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Solution processed large area field effect transistors from dielectrophoretically aligned arrays of carbon nanotubes

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We demonstrate solution processable large area field effect transistors (FETs) from aligned arrays of carbon nanotubes (CNTs). Commercially available, surfactant free CNTs suspended in aqueous solution were aligned between source and drain electrodes using ac dielectrophoresis technique. After removing the metallic nanotubes using electrical breakdown, the devices displayed *p*-type behavior with on-off ratios up to $\sim 2 \times 10^4$. The measured field effect mobilities are as high as $123 \text{ cm}^2/\text{V s}$, which is three orders of magnitude higher than typical solution processed organic FET devices. © 2009 American Institute of Physics. [DOI: 10.1063/1.3100197]

Solution processed electronic devices have attracted tremendous attention because of their ease of processability, low cost of fabrication, and ability to cover large areas. These devices may be useful for applications such as flexible displays, sensor sheets, radio frequency identification tags, and photovoltaics.^{1,2} A significant amount of effort has been dedicated to improve device performance of solution processed organic field effect transistors (FETs). However, typical field effect mobilities for these devices are usually on the order of $\sim 0.1 \text{ cm}^2/\text{V s}$ and can very rarely reach $\sim 1.0 \text{ cm}^2/\text{V s}$.¹⁻³ In addition, the mobilities are highly sensitive to detailed fabrication parameters. For example, solution processable FETs made from the most commonly used polymer, regio-regular poly(3-hexylthiophene), sensitively depends on the molecular weight, the dielectric-semiconductor interface, the solvent that it is spun from, surface treatments, postfilm formation treatment, and annealing.⁴⁻⁷ Furthermore, FETs made from polymers tend to degrade in air, adding another degree of difficulty to the procedure.⁸

An alternative route to fabricate high quality solution processed FETs that can be superior to polymer based devices may be the use of carbon nanotubes (CNTs) dispersed in solution. FETs from individual CNTs have displayed exceptional electrical properties including subthreshold swings as low as 60 mV/decade and mobilities reaching $79\,000 \text{ cm}^2/\text{V s}$.⁹ However, devices fabricated from arrays of CNTs can be advantageous over individual tube devices in certain cases, as they may provide more homogeneity from device to device and can cover large areas. In addition, devices fabricated with nanotube arrays contain hundreds of CNTs, which can increase current outputs (up to hundreds of microamperes). Large scale assembly of CNTs from solution can be achieved by several different techniques including chemical and biological patterning,^{10,11} Langmuir-Blodgett assembly,¹² bubble blown films,¹³ contact printing,¹⁴ inkjet printing,¹⁵ spin coating assisted alignment,¹⁶ and evaporation driven self-assembly.¹⁷ All of these techniques create CNT networks where charge transport needs to occur through a

large number of overlapping internanotube contacts.

Recently, dielectrophoresis (DEP) has been used for the directed assembly of individual, bundles, or networks of CNTs.¹⁸⁻²¹ However, high quality FETs from large area DEP assembled arrays have not been demonstrated. DEP assembled CNT-FET devices can be advantageous as every CNT connects between source and drain electrodes minimizing charge transport through CNT-CNT interconnects. Here we report on solution processed, large area high quality FETs from dielectrophoretically aligned arrays of CNTs. Commercially available, surfactant free CNT solution [suspended in de-ionized(DI) water]²² were assembled between source (S) and drain (D) electrodes patterned on a Si/SiO₂ substrate by applying an ac electric field. The highly doped Si substrate was used as a global back gate (G). After using an electrical breakdown technique to remove the metallic CNT pathways, the devices showed on-off ratios ($I_{\text{on}}/I_{\text{off}}$) up to $\sim 2 \times 10^4$ with *p*-type FET behavior. The measured mobilities are as high as $123 \text{ cm}^2/\text{V s}$, which is three orders of magnitude higher than typical solution processed organic FETs. Our technique represents a simple and convenient way to fabricate high quality solution processable FET devices.

Highly doped Si wafers with 250 nm capped layer of SiO₂ were used as substrates. Source and drain electrodes with spacing $L=5 \mu\text{m}$ and a width $W=200 \mu\text{m}$ were defined using electron beam lithography (EBL), then electron beam evaporation of Cr (5 nm) and Pd (30 nm), followed by standard lift-off in acetone. The sample was then placed in oxygen plasma cleaner for 10 min to remove the unwanted organic residues on the surface.

Surfactant free, highly purified CNTs suspended in DI water were obtained from Brewer Science Inc.²² The obtained solution has a concentration of $\sim 50 \mu\text{g}/\text{ml}$. The solution is further diluted in DI water to obtain a concentration of $\sim 1.0 \mu\text{g}/\text{ml}$. The assembly of CNTs was carried out in a probe station under ambient conditions. Figure 1(a) shows a schematic of the DEP assembly circuit. First, a 3 μl drop of the NT suspension was cast onto the electrode array. An ac voltage of 300 kHz, 5 V_{*p-p*} is applied between the source and gate electrode for 15 s. For high frequencies (*f*) of the ac voltage applied between source and gate, the impedance ($Z = 1/j\omega C_{\text{plate}}$, $\omega = 2\pi f$) reduces considerably. Therefore the drain becomes capacitively coupled to the gate electrode and

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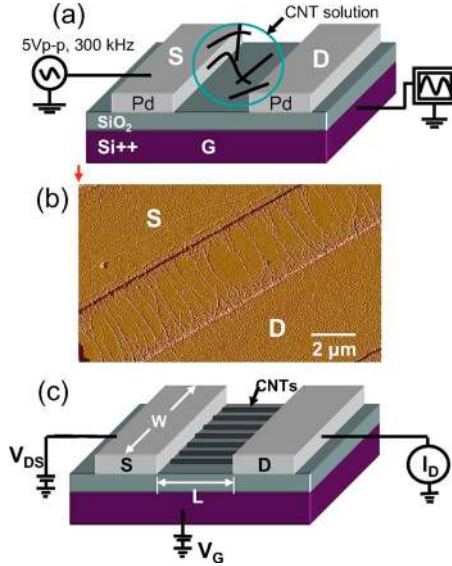


FIG. 1. (Color online) (a) Schematic of dielectrophoretic assembly. An ac voltage of 5 V at 300 kHz is applied to the source (S) electrode while the conducting Si substrate is monitored by an oscilloscope. (b) AFM image of a section of a device after assembly. (c) Cartoon for electronic transport measurements setup.

obtains a similar potential as the gate electrode creating the necessary potential difference between the source and drain electrodes. Hence, the ac voltage creates a time averaged dielectrophoretic force between source and drain to align the CNTs. For an elongated object, it is given by $F_{\text{DEP}} \propto \epsilon_m \text{Re}[K_f] \nabla E_{\text{rms}}^2$, $K_f = (\epsilon_p^* - \epsilon_m^*) / \epsilon_m^*$, $\epsilon_{p,m}^* = \epsilon_{p,m} - i(\sigma_{p,m} / \omega)$, where ϵ_p and ϵ_m are the permittivity of the nanotube and solvent respectively, K_f is the Clausius–Mossotti factor, and σ is the conductivity.¹⁸ The induced dipole moment of the nanotube interacting with the strong electric field causes the nanotubes to move in a translational motion along the electric field gradient and align between the source and drain electrodes. As the CNTs assemble between the source and drain electrodes, the parallel plate capacitance ($C_{\text{plate}} = \epsilon A / t_{\text{ox}}$, t_{ox} is the thickness and $\epsilon = 3.9\epsilon_0$ is the dielectric constant of the SiO₂ layer) of the electrode/SiO₂/Si stack increases due to an increase of the effective area $A = WL$. This causes a decrease of the impedance of the drain/SiO₂/Si stack. This was evident in the observation of the output signal on the oscilloscope as it increased by 30%–40% by the end of the assembly. Figure 1(b) shows an atomic force microscope (AFM) image for a portion of a device after the assembly. The density of the aligned array is ~ 1 CNT/ μm on average giving ~ 200 CNTs total in the channel. By varying the CNT density of the solution and the trapping time, it is possible to tune the number of CNTs per micron in the array. A detailed study of this assembly will be presented elsewhere. The diameter of the CNTs, measured by AFM varies from 1.5 to 6.0 nm. A total of 16 devices were measured, of which half of the devices are measured as-assembled without further processing (bottom contacted device). The other half was measured following an additional EBL step for which 30 nm thick Pd was evaporated to form a top contact (top contacted device).

After the assembly, the room temperature electronic transport measurements were carried out in a probe station using the Si substrate as a global back gate. Figure 1(c) shows a schematic of the electrical transport measure-

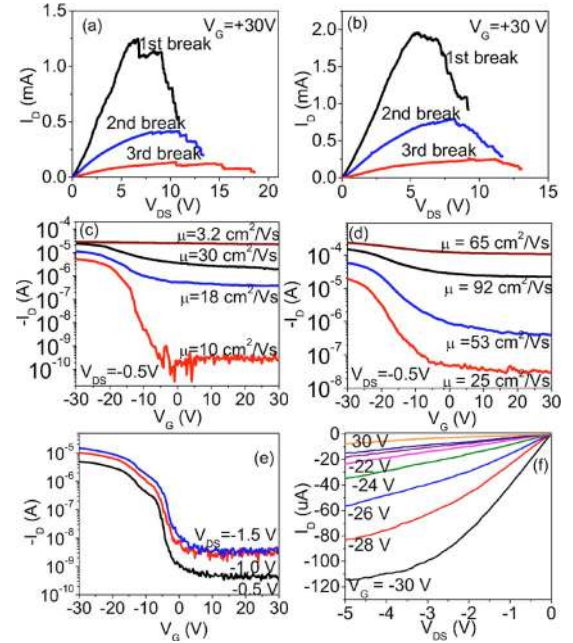


FIG. 2. (Color online) A representative plot of drain current (I_D) vs source-drain voltage (V_{DS}) for three sequential breakdowns (first, second, and third break) for (a) a bottom contacted and (b) a top contacted device (c) I_D vs back gate voltage V_G at constant V_{DS} of -0.5 V after the assembly and each breakdown (top to bottom of figure) for the bottom contacted device. (d) I_D vs V_G after assembly and each breakdown (top to bottom of figure) for top contacted device. (e) I_D vs V_G after third breakdown at different V_{DS} of -0.5 , -1.0 , and -1.5 V for the same device as in (c). The on-off ratio for this device is $\sim 2 \times 10^4$. (f) Output characteristics for the top contact device presented in (d) after third breakdown.

ment that was performed by means of a Keithley 2400 source-meter, 6517A electrometer, and a current preamplifier interfaced with LABVIEW. The initial two terminal resistance is typically in the range of 20–50 k Ω for bottom contacted devices and 2–5 k Ω for the top contacted devices. The mobility is calculated using the formula $\mu = (LA/WV_{\text{DS}}C) \times (dI_D/dV_G)$. The capacitance C of the CNT FET array device was approximated from $C = AD / \{C_Q^{-1} + (1/2\pi\epsilon) \ln[\sinh(2\pi t_{\text{ox}} D) / \pi R D]\}$, where $C_Q = 4 \times 10^{-10}$ F/m is the quantum capacitance, R is the radius of the nanotubes, and D is the linear density in CNTs per micron of the array.²³ According to this equation, the capacitance increases with increasing density of the nanotube and saturates to parallel plate capacitance value at high enough nanotube density. Here, we used $R = 1$ nm and $D = 1$ CNT/ μm .

The as-assembled aligned CNT array devices show semimetallic behavior with on-off ratios, $I_{\text{on}}/I_{\text{off}} \sim 1.3$ –3.0 and average mobilities of $\mu \sim 5.5$ and ~ 54 cm²/V s for the bottom contact and top contact devices, respectively. The low on-off ratio and modest mobility is due to the presence of large amount of metallic pathways in the array. Therefore to increase device performance, we performed an electrical breakdown procedure to controllably reduce the metallic pathways.²⁴

Figures 2(a) and 2(b) show a representative plot of drain current (I_D) versus source-drain voltage (V_{DS}) for three sequential breakdowns (first, second, and third break) for a bottom contacted and a top contacted device, respectively. The back gate was held constant at $V_G = +30$ V to deplete the carriers in the p -type semiconducting CNTs while we

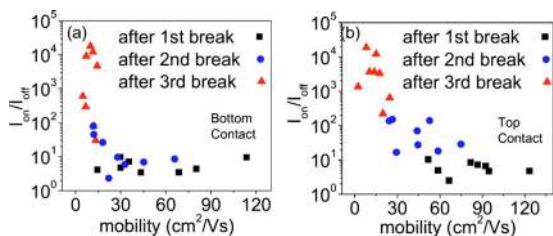


FIG. 3. (Color online) Plot of on-off ratios and corresponding mobility for all measured devices after each breakdown in (a) bottom contact and (b) top contact configuration.

ramped up V_{DS} to eliminate the metallic CNTs. As V_{DS} is ramped up, the CNTs start to breakdown and I_D begins to fall. In order to obtain reproducible results, each breakdown is stopped when I_D is about 50% of its peak value at which point V_{DS} is swept back to zero. When the third breakdown reaches $\sim 50\%$ of its peak value, I_D can range from ~ 0.03 to 0.12 mA.

Figures 2(c) and 2(d) show I_D versus V_G characteristics for a typical bottom and top contact device, respectively after each breakdown. The source-drain voltage is held constant at $V_{DS} = -0.5$ V. In Fig. 2(c), the uppermost curve is the initial sweep showing a mobility of 3.2 $\text{cm}^2/\text{V s}$ with very little on-off ratio (~ 1.1). After the first breakdown, the field effect behavior of the device is enhanced; both the mobility and on-off ratios are increased to 30 $\text{cm}^2/\text{V s}$ and ~ 10 , respectively, due to a reduction of metallic pathways. After the second breakdown, the mobility reduces a small amount to 18 $\text{cm}^2/\text{V s}$ and the on-off ratio increases to ~ 26 . Finally, after the third breakdown the mobility is reduced to 10 $\text{cm}^2/\text{V s}$; however, the on-off ratio increases a few orders of magnitude to $\sim 2 \times 10^4$. Figure 2(d) shows similar behavior for the top contacted device with $\mu = 65, 92, 53,$ and 25 $\text{cm}^2/\text{V s}$ and on-off ratios of $2.1, 6.6, 14,$ and 650 for the initial sample and then after the first, second, and third breakdown, respectively. We find that the top contacted devices show higher mobilities, which most likely is due to the reduced contact resistance. Figure 2(e) is a plot of I_D versus V_G at different V_{DS} after the third breakdown for the same device shown in Fig. 2(c). Figure 2(f) shows the detailed output characteristics, I_D versus V_{DS} at different V_G recorded for the sample presented in Fig. 2(d) after the third breakdown.

Figures 3(a) and 3(b) show the on-off ratio and corresponding mobility value for all of the bottom [Fig. 3(a)] and top contact [Fig. 3(b)] devices after each breakdown. In Fig. 3(a) the on-off ratio remains fairly constant and then increases more rapidly after the third breakdown with median on-off ratios after each breakdown of $4.6, 9.2,$ and 2.6×10^3 . Figure 3(b) shows a more steadily increase in on-off ratio after each breakdown with median values of $5.8, 50,$ and 3.5×10^3 . For the bottom contacted devices, the median mobilities are $50, 27,$ and 9.1 $\text{cm}^2/\text{V s}$ after first, second, and third breakdown, respectively. Top contacted devices yield median mobility values of $77, 41,$ and 15 $\text{cm}^2/\text{V s}$ after the three breakdowns, respectively. We found that the top contacted devices are more controllable and show better device

to device reproducibly after each breakdown. This is most likely due to the better contact resistance from the top contact. The highest mobility obtained from all the devices is 123 $\text{cm}^2/\text{V s}$. The mobility values reported here are up to three orders of magnitude higher than typical FET devices made from solution processed polymers.²

In conclusion, we have demonstrated solution processable large area FETs from aligned arrays of CNTs. The CNTs were aligned from a commercially available, surfactant free CNTs suspended in aqueous solution using ac DEP. After reducing the metallic pathways using electrical breakdown, the devices displayed on-off ratios up to $\sim 2 \times 10^4$. The devices showed p -type FET behavior with mobilities up to three orders of magnitude higher than typical solution processed organic FET devices. The ease of processing for the dielectrophoretically assembled devices presented here offers an alternative to solution processed polymer FET devices.

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- ¹A. Dodabalapur, *Mater. Today* **9**, 24 (2006).
- ²T. B. Singh and N. S. Sariciftci, *Annu. Rev. Mater. Res.* **36**, 199 (2006).
- ³S. K. Park, D. A. Mourey, S. Subramanian, J. E. Anthony, and T. N. Jackson, *Appl. Phys. Lett.* **93**, 043301 (2008).
- ⁴R. J. Kline, M. D. McGehee, E. N. Kadnikova, J. Liu, and J. M. J. Frechet, *Adv. Mater. (Weinheim, Ger.)* **15**, 1519 (2003).
- ⁵A. Facchetti, M.-H. Yoon, and T. J. Marks, *Adv. Mater. (Weinheim, Ger.)* **17**, 1705 (2005).
- ⁶J. F. Chang, B. Q. Sun, D. W. Breiby, M. M. Nielsen, T. I. Solling, M. Giles, I. McCulloch, and H. Sirringhaus, *Chem. Mater.* **16**, 4772 (2004).
- ⁷L. Ke, S. B. Dolmanan, L. Shen, C. Vijila, S. J. Chua, R.-Q. Png, P.-J. Chia, L.-L. Chua, and P. K.-H. Ho, *Appl. Phys. Lett.* **93**, 153507 (2008).
- ⁸B. Ong, Y. Wu, L. Jiang, P. Liu, and K. Murti, *Synth. Met.* **142**, 49 (2004).
- ⁹P. Avouris, Z. H. Chen, and V. Perebeinos, *Nat. Nanotechnol.* **2**, 605 (2007).
- ¹⁰S. Auvray, V. Derycke, M. Goffman, A. Filoramo, O. Jost, and J.-P. Bourgoin, *Nano Lett.* **5**, 451 (2005).
- ¹¹K. Keren, R. S. Berman, E. Buchstab, U. Sivan, and E. Braun, *Science* **302**, 1380 (2003).
- ¹²S. Jin, D. Whang, M. C. McAlpine, R. S. Friedman, Y. Wu, and C. M. Lieber, *Nano Lett.* **4**, 915 (2004).
- ¹³Y. Guihua, C. Anyuan, and C. M. Lieber, *Nat. Nanotechnol.* **2**, 372 (2007).
- ¹⁴A. Javey, S. W. Nam, E. S. Friedman, H. Yan, and C. M. Lieber, *Nano Lett.* **7**, 773 (2007).
- ¹⁵K. Kordás, T. Mustonen, G. Tóth, H. Jantunen, M. Lajunen, C. Soldano, S. Talapatra, S. Kar, R. Vajtai, and P. M. Ajayan, *Small* **2**, 1021 (2006).
- ¹⁶M. C. LeMieux, M. Roberts, S. Barman, Y. W. Jin, J. M. Kim, and Z. Bao, *Science* **321**, 101 (2008).
- ¹⁷M. Engel, J. P. Small, M. Steiner, M. Freitag, A. A. Green, M. C. Hersam, and P. Avouris, *ACS Nano* **2**, 2445 (2008).
- ¹⁸P. Stokes and S. I. Khondaker, *Nanotechnology* **19**, 175202 (2008).
- ¹⁹A. Vijayaraghavan, S. Blatt, D. Weissenberger, M. Oron-Carl, F. Henrich, D. Gerthsen, H. Hahn, and R. Krupke, *Nano Lett.* **7**, 1556 (2007).
- ²⁰S. Banerjee, B. White, L. Huang, B. J. Rego, S. O'Brien, and I. P. Herman, *Appl. Phys. A: Mater. Sci. Process.* **86**, 415 (2007).
- ²¹A. H. Monica, S. J. Papadakis, R. Osiander, and M. Paranjape, *Nanotechnology* **19**, 085303 (2008).
- ²²See <http://www.brewerscience.com> for more information about the CNT solution.
- ²³S. J. Kang, C. Kocabas, T. Ozel, M. Shim, N. Pimparkar, M. A. Alam, S. V. Rotkin, and J. A. Rogers, *Nat. Nanotechnol.* **2**, 230 (2007).
- ²⁴P. G. Collins, M. S. Arnold, and Ph. Avouris, *Science* **292**, 706 (2001).