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Room Temperature Activation of Methane and Dry Reforming with CO₂ on Ni-CeO₂(111) Surfaces: Effect of Ce³⁺ Sites and Metal-Support Interactions on C-H bond Cleavage

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Abstract

The results of core-level photoemission indicate that Ni-CeO₂(111) surfaces with small or medium coverages of nickel are able to activate methane at 300 K producing adsorbed CH_x and CO_x ($x = 2,3$) groups. Calculations based on density-functional

theory predict relatively low activation energy of 0.6–0.7 eV for the cleavage of the first C–H bond in the adsorbed methane molecule. Ni and O centers of ceria work in a cooperative way in the dissociation of the C–H bond at room temperature where a low Ni loading is crucial for the catalyst activity and stability. The strong electronic perturbations in the Ni nanoparticles produced by the ceria support of varying nature such as stoichiometric and reduced, result in a drastic change in their chemical properties towards methane adsorption and dissociation as well as the DRM reaction. The coverage of Ni had a drastic effect on the ability of the system to dissociate methane and catalyze the dry reforming process.

keywords: Methane activation, dry reforming, ceria, nickel, supported catalysts, support effect, XPS, DFT

Introduction

Methane is the main component of natural gas and biogas produced by the decomposition of waste in many landfills.¹ The activation of methane by heterogeneous catalysts will therefore play a key role in solving future energy needs and as a carbon source for the production of chemicals.² In principle, the activation of methane is difficult due to the high stability of its C–H bonds.³ Enabling low-temperature activation of methane is a major technological objective. It is known that enzymes such as the methane monooxygenase⁴ and some copper- and zinc-based inorganic compounds^{5–7} can activate C–H bonds near room temperature. In a previous study, we found that a Ni-CeO₂ high-surface area powder catalyst was very efficient for performing the dry reforming of methane reaction (DRM, $\text{CH}_4 + \text{CO}_2 \rightarrow 2\text{CO} + 2\text{H}_2$) at a relatively low temperature of 700 K.⁸ In this article, we show that a small coverage of Ni dispersed on ceria can actually activate the methane molecule at 300 K. We investigate at a molecular-level the phenomena responsible for the observed low barrier for C–H bond cleavage on this metal/oxide system. It is shown that they are associated with

metal-support interactions which are extremely sensitive to the coverage of Ni on the ceria substrate.

In general, pure metal surfaces display a low reactivity towards methane.⁹ They need high temperatures to activate the molecule and are deactivated by the deposition of carbon and coke formation. The dispersion of metal nanoparticles on oxide surfaces is of great interest for developing novel catalytic materials, because the assembly could be more reactive than the individual components.^{10,11} In heterogeneous catalysis, ceria (CeO_2)-supported noble and late transition metals have attracted much attention particularly due to their excellent catalytic activity in carbon monoxide oxidation and the low-temperature water-gas shift reactions.^{12–15} Among the factors which are assumed to be influencing the activity like the metal-oxide contact structure and the particle size, the first one has typically been considered the most important, because the perimeter interfaces around the metal particles are expected to be part of the active site where reactions take place. Moreover, ceria, with its capability for the easy conversion between Ce^{4+} and Ce^{3+} oxidation states, often stabilizes oxidic $\text{M}^{\delta+}$ cations by accommodating electrons transferred from the metal to the support in localized f-states.^{16–22} The change in the oxidation state of the adsorbed metallic species, resulting from strong metal-ceria interactions, has also been considered crucial to the systems' reactivity.^{12,13,22–25} A good anchoring of the metal to the ceria favors catalytic activity and stability as a consequence of strong metal-support interactions.^{26–29} In addition the morphology of the ceria support has an important impact on the performance of the catalyst.^{26,27} The Ni- CeO_2 system has received a lot of attention in the area of methane dry reforming.^{8,26,27,30–32} Previous works were mainly focused on studies at temperatures above 750 K, where significant activity is seen for DRM, and no attention was paid to the effects of nickel coverage or the nature of the interactions between nickel and ceria.^{8,26,27}

Here, we investigate the activation of methane at room temperature on a series of Ni- $\text{CeO}_2(111)$ model catalysts using X-ray photoelectron spectroscopy (XPS) and density functional theory (DFT)-based modeling.

We compare with results on the bare Ni(111) surface and studies reported in the literature⁴⁻⁷ for the dissociation of methane on different types of inorganic compounds. At small nickel coverages, the atoms in a Ni-CeO₂ interface work in a cooperative way and are able to break C–H bonds at room temperature. These model Ni-CeO₂ systems are also very good catalysts for methane dry reforming at the low temperature of 650 K, as it has been observed for the real powder catalysts at similar temperatures.⁸

Experimental and Theoretical Methods

Experimental Methods

We investigated the effect of Ni coverage on the performance of Ni-CeO₂(111) surfaces for methane activation and the reforming of the molecule with CO₂. The experimental data for these well-defined catalysts were collected in a set-up that combined an ultra-high vacuum (UHV) chamber for surface characterization and a micro-reactor for catalytic tests.^{13,15,28,33} The UHV chamber was equipped with instrumentation for X-ray photoelectron spectroscopy (XPS), low-energy electron diffraction (LEED), ion-scattering spectroscopy (ISS), and thermal-desorption mass spectroscopy (TDS).^{13,15,28,33}

The methodology followed for the preparation of the Ni-CeO₂(111) surfaces is described in detail in refs 15,33 Ni was vapor-deposited on the ceria substrate at 300 K and the system was characterized with core and/or valence photoemission.^{15,33} In the studies of methane activation, the sample was transferred to the reactor at ~300 K, then the reactant gas at 1 Torr of pressure was introduced. In the experiments testing the activity of the Ni-CeO₂(111) catalysts for the DRM process, the samples were exposed to a mixture of CH₄ (1 Torr) and CO₂ (1 Torr) at 300 K and were rapidly heated to the reaction temperature of 650 K. Product yields were analyzed by mass spectroscopy or gas chromatography.^{15,33} In our experiments data were collected at intervals of 5 min. The amount of molecules (CO or H₂) produced in the catalytic tests was normalized by the active area exposed by the sample and

the total reaction time. The kinetic experiments were done in the limit of low conversion (< 10%).

The near-ambient pressure (NAP) XPS studies with the Ni-CeO₂(111) catalysts were performed at the Advanced Light Source in Berkeley, CA (beamline 9.3.2). The C 1s and Ce 4d regions were probed with a photon energy of 490 eV and a resolution of ~ 0.2 eV. The Ce 4d photoemission lines were used for binding energy calibration based on the 122.8 eV satellite features.

Models and Computational Details

We modeled the interaction of individual Ni atoms, exploring several adsorption sites and different spin multiplets, on a ceria support of varying nature such as stoichiometric, CeO₂(111), and reduced, CeO_{2-x}(111) and Ce₂O₃(0001). Moreover, we considered three-dimensional (3D) pyramidal Ni₄ clusters on the stoichiometric and fully reduced ceria supports. The adsorption and dissociation of methane, CH₄→CH₃+H, was investigated on selected Ni-ceria systems and compared with Ni(111).

Spin-polarized DFT and supercell periodic models were used within the Vienna ab initio simulation package (vasp site, <http://www.vasp.at>; version vasp.5.3.5).^{34,35} We treated explicitly the Ce (4f, 5s, 5p, 5d, 6s), O (2s, 2p), and Ni (4s, 3d) electrons as valence states expanded in plane-waves with a cutoff energy of 415 eV, whereas the remaining electrons were kept frozen as core states in the projector-augmented wave (PAW) method.³⁶ Strong correlation effects due to charge localization are modeled by adding a Hubbard U -like term³⁷ ($U_{\text{eff}} = U - J$, that is, the difference between the Coulomb U and exchange J parameters, hereinafter referred to as simply U) to the Perdew, Burke and Ernzerhof (PBE) generalized gradient approximation (GGA) functional.³⁸ We used a value of $U = 4.5$ eV for the Ce 4f states.^{39,40}

The CeO₂(111) and Ce₂O₃(0001) surfaces were modeled by (2×2) unit cells, with calculated ceria bulk equilibrium lattice parameters [$a_0 = 5.485$ Å (CeO₂); $a_0/c_0 = 3.92/6.18$ Å

and internal parameters $u_{\text{Ce}}/u_{\text{O}} = 0.2471/0.6448$ (hexagonal A-type, ferromagnetic, Ce_2O_3)].⁴¹ In the case of $\text{CeO}_2(111)$, we have considered nine atomic layers (three O-Ce-O trilayers, TL), whereas $\text{Ce}_2\text{O}_3(0001)$ was modeled with fifteen atomic layers (three O-Ce-O-Ce-O quintuple layers, QL). The models for the defective $\text{CeO}_{2-x}(111)$ surfaces ($\Theta = 1/4, 1/2, 3/4,$ and 1) correspond to the most stable structures in ref 42 with one surface or subsurface oxygen vacancy, as well as with two, three and four subsurface vacancies (cf. Figure S1 b-f). Interestingly, we found that the removal of all subsurface O atoms ($\Theta = 1$) is stabilized by ~ 0.3 eV upon a reconstruction with a change in the stacking of the second O layer. The resulting structure corresponds to 1 QL of hexagonal (A-type) Ce_2O_3 on $\text{CeO}_2(111)$ (cf. Figure S1 g). We also considered 2 QL/2 TL $\text{Ce}_2\text{O}_3/\text{CeO}_2(111)$ (Figures S1 h). The Ni(111) surface was modeled by a (3×3) unit cell (bulk lattice constant of 3.52 \AA) and five Ni layers. In all models, consecutive slabs were separated by at least 12 \AA of vacuum space to avoid interaction with periodic images. Monkhorst-Pack grids with a $(3 \times 3 \times 1)$ and $(5 \times 5 \times 1)$ k -point sampling were used for the ceria-based systems and Ni(111) surface, respectively.

In all geometry optimizations, all atoms in the bottom CeO_2 TL or Ce_2O_3 QL were fixed at their optimized bulk positions, whereas the rest of the atoms were allowed to fully relax. For the case of Ni(111), the two bottom atomic layers were kept fixed. Calculated local density of states and magnetic moments were inspected in order to determine oxidation states.

The adsorption of molecular and dissociated methane and the $\text{CH}_4 \rightarrow \text{CH}_3 + \text{H}$ reaction were studied on the most stable $\text{Ni}_n/\text{CeO}_2(111)$ and $\text{Ni}_n/\text{Ce}_2\text{O}_3(0001)$ systems ($n = 1$ and 4), and on Ni(111). All of the reaction energies were referenced to CH_4 as gas-phase species for which Γ -point calculations were performed in a $12 \times 12 \times 12 \text{ \AA}^3$ box. To locate the transition state (TS) structures we employed the climbing image nudged elastic band method (CI-NEB)⁴³ with nine images for each reaction pathway. For all the TS reported in this work, we have found only one imaginary frequency.

Results and Discussions

Interaction of Methane with Ni-CeO₂(111) at Room Temperature

Ni and ceria can form solid solutions of the Ce_{1-x}Ni_xO_{2-y} ($x < 0.15$) which exhibit a fluorite structure with Ni in a 2+ oxidation state.¹⁵ Scanning tunneling microscopy (STM) and XPS has been used to study the growth mode of nickel on CeO₂(111).^{15,44} At 300 K and coverages below 0.2 ML, there is a large dispersion of Ni on ceria with small particles that exhibit an average height of 0.23 nm and an average diameter of 1.80 nm. XPS indicates that most of the Ni is in a +2 oxidation state. Upon annealing to 500K, some of the Ni migrates into the ceria substrate to form a Ce_{1-x}Ni_xO_{2-y} solid solution. At Ni coverages above 0.3 ML, there is still a tendency towards inter-mixing but now Ni grows forming 3D particles on the ceria substrate.^{15,44} As we will see below, these two different morphologies of the Ni-CeO₂(111) surface have a strong impact in the chemical and catalytic properties.

Figure 1 shows C 1s XPS spectra collected after exposing the clean CeO₂(111) and Ni/CeO₂(111) surfaces with admetal coverages of 0.15 and 0.4 ML to methane at 300 K. On the clean oxide surface weak features appear near 290 eV which can be assigned to emissions from a Ce 4s level. There is also some signal in the 284–286 eV region which could come from the dissociation of a few methane molecules on defect sites of the oxide substrate. After pre-covering the surface with 0.15 ML of Ni, exposure to methane at 300 K leads to clear peaks near 290 and 285.5 eV. These peaks have also been detected after the reaction of alcohols with nickel-ceria and can be assigned to CO_x and CH_x species.⁴⁵ The features near 285.5 eV are the product of the partial dissociation of methane into CH₃ and/or CH₂ groups.⁴⁵

On highly active sites of the nickel-ceria surface, the methane undergoes complete decomposition: CH₄ → CH₃ → CH₂ → CH → C, and the formed C reacts with O centers of ceria to yield adsorbed CO₂ or CO₃ species.⁴⁵ The compounds that produced the features near 290 and 285.5 eV were weakly bound to the nickel-ceria surface and disappeared upon heating to

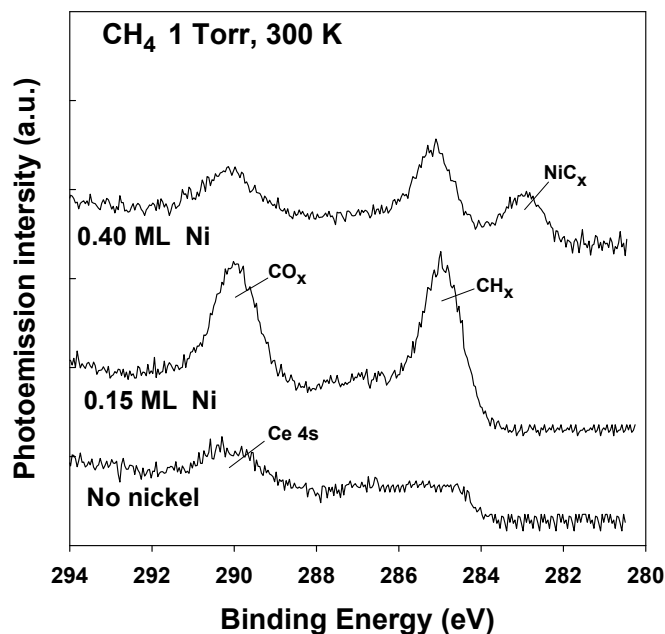


Figure 1: C 1s XPS spectra collected after exposing $\text{CeO}_2(111)$ and $\text{Ni/CeO}_2(111)$ surfaces with coverages of 0.15 and 0.4 monolayer (ML) to 1 Torr of methane at 300 K for 5 minutes.

temperatures above 400 K (Figure S2). The oxidation state of the nickel in $\text{Ni}_{0.15}/\text{CeO}_2(111)$ before and after adsorbing methane at room temperature was Ni^{2+} (Figure S3).

The surface with 0.4 ML of nickel (cf. Figure 1) exhibits a significant reduction in the intensity of the features for CO_x and CH_x species whereas a new peak appears near 282.5 eV which denotes the formation of NiC_x on the surface.^{45–47} At coverages above 0.5 ML, the feature for NiC_x was dominant in the C 1s region. Figure 2 shows the effect of nickel coverage on the C 1s signal for NiC_x and on the total signal including CH_x , CO_x , and NiC_x species.

The most important insight from Figure 1 is the presence of CH_x fragments on the surface of the catalyst at Ni coverages below 0.2 ML, when the Ni is present in the form of atoms or small particles in close contact with the ceria substrate. A drastic drop in the activation of methane is seen at Ni coverages above 0.2 ML. As the coverage of Ni increases, large 3D particles are formed on ceria and most of the admetal atoms are not able to activate methane at room temperature. They behave like atoms in extended surfaces of nickel.^{9,48}

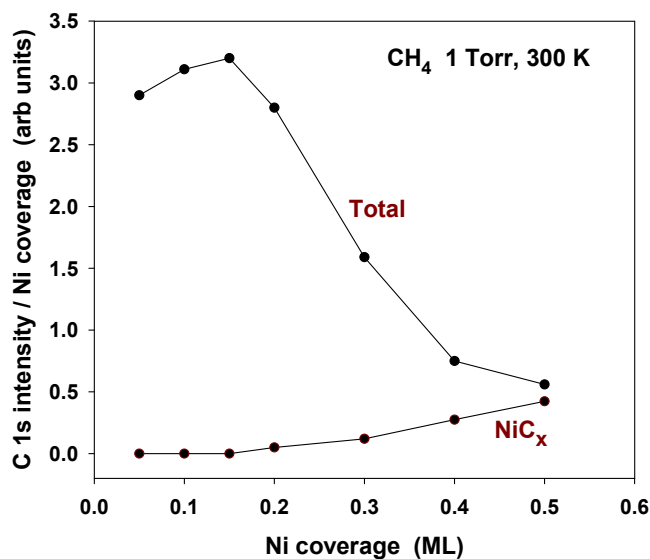


Figure 2: Effect of Ni coverage on the total C 1s intensity for CH_x , CO_x and NiC_x species. The C 1s intensities were normalized by the coverage of Ni on the surface. In the bottom of the figure is shown the normalized C 1s intensity for the NiC_x species only. The Ni/ $\text{CeO}_2(111)$ surfaces were exposed to 1 Torr of methane at 300 K for 5 minutes.

The Ni- $\text{CeO}_2(111)$ system shares with a few inorganic compounds⁵⁻⁷ the ability to activate the C–H bonds of methane at room temperature. This gives us the possibility of doing chemical transformations using the adsorbed CH_x fragments.

In a set of additional experiments, we explored the DRM process on the Ni- $\text{CeO}_2(111)$ systems at room temperature. We found activation of methane and plain chemisorption of CO_2 without achieving a catalytic cycle for the production of H_2 and CO. Catalytic activity for DRM was found at a temperature of 650 K, Figure 3. $\text{CeO}_2(111)$ did not display catalytic activity. The catalytic activity of the surface increased when nickel was added reaching a maximum at an admetal coverage of 0.15ML. These are the systems that display the highest activity in Figures 1 and 2 for the cleavage of C-H bonds in methane. At Ni coverages above 0.2 ML, there is a steady decline of the chemical and catalytic activity. At the same time post-reaction characterization of the catalysts with XPS showed an increase on the amount of NiC_x present on the surface, Figure 4. This is a bad sign, because the NiC_x species is not active for the reforming of hydrocarbons and can lead to coke formation.⁴⁵

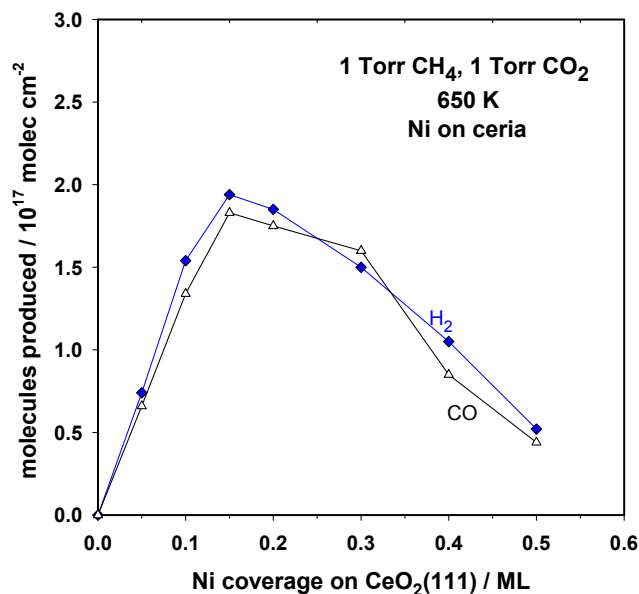


Figure 3: Catalytic activity for methane dry reforming of Ni-ceria surfaces as a function of nickel coverage. The figure reports the amount of H₂ and CO formed after exposing the nickel-ceria surfaces to 1 Torr of methane and 1 Torr of CO₂ at 650 K for 5 minutes.

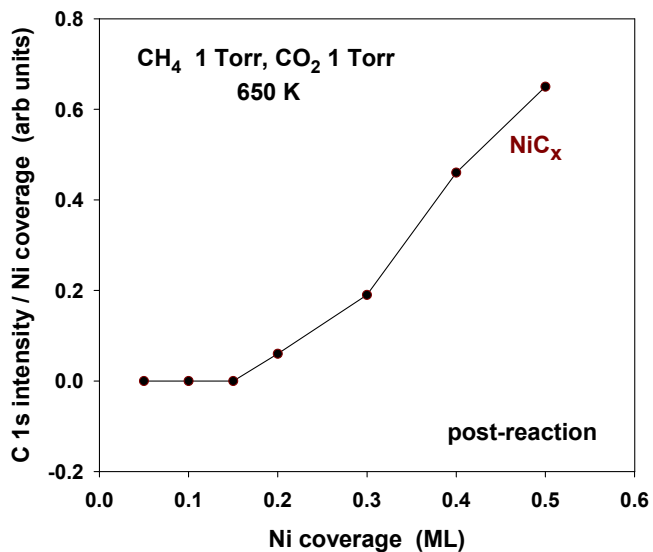
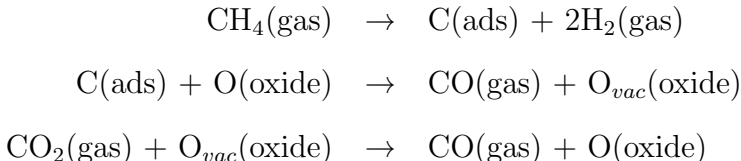


Figure 4: Effect of Ni coverage on the amount of NiC_x formed after exposing Ni-CeO₂(111) surfaces to 1 Torr of methane and 1 Torr of CO₂ at 650 K for 5 minutes. The C 1s intensity NiC_x was normalized by the coverage of Ni on the surface.

To gain a better understanding of the mechanism for DRM, we performed XPS experiments under near-ambient pressure. The study was done in a system with a low content of Ni, $\Theta_{\text{Ni}} \sim 0.1$ ML, to avoid NiC_x formation. The bottom of Figure 5 displays C 1s and Ce 4d photoemission spectra collected for a Ni-CeO₂(111) sample under a pressure of 100 mTorr of methane at 700 K. At this elevated temperature, there are no signals for CO_x or CH_x species on the surface. The C atoms produced by the full dissociation of methane react with O centers of the sample to produce CO that desorbs into gas phase. An analysis of the corresponding Ce 4d spectrum shows clear features for a reduced CeO_{2-x}(111) support. On the O vacancies of the system, the CO₂ molecules adsorb and dissociate yielding adsorbed O and CO gas to close a catalytic cycle:



In Figure 5, the adding of 100 mTorr of CO₂ to the reaction chamber produces a partial reoxidation in the Ce 4d region with the appearance of CO gas and an adsorbed CO_x species, peak at 291 eV, in the C 1s region.

Thus, from the results in Figures 5 and 6, we can conclude that the active phase in the DRM catalysts contained small particles of metallic Ni dispersed on partially reduced ceria. This nickel-ceria system operates as a DRM catalyst at a temperature that is much lower than that typically reported for metal oxide catalysts.¹ In its high-surface area powder form, it is highly active and stable (80 hours test).⁸ This is a consequence of synergistic interactions between the metal and oxide components of the system. In the next section, we use DFT based calculations to study the nature of these interactions and how they facilitate the activation of methane.

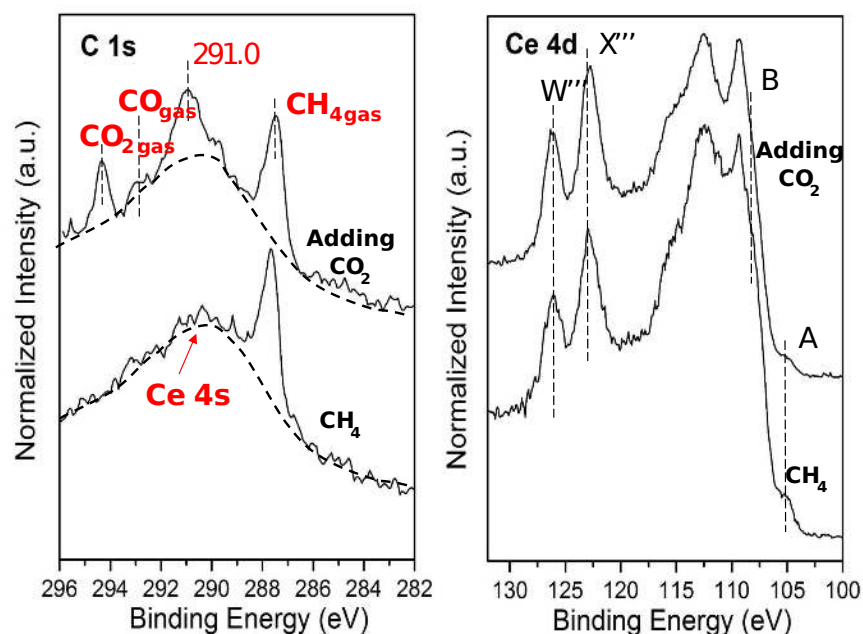


Figure 5: C 1s and Ce 4d NAP-XPS data collected while exposing a Ni-CeO₂(111) surface to 100 mTorr of methane at 700 K, spectra at the bottom, and after adding 100 mTorr of CO₂, spectra at the top. The Ni coverage on the ceria substrate was ~ 0.1 ML.

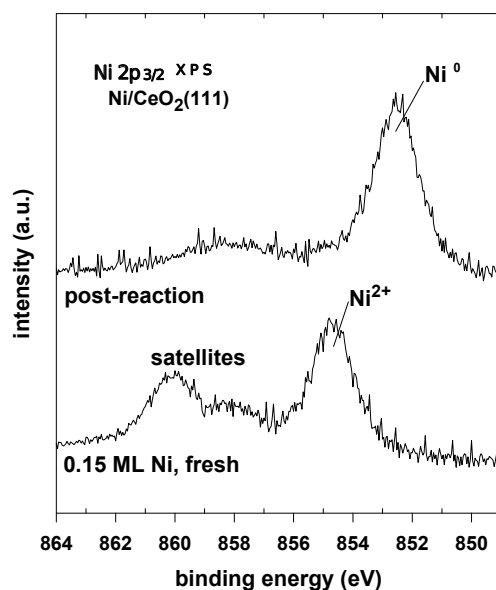


Figure 6: Ni 2p_{3/2} XPS spectra for a fresh, as prepared, Ni-CeO₂(111) catalyst and the same system after exposure to a mixture of 1 Torr of CH₄ and 1 Torr of CO₂ at 650 K under DRM reaction conditions.

Effect of Metal-Support Interactions on C–H Bond Cleavage

Interaction of Ni Species with $\text{CeO}_2(111)$, $\text{CeO}_{2-x}(111)$ and $\text{Ce}_2\text{O}_3(0001)$

An isolated Ni atom exhibits a $4s^2 3d^8$ electronic configuration. From the different possible adsorption sites on $\text{CeO}_2(111)$ (on-top, bridge, hollow) with varying total magnetization ($M = 0, 2,$ or 4) considered, we found that the O-hollow site where two Ni $4s$ electrons are transferred from Ni to the ceria support, yielding Ni^{2+} ($4s^0 3d^8$) and two Ce^{3+} species ($M = 4$), is the most stable site ($\text{Ni}^{2+}@O\text{-hollow}$), with -3.76 eV adsorption energy with respect to a Ni^0 atom in the gas phase (Figures 7 and 8), in agreement with previous work.²¹ Forcing a $M = 0$ state at this site, results in the spin flip of the transferred electrons with a negligible change in the adsorption energy (about 0.01 eV). The change in energy accompanying spin flips will be consistently small in all Ni/ceria systems considered. However, the location of the Ce^{3+} centers with respect to the Ni site in the Ni- CeO_2 systems, will have in general a larger effect (up to ~ 0.9 eV) (Table S2 and Figure S4).

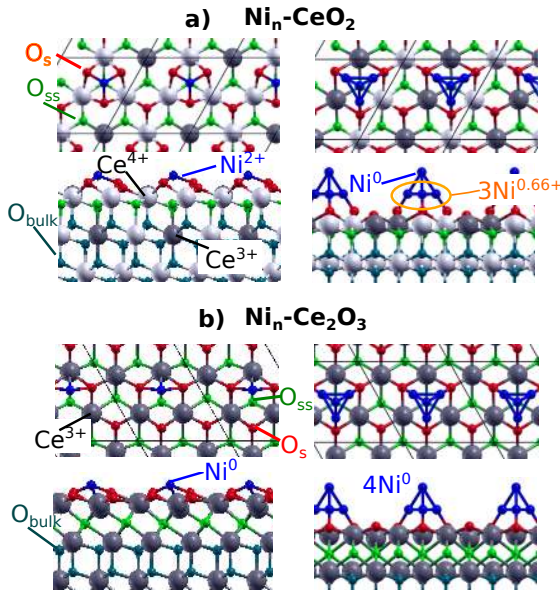


Figure 7: Adsorbed Ni atoms and pyramidal Ni_4 clusters on the $\text{CeO}_2(111)$ (a) and $\text{Ce}_2\text{O}_3(0001)$ (b) surfaces. Surface/subsurface oxygen atoms are depicted in red/green, Ce^{4+} in white and Ce^{3+} in gray. Those oxygen atoms in deeper layers are blue.

In addition, we found that Ni^{1+} ($4s^0 3d^9$, $M = 2$) species at the O-bridge site, resulting

from the transfer of only one electron from Ni to the support, and neutral Ni^0 ($4s^0 3d^{10}$, $M = 0$) species at the O-top, Ce-bridge, and Ce-top sites as well as Ni^0 ($4s^2 3d^8$, $M = 2$) species at the Ce-top site, are less stable than the Ni^{2+} @O-hollow site by about 0.4, 1.7, 1.7, 2.8, and 3.2 eV, respectively (Figure 8 and Table S2).

Thus, the electronic influence the $\text{CeO}_2(111)$ support exerts on highly dispersed Ni atoms results in oxidized Ni species with their most common oxidation state, +2. This is in line with the experimental observations that at 300 K and coverages below 0.2 ML, there is a large dispersion of Ni on $\text{CeO}_2(111)$ with most of the Ni in the +2 oxidation state.

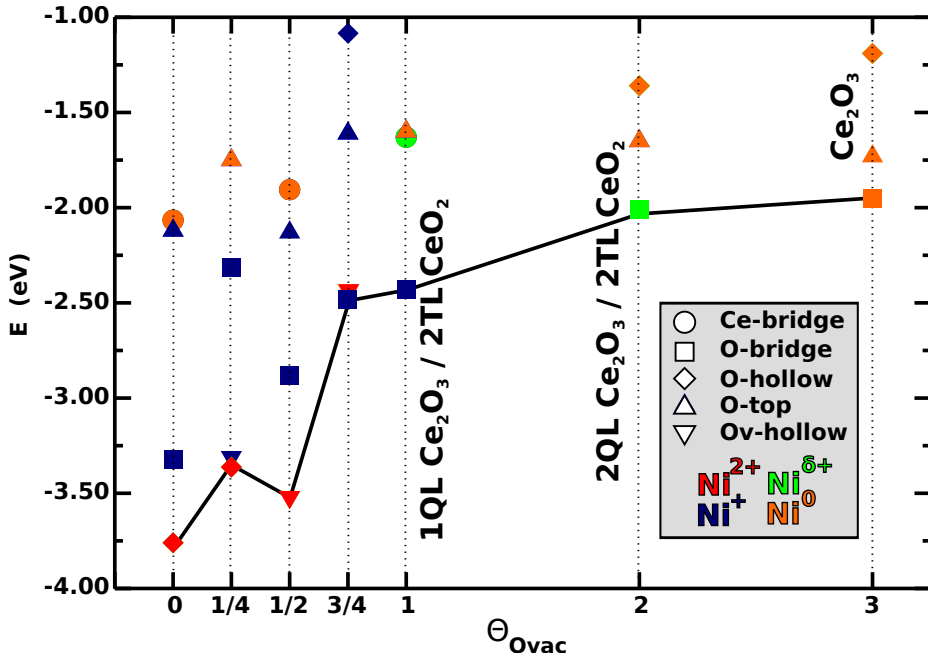


Figure 8: Adsorption energy of a Ni adatom at various sites on the $\text{CeO}_2(111)$ surface, $\text{CeO}_{2-x}(111)$ with subsurface oxygen vacancies ($\Theta = 1/4, 1/2$, and $3/4$), 1 and 2 QL $\text{Ce}_2\text{O}_3/2$ TL $\text{CeO}_2(111)$, and $\text{Ce}_2\text{O}_3(0001)$. The Ni oxidation state is color-coded ($0 < \delta < 1$).

Figure 8 shows that Ni adatoms gradually recover their metallic state, $\text{Ni}^{2+} \rightarrow \text{Ni}^{1+} \rightarrow \text{Ni}^{\delta+} \rightarrow \text{Ni}^0$, upon increasing oxygen removal from the ceria support, $\text{CeO}_2(111) \rightarrow \text{CeO}_{2-x}(111) \rightarrow \text{Ce}_2\text{O}_3(0001)$. This clearly reflects the synergistic interactions between nickel and the ceria support.

It is on the fully reduced $\text{Ce}_2\text{O}_3(0001)$ surface where no metal \rightarrow ceria charge transfer can take place and the most stable atomic Ni^0 ($4s^0 3d^{10}$, $M = 0$) species sit on a O-bridge site (Figure 7), with -1.95 eV adsorption energy (Figure 8). The O-top and O-hollow sites are

less stable than the Ni^0 @O-bridge site by about 0.2 and 0.8 eV, respectively (Figure 8 and Table S6).

Increasing the size of Ni nanoparticles to Ni_4 clusters with a pyramidal shape on the $\text{CeO}_2(111)$ and $\text{Ce}_2\text{O}_3(0001)$ surfaces, enables us to investigate the extent of the electronic influence of the ceria support in cases in which particles comprise atoms not in direct contact with it. The most stable site of a Ni_4 cluster on the stoichiometric $\text{CeO}_2(111)$ surface is on-top of a subsurface oxygen with -3.67 eV adsorption energy with respect to Ni^0 atoms in the gas phase (Figure 7), in agreement with previous work.²¹ These Ni_4 species also reduce the ceria support upon adsorption with the formation of two Ce^{3+} ions. These two electrons are transferred from the three Ni atoms forming the pyramid base, which are partially oxidized ($3\times\text{Ni}^{0.66+}$), whereas that at the top remains unaffected (Ni^0), reflecting a rapid decay of the metal-ceria interactions. Indeed, for a planar rombohedral Ni_4 species,²¹ two electrons are also transferred, but all Ni atoms are oxidized ($4\times\text{Ni}^{0.5+}$). Hence, at Ni coverages above 0.3 ML on $\text{CeO}_2(111)$, the 3D particles that are found to form,³⁰ comprise a large fraction of metal (Ni^0) atoms. Moreover, as in the case of the Ni adatoms, fully metallic pyramidal Ni_4 species ($4\times\text{Ni}^0$, Figure 7) are stabilized on the reduced $\text{Ce}_2\text{O}_3(0001)$ support.

Adsorption and Activation of Methane

The chemical bonding and associated charge transfer at the interface between the metal particles and the $\text{CeO}_2(111)$ support changes the chemical properties of the sites on the metal particles so that better catalytic effects towards methane activation can be achieved.

Figures 9a and d show the molecular binding of methane to isolated Ni^{2+} and partially oxidized larger Ni_4 species on the $\text{CeO}_2(111)$ surface with adsorption energies of approximately -0.2 eV. On $\text{Ni}^{2+}/\text{CeO}_2(111)$ the distance between Ni and the closest H atoms is increased by about 0.1 \AA as compared to that on the Ni_4 species. On both Ni-ceria systems, methane dissociation is hindered by a relatively low activation energy of $0.7\text{--}0.8$ eV (Figures 10a). In contrast, the molecular binding of methane to $\text{Ni}(111)$ is extremely weak,

decreasing the reaction probability, and dissociation is also difficult due to a larger energy barrier of about 0.9–1.1 eV.^{8,49–52} Therefore, the energy barriers of small Ni nanoparticles on CeO₂(111) are more accessible at room temperature than that of Ni(111), and methane dissociation is expected to occur, in agreement with the experiments shown in Figure 1.

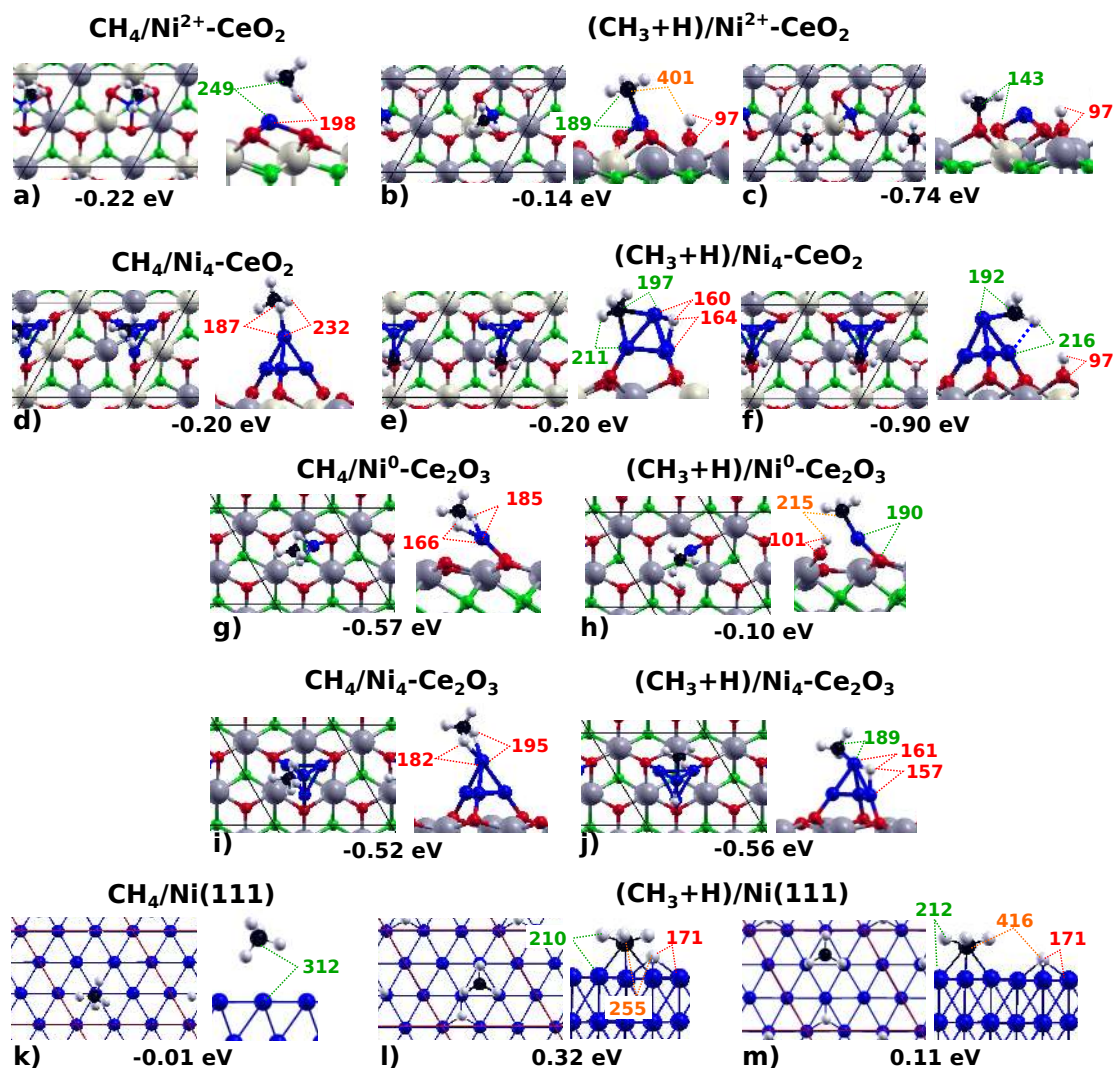


Figure 9: Top views of the most stable structures for methane adsorption on single Ni atoms (a-c) and Ni₄ clusters (d-f) on CeO₂(111) as well as Ni atoms (g-h) and a Ni₄ clusters (i-j) on Ce₂O₃(0001), and on Ni(111) (k-m). Adsorption energies are given with respect to gas-phase molecules in eV. Selected interatomic distances in pm are also given.

The dissociation products shown in Figure 10a indicate that for the dispersed Ni²⁺ species, Ni and O centers of ceria work in a cooperative way in the dissociation of the C–H bond. Methane adsorption sites with adjacent Ni²⁺ and surface O species do exist for

low Ni loadings (below 0.2 ML) for which a large dispersion of small Ni particles in direct contact with the CeO₂(111) support are observed.

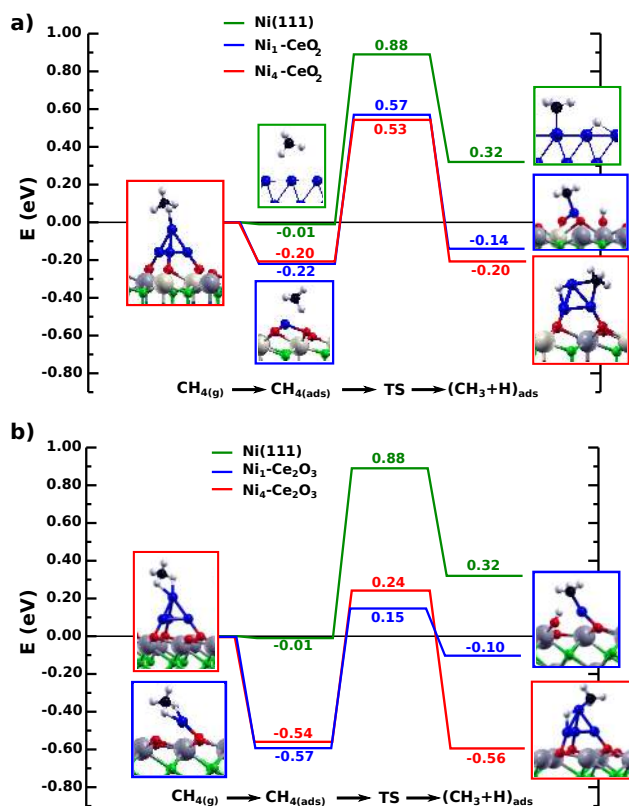


Figure 10: Reaction energy profile for the $\text{CH}_4 \rightarrow \text{CH}_3 + \text{H}$ reaction on isolated Ni atoms and Ni_4 clusters on the CeO₂(111) (a) and Ce₂O₃(0001) surfaces (b), compared to Ni(111). The structures shown on the left and right of the reaction pathways correspond to the side views of the optimized molecularly adsorbed and dissociated states used in the search of the transition state structure. All energies are relative to CH₄ in the gas phase.

Note that the final states shown in Figure 10a do not necessarily correspond to the lowest energy structures of the dissociated methane, but to stable states that are geometrically close to the transition state structures. For instance, on the Ni²⁺-CeO₂ system, a configuration with both the CH₃ methyl and the H atom adsorbed on-top of surface oxygen atoms is by 0.6 eV more stable than that with the CH₃ group on nickel and H forming a hydroxyl species (cf. Figures 9b and c). Also, on the Ni₄-CeO₂ system, diffusion of H to the ceria support further stabilizes the dissociated state by 0.7 eV (cf. Figures 9e and f).

Once CH₃ is formed, sequential decomposition of the methyl species into C should occur

very fast.⁵⁰⁻⁵² Indeed, after exposing the Ni-CeO₂(111) systems to methane at 300 K, the formation of CO_x species is observed (Figure 1). Yet, it is further observed that for low Ni loadings, below 0.2 ML, NiC_x does not form, whereas for higher loadings, for which 3D particles form, NiC_x also forms, and the activity towards methane dissociation drastically drops (Figure 2). The dependence on Ni loading of the ease with which NiC_x forms on Ni sites in the Ni-CeO₂ systems, is consistent with the calculated trend in the exothermicity of the nonoxidative adsorption of C atoms, namely, Ni²⁺/CeO₂(111) (-4.12) < Ni₄/CeO₂(111) (-6.54) ≈ Ni(111) (-6.78 eV), in agreement with previous work.²¹ That is, the 3D Ni₄ clusters behave like the extended surface with respect to the stabilization of C atoms, namely, strongly bound C species will stick to the surface of the existing 3D Ni particles at high Ni loading on CeO₂(111), resulting in the activity loss and ultimately in coke formation.

The strong electronic nickel-ceria interaction that is present at low nickel loadings yields not only active but also stable Ni-CeO₂ catalysts for methane activation at room temperature, enabling the use of the adsorbed CH_x compounds for chemical reactions. In addition, such nickel-ceria systems were also found to be active for CH₄ dry reforming with CO₂ at a relatively low temperature of ~ 700 K.⁸ At this temperature, as mentioned above, the C atoms resulting from the complete CH₄ dissociation, combine with lattice oxygen atoms producing CO that desorbs in the gas phase, leaving an oxygen vacancy behind. These vacancies help in the activation and dissociation of CO₂. Consistently, we find that the C/Ni²⁺-CeO₂(111) → CO/Ni²⁺-CeO_{2-x}(111) reaction on the Ni²⁺-CeO₂(111) system is highly exothermic $\Delta E = -3.58$ eV.²¹

Figures 5 and 6 show that under DRM reaction conditions, the Ni²⁺-CeO₂ → Ni⁰-CeO_{2-x} change occurs. In the previous section, we have shown that undoubtedly, as the reduction of the ceria support increases (Figure 8), the oxidation state of Ni particles in contact with the support changes from +2 to 0. The question arises of knowing what is the effect of this change on the methane activation barrier.

Figure 10b shows that the Ni⁰-Ce₂O₃(0001) systems have comparable barriers to those

of $\text{Ni}^{2+}\text{-CeO}_2(111)$ (Figure 10a). This means that the catalyst modifications induced by changing from room temperature to DRM operating conditions, can not result in an alteration of the activity towards C–H bond cleavage, in agreement with the experimental observations. Moreover, we notice that the chemisorbed methane molecules are more stable on $\text{Ni}^0\text{-Ce}_2\text{O}_3(0001)$ than on $\text{Ni}^{2+}\text{-CeO}_2$ by about 0.35 eV, which increases the probability for reaction on the actual DRM active system.

To gain insight into the change in the activity and stability of the active $\text{Ni}^0\text{-CeO}_{2-x}$ DRM catalysts with nickel loading (Figures 4 and 5), we newly calculate the binding of C atoms on the $\text{Ni}^0\text{-Ce}_2\text{O}_3(0001)$ model systems with highly dispersed Ni atoms and larger 3D Ni_4 clusters. As for the $\text{Ni}^{2+}\text{-CeO}_2(111)$ discussed above, we found a similar trend, $\text{Ni/Ce}_2\text{O}_3(0001) (-4.91) < \text{Ni}_4/\text{Ce}_2\text{O}_3(0001) (-5.92) \approx \text{Ni}(111) (-6.78 \text{ eV})$, in line with the observation that keeping low nickel loadings is crucial to avoid NiC_x formation that leads to deactivation during DRM reaction.

Conclusions

The experimental and theoretical results described above show that Ni-ceria catalysts with Ni atoms and small particles in direct contact with the ceria support are able to activate methane at room temperature. In this aspect, they match the activity reported for the methane monooxygenase enzyme⁴ and some copper- and zinc-based inorganic compounds.^{5–7} The Ni-ceria systems are also active for methane dry reforming with CO_2 at temperatures as low as 650 K.⁸ The systems that involved the smallest highly dispersed particles are the most efficient for methane activation and reforming as well as the most resistant to deactivation by NiC_x formation. Thus, selecting an appropriate Ni loading is vital when it comes to have a stable performance of the catalysts.

The activity and stability of the catalyst is associated with strong metal-support interactions that are more effective when Ni particles are small. The metal-support interactions

affect the charge transferred between the Ni and the ceria support which has a strong impact in the chemical and catalytic properties of the Ni-ceria systems. Small Ni particles on stoichiometric ceria experience large electronic perturbations, which result in a significant binding energy and a relatively low activation barrier for the cleavage of the first C–H bond in adsorbed methane molecules as compared to extended Ni surfaces, enabling methane activation at room temperature and the use of adsorbed CH_x species in chemical reactions. The nature of the support on which the Ni particles are fixed is important as Ni and O centers of ceria work in a cooperative way in the dissociation of the C–H bond at room temperature.

The methane C–H cleavage barrier also remains low upon changing the operating conditions from room temperature to those of methane dry reforming with CO_2 . This change produces a modification of the catalyst, $\text{Ni}^{2+}\text{-CeO}_2 \rightarrow \text{Ni}^0\text{-CeO}_{2-x}$, that does not affect the activation of methane but that of CO_2 . Oxygen vacancies formed on the reducible support strongly enhances the ability of the system to dissociate C–O bonds. Thus combining well dispersed small Ni particles that promote C–H activation with a support that allows for oxygen removal appears to be a winning approach for the DRM reaction.

Associated Content

Supporting Information

Reduced ceria and Ni/ceria models and Ni adsorption energies

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References

- (1) Pakhare, D.; Spivey, J. *Chem. Soc. Rev.* **2014**, *43*, 7813–7837.
- (2) Lavoie, J.-M. *Front. Chem.* **2014**, *2*, 81.
- (3) Wei, J.; ; Iglesia, E. *J. Phys. Chem. B* **2004**, *108*, 4094–4103.
- (4) Chan, S. I.; Yu, S. S.-F. *Acc. Chem. Res.* **2008**, *41*, 969–979.
- (5) Chan, S. I.; Lu, Y.-J.; Nagababu, P.; Maji, S.; Hung, M.-C.; Lee, M. M.; Hsu, I.-J.; Minh, P. D.; Lai, J. C.-H.; Ng, K. Y.; Ramalingam, S.; Yu, S. S.-F.; Chan, M. K. *Angew. Chem., Int. Ed. Engl.* **2013**, *52*, 3731–3735.
- (6) Xu, J.; Zheng, A.; Wang, X.; Qi, G.; Su, J.; Du, J.; Gan, Z.; Wu, J.; Wang, W.; Deng, F. *Chem. Sci.* **2012**, *3*, 2932–2940.
- (7) Grundner, S.; Markovits, M. A.; Li, G.; Tromp, M.; Pidko, E. A.; Hensen, E. J.; Jentys, A.; Sanchez-Sanchez, M.; Lercher, J. A. *Nature Commun.* **2015**, *6*, 7546.
- (8) Liu, Z.; Grinter, D. C.; Lustemberg, P. G.; Nguyen-Phan, T.-D.; Zhou, Y.; Luo, S.; Waluyo, I.; Crumlin, E. J.; Stacchiola, D. J.; Zhou, J.; Carrasco, J.; Busnengo, H. F.; Ganduglia-Pirovano, M. V.; Senanayake, S. D.; Rodriguez, J. A. *Angew. Chem., Int. Ed. Engl.* **2016**, *55*, 7455–7459.
- (9) Choudhary, T. V.; Aksoylu, E.; Goodman, D. W. *Catal. Rev.* **2003**, *45*, 151–203.

- (10) Freund, H.-J.; Pacchioni, G. *Chem. Soc. Rev.* **2008**, *37*, 2224–2242.
- (11) Haruta, M. *CATTECH* **2002**, *6*, 102–115.
- (12) Flytzani-Stephanopoulos, M.; Saltsburg, H.; Fu, Q. *Science* **2003**, *301*, 935–938.
- (13) Rodriguez, J.; Liu, P.; Hrbek, J.; Evans, J.; Pérez, M. *Angew. Chem., Int. Ed. Engl.* **2007**, *46*, 1329–1332.
- (14) Cargnello, M.; Doan-Nguyen, V. V. T.; Gordon, T. R.; Diaz, R. E.; Stach, E. A.; Gorte, R. J.; Fornasiero, P.; Murray, C. B. *Science* **2013**, *341*, 771–773.
- (15) Senanayake, S. D.; Evans, J.; Agnoli, S.; Barrio, L.; Chen, T.-L.; Hrbek, J.; Rodriguez, J. A. *Top. Catal.* **2011**, *54*, 34–41.
- (16) Hernandez, N. C.; Grau-Crespo, R.; de Leeuw, N. H.; Sanz, J. F. *Phys. Chem. Chem. Phys.* **2009**, *11*, 5246–5252.
- (17) Camellone, M. F.; Fabris, S. *J. Am. Chem. Soc.* **2009**, *131*, 10473–10483.
- (18) Zhang, C.; Michaelides, A.; Jenkins, S. J. *Phys. Chem. Chem. Phys.* **2011**, *13*, 22–33.
- (19) Nolan, M. *J. Chem. Phys.* **2012**, *136*.
- (20) Bruix, A.; Neyman, K. M.; Illas, F. *J. Phys. Chem. C* **2010**, *114*, 14202–14207.
- (21) Carrasco, J.; Barrio, L.; Liu, P.; Rodriguez, J. A.; Ganduglia-Pirovano, M. V. *J. Phys. Chem. C* **2013**, *117*, 8241–8250.
- (22) Bruix, A. et al. *Angew. Chem., Int. Ed. Engl.* **2014**, *53*, 10525–10530.
- (23) Rodriguez, J. A.; Ma, S.; Liu, P.; Hrbek, J.; Evans, M., J. and Pérez *Science* **2007**, *318*, 1757–1760.
- (24) Rodriguez, J.; Graciani, J.; Evans, J.; Park, J.; Yang, F.; Stacchiola, D.; Senanayake, S.; Ma, S.; Pérez, M.; Liu, P.; Sanz, J.; Hrbek, J. *Angew. Chem., Int. Ed. Engl.* **2009**, *48*, 8047–8050.
- (25) Park, J. B.; Graciani, J.; Evans, J.; Stacchiola, D.; Ma, S.; Liu, P.; Nambu, A.; Sanz, J. F.; Hrbek, J.; Rodriguez, J. A. *Proc. Natl. Acad. Sci. U. S. A.* **2009**, *106*, 4975–4980.
- (26) Wang, N.; Qian, W.; Chu, W.; Wei, F. *Catal. Sci. Technol.* **2016**, *6*, 3594–3605.

- (27) Du, X.; Zhang, D.; Shi, L.; Gao, R.; Zhang, J. *J. Phys. Chem. C* **2012**, *116*, 10009–10016.
- (28) Carrasco, J.; López-Durán, D.; Liu, Z.; Duchoň, T.; Evans, J.; Senanayake, S. D.; Crumlin, E. J.; Matolín, V.; Rodríguez, J. A.; Ganduglia-Pirovano, M. V. *Angew. Chem., Int. Ed. Engl.* **2015**, *54*, 3917–3921.
- (29) Bruix, A.; Rodriguez, J. A.; Ramírez, P. J.; Senanayake, S. D.; Evans, J.; Park, J. B.; Stacchiola, D.; Liu, P.; Hrbek, J.; Illas, F. *J. Am. Chem. Soc.* **2012**, *134*, 8968–8974, PMID: 22563752.
- (30) Hahn, K. R.; Seitsonen, A. P.; Iannuzzi, M.; Hutter, J. *ChemCatChem* **2015**, *7*, 625–634.
- (31) Xie, T.; Zhao, X.; Zhang, J.; Shi, L.; Zhang, D. *Int. J. Hydrogen Energy* **2015**, *40*, 9685 – 9695.
- (32) Zhao, X.; Li, H.; Zhang, J.; Shi, L.; Zhang, D. *Int. J. Hydrogen Energy* **2016**, *41*, 2447 – 2456.
- (33) Xu, W.; Liu, Z.; Johnston-Peck, A. C.; Senanayake, S. D.; Zhou, G.; Stacchiola, D.; Stach, E. A.; Rodriguez, J. A. *ACS Catal.* **2013**, *3*, 975–984.
- (34) Kresse, G.; Hafner, J. *Phys. Rev. B* **1993**, *47*, 558–561.
- (35) Kresse, G.; Furthmüller, J. *Phys. Rev. B* **1996**, *54*, 11169–11186.
- (36) Kresse, G.; Joubert, D. *Phys. Rev. B* **1999**, *59*, 1758–1775.
- (37) Dudarev, S.; Botton, G.; Savrasov, S.; Humphreys, C.; Sutton, A. *Phys. Rev. B* **1998**, *57*, 1505–1509.
- (38) Perdew, J.; Burke, K.; Ernzerhof, M. *Phys. Rev. Lett.* **1996**, *77*, 3865–3868.
- (39) Fabris, S.; de Gironcoli, S.; Baroni, S.; Vicario, G.; Balducci, G. *Phys. Rev. B* **2005**, *71*, 041102.
- (40) Cococcioni, M.; de Gironcoli, S. *Phys. Rev. B* **2005**, *71*, 035105.
- (41) Da Silva, J.; Ganduglia-Pirovano, M.; Sauer, J.; Bayer, V.; Kresse, G. *Phys. Rev. B* **2007**, *75*, 045121.

- (42) Murgida, G. E.; Ganduglia-Pirovano, M. V. *Phys. Rev. Lett.* **2013**, *110*, 246101.
- (43) Henkelman, G.; Uberuaga, B. P.; Jónsson, H. *J. Chem. Phys.* **2000**, *113*, 9901–9904.
- (44) Zhou, Y.; Zhou, J. *J. Phys. Chem. C* **2012**, *116*, 9544–9549.
- (45) Liu, Z.; Duchoň, T.; Wang, H.; Peterson, E. W.; Zhou, Y.; Luo, S.; Zhou, J.; Matolín, V.; Stacchiola, D. J.; Rodriguez, J. A.; Senanayake, S. D. *J. Phys. Chem. C* **2015**, *119*, 18248–18256.
- (46) Kovács, G.; Bertóti, I.; Radnóczy, G. *Thin Solid Films* **2008**, *516*, 7942 – 7946.
- (47) Czekaj, I.; Loviat, F.; Raimondi, F.; Wambach, J.; Biollaz, S.; Wokaun, A. *Appl. Catal., A* **2007**, *329*, 68 – 78.
- (48) Yuan, K.; Zhong, J.-Q.; Zhou, X.; Xu, L.; Bergman, S. L.; Wu, K.; Xu, G. Q.; Bernasek, S. L.; Li, H. X.; Chen, W. *ACS Catal.* **2016**, *6*, 4330–4339.
- (49) Abild-Pedersen, F.; Lytken, O.; Engbæk, J.; Nielsen, G.; Chorkendorff, I.; Nørskov, J. K. *Surf. Sci.* **2005**, *590*, 127 – 137.
- (50) Nave, S.; Tiwari, A. K.; Jackson, B. *J. Chem. Phys.* **2010**, *132*, 054705:1–12.
- (51) Jiang, B.; Liu, R.; Li, J.; Xie, D.; Yang, M.; Guo, H. *Chem. Sci.* **2013**, *4*, 3249–3254.
- (52) Li, J.; Croiset, E.; Ricardez-Sandoval, L. *Chem. Phys. Lett.* **2015**, *639*, 205 – 210.

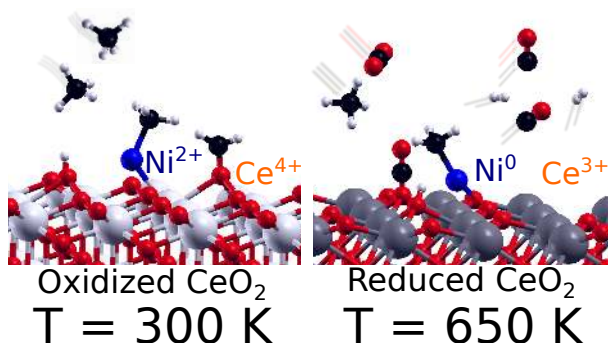


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