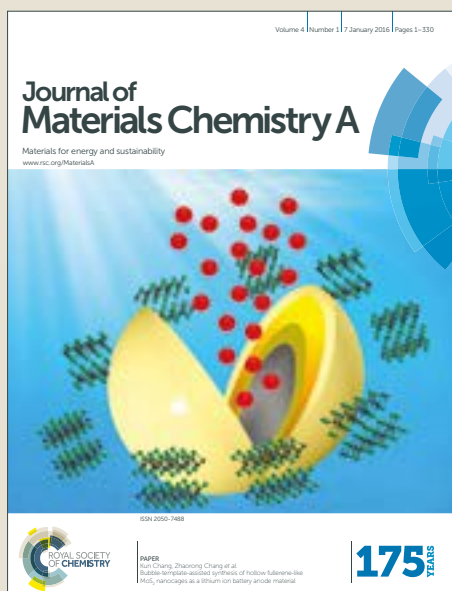


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1 **Recent Advances in Transition Metal-Based Electrocatalysts for** View Article Online
DOI: 10.1039/C9TA03220G

2 **Alkaline Hydrogen Evolution**

3
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26 **Abstract**

27 With the increasing demands in energy consumption and raising environmental concerns,
28 it is of vital significance for developing renewable and clean energy sources to substitute
29 traditional fossil fuels. As an outstanding candidate, hydrogen is recognized as a green
30 energy carrier for its high gravimetric energy density, zero carbon footprints, and earth-
31 abundance. Currently, water splitting in alkaline electrolytes represents one of the most
32 promising methods for sustainable hydrogen production, and the key challenge lies in
33 the development of high-performance electrocatalysts for hydrogen evolution reaction
34 (HER). Given the rapid advances in the design and development of efficient catalysts
35 towards alkaline HER, especially the capable transition metal (TM)-based materials, this
36 review aims to summarise recent progress in the theoretical understanding of alkaline
37 HER and the TM-based electrocatalysts. TM-based catalysts classified by their different
38 anionic compositions (metals, alloys, oxides, hydroxides, sulfides, selenides, tellurides,
39 nitrides, phosphides, carbides, and borides) are comprehensively showcased. Special
40 attention is given to the mainstream strategies that can upgrade the catalytic properties
41 of each category, as well as the underlying structure-activity regimes. Additionally, the
42 challenges for the future development of novel catalysts are also analyzed.

43

44 **Keywords:** water splitting; alkaline hydrogen evolution reaction; transition metals;
45 electrocatalysts

46

47 **1. Introduction**

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48 Developing sustainable and clean energy supplies to substitute traditional fossil fuels has
49 aroused great concerns because of the approaching environmental and energy crisis.^{1, 2}
50 In the last few decades, renewable energy resources (e.g., solar, wind, and geothermal)
51 have been extensively explored as sustainable alternatives for conventional fossil fuels.
52 However, the spatial and temporal distribution of these energy sources has deeply limited
53 their large-scale application.^{3, 4} Hydrogen has been recognized as a green energy carrier
54 for its high gravimetric energy density, environmental friendliness, and earth-
55 abundance.^{4, 5} Hitherto, industrial hydrogen production is largely dominated by steam
56 methane reforming and coal gasification, whereas these processes are intensive energy
57 consuming and will result in low hydrogen purity with massive pollutant/carbon
58 emissions.^{6, 7} Alternatively, water electrocatalysis is a clean and sustainable approach to
59 produce high-purity hydrogen (>99%).

60
61 Electrocatalytic water splitting can be conducted in both acidic and alkaline solutions.
62 Currently, acidic HER is limited by the unsatisfactory stability of the metal catalyst and
63 poor membrane permanence.⁸ In contrast, alkaline electrolyzers are technologically well
64 developed and are commercially available.⁹ Furthermore, alkaline hydrogen evolution is
65 also an important reaction in the chlor-alkali process.¹⁰ Herein, renovating alkaline HER
66 techniques is of crucial importance to fulfill the enormous demands for hydrogen energy.
67 Unfortunately, HER in alkaline electrolytes is strictly hampered by the sluggish kinetics.⁵
68 To accelerate HER, numerous catalysts have been developed in recent years. Platinum
69 group metals (PGMs, e.g., Pt and Pd) based materials are the most attractive HER
70 catalysts, which possess low overpotentials, low Tafel slopes, high exchange current
71 densities, and remarkable stabilities in water reduction reactions. Nevertheless, the

72 prohibitive costs and scarcity of these noble metals limit the commercial application of
73 PGMs catalysts. The earth-abundant transition metals (TMs) based catalysts also show
74 great potentials for alkaline HER. The commonly applied TMs based electrocatalysts
75 include TMs, TM alloys, and TMXs (X= O, S, Se, Te, N, P, C, and B). These materials
76 have been extensively studied, and to our delight, some exhibit extremely high capability
77 in HER even comparable to that of PGMs-based catalysts.

78

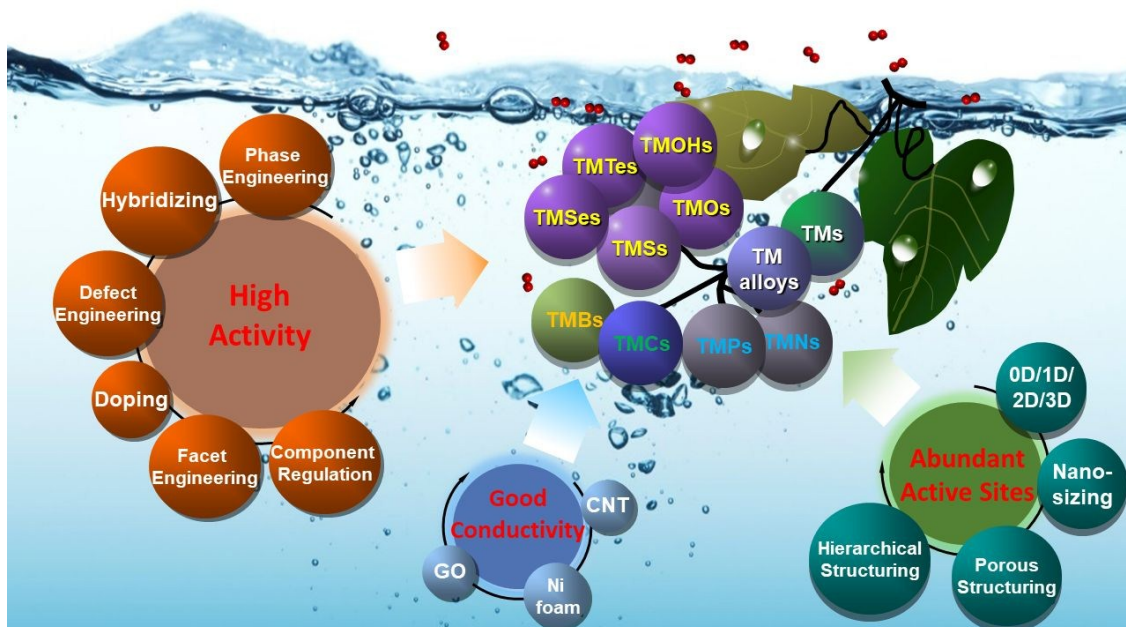
79 To date, a handful of reviews have been dedicated with focusses on the mechanisms,
80 challenges, and emerging strategies in alkaline HER.^{7, 11, 12} For electrocatalysts, Jamesh
81 et al. analyzed the earth-abundant-elements based catalysts for alkaline HER.¹³ Gong et
82 al. reviewed the Ni-based catalysts,⁴ and Wei et al. summarised the heterostructure
83 materials including Pt- (Pt/Ni(OH)₂) and MoS₂ -based catalysts.³ Lately, the compelling
84 role of anions in electrocatalysts has been extensively researched.¹⁴⁻¹⁹ This review aims
85 to not only offer a comprehensive overview of the latest progress of TMs (mainly refer
86 to non-noble metals) based catalysts grouped by different anionic elements but also
87 identify the advanced strategies to precisely regulate the atomic structure and electronic
88 configurations of the electrocatalysts toward alkaline HER.

89

90 Herein, we will first discuss the electrochemistry of alkaline HER and several popular
91 parameters/indexes for evaluating the properties of catalysts. Second, fruitful
92 achievements of TM-based catalysts will be described classified by different anionic
93 elements, including TMs, TM alloys, TMOs, LDHs, TMSs, TMSes, TMTes, TMNs,
94 TMPs, TMCs, and TMBs. Particularly, the state-of-the-art strategies for tailoring the
95 catalytic performance and features of different classified catalysts are highlighted
96 (**Scheme 1**). The underlying structure-performance mechanisms will be unveiled. Finally,

97 we present the major challenges and future perspectives in this booming field. We hope
 98 that this timely and comprehensive review will inspire more future studies to advance
 99 the design principles and promote the application of TMs based catalysts with respectful
 100 performance for HER in alkaline electrolytes.

101



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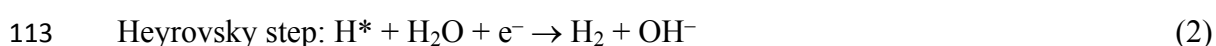
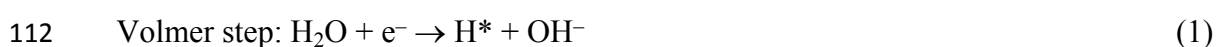
103 **Scheme 1.** Transition Metal-Based Electrocatalysts for Alkaline Hydrogen Evolution

104

105 **2. Electrochemistry of hydrogen evolution in alkaline solution**

106 **2.1 General mechanisms of alkaline HER**

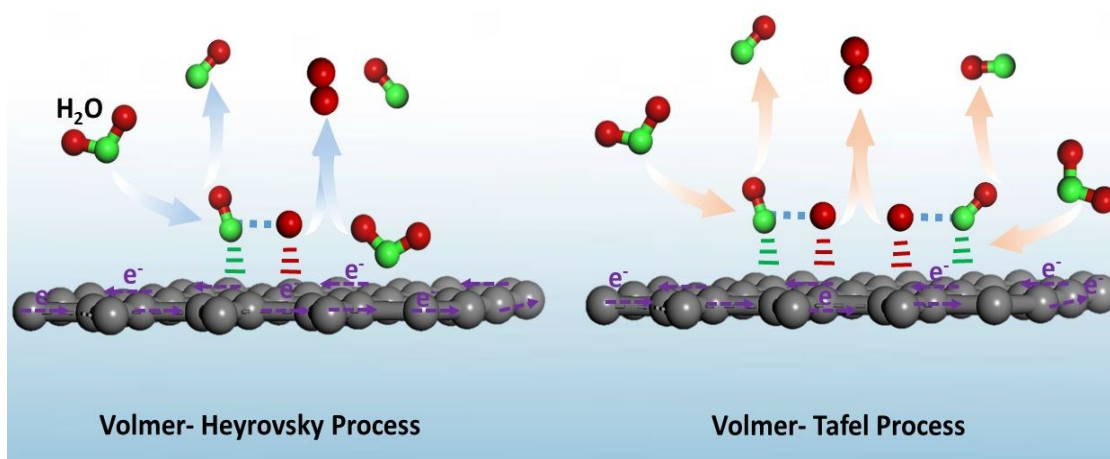
107 The HER in alkaline solution begins with water adsorption and dissociation process
 108 which can generate H* by breaking the H-O-H bonds. It is widely recognized that this
 109 process is the rate-determining step (RDS) in HER because extra energy is required to
 110 produce protons. Commonly, the alkaline HER follows the classical Volmer-Heyrovsky
 111 process or the Volmer-Tafel process (**Figure 1**, Equations 1-3).



114 Tafel step: $2\text{H}^* \rightarrow \text{H}_2$

115 Apart from the reduction of protons into adsorbed H^* , the Volmer step in alkaline
116 electrolytes involves two key steps, the adsorption and dissociation of water molecules
117 over the catalyst surface.⁴ These multi-step surface reactions make the Volmer step more
118 intricate than that of the acidic HER in which the sophisticated “volcano plot”
119 successfully correlates the intrinsic catalytic activity of the catalysts with the adsorption
120 energy of H atoms. However, there have not yet highly developed principles to describe
121 the complicated alkaline HER process.

122



123

124 **Figure 1.** Schematic diagram of the Volmer-Heyrovsky and Volmer-Tafel processes on
125 a catalyst surface.

126

127 Most of the research articles and reviews reported that the inferior performance of
128 electrocatalysts in alkaline solution to the acidic one is due to the extra energy required
129 in the Volmer step.^{4, 20} However, several emerging catalysts (e.g., Ru-MoO₂,²¹ MoO₂/α-
130 Mo₂C,²² Mo₂C,^{23, 24} Mo₂C/G-NCS,²⁵ C-CWC,²⁶ DR-MoP,²⁷ NiCo₂P₂-ACNT,²⁸ NCP,²⁹
131 NiCo₂P_x,³⁰ Ni-Cu-P³¹) have been developed which are active both in acid and alkaline
132 media and demonstrate better catalytic capability in alkaline electrolytes than in acidic
133 media. Nevertheless, there is a lack of understanding of their better performance in
134 alkaline solutions.

135

View Article Online
DOI: 10.1039/C9TA03220G136 **2.2 Energetics of alkaline HER**

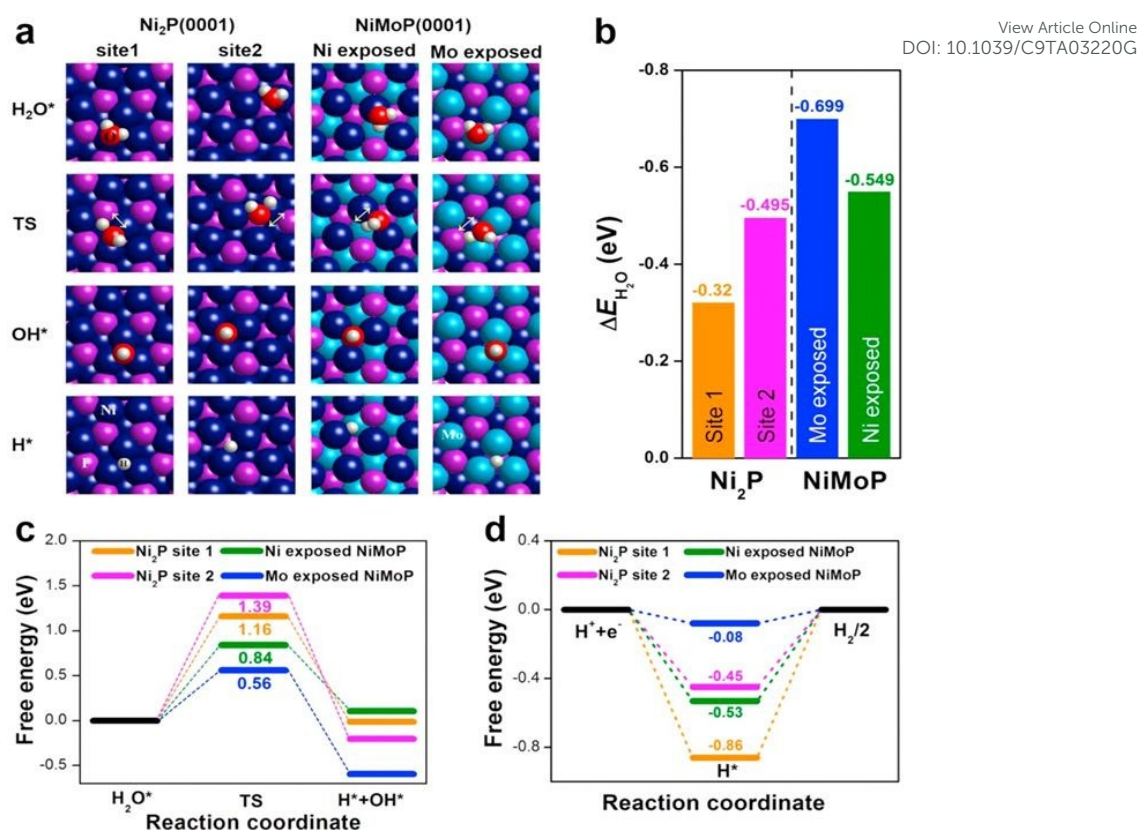
137 As depicted in **Figure 1**, HER in basic solutions may proceed via two pathways, i.e.,
138 either Volmer-Tafel or Volmer-Heyrovsky reactions. Accordingly, the conversion of
139 water molecules to hydrogen molecules contains three successive stages, including water
140 adsorption, water dissociation, and hydrogen generation. Generally, three corresponding
141 computational descriptors, water adsorption energy (E_{ad}), the activation energy of water
142 dissociation (E_{ac}), and adsorption-free energy of H^* (ΔG_{H^*}) are employed to evaluate the
143 ease of a catalyst to initiate the reactions.

144

145 ΔG_{H^*} is the most frequently documented one among the three parameters, which can
146 illustrate the binding strength of H^* on the catalyst surface both in acidic and alkaline
147 media. The hydrogen generation is most feasible if the value of ΔG_{H^*} is close to zero,
148 because of the optimal balance between absorption and desorption of hydrogen atoms on
149 the catalyst surface.⁸ Nevertheless, the alkaline HER cannot be described merely by
150 ΔG_{H^*} .^{8,32} Recently, the E_{ad} and E_{ac} have attracted increasing attention due to their critical
151 roles in the Volmer step of the whole alkaline HER process. For E_{ad} , a lower value
152 indicates a better affinity between the water molecule and catalyst surface, thus
153 facilitating the following reactions. A smaller E_{ac} represents a faster water dissociation
154 process because less activation energy is required. Therefore, a desirable catalyst should
155 possess a low E_{ad} , a minor E_{ac} , and a small absolute value of ΔG_{H^*} . To give an example,
156 Yu et al. investigated the catalytic performance of $Ni_{2(1-x)}Mo_{2x}P$,³³ and the computed
157 energetic profiles are presented in **Figure 2**. Firstly, the E_{ad} of ternary NiMoP were lower
158 than those of Ni_2P (**Figure 2a and b**), especially for the Mo exposed plane (-0.699 eV).
159 Secondly, Mo exposed plane of NiMoP also possesses the lowest E_{ac} over the other

160 selected configurations (**Figure 2c**). Therefore, the Volmer step on NiMoP surface is
161 more favourable than that on Ni₂P catalyst. Finally, the ΔG_{H^*} values also support that
162 Mo exposed NiMoP owns the best hydrogen generation capability due to its lowest
163 $|\Delta G_{H^*}|$, thus benefiting the Heyrovsky step (**Figure 2d**). The E_{ad} , E_{ac} , and ΔG_{H^*} of
164 different samples collaboratively suggest that NiMoP manifests better HER performance.
165 It is worth noting that the obtained energies of different active sites/facets on the catalyst
166 surface are important indicators of the underlying mechanisms. Also, they provide
167 valuable guidance for the designing of desirable catalysts.

168
169 To gain a better understanding of alkaline HER, comprehensively computational
170 energetics are required. The values of E_{ad} , E_{ac} , and ΔG_{H^*} of catalysts are sound indicators
171 of their corresponding properties of water adsorption, water dissociation, and hydrogen
172 generation. For instance, the catalytic activity of Pt in alkaline media is about two orders
173 of magnitude lower than that in acidic solutions because of the sluggish water
174 dissociation process (high E_{ac}).⁸ Under this circumstance, the presence of a cocatalyst to
175 effectively cleavage the O-H bonds is highly favourable, such as the layered double
176 hydroxides (LDHs)-based materials.³ Thus, the Pt-Ni(OH)₂ hybrids possess a perfect
177 combination of E_{ad} , E_{ac} , and ΔG_{H^*} and exhibits appealing alkaline HER ability. Therefore,
178 the computational energies are handy descriptors in the design of high-performance
179 electrocatalysts.



180

181 **Figure 2.** Density functional theory (DFT) calculations. (a) Chemisorption models of
 182 H₂O adsorption, OH adsorption, and H adsorption for the calculated free energies over
 183 Ni₂P (0001) and NiMoP (0001) terminations. Here TS represents a transition state of
 184 H₂O activation. (b) The corresponding adsorption free energy of H₂O, (c) free energy
 185 diagram for H₂O activation (cleavage of O-H bonds of H₂O molecules), and (d) free
 186 energy diagram for H adsorption (ΔG_{H}) on the (0001) surfaces of Ni₂P and NiMoP.
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 188

189 2.3 Key parameters for evaluating catalysts in alkaline HER

190 As reported, many electrocatalytic parameters can be used to assess the overall
 191 electrochemical performance of the developed catalysts from different aspects, although
 192 it is difficult to compare the performance of different catalysts for alkaline HER.³⁴ The
 193 most frequently reported parameters are the overpotential and Tafel slope, which can
 194 offer key information of catalytic activities as well as HER mechanisms.³⁵ Other valuable
 195 parameters are also evaluated including the exchange current density, turnover frequency,
 196 Faradaic efficiency, mass and specific activities, and stability. In this part, the
 197 aforementioned parameters are briefly introduced and discussed, and more information

198 about these parameters can be referred to a comprehensive review reported by S
199 Anantharaj et al.³⁶ New Article Online
DOI: 10.1039/C9TA03220G

200

201 2.3.1 Overpotential

202 Theoretically, the overall water-splitting processes only require a cell voltage of 1.23V
203 (0 V is for HER and 1.23V for oxygen evolution reaction (OER)). However, due to the
204 reaction hindrances in the real system, both HER and OER require additional potentials
205 to initiate the reactions.¹⁴ The extra potential is called overpotential (η), which is a key
206 parameter to evaluate the activity of catalysts. Typically, the overpotentials at a fixed
207 current density such as 10 mA cm⁻² (η_{10}) and/or 100 mA cm⁻² (η_{100}) of different catalysts
208 are compared, and a smaller η represents a better electrocatalytic activity. Commonly,
209 the iR corrected overpotentials are reported, but an actual water electrolysis cell does not
210 run according to iR corrected overpotentials. Hence, it is necessary to report the iR
211 uncompensated data in addition to the iR corrected overpotential to further simulate the
212 real water electrolysis system.³⁶

213

214 2.3.2 Tafel slope and exchange current density

215 Tafel slope is one of the primary indicators for evaluating the activity of catalysts, which
216 is beneficial to predict the kinetics of alkaline HER process.³⁶ It means the rising rates of
217 current density with the increase of overpotential which can be obtained from the Tafel
218 plot. The Tafel plot is derived from the linear sweep voltammetry (LSV) results and
219 recorded with the linear portion at relatively low overpotentials fitted by the Tafel
220 equation ($\eta = a + b \log j$, where η is the overpotential, b is the Tafel slope, and j is the
221 current density).¹⁴ The calculated Tafel slope is in inverse ratio to the charge transfer
222 coefficient, and catalysts with high charge transfer ability favour a small Tafel slope.³⁷

223 Also, the Tafel slope is an indicator of the reaction pathway and RDS. Specifically, the
224 Tafel slopes of the Volmer, Heyrovsky, and Tafel reactions are calculated to be 118.2
225 mV dec⁻¹, 39.4 mV dec⁻¹, and 29.6 mV dec⁻¹, respectively.³⁰ For a typical catalyst,
226 Volmer reaction is the rate-determining step when the calculated b is close to 118.2 mV
227 dec⁻¹, and kinetics for adsorption of H atoms onto catalyst surface are sluggish.³⁸ If the b
228 is around 39.4 mV dec⁻¹, Heyrovsky while the generation of H₂ is principally controlled
229 by desorption process. If b approaches 29.6 mV dec⁻¹, the combination of adsorbed H
230 atoms and the desorption of H₂ will be rate-limiting steps. As a result, it is convenient to
231 judge the hydrogen evolution pathway through the calculated value of the Tafel slope.
232 For example, the Tafel slope of V₈C₇ NMs/GR is about 89.4 mV dec⁻¹ which falls within
233 the range of 39.4-118.2 mV dec⁻¹, illustrating the Volmer-Heyrovsky processes.³⁹
234 Similarly, the NiCo₂P_x possesses a small Tafel slope of 34.3 mV dec⁻¹, implying a
235 Volmer-Tafel mechanism.³⁰

236

237 Plotting $\log(I/R_{ct})$ vs. η (where R_{ct} is the charge transfer resistance) from the
238 electrochemical impedance spectroscopy (EIS) data is another method to obtain the Tafel
239 slope. It is suggested that the latter method can simply reveal the charge transfer kinetics,
240 while Tafel slopes obtained from the former one may include contributions from catalyst
241 and electrode resistance.⁴⁰ However, the value of Tafel slope is greatly influenced by the
242 selected experimental parameters (e.g., potential ranges). Tafel slopes over a narrow low-
243 potential range can attain very low values.^{34, 37}

244

245 When $\eta = 0$, the acquired current density from the Tafel equation is named exchange
246 current density (j_0). It can be obtained by extrapolating the linear part of Tafel plots.
247 Normally, j_0 indicates the intrinsic activity of the catalysts under the equilibrium state,

248 the value is similar to the rate constants employed in the heterogeneously electrocatalytic
249 process.³⁸ The j_0 is determined by both thermodynamics and kinetics. The intrinsic
250 property of electrocatalysts and experimental conditions (e.g., electrolyte composition
251 and temperature) will also affect the value.³⁸ Accordingly, a desirable high-performance
252 electrocatalyst possesses a low b and a high j_0 . A detailed discussion of Tafel slope and
253 j_0 is presented in a recent review.³⁶

254

255 2.3.3 Mass and specific activities

256 The mass and specific activities are quantitative parameters to illustrate the catalytic
257 reactivity of electrocatalysts. The current normalized by the catalyst loading is the mass
258 activity, which is normally expressed in amperes per gram (A/g). The current normalized
259 by the Brunauer-Emmett-Teller (BET) surface area or the electrochemical surface area
260 (ECSA) is the specific activity. As suggested, the BET strategy often leads to a highly
261 inexact result, while ECSA normalized activity has gained more attentions in recent
262 studies.³⁶ ECSA is considered to disclose the intrinsic surface area of the catalyst exposed
263 to the electrolyte. Hence, the activity calculated by ECSA appears to be more precise. In
264 other words, this method offers an accurate measurement of the catalytic ability of
265 different catalysts with distinguished components, morphology, size, shape, topography
266 and porous structure.³⁶ In some cases, the double layer capacitance (C_{dl}) of the catalyst
267 is offered in the literature because of its positive correlation with ECSA. A detailed
268 discussion involves mass and specific activities can be found in a previous review.³⁷

269

270 2.3.4 Turnover frequency

271 The turnover frequency (TOF) is an important kinetic parameter defined as the number
272 of reactants that can be converted by the catalyst to the desired product per catalytic site

273 per unit of time. Hence, independent of the active surface area, TOF can rigorously reflect
274 the intrinsic activity of active centers on a catalyst.^{14, 34} TOF can be obtained from the
275 equation: $\text{TOF} = (j \times A)(\alpha \times F \times n)^{-1}$, where j is the current density at a given
276 overpotential, A is the surface area of the working electrode, α is the electron transfer
277 number at a given overpotential, F is the Faraday constant, and n is the number of moles
278 of coated metal atoms on the electrode. It should be mentioned that not all the atoms in
279 the catalyst deliver catalytic activities and can be equally accessible. Nevertheless, the
280 calculated TOF is still a handy tool for comparing the activities of similar catalysts.

281

282 2.3.5 Faradaic efficiency

283 Faraday efficiency (FE) is used to quantify the number of electrons that are engaged in
284 the target reactions instead of the side reactions. It is the ratio of the quantity of
285 experimentally produced H_2 to the theoretical calculated H_2 amount in HER. The amount
286 of the product (H_2) can be measured by gas chromatography (GC) or the water
287 displacement method. Normally, the reported FE in HER are close to 100%, and a higher
288 FE represents a better selectivity for the alkaline HER process.

289

290 2.3.6 Stability

291 Apart from those aforementioned parameters that focus on the activity of catalysts,
292 stability is another vital issue which can determine the feasibility of an electrocatalyst in
293 practical applications. There are two popular methods to evaluate the long-term
294 durability of catalysts. One is to record the chronopotentiometry curves or
295 chronoamperometry curves at constant current densities in a long operation (>12 h). The
296 decrease of the current density at a fixed potential or the rise of the overpotential at a
297 fixed current density (commonly 10 mA cm^{-2}) can reveal the durability of the catalysts.

298 The less variation of current density or the overpotential marks better longevity. Another
299 method is the accelerated degradation test, which records and compares the cyclic
300 voltammetry (CV) curves or LSV curves before and after thousands of cycles. The shift
301 of the onset overpotential and the overpotential at a constant current density indicates the
302 durability of the catalyst. The smaller shift, the better the stability.

303

304 **3. Transition metal-based electrocatalysts for hydrogen evolution in** 305 **alkaline media**

306 Transition metals have significantly contributed to the development of advanced
307 electrocatalysts because of their high abundance, impressive activities, and easy
308 accessibility. Lately, researchers perform intensive and extensive studies on the design
309 and application of TM-based catalysts for alkaline HER. To our delight, grand progress
310 on both catalyst synthesis and mechanism explorations rapidly push the boundary of the
311 field to a high level. **Table 1** lists the key information (e.g., main catalytic performance
312 and applied strategies) of typical catalysts that discussed in this review, some of which
313 are comparable to the PGMs-based catalysts. Obviously, part of these TM-catalysts are
314 promising alternatives to the noble-based materials and provide new concepts to design
315 cost-efficient electrocatalysts for alkaline HER. Notably, TMs, TM alloys, and TMXs
316 (X= O, S, Se, N, P, C, and B) all exhibit outstanding catalytic activities, while the design
317 principles and underlying mechanisms are different. In this section, the recent
318 achievements of TM-based catalysts will be showcased and classified by the anionic
319 elements.

320

321

Table 1 TM-based electrocatalysts for alkaline HER.

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Category	Electrocatalyst	Key strategy	Electrolyte	η_{10} / mV	Tafel slope/ mV dec ⁻¹
TMs	Ni-rGO/Ni foam ⁴¹	Hybridization	1 M NaOH	36	77
	N,P-doped Ni ⁴²	Co-doping	1 M KOH	24	34
	Mn-doped <i>hcp</i> Ni ⁴³	Phase engineering, Doping	1 M KOH	80	68
	W-SAC ⁴⁴	Single-Atom catalysis	0.1 M KOH	85	53
	Ru-NC-700 ⁴⁵	Single-Atom catalysis	0.1 M KOH	47	14
TM alloys	Ni ₃ Fe@N-C	Hybridization;	1 M KOH	72	98
	NT/NFs ²	Hierarchical structure			
	RuCo@NC ⁴⁶	Hybridization	1 M KOH	28	31
TMOs	Ni ₃ S ₂ /VO ₂ CSN ⁴⁷	Hybridization	1 M KOH	100	114
	Ni ₅ P ₄ @NiCo ₂ O ₄ ⁴⁸	Hybridization	1 M KOH	27	27
	NiCo ₂ O ₄ nanosheet ⁴⁹	Facet engineering	1 M KOH	105	62.1
	Reduced NiCo ₂ O ₄ ⁵⁰	Defect engineering	1 M KOH	135	52
	NiCo/Ar-EA ⁵¹	Defect engineering	1 M KOH	317 (η_{360})	132
	Fe- NiCo ₂ O ₄ @HNCP ⁵²	Doping	1 M KOH	124	47
	P-doped β -CoMoO ₄ ⁵³	Doping	1 M KOH	138	68.74
	Co ₂ Mo ₃ O ₈ /Co/NF ⁵⁴	Hybridization; Hierarchical structure	1 M KOH	50	49
TMOHs	β -Ni(OH) ₂ /Pt ⁵⁵	Hybridization	0.1 M KOH	92	42
	NiFeRu-LDH ⁵⁶	Doping	1 M KOH	29	31
	MoS ₂ /NiCo-LDH ⁵⁷	Hybridization	1 M KOH	78	76.6
TMSs	N-doped Ni ₃ S ₂ /NF ⁵⁸	Doping	1 M KOH	155	113
	N-doped Ni ₃ S ₂ /VS ₂ ⁵⁹	Doping	1 M KOH	151	107.5
	Sn-Ni ₃ S ₂ /NF ⁶⁰	Doping	1 M KOH	170 (η_{100})	55.6
	Fe _{17.5%} -Ni ₃ S ₂ /NF ⁶¹	Doping	1 M KOH	47	95

	Co_3S_4 PNS _{vac} ⁶²	Defect engineering	1 M KOH	63	58
	CoMoS ⁶³	Defect engineering	1 M KOH	98	82
	Cu NDs/Ni ₃ S ₂ NTs-CF _S ³²	Hybridization	1 M KOH	128	76.2
	NiCo ₂ S ₄ /Ni ₃ S ₂ /NF ⁶⁴	Hybridization	1 M KOH	119	105.2
	Co-WSe ₂ /MWNTs ⁶⁵	Doping	1 M KOH	241	-
	Ni-MoSe ₂ ⁶⁶	Doping	1 M KOH	206	81
	N-NiSe ₂ ⁶⁷	Doping	1 M KOH	86	69
	P-CoSe ₂ ⁶⁸	Doping	1 M KOH	92	90
	CoSe ₂ -MoSe ₂ /G ⁶⁹	Hybridization	1 M KOH	198	79
	NiCoSe ₂ /CC ⁷⁰	Component optimization	1 M KOH	112.7	65
TMSes	CoNiSe ₂ @CoNi-LDHs/NF ⁷¹	Hybridization	1 M KOH	106	74
	MoSe ₂ -CoSe ₂ ⁷²	Hybridization	1 M KOH	127 (η_0)	89
	c-CoSe ₂ ⁷³	Phase engineering	1 M KOH	190	85
	o-CoSe ₂ P ⁷⁴	Phase engineering; Doping	1 M KOH	104	69
	Ni ₃ Se ₂ /NiSe ⁷⁵	Phase engineering	1 M KOH	168 (η_{100})	72.1
TMTes	Co _{1.11} Te ₂ /C ⁷⁶	Component optimization	1 M KOH	178	77.3
	Mo ₅ N ₆ ⁷⁷	Component optimization	1 M KOH	94	66
	Ni ₃ N@CQDs ⁷⁸	Hybridization	1 M KOH	69	108
TMNs	Ni ₃ N/Ni ⁷⁹	Hybridization	1 M KOH	12	29.3
	NC-NiCu-NiCuN ⁸⁰	Hybridization	1 M KOH	93	55
	Ni ₃ N-CeO ₂ /TM ⁸¹	Hybridization	1 M KOH	80	122
	Ni ₉₀ P ₁₀ /Ti ⁸²	Component optimization	1 M KOH	125	55.7
TMPs	Ni _{0.67} Co _{1.33} P/N-CNFs ⁸³	Component optimization	1 M KOH	130	70
	Ni _{1.8} Cu _{0.2} -P/NF ⁸⁴	Doping	1 M KOH	78	70
	O-NiMoP ₂ /Ni ⁸⁵	Doping	1 M KOH	31	62.11
	N-Co ₂ P/CC ⁸⁶	Doping	1 M KOH	34	51

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					View Article Online DOI: 10.1039/C9TA03220G
	O,Cu-CoP ⁸⁷	Co-doping	1 M KOH	72	57.6
	Ni@Ni ₂ P-Ru ⁸⁸	Hybridization	1 M KOH	31	41
	FeP/Ni ₂ P ⁸⁹	Hybridization	1 M KOH	14	24.2
	Mo ₂ C/G3-NCS750 ²⁵	Porous structure; Doping	1 M KOH	66	37
	N ₃ P-Mo _x C NF ⁹⁰	Co-doping; Hybridization	1 M KOH	135	57.1
TMCs	Mo ₂ N-Mo ₂ C/HGr-3 ⁹¹	Hybridization	1 M KOH	154	68
	CoP/Mo ₂ C-NC ⁹²	Hybridization	1 M KOH	67.2	66
	Mo/Mo ₂ C-HNS-750 ⁹³	Hybridization	1 M KOH	79	62.86
TMBs	Ni _x B/f-MWCNT ⁹⁴	Porous structure; Hybridization	1 M KOH	116	70.4
	Co-B/Ni ⁹⁵	Porous structure	1 M KOH	70	68

322

323 3.1 Transition metals

324 Due to the promising electrocatalytic activities and durability, transition metals have
 325 been widely employed in electrocatalysis, especially HER in basic solutions. To date,
 326 earth-abundant TMs (Ni, Co, W, Fe, etc.) have exhibited excellent performance for HER.
 327 Ni is the most intensively researched TM and miscellaneous approaches have been
 328 engaged to achieve the desirable Ni-based electrocatalysts. For example, diverse Ni-
 329 carbon (reduced graphene oxide (rGO),⁴¹ carbon quantum dot,⁹⁶ carbon nanotube⁹⁷)
 330 composites were fabricated act as high-performance catalysts for alkaline HER. These
 331 sophisticated hybrids take advantage of the intrinsic activity of Ni and the features of
 332 carbon materials with good conductivity, large surface area, easy surface
 333 functionalization, and high stability.⁹⁶ Take Ni-rGO as an example, Wang et al. applied
 334 supergravity electrodeposition to construct the Ni-rGO/NF catalyst. The catalyst showed
 335 great electrocatalytic activities with a low Tafel slope (77 mV dec⁻¹), a small
 336 overpotential (η_{10} = 36 mV), as well as a high exchange current density (j_0 = 3.408 mA

337 cm⁻²).⁴¹ The excellent performance derives from the large specific surface area, improved
338 electrical conductivity, and synergistic effect between Ni particles and rGO sheets.
339 Additionally, DFT calculations suggest that Ni-rGO possessed superior interfacial
340 activities in adsorption/desorption of H* than the pure Ni and rGO sheet.

341

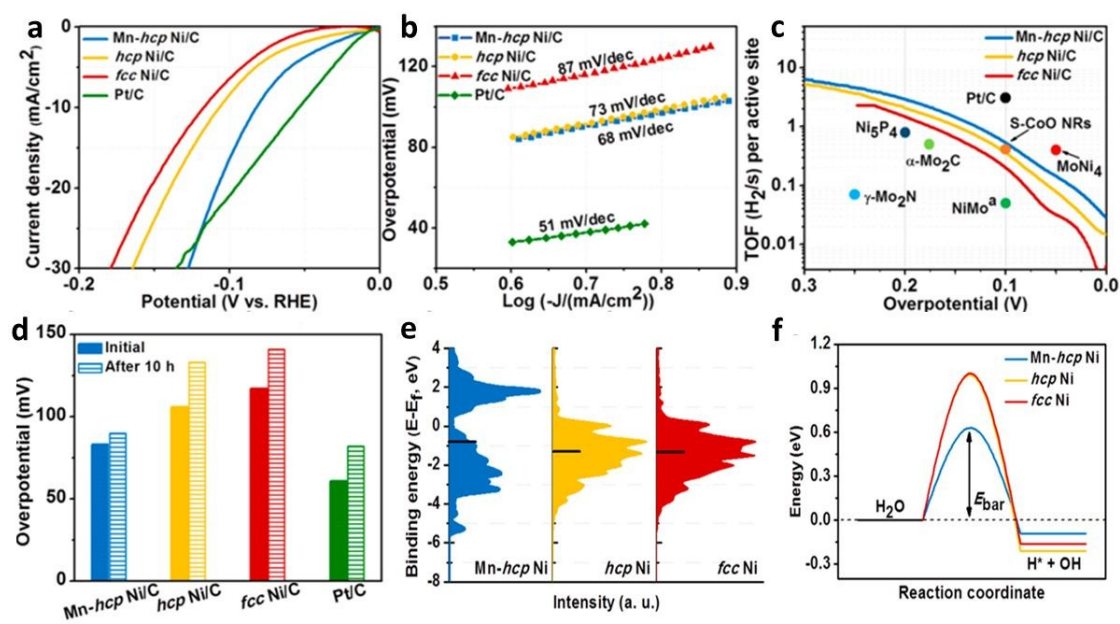
342 Doping transition metals with heteroatoms arises as an efficient strategy to boost the
343 catalytic performance, because doping can regulate the electronic structure of the host
344 metals by tailoring the local charge redistribution without changing the desired intrinsic
345 features of the host elements.⁴² Very recently, Jin et al. found that the nitrogen and
346 phosphorus dual-doped Ni exhibited better electrocatalytic activities than the pure nickel,
347 nickel compounds (Ni₃N, Ni₂P), as well as other doped Ni samples (N-Ni, P-Ni, S-Ni, S-
348 N-Ni, S-P-Ni), with η_{10} =24 mV in alkaline HER.⁴² Experimental results and DFT
349 calculations indicated that the alien atoms could induce charge redistribution on the Ni
350 surface and manipulate the electronic structure of Ni metal in catalysis. In addition,
351 chronoamperometry tests, spectroscopic and TEM measurements suggested that the N-
352 P-Ni was quite stable under the operating condition. This doping scheme was also
353 applicable to Co metal catalysts, thus providing a new strategy for activating transition
354 metal catalysts.

355

356 It is suggested that the exposed crystal phase of metal crystals has a dominant impact on
357 their performance because of the variations in packing density, reactivity, and durability.
358 Shao et al. compared the electrocatalytic properties of hexagonal-close-packed (*hcp*)
359 Ni,⁴³ face-centered cubic (*fcc*) Ni, and Mn-doped *hcp* Ni. The experimental results
360 disclosed that the *hcp* Ni owned a higher HER activity than the *fcc* Ni, and the
361 introduction of Mn further improved the catalytic activity of *hcp* Ni (**Figure 3a-d**).

362 Surprisingly, the Mn-doped *hcp* Ni exhibited the lowest overpotential ($\eta_{10} = 80$ mV) and
 363 Tafel slope (68 mV dec⁻¹), the highest TOF (0.53 H₂ s⁻¹@ $\eta = 100$ mV), as well as the best
 364 stability. Further study indicated that the Mn-*hcp* Ni possessed a higher *d*-band center
 365 and a lower water dissociation energy (Figure 3e-f). As a result, the formed MnO/*hcp*
 366 Ni surface remarkably boosted the HER activity through favourable water adsorption and
 367 fast water dissociation.

368



369

370 **Figure 3.** (a) HER polarization plots and (b) the Tafel slopes of Mn-*hcp* Ni/C, *hcp* Ni/C,
 371 *fcc* Ni/C, and Pt/C. (c) TOF plots of Mn-*hcp* Ni/C, *hcp* Ni/C, and *fcc* Ni/C against
 372 overpotential and TOF values of the commercial Pt/C and the reported electrocatalysts
 373 at specific overpotentials. (d) The comparison of overpotentials at 10 mA cm⁻² before
 374 and after 10 h chronopotentiometry tests of Mn-*hcp* Ni/C, *hcp* Ni/C, *fcc* Ni/C, and Pt/C.
 375 (e) Surface valence band photoemission spectra of Mn-*hcp* Ni, *hcp* Ni, and *fcc* Ni. The
 376 black lines point to the locations of *d*-band center. (f) Calculated energy diagram of water
 377 dissociation on the three metal slabs. Reproduced with permission.⁴³ Copyright 2018,
 378 American Chemical Society.

379

380 Lately, single atom catalysts (SACs) have attracted enormous attention because of their
 381 unique catalytic activities and selectivity, as well as high atomic utilization. SACs are
 382 defined as catalysts that contain singly isolated metal atoms anchored on substrates
 383 without appreciable interactions with other similar atomically dispersed metal atoms.^{45,}

384 ⁹⁸ For example, Chen et al. anchored single W atoms in N-doped carbon matrix from View Article Online
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385 metal-organic framework (MOF) for alkaline HER.⁴⁴ The obtained catalysts exhibited a
386 low η_{10} of 85 mV, a low Tafel slope of 53 mV dec⁻¹ and an extremely high TOF of 6.35
387 H₂ s⁻¹ ($\eta=120$ mV). A computational study revealed that the W₁N₁C₃ moiety owned a
388 low Gibbs free energy of hydrogen adsorption (ΔG_{H^*}) of 0.033 eV, which demonstrated
389 that the W₁N₁C₃ moiety played a crucial role in improving the HER activity. Additionally,
390 the unique structure of W₁N₁C₃ moiety possesses high stability, because of the strong
391 interaction between the W atoms and the carbon support. In similar research, Lu and co-
392 workers found that the RuC_xN_y moieties in the Ru and N co-doped carbon materials was
393 the primary active sites in the alkaline HER, while the Ru nanoparticles played a less
394 significant role.⁴⁵ The Ru atoms and C atoms adjacent to the Ru centers possible served
395 as the catalytic centres for the outstanding HER performance. These atomically dispersed
396 metals possess great potential for catalyzing water splitting in alkaline solutions.
397 Currently, the fabrication process of single atom catalysts is still the main obstacle that
398 hinders their wide applications. In this regard, many efficient methods have been
399 developed to fabricate SACs, such as the mass-selected soft-landing technique and the
400 wet-chemistry method.⁹⁸

401

402 **3.2 Transition metal alloys**

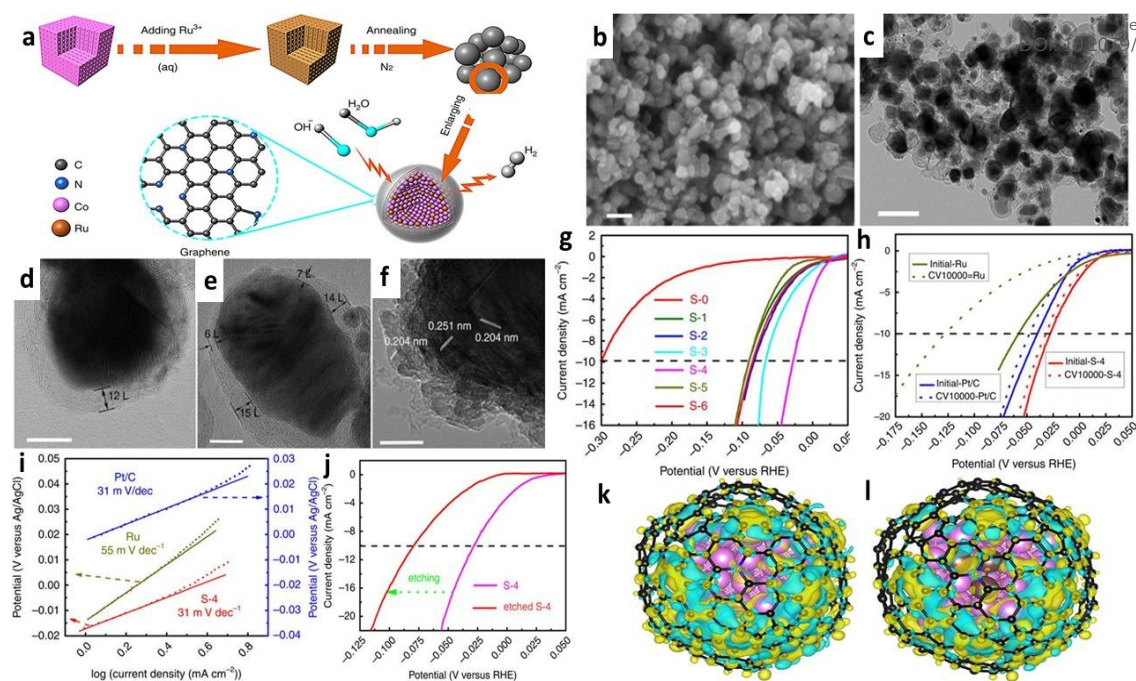
403 Alloying is claimed as a sagacious way to upgrade the catalytic activity and longevity of
404 TMs.^{99, 100} TM alloying can raise the electrocatalytic efficiency by a synergistic
405 combination of electrocatalytic metals or enlarging the ratios of the real and the
406 geometric surface areas.¹⁰⁰ Currently, Ni-based alloys are the most reported TM alloys
407 because of their high catalytic performance in alkaline HER and low price. For instance,
408 Li et al. applied the electrospinning strategy to construct a 1D hierarchical

409 nanoarchitecture comprising Ni₃Fe nanoalloy-encapsulated carbon nanotubes grown
410 onto N-doped carbon nanofibers (Ni₃Fe@N-C NT/NFs).² The immobilized Ni₃Fe
411 nanoparticles worked as catalytic sites and the N-doped carbon nanostructures facilitated
412 the electron transport and mass diffusion. Moreover, the Ni₃Fe nanoalloys encapsulated
413 in carbon nanotubes could remarkably optimize the electron structure of the surrounding
414 carbons because of the electron-penetration effect, thus elevating the electrocatalytic
415 activity. Compared with Ni-based and Fe₂O₃-based catalysts, i.e., Ni@N-C NT/NFs and
416 Fe₂O₃@N-CNFs, Ni₃Fe@N-C NT/NFs possessed the lowest η_{10} of 72 mV, the lowest
417 Tafel slope of 98 mV dec⁻¹, as well as the highest ECSA of 407.5 cm². Additionally, the
418 computational results revealed that the Ni₃Fe@N-C NT/NFs attained a favourable ΔG_{H^*}
419 of -0.14 eV, thus promoting the HER kinetics. Similarly, Ni-Cu alloys,^{6, 101} Ni-Mo
420 alloys,¹⁰²⁻¹⁰⁴ and Co-based alloys^{46, 105} also exhibited decent catalytic capabilities for
421 alkaline HER.

422
423 Apart from those earth abundant TMs, noble metal-based alloys have enticed intensive
424 attention recently due to their distinctive catalytic behaviors. Alloying noble metals with
425 other TMs (e.g., Ni, Fe, Co) to obtain the multicomponent electrocatalysts is a smart
426 strategy to increase the atomic utilization efficiency of noble metals and decrease the
427 consumption of noble metals by up to an order of magnitude.^{46, 106} More importantly,
428 noble-transition alloys significantly enhance catalytic capability because of the
429 redistribution of charge and tailored surface properties during the formation process of
430 alloys.^{46, 107} Lately, Ru, a noble metal cheaper than Pt, Pd, and Ir, evoking intense
431 excitement to the researchers for its high catalytic activities for HER and relatively low
432 price. For example, Su and co-workers developed a high-performance electrocatalyst
433 composed of Ru-Co nanoalloy encapsulated in nitrogen-doped graphene layers.⁴⁶ The

434 synthetic route is presented in **Figure 4a**. The FESEM results indicated that the obtained
435 catalysts possessed an irregular morphology (**Figure 4b**). TEM and high-resolution TEM
436 images showed that the alloy particles were coated with thin N-doped graphene shells
437 which were beneficial to enhancing the catalytic activity (**Figure 4c-f**). The sample S4
438 (3.58 wt. % Ru in RuCo alloy) exhibited the best HER ability in 1 M KOH compared to
439 other samples with different Ru contents. Specifically, S4 owned the smallest
440 overpotential ($\eta_{10} = 28$ mV) and Tafel slope (31 mV dec^{-1}), as well as the best durability
441 (**Figure 4g-i**), which is better than the benchmark Pt/C (20 wt.%) catalysts. In addition,
442 the results of the etching test indicated that the metallic Co played an indispensable role
443 in the outstanding HER performance in alkaline media (**Figure 4j**). Further
444 computational results suggested that the number of electrons transferred from the alloy
445 core to the graphene shell was higher than that of the metal core, which implied lower
446 ΔG_{H^*} and thus better catalytic activities (**Figure 4k and l**). Furthermore, the frequently
447 documented Pt-based alloys also present appealing catalytic capability for alkaline
448 HER.^{106, 108-111} These results declared that alloying noble metals with TMs are an
449 effective and economic strategy for fabricating high-performance HER electrocatalysts
450 with a lower cost.

451



452

453 **Figure 4.** (a) Schematic illustration of the synthetic route and model of the RuCo
 454 nanoalloys encapsulated in nitrogen-doped graphene layers. (b, c) The FESEM and TEM
 455 images of S-4. Scale bars, 100 nm. (d-f) HRTEM images of S-4. Scale bars, 10 nm. (g)
 456 HER polarization curves of RuCo@NC samples with the same mass loading. (h)
 457 HER polarization curves of S-4, Ru and Pt/C with the same mass loading and durability test
 458 after 10,000th cycles. (i) Tafel plots of S-4, Ru and Pt/C. (j) HER polarization curve of
 459 etched S-4 by 1M HCl. (k, l) Calculated charge-density differences between Co and
 460 Co₃Ru models, respectively. The isosurface value of the color region is 0.01 eÅ⁻³. The
 461 yellow and cyan regions refer to increased and decreased charge distributions,
 462 respectively. Reproduced with permission.⁴⁶ Copyright 2018, Nature Publishing Group.
 463

464 The electrochemical stability of catalysts is a key criterion that determines the
 465 practicability of a designed electrocatalyst in industrial applications. The oxidation and
 466 corrosion of TM alloys has been a concern in the alkaline HER process, whilst a vast
 467 number of the TM electrocatalysts have been reported to deliver a great stability. Apart
 468 from the conventional chronopotentiometry and chronoamperometry methods
 469 introduced before, other techniques are also applied to evaluating the durability of
 470 catalysts, including X-ray diffraction (XRD), X-ray photoelectron spectra (XPS),
 471 transmission electron microscopy (TEM), scanning electron microscopy (SEM), and
 472 Fourier-transform infrared spectroscopy (FTIR).^{102, 112-117} These tests along with the
 473 electrochemical measurements can provide comprehensive information (e.g., catalytic

474 activities, components, morphology, surface properties) of the materials before and after
475 durability testing. For example, Ahn et al. investigated the stability of Ni-Co-Fe alloys
476 derived from the Prussian blue analog.¹¹² Firstly, chronopotentiometry tests were
477 conducted and the catalysts showed an initial HER potential of -0.34V vs. RHE, and a
478 voltage loss of merely 0.05V after 20000s. Then, TEM characterization was further
479 performed to verify the structural durability of the ternary alloy. The main nanocuboid
480 morphology was maintained during the HER tests, although the edges were slightly
481 roughened because of the formation of metal hydroxides. The favourable electrochemical
482 stability is mainly due to the particularly rigid porous structure of the robust MOF, and
483 the unique architecture facilitates the gas bubbling and retains the framework.

484

485 3.3 Oxygen group transition metal catalysts

486 3.3.1 Transition metal oxides

487 Electrocatalysts based on transition metal oxides (TMOs) have gained enormous interests
488 due to their low toxicity, earth-abundance, low cost, and high activities.¹¹⁸ Previously,
489 pure TMOs exhibit superb potentials for OER yet poor reactivity toward HER because
490 of their unappealing hydrogen desorption ability.^{118, 119} Encouragingly, great efforts have
491 been made to shape TMOs into appropriate catalysts for alkaline HER, affording the
492 engineered TMOs good candidates for overall water splitting.

493

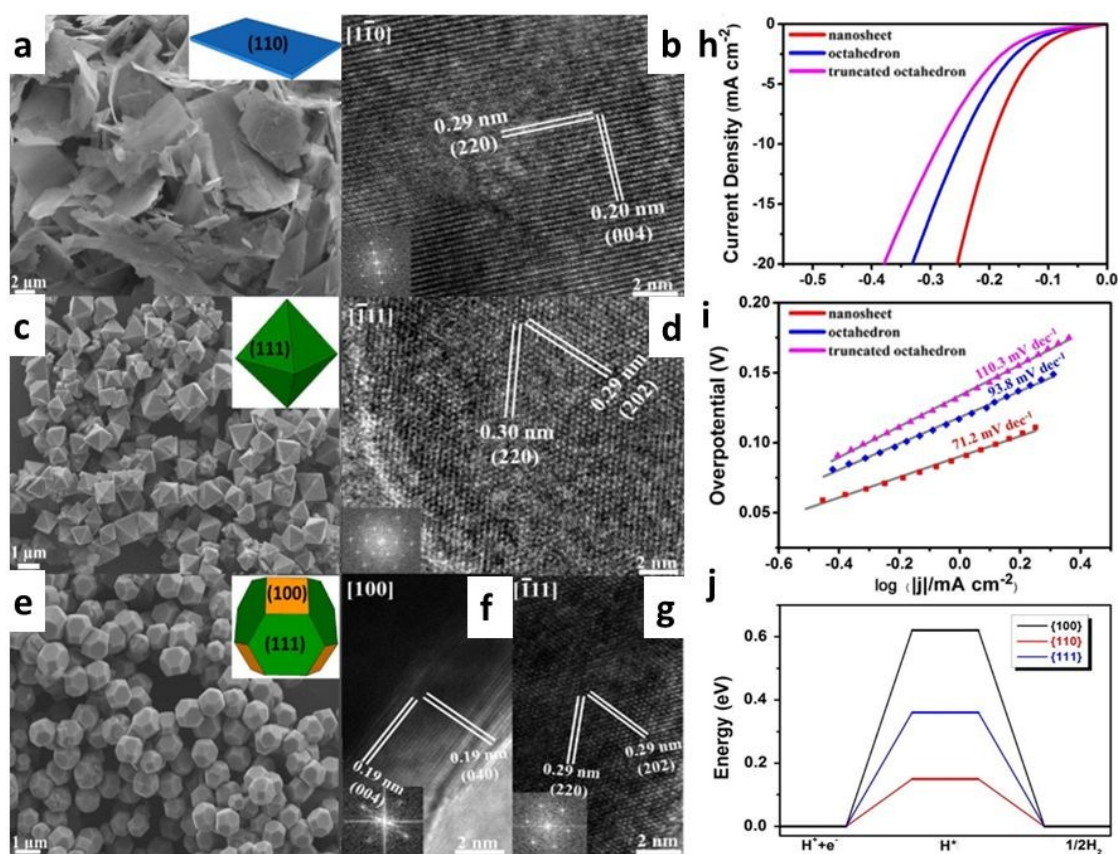
494 Hitherto, CoO_x ,¹¹⁹⁻¹²¹ VO_2 ,⁴⁷ MnO_2 ,^{21, 122} and NiO ,^{118, 123} are representatives that possess
495 high catalytic performance after modifications. Hybridizing TMOs with other
496 electroactive materials (e.g., TMOs,¹²⁴⁻¹²⁶ TMPs,^{48, 127} TMs,^{123, 128, 129} TM alloys,^{130, 131}
497 TMSs^{132, 133}) is an advisable approach to improve the catalytic performance of bare
498 TMOs. Take VO_2 as an example, compositing VO_2 with Ni_3S_2 could yield an appealing

499 hybrid for overall water splitting, which only took a low potential of 1.42 V to attain a
500 current density of 10 mA cm⁻².⁴⁷ The astonishing catalytic activity originated from the
501 interfaces which gave rise to a low-energy-level *d* band center and a low Gibbs free
502 energy. Lately, Zhang and Co-workers constructed nanometric Ni₅P₄ clusters on
503 NiCo₂O₄ via a phosphating process.⁴⁸ The Ni₅P₄@NiCo₂O₄ exhibited superior catalytic
504 performance than Ni₅P₄ and NiCo₂O₄, with a low Tafel slope of 27 mV dec⁻¹ and an
505 overpotential of 27 mV at 10 mA cm⁻². DFT calculations implied that NiCo₂O₄
506 significantly improved the water dissociation step, and Ni₅P₄ facilitated the hydrogen
507 adsorption and desorption process. Additionally, the presence of Ni₅P₄ promoted the
508 electron transfer within the NiCo₂O₄ nanoflakes, which further ameliorated the overall
509 electrochemical reactivity. The hybridization of Ni₅P₄ and NiCo₂O₄ thus delivered an
510 excellent HER activity in alkaline electrolytes.

511
512 Bimetallic oxides of M₁M₂O_x (M₁, M₂: transition metal) also manifest great potentials
513 for alkaline HER. In the following part, we mainly focus on the application of two series
514 of typical oxides, i.e., spinel TMOs and MMoO₄. Spinel bimetallic TMOs (e.g., NiCo₂O₄)
515 have a more flexible redox property, and higher conductivity than the corresponding
516 individual metal oxides, herein manifesting a better catalytic activity in
517 electrochemistry.⁴⁸ However, the electrocatalytic ability of such pristine TMOs in HER
518 is still unsatisfactory. To break the bottleneck, several approaches have been developed,
519 including doping,^{52, 134, 135} defect (oxygen vacancy) engineering,^{50, 51} crystal-plane
520 engineering,⁴⁹ compositing^{48, 124, 126, 136} and hierarchical structuring.¹³⁷ Recently, Fang et
521 al. fabricated NiCo₂O₄ nanocrystals exposing various crystal planes, including NiCo₂O₄
522 nanosheet exposing (1 1 0) crystal planes,⁴⁹ NiCo₂O₄ octahedron exposing (1 1 1) crystal
523 planes and NiCo₂O₄ truncated octahedron exposing (1 1 1) and (1 0 0) crystal planes

524 (Figure 5a-g). The experimental results suggested that NiCo₂O₄ nanosheets exhibited
 525 the best catalytic activity for alkaline HER, followed by NiCo₂O₄ octahedron and
 526 NiCo₂O₄ truncated octahedron (Figure 5h and i). Moreover, density functional theory
 527 (DFT) calculations revealed that the (1 1 0) crystal planes have the lowest ΔG_{H^*} of 0.15
 528 eV relative to (1 0 0) (0.62 eV) and (1 1 1) (0.36 eV) planes, implying that the (1 1 0)
 529 surface was more active than (1 0 0) and (1 1 1) planes for HER (Figure 5j). These
 530 fruitful findings disclose that selectively exposing favourable grain boundaries of
 531 electrocatalysts the can achieve high activities in HER.

532



533

534 **Figure 5.** (a- b) The SEM and HRTEM images of the NiCo₂O₄ nanosheet exposing (1 1
 535 0) crystal planes. (c- d) The SEM and HRTEM images of NiCo₂O₄ octahedron exposing
 536 (1 1 1) crystal planes. (e- g) The SEM and HRTEM images of NiCo₂O₄ truncated
 537 octahedron exposing (1 1 1) and (1 0 0) crystal planes. (h) Polarization curves and (i) the
 538 corresponding Tafel slopes of the NiCo₂O₄ nanosheet, NiCo₂O₄ octahedron, NiCo₂O₄
 539 truncated octahedron. (j) The HER free energy change diagram of NiCo₂O₄ (1 0 0), (1 1
 540 0) and (1 1 1) crystal planes. Inset images in (a, c, e) are the models of crystals.

541 images in (b, d, f, g) are the corresponding fast Fourier transform patterns. Reproduced
542 with permission.⁴⁹ Copyright 2018, Elsevier Ltd. View Article Online
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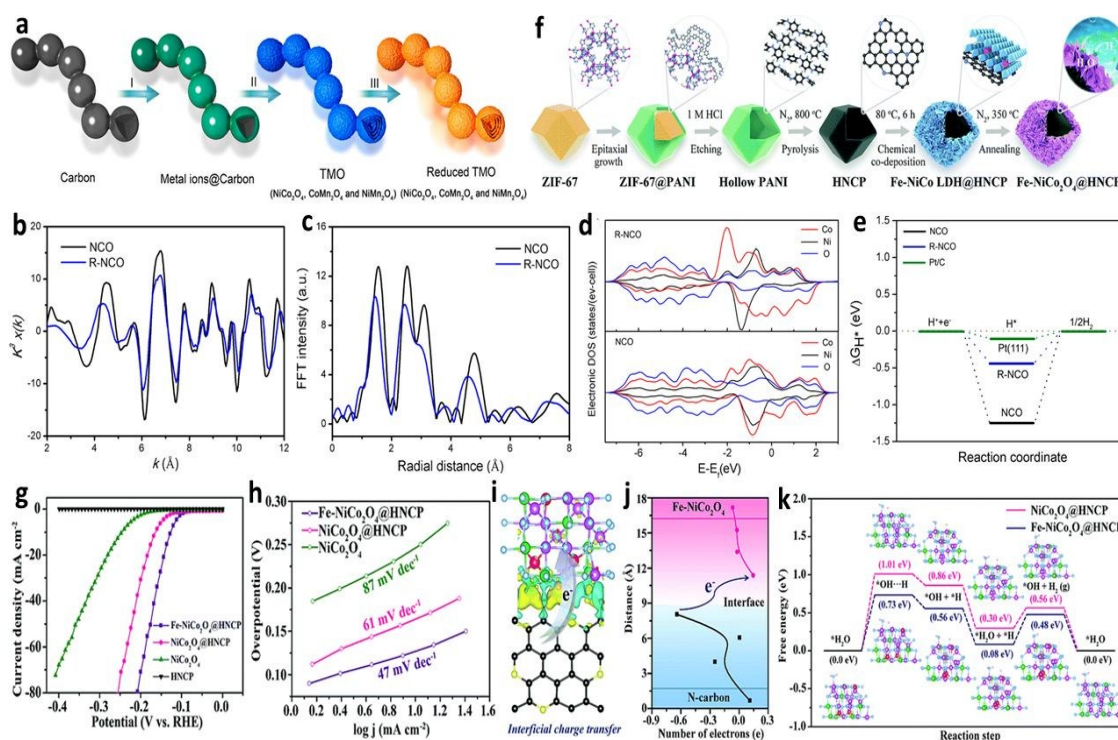
544 Oxygen vacancy engineering is another efficient method to simultaneously modulate the
545 electronic structure, conductivity, and the reactive species' adsorption energy.^{50, 51, 118} As
546 a result, the presence of oxygen vacancies can profoundly boost intrinsic catalytic
547 activities. Lately, Feng et al. developed an "adsorption-calcination-reduction" strategy to
548 fabricate a series of spinel TMOs (NiCo₂O₄, CoMn₂O₄, and NiMn₂O₄) (**Figure 6a**).⁵⁰
549 The XAFS analysis revealed the existence of abundant oxygen vacancies in the reduced
550 NiCo₂O₄, leading to the decline in Co coordination numbers (**Figure 6b** and **c**).
551 Compared to unmodified NiCo₂O₄, the reduced sample possessed a lower overpotential
552 (135 vs. 236 mV, η_{10}) and a smaller Tafel slope (52 vs. 95 mV dec⁻¹). Furthermore, the
553 projected density of states (PDOS) of Co *d* orbital in the reduced NiCo₂O₄ shifted to the
554 low-energy direction. This leftward shift indicated that the distribution of electrons in the
555 *d* band drifted away from the Fermi level (**Figure 6d**), indicating that the catalyst attained
556 weakened chemical bonding with intermediates with lower adsorption energy. These
557 findings were further verified by the calculated ΔG_{H^*} (**Figure 6e**). In another study, Liu
558 et al. found that the oxygen vacancy content had a prominent effect on the catalytic
559 capability of NiCo₂O₄.⁵¹ Specifically, increasing the oxygen vacancy concentration
560 would decrease the adsorption energy and the dissociation energy barrier of H₂O
561 molecules on catalyst surfaces, thus improving the catalytic ability in alkaline HER.

562

563 Doping is a conventional strategy to upgrade the activity of catalysts by increasing active
564 sites, optimizing the electronic structure, inducing phase transformation, and regulating
565 reaction energy barrier, etc. Very recently, Lai et al. investigated the catalytic
566 performance of TM-doped NiCo₂O₄ (TM= Fe, Co, Ni, Zn, Mn, Cu),⁵² and the preparation

567 route is depicted in **Figure 6f**. The alkaline HER activity of Fe-doped NiCo_2O_4 was
 568 higher than that of pristine NiCo_2O_4 , with a low η_{10} of 124 mV and a small Tafel slope
 569 of 47 mV dec^{-1} (**Figure 6g and h**), as well as a high TOF value of 0.39 s^{-1} at an
 570 overpotential of 200 mV. The DFT analysis indicated that the electrons transferred from
 571 the N-doped carbon polyhedron (HNCP) to the partially charge-delocalized Fe- NiCo_2O_4 ,
 572 which increased the electronic states near the Fermi level of Fe/Co/Ni *d* orbitals (**Figure**
 573 **6i and j**). This charge transfer process improves the overall properties of catalysts.
 574 Compared with NiCo_2O_4 @HNCP, the obtained energy barriers of Fe-doped
 575 NiCo_2O_4 @HNCP revealed the improved thermodynamic and kinetic performance
 576 (**Figure 6k**). Moreover, Cu, Mn, and Zn doped NiCo_2O_4 also exhibited higher catalytic
 577 activities than the pristine one.

578



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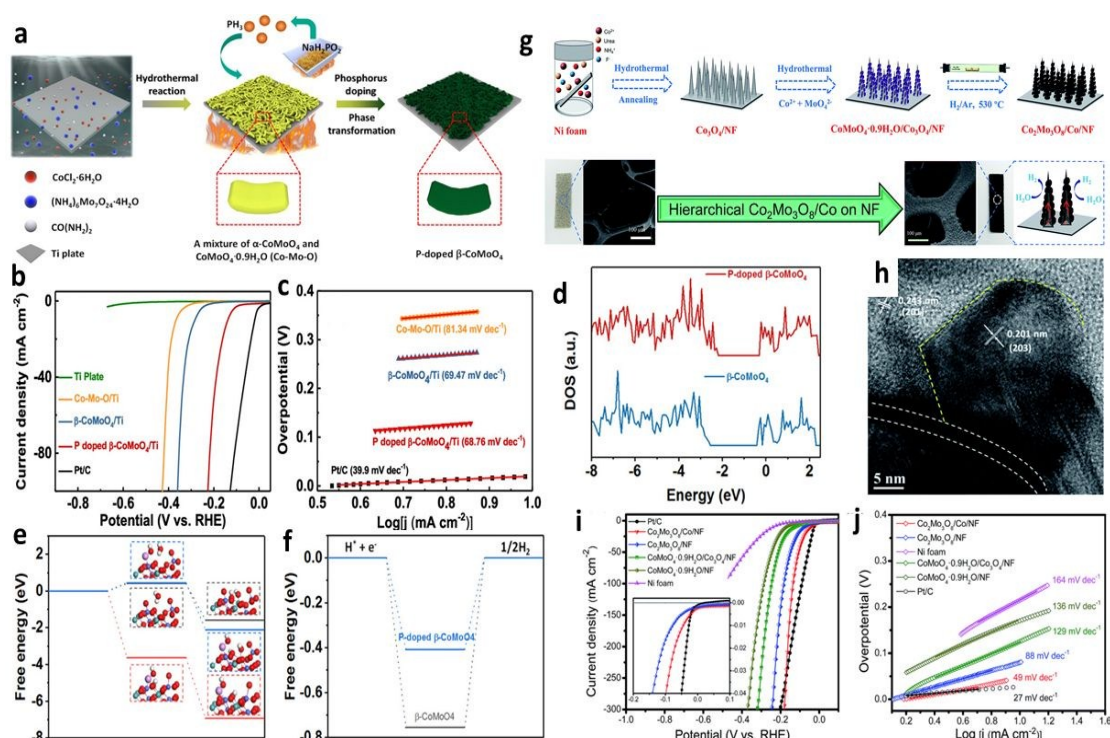
580 **Figure 6.** (a) Schematic illustration of the formation process of R-TMO with a necklace-
 581 like multishelled hollow structure: (I) The absorption of metal ions on the carbon, (II)
 582 calcination of the absorbed carbon, and (III) reduction of the TMO to obtain R-TMO
 583 with a necklace-like multishelled hollow structure. (b) Co K-edge EXAFS data. (c) The
 584 corresponding k^3 -weighted Fourier-transformed data of pristine NiCo_2O_4 and R-

585 NiCo₂O₄. (d) Calculated DOS curves for pristine NiCo₂O₄ and R-NiCo₂O₄. (e)
586 Calculated free energy diagram of the HER on pristine NiCo₂O₄ and R-NiCo₂O₄. (a- e).
587 Reproduced with permission.⁵⁰ Copyright 2018, American Chemical Society. (f)
588 Schematic illustration of the synthesis process of Fe-NiCo₂O₄@HNCP. (g) Polarization
589 curves and (h) Tafel curves of Fe-NiCo₂O₄@HNCP and the compared samples in 1.0 M
590 KOH solution for HER with a scan rate of 5 mV s⁻¹. (i) Charge density distribution of
591 Fe-NiCo₂O₄@HNCP around the interface. (j) Bader charge analysis of average atoms
592 near the interface of N-carbon and Fe-NiCo₂O₄. (k) Calculated free energy diagram of
593 HER on Fe-NiCo₂O₄, and NiCo₂O₄. (f- k). Reproduced with permission.⁵² Copyright
594 2019, The Royal Society of Chemistry.
595

596 Mo-based materials are promising HER catalysts among which transition metal
597 molybdates (MMoO₄) have been demonstrated as efficient electrocatalysts for alkaline
598 HER due to their high activities and adjustable electronic structures by the synergy of
599 Mo and M.⁵⁴ Lately, Li and co-workers investigated the catalytic properties of P-doped
600 β-CoMoO₄ prepared by a two-step strategy (**Figure 7a**).⁵³ The doped sample showed a
601 better performance than β-CoMoO₄ and Co-Mo-O, with a small overpotential ($\eta_{10} = 138$
602 mV) and a low Tafel slope (68.76 mV dec⁻¹) (**Figure 7b** and **c**). The density of states
603 (DOS) of different samples indicated that the P doping narrowed the bandgap of β-
604 CoMoO₄ from 2.09 eV to 1.87 eV, introduced more charge carriers and ameliorated the
605 electrical conductivity (**Figure 7d**). Furthermore, DFT results indicated that P-doped β-
606 CoMoO₄ owned a similar energy barrier in the Volmer step ($\Delta G(\text{H}_2\text{O}) = 0.45$ eV) to pure
607 β-CoMoO₄ ($\Delta G(\text{H}_2\text{O}) = 0.41$ eV), while the ΔG_{H^*} of the P-doped CoMoO₄ was quite
608 lower than the pristine one (0.41 eV vs. 0.76 eV) (**Figure 7e** and **f**). Hence, the optimized
609 electrical conductivity and hydrogen adsorption free energy resulted in the improved
610 catalytic activity. Interestingly, Ou et al. used CoMoO₄•0.9H₂O as a precursor to produce
611 Co₂Mo₃O₈ (**Figure 7g**).⁵⁴ The prepared hierarchical structures are composed of internal
612 Co nanowires and Co₂Mo₃O₈ outer layers (**Figure 7h**). Compared with the pristine
613 CoMoO₄•0.9H₂O, Co₂Mo₃O₈ demonstrated a better catalytic activity (**Figure 7i** and **j**).
614 Furthermore, the authors declared that the bimetallic suboxides with lower-valence Mo

615 species (Mo^{4+} , Mo^{5+}) facilitated the water adsorption and intermediate formation
 616 processes. This approach may be beneficial to the design of desirable electrocatalysts of
 617 bimetallic suboxides reduced from the corresponding oxide precursors.

618



619

620 **Figure 7.** (a) Schematic illustration of the synthesis route of P-doped $\beta\text{-CoMoO}_4$ on a Ti
 621 plate. (b) Linear sweep voltammetry (LSV) polarization curves of Ti plate, Co-Mo-O/Ti,
 622 $\beta\text{-CoMoO}_4/\text{Ti}$, P-doped $\beta\text{-CoMoO}_4/\text{Ti}$, and Pt/C. (c) Tafel plots for Co-Mo-O/Ti, $\beta\text{-CoMoO}_4/\text{Ti}$,
 623 P-doped $\beta\text{-CoMoO}_4/\text{Ti}$, and Pt/C. (d) Total density of states of $\beta\text{-CoMoO}_4$
 624 and P-doped $\beta\text{-CoMoO}_4$. The Fermi level is set at 0 eV. (e) Calculated free energy
 625 diagram for the Volmer step on the original P-doped $\beta\text{-CoMoO}_4$ model (red line), revised
 626 P-doped $\beta\text{-CoMoO}_4$ model (grey line), and $\beta\text{-CoMoO}_4$ model (blue line). (f) Calculated
 627 free energy diagram for the Tafel step. (a- f). Reproduced with permission.⁵³ Copyright
 628 2018, American Chemical Society. (g) Schematic illustration of the process for
 629 fabricating the $\text{CoMoO}_4 \cdot 0.9\text{H}_2\text{O}/\text{Co}_3\text{O}_4$ composite on Ni foam and reducing it to
 630 $\text{Co}_2\text{Mo}_3\text{O}_8/\text{Co}/\text{NF}$ under a H_2/Ar atmosphere. (h) The HRTEM image of $\text{Co}_2\text{Mo}_3\text{O}_8/\text{Co}$.
 631 The white dash shows the outline of the tip region of a Co nanowire and the yellow dash
 632 shows the outline of a $\text{Co}_2\text{Mo}_3\text{O}_8$ nanosheet. (i) LSV curves and (j) Tafel plots of
 633 $\text{Co}_2\text{Mo}_3\text{O}_8/\text{Co}/\text{NF}$, $\text{Co}_2\text{Mo}_3\text{O}_8/\text{NF}$, $\text{CoMoO}_4 \cdot 0.9\text{H}_2\text{O}/\text{Co}_3\text{O}_4/\text{NF}$, $\text{Co}_2\text{Mo}_3\text{O}_8/\text{NF}$, Ni
 634 foam, and Pt/C. (g- j). Reproduced with permission.⁵⁴ Copyright 2018, The Royal
 635 Society of Chemistry.

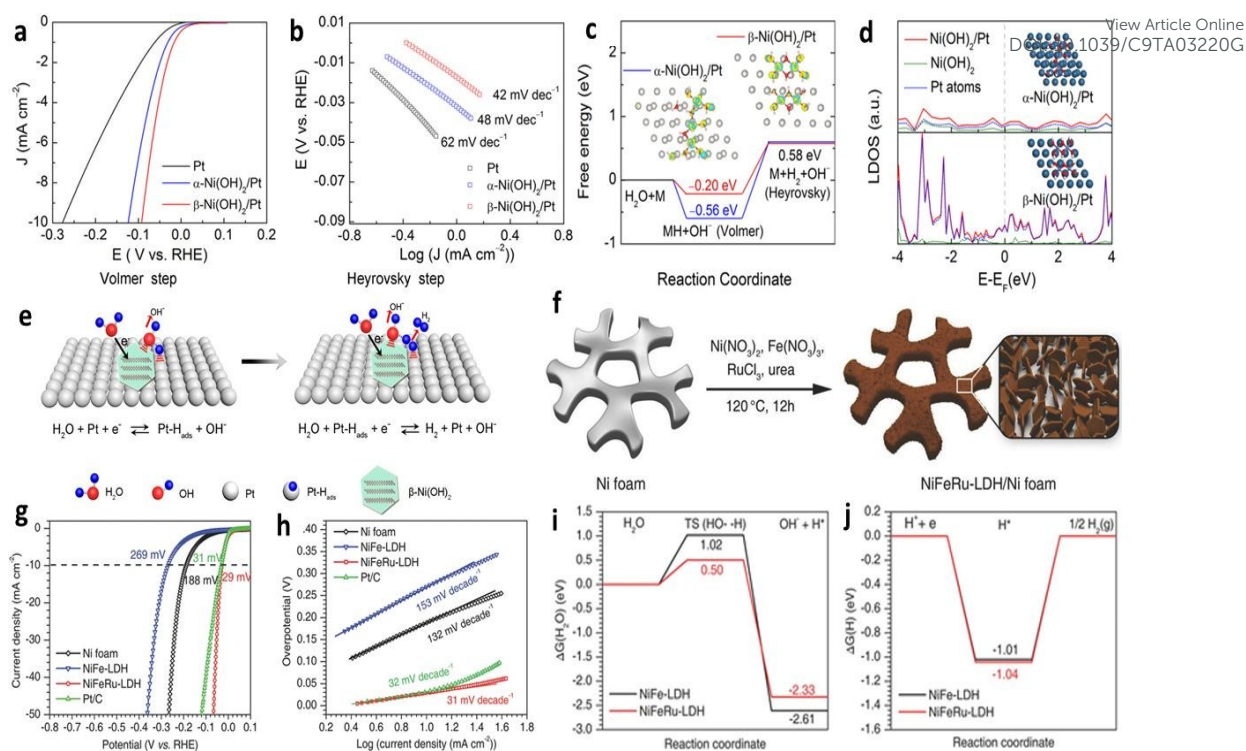
636

637 3.3.2 Transition metal (oxy)hydroxides

638 Transition metal (oxy)hydroxides (TMOHs), including the special class of layered
639 double hydroxide (LDH) materials, possess layered structures with high specific surface
640 area, distinctive electron distribution, outstanding catalytic activity and stability in
641 alkaline solutions, as well as low cost.^{35, 138} Pristine TMOHs own excellent activities in
642 OER, which are comparable to TMOs. Moreover, TMOHs are also great candidates for
643 alkaline HER because they can effectively adsorb hydroxyl species and subsequently
644 catalyze their dissociation.⁵⁷ Therefore, TMOHs can largely facilitate the Volmer step.
645 On the other hand, the following hydrogen evolution step (Heyrovsky step or Tafel step)
646 requires another active substance for the adsorption and recombination of reactive
647 hydrogen intermediates.^{55, 139} Therefore, hybridizing TMOHs with those conventional
648 HER electrocatalysts (X) is an efficient approach to obtain high-performance catalysts
649 for alkaline HER. Successful applications “TMOHs + X” hybrids have been extensively
650 reported, and several typical samples are detailed in this part.

651
652 Combining TMOHs with noble metals can fabricate high-activity catalysts. The derived
653 composites can take advantage of high catalytic activities of noble metals and favourable
654 water dissociation ability of TMOHs. To date, Pt-Ni(OH)₂,^{55, 140-142} Ni(OH)₂-PtO₂,¹⁴³ Pt-
655 Co(OH)₂,¹⁴⁴ Co(OH)₂-Au-Ni(OH)₂,¹⁴⁵ NiFeRu-LDH,⁵⁶ PtO₂-CoOOH,¹⁴⁶ etc. have been
656 studied as excellent catalysts in alkaline HER. Lately, Yu et al. discussed the influence
657 of crystal structure (α - and β -Ni(OH)₂) on the performance of Ni(OH)₂/Pt.⁵⁵ The
658 experimental results indicated that β -Ni(OH)₂/Pt exhibited the best catalytic activity,
659 followed by α -Ni(OH)₂/Pt and Pt (**Figure 8a and b**). Further study indicated that the
660 adsorption of H₂O on the β -Ni(OH)₂/Pt surface was more favourable than that on the α -
661 Ni(OH)₂/Pt surface, declaring the stronger hydrogen bonding interactions between β -
662 Ni(OH)₂ and water molecules. The calculated free energies of β -Ni(OH)₂/Pt were lower

663 than those of α -Ni(OH)₂/Pt, revealing more feasible Volmer and Heyrovsky steps at the
664 β -Ni(OH)₂/Pt surface. Moreover, the charge difference isosurface of the α - or β -
665 Ni(OH)₂/Pt surface disclosed noticeable charge transfers between Pt and α - or β -Ni(OH)₂
666 (**Figure 8c**). The local density of states (LDOS) of β -Ni(OH)₂/Pt exhibited more
667 significant peaks, confirming β -Ni(OH)₂/Pt was more active than the α -Ni(OH)₂/Pt
668 (**Figure 8d**). The high activity of the β -Ni(OH)₂/Pt hybrid mostly aroused from the high
669 edging catalytic activities of β -Ni(OH)₂ and its strong interaction with Pt substrate that
670 significantly improved the catalytic activity of Pt for alkaline HER.⁵⁵ Shortly, the better
671 performance of β -Ni(OH)₂/Pt results from the coupling effect with optimized electron
672 configures toward favourable binding with H₂O molecules (**Figure 8e**). Using noble
673 metals as heteroatom dopants can further tailor the activity of LDH. Take Ru as an
674 example, Chen et al. designed a Ru-doped NiFe-LDH with a one-pot hydrothermal
675 approach (**Figure 8f**).⁵⁶ The as-prepared NiFeRu-LDH sample displayed higher catalytic
676 activities than the pristine NiFe-LDH and Pt/C. Surprisingly, the low overpotential (η_{10}
677 = 29 mV) and Tafel slope (31 mV dec⁻¹) were achieved which surpassed most of the
678 reported catalysts (**Figure 8g and h**). Theoretical calculations demonstrated that the
679 water dissociation energy barrier of NiFeRu-LDH was quite lower than that of NiFe-
680 LDH (0.50 eV vs. 1.02 eV). Consequently, the presence of Ru highly improved the
681 sluggish Volmer step of NiFe-LDH. However, little difference was found in the hydrogen
682 adsorption free energy, so the Tafel step was merely influenced (**Figure 8i and j**).
683



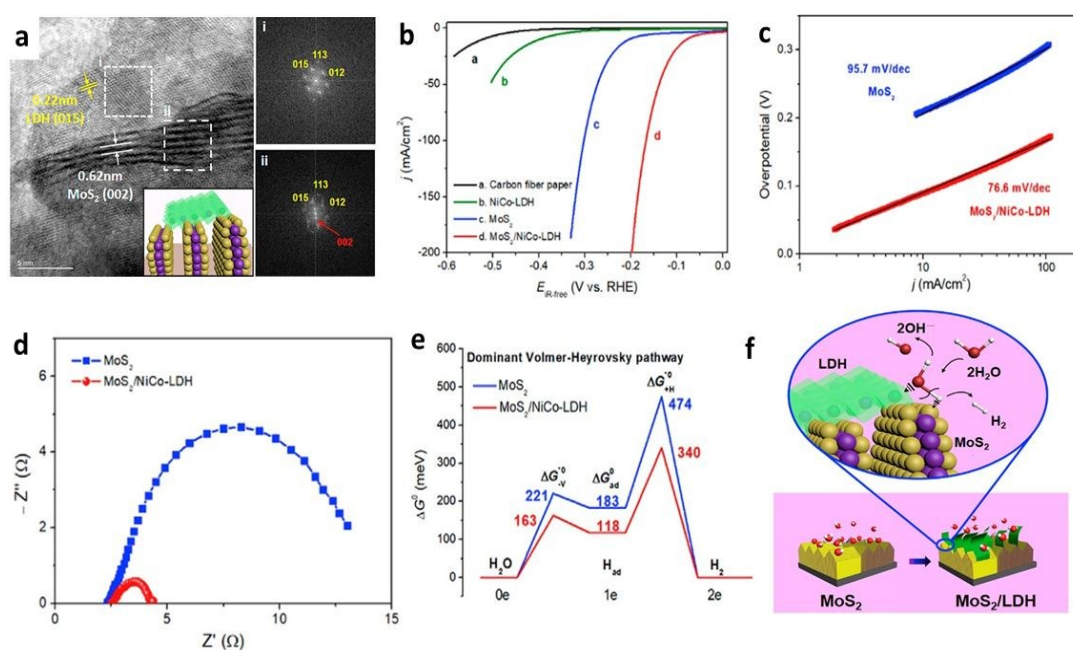
684

685 **Figure 8.** (a) LSV curves and (b) Tafel plots of Pt, α -Ni(OH)₂/Pt, and β -Ni(OH)₂/Pt
 686 electrodes for HER in 0.1 M KOH. (c) Adsorption free energy diagram for the Volmer
 687 and Heyrovsky steps. (d) LDOS of the α - or β -Ni(OH)₂/Pt electrode. (e) Schematic
 688 diagrams of the Ni(OH)₂/Pt electrode (only a β -Ni(OH)₂ sheet is shown as an example)
 689 for Volmer and Heyrovsky steps during HER. (a- e). Reproduced with permission.⁵⁵
 690 Copyright 2018, American Chemical Society. (f) Schematic illustration for in situ growth
 691 of the NiFeRu-LDH on the Ni foam. (g) Polarization curves and (h) corresponding Tafel
 692 plots of the NiFeRu-LDH, NiFe-LDH, nickel foam, and Pt/C electrocatalysts. (i)
 693 Calculated adsorption free energy diagrams for the Volmer step and (j) the Tafel step on
 694 the as-built NiFe-LDH and NiFeRu-LDH models. (f- j). Reproduced with permission.⁵⁶
 695 Copyright 2018, Wiley-VCH.

697 Integrating TMOHs and TM-based active compounds such as transition metal sulfides
 698 (TMSs) can produce high-performance and low-cost catalysts. For example, Hu et al.
 699 designed a MoS₂/NiCo-LDH hybrid by a two-step hydrothermal process.⁵⁷ The HRTEM
 700 image revealed the co-existence of MoS₂ and NiCo-LDH, as well as the interface
 701 between the (015) facet of NiCo-LDH and the neighboring (002) facet of MoS₂ (**Figure**
 702 **9a**). The hybrids exhibited better catalytic activities than sole MoS₂ and NiCo-LDH,
 703 suggesting the synergistic effect between MoS₂ and NiCo-LDH (**Figure 9b** and **c**). The
 704 enhanced alkaline HER kinetics was also implied by the electrochemical impedance

705 spectroscopy (EIS) that the hybrids owned a quite lower charge-transfer resistance than
 706 MoS₂ (1.7 Ω vs. 11.2 Ω, $\eta = 200$ mV) (**Figure 9d**). DFT calculations demonstrated the
 707 activation energies for different steps of MoS₂/NiCo-LDH were all lower than those of
 708 the bare MoS₂. Accordingly, hybridizing of MoS₂ with NiCo-LDH can facilitate the
 709 overall HER process through synergistic chemisorption of H on MoS₂ and OH on LDH
 710 (**Figure 9e and f**). Apart from MoS₂/LDHs, MoS₂-CoOOH,¹⁴⁷ WS₂,¹⁴⁸ NiS₂/Ni(OH)₂,¹⁴⁹
 711 Ni₃S₂/NiOOH,¹⁵⁰ and Ni(OH)₂-CoS₂¹⁵¹ are also appealing catalysts which realised
 712 promoted activities in alkaline HER. Apart from the TMSs, electroactive materials with
 713 other anions also exhibit great potentials as co-catalysts, such as CoNiSe₂@CoNi-LDHs,⁷¹
 714 Ni(OH)₂/NiSe₂,¹⁵² Ni(OH)₂-Ni₃N,¹⁵³ Ni(OH)₂-WP,¹⁵⁴ Ni(OH)₂-Ni₂P,¹⁵⁵ NiFe-LDH/CeO_x.¹⁵⁶ The
 715 combination may vary from one to another, but the origin of synergy is similar.
 716 Specifically, the present of TMOHs can optimize the dissociation of water molecules and
 717 concomitant generation of hydrogen intermediates (H_{ad}) which subsequently adsorb on
 718 catalyst surfaces and finally recombine into H₂.

719



720

721 **Figure 9.** (a) HRTEM image of the MoS₂/NiCo-LDH composite. The corresponding
 722 FFT patterns of the selected areas marked by white dashed squares (i and ii) are shown.

723 Inset: schematic illustration of the designed MoS₂/NiCo-LDH heterostructure. (b)
724 Polarization curves of the CFP substrate, bare NiCo-LDH, MoS₂, and MoS₂/NiCo-LDH
725 composite catalysts. (c) Tafel plots for the bare MoS₂ and MoS₂/NiCo-LDH composite
726 catalysts. (d) Nyquist plots of the bare MoS₂ and MoS₂/NiCo-LDH composite catalysts
727 at the overpotential of 200 mV. (e) Free energy diagram of the dominant Volmer-
728 Heyrovsky pathway for HER in the alkaline electrolyte for bare MoS₂ (blue) and
729 MoS₂/NiCo-LDH composite (red) catalysts. (f) Schematic illustration of the HER in
730 MoS₂/LDH interface in an alkaline environment. The synergistic chemisorption of H (on
731 MoS₂) and OH (on LDH) benefits the water dissociation step. Reproduced with
732 permission.⁵⁷ Copyright 2018, Elsevier Ltd.
733

734 3.3.3 Transition metal sulfides

735 Moreover, TMSs themselves are outstanding catalysts for alkaline HER due to the high
736 electrocatalytic activity and natural abundance.³⁷ Researchers hold different opinions on
737 the role of S atoms in TMSs in their excellent catalytic activities. On one hand, catalytic
738 activities was determined by the electronic properties of S atoms. S atoms in TMSs can
739 withdraw electrons from the transition metals due to the high electronegativity, and then
740 sulphur can act as the active sites to stabilise the reaction intermediates.¹⁵⁷ On the other
741 hand, S atoms may play an indirect role in alkaline HER by creating S-vacancies to tailor
742 the electron density of TMs or improving water dissociation via the S^{δ-}-TMⁿ⁺-H₂O
743 network.^{157, 158} Currently, MoS₂, Ni_xS_y, and Co_xS_y are the most reported TMSs, and the
744 fabrication and catalysis of MoS₂ have been extensively reviewed.^{3, 18, 159, 160} Therefore,
745 in this part, we put the emphasis on the alkaline HER application of other TMSs.

746

747 Ni_xS_y (NiS, NiS₂, Ni₃S₂, etc.) manifests good structural stabilities and appealing catalytic
748 activities which exhibits great potentials for alkaline HER, especially the Ni₃S₂. To
749 further elevate the catalytic performance of Ni₃S₂, doping appears as an attractive method.
750 Take nitrogen as an example, Kou et al. prepared an N-doped Ni₃S₂/NF through
751 hydrothermal reaction and ammonia treatment (**Figure 10a**).⁵⁸ Compared to the pristine
752 Ni₃S₂, the modified one exhibited a better alkaline HER ability with a lower overpotential
753 and Tafel slope (**Figure 10b** and **c**). In addition, theoretical analysis implied that the

754 outstanding catalytic activity of N-doped Ni₃S₂ could be attributed to the enriched active sites on the catalyst surface and the favourable ΔG_{H^*} (**Figure 10d**). Interestingly, it was
755 unveiled that the activity of Ni₃S₂ particularly relied on the coordination number of the
756 surface S atoms, as well as the charge depletion of the neighboring Ni atoms. Yu et al.
757 also suggested that N-doping could significantly improve the electrocatalytic activity of
758 Ni₂S₃.¹⁶¹ The N-doped Ni₂S₃ possessed a high surface area, and the great differences in
759 electronegativity between H and N facilitated the adsorption of H. These features
760 significantly upgraded the catalytic performance of N-doped Ni₂S₃, with a lower
761 overpotential compared to Ni₂S₃ (η_{10} = 105 mV vs. 228 mV). Similarly, Zhong and co-
762 workers investigated the electrocatalytic performance of N-doped Ni₃S₂/VS₂.⁵⁹
763 Compared with bare Ni₃S₂/VS₂, the N-doped sample exhibited improved conductivity
764 and larger catalytically active area. Consequently, the N-doped Ni₃S₂/VS₂ exhibited a
765 lower overpotential (η_{10} = 151 mV) and a lower Tafel slope (107.5 mV dec⁻¹) than other
766 samples (e.g., Ni₃S₂/VS₂, N-doped Ni₃S₂, and Ni₃S₂). Briefly, the N-doping method can
767 enrich the catalytic active sites, improve the conductivity and optimize the adsorption
768 energy of reaction intermediates.

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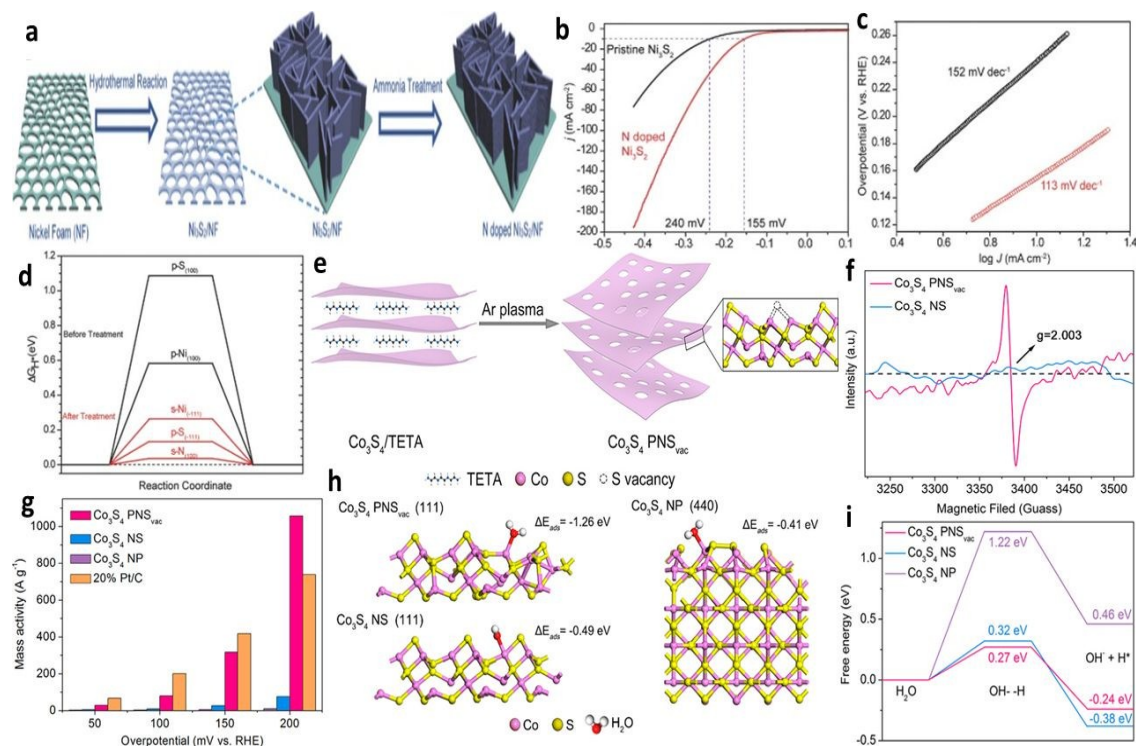
771 Other than anion-doping, introducing metal atoms into Ni₃S₂ frameworks also can
772 improve the electrocatalytic performance. Jian et al. fabricated Sn-doped Ni₃S₂
773 nanosheets by a facile hydrothermal process.⁶⁰ The modified Ni₃S₂ owned a small
774 overpotential (η_{100} = 171 mV) and a low Tafel slope (33.8 mV dec⁻¹). Impressively, the
775 Sn-Ni₃S₂/NF only took a very low potential of 1.46V at 10 mA cm⁻² for the overall water
776 splitting. The Sn doping changed the morphology of pristine Ni₃S₂ and improved the
777 intrinsic catalytic performance. Apart from Sn, other metals (e.g., Fe,⁶¹ V,¹⁶² Mn¹⁶³) also
778 act as effective dopants that can improve the alkaline HER activity of Ni₃S₂. For example,

779 Zhang et al. disclosed the mechanism of for the enhanced electrocatalytic performance
780 of Fe-Ni₃S₂/NF by systematic experiments and DFT calculations.⁶¹ The Fe-doping could
781 enlarge the active surface area, improve the electronic conductivity, boost the water
782 adsorption ability, and optimize the H adsorption energy of Ni₃S₂. These advantages
783 cooperatively benefit the overall electrocatalytic performance of Fe_{17.5%}-Ni₃S₂/NF, with
784 a low Tafel slope of 95 mV dec⁻¹ and a low overpotential of 47 mV at 10 mA cm⁻².

785

786 Defect engineering by constructing S vacancies can further tailor the catalytic
787 performance of TMSs. Theoretically, the presence of vacancies can modify the
788 physicochemical properties and the electronic configuration of TMSs,¹⁶⁴ thus refining
789 the intrinsic activity of catalysts. Zhang and co-workers fabricated non-layered structure
790 of Co₃S₄ ultrathin porous nanosheets with plentiful S vacancies (Co₃S₄ PNS_{vac}) by Ar
791 plasma-induced dry exfoliation (**Figure 10e**).⁶² The obvious electron paramagnetic
792 resonance (EPR) signal with g=2.003 implied the abundant S vacancies (**Figure 10f**).
793 Electrochemical experiments indicated that the mass activity of Co₃S₄ PNS_{vac} was higher
794 than that of pristine Co₃S₄ samples. As a result, the modified Co₃S₄ owned an
795 extraordinarily large mass activity of 1056.6 A g⁻¹ at an overpotential of 200 mV, which
796 was superior to the Pt/C (20 wt.%) electrode (**Figure 10g**). In addition, Co₃S₄ PNS_{vac}
797 possessed a lower water adsorption energy than Co₃S₄, so S vacancies could efficiently
798 expedite the initial step of alkaline HER (**Figure 10h**). The activation energy barrier of
799 water dissociation process for Co₃S₄ PNS_{vac} was also the lowest, which could benefit the
800 formation of catalyst-H_{ads} (**Figure 10i**). Altogether, the kinetics of alkaline HER can be
801 impressively boosted. Moreover, Wu et al. suggested that the rich S vacancies/defect
802 sites in CoMoS layers could enhance the reactive sites for alkaline HER,⁶³ because the
803 catalytic active sites located at the basal edges of catalysts. More relevant information on

804 this topic can be found in the previous reviews by Jia et al.,¹⁶⁵ Yang et al.,¹⁶⁶ and Zhang
 805 et al.¹⁶⁴
 806



807

808 **Figure 10.** (a) Schematic illustration of the synthesis of Ni₃S₂/NF and N-doped Ni₃S₂/NF.
 809 (b) HER polarization curves of the pristine (black) and N-doped Ni₃S₂/NF (red) collected
 810 in 1 M KOH at a scan rate of 5 mV s⁻¹ (dashed lines highlight the potential at 10 mA
 811 cm⁻²). (c) Tafel plots of the pristine and N-doped Ni₃S₂/NF. (d) Reaction energy of H
 812 adsorption, ΔG_{H^*} , displayed for sites present before (black lines) and after treatment (red
 813 lines). (a- d). Reproduced with permission.⁵⁸ Copyright 2018, Wiley-VCH. (e) Scheme
 814 for the preparation of Co₃S₄ PNS_{vac}. (f) EPR spectra of Co₃S₄ PNS_{vac} and Co₃S₄ NS. (g)
 815 The mass activity of different samples as a function of overpotential. (h) The adsorption
 816 energies (ΔE_{ads}) and (i) the activation energy barriers of an H₂O molecule on three models
 817 surfaces. (e- i). Reproduced with permission.⁶² Copyright 2018, American Chemical
 818 Society.
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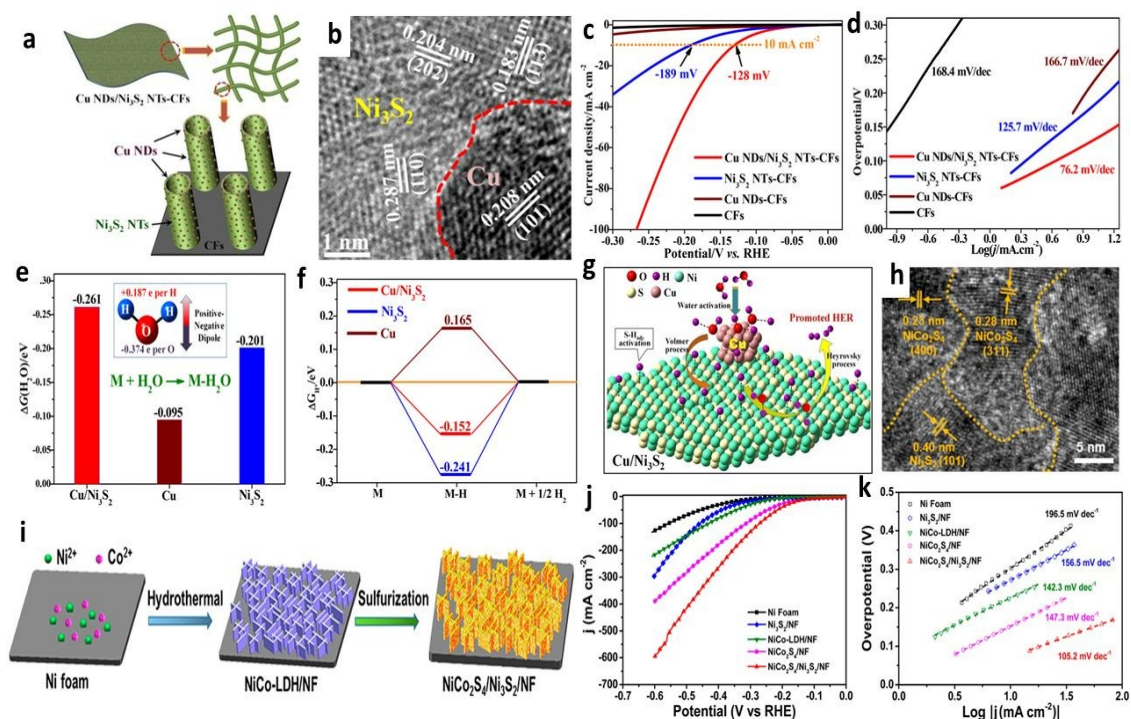
820 Recently, a growing number of high-performance TMSs are fabricated as hybrids, such
 821 as MoS₂-CoS₂@MoO₂,¹⁶⁷ MoS₂/LaNiO_{3- δ} ,¹⁶⁸ MoS₂-Co(OH)₂,¹⁶⁹ MoS₂/Ni₃S₂/Ni,¹⁷⁰
 822 MoS₂/MoP/NC¹⁷¹ and Pt₃Ni/NiS¹⁷². These composites commonly consist of two or more
 823 active components. Strong interactions between the different compounds can induce the
 824 electron transfer which will significantly optimize the intrinsic catalytic activity of the

825 metallic sites in TMSs.³² Recently, Feng et al. reported a Cu-Ni₃S₂ hybrid for alkaline
826 HER (**Figure 11 a and b**).³² Compared with the single Ni₃S₂ and Cu-based samples, the
827 metal/metal sulfide hybrids manifested a smaller overpotential (η_{10} = 128 mV) and a
828 lower Tafel slope (76.2 mV dec⁻¹) (**Figure 11c and d**). The DFT calculations indicated
829 that the electron density of Cu increased while the electron density of Ni₃S₂ decreased.
830 The positively charged Cu could adsorb and activate H₂O molecules by capturing O
831 atoms of water. As a result, the Cu/Ni₃S₂ hybrids owned a lower water adsorption energy
832 and could promote H₂O adsorption for alkaline HER (**Figure 11e**). In addition, the
833 Cu/Ni₃S₂ hybrids attained appropriate adsorption energy of H (**Figure 11f**), which was
834 beneficial to H desorption. In short, the hybridization of Cu with Ni₃S₂ can facilitate the
835 entire process of the alkaline process (**Figure 11g**).

836
837 In addition, the mixed TMSs possess more active sites and the electron tunneling effects
838 between different TMSs also benefit the alkaline HER. For instance, Liu et al. fabricated
839 a 3D heteromorphous NiCo₂S₄/Ni₃S₂/NF network with NiCo-LDH as the precursor
840 (**Figure 11i**).⁶⁴ The HRTEM image elucidated the interfacial heterostructure between
841 NiCo₂S₄ and Ni₃S₂ terminations (**Figure 11h**). The electrochemical tests indicated that
842 NiCo₂S₄/Ni₃S₂/NF manifested a better catalytic activity towards HER with the lowest
843 overpotential (η_{10} = 119 mV) and smallest Tafel slope (105.2 mV dec⁻¹) compared to
844 other samples in **Figure 11j and k**. Moreover, recently documented hybrids, like
845 NiS₂/MoS₂,¹⁷³ MoS₂/Co₉S₈,¹⁷⁴ MoS₂/NiS,^{175, 176} NiS/NiS₂/Ni₃S₂,¹⁷⁷ MoS₂/Ni₃S₂,¹⁷⁸
846 NiCo₂S₄/Co₉S₈¹⁷⁹ also exhibit distinguished catalytic performance in alkaline HER.

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Figure 11. (a) Schematic illustration of the microstructure of Cu NDs/Ni₃S₂ NTs-CFs. (b) HRTEM image of Cu/Ni₃S₂ border. (c) Polarization curves and (d) Tafel plots of Cu NDs/Ni₃S₂ NTs-CFs, Ni₃S₂ NTs-CFs, Cu NDs-CFs, and CFs. (e) The calculated adsorption free energy changes of H₂O on Cu/Ni₃S₂, Ni₃S₂, and Cu. (f) H adsorption free energy profiles of Cu NDs/Ni₃S₂ hybrid, Ni₃S₂, and Cu. (g) Schematic illustration of water adsorption, water activation, and hydrogen generation processes on Cu/Ni₃S₂. (a-g). Reproduced with permission.³² Copyright 2018, American Chemical Society. (h) HRTEM images of NiCo₂S₄/Ni₃S₂/NF. (i) Schematic illustration of 3D NiCo₂S₄/Ni₃S₂/NF. (j) Polarization curves and (k) the corresponding Tafel slopes of NiCo₂S₄/Ni₃S₂/NF, NiCo-LDH/NF, NiCo₂S₄/NF, Ni₃S₂/NF, and bare Ni foam at a scan rate of 5 mV s⁻¹ without IR correction. (h-k). Reproduced with permission.⁶⁴ Copyright 2018, American Chemical Society.

863 3.3.4 Transition metal selenides

864 Transition metal selenides (TMSes) have allured increasing attention in electrocatalysis
865 due to their metallic nature with low intrinsic electrical resistivity. Compared to TMOs
866 and TMSs, TMSes manifests a faster electron-transfer capacity so as to facilitate the
867 electrocatalytic processes. Mo_xSe_y, Co_xSe_y, and Ni_xSe_y are the most popular
868 electrocatalysts for alkaline HER in this series. However, the bare TMSes still requires
869 improvement to compete with the noble metal-based electrocatalysts.

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871 Doping TMSes with heteroatoms is a popular approach to enhance the intrinsic catalytic
872 activity. The introduction of a second cation or anion may have different influences on
873 the properties of original TMSes, giving rise to modulated catalytic performance. For
874 example, Zhang et al. explored the role of Co in the Co-WSe₂/MWNTs, and they found
875 that the introduction of Co could improve the inherent activity of the active sites.⁶⁵
876 However, the Co-doped catalyst showed a poor activity-stability relationship in alkaline
877 solutions. The main reason is that the Co incorporation resulted in a high proportion of
878 metal atoms being exposed on the surface, rendering the catalyst thermodynamically
879 metastable and vulnerable to oxidation. In another study, Zhao and co-authors unveiled
880 the importance of Ni dopant in MoSe₂.⁶⁶ The dopants can not only facilitate water
881 adsorption but also optimize H adsorption. Similarly, Jing et al. suggested that the N-
882 doped NiSe₂ had a lower ΔG_{H^*} and water adsorption energy compared to the bare NiSe₂.⁶⁷
883 Consequently, the N-doped catalysts demonstrate better electrocatalytic performance
884 with a low overpotential of 86 mV at 10 mA cm⁻². Interestingly, the role of P in the P-
885 substituted CoSe₂ is quite different.⁶⁸ Zhu et al. suggested that the presence of P could
886 generate more vacancies/defects in the CoSe₂ and significantly benefited the structural
887 transformation into metallic cobalt which is the intrinsic catalytic species for HER.

888

889 Hybridizing with conductive carbon-based materials/substrates (e.g., carbon cloth,¹⁸⁰ N-
890 doped carbon framework,¹⁸¹ carbon fiber paper,¹⁸² N-doped graphene,¹⁸³ N-doped carbon
891 nanotube,¹⁸⁴ N-doped porous carbon,^{185, 186} carbon microspheres¹⁸⁷) can further enhance
892 the conductivity of TMSes. Meanwhile, the carbon skeleton can prevent TMSes particles
893 from self-aggregation and corrosion.^{70, 181, 182, 187} Moreover, coupling with carbons can
894 regulate the electron spin density and charge distribution of the TMSes surface, which

895 manipulated the intrinsic activity of the hybrids.¹⁸¹ Recently, Wang et al. prepared
896 CoSe₂-MoSe₂ hybrids with reduced graphene oxide and amorphous carbon (CS-
897 MS/rGO-C) by a scalable spray-drying and selenization process.¹⁸⁷ Compared with CS-
898 MS, CS-MS/C, and CS-MS/rGO, CS-MS/rGO-C exhibited the best catalytic
899 performance in alkaline HER (Tafel slope of 83.2 mV dec⁻¹, η_{10} of 215 mV). Further
900 study indicated that the appealing catalytic activity of CS-MS/rGO-C stemmed from two
901 aspects. Firstly, the hollow porous microspheres significantly increased the contact areas
902 between the electrolyte and the catalyst, thus enabling enormous active sites exposed into
903 the electrolyte. Secondly, the conductivity of the composite could be remarkably
904 improved by the highly conductive carbon substrate.

905

906 Bimetallic selenides own obvious advantages over those monometallic selenides due to
907 the optimized electronic structure by two metal elements.⁷⁰ Taking NiCoSe₂ as an
908 example, Yu and co-workers fabricated the TMSes by an electrodeposition technique
909 (**Figure 12a**).⁷⁰ The HRTEM image displayed a clear lattice fringe of hexagonal NiCoSe₂
910 (**Figure 12b**). The results of electrochemical experiments revealed that bimetallic
911 NiCoSe₂ possessed a better catalytic performance than CoSe, NiSe, and NiCo-OH, with
912 a lower overpotential (η_{10} = 112.7 mV) and a smaller Tafel slope (65 mV dec⁻¹) (**Figure**
913 **12c**). The electronic structure of NiCoSe₂ showed that Ni, Co, and Se contributed
914 cooperatively to the total DOS (TDOS), and the overlap *d*-orbitals of Ni and Co implied
915 a covalent interaction between the Co and Ni atoms. What's more, the PDOS of NiCoSe₂
916 revealed that all the *d* orbitals from Ni, Co and Se contributed dominantly to the TDOS,
917 and the Se *p*- orbitals contributed to the covalent interactions among the three elements
918 (**Figure 12d** and **e**). Such regulated electronic structure vastly enhanced the intrinsic
919 electrocatalytic activity of NiCoSe₂. Moreover, the structurally engineered mixed metal

920 selenides manifested a high superhydrophilicity, which facilitated the water adsorption
921 process (**Figure 12f** and **g**).

922

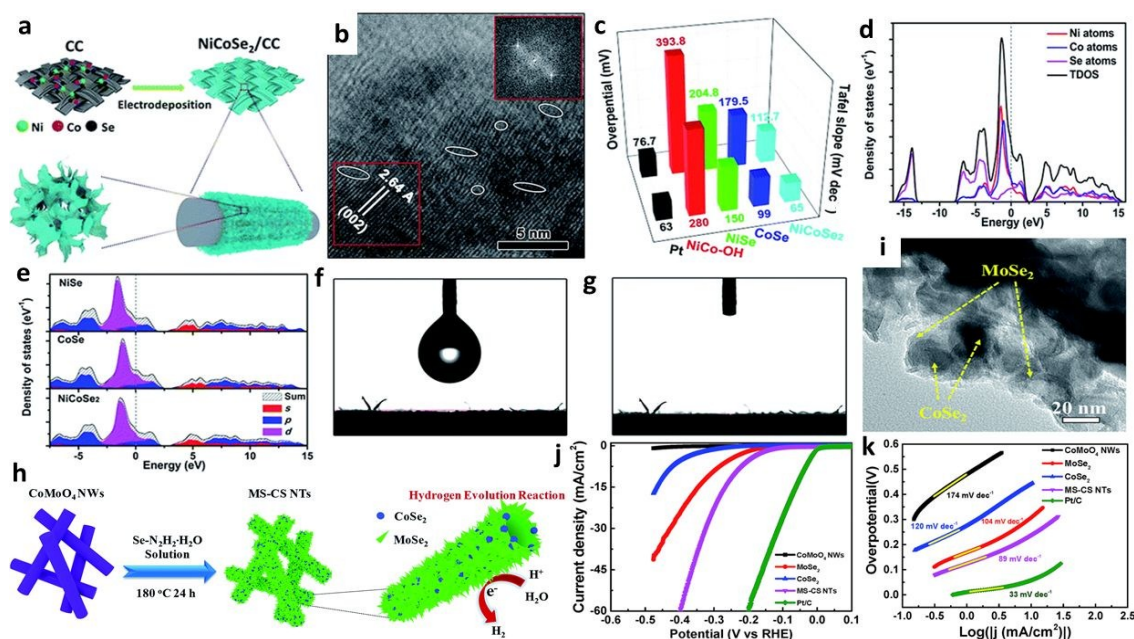
923 Hybridizing TMSeSes with other electroactive materials is well recognized as an
924 appropriate method to further improve their electrocatalytic activity. Typically,
925 TMSeSes/LDHs (e.g., $\text{CoNiSe}_2@\text{CoNi-LDHs/NF}$,⁷¹ $\text{CoSe}@\text{NiFe-LDH/NF}$,¹⁸⁸
926 $\text{SWCNTs/ex-MoSe}_2:\text{NiCl}_2$ ¹⁸⁹), TMSeSes/TMSeSes (e.g., $\text{MoSe}_2\text{-CoSe}_2$,^{69, 72, 187, 190} $\text{MoSe}_2\text{-}$
927 CoSe ,¹⁸⁵ $\text{MoSe}_2\text{-NiSe}$,¹⁹¹, $\text{MoSe}_2\text{-Ni}_{0.85}\text{Se}$ ¹⁹²), and TMSeSes/TMCs (e.g.,
928 $\text{SWCNTs/MoSe}_2\text{-2:Mo}_2\text{C}$ ¹⁹³) have been applied as high-performance catalysts for
929 alkaline HER. Taking $\text{CoNiSe}_2@\text{CoNi-LDHs/NF}$ as an example, Yang et al. found that
930 the combination of 1D CoNiSe_2 with 2D CoNi-LDHs could notably improve the
931 electrocatalytic activity, with an overpotential of $\eta_{10} = 215$ mV.⁷¹ DFT calculations and
932 in-situ Raman tests revealed that the LDHs-CoNiSe_2 interfaces could significantly
933 facilitate water adsorption and dissociation to generate H_{ads} because of the strong
934 electronic interactions through the interface, thus enhancing the alkaline HER process.

935

936 Constructing TMSe/TMSe heterostructures also can enhance the catalytic activity of the
937 bare TMSe catalysts, especially for the MoSe_2 -based materials. For instance, Zhao et al.
938 reported that the $\text{CoSe}_2/\text{MoSe}_2$ heterostructures delivered a better activity in alkaline
939 HER than MoSe_2 .¹⁹⁰ CoSe_2 species was revealed to be able to facilitate the water
940 adsorption and subsequent dissociation processes. Meanwhile, MoSe_2 species provided
941 the active sites for adsorption and combination of adsorbed hydrogen (H^*). In another
942 study, Wang et al. prepared hierarchical $\text{MoSe}_2\text{-CoSe}_2$ nanotubes (MS-CS NTs) by a
943 facile hydrothermal selenization process (**Figure 12h**).⁷² The prepared nanotubes
944 consisted of few-layered MoSe_2 nanosheets and CoSe_2 nanoparticles (**Figure 12i**).

945 Electrochemical tests marked that the MS-CS NTs owned a higher catalytic activity than
 946 MoSe₂ and CoSe₂ in 1M KOH (**Figure 12j** and **k**). The authors concluded that the highly
 947 conductive CoSe₂ particles in the few-layered MoSe₂ nanosheets significantly refined the
 948 charge transfer process, thus improving the electrocatalytic performance.

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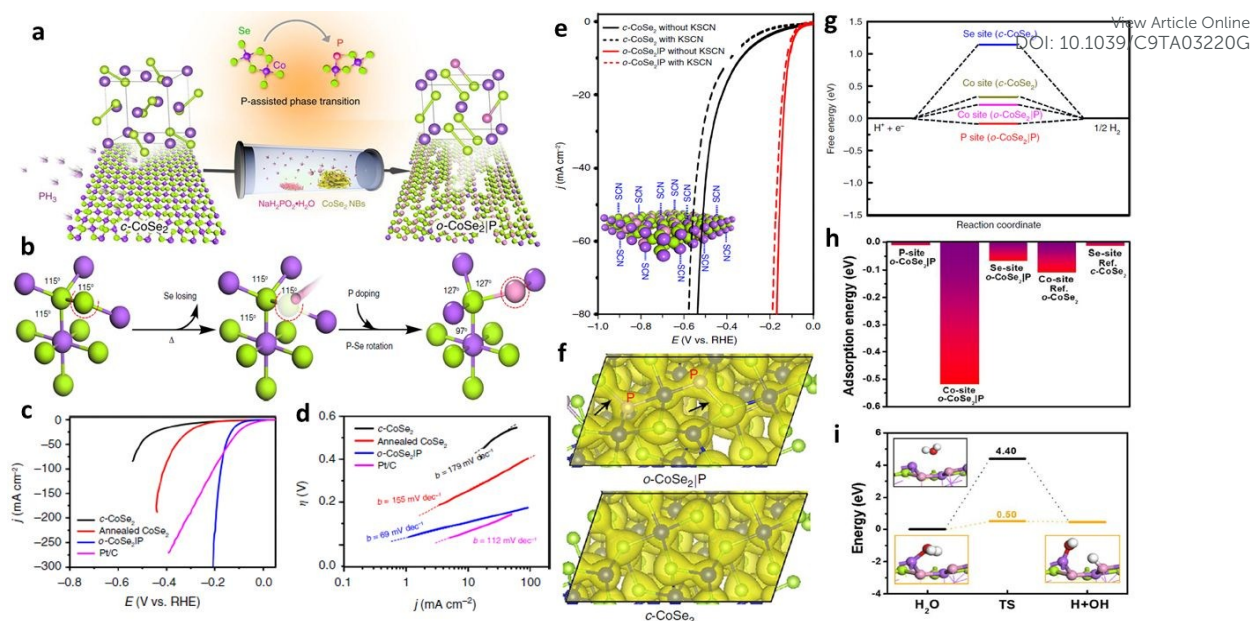
951 **Figure 12.** (a) Schematic diagram of the construction of the NiCoSe₂ nanosheet network.
 952 (b) HRTEM image of the NiCoSe₂. The inset of (b) shows the corresponding fast Fourier
 953 transform (FFT) pattern. (c) Comparison of the overpotentials required at 10 mA cm⁻²
 954 current density and Tafel slopes. (d) Calculated total and partial electronic density of
 955 states of NiCoSe₂. (e) The electronic density of states for NiSe, CoSe, and NiCoSe₂. The
 956 Fermi level is set at 0 eV. The images (f) before and (g) after the water drop fell on the
 957 surface of NiCoSe₂/CC. (a- g). Reproduced with permission.⁷⁰ Copyright 2018, The
 958 Royal Society of Chemistry. (h) Schematic illustration of the preparation of MS-CS NTs.
 959 (i) TEM image of MS-CS NTs. (j) Polarization curves and (k) Tafel plots of CoMoO₄
 960 NW, MoSe₂, CoSe₂, MS-CS NT and commercial Pt/C catalysts in 1 M KOH. (h- k).
 961 Reproduced with permission.⁷² Copyright 2018, The Royal Society of Chemistry.

962

963 It is noteworthy to mention that the catalytic performance of materials considerably
 964 relies on their phase. For instance, 1T-MoS₂ exhibits a much higher reactivity than the
 965 thermodynamically favoured 2H-MoS₂.²⁰ Similarly, phase engineering can be utilized to
 966 tune the intrinsic catalytic activity of TMSe. Chen et al. found that the cubic phase CoSe₂
 967 (c-CoSe₂) manifested notably improved activities over the orthorhombic phase CoSe₂ (o-

968 CoSe₂) in alkaline HER.⁷³ This is because c-CoSe₂ possessed a superior electrical
969 conductivity, the ideal water adsorption energy, as well as faster transformation
970 efficiency of H_{ads} into molecular hydrogen. In contrast, Zheng et al. found that the
971 catalytic properties of c-CoSe₂ were inferior to P-doped o-CoSe₂ (o-CoSe₂|P) by a P-
972 assisted transition process (**Figure 13a**).⁷⁴ Combined with XRD patterns, X-ray
973 photoelectron spectroscopy (XPS) spectra, EDX spectra, and Raman spectra, the phase
974 transformation process was illustrated in **Figure 13b**. The electrochemical tests
975 suggested that o-CoSe₂|P outperformed c-CoSe₂ and annealed CoSe₂, and was
976 comparable to that of Pt/C (**Figure 13c and d**). The SCN⁻ probing tests identified that P
977 atoms were intrinsic active sites (**Figure 13e**). The DFT calculations suggested that the
978 enriched charge density of o-CoSe₂|P, especially at the P-doped regions is much
979 beneficial to HER process (**Figure 13f**). The obtained ΔG_{H^*} also revealed that the P site
980 owned the best activity for water reduction (**Figure 13g**). In addition, the Co sites in o-
981 CoSe₂|P exhibited better water adsorption and dissociation abilities than other
982 components in **Figure 13h and i**. The synergistic effect of P and Co in o-CoSe₂|P gave
983 rise to the best catalytic performance. Additionally, a partial phase transition from Ni₃Se₂
984 to NiSe created the phase junctions (Ni₃Se₂/NiSe) and demonstrated a better
985 electrocatalytic activity for alkaline HER than the single-phase counterparts.⁷⁵ This is
986 mainly attributed to the maximum exposure of the active sites, fast charge transport, and
987 favourable OH⁻ and H⁺ adsorptions. Therefore, the strategy of constructing
988 heterostructures by partial phase transformation is inspiring for the rational design of
989 other high-activity electrocatalysts.

990



991

992 **Figure 13.** (a) Synthesis and physical characterization of o-CoSe₂|P. (b) Schematic of
 993 the P-doping-assisted phase-transition process from cubic to orthorhombic phases in
 994 CoSe₂. (c) Polarization curves and (d) Tafel plots for the HER on c-CoSe₂, annealed c-
 995 CoSe₂, o-CoSe₂|P, and commercial Pt/C catalyst (20 wt %). Catalyst loading: ~1.02 mg
 996 cm⁻². Sweep rate: 10 mV s⁻¹. (e) Comparison of SCN⁻ ions effects on the HER activities
 997 of c-CoSe₂ and o-CoSe₂|P, respectively. The inset shows the schematic of Co centers
 998 blocked by the SCN⁻ ions. (f) Calculated charge density distribution for o-CoSe₂|P (up)
 999 and c-CoSe₂ (down) catalysts. (g) Free energy diagrams for hydrogen adsorption at
 1000 different sites on the (111) surface of o-CoSe₂|P (8 wt%) and c-CoSe₂. (h) Adsorption
 1001 energy of H₂O molecule on o-CoSe₂|P surface. (i) H₂O dissociation in a vacuum and on
 1002 o-CoSe₂|P surface. Reproduced with permission.⁷⁴ Copyright 2018, Nature Publishing
 1003 Group.

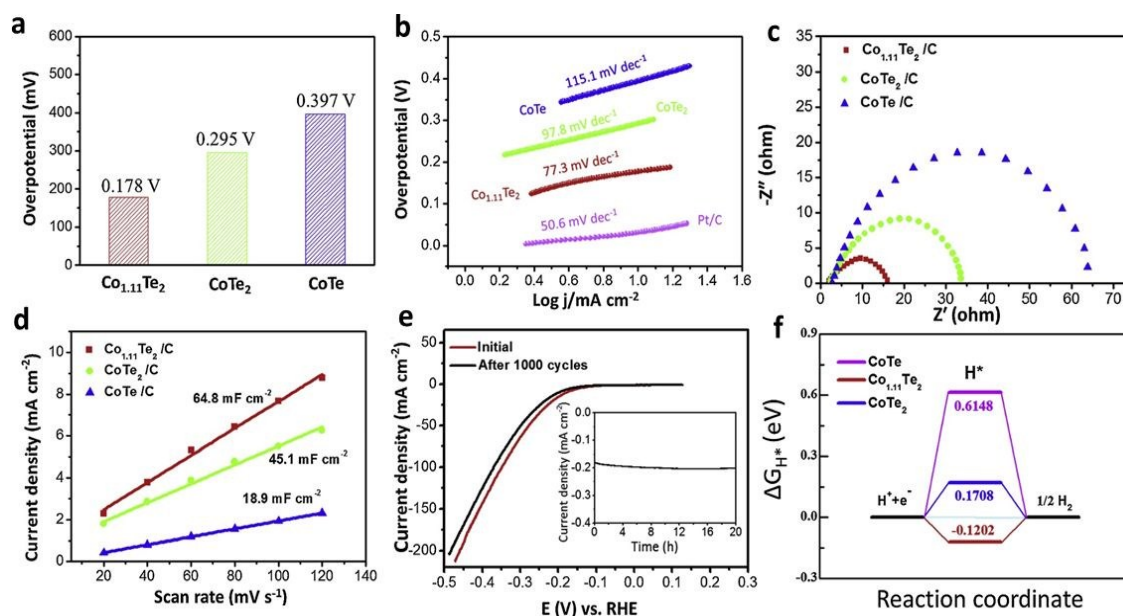
1004

1005 3.3.5 Transition metal tellurides

1006 Transition metal tellurides (TMTes) are emerging catalysts for alkaline HER, although
 1007 only a few of TMTes have been reported. Compared with the congeners (O, S, Se),
 1008 tellurium possesses a higher metallic characteristic, endowing a better electronic
 1009 conductivity to TMTes and a higher degree of covalency in the metal-chalcogen bonds.⁷⁶
 1010 ¹⁹⁴ These merits can benefit HER in alkaline solutions. To date, Co_xTe_y,^{76, 195, 196}
 1011 Ni_xTe_y,^{194, 197, 198} and Mo_xTe_y¹⁹⁹ have demonstrated respectful catalytic properties
 1012 towards alkaline HER. In a recent study, Wang et al. prepared component-controllable
 1013 cobalt telluride nanoparticles by chemical vapor deposition.⁷⁶ Compared with CoTe/C
 1014 and CoTe₂/C, the component optimized Co_{1.11}Te₂/C yielded more reducible Co species

1015 and a higher surface dispersion of Co ions. Electrochemical tests stated that the catalytic
 1016 activity of $\text{Co}_{1.11}\text{Te}_2/\text{C}$ was higher than CoTe/C and CoTe_2/C , with a low overpotential
 1017 ($\eta_{10} = 178 \text{ mV}$) and a small Tafel slope (77.3 mV dec^{-1}) (**Figure 14a** and **b**). Further,
 1018 $\text{Co}_{1.11}\text{Te}_2/\text{C}$ showed the smallest charge resistance and the largest double layer
 1019 capacitance, which implied a fast electron-migration capacity and a large active surface
 1020 area, respectively. (**Figure 14c** and **d**). $\text{Co}_{1.11}\text{Te}_2/\text{C}$ also presented a good cycling
 1021 capability with negligible performance loss in **Figure 14e**. The DFT calculations showed
 1022 that ΔG_{H^*} of $\text{Co}_{1.11}\text{Te}_2$ was the smallest one, suggesting the optimal H adsorption on the
 1023 $\text{Co}_{1.11}\text{Te}_2$ surface in the alkaline HER process. The order of computational $|\Delta G_{\text{H}^*}|$
 1024 successfully verified the experimental results. As aforementioned, the catalytic
 1025 applications of TMTes are at the early stage, so further explorations are encouraged to
 1026 develop high-performance TMTes toward alkaline HER.

1027



1028

1029 **Figure 14.** (a) Overpotentials of $\text{Co}_{1.11}\text{Te}_2/\text{C}$, CoTe_2/C and CoTe/C at the current of 10 mA cm^{-2} .
 1030 (b) Tafel slopes of $\text{Co}_{1.11}\text{Te}_2/\text{C}$, CoTe_2/C and CoTe/C . (c) Nyquist plots. (d)
 1031 The plot of double-layer charging current density versus scan rate. (e) CV curves of the
 1032 $\text{Co}_{1.11}\text{Te}_2/\text{C}$ catalyst before and after 1000 cycles and the insert showing
 1033 chronoamperometric curves at the current of 10 mA cm^{-2} . (f) Gibbs free energy diagram
 1034 of CoTe_2 , $\text{Co}_{1.11}\text{Te}_2$ and CoTe . Reproduced with permission.⁷⁶ Copyright 2019, Elsevier
 1035 Ltd.

1036

1037 **3.4 Nitrogen group transition metal catalysts**

1038 3.4.1 Transition metal nitrides

1039 Transition metal nitrides (TMNs) are new favourites in alkaline HER which possess
1040 unique metallic physiochemical properties and electronic features.²⁰⁰ The nitrogen
1041 elements can raise the *d*-electron density and lead to the contraction of *d*-band of TMs,
1042 affording TMNs with the similar electronic structure to the noble metals (i.e., Pd and
1043 Pt).²⁰⁰ Additionally, the distinguished conductivity and good corrosion resistance also
1044 make TMNs high-performance electrocatalysts.²⁰¹

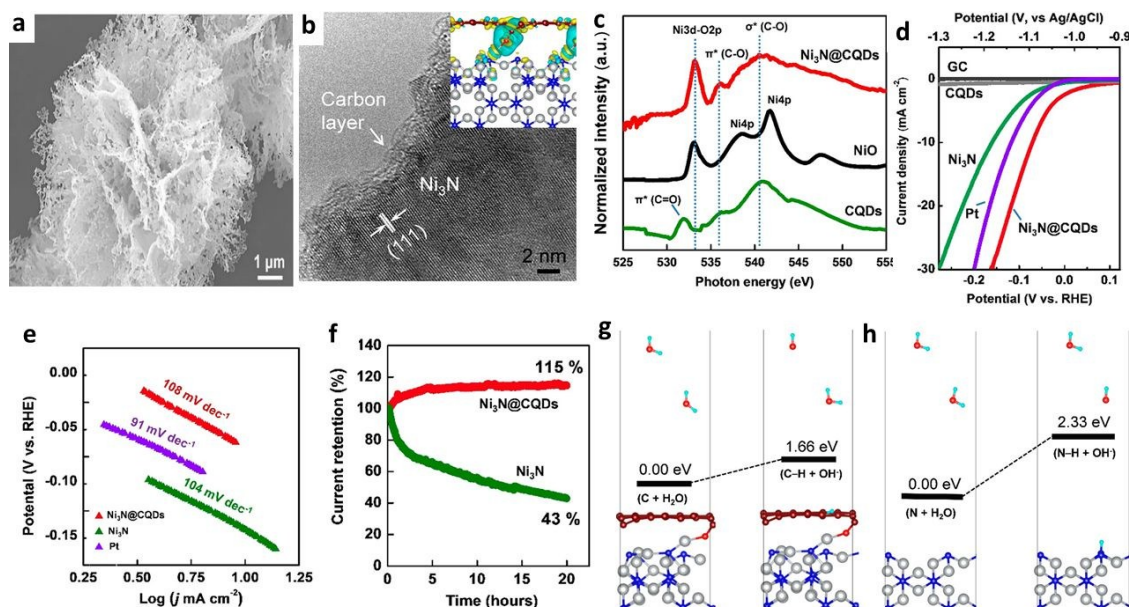
1045

1046 Ni_xN_y and Mo_xN_y are the primary documented catalysts. Commonly, the *y/x* is less than
1047 1 in TM_xN_y, such as Ni₃N,^{5, 78, 79, 81} Mo₂N,^{202, 203} Cu_xNi_yFe_{4-x-y}N,²⁰⁴ and Ni₃FeN,²⁰⁵
1048 because incorporation of N atoms into a transition metal lattice is thermodynamically
1049 unfavourable.⁷⁷ The low valance state of the metal atoms may result in oxidation of
1050 TMNs during the electrochemical tests.²⁰⁶ Thus, creating N-rich TMNs (*x/y* < 1) is a
1051 great strategy to ameliorate the longevity and intrinsic activity of catalysts. For example,
1052 Jin and co-workers designed Mo₅N₆ nanosheets by a Ni-inducing growth method.⁷⁷
1053 Compared with Ni_{0.2}Mo_{0.8}N, MoN and Ni₃N, Mo₅N₆ possessed a higher electrocatalytic
1054 activity ($\eta_{10} = 94$ mV, Tafel slope 66 mV dec⁻¹) and impressive durability in 1M KOH.
1055 Further study indicated that the appealing capability of Mo₅N₆ originated from its Pt-like
1056 electronic structure and the high valance state of the Mo atoms. Moreover, bimetallic and
1057 trimetallic nitrides are recognized to exhibit better abilities than the corresponding single
1058 metal nitrides due to the regulated electrical conductivity and the synergy
1059 between/among the adjacent heteromaterials.²⁰⁴ The currently reported P-NiMo₄N₅,²⁰⁷

1060 $\text{Cu}_x\text{Ni}_y\text{Fe}_{4-x-y}\text{N}$,²⁰⁴ $\text{Cu}_x\text{Ni}_{4-x}\text{N}$,²⁰⁴ FeNi-N ,²⁰⁵ and Ni-Mo-N ⁹ are identified as high View Article Online
DOI: 10.1039/C9TA03220G
1061 performance and low-cost catalysts towards alkaline HER.
1062
1063 Hybridizing TMNs with other components (i.e., carbon materials, metals/alloys, TMXs),
1064 especially metals, can further upgrade the electrocatalytic performance. In this part,
1065 TMN/C (TMN/carbon) and TMN/M (TMN/alloy) will be detailed. Generally, carbon
1066 materials prominently work as a conductive substrate, and they can also disperse active
1067 sites, protect metallic catalysts from corrosion and regulate the electron culture of the
1068 supported TMNs nanoparticles.²⁰⁰ As a consequence, the addition of carbon materials
1069 can significantly improve the catalytic performance of TMNs.^{78, 203, 205, 208, 209} For
1070 instance, Zhou et al. prepared a carbon layer-coated Ni_3N nanocomposite ($\text{Ni}_3\text{N}@$ CQDs)
1071 which possessed a porous structure (**Figure 15a**) and a few atoms thick amorphous
1072 carbon layers coated on the Ni_3N surface (**Figure 15b**).⁷⁸ It could be found that O atoms
1073 covalently bridged C and Ni atoms and formed Ni-O-C structures (**Figure 15c**).
1074 Electrochemical experiments disclosed that $\text{Ni}_3\text{N}@$ CQDs exhibited higher catalytic
1075 activities than GC, CQDs, and Ni_3N , with a lower overpotential ($\eta_{10} = 69$ mV) and a
1076 small Tafel slope (108 mV dec^{-1}) (**Figure 15d** and **e**). Compared to Ni_3N , $\text{Ni}_3\text{N}@$ CQDs
1077 manifested much better stability (**Figure 15f**), and the advancement of current density
1078 could be attributed to the increased number of utilizable active sites due to the activation
1079 effect. Hence, the CQDs coating not only upgraded HER activities for the Ni_3N surfaces
1080 but also stabilized the catalysts in the alkaline media. Further computational results
1081 revealed that the presence of negatively charged C sites became catalytic centers which
1082 obviously decreased the energy barrier for the water dissociation step, from 2.33 eV to
1083 1.66 eV, thus accelerating the Volmer step (**Figure 15g** and **h**). Therefore, combining

1084 TMNs with carbon materials into hybrids is an effective approach to develop robust and
 1085 durable alkaline HER electrocatalysts.

1086



1087

1088 **Figure 15.** (a) SEM of Ni₃N@CQDs. (b) HRTEM image of Ni₃N@CQDs. The inset is
 1089 the simulated charge distributions in the proposed Ni₃N@CQDs structural model. The
 1090 loss and the gain of the charge are denoted by yellowish and bluish colors, respectively,
 1091 with the isosurface values of $\Delta\rho = \pm 2 \times 10^{-3} \text{ e}/\text{\AA}^3$. (c) O K-edge XANES spectra of
 1092 Ni₃N@CQDs, commercial NiO, and the CQDs treated in NH₃ at 370 °C. (d) LSV
 1093 polarization curves of Ni₃N@CQDs in comparison with platinum (Pt) electrode, Ni₃N,
 1094 CQDs, and glassy carbon (GC) electrode in a 1 M KOH aqueous solution. (e) Tafel slopes
 1095 of Ni₃N@CQDs, Ni₃N and Pt electrode. (f) Normalized HER amperometric I-t curves of
 1096 Ni₃N@CQDs and Ni₃N at a constant overpotential of 77 mV (-1.1 V vs Ag/AgCl). (g, h)
 1097 Comparison of HER Volmer reaction step and the resultant binding energies on carbon-
 1098 coated Ni₃N(110) (g) and Ni₃N(110) (h) surfaces, N, Ni, C, O, and H atoms are marked
 1099 in blue, gray, brown, red, and cyan colors, respectively. Reproduced with permission.⁷⁸
 1100 Copyright 2018, American Chemical Society.

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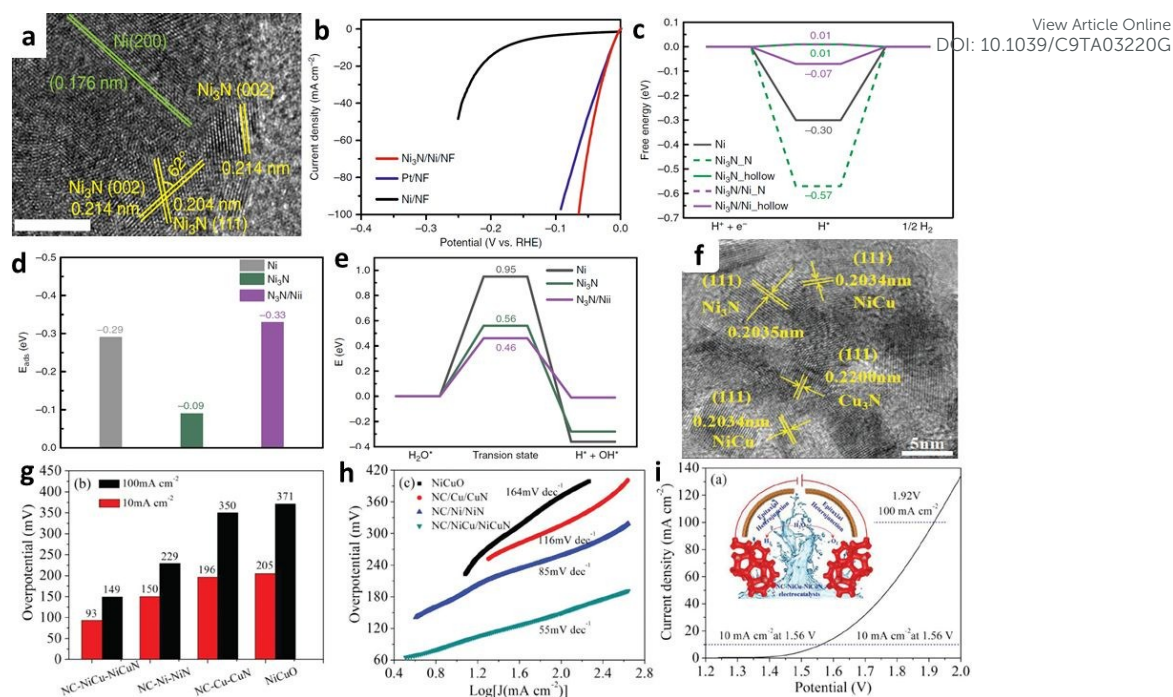
1102 Heterojunctions can integrate multiple catalytic components for alkaline HER with
 1103 increased active sites and ungraded electrical conductivity.⁸⁰ Furthermore, the interfaces
 1104 between different active components can further improve the catalytic performance of
 1105 the hybrids.²⁰⁹ Novel nanocomposites of TMN and metal/alloy such as Co-Ni₃N,²¹⁰
 1106 Ni₃N/Ni,⁷⁹ Co-Mo₂N,²⁰² NC-NiCu-NiCuN,⁸⁰ Co/CoN,²⁰⁹ Pt-Ni₃N,⁵ Ni/NiMo₄N₅,⁹ and
 1107 Co/VN²¹¹ have exhibited superior electrocatalytic capabilities. For example, Song et al.

1108 developed a Ni₃N/Ni interfacial electrocatalysts which owned higher intrinsic specific
1109 activities than Pt/C in 1.0M KOH (**Figure 16a and b**).⁷⁹ The in-depth DFT study
1110 demonstrated that H atoms preferred to adsorb along the interfaces between Ni₃N and Ni
1111 (**Figure 16c**). Moreover, Ni₃N/Ni possessed a higher water adsorption energy of 0.33 eV
1112 than Ni₃N (0.09 eV) and Ni (0.29 eV), as well as the smallest activation energy barrier
1113 of water dissociation (**Figure 16d and e**). The computational results revealed that the
1114 interfacial sites of Ni₃N/Ni provided appropriate binding affinities towards H₂O and H,
1115 and facilitated the water dissociation. In another study, Hou et al. fabricated NC-NiCu-
1116 NiCuN catalysts with carbonization-/nitridation-induced *in situ* growth strategies.⁸⁰ The
1117 presence of Ni₃N, Cu₃N, and NiCu offered the catalyst more active sites (**Figure 16f**).
1118 The electrocatalytic activity of TMN/alloy hybrids ($\eta_{100} = 149$ mV, Tafel slope 55 mV
1119 dec⁻¹) was better than the TMN/M composites with the single metal compound (**Figure**
1120 **16g and h**). Surprisingly, the NC-NiCu-NiCuN also exhibited outstanding activities for
1121 OER, and it only took 1.56 eV to deliver 10 mA cm⁻² water splitting current (**Figure 16i**).
1122 The superior catalytic performance could be ascribed to the improved electrical
1123 conductivity and enhanced accessibility of the active sites.

1124

1125 Apart from TMN/M and TMN/alloy hybrids, combining TMN with other electroactive
1126 TMXs (i.e., TMOs,^{81, 203} TMCs²⁰³) also manifests talented potentials. Take CeO₂ as an
1127 example, Sun et al. found that the Ni₃N-CeO₂/TM nanohybrids performed better than
1128 CeO₂/TM, Ni₃N/TM,⁸¹ and NiO-CeO₂/TM, with a low overpotential ($\eta_{10} = 80$ mV) and
1129 a small Tafel slope (122 mV dec⁻¹). Further study indicated that the incorporation of
1130 CeO₂ could refine the electrical conductivity and interfacial synergy between Ni₃N and
1131 CeO₂.

1132



1133

1134 **Figure 16.** (a) HRTEM image of Ni₃N/Ni interface. Scale bar, 5 nm. (b) Linear sweep
 1135 voltammetry (LSV) curves of Ni₃N/Ni/NF, Ni/NF, and optimized Pt/NF (Pt/C: 2.5 mg
 1136 cm⁻²) for HER in 1.0 M KOH with the current density normalized by the geometric
 1137 electrode area. (c) Hydrogen adsorption free energy (ΔG_{H^*}) on Ni, Ni₃N, and Ni₃N/Ni.
 1138 (d) Adsorption energy of water on Ni, Ni₃N, and Ni₃N/Ni. (e) Energy barrier for water
 1139 dissociation on Ni, Ni₃N, and Ni₃N/Ni. (a - e). Reproduced with permission.⁷⁹ Copyright
 1140 2018, Nature Publishing Group. (f) TEM images of NC-NiCu-NiCuN nanowires. (g)
 1141 Overpotentials at 10 and 100 mA cm⁻². (h) Tafel slopes of NiCuO, NC-Cu-CuN, NC-Ni-
 1142 NiN, and NC-NiCu-NiCuN electrodes. (i) Overall-water-splitting performance of NC-
 1143 NiCu-NiCuN array with two-electrode configuration. (f - i). Reproduced with
 1144 permission.⁸⁰ Copyright 2018, Wiley-VCH.

1145

1146 3.4.2 Transition metal phosphides

1147 Transition metal phosphides (TMPs) are the most extensively spotlighted candidates for
 1148 alkaline HER because of their impressive catalytic performance and low cost. The origin
 1149 of their excellent catalytic activities mainly due to the electronic features of P.
 1150 Phosphorus atoms in TMPs can extract electrons from the neighbouring transition metals
 1151 due to the high electronegativity. The negatively charged P atoms can trap the positively
 1152 charged proton as a base.¹⁵⁷ Therefore, the presence of P atoms leads to the moderate
 1153 bonding with the reaction intermediates/products over the catalyst surface.¹⁶ Thus, TMPs
 1154 typically possess higher electrocatalytic activities than TMXs with other anions.²¹² In

1155 this part, the application features, as well as effective strategies to boost the catalytic
1156 activity of TMPs in alkaline HER, will be discussed.

1157

1158 Single TM-based phosphides are the most reported TMPs. To date, Co_xP_y , Ni_xP_y , Fe_xP_y ,
1159 Mo_xP_y , Cu_xP_y , and W_xP_y have shown great activities to HER in basic media. Interestingly,
1160 the high conductive 2D TMP monolayer and corresponding oxidized counterpart are
1161 predicted as promising HER catalysts based on the DFT computations.²¹³ Among a series
1162 of calculated TMPs, $\text{Mo}_2\text{P}-2\text{H}$ and $\text{Fe}_2\text{P}-2\text{H}$ have a favourable ΔG_{H} which indicates a
1163 better HER performance, which is further verified by the experiment that these catalysts
1164 indeed exhibit high-performance in HER. Therefore, the advanced DFT calculations are
1165 great indicators for discovering innovative electrocatalysts for HER.

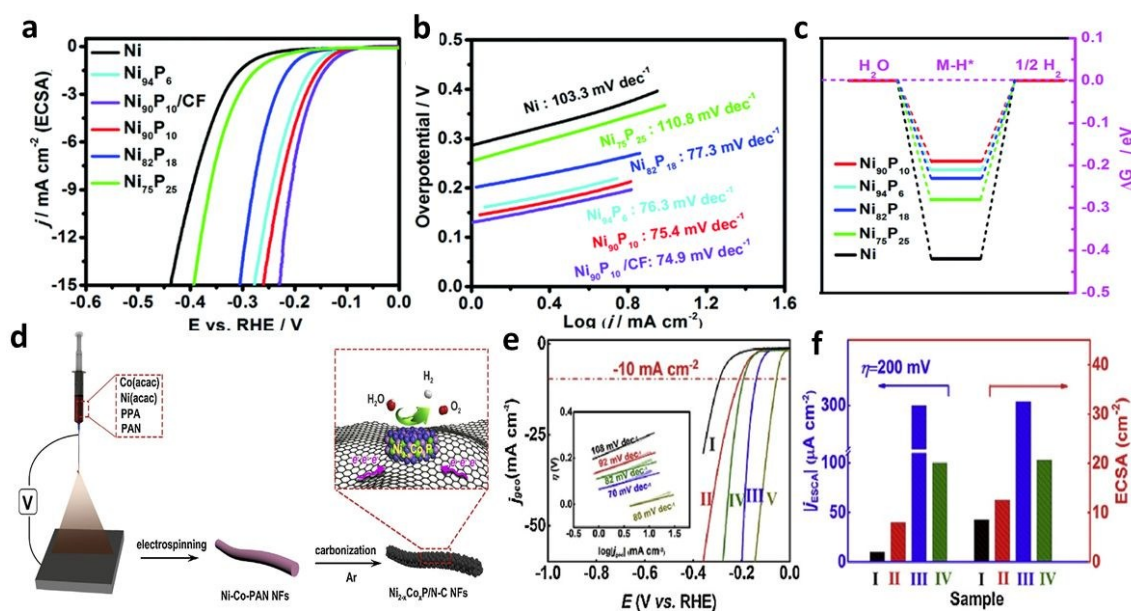
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1167 It is suggested that the variation of component can alter the catalytic performance of
1168 TMPs significantly because of the different roles of different elements. For example, Liu
1169 et al. designed a series of Ni_xP_y micro-spheres,⁸² and the catalytic properties of $\text{Ni}_{90}\text{P}_{10}$
1170 were the best among the as-prepared TMPs (**Figure 17a-b**). Further study revealed that
1171 the ΔG_{H^*} of $\text{Ni}_{90}\text{P}_{10}$ was the smallest (**Figure 17c**), which well supported the
1172 experimental results. The high P ratio can lead to favourable bonding strength of reaction
1173 intermediates/products with catalyst surfaces which indicates a better HER performance.

1174

1175 Beyond single TM-based phosphides, bimetallic phosphides own better activities for
1176 alkaline HER. Taking advantages of the synergistic effects of bimetallic sites with
1177 modulated electronic configurations and structural flexibility, bimetallic phosphides
1178 performed better than the single TM-based phosphides.^{29, 83} Among all the reported
1179 bimetallic phosphides (e.g., CoFeP ,^{214, 215} NiCuP ,⁸⁴ NiCoP ,^{29, 83, 216-218} NiMoP ^{33, 85}),

1180 NiCoP is the most frequently documented one. Recently, Mo et al. designed bimetallic
 1181 $\text{Ni}_{2-x}\text{Co}_x\text{P}$ catalysts by electrospinning and pyrolysis (**Figure 17d**).⁸³ They found that the
 1182 $\text{Ni}_{0.67}\text{Co}_{1.33}\text{P}/\text{N-C}$ exhibited the best catalytic activity (i.e., the smallest η_{10} , the lowest
 1183 Tafel slope, and the highest ECSA) over other NiCoPs with different Ni/Co ratios
 1184 (**Figure 17e and f**). Furthermore, the computational results stated that the regulated
 1185 electronic structure resulted in the boosted hydrogen binding and the improved alkaline
 1186 HER kinetics. More discussion involves this topic can be referred to a previous review.¹⁶
 1187



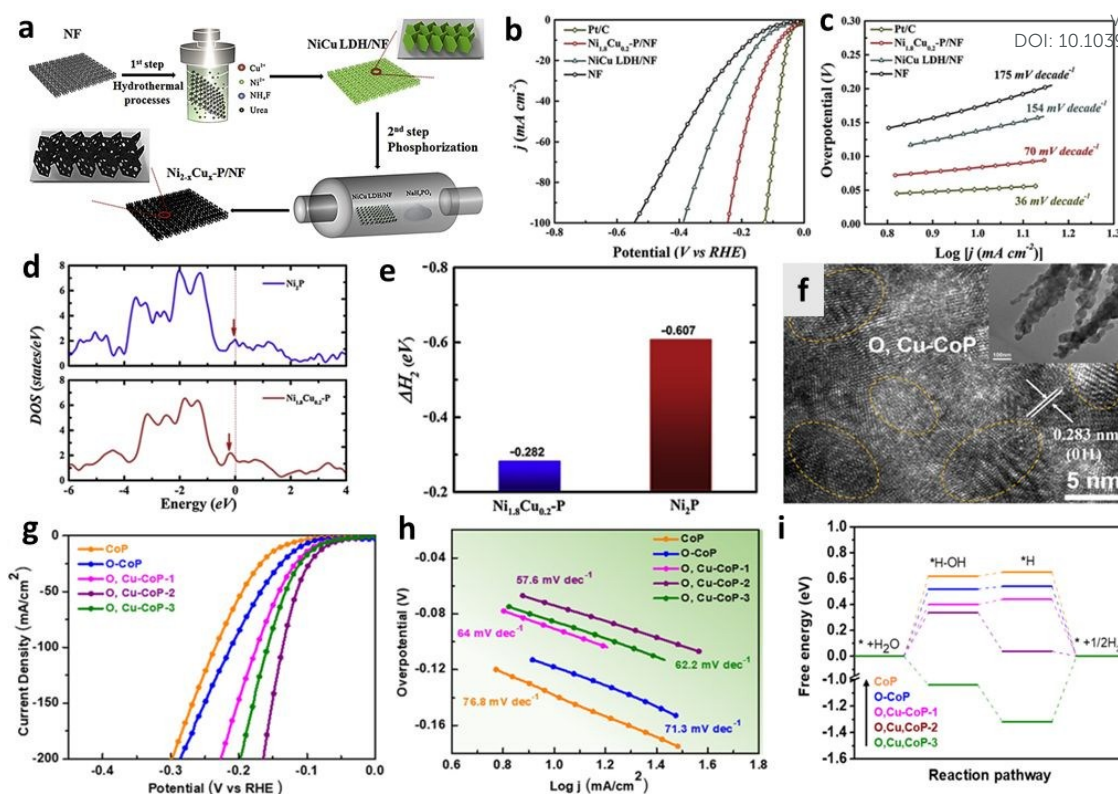
1188
 1189 **Figure 17.** (a) LSV curves and (b) the corresponding Tafel plots for Ni, Ni_{94}P_6 , $\text{Ni}_{90}\text{P}_{10}$,
 1190 $\text{Ni}_{82}\text{P}_{18}$, and $\text{Ni}_{75}\text{P}_{25}$ on Ti and $\text{Ni}_{90}\text{P}_{10}$ on CF. (c) H adsorption free energy of $\text{Ni}_{90}\text{P}_{10}$ and
 1191 Ni. (a- c). Reproduced with permission.⁸² Copyright 2018, The Royal Society of
 1192 Chemistry. (d) Schematic illustration for the fabrication of $\text{Ni}_{2-x}\text{Co}_x\text{P}/\text{N-C}$ NFs. (e)
 1193 Polarization curves and (f) corresponding ECSA and j_{ECSA} at $\eta=200 \text{ mV}$ of phosphide-
 1194 modified GCEs in 1.0M KOH, along with that of commercial 40% Pt/C (I:
 1195 $\text{Ni}_{1.67}\text{Co}_{0.33}\text{P}/\text{N-C}$, II: $\text{Ni}_{1.00}\text{Co}_{1.00}\text{P}/\text{N-C}$, III: $\text{Ni}_{0.67}\text{Co}_{1.33}\text{P}/\text{N-C}$, IV: $\text{Ni}_{0.33}\text{Co}_{1.67}\text{P}/\text{N-C}$, V:
 1196 40% Pt/C). (d- f). Reproduced with permission.⁸³ Copyright 2019, Elsevier Ltd.
 1197

1198 Doping becomes an efficient approach to upgrade the capability of TMPs. For TMPs, the
 1199 dopants can be metals, non-metals, as well as both of them. It is highly suggested that
 1200 the alkaline HER process can be affected by the atomic type and/or the exposed crystal
 1201 phase of the catalyst.²¹⁹ Hence, the dopants can have distinguishable impacts on the

1202 catalytic behavior of TMPs. Firstly, metal doping has prominent influence on the
1203 electronic structure, stability and surface properties of TMPs, resulting in a better
1204 catalytic performance. For example, Chu et al. designed a Cu-doped Ni₂P with the NiCu
1205 layered double hydroxide as a precursor (**Figure 18a**).⁸⁴ Experimental results implied
1206 that the Ni_{1.8}Cu_{0.2}-P/NF just required a small overpotential of 78mV to reach a current
1207 density of 10 mA cm⁻² and a low Tafel slope of 70 mV dec⁻¹ for the HER in 1.0 M KOH
1208 (**Figure 18b** and **c**). DFT study indicated that the Cu doping tuned the electronic structure
1209 of the Ni₂P by offering extra electrons in the valence band and near Fermi surface of
1210 Ni_{1.8}Cu_{0.2}P to raise the Fermi level to a higher energy level (**Figure 18d**). Also, the
1211 calculated ΔG_{H^*} of Ni_{1.8}Cu_{0.2}P (-0.282 eV) and Ni₂P (-0.607 eV) signified that Cu doping
1212 decrease the energy barrier for HER (**Figure 18e**). Lately, anions doping demonstrate
1213 great potentials in upgrading the catalytic activity of TMPs. Inspired by the impressive
1214 role of metal oxides and hydroxides in water dissociation and adsorption of hydrogen
1215 intermediates, oxygen has been recognized as an ideal dopant to ameliorate the catalytic
1216 ability of TMPs. Zhang and co-workers fabricated oxygen-incorporated NiMoP₂
1217 nanowire arrays on the Ni foam substrate.⁸⁵ Surprisingly, the as-prepared O-NiMoP₂/Ni
1218 manifested a geometrical catalytic current density of 10 mA cm⁻² at a low overpotential
1219 of 31 mV with a Tafel slope of 62.11 mV dec⁻¹. The theoretical calculations revealed that
1220 oxygen incorporation not only improved the interaction between H* and metal atoms
1221 toward a more favourable calculated ΔG_{H^*} , but remarkably promoted H₂O adsorption.
1222 Similarly, the role of N in the N-Co₂P/CC is almost the same as O. Men et al. found that
1223 N doping could tune the electronic structure of Co₂P, resulting in optimised adsorption
1224 free energies of water ($\Delta G_{H_2O^*}$) and hydrogen (ΔG_{H^*}), facilitating alkaline HER through
1225 the Volmer-Heyrovsky process.⁸⁶
1226

1227 Based on the benefits of both metal and non-metal doping strategies, the dual-doping
1228 approach also evoked increasing interests among the researchers. For instance, Xu et al.
1229 recently developed new oxygen and copper co-doped CoP nanowire array electrode by a
1230 low-temperature phosphatization method.⁸⁷ They found that the lattices of the co-doped
1231 sample became curvy and locally distorted (**Figure 18f**), which would induce more
1232 exposure of surface active sites. The alkaline HER performance of prepared samples
1233 signified that the co-doped catalysts were better than the un-doped CoP and the oxygen-
1234 doped CoP (**Figure 18g and h**). The calculated free energy diagram indicated that both
1235 the water adsorption energy and the Gibbs free energy of hydrogen adsorption of O, Cu-
1236 CoP were lower than those of CoP (**Figure 18i**). The enhancement is due to the co-
1237 incorporation of oxygen and copper atoms in CoP, which results in the exposure of more
1238 active sites and optimization of activated water dissociation energy, as well as binding
1239 free energy of H* intermediates. To date, most of the reported catalysts are single doped,
1240 so more efforts are encouraged to explore the optimal categories and dosages of co-
1241 dopants to attain desirable catalysts.

1242

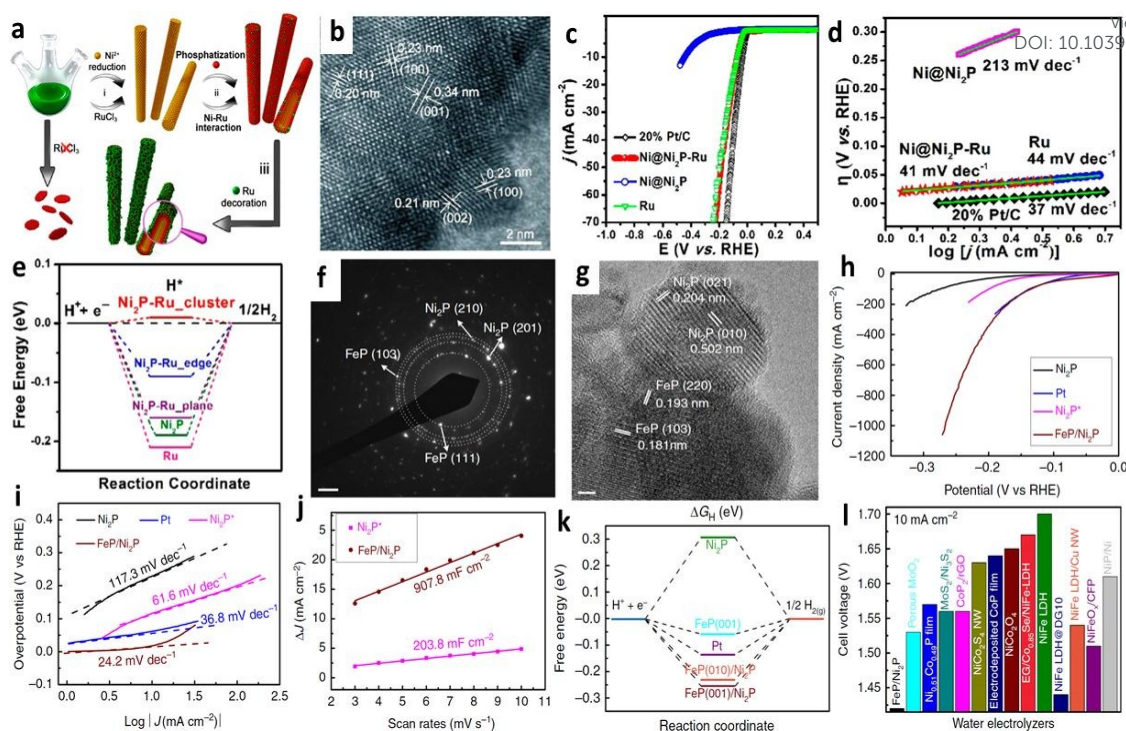


1243

1244 **Figure 18.** (a) Process steps in the synthesis of high-performance $\text{Ni}_{2-x}\text{Cu}_x\text{-P}$
 1245 electrocatalysts. (b) Polarization curves of HER and (c) corresponding Tafel plots
 1246 derived from the polarization curves. (d) Calculated DOS for Ni_2P and $\text{Ni}_{1.8}\text{Cu}_{0.2}\text{-P}$. (e)
 1247 HER free-energy change of Ni_2P doped with Cu. (a- e). Reproduced with permission.⁸⁴
 1248 Copyright 2019, Elsevier Ltd. (f) HRTEM images of O, Cu-CoP. Inset: Corresponding
 1249 TEM images. The circles indicate the lattice distortion. (g) IR-corrected polarization
 1250 curves and (h) Tafel plots of the O, Cu-CoP nanowire, O-CoP nanowire, CoP nanowire
 1251 electrodes, and bare carbon cloth in 1 M KOH electrolyte with Ag/AgCl as the reference
 1252 electrode and a graphite bar as the counter electrode. (i) HER diagram on the (011)
 1253 surface of clean CoP and oxygen- and copper-doped CoP. (f- i). Reproduced with
 1254 permission.⁸⁷ Copyright 2018, American Chemical Society.
 1255

1256 Theoretically, the hybrid materials would inherit the merits of each component and may
 1257 manifest new features due to the coupling effect, thus demonstrating higher
 1258 electrocatalytic activities than the single components. To date, novel composites of
 1259 TMPs/C (i.e. $\text{Ni}_{2-x}\text{Co}_x\text{P}/\text{N-C}$ NFs,⁸³ CQDs/MoP,²²⁰ $\text{Co}_2\text{P}@/\text{NPC}/\text{CC}$,²²¹ MoP/CNT,²²²
 1260 $\text{Ni}_2\text{P}@/\text{NPCNFs}$,²²³ $\text{Ni}_2\text{P}@/\text{PCG}$,²²⁴ $\text{Ru}_2\text{P}/\text{RGO}$ ²²⁵), TMPs/Ms (i.e. Co/CoP,^{226, 227}
 1261 $\text{Cu}@/\text{Cu}_3\text{P}$,²²⁸ $\text{Ni}@/\text{Ni}_2\text{P}-\text{Ru}$ ⁸⁸) and TMPs/TMXs (i.e. $\text{V}_4\text{P}_{6.98}/\text{VO}(\text{PO}_3)_2$,²²⁹ $\text{Ni}_3\text{S}_2/\text{Ni}_2\text{P}$,²³⁰
 1262 CoP/CeO_2 ,¹²⁷ $\text{CoP}(\text{MoP})-\text{CoMoO}_3@/\text{CN}$,²³¹ $\text{Ni}_2\text{P}/\text{Fe}_2\text{P}$,²³² $\text{FeP}/\text{Ni}_2\text{P}$,⁸⁹ MoC-
 1263 MoP/BCNC NFs,²³³ $\text{CoP}/\text{Mo}_2\text{C}-\text{NC}$ ⁹²) have exhibited superb potentials for alkaline HER.

1264 In this part, representative catalysts of TMPs/Ms and TMPs/TMXs will be discussed
1265 Recently, Liu et al. fabricated a unique multi-heterogeneous Ni@Ni₂P-Ru nanorod with
1266 a simple one-pot synthetic strategy (**Figure 19a**).⁸⁸ The as-prepared hybrids contained
1267 Ni₂P, Ni and Ru phases (**Figure 19b**), and the introduced Ru could regulate the
1268 phosphating process of Ni by a Ru-Ni coordination effect to obtain Ni@Ni₂P-Ru
1269 heterogeneous nanorods. Surprisingly, the electrocatalytic activity of Ni@Ni₂P-Ru was
1270 comparable to that of Pt/C (20 wt.%), with a low overpotential ($\eta_{10} = 31$ mV) and a small
1271 Tafel slope (41 mV dec⁻¹) (**Figure 19c** and **d**). Additionally, the DFT calculations
1272 revealed that Ru sites improved the H adsorption/desorption behaviors over Ni₂P (**Figure**
1273 **19e**). In another study, Yu et al. prepared a TMP/TMP (FeP/Ni₂P) hybrid by a simple
1274 thermal treatment process.⁸⁹ Numerous FeP and Ni₂P nanocrystals were distributed
1275 uniformly on the surface, forming abundant active sites on the catalyst (**Figure 19f** and
1276 **g**). The obtained hybrids showed higher electrocatalytic capability ($\eta_{10} = 14$ mV, Tafel
1277 slope 24.2 mV dec⁻¹) than the corresponding single components, even the Pt electrode
1278 (**Figure 19h** and **i**). Compared to Ni₂P, FeP/Ni₂P possessed a much higher C_{dl}, suggesting
1279 that the hybrids owned a high active surface area (**Figure 19j**). In-depth DFT
1280 investigations revealed that the introduction of FeP could significantly reduce the ΔG_{H^*}
1281 on Ni₂P surface(**Figure 19k**), which further supported the experimental results.
1282 Surprisingly, FeP/Ni₂P also presented marvelous potentials for OER, and it only took a
1283 very low cell voltage of 1.42 V to afford 10 mA cm⁻² for overall water splitting (**Figure**
1284 **19l**). Thus, low-cost and high-performance hybrids hold great potential for large-scale
1285 applications.



1286

1287 **Figure 19.** (a) Schematic illustration of the formation of Ni@Ni₂P-Ru HNRs. (2)
 1288 HRTEM image of Ni@Ni₂P-Ru. (c) LSV HER curves (scan rate: 5 mV s⁻¹) and (d) Tafel
 1289 plots of the different catalysts in 1.0 M KOH. (e) Computed free energy diagram of HER.
 1290 (a- e). Reproduced with permission.⁸⁸ Copyright 2018, American Chemical Society. (f)
 1291 The SAED pattern taken from the FeP/Ni₂P catalysts. Scale bar, 2 1/nm. (g) A typical
 1292 HRTEM image taken from the FeP/Ni₂P catalysts. Scale bar, 2 nm. (h) The HER
 1293 polarization curves of different catalysts. (i) The relevant Tafel plots. (j) Double-layer
 1294 capacitance measurements for determining electrochemically active surface areas of
 1295 Ni₂P and FeP/Ni₂P electrodes. (k) Free energy diagram for ΔG_{H} , the hydrogen adsorption
 1296 free energy at pH= 14 on FeP/Ni₂P catalyst in comparison with Ni₂P and benchmark Pt
 1297 catalysts. (l) Comparison of the cell voltages to achieve 10 mA cm⁻² among different
 1298 water alkaline electrolyzers. (f- l). Reproduced with permission.⁸⁹ Copyright 2018,
 1299 Nature Publishing Group.

1300

1301 3.5 Transition metal carbides

1302 Transition metal carbides (TMCs) have shown impressive catalytic performance towards
 1303 alkaline HER thanks to their noble-metal-like electronic structures, structural robustness,
 1304 and earth abundance.²³⁴ The *d* orbitals of M atoms will be broadened with the help of *s*-
 1305 and *p*- orbitals of C atoms, thus forming a similar Pt *d*-band state.²³⁵ Recently, TMCs
 1306 like Mo_xC_y,^{23, 235, 236} W_xC_y,^{237, 238} V_xC_y,^{39, 239} and Ni_xC_y²⁴⁰ have been applied in alkaline
 1307 HER, especially Mo₂C. An informative summary can be found in the latest review.²³⁵

1308

1309 Despite these advantages, the relatively poor conductivity of TMCs requires a highly
1310 conductive substrate. In most cases, carbon materials function well and can promote the
1311 electrocatalytic properties. Lately, Wei et al. prepared a Mo₂C/G-NCS (graphene
1312 wrapping N-doped porous carbon microspheres) catalyst by a two-step process.²⁵ The
1313 ultrafine nanoparticles with a size about 4 nm were embedded into the carbon matrix.
1314 The designed porous structure enhanced the electrode-electrolyte contact points and the
1315 charge transfer process. The XPS results indicated the presence of graphitic N, pyridinic
1316 N, and Mo-N. The pyridinic N was the primary N species to benefit the HER process.
1317 The formation of Mo-N bonds implied that N atoms were doped into Mo₂C, and this
1318 electron-rich dopant could downshift the density of empty *d*-band in Mo₂C, thus
1319 weakening the Mo-H strength. Similar phenomena were also reported in other studies.^{24,}
1320 ^{241, 242} The electrochemical tests indicated that the Mo₂C/G3-NCS750 sample possessed
1321 an outstanding alkaline HER activity ($\eta_{10} = 66$ mV, Tafel slope 37 mV dec⁻¹). Therefore,
1322 graphene wrapping, good conductivity, the well-defined porous microspherical structure,
1323 ultrafine Mo₂C nanocrystal, and regulated Mo-H strength all benefit the alkaline HER
1324 process.

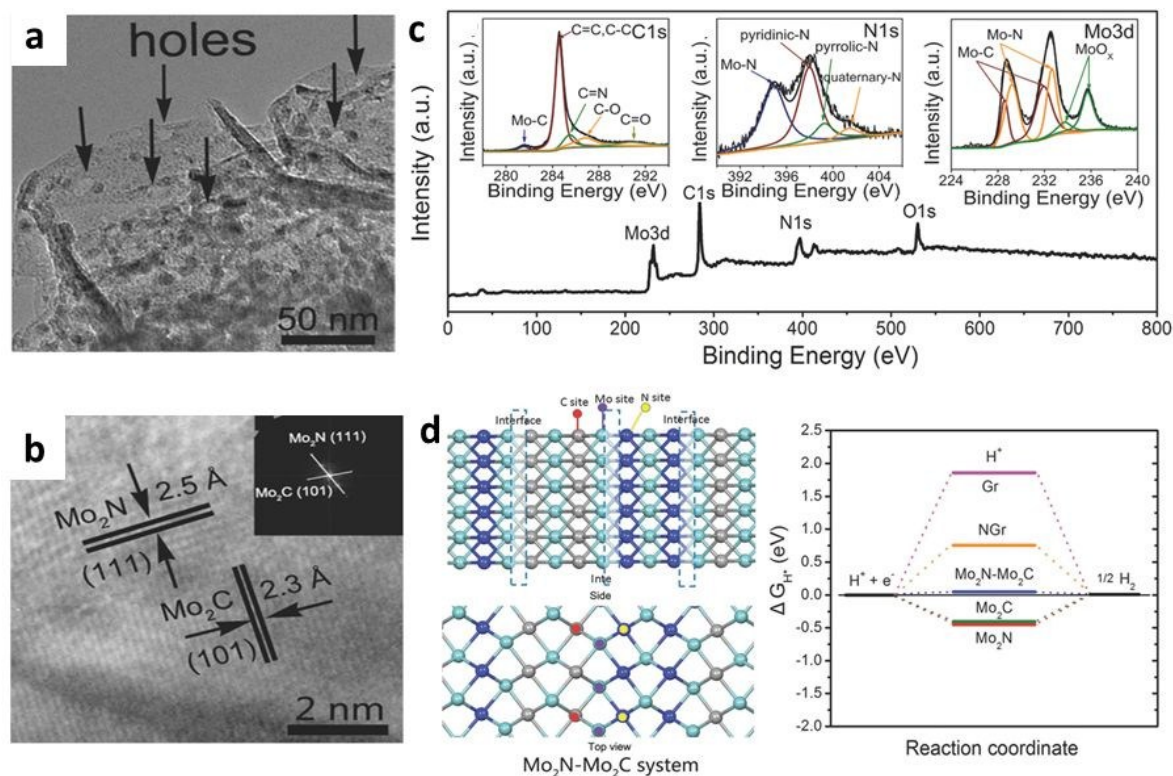
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1326 Other sophisticated methods are also efficient to promote the catalytic performance of
1327 TMCs, such as heteroatom doping and hybridizing with TMXs. In doping strategies, the
1328 universally employed dopants include N,^{24, 243} P,^{244, 245} Ni,²⁴⁶ and the combinations of
1329 these single elements.^{90, 246} Recently, Ji and co-authors prepared N, P-codoped
1330 Mo₂C/MoC nanofibers via pyrolysis of phosphomolybdic acid-doped polyaniline
1331 nanofibers.⁹⁰ Further experimental results indicated electron transfer from Mo to P and
1332 N. Specifically, for Mo₂C, the charge migration can enhance H adsorption onto Mo sites,

1333 whilst for MoC, the N and P act as the basic sites to trap positively protons and mediate
1334 proton transfer to the adjacent Mo sites to boost the HER process.
1335
1336 TMCs/TMXs hybrids, such as Mo/Mo₂C,⁹³ Co/ β -Mo₂C,²⁴⁷ MoO₂/ α -Mo₂C,²²
1337 Mo₂N/Mo₂C,^{91, 241, 248} Mo₂C/Mo₃Co₃C,²⁴⁹ MoC/Mo₂C,^{234, 250} and Mo₂C/CoP⁹² all
1338 exhibited appealing electrocatalytic activities for alkaline HER. In a recent study, Yan et
1339 al. applied an *in situ* catalytic etching method to prepare a Mo₂N-Mo₂C heterostructure.⁹¹
1340 Many holes were spotted to present adjacent to nanoparticles, and intimate contacts of
1341 Mo₂C with Mo₂N were identified by the HRTEM image in **Figure 20a** and **b**. The shift
1342 in XPS results further confirmed the electron transport between Mo₂C and Mo₂N, which
1343 benefited the catalytic reaction (**Figure 20c**). The obtained Mo₂N-Mo₂C/HGr-3 sample
1344 delivered a magnificent catalytic performance ($\eta_{10} = 154$ mV, Tafel slope 68 mV dec⁻¹)
1345 and good durability. DFT calculations implied that N atoms at the interface of N-Mo-C
1346 should be the dominant active sites in the Mo₂N-Mo₂C hybrid because of the smallest
1347 ΔG_{H^*} of 0.046 eV (**Figure 20d**). Analogously, interfaces in catalysis is also highlighted
1348 in other reports, which outperforms the catalytic properties of each part in the
1349 heterojunction. For example, Luo et al. suggested that the catalytic performance of
1350 CoP/Mo₂C-NC was better than CoP and Mo₂C-NC.⁹² The main reason was that the
1351 electron cloud transferred from Co to Mo through Co-P-Mo bonds, resulting in the
1352 generation of high valence state for Co³⁺ species and the low valence states for Mo²⁺ and
1353 Mo³⁺ species, providing rich active sites for alkaline HER. Interestingly, Xiong et al.
1354 unveiled that the Mo/Mo₂C heterointerfaces could not only supply a platform to gather
1355 the double-phase interface consisted of catalytic sites (exposed β -Mo₂C defects) and the
1356 electron-transfer channel (metallic Mo), but also optimize the intrinsic binding energy
1357 between H_{ads} and catalyst surface.⁹³ Thus, the Mo/Mo₂C hybrid exhibited superior

1358 alkaline HER activities ($\eta_{10} = 79$ mV, Tafel slope 62.86 mV dec^{-1}) than the sole Mo and
 1359 β - Mo_2C .

1360



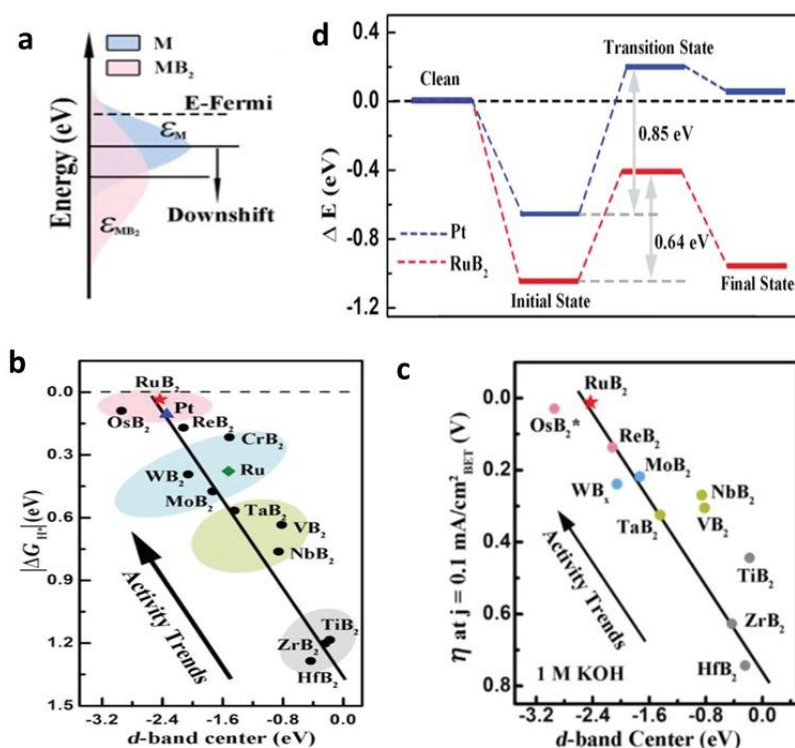
1361

1362 **Figure 20.** (a) Magnified TEM image, (b) HRTEM image (inset: FFT of Mo_2N and
 1363 Mo_2C), and (c) XPS survey spectrum of the Mo_2N - Mo_2C /HGr-3 hybrid. (d) Possible
 1364 adsorption sites of H^* on the Mo_2N - Mo_2C system and calculated free-energy diagram
 1365 for HER based on the various studied systems. Reproduced with permission.⁹¹ Copyright
 1366 2018, Wiley-VCH.
 1367

1368 3.6 Transition metal borides

1369 Transition metal borides (TMBs) have been extensively exploited as active catalysts for
 1370 OER, while their performance in HER is seldom documented. Recently, the good
 1371 electrocatalytic abilities of TMBs have attracted enormous attention, and Mo_xB_y ,²⁵¹
 1372 Ni_xB_y ,^{94, 252, 253} Co_xB_y ,²⁵⁴ Fe_xB_y ,²⁵⁵ Ru_xB_y ,²⁵⁶ etc. have been applied in alkaline HER. It
 1373 is suggested that the coupling interactions between s -, p - orbitals of B and d - orbitals of
 1374 M can regulate the d -orbitals center of TMBs downshift away from the Fermi level
 1375 (**Figure 21a**), thus obtaining an optimized M-H affinity and promoted HER

1376 performance.²⁵⁶ Li et al. studied the catalytic trends of 12 TMBs,²⁵⁶ from Group IV B to
 1377 Group VIII metal diborides. The computational results indicated that the *d*-band center
 1378 of TMBs was a good descriptor of the HER activity. A generally linear relationship can
 1379 be established between the *d*-band center and the absolute value of ΔG_{H^*} (**Figure 21b**).
 1380 The experimental results were in line with the DFT calculations (**Figure 21c**), and the
 1381 catalytic properties of RuB₂ were better than those of Pt/C (20 wt.%) in 1 M KOH. In
 1382 addition, the activation energy of water dissociation step on RuB₂ (001) plane was lower
 1383 than that on Pt (111) surface (0.64 eV vs. 0.85 eV), revealing a more favourable kinetic
 1384 process for RuB₂ (**Figure 21d**). Therefore, the theoretical and experimental study figured
 1385 out the importance of rationally tailoring the electronic structure of TMBs on regulating
 1386 their catalytic activities.
 1387



1388 **Figure 21.** (a) Schematic explanation of the boron effect on the *d*-bands of metal. ϵ_M and
 1390 ϵ_{MB_2} represent *d*-band centers of metal and metal diboride, respectively. (b) A fitted linear
 1391 relationship between hydrogen adsorption free energy (ΔG_{H^*}) and *d*-band center of MB₂,
 1392 Pt, and Ru. (c) A fitted linear relationship between the measured overpotential and *d*-
 1393 band center at a current density of 0.1 mA cm⁻² (normalized by BET surface area) in 1.0

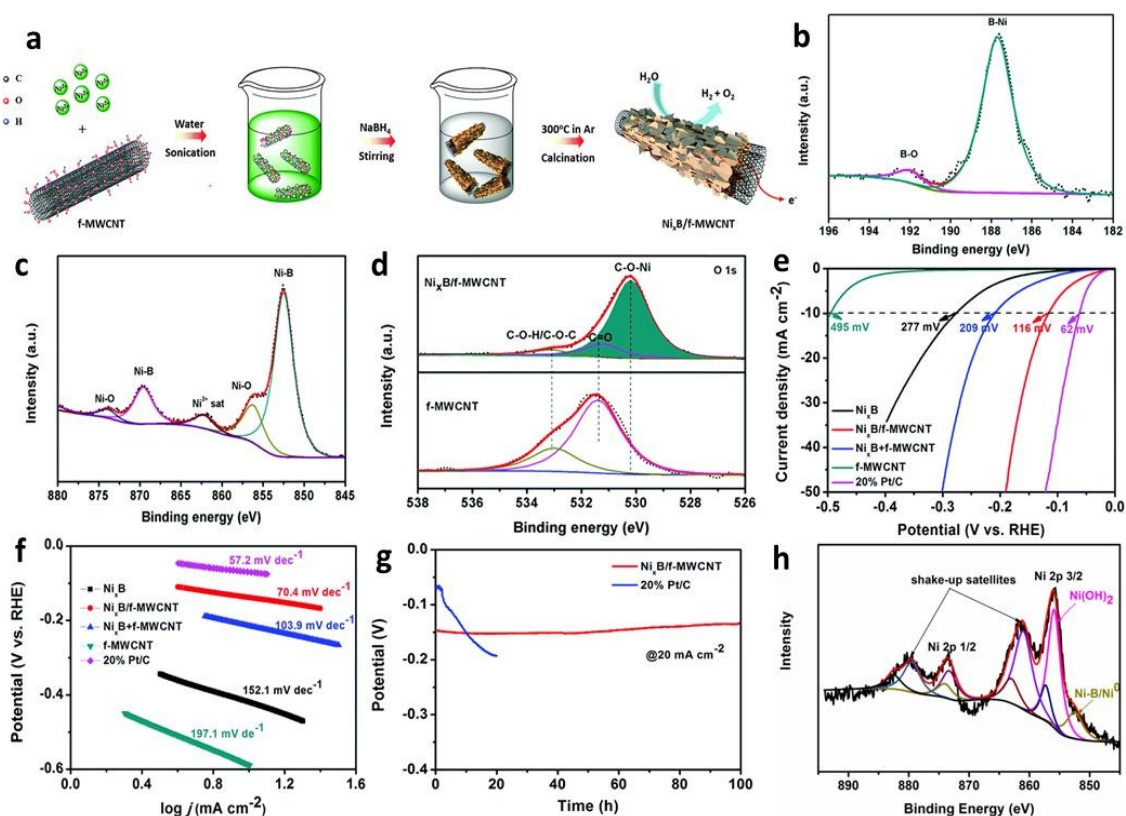
1394 M KOH solution. (d) Reaction pathways for water molecule dissociation on the RuB₂ (001) and Pt (111) surfaces. Reproduced with permission.²⁵⁶ Copyright 2018, Wiley-VCH.

1398 It is worth mentioning that amorphous TMBs are inevitably oxidized when exposed to
1399 air and water, thus forming oxides/hydroxides on the surface.^{94, 252, 254, 257} The presence
1400 of oxides/hydroxides can benefit the alkaline HER process. For instance, Chen et al.
1401 fabricated the Ni_xB/f-MWCNT hybrid by a room-temperature approach (**Figure 22a**).⁹⁴
1402 The XPS peak of B-O confirmed that the surface oxidation of Ni_xB, the Ni-O bonds
1403 originated from the interactions of Ni_xB with the O-containing functional groups on the
1404 MWCNT surface. The O 1s spectra (C-O-Ni) also marked a strong coupling effect by
1405 the oxygen bridge (**Figure 22b-d**). Electrochemical tests illustrated that Ni_xB/f-MWCNT
1406 possessed higher catalytic activities (η_{10} = 116 mV, Tafel slope 70.4 mV dec⁻¹) than Ni_xB,
1407 f-MWCNT, and their mechanical mixture (**Figure 22e and f**). Additionally, the Ni_xB/f-
1408 MWCNT owned better stability than Pt for alkaline HER (**Figure 22g**). The XPS results
1409 of the Ni_xB/f-MWCNT sample after the 100-hour HER test indicated the formation of
1410 Ni(OH)₂ on the catalyst surface (**Figure 22h**). Totally, the Ni(OH)₂/Ni_xB hybrids worked
1411 as the active component. The superb activity of Ni(OH)₂ for water dissociation and
1412 H*/OH* adsorption could significantly improve the catalytic performance of Ni_xB/ f-
1413 MWCNT. Accordingly, partial oxidation can be utilized as a facile strategy in the design
1414 of high-performance TMBs catalysts.

1415

1416 Apart from mono-metal borides, ternary borides also exhibit impressive catalytic
1417 performance. The introduction of a secondary metal/non-metal element can enhance the
1418 catalytic HER activity because of the regulated electronic structure and synergistic
1419 effect.²⁵⁸ Hitherto, Co-Mo-B,^{258, 259} Co-Ni-B,^{257, 260} Co-W-B,²⁶¹ and Co-B-P²⁶² have
1420 shown decent catalytic properties towards HER in basic media.

1421

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1422

1423 **Figure 22.** (a) Schematic of the synthesis of $\text{Ni}_x\text{B}/\text{f-MWCNT}$ bifunctional
 1424 electrocatalysts. Core-level XPS spectra of $\text{Ni}_x\text{B}/\text{f-MWCNT}$ (b) B 1s spectrum of $\text{Ni}_x\text{B}/\text{f-MWCNT}$, (c) Ni 2p spectrum of $\text{Ni}_x\text{B}/\text{f-MWCNT}$, (d) O 1s spectra of f-MWCNT and
 1426 $\text{Ni}_x\text{B}/\text{f-MWCNT}$. (e) Polarization curves and (f) corresponding Tafel plots of $\text{Ni}_x\text{B}/\text{f-MWCNT}$, Ni_xB , $\text{Ni}_x\text{B} + \text{f-MWCNT}$, f-MWCNT, and commercial 20 wt% Pt/C catalysts
 1428 in 1 M KOH electrolyte. (g) HER chronopotentiometry curves of $\text{Ni}_x\text{B}/\text{f-MWCNT}$ and
 1429 Pt/C under the current density of 20 mA cm^{-2} over 100 hours of operation. (h) Core-level
 1430 XPS spectra near the region of Ni 2p of $\text{Ni}_x\text{B}/\text{f-MWCNT}$ after the 100-hour HER test in
 1431 H_2 -saturated 1 M KOH electrolyte under the current density of 20 mA cm^{-2} . Reproduced
 1432 with permission.⁹⁴ Copyright 2018, The Royal Society of Chemistry.
 1433

1434 Apart from those aforementioned strategies mainly focus on enhancing the intrinsic
 1435 catalytic activity of the catalysts, constructing porous nanostructures has also been
 1436 proved to be an efficient strategy for enhancing the catalytic performance of the
 1437 catalysts.²⁶³⁻²⁶⁷ Generally, a porous structures can increase the specific surface area with
 1438 the maximum exposed active sites and provide more electrolyte-electrode contact
 1439 points.²⁴ Additionally, the unique architecture facilitate the charge transfer, and improve
 1440 the mass transportation/diffusion during the alkaline HER process.^{263, 268} For instance,

1441 Hao et al. developed a simple electroless plating process to fabricate a Co-B/Ni catalyst
1442 with a unique porous nodule structure and abundant internal spaces for catalytic
1443 reactions.⁹⁵ Further experimental results indicated that the HER activity decay was highly
1444 relevant to the decrease and blockage of internal pores, which verified the importance of
1445 the porous architecture. Meanwhile, the porous structure guarantees good mechanical
1446 adhesion and provides enormous vent channels for the gaseous product (H₂). Therefore,
1447 the porous structures of TMBs play a key role in the enhancement of electrocatalytic
1448 performance in HER.^{94, 257, 262, 269, 270}

1449

1450 **4. Concluding Remarks and Outlook**

1451 Nowadays, tremendous research enthusiasms dedicated to advancing the technology for
1452 hydrogen production, especially the water splitting in alkaline electrolytes. Therefore, it
1453 is of critical significance to create sufficient low-cost electrocatalysts to substitute noble
1454 metals. In this regard, TM-based materials hold impressive potentials. In this review, we
1455 summarise the recent progress on TM-based electrocatalysts for the alkaline hydrogen
1456 evolution reaction. According to their featured anions, 1) TMs; 2) TM alloys; 3) oxygen
1457 group TM-based catalysts (TMOs, LDHs, TMSs, TMSes, and TMTes); 4) nitrogen group
1458 TM-based catalysts (TMNs, TMPs); 5) TMCs; and 6) TMBs are comprehensively
1459 presented. We also highlight the mainstream strategies to upgrade the catalytic properties
1460 of each TMX. These state-of-the-art strategies aim at the following three goals: 1)
1461 improving the inherent catalytic activity by component regulation, defect (vacancy)
1462 engineering, doping, phase engineering, facet engineering, and hybridizing (interface
1463 engineering), etc.; 2) enriching active sites through nanosizing, designing dimensional
1464 architectures (0-3D), and hierarchical/porous structuring, etc.; 3) enhancing conductivity
1465 with the assistance of highly conductive substrates, such as graphene, Ni foam, and

1466 carbon materials. For a typical kind of TMX, selected strategies are highlighted in this
1467 review, but all of the aforementioned strategies are effective for all category of TMXs in
1468 constructing high-performance electrocatalysts.

1469

1470 To date, although remarkable progress has been achieved in the development of
1471 respectful electrocatalysts for alkaline HER, there are still some issues that deserve
1472 further explorations. Firstly, the design of novel electrocatalysts requires novel strategies
1473 to transform the mediocre materials into a high-performance catalyst through increasing
1474 the number of active sites and improving the conductivity, intrinsic activity, as well as
1475 durability. The “all-in-one” strategy offers the pristine material impressive catalytic
1476 properties. Nevertheless, how to assemble the related approaches logically is a key issue,
1477 which needs more efforts. In addition, computational methods can provide valuable
1478 guidelines to rapidly discover novel catalysts for alkaline HER. Combining experimental
1479 tests and computational results will improve the research efficiency and avoid the time-
1480 consuming trial-and-error route.²⁷¹ Another vital factor should be considered is the
1481 catalyst preparation procedures, where the reaction time and temperature, safety issues,
1482 and environmental consequences should be carefully evaluated.

1483

1484 Secondly, the catalytic mechanisms of alkaline HER require more in-depth investigations.
1485 The *In situ* and operando spectroscopic and microscopic techniques are highly
1486 recommended to understand the catalytic phenomena at the catalyst surface region.
1487 Moreover, DFT calculations are valuable tools to unveil the catalytic processes at the
1488 atomic scale. The computational results can provide significant information about
1489 reaction energetics, reaction routes, intermediate evolution, and charge distributions etc.
1490 Additionally, the theoretical calculation plays a critical role in prediction and explanation

1491 of the catalytic performance of the designed materials. Under this circumstance, the
1492 rational design of model catalyst by DFT studies is an important step to match and guide
1493 the real catalysts in HER processes.

1494

1495 Thirdly, parameters for catalysts evaluation should be further standardized. To date,
1496 overpotentials at 10 mA cm⁻² (η_{10}) and Tafel slope are the most popular descriptors and
1497 used by almost all the reported papers. However, other factors are often documented
1498 randomly, such as the specific/mass activity, turnover frequency, stability, and onset
1499 potential. These issues result in unfair performance comparisons among different
1500 catalysts. Additionally, mass loading has a major influence on the evaluating parameters.

1501

1502 Finally, it is surprising to find that many of HER electrocatalysts reported above are great
1503 candidates for OER and/or oxygen reduction reaction (ORR). These multifunctional
1504 catalysts can further decrease the running cost and facilitate the experimental setups in
1505 overall water splitting and batteries. Hence, it is of great significance to analyse the
1506 origins of versatile abilities and offer valuable guidance for the design of future high-
1507 performance electrocatalysts.

1508

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1513

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