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Single step electrodeposition of superhydrophobic black NiO thin films

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Highlights

- Electrodeposition of superhydrophobic black NiO films on Cu is studied.
- Superhydrophobicity is attributed to the formation of a rough surface.
- An improvement of the corrosion resistance of Cu sub-layers is observed.

Abstract

Black finished surfaces have extensive applications in many domains, such as optics, solar cells, and aerospace. The single step electrodeposition of superhydrophobic black NiO films from a dimethyl sulfoxide based electrolyte is described in this paper. The physicochemical properties of the obtained film were characterized using Scanning Electron Microscopy, X-ray Diffraction, and electrochemical tests (Electrochemical Impedance Spectroscopy and potentiodynamic polarization). A rough surface with a low reflection of light was formed after the deposition process that increased the contact angle of water from about 87° (for bare Cu) to 163° (in presence of the black coating), which improved the corrosion resistance of the Cu substrate by about 30%. The formed black NiO film revealed a notably high stability and kept its appearance even after corrosion tests.

Keywords: Black NiO, Superhydrophobic, Corrosion, Electrodeposition, thin films

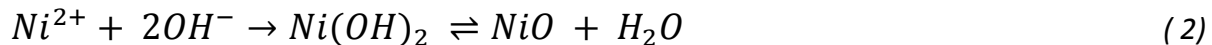
1. Introduction

Black finished surfaces are useful for optical instruments, solar cells, black decorative coatings, and for defense and aerospace industries [1,2]. There are several studies regarding black finished surfaces, including black Si, black Cr, and black Ni [2–4]. Black Ni surfaces are normally obtained by etching a low P content Ni-P (1-4 wt.% P) in an oxidizing acid [5,6]. Etching forms several cracks on the surface of the Ni-P film. These cracks, providing multiple reflection and absorption sites for incident light, are responsible for the black aspect of the surface [7]. However, the film also becomes very vulnerable to corrosion due to these cracks

1 [1]. Furthermore, blackening the surface by chemical etching induces a huge
2 material loss that is not economical [8].

3 The corrosion of metals can be effectively blocked by decreasing the contact of the
4 surface with the corrosive media. Superhydrophobic materials, i.e. surfaces with a
5 water contact angle larger than 150° [9], reduce the corrosion, because the contact
6 between the oxidizing media and the surface of the sample is reduced. Therefore,
7 superhydrophobic films are known to offer an outstanding corrosion resistance
8 [10–12]. Superhydrophobic films also offer self-cleaning, anti-fouling, and frost
9 prevention properties. Superhydrophobicity can be obtained by modification of a
10 surface owing to the surface energy decrease and the formation of a rough
11 hierarchical surface morphology [13].

12 NiO is formed by oxidation at the surface of Ni-P during chemical etching. NiO is
13 a p-type semi-conducting oxide with a fcc crystalline structure exhibiting a good
14 chemical stability [14,15]. Several methods including annealing [17], sol-gel
15 chemistry [18], ionic layer adsorption and reaction (SILAR) [19], and
16 electrodeposition [20] have been investigated to form NiO. Electrodeposition has
17 various advantages such as the ability to coat large surfaces, a homogenous
18 deposition, a controllable morphology and thickness, high adhesion to the
19 substrate, and few problems related to inter-diffusion [20]. Koussi *et al.* [20] have
20 successfully electrodeposited NiO from a dimethyl sulfoxide (DMSO) based
21 electrolyte. The mechanism of the electrodeposition of NiO in DMSO was
22 described by the following equations in the absence (1) and in the presence of a
23 low content of water (2), respectively:



27
28 In this paper, we report for the first time the single-step electrodeposition of a
29 black superhydrophobic NiO coating from a DMSO-based electrolyte. The coating
30 was characterized using Scanning Electron Microscopy (SEM), grazing incidence
31 X-Ray Diffraction (grazing XRD), Water Contact Angle measurements (WCA),
32 Electrochemical Impedance Spectroscopy (EIS) and potentiodynamic polarization
33 tests.

34
35 **2. Experimental procedure**

1 The electrodeposition electrolyte was a DMSO based solution at 50°C, containing
2 190 g.L⁻¹ NiSO₄. 6H₂O, 10 g.L⁻¹ NiCl₂. 6H₂O, 8 g.L⁻¹ H₃PO₃, 16 g.L⁻¹ H₃BO₃, and
3 5 mL.L⁻¹ H₃PO₄. All the chemicals, provided by Sigma-Aldrich, were highly pure
4 and used as received.

5 Cyclic voltammetry and potentiostatic electrodeposition experiments were
6 performed with a potentiostat/galvanostat (VersaSTAT 3). Before the deposition,
7 the Cu substrate was pre-treated according to reference [21]. A common 3-
8 electrode setup was used consisting of the Cu substrate as working electrode, a Pt
9 plate as counter electrode, and an Ag/AgCl (KCl saturated) as reference electrode.
10 For cyclic voltammetry tests, the potential was gradually decreased from 0 V vs.
11 the reference electrode, to -1.5 V and then increased to 0 V. The scan rate was 20
12 mV.s⁻¹. The electrodeposition was achieved by chronoamperometry. A constant
13 potential of -1V vs. the reference electrode was applied on the working electrode
14 for 1200 s, while the solution was stirred with a speed of 650 rpm.

15 Two scanning electron microscopes (Philips XL 30 ESEM, and CARL
16 ZEISS/Ultra 55) were used to study the surface and cross-section morphologies,
17 respectively. The chemical composition of the films was investigated by Energy
18 Dispersive Spectroscopy (EDS).

19 The sample reflectance was recorded using an UV-visible spectrophotometer
20 (Varian Cary 300) equipped with an integrating sphere DRA-CA-30I in the
21 380–800 nm range. Measurements were performed 4 times for each sample by
22 rotating the sample holder in order to probe the whole surface of the samples.

23 A Siemens D5000 diffractometer was employed to investigate the crystalline
24 structure. XRD patterns were obtained over the 2θ range of 30–60° (with 0.04°
25 step size) using Cu Kα radiation (λ = 0.15406 nm) generated at 30 mA and 40 kV.

26 A tensiometer (Adimec MX12P) was used to do the water contact angle
27 measurements (2 μL droplets) at five different locations. Drop analysis LB-ADSA
28 plugin in ImageJ software was used to measure the contact angle [22].

29 The corrosion resistance of the films was studied after 60 minutes immersion in a
30 3% NaCl solution by EIS and Tafel tests. The EIS measurements were done at
31 Open Circuit Potential (OCP) in a frequency range from 100 kHz to 10 mHz with
32 10 mV peak-to-peak voltage amplitude. Zview was used to analyze the obtained
33 EIS data. Potentiodynamic polarization experiments were carried out at a scan rate

1 of 0.5 mV/s from -300 mV (vs. OCP) to 500 mV (vs. reference electrode). The
2 results were normalized by the exposed area to the corrosive media.

3 4 **3. Results and Discussion**

5 • Cathodic electrodeposition

6 The cyclic voltammetry curves of the cathodic electrodeposition of Ni from the
7 DMSO based electrolyte are presented in Figure 1. It can be seen that the current
8 density at -1 V is relatively low. Such a low value makes it kinetically favorable
9 for the nucleation and growth to occur at preferential sites. Such type of 3D
10 “island” growth (described by the Volmer-Weber model) is known to form a rough
11 deposit. Figure 2 depicts the current density during the deposition of the film in
12 DMSO at -1 V. The current density for NiO deposition from DMSO is notably
13 lower than previously reported for Ni deposition from an aqueous electrolyte [21].
14 This observation points to the higher resistivity and lower throwing power in
15 DMSO than in aqueous electrolytes, leading to a reduced deposition rate.

16 • Surface and cross-section morphology

17 Figure 3 shows the surface and cross-section morphologies of the black coating. In
18 Figure 3A, the black appearance of the coating is obvious. The SEM image
19 presents a surface made of several hierarchical island shape features that are
20 regularly distributed over the surface. Such a morphology (that is similar to the
21 known lotus leave morphology [23]) highly decreases the light reflection (Figure
22 4) and thus makes the surface look black. The increase in the reflection of the Cu
23 substrate at around 550 nm is due to the natural color of Cu. Moreover, such a
24 morphology offers a high roughness that increases the water contact angle (see
25 below). The cross-section shows several well-distributed nanometric island shape
26 features on the surface. Moreover, the formed black film was notably adhesive,
27 since no cracks or delamination was observed at the interface between Cu and the
28 coating.

29 • Chemical composition and structure

30 The chemical composition of the film is summarized in Table 1. A film composed
31 of a mixture of NiO and Ni-P is formed on pure Cu. The high Cu content is due to
32 the low thickness of the black film. The presence of S and C in the chemical
33 composition is due to the incorporation of DMSO inside the film [20].

1 Figure 5 shows the grazing incidence XRD pattern of the black film. The strong
2 sharp peak at 43.5 ° is attributed to the NiO (200) reflection (NiO presents a NaCl-
3 type structure) [14,16] that shows the preferential growth of the film. The absence
4 of Ni peaks can be attributed to the amorphous nature of the Ni-P film. The
5 average grain size D was estimated to be around 100 nm using the Debye Scherrer
6 formula [14],

$$7 \quad D = \frac{0.9\lambda}{\beta \cos\theta} \quad (3)$$

8 where λ is the X-ray wavelength, β is the full width at half maximum of the peak,
9 and θ is the diffraction angle. The grain size of the black film is notably bigger
10 than that previously reported elsewhere [14,16,24]. The formation of bigger grains
11 might be related to the low deposition rate from DMSO.

12 • Water contact angle

13 The water contact angle of the Cu substrate and the black film are shown in Figure
14 6. The contact angle values for the Cu substrate and the film were $83 \pm 6^\circ$ and 161
15 $\pm 3^\circ$, respectively. Therefore, applying the black NiO film changed the hydrophilic
16 Cu surface to a superhydrophobic surface.

17 The hydrophobicity of these films was previously found to be due to the roughness
18 of the surface [25]. The superhydrophobicity of the surface comes from the
19 heterogeneous wetting, which means that the trapped air prevents the liquid from
20 penetration [26]. This can be expressed by the following equation:

$$21 \quad \cos \theta^* = f(1 + \cos \theta) - 1 \quad (4)$$

22 Where f is the area fraction of the droplet in contact with the solid (and thus 1-f is
23 the area fraction of the droplet in contact with the trapped air), and θ^* and θ are the
24 contact angles of the rough coating and the substrate (or flat surface). The f value
25 calculated for the black NiO film is 0.049: it has a very low wetting ability, notably
26 lower than some previously reported values for superhydrophobic Ni films (0.25
27 [27], and 0.13 [28]).

28 • Corrosion resistance

29 A typical impedance spectrum of the black NiO film is depicted in Figure 7. The
30 film presents one time constant in its EIS data. The Randles model, which is the
31 typical model to fit the EIS data of electrodeposited superhydrophobic NiO films
32 [12,25,29], was employed to analyze the black NiO film. A Constant Phase

1 Element (CPE) was used to describe the non-ideal behavior of the interface. The
2 impedance of a CPE is defined as:

$$3 \quad Z_{CPE} = \frac{1}{Y_0(i\omega)^n} \quad (5)$$

4 Y_0 is the constant of admittance, i is the imaginary unit, ω is the angular frequency,
5 and n is the CPE exponent [30]. n is also sometimes considered as the roughness
6 factor since it is affected by the surface roughness [31]. The analyzed data of the
7 black NiO film and Cu substrate (that is reported in [21]) are presented in Table 2.
8 Note that the Cu substrate had an extra Warburg element in its equivalent circuit
9 that expressed the diffusion-limitation of corrosion. The value of n for the Cu
10 substrate and the black NiO film were 0.80 and 0.76, respectively. The n value for
11 the substrate is low due to the presence a Warburg element. This observation is
12 pointing out that the black NiO has a rougher surface. The modulus Y° is higher
13 than the value expected for a “metallic” electrode. Rough surfaces have a higher
14 specific capacitance than flat ones, and increasing the roughness of thin films will
15 enhance their capacitance [32]. Moreover, electrodeposited NiO has been shown to
16 have a high pseudocapitance, useful for supercapacitor applications [33–35],
17 attributed to some contribution of chemical capacitances.

18 Applying the thin black NiO film increased the corrosion resistance by about 25%
19 (8.1 vs. 6.1 kΩ.cm² for NiO and Cu, respectively).

20 Figure 8 shows the polarization curves performed on the black NiO film and the
21 bare Cu substrate. The Tafel extrapolation method was used in a potential range of
22 ± 250 mV around OCP, to obtain the corrosion potential (E_{corr}), corrosion current
23 density (i_{corr}), and anodic and cathodic slopes (β_a and β_c). The Stern-Geary equation
24 was used to calculate the polarization resistance [36]. Table 3 summarizes these
25 data for the black NiO film.

$$26 \quad R_p = \frac{\beta_a \beta_c}{2.303(\beta_a + \beta_c) i_{corr}} \quad (6)$$

27 In comparison with the reported data for the Cu substrate [21], the black NiO film
28 notably shifted the corrosion potential (+155 mV) to more anodic regions. This
29 means a significant decrease in corrosion inclination [37]. The polarization
30 resistance (6.4 and 9.5 kΩ for Cu and black NiO, respectively) shows about 33%
31 increase, in good agreement with EIS results. The black NiO coating presents
32 especially a lower current density at high potentials showing its high chemical
33 stability.

1 Figure 9 shows that the surface morphology of the black NiO after the corrosion
2 tests is clearly unchanged. The chemical composition of the surface after the
3 corrosion tests is presented in Table 1. A comparison of the chemical composition
4 of the black NiO film before and after the corrosion tests reveals that the amount of
5 O increased, while the content of Ni decreased. A part of the Ni of Ni-P was
6 dissolved, so the surface was enriched with other components (i.e. O and S), in
7 agreement with the common corrosion mechanism of Ni films [37]. Therefore, the
8 superhydrophobicity of the electrodeposited black NiO film strengthens the
9 chemical stability and substantially improves the corrosion resistance.

10

11 **4. Conclusions**

12 The single step electrodeposition of superhydrophobic black NiO coatings from a
13 DMSO based electrolyte is presented in this paper. A very rough surface
14 containing many well-distributed island shaped features is formed that decreases
15 the light reflection and increases the water contact angle.

16 The corrosion resistance increases by about 30% due to the superhydrophobicity of
17 the surface that limits the contact between the surface and corrosive media. The
18 effect of the deposition parameters on the properties of the black NiO film and the
19 possibility to form it on other substrates, such as Al or Ti, is interesting for
20 aerospace applications and will be the subject of a future study.

21

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26

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2 **Tables**

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4 *Table 1. Chemical composition of the black coating before and after corrosion tests*

Element	Before corrosion tests: Wt. %	After corrosion tests: Wt. %
C	3.6	3.2
O	11.0	13.9
P	2.6	2.8
S	7.8	8.8
Ni	61.8	56.2
Cu	13.2	15.1

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7 *Table 2. EIS fit values of the black NiO film and Cu substrate*

Sample	R_s ($\Omega.cm^2$)	CPE_{dl}		R_{ct} ($k\Omega.cm^2$)	W		
		Q ($\mu F.cm^{-2}.s^{n-1}$)	n		W-R ($k\Omega.cm^2$)	W-T s	W-P
Black NiO	7.5	578.2	0.76	8.1	-	-	-
Cu Sub. [21]	20.3	129.8	0.80	0.02	6.1	165.9	0.43

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10 *Table 3. Corrosion current density, corrosion potential, anodic and cathodic slopes, and polarization resistance of the black NiO film*

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Sample	i_{corr} ($\mu A.cm^{-2}$)	E_{corr} (mV) vs. Ag/AgCl	Average Tafel slope (mV. dec ⁻¹)		R_p ($k\Omega.cm^2$)
			β_a	β_c	
Black NiO	1.64	-45	44	149	9.5
Cu Sub. [21]	3.32	-200	66	195	6.4

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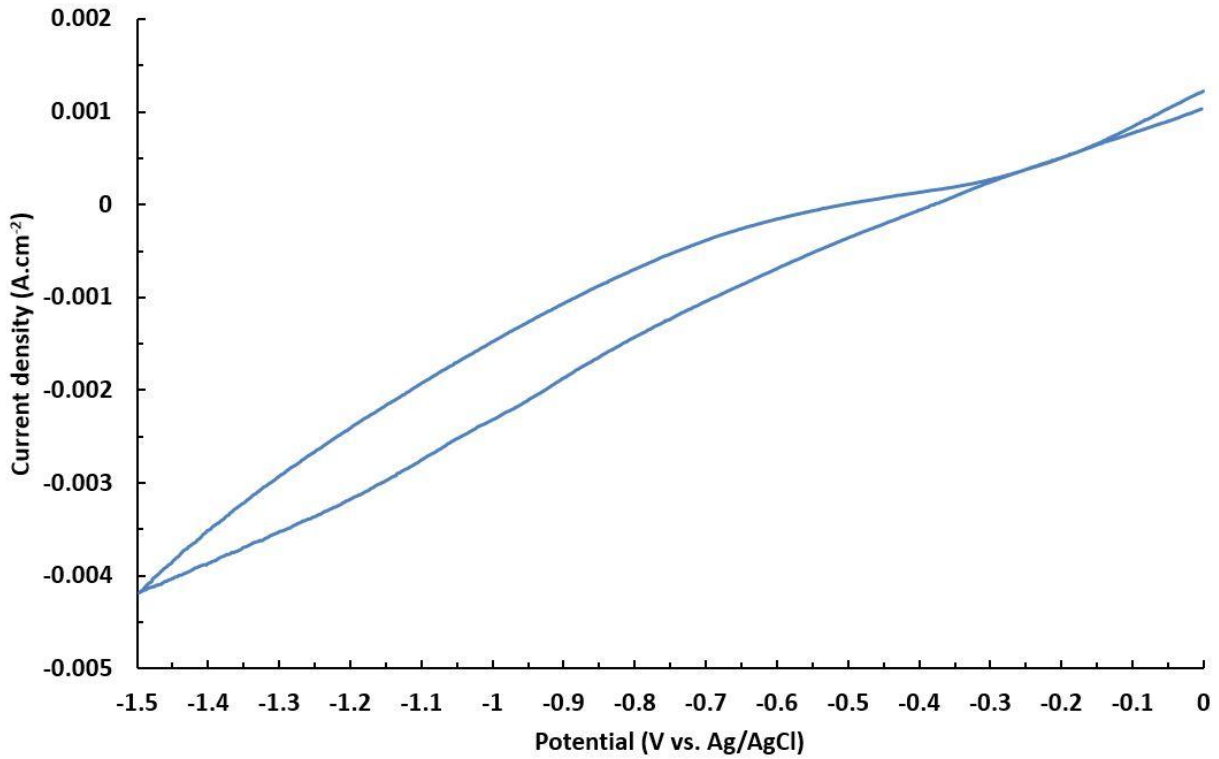
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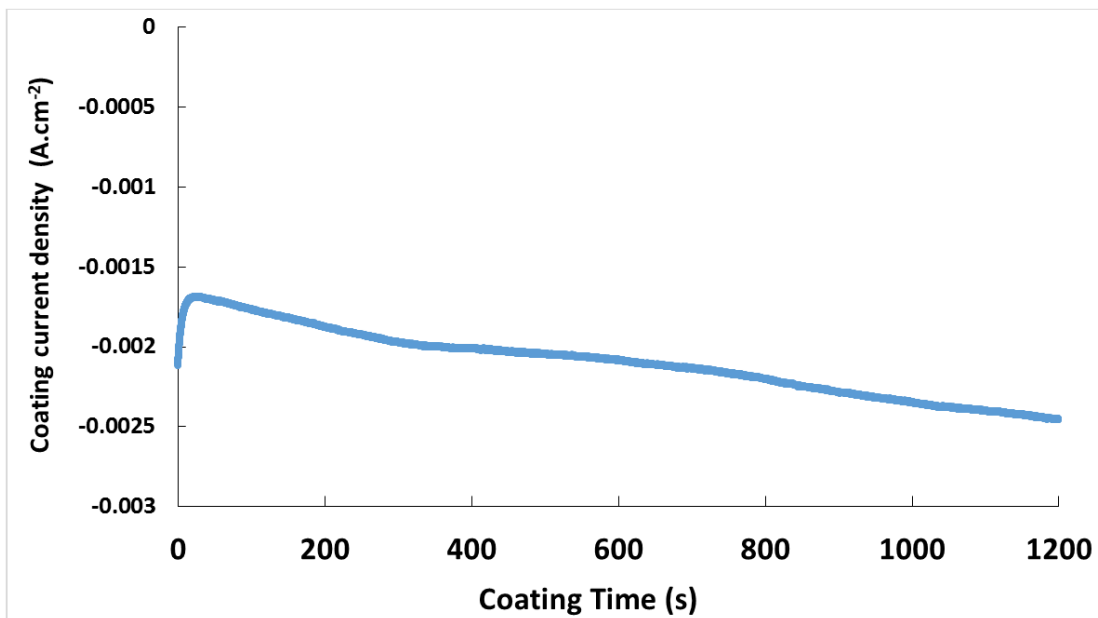
2 Figures

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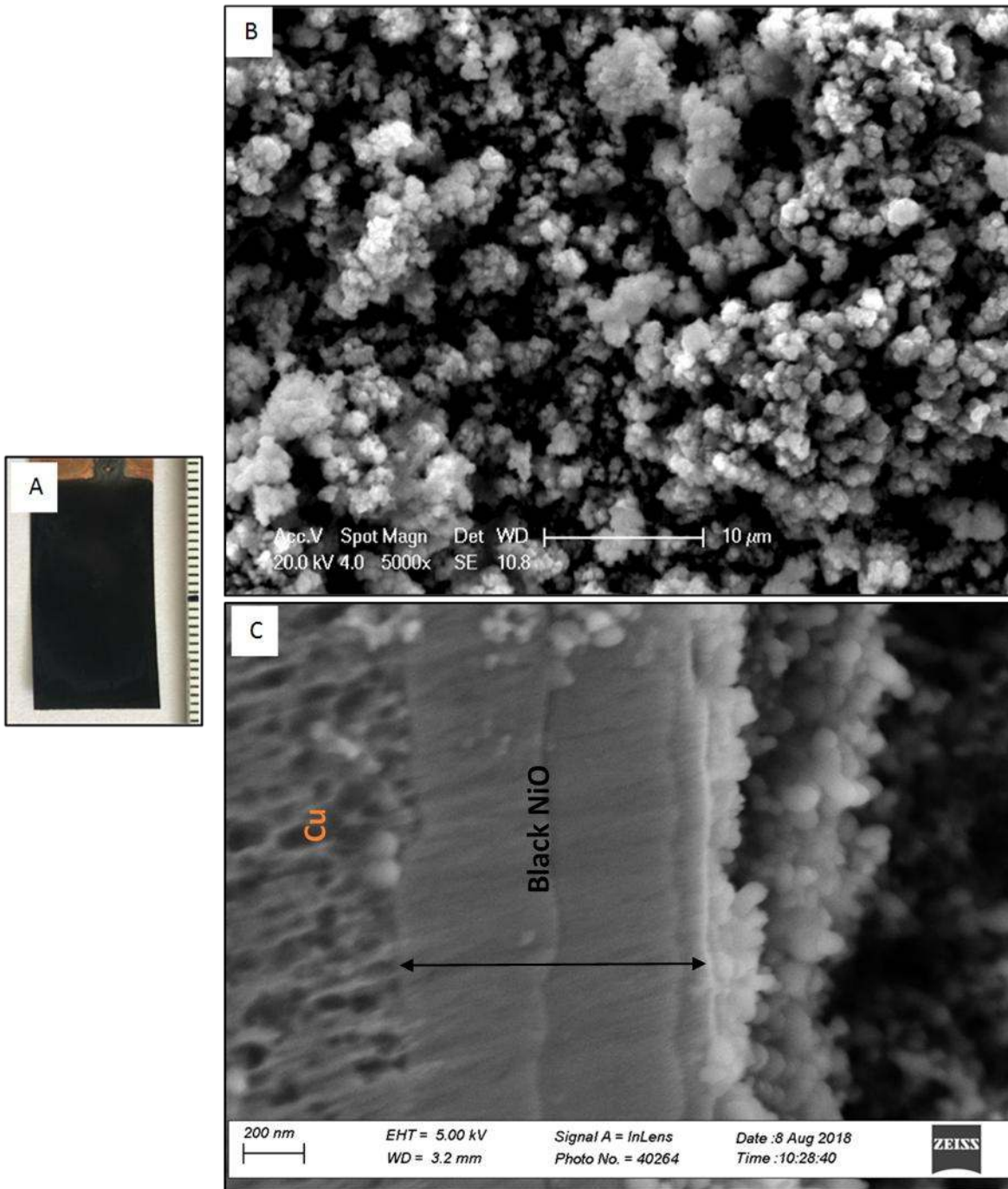
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5 *Figure 1. CV curve for the cathodic electrodeposition of Ni from the DMSO based electrolyte on Cu with a scan rate of 20 mV.s⁻¹.*



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7 *Figure 2. Current density as a function of the coating time for a black NiO film obtained in DMSO.*



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Figure 3. (A) Black NiO film on Cu and its surface (B) and cross-section (C) morphology.

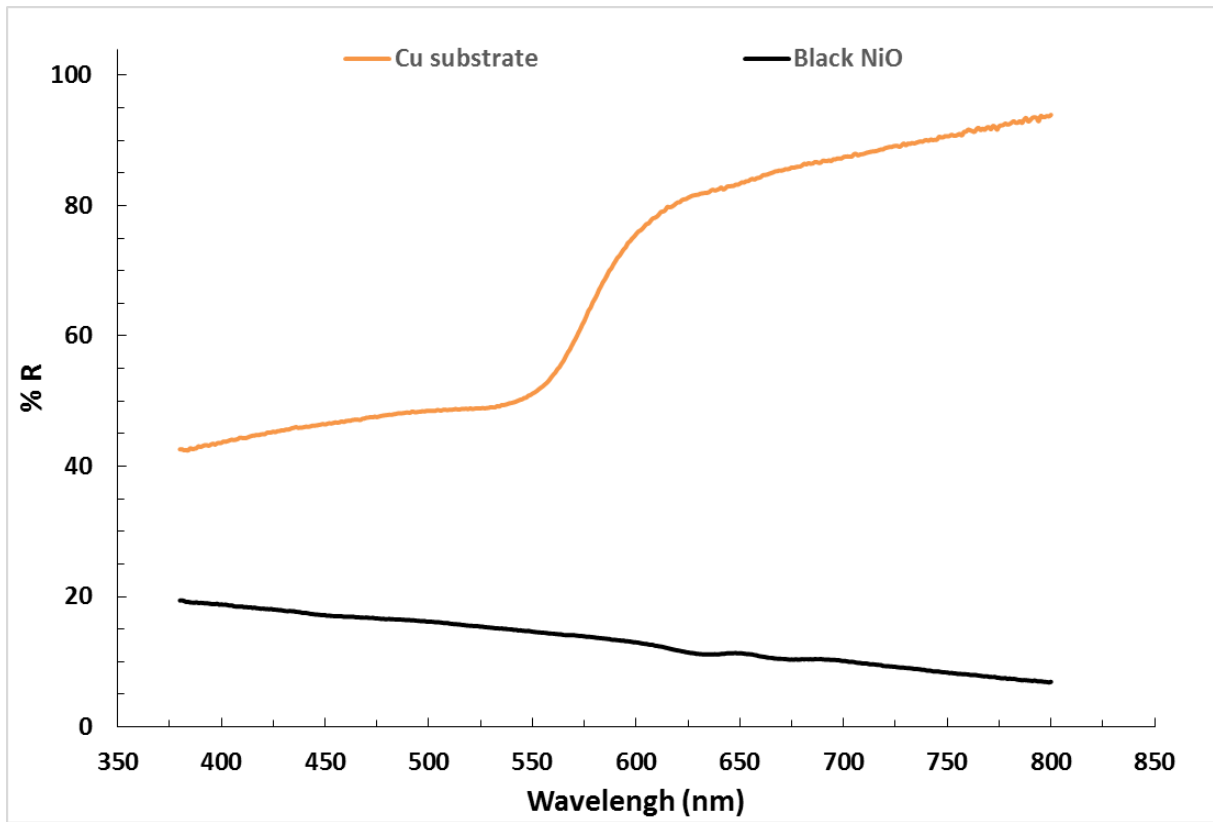
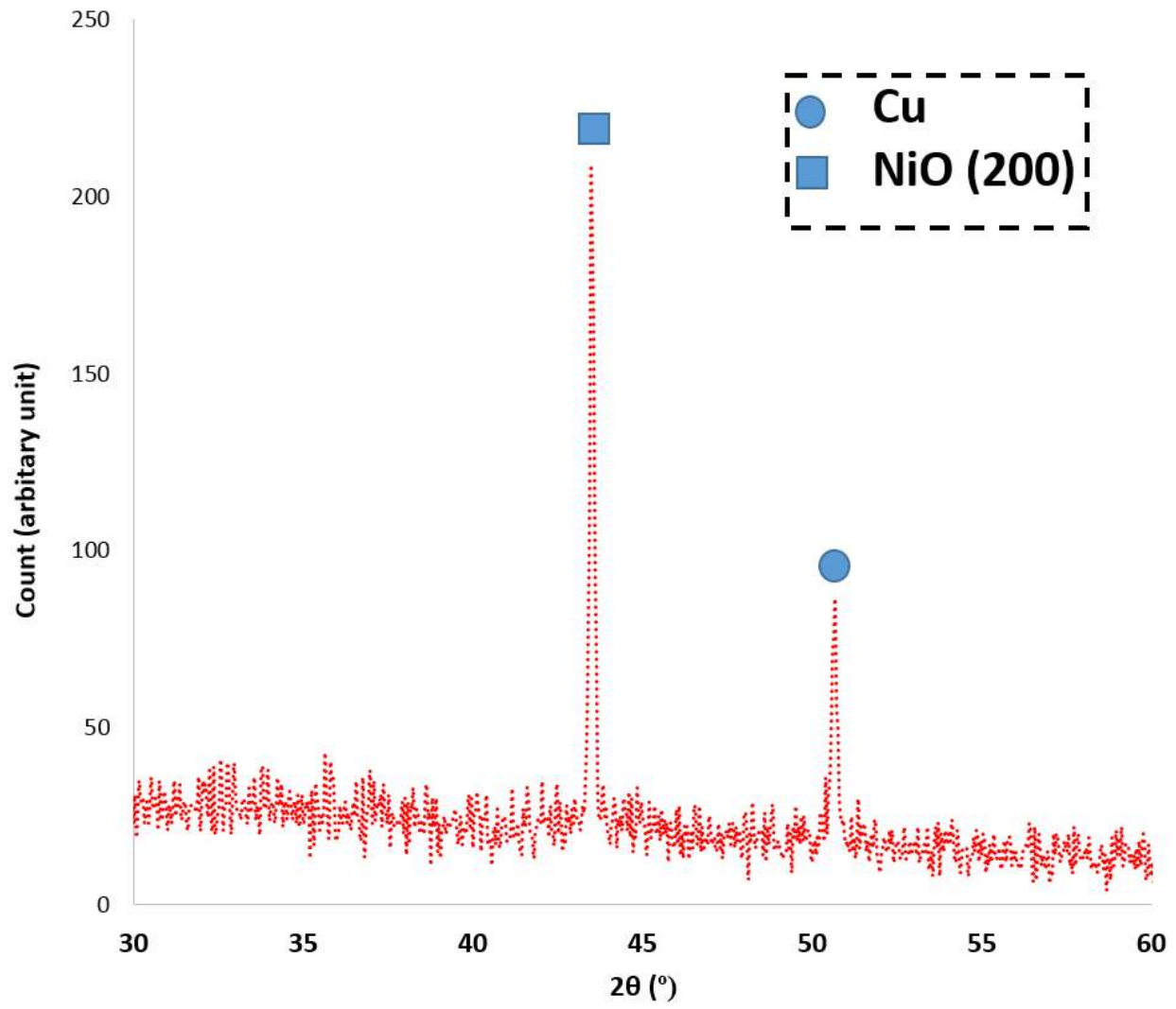


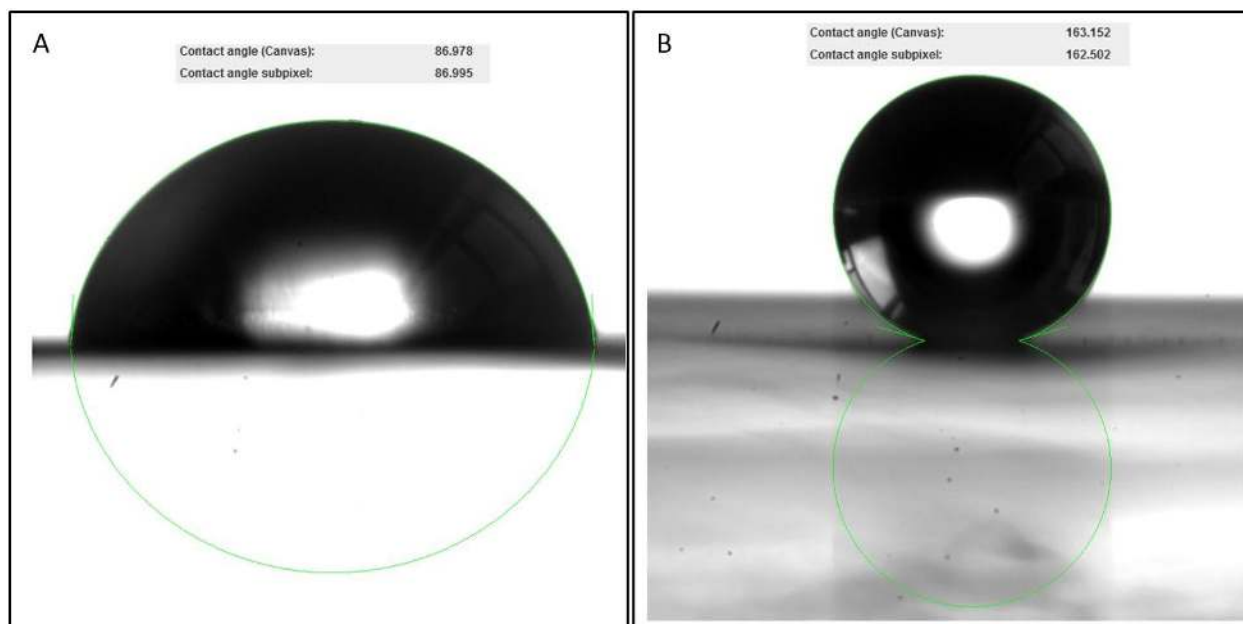
Figure 4. Reflectance of Cu substrate and black NiO film as a function of the wavelength of light.

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2 *Figure 5. Grazing incidence XRD pattern of the black NiO coating.*



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2 *Figure 6. Contact angle between a 2 μ L water droplet and (A) the Cu substrate, (B) the black NiO film.*

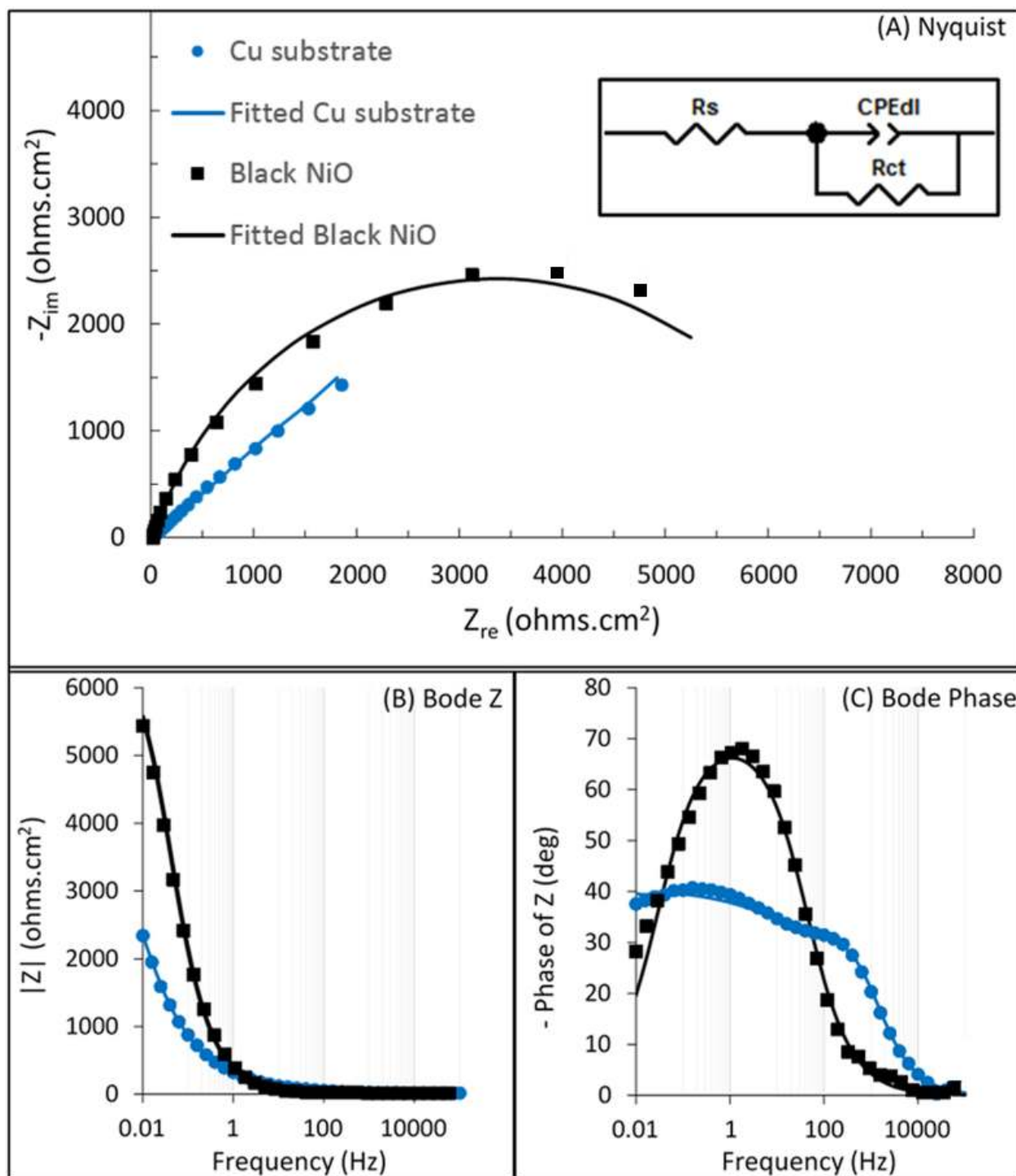
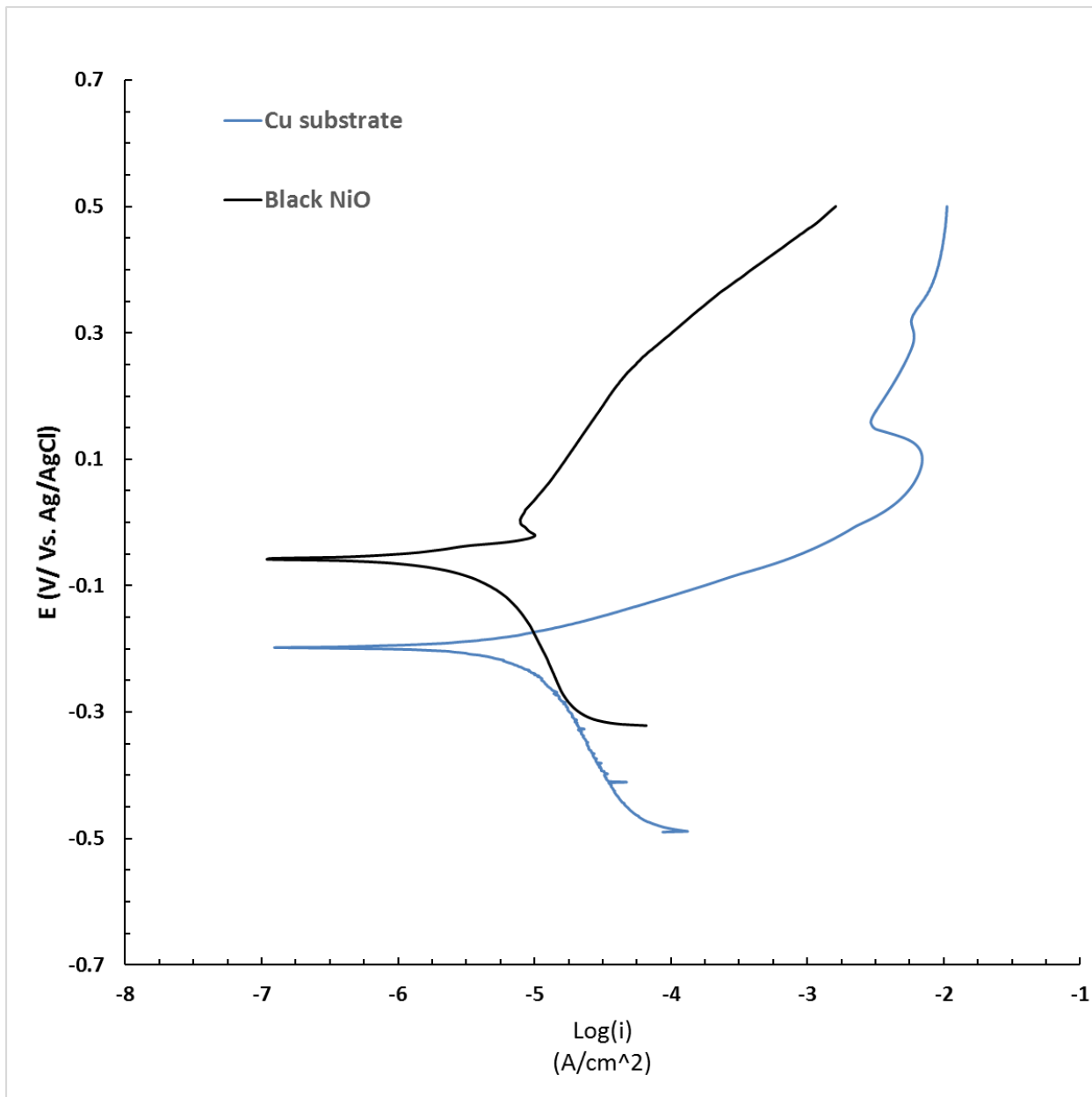


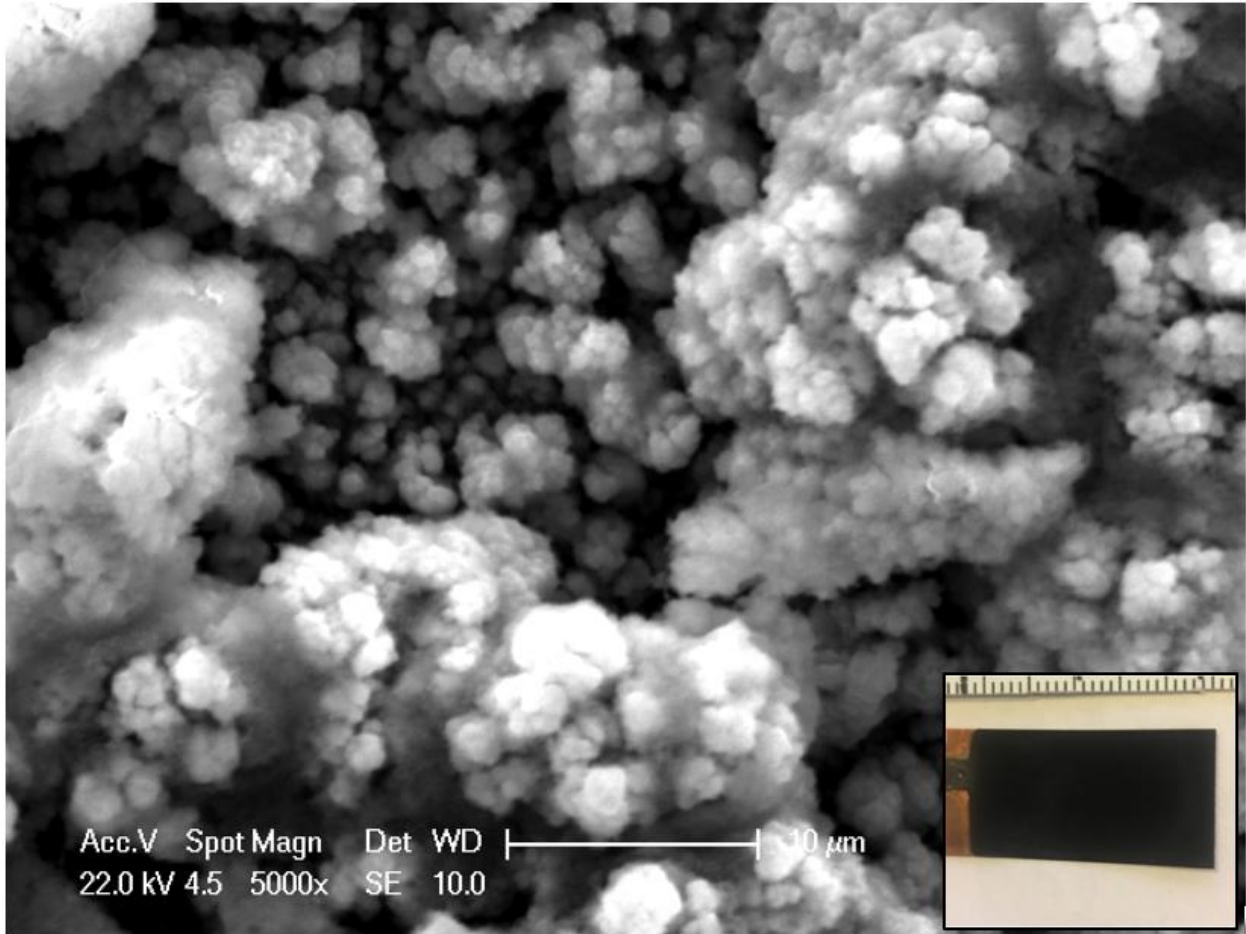
Figure 7. (A) Nyquist, (B) bode Z, and (C) bode phase plots of the black NiO film and the Cu substrate.

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2 *Figure 8. Tafel plots of the black NiO film and the Cu substrate.*



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2 *Figure 9. The general aspect and surface morphology of the black NiO film after the corrosion tests.*