated materials, and encapsulated nanoparticles in each petalous heterostructure, all of them could be tuned straightforwardly by varying the concentration of copper salt, the nature and concentration of used nanoparticles during synthesis of the Au-Cu<sub>2</sub>O core-shell NPs (shown in Fig. 18d-g).

Transition metal core-semiconductor shell nanoparticles are also important candidate in this topic. For example, Chang *et al.*<sup>169</sup> have been reported the synthesis of a Cu-ZnO metal-semiconductor core-shell nanoparticles containing Cu nanoparticles as core with an average diameter of  $44.4 \pm 4.3$  nm and coated with  $4.8 \pm 0.5$  nm-thick ZnO nanoshell. Chen's group has been reported a new and facile route to develop the fabrication of Zn-ZnO core-shell metal-semiconductor nanostructures on a large scale.<sup>170</sup> Recently, Kudrynskiy's group has been reported a route for the controlled synthesis of greatly ordered core-shell nickel-carbon nanoparticle arrays.<sup>171</sup> Ivanov *et al.*<sup>172</sup> have been synthesized core-shell nanowires with a Fe core coated by an iron oxide shell *via* a facile low-cost fabrication method. They have demonstrated that the magnetic properties of the core-shell nanostructures can be altered by changing magnetite shell thicknesses to obtain significantly multi-functionality and new properties.

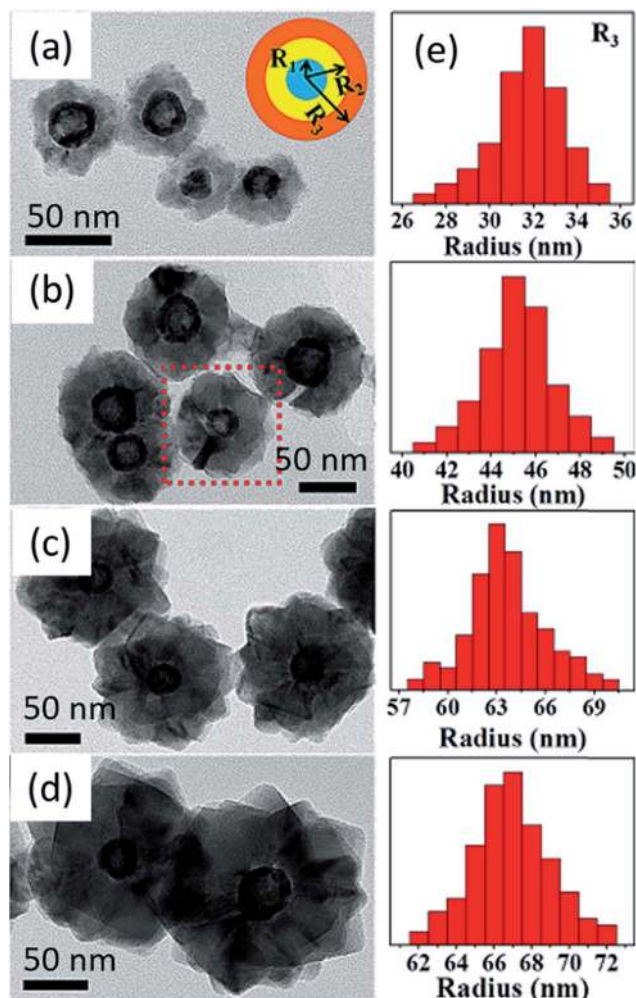


Fig. 16 TEM micrographs of hollow Au-Cu<sub>2</sub>O core-shell nanostructures with adjustable shell thicknesses. The inset in (a) shows the scheme of cross-sectional drawing of core-shell nanoparticle where  $R_1$ ,  $R_2$ , and  $R_3$  are the average inner radius and outer radius of core and overall radius of core-shell structure, respectively. The thickness was tuned by mixing different volumes of colloidal hollow gold nanospheres in the reaction (a) 1.8, (b) 1.0, (c) 0.4, and (d) 0.3 mL. (e) Histogram describing size distribution ( $R_3$ ) of the aforesaid core-shell nanoparticles with different of Cu<sub>2</sub>O shell thicknesses. Reprinted with permission from ref. 162. Copyright (2016) American Chemical Society.<sup>162</sup>

Interestingly, metal-metal oxide core-shell nanoparticles systems sometime confirmed photocatalytically more effective owing to the fact of doping of the core-shell systems, such as doping in TiO<sub>2</sub> shell by Zn, F, S or C *etc.*, nanoparticles. For example, it has been observed that a simple doping in TiO<sub>2</sub> with Zn<sup>173</sup> or F<sup>174</sup> gives higher photocatalytic activity in systems with Pd-TiO<sub>2</sub> nanoparticles. Recently, Er<sup>3+</sup>, Yb<sup>3+</sup> doped core-shell nanostructured BiVO<sub>4</sub> and their near-infrared photocatalytic prospects have been reported by Shan's group.<sup>175</sup> Along the same concept, Pihosh *et al.*<sup>176</sup> have doped WO<sub>3</sub> into BiVO<sub>4</sub> system and have reported enhanced photocatalytic activity.

Moreover, core-shell nanostructures with other metals and metal oxides have also been described, but, many of them do

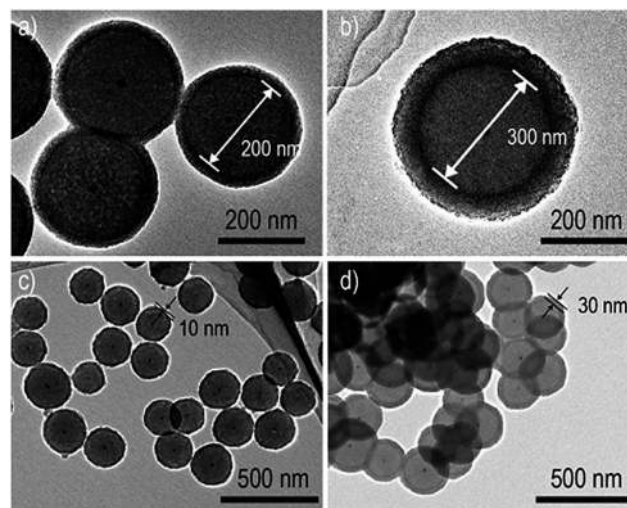


Fig. 17 TEM micrograph shows Au/carbon/Al<sub>2</sub>O<sub>3</sub> (a and b) different carbon sphere diameter and several alumina shell thickness (c and d). Reprinted with permission from ref. 163. Copyright (2015) American Chemical Society.<sup>163</sup>

not have well-defined core-shell structures and failed to achieve uniform particle sizes.

## 5. Environmental applications of the core-shell photocatalysts

Semiconductor nanostructured mediated photocatalytic degradation of organic pollutants have revealed excessive prospective as a wastewater management technology. This method can be employed to eliminate persistent organic compounds and various micro-organisms that are involved water pollution, and it has been established extensively.<sup>177-181</sup> Semiconductors are encouraging catalysts for this technology since they are environmentally friendly and can perform great photocatalysis of waste at small cost in ambient conditions.<sup>182</sup> Although, there are few disadvantages to the usage of semiconductors photocatalysts that bounds their wide spread acceptability for extensive photocatalytic developments. For example, they have a high rate of recombination of photogenerated electron-hole charge pairs and on top of that they show poor recovery after photocatalysis.

To address both issues with photocatalytic semiconductors, core-shell nanomorphology of the metal-semiconductor photocatalysts is a feasible option. Nanooscopic metal core and semiconductor shell are able to decrease the rate of recombination rate of electron-hole pairs. Without a doubt, there is a fast transfer of electrons from the conduction band of the semiconductor to that of the metal and slow recombination of charge carriers as there is defect state created below the conduction and above the fermi energy levels owing to the metal-semiconductor bandgap modification.<sup>183</sup> A summary of the nanostructured metal-semiconductor core-shell photocatalysts and their use towards various environmental applications



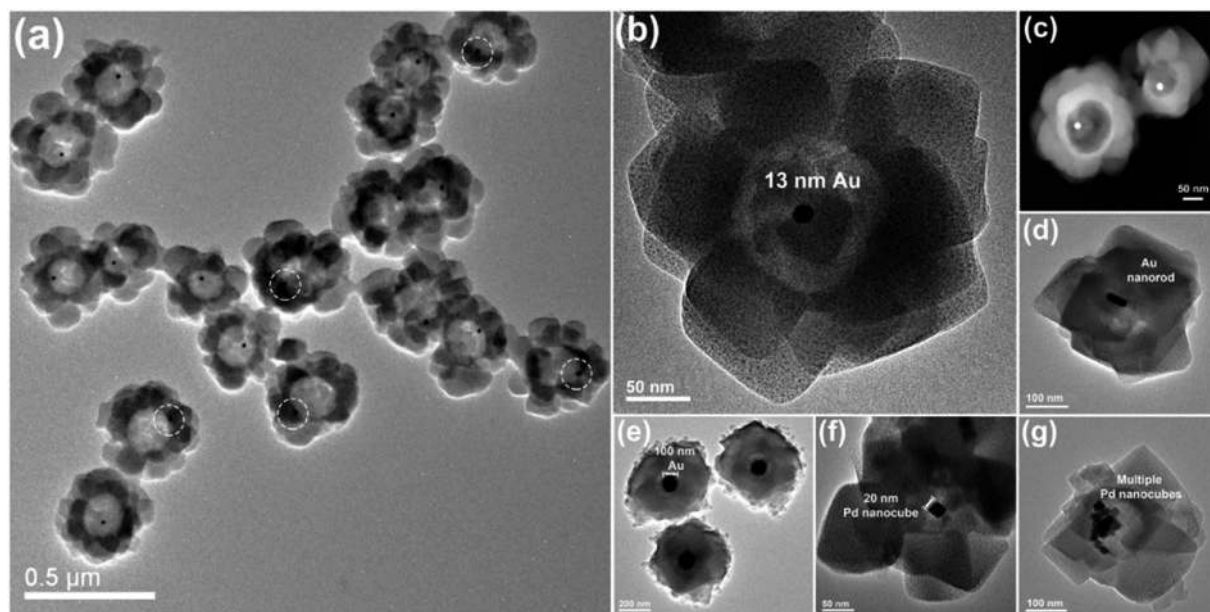


Fig. 18 Surface morphology of yolk-shell nanoparticles petal-like heterostructures: (a and b) TEM micrographs and (c) STEM dark-field micrograph of the Au petal-like heterostructures displaying good product concentration and consistency. TEM micrograph shows the heterostructures encapsulating (d) a single gold nanorod, (e) a single Au nanoparticle, (f) a single Pd nanocube, and (g) several Pd nanocubes, respectively. Reprinted with permission from ref. 168. Copyright (2014) American Chemical Society.<sup>168</sup>

based on visible and UV light mediated photocatalysis is explained in Table 1.

### 5.1 Wastewater treatment and air purification with core-shell nanostructured photocatalysis

Conventional metal-oxide photocatalysts have been widely considered for wastewater management and polluted air purification application and it is well acknowledged to be an active scheme to treat a number of hazardous materials in polluted water and also the air. More emphasis is specified at the present time to core-shell metal-metal oxide-based photocatalysis and their use concerning the remediation of controlled and emergent pollutants of concern. Owing to the dissimilarity between the atomic electronegativity, the metallic nanostructured layer in the double-layered configuration helps the photogenerated

charge transfer of in the conduction bands of the semiconductor photocatalyst, permitting electrons to counter with reactants efficiently.

Recently, since silicon nanowires have been confirmed to be very useful in the degradation of organic dyes by photocatalysis, Jiang and coworkers<sup>195</sup> have studied a system where they have fabricated Si/SiO<sub>x</sub> metal-semiconductor photocatalyst and applied for wastewater treatment. They have purposely designed SiO<sub>2</sub> shells on the surface of Si nanowires to yield Si/SiO<sub>x</sub> core-shell nanowire photocatalyst. The photo degradation study was done on indigo carmine dye which is a model organic contaminant in wastewater and interestingly the core-shell photocatalyst was observed to be very active towards the photocatalysis. In another study, Zhang's group<sup>196</sup> has prepared Zr-doped silica shell/titania core nanoparticle photocatalyst through extended channels into polyvinylidene fluoride (PVDF)

Table 1 Table for core-shell nanostructured photocatalyst and environment applications

Core-shell photocatalyst	Average particle size	Light irradiated	Application	Literature
Au-TiO <sub>2</sub>	100 nm (core-Au (25 nm))	UV	Photocatalysis of ethanol	Goebel <i>et al.</i> <sup>184</sup>
Au-Cu/TiO <sub>2</sub>	5.4 nm (core-Au (4.5 nm))	Visible	Selective oxidation of amines	Sato <i>et al.</i> <sup>185</sup>
Pt-SnO <sub>2</sub>	12 nm (core-Pt (7 nm))	Visible	Degradation of formaldehyde	Chang <i>et al.</i> <sup>186</sup>
Pt-TiO <sub>2</sub>	90 nm (core-Pt (30 nm))	Visible	Photocatalysis	Fang <i>et al.</i> <sup>187</sup>
Ag-TiO <sub>2</sub>	15 nm (core-Ag (5 nm))	UV	Destruction of methylene blue	Chen <i>et al.</i> <sup>188</sup>
Ag-TiO <sub>2</sub>	37.33 nm (core-Ag (33.63 nm))	UV and visible	Degradation of azo dyes in wastewater	Khanna <i>et al.</i> <sup>189</sup>
Ag-SiO <sub>2</sub> /TiO <sub>2</sub>	80 nm (core-Ag (60 nm))	Visible	Photocatalysis	Zhang <i>et al.</i> <sup>190</sup>
Ag-Cu <sub>2</sub> O	100 nm (core-Ag (31 nm))	Visible	Photocatalysis	Li <i>et al.</i> <sup>191</sup>
AgAu alloy-TiO <sub>2</sub>	30 nm (core-Ag (10 nm))	UV	Photocatalysis of methylene blue	Xiao-yu <i>et al.</i> <sup>192</sup>
Ag-ZnO	100 nm (core-Ag nanowire (83 nm))	Visible	Degradation of rhodamine B	Xiong <i>et al.</i> <sup>193</sup>
Ag-ZnO	15 nm (core-Ag (10 nm))	Visible	Disinfection of bacterium <i>Vibrio cholerae</i> 569B	Das <i>et al.</i> <sup>194</sup>

membrane to improve mass transfer depending on immobilizing and recycling  $\text{TiO}_2$  in the course of photocatalysis of methyl orange dye solution and oil in wastewater comprising oil, respectively. They have demonstrated that the photocatalytic membranes own encouraging requests in the immobilization and recycling of photocatalysts with improved photocatalytic activity for wastewater treatment. It can be interesting to note that Xu's group<sup>4,197</sup> is the foremost one who has fabricated noble metal core and semiconductor shell nanostructured photocatalysts and put on them as a photocatalysts for selective oxidation of alcohols to aldehydes. They have synthesized an aqueous phase Pt– $\text{CeO}_2$  photocatalyst with controllable core-shell and yolk-shell nanomorphologies by template-free hydrothermal method which can assist as a competent visible light driven photocatalyst towards wastewater treatment.

Additionally, to the examples with core-shell nanocomposites as photocatalysts for wastewater treatment and selective oxidation reaction stated earlier, Liu and other coworkers<sup>199</sup> have designed a Ni–NiO core-shell nanostructure modified with nitrogen doped  $\text{InTaO}_4$  photocatalyst and explored its photocatalytic action for the choosy reduction of  $\text{CO}_2$  to methanol which is as example of treatment of pollutant air *via* photocatalyst. They have fabricated  $\text{InTaO}_4$  followed by its sintering in an ammonia atmosphere at reasonably high temperature for confirming nitrogen doping. The nitrogen doped catalyst Ni–NiO/ $\text{InTaO}_4$ -N shows the maximum photocatalytic activity as compared to without doped one under the exposure of visible light rendering to the rate of methanol formation. The improved photocatalysis can be attributed due to three causes. First, doping of nitrogen can thinner the band gap of core-shell  $\text{InTaO}_4$  photocatalyst, thereby growing both the wavelength and the quantity of light energy absorbed. Second, the  $\text{O}_2$  atoms could be substituted by the nitrogen atoms due to doping, causing in the creation of some oxygen vacancies, which are important to augment the photocatalytic activity. Lastly, chemical alteration of  $\text{InTaO}_4$ -N by putting the Ni–NiO core-shell constituent will give rise to an additional photoactivity enrichment. Rabbani's group<sup>200</sup> has developed a  $\text{Zn-Fe}_2\text{O}_4$ @ $\text{ZnO}$  core-shell structured hollow nanospheres and demonstrated their visible light photocatalytic activity by degradation of methylene blue dye. The study reveals that due to magnetic properties  $\text{Fe}_2\text{O}_4$  the catalysts are very efficient and easily recyclable. Recently, Singh *et al.*<sup>198</sup> studied the mesoporous, hollow  $\text{TiO}_2$  nanofibers fabricated by coaxial electrospinning method for the photocatalytic decomposition of *para*-nitrophenol (4-NP) dye which is a familiar model water pollutant dye. They have sensitized hollow titania nanofibers by cadmium sulphide (CdS) quantum dots (QDs) over successive ion layer adsorption and reaction (SILAR) technique for different deposition cycles (the fabricated nanofibers are shown in Fig. 19a–h). The CdS QDs loaded hollow titanium dioxide nanofibers yield catalytic spots at the QDs and  $\text{TiO}_2$  interface which benefits in improved exciton separation. It was reported that  $\text{TiO}_2$ /CdS photocatalyst for 3 SILAR deposition cycles is three times more photocatalytically efficient than hollow  $\text{TiO}_2$  nanofibers and eight times effective than the pristine solid nanofibers.

## 5.2 Water disinfection with core-shell photocatalysts

Over the past few years visible light induced photocatalytic decontamination of wastewater has gained substantial research consideration.<sup>201–203</sup> Various semiconductor photocatalysts have also been explored for a variety of decontamination uses, together with water cleansing. Thirty years later Matsunaga *et al.*<sup>204</sup> reported the first work regarding disinfection of wastewater using titania/Pt nanoparticles photocatalysis for a variety of organisms (*Lactobacillus acidophilus* (bacteria), *Saccharomyces cerevisiae* (yeast), and *Escherichia coli* (bacteria), Yu *et al.*<sup>205</sup> published paper on decontamination of the wastewater containing Gram positive bacterium *Micrococcus lylae*. In this work sulfur doped  $\text{TiO}_2$  was irradiated to 100 W tungsten halogen lamp and wavelengths less than 420 nm eliminated using a suitable glass filter.

The study (Saito *et al.*<sup>206</sup>) revealed that that photo activated photocatalysts triggered split of the cell membrane of the microorganisms, as shown in representative Fig. 20 and validated by intracellular leakage of  $\text{K}^+$  ions that correspondence to cell damage. Added confirmation of this microorganism cell death mechanism was establish by Sunada and co-workers,<sup>207</sup> who showed that semiconductor mediated photocatalysis damage of endotoxin, an vital constituent of the external membrane of bacteria. Maness *et al.*<sup>208</sup> further revealed that the existence of lipid peroxidation and the concurrent damage of both membrane-dependent respirational movement and cell feasibility rigorously rest on both the exposed light energy and photocatalyst. It has also been studied (by Huang *et al.*<sup>209</sup>) the positions of cellular destruction and their impact to cell death in *Escherichia coli* with photoexcited titanium dioxide catalyst particle. Her it was suggested that the surface of photocatalyst first creates contact with whole cells and immediately after the oxidative destruction happens to the cell wall. Once the photocatalysis gradually amplified cell permeability and then subsequently, open up intracellular constituents, and permits the catalyst nanoparticles an easy entrance and photo-oxidation of intracellular components, thus speed up the cell death<sup>209,210</sup> (as shown in Fig. 20).

It is well acknowledged that UV radiation creates damage in DNA, however, once cell wall mutilation is instigated by the photo-oxidation of photocatalysts, they can also yield further damage to the intracellular modules.<sup>211,212</sup> In light of this context, Hidaka and co-workers<sup>213</sup> implemented *in vitro* tests and monitored the outcome of DNA, RNA, and their purine and pyrimidine bases as soon as exposed by UVA light in presence of photocatalyst nanoparticles. Interestingly, till to date, the photochemical biocidal mechanism is mostly unclear.<sup>214</sup> It is still unclear what exact reactive oxygen species are openly intricate the photocatalytic cell killing procedure, particularly the characteristics of the key reactive oxygen species (ROS), which not only contain the  $\text{OH}^\cdot$ , but also  $\text{H}_2\text{O}_2$ , and  $\text{O}_2^-$ . Interestingly, Ag– $\text{ZrO}_2$  core-shell photocatalysts have also prepared very recently and verified for the antibacterial property towards various organisms (*Escherichia coli* and *Staphylococcus aureus*) and the antifungal effect against *Candida glabrata*, *Candida albicans*, *Aspergillus flavus*, and *Aspergillus*

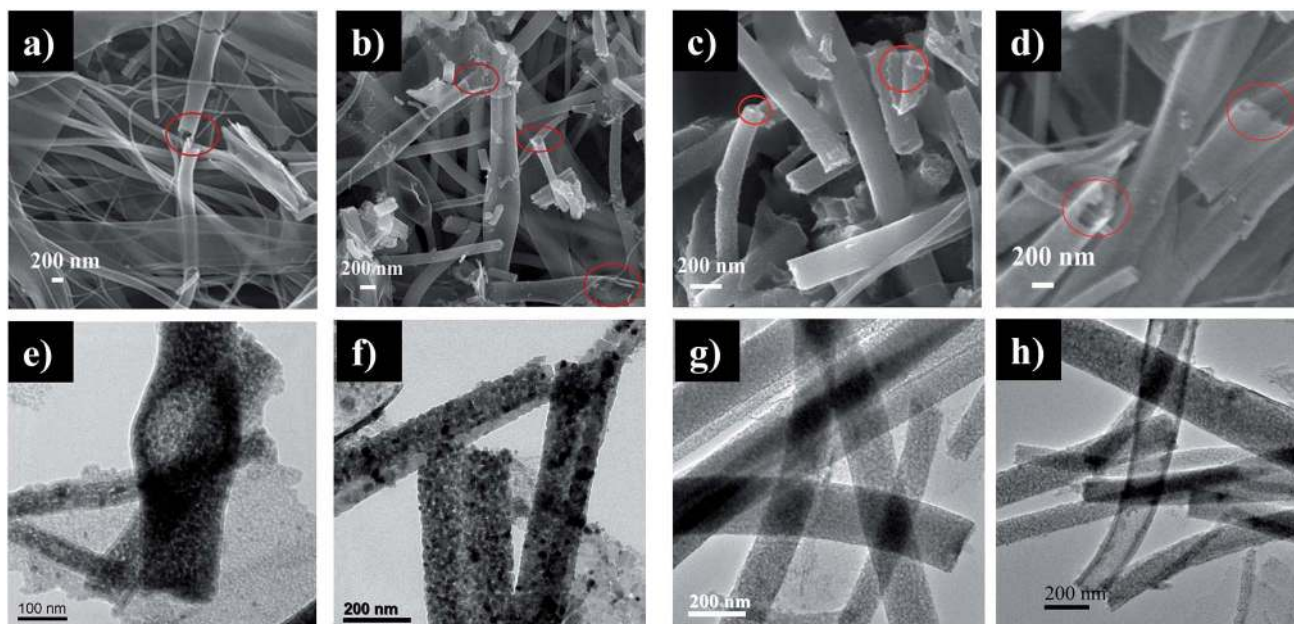


Fig. 19 FESEM images (a, b, c and d) showing CdS loaded TiO<sub>2</sub> hollow nanofibers after 1, 2, 3 and 5 SILAR cycles, respectively (marked regions show the hollow morphology of nanofibers); TEM images (e, f, g and h) reveal CdS loaded TiO<sub>2</sub> hollow nanofibers after same SILAR cycles, respectively. Reproduced with permission from ref. 198. Copyright 2016, Royal Society of Chemistry.<sup>198</sup>

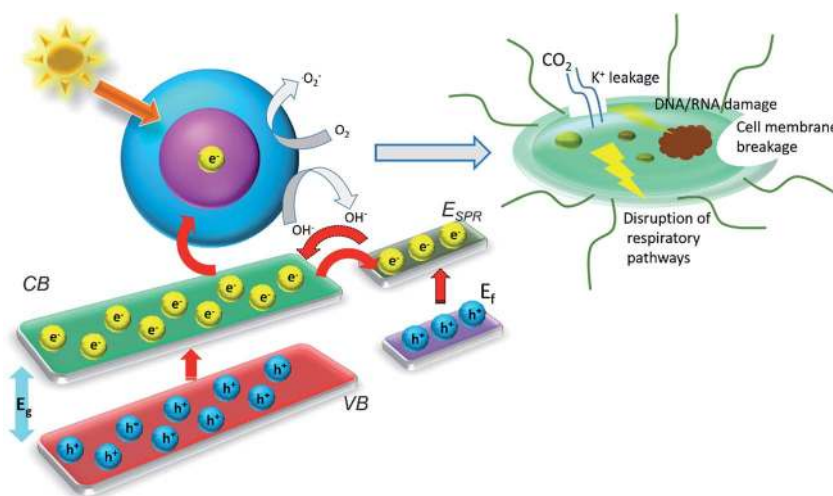


Fig. 20 Proposed bacterial disinfection mechanism during solar excitation.

*niger* fungi by the agar diffusion technique.<sup>215</sup> DNA intercalation studies were also performed in CT-DNA in presence of the photocatalyst. The study reveals that semiconductor ZrO<sub>2</sub> supported on the Ag nanosurfaces not only prohibited agglomeration of nanoparticles, but also showed superior DNA intercalation and antimicrobial activity than the pristine silver nanoparticles which could have encouraging application as antimicrobial materials for microbicides disinfection in wastewater. Meena *et al.*<sup>216</sup> demonstrated core-shell Au-SiO<sub>2</sub> metal-semiconductor nanoparticles by Stober's synthesis and their result reveals that improved core-shell nanoparticles displays that the formation of singlet oxygen. The cell viability of the core-shell nanoparticles in contrast to

HeLa cell lines were deliberated by MTT assay technique. The consequences of their study point out that, the Au-SiO<sub>2</sub> core-shell nanoparticles are enormously stable with a high photodynamic efficacy in visible light.

Recently, Karunakaran's group has been synthesized core/shell Fe<sub>3</sub>O<sub>4</sub>/Ag-ZnO nanostructured composite photocatalyst nanoflakes *via* hydrothermal method followed photodeposition technique.<sup>217</sup> The magnetically recoverable core-shell catalyst was used for photocatalytic and antibacterial activities for dye-degradation and *E. coli* bacteria disinfection. In another study, Dhanalekshmi *et al.*<sup>218</sup> have prepared core-shell type Ag-TiO<sub>2</sub> nanoparticles of size <50 nm by reduction of silver nitrate salt and hydrolysis of Ti(IV) isopropoxide. The antibacterial



properties of Ag–TiO<sub>2</sub> core–shell nanostructured photocatalysts were demonstrated towards *Escherichia coli* and *Staphylococcus aureus* bacteria disinfection in wastewater. They have claimed that Ag helps in antibacterial activity while TiO<sub>2</sub> serves the photocatalytic activity in contaminated water. They have also compared the antibacterial activity of Ag–TiO<sub>2</sub> core–shell with Ag–SiO<sub>2</sub> core–shell photocatalysts and demonstrated that the antibacterial action of Ag–SiO<sub>2</sub> is greater than the Ag–TiO<sub>2</sub> owing to their superior surface area.<sup>219</sup> Positively charged core of Ag nanostructures responded straightforwardly with Gram negative bacteria reasonably than gram positive bacteria and further successfully killed *Escherichia coli* and *Staphylococcus aureus*. Antimicrobial activity of nanocomposite TiO<sub>2</sub>–NiFe<sub>2</sub>O<sub>4</sub> with a photocatalytic titania shell covering a magnetic nickel ferrite core is reported by Misra's group.<sup>220</sup> TiO<sub>2</sub>-coated NiFe<sub>2</sub>O<sub>4</sub> nanoparticle photocatalysts display remarkable anti-microbial activity when brought near UV light irradiation. It was also found that the bacterial disinfection reaction of anatase TiO<sub>2</sub> coated nickel ferrite is greater than brookite phase of titania, even when the photocatalytic reaction rates are identical.

### 5.3 Other applications of metal–semiconductor core–shell nanostructures

Core–shell nanoparticles can be used in other fields like reusable supply, prosthetic kit, dental devices, implant material, wound dressings, external corporeal devices, encapsulating medium, devices nanoparticle-polymeric drug delivery, tissue engineering, and other biomedical applications. Additionally, the core–shell nanostructured materials have revealed an extensive range of new requests in the fields of biochemistry, biomaterial sciences due to their better physicochemical behaviors. Core–shell nanomaterials also play vital parts in emerging medical and biological uses. Suitably designed core–shell structures can deliver biocompatibility, environmental safety, and better surface functionalization that is compulsory for particular applications *in vivo* and *in vitro* systems. For example, cancer and tumor treatment is a well-known field where core–shell nanomaterials have proved their ability. In light of this, we know that in case of treatment of hyperthermia with size-dependent iron oxide magnetic nanoparticles (SIONPs) comprises a local rise in temperature up to 45 °C when the magnetic nanoparticles are subjected to a strong external magnetic field. Such an increase in temperature could be deadly for temperature-sensitive biological cells, for instance cancerous cells. Modern studies by calorimetry confirms that the rate of heating depends on the size of the SIONPs. Thus the nanoparticles should be fabricated which have optimum particle size with narrow-size distribution for improved hyperthermia treatment.<sup>221</sup>

There are also some reports about the toxic and cytotoxic potential of core–shell hybrid nanostructured systems. Guo *et al.*<sup>222</sup> have prepared sulfhydryl-modified Fe<sub>3</sub>O<sub>4</sub>–SiO<sub>2</sub> core–shell magnetic nanocomposites to evaluate their toxicity *in vitro*, and demonstrate their potential application in the biomedical fields. Atif *et al.*<sup>223</sup> demonstrated *in vitro* cytotoxicity of mesoporous SiO<sub>2</sub>–Eu(OH)<sub>3</sub> core–shell nanospheres in human

breast cancer cells (MCF-7). Also, Fan's group has reported the synthesis of multifunctional Fe<sub>3</sub>O<sub>4</sub>@SiO<sub>2</sub>–GdVO<sub>4</sub>:Dy<sup>3+</sup> core–shell nanoarchitecture and characterize for their cytotoxicity property as an efficient drug carrier.<sup>224</sup>

Recently, Sharma *et al.*<sup>225</sup> have fabricated monodisperse iron–iron oxide core–shell magnetic nanoparticles and have been investigated their uptake by the cancer cells (LX-1 small cell lung cancer) and the stability in aqueous solution. Their findings have shown that core–shell metal–metal oxide semiconductor iron nanoparticles are very resistive to oxidation even after one week in presence of water, corroborating these nanoparticle assembly is a potential candidate for cancer treatment by hyperthermia.

## 6. Photocatalyst recycling

A mechanical pump is placed between the reactor and the reservoir for pumping the reactant *via* a flow-through cuvette located inside a photometer for nonstop on-line data collection of the catalyzed product (shown in Fig. 21). The reactor consists of quartz glass tubes layered on its exterior surface with catalyst particles. The photocatalytic reactor is designed in such a way so that it could get sufficient dose of light energy. Few glass valves are used between the water and reactant reservoir for primary null setting of the analytical instrumentation part prior to the start of an experiment, feed of the reactant into the system, removal of air bubbles occurred throughout the reaction cycle, and finishing purging of the complete system.

The semiconductor photocatalyst can be introduced into reactor either in a colloid form or in the form of an immobilized film. In a typical photoreactors run with catalysts in the form of slurry, the reaction rate is mostly controlled by the intensity of exposed light on the catalyst surface, the quantum efficiency of the photocatalyst particles and the adsorption nature of the reactant and non-reactant species present in the reaction environment.<sup>226</sup> On the other hand, the usage of suspended catalyst needs the separation and recycling of the nanocatalyst from the treated solution and can be a difficult, time intense costly method.<sup>227</sup> Furthermore, the penetration depth of UV radiation is restricted for the reason that of strong absorptions by both the catalyst and thawed organic compounds. These difficulties could be escaped in photocatalytic reactors where catalyst nanoparticles are immobilized on a support material, however, immobilization of photocatalyst particles on a supporter creates

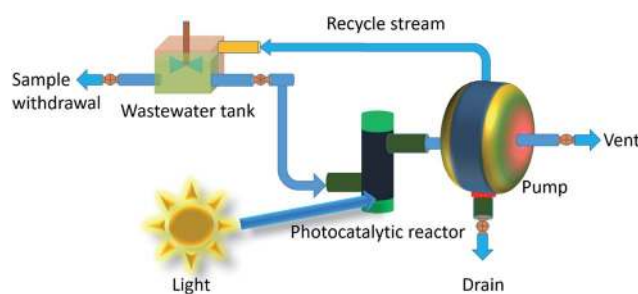


Fig. 21 Schematic diagram of a tubular photocatalytic reactor.

a distinctive difficulty. Since the reaction happens at the solid-liquid interface and mass transfer from the catalyst surface to the reactant may now be hindered and could make the process slower. However, the design of reactor, use of high surface area porous supporter and suitable nanostructured catalyst along with their recycling process is essential for an efficient wastewater management.

The reusability and recovery of the catalyst is a vital issue in photocatalysis ever since it can subsidize considerably the operating cost of the treatment process, therefore making photocatalysts recyclable is a striking way for wastewater treatment. In this context, several repeated decolonization of waste are accomplished, for each time the same catalyst is used and with a fresh waste solution.

Membrane filtration is a smart tactic for solvable catalyst recycling. Applications of nanofiltration and ultra-filtration have established their excessive prospective as a technique for process intensification in organic, enzymatic, and homogeneous catalysis, both in laboratory exercise and on an industrial scale.<sup>228</sup> Continuous flow nanofiltration is an interesting option for photocatalyst recycling. In addition to the catalyst recycling, continuous process operation with a combined membrane filtration can have many advantages. As of an economic perspective, an integrated process such as a continuous flow membrane reactor can be helpful towards lowering entire investment costs and total energy consumption. With regard to the chemistry point of view, the nonstop withdrawal of product can reduce conceivable product inhibition and undesirable repeated reactions. This may end result in greater reaction rates and a cleaner stream of product by simplifying product isolation. Interestingly, organic, aqueous or organic/aqueous photocatalysis permits stress-free separation of a homogeneous catalyst from the product phase, but is frequently incompetent when hydrophobic substrates are employed. A recycling system based on switchable water biphasic in presence of CO<sub>2</sub> and is single phase in the absence of it. Photocatalysis can be executed first in the single phase solvent, and then swapped to a biphasic system, extracting catalyst from product phase. Upon removal of CO<sub>2</sub> permits an easy recycling of the catalyst particles. The catalyst could be recycled numerous times with minimal loss of catalytic activity.<sup>229</sup> In this context, Myakonkaya *et al.*<sup>230</sup> have demonstrated a method for separation, recycling and reuse of a highly-active Au and Au-Pd nanoparticle catalysts using the practically simple method of solvent quality tuning. In another study magnetic metal-oxide core (Fe<sub>3</sub>O<sub>4</sub>) and polymeric shell (PVA polymer) nanoparticles were efficaciously prepared and conjugated with heparin (HEP) in order to offer anticoagulation. The magnetic properties of the core-shell nanostructured catalysts helps in easy recycling by applying magnetic fields.<sup>231</sup> Furthermore, magnetic nanostructured powders and nanostructured arrays films are other two kinds of photocatalysts which could be recycled easily. Most recently, Zhang *et al.*<sup>232</sup> have been reported magnetic properties of CoFe<sub>2</sub>-CoFe<sub>2</sub>O<sub>4</sub> core-shell composite nanopowder formed *via* oxidation route. Interesting core-shell nanostructure in a Ge<sub>0.9</sub>Mn<sub>0.1</sub> magnetic film has also been demonstrated by Réotier *et al.*<sup>233</sup> Similarly, 1D magnetic

Ni/Ni<sub>3</sub>C core-shell nanoball chains with an average diameter ~ 30 nm were produced *via* a mild chemical solution route by Chen's group.<sup>234</sup>

## 7. Conclusions and future prospective

After reviewing the latest representative literature reports, the role of various metal-semiconductor core-shell nanostructured photocatalysts towards environmental remediation application. It was found that there are different varieties of core-shell nanostructures. Depending on the selection of core and shell materials they can be categorized in several types like metal, nonmetal, semiconductor and polymers *etc.*, core-shell nanostructures. The core-shell particles are reasonably stable and could retain their structural integrity when redispersed in aqueous medium. The core-shell structures signify a new class of metal-semiconductor composite nanomaterials that can be applied for catalytic, and biomedical, and photocatalytic usage using extraordinary electronic properties. The metal and metal-oxide semiconductor core-shell nanostructures, and metal-semiconductor heterostructures or hybrid nanoparticles with added functionalities can be used for the several applications including environmental remediation and bacterial disinfection. Core-shell nanostructures and hybrid core-shell nanoparticles will be occupying an important role in designing efficient photocatalytic reactors, medical devices, drug delivery vehicles, tissue engineering, and especially in water disinfection by germs and bacteria killing. In recent times, much consideration has been dedicated on metal nanoparticles such as Ag, Au, Pt and Pd as core material because of their physicochemical, and electronic properties such as plasmonic property which may significantly vary from their individual bulk counterparts. The metal-oxides are the interesting option as a semiconductor shell material as they are economic and easy to engineer on core particles. These shell particles have good electronic, magnetic, optoelectronic, photocatalytic and photon harvesting capability which helps in designing competent transistors, sensors, drug delivery, bioimagination and photovoltaics devices. It can be predicted that the metal-semiconductor core-shell nanostructure based core-shell functional nanoparticles have potential future compared to the other assemblies in environmental remediation application *via* photocatalytic wastewater treatment since they can perform efficient solar light driven photocatalysis. Moreover, since metal-core such as silver, gold have antimicrobial property and metal-oxide photocatalysts are known to kill germs by using UV light, the metal-semiconductor core-shell nanoparticles are promising candidates for bacterial disinfection and air purification. Additionally, the interesting physicochemical properties like nanoscale morphology, porosity, magnetic property *etc.* of the core-shell nanocatalysts comforts in easy recycling and reusability.

In order to many fold increase the activity of core-shell nanoparticles in various uses, it is essential to choose the constituents of the core-shell nanoparticles with superior catalytic, photocatalytic and electrochemical activity to interact

with the target molecules. Also manipulate more elegant nanoarchitecture without hampering the physicochemical properties of core-shell materials, like electrical and thermal conductivities, surface area, pore morphology, etc., that are promising ways for future research. Also, the surface plasmonic property of noble metals core along with metal-oxide semiconductor shell can be employed to further increase the bio- and gas sensing efficiency of core-shell nanostructures under visible light. Therefore, we need to focus for developing more reliable and simplistic technique for the synthesis of good class core-shell particles with more control over shape, size, structure and better functionality, achievable at reasonably low cost.

## Acknowledgements

Authors gratefully acknowledge to the Department of Science and Technology (DST), New Delhi, India. KM wants to acknowledge Professor Michael D Dickey of the Department of Chemical and Biomolecular Engineering, North Carolina State University, Raleigh, North Carolina-27695, United States of America for his useful suggestion during preparation of the manuscript.

## References

- 1 R. Hayes, A. Ahmed, T. Edge and H. Zhang, *J. Chromatogr. A*, 2014, **1357**, 36–52.
- 2 P. Rai, S. M. Majhi, Y.-T. Yu and J.-H. Lee, *RSC Adv.*, 2015, **5**, 76229–76248.
- 3 C. Burda, X. Chen, R. Narayanan and M. A. El-Sayed, *Chem. Rev.*, 2005, **105**, 1025–1102.
- 4 N. Zhang, S. Liu and Y.-J. Xu, *Nanoscale*, 2012, **4**, 2227–2238.
- 5 L. Lingyan, J. Luo, Q. Fan, M. Suzuki, I. S. Suzuki, M. H. Engelhard, Y. Lin, N. Kim, J. Q. Wang and C.-J. Zhong, *J. Phys. Chem. B*, 2005, **109**, 21593–21601.
- 6 L. Song, K. Mao, X. Zhou and J. Hu, *Talanta*, 2016, **146**, 285–290.
- 7 X. Su, B. C. Riggs, M. Tomozawa, J. K. Nelson and D. B. Chrisley, *J. Mater. Chem. A*, 2014, **2**, 18087–18096.
- 8 M. Zhu, X. Huang, K. Yang, X. Zhai, J. Zhang, J. He and P. Jiang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 19644–19654.
- 9 K. Chatterjee, S. Sarkar, K. Jagajjanani Rao and S. Paria, *Adv. Colloid Interface Sci.*, 2014, **209**, 8–39.
- 10 C. Yujin, Z. Chunling and W. Taihong, *Nanotechnology*, 2006, **17**, 3012.
- 11 F. Douglas, R. Yañez, J. Ros, S. Marín, A. Escosura-Muñiz, S. Alegret and A. Merkoçi, *J. Nanopart. Res.*, 2008, **10**, 97–106.
- 12 C. Li, C. Ma, F. Wang, Z. Xi, Z. Wang, Y. Deng and N. He, *J. Nanosci. Nanotechnol.*, 2012, **12**, 2964–2972.
- 13 R. Ghosh Chaudhuri and S. Paria, *Chem. Rev.*, 2012, **112**, 2373–2433.
- 14 P. Paik, *J. Nanopart.*, 2013, **2013**, 672059.
- 15 M. B. Gawande, A. Goswami, T. Asefa, H. Guo, A. V. Biradar, D.-L. Peng, R. Zboril and R. S. Varma, *Chem. Soc. Rev.*, 2015, **44**, 7540–7590.
- 16 S. Pyne, P. Sarkar, S. Basu, G. P. Sahoo, D. K. Bhui, H. Bar and A. Misra, *J. Nanopart. Res.*, 2010, **13**, 1759–1767.
- 17 B. Mandal, H. Bhattacharjee, N. Mittal, H. Sah, P. Balabathula, L. A. Thoma and G. C. Wood, *Nanomedicine: Nanotechnology, Biology and Medicine*, 2013, **9**, 474–491.
- 18 X. Xie, N. Gao, R. Deng, Q. Sun, Q.-H. Xu and X. Liu, *J. Am. Chem. Soc.*, 2013, **135**, 12608–12611.
- 19 S. Kayal and R. V. Ramanujan, *J. Nanosci. Nanotechnol.*, 2010, **10**, 5527–5539.
- 20 W. Zhu, S.-J. Lee, N. J. Castro, D. Yan, M. Keidar and L. G. Zhang, *Sci. Rep.*, 2016, **6**, 21974.
- 21 G. Guiochon and F. Gritti, *J. Chromatogr. A*, 2011, **1218**, 1915–1938.
- 22 S. K. Lee, J. M. Cho, Y. Goo, W. S. Shin, J.-C. Lee, W.-H. Lee, I.-N. Kang, H.-K. Shim and S.-J. Moon, *Chem. Commun.*, 2011, **47**, 1791–1793.
- 23 H. N. Tsao, D. Cho, J. W. Andreasen, A. Rouhanipour, D. W. Breiby, W. Pisula and K. Müllen, *Adv. Mater.*, 2009, **21**, 209–212.
- 24 L.-L. Chua, J. Zaumseil, J.-F. Chang, E. C. W. Ou, P. K. H. Ho, H. Sirringhaus and R. H. Friend, *Nature*, 2005, **434**, 194–199.
- 25 L. Yang, W. Luo and G. Cheng, *ACS Appl. Mater. Interfaces*, 2013, **5**, 8231–8240.
- 26 C. Li, S. Bolisetty, K. Chaitanya, J. Adamcik and R. Mezzenga, *Adv. Mater.*, 2012, **25**, 1010–1015.
- 27 Y. Xu, Y. Dong, J. Shi, M. Xu, Z. Zhang and X. Yang, *Catal. Commun.*, 2011, **13**, 54–58.
- 28 S. Mathew, B. S. Bhardwaj, A. D. Saran, P. Radhakrishnan, V. P. N. Nampoori, C. P. G. Vallabhan and J. R. Bellare, *Opt. Mater.*, 2015, **39**, 46–51.
- 29 A. M. El-Toni, M. A. Habila, J. P. Labis, Z. A. Allothman, M. Alhoshan, A. A. Elzatahry and F. Zhang, *Nanoscale*, 2016, **8**, 2510–2531.
- 30 X.-L. Zhang, H.-Y. Niu, W.-H. Li, Y.-L. Shi and Y.-Q. Cai, *Chem. Commun.*, 2011, **47**, 4454–4456.
- 31 N. Insin, J. B. Tracy, H. Lee, J. P. Zimmer, R. M. Westervelt and M. G. Bawendi, *ACS Nano*, 2008, **2**, 197–202.
- 32 X. Lai, J. Li, B. A. Korgel, Z. Dong, Z. Li, F. Su, J. Du and D. Wang, *Angew. Chem., Int. Ed.*, 2011, **50**, 2738–2741.
- 33 J. Liu, S. Z. Qiao, J. S. Chen, X. W. Lou, X. Xing and G. Q. Lu, *Chem. Commun.*, 2011, **47**, 12578–12591.
- 34 M. D. Brown, T. Suteewong, R. S. S. Kumar, V. D-Innocenzo, A. Petrozza, M. M. Lee, U. Wiesner and H. J. Snaith, *Nano Lett.*, 2011, **11**, 438–445.
- 35 J. Peng, X. Xu, Y. Tian, J. Wang, F. Tang and L. Li, *Appl. Phys. Lett.*, 2015, **105**, 173301.
- 36 L. Zhang, R. Tu and H. Dai, *Nano Lett.*, 2006, **6**, 2785–2789.
- 37 H. M. Fahad, C. E. Smith, J. P. Rojas and M. M. Hussain, *Nano Lett.*, 2011, **11**, 4393–4399.
- 38 O. M. Sadek, S. M. Reda and R. K. Al-Bilali, *Adv. Nanopart.*, 2013, **2**, 165–175.
- 39 T. Jiang, X. Wang, J. Zhou, D. Chen and Z. Zhao, *Nanoscale*, 2016, **8**, 4908–4914.
- 40 N. Yeole, D. Hundiwale and T. Jana, *J. Colloid Interface Sci.*, 2011, **354**, 506–510.



- 41 C. Norakankorn, Q. Pan, G. L. Rempel and S. Kiatkamjornwong, *J. Appl. Polym. Sci.*, 2009, **116**, 1291–1298.
- 42 F. Weis, M. Seipenbusch and G. Kasper, *Materials*, 2015, **8**, 966–976.
- 43 C.-W. Lai, Y.-H. Wang, B. P. Uttam, Y.-C. Chen, J.-K. Hsiao, C.-L. Liu, H.-M. Liu, C.-Y. Chen and P.-T. Chou, *Chem. Commun.*, 2008, 5342–5344.
- 44 J. Zheng, Z. Q. Liu, X. S. Zhao, M. Liu, X. Liu and W. Chu, *Nanotechnology*, 2012, **23**, 165601.
- 45 V. A. Bershtein, V. M. Gun'ko, L. M. Egorova, N. V. Guzenko, E. M. Pakhlov, V. A. Ryzhov and V. I. Zarko, *Langmuir*, 2010, **26**, 10968–10979.
- 46 A. Yang, S. Li, Y. Wang, L. Wang, X. Bao and R. Yang, *Sci. China: Technol. Sci.*, 2015, **58**, 881–888.
- 47 B. Lu, A. Liu, H. Wu, Q. Shen, T. Zhao and J. Wang, *Langmuir*, 2016, 3085–3094.
- 48 S. Duan and R. Wang, *NPG Asia Mater.*, 2014, **6**, e122.
- 49 A. M. Smith and S. Nie, *Acc. Chem. Res.*, 2010, **43**, 190–200.
- 50 Z. Jiatao, J. Muwei, L. Jiajia and X. Meng, *Metal/Semiconductor Hybrid Nanocrystals and Synergistic Photocatalysis Applications*, 2016.
- 51 R. Jiang, B. Li, C. Fang and J. Wang, *Adv. Mater.*, 2014, **26**, 5274–5309.
- 52 C. Clavero, *Nat. Photonics*, 2014, **8**, 95–103.
- 53 N. Zhang, S. Liu, X. Fu and Y.-J. Xu, *J. Phys. Chem. C*, 2011, **115**, 9136–9145.
- 54 S. Kandula and P. Jeevanandam, *Eur. J. Inorg. Chem.*, 2016, 1548–1557.
- 55 P. Singh, K. Mondal and A. Sharma, *J. Colloid Interface Sci.*, 2013, **394**, 208–215.
- 56 K. Mondal, S. Bhattacharyya and A. Sharma, *Ind. Eng. Chem. Res.*, 2014, **53**, 18900–18909.
- 57 Y. A. Ilan, D. Meisel and G. Czapski, *Isr. J. Chem.*, 1974, **12**, 891–895.
- 58 H. J. H. Fenton, *J. Chem. Soc., Trans.*, 1894, **65**, 899–910.
- 59 J. J. Pignatello, E. Oliveros and A. MacKay, *Crit. Rev. Environ. Sci. Technol.*, 2006, **36**, 1–84.
- 60 S. H. Bossmann, E. Oliveros, S. Gob, S. Siegwart, E. P. Dahlen, L. Payawan, M. Straub, M. Worner and A. M. Braun, *J. Phys. Chem. A*, 1998, **102**, 5542–5550.
- 61 R. V. Prihod'ko and N. M. Soboleva, *J. Chem.*, 2013, **8**.
- 62 P. Ciesla, P. Kocot, P. Mytych and Z. Stasicka, *J. Mol. Catal. A: Chem.*, 2004, **224**, 17–33.
- 63 M. I. Litter, *Appl. Catal., B*, 1999, **23**, 89–114.
- 64 C. Hariharan, *Appl. Catal., A*, 2006, **304**, 55–61.
- 65 M. Ye, Q. Zhang, Y. Hu, J. Ge, Z. Lu, L. He, Z. Chen and Y. Yin, *Chem.–Eur. J.*, 2010, **16**, 6243–6250.
- 66 J. C. Colmenares, R. Luque, J. M. Campelo, F. Colmenares, Z. Karpinski and A. A. Romero, *Materials*, 2009, **2**, 2228.
- 67 M. Anpo, T. Shima, S. Kodama and Y. Kubokawa, *J. Phys. Chem.*, 1987, **91**, 4305–4310.
- 68 H. Lin, C. P. Huang, W. Li, C. Ni, S. I. Shah and Y.-H. Tseng, *Appl. Catal., B*, 2006, **68**, 1–11.
- 69 D. S. Kim and S.-Y. Kwak, *Appl. Catal., A*, 2007, **323**, 110–118.
- 70 E. Auyeung, W. Morris, J. E. Mondloch, J. T. Hupp, O. K. Farha and C. A. Mirkin, *J. Am. Chem. Soc.*, 2015, **137**, 1658–1662.
- 71 S. Gheorghiu and M.-O. Coppens, *AIChE J.*, 2004, **50**, 812–820.
- 72 E. Pelizzetti and C. Minero, *Comments Inorg. Chem.*, 1994, **15**, 297–337.
- 73 T. Hisatomi, J. Kubota and K. Domen, *Chem. Soc. Rev.*, 2014, **43**, 7520–7535.
- 74 M. R. Hoffmann, S. T. Martin, W. Choi and D. W. Bahnemann, *Chem. Rev.*, 1995, **95**, 69–96.
- 75 M. M. Khan, S. F. Adil and A. Al-Mayouf, *J. Saudi Chem. Soc.*, 2015, **19**, 462–464.
- 76 A. Gupta, K. Mondal, A. Sharma and S. Bhattacharya, *RSC Adv.*, 2015, **5**, 45897–45907.
- 77 L. Matt, G. Joshua and Y. Peidong, *Annu. Rev. Mater. Res.*, 2004, **34**, 83–122.
- 78 Z. W. Pan, Z. R. Dai and Z. L. Wang, *Science*, 2001, **291**, 1947–1949.
- 79 H. B. Wu, H. H. Hng and X. W. Lou, *Adv. Mater.*, 2012, **24**, 2567–2571.
- 80 G. K. Mor, O. K. Varghese, M. Paulose, N. Mukherjee and C. A. Grimes, *J. Mater. Res.*, 2003, **18**, 2588–2593.
- 81 X. Kang and S. Chen, *J. Mater. Sci.*, 2010, **45**, 2696–2702.
- 82 C. Ratanatawanate, A. Chyao and K. J. Balkus, *J. Am. Chem. Soc.*, 2011, **133**, 3492–3497.
- 83 Z.-M. Huang, Y. Z. Zhang, M. Kotaki and S. Ramakrishna, *Compos. Sci. Technol.*, 2003, **63**, 2223–2253.
- 84 N. A. M. Nor, J. Jaafar, M. H. D. Othman and M. A. Rahman, *J. Teknol.*, 2013, **65**, 83–88.
- 85 P. Suresh Kumar, V. Aravindan, J. Sundaramurthy, V. Thavasi, S. G. Mhaisalkar, S. Ramakrishna and S. Madhavi, *RSC Adv.*, 2012, **2**, 7983–7987.
- 86 R. Ostermann, D. Li, Y. Yin, J. T. McCann and Y. Xia, *Nano Lett.*, 2006, **6**, 1297–1302.
- 87 M. A. Ali, K. Mondal, C. Singh, B. Dhar Malhotra and A. Sharma, *Nanoscale*, 2015, **7**, 7234–7245.
- 88 T. Cao, Y. Li, C. Wang, C. Shao and Y. Liu, *Langmuir*, 2011, **27**, 2946–2952.
- 89 P. Zhang, C. Shao, Z. Zhang, M. Zhang, J. Mu, Z. Guo, Y. Sun and Y. Liu, *J. Mater. Chem.*, 2011, **21**, 17746–17753.
- 90 A. Kumar, R. Jose, K. Fujihara, J. Wang and S. Ramakrishna, *Chem. Mater.*, 2007, **19**, 6536–6542.
- 91 T. Krishnamoorthy, V. Thavasi, M. G. Subodh and S. Ramakrishna, *Energy Environ. Sci.*, 2011, **4**, 2807–2812.
- 92 A. Greiner and J. H. Wendorff, *Angew. Chem., Int. Ed.*, 2007, **46**, 5670–5703.
- 93 D. Li and Y. Xia, *Adv. Mater.*, 2004, **16**, 1151–1170.
- 94 S. K. Choi, S. Kim, S. K. Lim and H. Park, *J. Phys. Chem. C*, 2010, **114**, 16475–16480.
- 95 N. Qin, Y. Liu, W. Wu, L. Shen, X. Chen, Z. Li and L. Wu, *Langmuir*, 2015, **31**, 1203–1209.
- 96 P. Li, Y. Zhou, W. Tu, Q. Liu, S. Yan and Z. Zou, *ChemPlusChem*, 2013, **78**, 274–278.
- 97 Q. Liu, Y. Zhou, J. Kou, X. Chen, Z. Tian, J. Gao, S. Yan and Z. Zou, *J. Am. Chem. Soc.*, 2010, **132**, 14385–14387.

- 98 Y. Zhou, Z. Tian, Z. Zhao, Q. Liu, J. Kou, X. Chen, J. Gao, S. Yan and Z. Zou, *ACS Appl. Mater. Interfaces*, 2011, **3**, 3594–3601.
- 99 W. Tu, Y. Zhou, Q. Liu, Z. Tian, J. Gao, X. Chen, H. Zhang, J. Liu and Z. Zou, *Adv. Funct. Mater.*, 2012, **22**, 1215–1221.
- 100 X. Chen, Y. Zhou, Q. Liu, Z. Li, J. Liu and Z. Zou, *ACS Appl. Mater. Interfaces*, 2012, **4**, 3372–3377.
- 101 S. Feng, X. Chen, Y. Zhou, W. Tu, P. Li, H. Li and Z. Zou, *Nanoscale*, 2014, **6**, 1896–1900.
- 102 M. Xing, W. Fang, X. Yang, B. Tian and J. Zhang, *Chem. Commun.*, 2014, **50**, 6637–6640.
- 103 L. Cheng, M. Shao, X. Wang and H. Hu, *Chem.–Eur. J.*, 2009, **15**, 2310–2316.
- 104 H. Zhang, G. Du, W. Lu, L. Cheng, X. Zhu and Z. Jiao, *CrystEngComm*, 2012, **14**, 3793–3801.
- 105 Z. Jiao, Y. Zhang, S. Ouyang, H. Yu, G. Lu, J. Ye and Y. Bi, *ACS Appl. Mater. Interfaces*, 2014, **6**, 19488–19493.
- 106 B. Jin, X. Zhou, X. Xu, L. Ma, Z. Wu and Y. Huang, *World J. Nano Sci. Eng.*, 2013, **3**, 1–5.
- 107 K. Mondal, J. Kumar and A. Sharma, *Nanomater. Energy*, 2013, **2**, 121–133.
- 108 M. Ahmed and G. Xinxin, *Inorg. Chem. Front.*, 2016, **3**, 578–590.
- 109 J. Zhu and M. Zäch, *Curr. Opin. Colloid Interface Sci.*, 2009, **14**, 260–269.
- 110 H. Gerischer and M. Lübke, *J. Electroanal. Chem. Interfacial Electrochem.*, 1986, **204**, 225–227.
- 111 A. J. Bard and M. A. Fox, *Acc. Chem. Res.*, 1995, **28**, 141–145.
- 112 R. Baba, S. Nakabayashi, A. Fujishima and K. Honda, *J. Phys. Chem.*, 1985, **89**, 1902–1905.
- 113 K. Vinodgopal and P. V. Kamat, *Environ. Sci. Technol.*, 1995, **29**, 841–845.
- 114 T. Jiang, X. Qin, Y. Sun and M. Yu, *RSC Adv.*, 2015, **5**, 65595–65599.
- 115 L. Zang, W. Macyk, C. Lange, W. F. Maier, C. Antonius, D. Meissner and H. Kisch, *Chem.–Eur. J.*, 2000, **6**, 379–384.
- 116 Y. Nosaka, K. Norimatsu and H. Miyama, *Chem. Phys. Lett.*, 1984, **106**, 128–131.
- 117 S. T. Martin, C. L. Morrison and M. R. Hoffmann, *J. Phys. Chem.*, 1994, **98**, 13695–13704.
- 118 W. Choi, A. Termin and M. R. Hoffmann, *J. Phys. Chem.*, 1994, **98**, 13669–13679.
- 119 N. R. de Tacconi, J. Carmona and K. Rajeshwar, *J. Phys. Chem. B*, 1997, **101**, 10151–10154.
- 120 V. Subramanian, E. Wolf and P. V. Kamat, *J. Phys. Chem. B*, 2001, **105**, 11439–11446.
- 121 H. Yu, M. Chen, P. M. Rice, S. X. Wang, R. L. White and S. Sun, *Nano Lett.*, 2005, **5**, 379–382.
- 122 R. Costi, G. Cohen, A. Salant, E. Rabani and U. Banin, *Nano Lett.*, 2009, **9**, 2031–2039.
- 123 C. Wang, H. Daimon and S. Sun, *Nano Lett.*, 2009, **9**, 1493–1496.
- 124 Z.-P. Liu, X.-Q. Gong, J. Kohanoff, C. N. Sanchez and P. Hu, *Phys. Rev. Lett.*, 2003, **91**, 266102.
- 125 W. J. Huang, R. Sun, J. Tao, L. D. Menard, R. G. Nuzzo and J. M. Zuo, *Nat. Mater.*, 2008, **7**, 308–313.
- 126 C. Wang, H. Yin, S. Dai and S. Sun, *Chem. Mater.*, 2010, **22**, 3277–3282.
- 127 V. Subramanian, E. E. Wolf and P. V. Kamat, *J. Am. Chem. Soc.*, 2004, **126**, 4943–4950.
- 128 Z. Bian, T. Tachikawa, P. Zhang, M. Fujitsuka and T. Majima, *J. Am. Chem. Soc.*, 2014, **136**, 458–465.
- 129 S. Linic, P. Christopher and D. B. Ingram, *Nat. Mater.*, 2011, **10**, 911–921.
- 130 F. Wang, Y. Jiang, A. Gautam, Y. Li and R. Amal, *ACS Catal.*, 2014, **4**, 1451–1457.
- 131 J. S. DuChene, B. C. Sweeny, A. C. Johnston-Peck, D. Su, E. A. Stach and W. D. Wei, *Angew. Chem., Int. Ed.*, 2014, **53**, 7887–7891.
- 132 M. Farnesi Camellone and D. Marx, *J. Phys. Chem. Lett.*, 2013, **4**, 514–518.
- 133 Z. W. Seh, S. Liu, M. Low, S.-Y. Zhang, Z. Liu, A. Mlayah and M.-Y. Han, *Adv. Mater.*, 2012, **24**, 2310–2314.
- 134 S. K. Dutta, S. K. Mehetor and N. Pradhan, *J. Phys. Chem. Lett.*, 2015, **6**, 936–944.
- 135 A. Furube, L. Du, K. Hara, R. Katoh and M. Tachiya, *J. Am. Chem. Soc.*, 2007, **129**, 14852–14853.
- 136 J.-S. Lee, E. V. Shevchenko and D. V. Talapin, *J. Am. Chem. Soc.*, 2008, **130**, 9673–9675.
- 137 M. T. Sheldon, P.-E. Trudeau, T. Mokari, L.-W. Wang and A. P. Alivisatos, *Nano Lett.*, 2009, **9**, 3676–3682.
- 138 J. W. Ha, T. P. A. Ruberu, R. Han, B. Dong, J. Vela and N. Fang, *J. Am. Chem. Soc.*, 2014, **136**, 1398–1408.
- 139 J. Zhang, Y. Tang, K. Lee and M. Ouyang, *Nature*, 2010, **466**, 91–95.
- 140 K. Wu, W. E. Rodriguez-Cordoba, Y. Yang and T. Lian, *Nano Lett.*, 2013, **13**, 5255–5263.
- 141 L. Amirav and A. P. Alivisatos, *J. Phys. Chem. Lett.*, 2010, **1**, 1051–1054.
- 142 X. Yu, A. Shavel, X. An, Z. Luo, M. Ibanez and A. Cabot, *J. Am. Chem. Soc.*, 2014, **136**, 9236–9239.
- 143 T. Hirakawa and P. V. Kamat, *J. Am. Chem. Soc.*, 2005, **127**, 3928–3934.
- 144 L. M. Liz-Marzan and P. Mulvaney, *J. Phys. Chem. B*, 2003, **107**, 7312–7326.
- 145 T. Ung, L. M. Liz-Marzán and P. Mulvaney, *Langmuir*, 1998, **14**, 3740–3748.
- 146 F. Caruso, M. Spasova, V. Salgueiriño-Maceira and L. M. Liz-Marzán, *Adv. Mater.*, 2001, **13**, 1090–1094.
- 147 G. Oldfield, T. Ung and P. Mulvaney, *Adv. Mater.*, 2000, **12**, 1519–1522.
- 148 Q. Zhang, T. Zhang, J. Ge and Y. Yin, *Nano Lett.*, 2008, **8**, 2867–2871.
- 149 I. Lee, J. B. Joo, Y. Yin and F. Zaera, *Angew. Chem., Int. Ed.*, 2011, **50**, 10208–10211.
- 150 T. Berger, M. Sterrer, O. Diwald, E. Knözinger, D. Panayotov, T. L. Thompson and J. T. Yates, *J. Phys. Chem. B*, 2005, **109**, 6061–6068.
- 151 M. S. Lee, S.-S. Hong and M. Mohseni, *J. Mol. Catal. A: Chem.*, 2005, **242**, 135–140.
- 152 T. Hirakawa and P. V. Kamat, *Langmuir*, 2004, **20**, 5645–5647.

- 153 N. Zhang, X. Fu and Y.-J. Xu, *J. Mater. Chem.*, 2011, **21**, 8152–8158.
- 154 N. Zhang, S. Liu, X. Fu and Y.-J. Xu, *J. Phys. Chem. C*, 2011, **115**, 22901–22909.
- 155 L. Zhang, H. Jing, G. Boisvert, J. Z. He and H. Wang, *ACS Nano*, 2012, **6**, 3514–3527.
- 156 A. Guerrero-Martínez, J. Pérez-Juste and L. M. Liz-Marzán, *Adv. Mater.*, 2010, **22**, 1182–1195.
- 157 Z. L. Wang, *Nano Today*, 2010, **5**, 540–552.
- 158 Y. Zhao, S. Li, Y. Zeng and Y. Jiang, *APL Mater.*, 2015, **3**, 086103.
- 159 H. Sun, J. He, J. Wang, S.-Y. Zhang, C. Liu, T. Sritharan, S. Mhaisalkar, M.-Y. Han, D. Wang and H. Chen, *J. Am. Chem. Soc.*, 2013, **135**, 9099–9110.
- 160 L. Xu, M.-L. Yin and S. Liu, *J. Alloys Compd.*, 2015, **623**, 127–131.
- 161 L. Xu, M.-L. Yin and S. Liu, *Sci. Rep.*, 2014, **4**, 6745.
- 162 B. Lu, A. Liu, H. Wu, Q. Shen, T. Zhao and J. Wang, *Langmuir*, 2016, **32**, 3085–3094.
- 163 W. Zhang, X.-J. Lin, Y.-G. Sun, D.-S. Bin, A.-M. Cao and L.-J. Wan, *ACS Appl. Mater. Interfaces*, 2015, **7**, 27031–27034.
- 164 J. Hu, Z. Wen, Q. Wang, X. Yao, Q. Zhang, J. Zhou and J. Li, *J. Phys. Chem. B*, 2006, **110**, 24305–24310.
- 165 N. Zhang and Y.-J. Xu, *Chem. Mater.*, 2013, **25**, 1979–1988.
- 166 B. Y. Kim, I.-B. Shim, Z. O. Araci, S. S. Saavedra, O. L. A. Monti, N. R. Armstrong, R. Sahoo, D. N. Srivastava and J. Pyun, *J. Am. Chem. Soc.*, 2010, **132**, 3234–3235.
- 167 Z. Zhuang, W. Sheng and Y. Yan, *Adv. Mater.*, 2014, **26**, 3950–3955.
- 168 Y. Liu, W. Zhang, S. Li, C. Cui, J. Wu, H. Chen and F. Huo, *Chem. Mater.*, 2014, **26**, 1119–1125.
- 169 S.-H. Chang, P.-Y. Yang, C.-M. Lai, S.-C. Lu, G.-A. Li, W.-C. Chang and H.-Y. Tuan, *CrystEngComm*, 2016, **18**, 616–621.
- 170 B. Zhong, X. Tang, X. Huang, L. Xia, X. Zhang, G. Wen and Z. Chen, *CrystEngComm*, 2015, **17**, 2806–2814.
- 171 A. P. Bakhtinov, V. M. Vodopyanov, Z. R. Kudrynskiy, M. Z. Kovalyuk, V. V. Netyaga, V. L. Karbivskyy, V. V. Vishniak and O. S. Lytvyn, *Phys. Status Solidi A*, 2014, **211**, 342–350.
- 172 Y. P. Ivanov, A. Alfadhel, M. Alnassar, J. E. Perez, M. Vazquez, A. Chuvilin and J. R. Kosel, *Sci. Rep.*, 2016, **6**, 24189.
- 173 B. Rico-Oller, A. Boudjemaa, H. Bahruji, M. Kebir, S. Prashar, K. Bachari, M. Fajardo and S. Gomez-Ruiz, *Sci. Total Environ.*, 2016, **563–564**, 921–932.
- 174 S. Lazaro-Navas, S. Prashar, M. Fajardo and S. Gomez-Ruiz, *J. Nanopart. Res.*, 2015, **17**, 1–14.
- 175 L. Shan and Y. Liu, *J. Mol. Catal. A: Chem.*, 2016, **416**, 1–9.
- 176 Y. Pihosh, I. Turkevych, K. Mawatari, J. Uemura, Y. Kazoe, S. Kosar, K. Makita, T. Sugaya, T. Matsui, D. Fujita, M. Tosa, M. Kondo and T. Kitamori, *Sci. Rep.*, 2015, **5**, 11141.
- 177 M. N. Chong, B. Jin, C. W. K. Chow and C. Saint, *Water Res.*, 2010, **44**, 2997–3027.
- 178 V. Scuderi, G. Impellizzeri, L. Romano, M. Scuderi, M. V. Brundo, K. Bergum, M. Zimbone, R. Sanz, M. A. Buccheri, F. Simone, G. Nicotra, B. G. Svensson, M. G. Grimaldi and V. Privitera, *Nanoscale*, 2014, **6**, 11189–11195.
- 179 V. Scuderi, G. Impellizzeri, L. Romano, M. Scuderi, G. Nicotra, K. Bergum, A. Irrera, B. G. Svensson and V. Privitera, *Nanoscale Res. Lett.*, 2014, **9**, 458.
- 180 G. Impellizzeri, V. Scuderi, L. Romano, E. Napolitani, R. Sanz, R. Carles and V. Privitera, *J. Appl. Phys.*, 2015, **117**, 105308.
- 181 G. Impellizzeri, V. Scuderi, L. Romano, P. M. Sberna, E. Arcadipane, R. Sanz, M. Scuderi, G. Nicotra, M. Bayle, R. Carles, F. Simone and V. Privitera, *J. Appl. Phys.*, 2014, **116**, 173507.
- 182 A. O. Ibhaddon and P. Fitzpatrick, *Catalysts*, 2013, **3**, 189–218.
- 183 O. Bar-Elli, E. Grinvald, N. Meir, L. Neeman and D. Oron, *ACS Nano*, 2015, **9**, 8064–8069.
- 184 J. Goebel, J. B. Joo, M. Dahl and Y. Yin, *Catal. Today*, 2014, **225**, 90–95.
- 185 Y. Sato, S.-i. Naya and H. Tada, *APL Mater.*, 2015, **3**, 104502.
- 186 Y.-C. Chang, C.-Y. Yan and R.-J. Wu, *J. Chin. Chem. Soc.*, 2014, **61**, 345–349.
- 187 J. Fang, L. Yin, S. Cao, Y. Liao and C. Xue, *Beilstein J. Nanotechnol.*, 2014, **5**, 360–364.
- 188 Y.-W. Chen and D.-S. Lee, *Open Access Library Journal*, 2014, **1**, 1–14.
- 189 A. Khanna and V. K. Shetty, *Sol. Energy*, 2014, **99**, 67–76.
- 190 X. Zhang, Y. Zhu, X. Yang, S. Wang, J. Shen, B. Lin and C. Li, *Nanoscale*, 2013, **5**, 3359–3366.
- 191 J. Li, S. K. Cushing, J. Bright, F. Meng, T. R. Senty, P. Zheng, A. D. Bristow and N. Wu, *ACS Catal.*, 2013, **3**, 47–51.
- 192 X.-Y. Zhang, S.-L. Yuan, Y.-Z. Yuan and X. Li, *Optoelectron. Lett.*, 2015, **11**, 1–4.
- 193 J. Xiong, Q. Sun, J. Chen, Z. Li and S. Dou, *CrystEngComm*, 2016, **18**, 1713–1722.
- 194 S. Das, S. Sinha, M. Suar, S.-I. Yun, A. Mishra and S. K. Tripathy, *J. Photochem. Photobiol., B*, 2015, **142**, 68–76.
- 195 Y. Cao, X. Gu, H. Yu, W. Zeng, X. Liu, S. Jiang and Y. Li, *Chemosphere*, 2015, **144**, 836–841.
- 196 Y. Zhang and P. Li, *RSC Adv.*, 2015, **5**, 98118–98129.
- 197 N. Zhang, S. Liu, X. Fu and Y.-J. Xu, *J. Mater. Chem.*, 2012, **22**, 5042–5052.
- 198 N. Singh, K. Mondal, M. Misra, A. Sharma and R. K. Gupta, *RSC Adv.*, 2016, **6**, 48109–48119.
- 199 C.-W. Tsai, H. M. Chen, R.-S. Liu, K. Asakura and T.-S. Chan, *J. Phys. Chem. C*, 2011, **115**, 10180–10186.
- 200 R. Rahimi, M. Heidari-Golafzani and M. Rabbani, *Superlattices Microstruct.*, 2015, **85**, 497–503.
- 201 M. Pelaez, N. T. Nolan, S. C. Pillai, M. K. Seery, P. Falaras, A. G. Kontos, P. S. M. Dunlop, J. W. J. Hamilton, J. A. Byrne, K. O'Shea, M. H. Entezari and D. D. Dionysiou, *Appl. Catal., B*, 2012, **125**, 331–349.
- 202 S. Malato, J. Blanco, D. C. Alarcon, M. I. Maldonado, P. Fernandez-Ibanez and W. Gernjak, *Catal. Today*, 2007, **122**, 137–149.
- 203 D. Gumy, A. G. Rincon, R. Hajdu and C. Pulgarin, *Sol. Energy*, 2006, **80**, 1376–1381.
- 204 T. Matsunaga, R. Tomoda, T. Nakajima and H. Wake, *FEMS Microbiol. Lett.*, 1985, **29**, 211–214.



- 205 J. C. Yu, W. Ho, J. Yu, H. Yip, P. K. Wong and J. Zhao, *Environ. Sci. Technol.*, 2005, **39**, 1175–1179.
- 206 T. Saito, T. Iwase, J. Horie and T. Morioka, *J. Photochem. Photobiol., B*, 1992, **14**, 369–379.
- 207 K. Sunada, Y. Kikuchi, K. Hashimoto and A. Fujishima, *Environ. Sci. Technol.*, 1998, **32**, 726–728.
- 208 P.-C. Maness, S. Smolinski, D. M. Blake, Z. Huang, E. J. Wolfrum and W. A. Jacoby, *Appl. Environ. Microbiol.*, 1999, **65**, 4094–4098.
- 209 Z. Huang, P.-C. Maness, D. M. Blake, E. J. Wolfrum, S. L. Smolinski and W. A. Jacoby, *J. Photochem. Photobiol., A*, 2000, **130**, 163–170.
- 210 K. Sunada, T. Watanabe and K. Hashimoto, *J. Photochem. Photobiol., A*, 2003, **156**, 227–233.
- 211 N.-P. Huang, X. Min-hua, C.-W. Yuan and Y. Rui-rong, *J. Photochem. Photobiol., A*, 1997, **108**, 229–233.
- 212 S. Kim, T. Taguchi, M. Nishioka and M. Taya, *Biochem. Eng. J.*, 2004, **22**, 81–87.
- 213 H. Hidaka, S. Horikoshi, N. Serpone and J. Knowland, *J. Photochem. Photobiol., A*, 1997, **111**, 205–213.
- 214 D. M. Blake, P.-C. Maness, Z. Huang, E. J. Wolfrum, J. Huang and W. A. Jacoby, *Sep. Purif. Methods*, 1999, **28**, 1–50.
- 215 K. I. Dhanalekshmi and K. S. Meena, *Mater. Sci. Eng., C*, 2016, **59**, 1063–1068.
- 216 K. S. Meena, K. I. Dhanalekshmi and K. Jayamoorthy, *Mater. Sci. Eng., C*, 2016, **63**, 317–322.
- 217 C. Karunakaran and P. Vinayagamoorthy, *New J. Chem.*, 2016, **40**, 1845–1852.
- 218 K. I. Dhanalekshmi, K. S. Meena and I. Ramesh, *Int. J. Nanotechnol. Appl.*, 2013, **3**, 5–14.
- 219 K. I. Dhanalekshmi and K. S. Meena, *Spectrochim. Acta, Part A*, 2014, **128**, 887–890.
- 220 S. Rana, J. Rawat and R. D. K. Misra, *Acta Biomater.*, 2005, **1**, 691–703.
- 221 Z. Liao, H. Wang, R. Lv, P. Zhao, X. Sun, S. Wang, W. Su, R. Niu and J. Chang, *Langmuir*, 2011, **27**, 3100–3105.
- 222 X. Guo, F. Mao, W. Wang, Y. Yang and Z. Bai, *ACS Appl. Mater. Interfaces*, 2015, **7**, 14983–14991.
- 223 M. Atif, M. Fakhar-e-Alam, N. Abbas, M. A. Siddiqui, A. A. Ansari, A. A. Al-Khedhairi and Z. M. Wang, *J. Nanomater.*, 2016, **6**.
- 224 B. Li, H. Fan, Q. Zhao and C. Wang, *Materials*, 2016, **9**, 149.
- 225 A. Sharma, Y. Qiang, D. Meyer, R. Souza, A. McConnaughoy, L. Muldoon and D. Baer, *J. Appl. Phys.*, 2008, **103**, 07A308.
- 226 R. W. Matthews, *Pure Appl. Chem.*, 1992, **64**, 1285.
- 227 A. K. Ray and A. A. C. M. Beenackers, *Catal. Today*, 1998, **40**, 73–83.
- 228 M. Janssen, C. Muller and D. Vogt, *Green Chem.*, 2011, **13**, 2247–2257.
- 229 S. M. Mercer, T. Robert, D. V. Dixon and P. G. Jessop, *Catal. Sci. Technol.*, 2012, **2**, 1315–1318.
- 230 O. Myakonkaya, B. Deniau, J. Eastoe, S. E. Rogers, A. Ghigo, M. Hollamby, A. Vesperinas, M. Sankar, S. H. Taylor, J. K. Bartley and G. J. Hutchings, *J. Colloid Interface Sci.*, 2010, **350**, 443–446.
- 231 T.-Y. Liu, L.-Y. Huang, S.-H. Hu, M.-C. Yang and S.-Y. Chen, *J. Biomed. Nanotechnol.*, 2007, **3**, 353–359.
- 232 Y. Zhang, B. Yan, J. Ou-Yang, B. Zhu, S. Chen, X. Yang, Y. Liu and R. Xiong, *Ceram. Int.*, 2015, **41**, 11836–11843.
- 233 P. Dalmas de Réotier, E. Prestat, P. Bayle-Guillemaud, M. Boukhari, A. Barski, A. Marty, M. Jamet, A. Suter, T. Prokscha, Z. Salman, E. Morenzoni and A. Yaouanc, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 245408.
- 234 W. Zhou, K. Zheng, L. He, R. Wang, L. Guo, C. Chen, X. Han and Z. Zhang, *Nano Lett.*, 2008, **8**, 1147–1152.