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# Solution-Processed Low Threshold Vertical Cavity Surface Emitting Lasers from All-Inorganic Perovskite Nanocrystals

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1	Solution-processed low threshold vertical cavity surface emitting lasers
2	from all-inorganic perovskite nanocrystals
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15	Abstract: Recently, all-inorganic cesium lead halide perovskite nanocrystals (IPNCs) (CsPbX <sub>3</sub> ,
16	X = Cl, Br, I) were discovered to possess superior optical gain properties appealing for solution-
17	processed cost-effective lasers. Yet, the potential of such materials has not been exploited for
18	practical laser devices, rendering the prospect as laser media elusive. Herein, we realized for the
19	first time the challenging but practically desirable vertical cavity surface emitting lasers (VCSELs)
20	based on the CsPbX <sub>3</sub> IPNCs, featuring low threshold (9 $\mu$ J/cm <sup>2</sup> ), unidirectional output (beam
21	divergence of $\sim 3.6^{\circ}$ ) and favorable stability. The lasing wavelength can be tuned across red, green
22	and blue region maintaining comparable thresholds, which is promising in developing single
23	source-pumped full-color visible lasers. It is fully demonstrated that the characteristics of the

24	VCSELs can be versatilely engineered by independent adjustment of the cavity and solution
25	processable nanocrystals. Our results represent a significant leap towards practical laser sources
26	leveraging on the advantageous CsPbX <sub>3</sub> IPNCs.
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43 Solution-processed gain materials are being pursued with the promise to revolutionize the 44 vacuum-based epitaxial semiconductor lasers aiming at developing low-cost yet highperformance laser source<sup>1</sup>. Colloidal quantum dot (CQD) has been recognized as the ideal 45 candidate due to the tunable emission color, enhanced optical properties and facile solution 46 processibility<sup>2-4</sup>. Ever since the first demonstration of stimulated emission from metal-47 chalcogenide CQDs in  $2000^2$ , significant progress has been made in improving the lasing 48 performance<sup>5, 6</sup>. However, the relatively low absorption cross-section and the high optical loss 49 due to carrier trapping and Auger recombination (AR) still hinder the advance of CQD-lasers, 50 51 especially in the short visible spectral range of blue and green<sup>7,8</sup>.

52 In the past years, the organic-inorganic halide perovskites, which have demonstrated impressive photovoltaic performance<sup>9, 10</sup>, were also discovered to be potential optical gain 53 media<sup>11, 12</sup>. However, the commercial aspirations may be stifled by the inherent instability of 54 these hybrid perovskites<sup>11, 12</sup>. More recently, the newly engineered all-inorganic cesium lead 55 56 halide (CsPbX<sub>3</sub> (X=Cl, Br and I)) perovskite nanocrystals are resurging as a superior optical gain material because of the large optical absorption cross-section, large exciton binding energy, high 57 photoluminescence (PL) quantum yield and relatively low AR loss<sup>13, 14</sup>. Importantly, these all-58 inorganic perovskite nanocrystals (IPNCs) show much enhanced endurance to ambient 59 environment than the organic-inorganic analogues<sup>13, 15, 16</sup>. Compared to traditional metal-60 chalcogenide CODs, these CsPbX<sub>3</sub> IPNCs feature in narrow emission spectra, simple fabrication 61 process, ease of color tunability by anion exchange, superior optical gain properties and so on<sup>17-20</sup>. 62 As a result, these new emerging CsPbX<sub>3</sub> IPNCs can be envisioned as the promising solution 63 64 towards inexpensive laser sources in the future.

So far, lasers based on single CsPbX<sub>3</sub> nanowire have been reported<sup>21, 22</sup>. Regarding lasers 65 made from the ensemble of inorganic perovskite nanocrystals, only whispering gallery mode and 66 random lasers which are among the easiest configurations for lasers have been demonstrated as a 67 proof of concept<sup>13, 14</sup>. However, both of these laser types lack directionality, one of the most 68 important advantages of a laser. Moreover, all of the above lasing demonstrations fall in multi-69 mode operation<sup>13, 14</sup>. In laser physics and applications, obtaining single-mode lasing is crucial 70 since multi-mode operation will deteriorate the color purity and temporal stability by mode 71 oscillation<sup>23, 24</sup>. Due to the typical large cavity length to provide sufficient gain for lasing action 72 and the relatively broad gain bandwidth of traditional lasing materials, achieving single-mode 73 lasing is still challenging<sup>23, 24</sup> and remains unaccomplished for CsPbX<sub>3</sub>IPNCs. 74

75 Vertical cavity surface emitting laser (VCSEL) is an important and much desired laser type 76 which finds a broad range of applications like optical communication, high density optical storage, 77 laser display, parallel optical computing and signal processing because of its ability to form 2dimensional arrays, surface normal emission characteristics and high beam quality output 78 ensuring easy coupling into an optical fibre<sup>25, 26</sup>. For any given gain materials, VCSELs represent 79 the most tough laser configuration determined by the stringent criteria between gain and  $loss^{6, 25}$ . 80 81 As such, most surface normal devices were previously developed from complicated epitaxial 82 growth of semiconductor heterostructures of many layers in which both bandgap alignment and lattice mismatch are big concerns<sup>27-30</sup>. The unavailability of independently adjusting the cavity 83 and the active gain materials in an epitaxy based VCSEL makes material selection a constraint 84 and the device cost ineffective. Thus, alternative fabrication of VCSELs with simplified 85 86 processing is very appealing. Herein, we realized for the first time the high-performance VCSELs 87 from solution-processed CsPbX<sub>3</sub> IPNCs. A clear evolution from spontaneous emission to lasing 88 in the device upon optical pumping was manifested by the spectral narrowing, nonlinear increase of the PL intensity, drastic reduction of PL lifetime and remarkable decrease of output beam 89 90 divergence. Both multi-mode and single-mode lasing operation have been achieved by tuning the 91 cavity length. Thanks to the superior optical gain properties of the CsPbBr<sub>3</sub> IPNCs and the good 92 match between the gain profile and the high reflectivity band of the distributed Bragg reflectors 93 (DBRs), the lasing threshold of the IPNC-VCSELs is so low that quasi-continuous wave (q-CW) 94 pumping is feasible. In contrast to traditional metal-chalcogenide CQD based lasers, where the 95 thresholds for the green and blue are typically much higher than that of the red<sup>6</sup>, these CsPbX<sub>3</sub> IPNC-VCSELs lase with comparable thresholds across the whole visible spectral range, which is 96 97 promising in achieving single source-pumped full-color lasers. Our results highlight the resurging CsPbX<sub>3</sub> IPNCs in developing the challenging but practical lasers and shed light on the feasibility 98 99 of independent adjustment of cavity and gain materials for the IPNC-VCSELs, which represent a great progress towards advanced laser sources based on the advantageous CsPbX<sub>3</sub> IPNCs. 100

101 Results

102 Modification of spontaneous emission from CsPbBr<sub>3</sub> IPNCs by microcavity effect. The 103 CsPbX<sub>3</sub> IPNCs adopted here were synthesized following the method reported by Protesescu et 104 al.<sup>17</sup> with slight adjustment<sup>13</sup>. The absorption and emission spectra of the CsPbBr<sub>3</sub> IPNCs are 105 shown in Supplementary Fig. 1b, revealing the emission peak of ~504 nm with a narrow full-106 width at half maximum (FWHM) of ~21 nm. The corresponding transmission electron 107 microscope (TEM) image (Supplementary Fig. 1a) displays the cubic shape with an edge length

of ~9 nm. To build the vertical microcavity, the commercially available dielectric DBRs 108 consisted of 25 pairs of SiO<sub>2</sub>/TiO<sub>2</sub> quarter-wave layers were employed as the high-reflective 109 110 mirrors. Figure 1a displays the reflection spectrum of a DBR used here, which exhibits a stop-111 band from 430 nm to 570 nm, matching the PL spectrum of CsPbBr<sub>3</sub> IPNCs (see broadband reflection spectrum in Supplementary Fig. 2). The reflectivity was determined to be as high as 112 99.6% at ~500 nm. To sandwich the CsPbBr<sub>3</sub> IPNCs between two DBRs so as to form the 113 prototypical VCSEL, the highly concentrated CsPbBr<sub>3</sub> IPNCs solution was spin-coated onto the 114 115 top surface of a DBR, then, another DBR was brought upside-down in contact to the CsPbBr<sub>3</sub> IPNC film and finally fixed with glue. The schematic diagram (Fig. 1b), photograph 116 (Supplementary Fig. 3) and the corresponding reflection spectrum ((Supplementary Fig. 2) of the 117 118 final device are presented. Prior to the lasing investigation, the modification of spontaneous emission from CsPbX<sub>3</sub> IPNCs by the optical microcavity was interrogated, which could reflect 119 the quality of the complete DBR resonator. Figure 1a shows the typical PL spectrum from the 120 CsPbBr<sub>3</sub> IPNCs inserted within the cavity collected at 0° with respect to the surface normal (see 121 122 Methods for details). In contrast to the PL spectrum from CsPbX<sub>3</sub> IPNCs spin-coated on a single 123 DBR substrate, the PL within the cavity manifests multiple interference peaks, visualizing the 124 presence of Fabry-Perot cavity effect. It is worth to note that the spiked emission is not laser but 125 spontaneous emission modified by interference effect, which is further demonstrated by the following lifetime measurements (Fig. 2c). The resonant peaks can be assigned to mode numbers 126 127 indexed as 29-33 based on the equation<sup>31</sup>:  $m\lambda = 2nL$ , where *m* is the mode number,  $\lambda$  is the emission wavelength, n is the refractive index of the CsPbX<sub>3</sub>IPNC film, L is the effective cavity 128 length. L is estimated to be 4  $\mu$ m assuming  $n=2^{13}$ . By changing the thickness of the CsPbBr<sub>3</sub> 129 IPNC film, the optical modes of the cavity and thus the PL spectrum can be tuned as exemplified 130 131 in Supplementary Fig. 4. The slight red-shift of the envelop of the emission comb relative to the 132 free-space emission spectrum can be attributed to the reabsorption effect due to the multipass of the radiation inside the cavity<sup>24</sup>. Furthermore, the spatial distribution of the radiation and the 133 134 angle ( $\theta$ ) dependent PL spectrum from the device were examined by angle-resolved PL 135 measurement (see Methods for details). It is found that the optical mode wavelengths gradually shift to blue as the detection angle increases (Fig. 2a), which quantitatively follows the equation: 136  $m\lambda = 2nL\sqrt{n^2 - sin^2\theta}$  (Supplementary Fig. 5). Such behavior states nothing but that the 137 radiation propagation angle inside the cavity and the detection angle are correlated by Snell's law 138 (see detailed deviations in Supporting Information)<sup>1, 31</sup>, further confirming that the discrete peaks 139 140 originate from the Fabry-Perot interference. Notably, the radiation from the microcavity is much 141 narrowed in space with respect to that from free-space CsPbBr<sub>3</sub> IPNCs (Fig. 2b), indicating that spatially confined emission is obtained by the microcavity effect, which may shed light on developing CsPbX<sub>3</sub> IPNC-based light emitting diodes with directional radiation. Finally, the carrier dynamics of CsPbBr<sub>3</sub> IPNCs within/without microcavity was inspected by a streak camera system (Fig. 2c) (see the corresponding spectrograms in Supplementary Fig. 6). It is found that the PL decay curve within microcavity is almost identical to that in free space. A reduction of PL lifetime due to Purcell effect is not observed which can be attributed to the long cavity length and that the cavity mode only confined in one dimension<sup>32</sup>.

149 Vertical cavity surface emitting lasers from CsPbBr<sub>3</sub> IPNCs. The stimulated emission from the close-packed thin film of the CsPbBr<sub>3</sub> IPNCs was first studied by stripe pumping 150 configuration<sup>13, 33</sup> to exam the optical gain. As increasing the pumping density, the development 151 of stimulated emission revealed by a much narrowed emission peak (FWHM: ~6.5 nm) can 152 readily occur from our sample with a threshold of  $\sim 18 \mu J/cm^2$  (Supplementary Fig. 7). Such a 153 low-threshold optical gain from CsPbBr<sub>3</sub> IPNCs in combination with the favorable coupling 154 155 between the emission and the cavity modes motivates us to pursue CsPbBr<sub>3</sub> IPNC-based VCESLs. In doing so, the pump laser beam (400 nm, 100 fs) was focused by a circular lens (focus length: 5 156 157 cm) onto the device vertically, and the PL signal was collected from the surface normal of the cavity. A 30% reflection by the DBR for pump wavelength was extracted and counted in 158 159 determining the final pump intensity. Figure 3a displays the evolution of PL spectra from the device as a function of pump intensity. Under relatively low pump intensities ( $< 11 \mu J/cm^2$ ), the 160 PL spectra are dictated by spontaneous emission shaped by the cavity mode. With the increase of 161 pump intensity, one of the discrete peaks with wavelength close to the stimulated emission 162 163 spectra grows much faster than the others, which is accompanied by dramatic narrowing of the line-width down to ~0.6 nm. Figures 3b and c depict the change of PL intensity and the FWHM 164 165 of the peaks corresponding to mode numbers from 29 to 33, respectively. The abrupt increase of 166 the PL intensity and sudden decrease of the line-width of a particular peak indicate the achievement of lasing action<sup>1, 34, 35</sup>. The lasing threshold ( $P_{th}$ ) is derived to be as low as 11  $\mu$ J/cm<sup>2</sup>, 167 which is much lower than that of the CdSe/CdZnS CQDs (65  $\mu$ J/cm<sup>2</sup>), under similar conditions<sup>6</sup>. 168 Importantly, due to the much larger free spectral range (FSR) of  $\sim 16$  nm determined by the short 169 cavity length than the stimulated emission line-width (6.5 nm), single-mode lasing operation was 170 171 achieved. It should be noted that although the lasing peak locates at the blue side of the emission comb, it is actually on the red side of the spontaneous emission maximum of CsPbBr<sub>3</sub> IPNCs 172 (Supplementary Fig. 8), in agreement with that the optical transition of the laser originates from 173 the biexciton recombination<sup>13, 36, 37</sup>. 174

To further verify the development of lasing, the PL dynamics as a function of pump intensity was investigated. Under pump intensity (0.1  $P_{th}$ ) of much lower than the threshold, the PL decay resembles the spontaneous emission trace. With the increase of pump intensity (0.8  $P_{th}$ ) approaching the threshold, a much faster decay channel, corresponding to the Auger recombination, appears. As the pump intensity (1.3  $P_{th}$ ) surpasses the threshold, the PL decay suddenly collapses to <50 ps, limited by the temporal resolution of the streak camera system, indicating the onset of lasing action<sup>38</sup>.

182 Coherence and directionality are two basic features of a laser but are rarely examined in 183 CQDs based lasers. Herein the directionality of the output radiation above and below the 184 threshold is assessed by a CCD camera located 15 mm away from the laser device (inset in Fig. 185 3a) as well as angle-resolved PL measurement (Supplementary Fig. 9). We can see that when the pump intensity (1.3  $P_{th}$ ) exceeds the threshold, the divergence of the output signal remarkably 186 decreases from 15° for spontaneous emission below  $P_{th}$  to 3.6°, thus a directional emission was 187 obtained in our IPNC-VCSEL. Moreover, a clear interference pattern (Supplementary Fig. 10a) 188 can be observed by using the conventional Michelson interference experiment<sup>39, 40</sup>. From the plot 189 of the visibility (V =  $\frac{I_{max} - I_{min}}{I_{max} + I_{min}}$ , where  $I_{max}$  and  $I_{min}$  represent the intensities at the fringe maxima 190 and minimum) as a function of detuning time or optical path difference<sup>39, 40</sup>, the coherence time  $(\tau)$ 191 of  $\sim$ 1 ps can be derived (Supplementary Fig. 10c). 192

Finally, we test the stability of the device by monitoring the lasing peak intensity versus time at pump intensity of 1.5  $P_{th}$  (pulse-width: 100 fs; repetition rate: 1KHz) (Supplementary Fig. 11). Thanks to the robustness of the CsPbBr<sub>3</sub> IPNCs, the output signal can maintain 80% of its initial value for more than one hour, which far excels the CdSe/CdZnS CQD-lasers<sup>6</sup>.

Inspired by the excellent laser performance under femtosecond laser pump, it is highly 197 promising to explore our IPNC-VCSEL pumped in the q-CW regime by a compact nanosecond 198 laser<sup>7, 41</sup>, which would be more practical and cost-effective. A ns laser pump is called the q-CW 199 just because the pump duration is much longer than that of the effective gain  $(\sim 35 \text{ ps})^{42}$ . Figure 4a 200 shows the pump intensity dependent PL spectra pumped by a Q-switched nanosecond laser 201 202 (pulse-width: 5 ns; repetition rate: 20 Hz; wavelength: 400 nm). The onset of lasing action is 203 unambiguously evidenced by the spectral narrowing and nonlinear increase of the PL intensity with respect to the pump intensity (inset in Fig. 4a) with a low threshold of 900  $\mu$ J/cm<sup>2</sup>. To the 204 205 best of our knowledge, this is the first demonstration of CQD-VCSEL operating in q-CW regime, 206 which represents a significant step towards continuous wave and electrical pumping. The lasing stability under nanosecond pumping is also examined (Supplementary Fig. 11). It is found that the lasing can last over one hour with losing only 50% of initial peak intensity. The faster decrease of the lasing intensity compared to that under femtosecond pump suggests the more serious thermal issue for long pump pulse duration<sup>36</sup>. So the effective thermal management (heat roll-off) shall be one of the major issues to address for future work toward stable CW operation<sup>36,</sup>  $^{43}$ .

Furthermore, we demonstrate that multi-mode lasing operation can be produced in our IPNC-VCSELs via tuning the *FSR* to be smaller than the stimulated emission bandwidth which is enabled by increasing the effective cavity length (*L*) or film thickness of CsPbBr<sub>3</sub> IPNCs. Figure 4b illustrates the typical double mode lasing from the IPNC-VCSEL with *FSR* of ~6.2 nm and *L* of ~10.3  $\mu$ m. Due to the longer cavity path and thus the lager round-trip gain, the lasing threshold further reduces to ~9.0  $\mu$ J/cm<sup>2</sup> (inset in Fig. 4b).

219 Full-color VCSELs from CsPbX<sub>3</sub> IPNCs with comparable thresholds. For traditional metal-220 chalcogenide CQDs, the lasing wavelength is generally tuned by changing the dot size because of the band gap limit<sup>2, 6</sup>. For example, smaller dot size is necessitated to obtain green and blue lasers 221 than that for red ones<sup>6</sup>. However, the optical loss including nonradiative AR and carrier trapping 222 becomes more serious and the absorption cross-section reduces as the dot decreases, making the 223 CQDs lase in short visible region challenging<sup>2, 8</sup>. Despite that lasing has been demonstrated from 224 CQDs across the full visible range<sup>6</sup>, the pump thresholds differ dramatically for red, green and 225 blue colors, which hinders the realization of integrated full-color lasers with single pumping 226 source<sup>44</sup>. In contrast, the emission color of CsPbX<sub>3</sub> IPNCs can be facilely tailored by composition 227 control<sup>17, 18</sup>. By inserting the blue-emitting CsPb(Br/Cl)<sub>3</sub> IPNCs and red-emitting CsPb(I/Br)<sub>3</sub> 228 229 IPNCs into the DBR resonators (for the red-emitting CsPb(I/Br)<sub>3</sub> IPNC-VCSEL, the DBR with 230 stop-band centered at 590 nm was employed (Supplementary Fig. 12)), we successfully achieved 231 IPNC-VCSELs across the full visible region (Fig. 5) (see full-range spectra in Supplementary Fig. 13 and Supplementary Fig. 14 for blue and red VCSELs, respectively). Notably, the lasing 232 thresholds for the red (19.0  $\mu$ J/cm<sup>2</sup>) and blue (25.5  $\mu$ J/cm<sup>2</sup>) are comparable to that of the green, 233 which indicates our CsPbX<sub>3</sub> IPNC-VECSELs may hold great promise for developing the single 234 235 source-pumped full-color visible and white lasers.

#### 236 Discussion

We for the first time realized the tough yet practically desirable VCSELs based on the emerging
solution-processable CsPbX<sub>3</sub> IPNCs. The CsPbX<sub>3</sub> IPNC-VCSELs operate at a very low threshold,

239 so that the q-CW pumping is made feasible. Such a low lasing threshold can be mainly attributed 240 to the large absorption cross-section of the  $CsPbX_3$  IPNCs, high PL quantum yield, relatively low Auger loss and the good match between the gain profile and the stop-band of the DBRs<sup>13, 14</sup>. 241 Especially, the absorption cross-section of the CsPbX<sub>3</sub> IPNCs was disclosed to be orders of 242 magnitude higher than those of the metal-chalcogenide CQDs<sup>13, 14, 42</sup>, which allows for the 243 generation of excitons with denser concentration under the same pump density. Therefore, even 244 though the CsPbX<sub>3</sub> IPNC-VCSELs stem from the biexciton recombination<sup>13, 37</sup>, the pump 245 thresholds for lasing still remain several times lower than that of the CdSe/CdZnS CQDs under 246 similar conditions which was claimed to lase in single exciton regime<sup>6</sup>. Another advantage of the 247 CsPbBr<sub>3</sub> IPNCs may be its cubic crystal shape. In general, dense packing of the nanocrystals in 248 249 the ensemble film is essential for high effective gain and thus a low lasing threshold. In this 250 regard, the cubic shape of CsPbBr<sub>3</sub> IPNCs is advantageous over the spherical ones of II-VI group 251 metal-chalcogenide CQDs. The directionality and coherence from the laser device was clearly 252 revealed. Taking advantage of the short cavity length and the narrow gain spectrum ( $\sim 6.5$  nm), single-mode lasing was achieved, attractive for various application fields. Noticeably, it is facile 253 254 to independently choose the cavity characteristics and CsPbX<sub>3</sub> IPNCs for optimal match so that 255 red, green and blue VCSELs can be realized with comparable pump thresholds, which is 256 extremely difficult for epitaxial semiconductors and traditional metal-chalcogenide CQDs. We 257 envisage that further optimization of the CsPbX<sub>3</sub> IPNCs, such as engineering the core/alloyedshell heterostructure to mitigate the Auger and trapping loss<sup>45</sup>, as well as adoption of proper 258 thermal management in the device<sup>36</sup> could further reduce the lasing threshold and may eventually 259 260 enable the continuous wave operation in the future.

#### 261 Methods

Synthesis of CsPbX<sub>3</sub> IPNCs.  $Cs_2CO_3$  (0.8 g), OA (oleic acid, 2.5 mL) and of ODE (octadecene, 30 mL) were mixed and kept in an argon atmosphere. The mixture was heated up to 130 °C for one hour. After that, the mixture was further heated up to 150 °C for reaction and lasted for 0.5 hour. Then, the solution was cooled down to room temperature. On the other side, 10 mL ODE, 1 mL OA and 0.36 mmol of PbX<sub>2</sub> (X=Cl, Br and I) were put together and kept in an argon atmosphere for one hour at temperature of 130 °C. Then, the solution was further heated up to 160 °C for reaction for 10 minutes. Finally, the Cs-precursor (1 mL) was quickly added into the solution and cooled down by ice. More information about the synthesis of CsPbX<sub>3</sub> IPNCs can be
found in reference 13.

271 **Optical characterization.** To investigate the modification of spontaneous emission from  $CsPbX_3$ 272 IPNCs by microcavity effect, the same femtosecond laser source (excitation wavelength: 400 nm, pulse-width: 100 fs, repetition rate: 1000 Hz) was employed as that used in the following lasing 273 274 studies to maintain consistency. The excitation intensity was kept very low (0.5  $\mu$ J/cm<sup>2</sup>), far less than the threshold. We also have tried the continuous wave He-Cd laser as the excitation source, 275 276 where similar spiked emission was observed, further confirming the spiked emission is not laser. 277 The angle dependent PL signal was recorded as a function of detection direction by a home-built 278 fiber-optics system. In particular, the collection fiber with diameter of 200 µm was attached to a 279 50 mm diameter rotating stage with the sample mounted at its center. The distance between the 280 fiber and the sample is set as 15 mm. The other side of the fiber was coupled to a 320 mm 281 monochromator combined with a charge coupled device detector. For the lasing investigation, the 282 laser beam with wavelength of 400 nm generated by second harmonic generation from the seed 283 with wavelength of 800 nm was focused by a circular lens (focus length: 5 cm) onto the VCSEL 284 vertically, and the spot diameter on the sample was  $\sim 90 \,\mu$ m. The output signal was collected from 285 the other side of the VCSEL by the above-mentioned fiber-optics system.

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Figure 1| Modification of spontaneous emission from CsPbBr<sub>3</sub> IPNCs by presence of microcavity. (a) PL spectra from CsPbBr<sub>3</sub> IPNCs within (red line)/without (black line) microcavity under low excitation intensity of 0.5  $\mu$ J/cm<sup>2</sup>. The blue dashed line represents the reflectivity of the adopted DBR. (b) Schematic configuration of the CsPbBr<sub>3</sub> IPNC-VCSEL.



Figure 2| Angle-resolved PL from CsPbBr<sub>3</sub> IPNCs within microcavity and PL dynamics. (a)
Detection angle dependent PL spectra from the CsPbBr<sub>3</sub> IPNCs within microcavity. (b) Spatial
distribution of the irradiation from CsPbBr<sub>3</sub> IPNCs within/without microcavity. The PL signal
greatly narrows with the presence of microcavity. The divergence is estimated to be 15° by
Gaussian fitting (red line). (c) PL decay curves of CsPbBr<sub>3</sub> IPNCs within microcavity (mode
number from 30 to 32) and that without microcavity.



Figure 31 Vertical cavity surface emitting lasers from CsPbBr<sub>3</sub> IPNCs. (a) Pump intensity dependent PL spectra from the device. Insets shows the output beam profiles below  $(5.4 \,\mu\text{J/cm}^2)$ (upper one) and above  $(25.8 \,\mu\text{J/cm}^2)$  (bottom one) threshold. Scale bar is 1 mm. (b) Pump intensity dependent FWHM of the modes indexed from 29 to 33. (c) Pump intensity dependent PL intensity of the modes indexed from 29 to 33. (d) Pump intensity dependent PL decay curves of the device  $(0.1 P_{th}$  for the black curve,  $0.8 P_{th}$  for the red curve and  $1.3 P_{th}$  for the green curve)

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Figure 4 Lasing operation in quasi-continuous wave regime and multi-mode lasing. (a) Nanosecond laser pumped PL spectra as a function of pump intensity. The inset shows the integrated PL intensity over the sharp peak versus pump intensity. (b) Multi-mode lasing from the CsPbBr<sub>3</sub> IPNC-VCSEL. The inset shows the integrated PL intensity over the sharp peaks versus pump intensity.

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**Figure 5** Red and blue vertical cavity surface emitting lasers from CsPbX<sub>3</sub> IPNCs. Red and blue lasing spectra of vertical cavity surface emitting lasers from CsPb(I/Br)<sub>3</sub> and CsPb(Br/Cl)<sub>3</sub> IPNCs under pump intensity of 30.5 and 38.2  $\mu$ J/cm<sup>2</sup>, respectively. The corresponding photograph of the device in operation is attached inside the plot.

352	1.	Tessler, N., Denton, G.J. & Friend, R.H. Lasing from conjugated-polymer microcavities.
353		<i>Nature</i> <b>382</b> , 695-697 (1996).
354	2.	Klimov, V.I. et al. Optical gain and stimulated emission in nanocrystal quantum dots.
355		Science <b>290</b> , 314-317 (2000).
356	3.	Wang, Y. et al. Stimulated Emission and Lasing from CdSe/CdS/ZnS Core-Multi-Shell
357		Quantum Dots by Simultaneous Three-Photon Absorption. Adv. Mater. 26, 2954–2961
358		(2014).
359	4.	Grim, J.Q. et al. Continuous-wave biexciton lasing at room temperature using solution-
360		processed quantum wells. Nature Nanotech. 9, 891-895 (2014).
361	5.	Grivas, C. et al. Single-mode tunable laser emission in the single-exciton regime from
362		colloidal nanocrystals. Nature Commun. 4, 2376 (2013).
363	6.	Dang, C. et al. Red, green and blue lasing enabled by single-exciton gain in colloidal
364		quantum dot films. Nature Nanotech. 7, 335-339 (2012).
365	7.	Wang, Y. et al. Blue Liquid Lasers from Solution of CdZnS/ZnS Ternary Alloy Quantum
366		Dots with Quasi-Continuous Pumping. Adv. Mater. 27, 169-175 (2015).
367	8.	Klimov, V.I., Mikhailovsky, A.A., McBranch, D.W., Leatherdale, C.A. & Bawendi, M.G.
368		Quantization of multiparticle Auger rates in semiconductor quantum dots. Science 287,
369		1011-1013 (2000).
370	9.	Lee, M.M., Teuscher, J., Miyasaka, T., Murakami, T.N. & Snaith, H.J. Efficient Hybrid
371		Solar Cells Based on Meso-Superstructured Organometal Halide Perovskites. Science
372		<b>338</b> , 643-647 (2012).
373	10.	Zhou, H. et al. Interface engineering of highly efficient perovskite solar cells. Science
374		<b>345</b> , 542-546 (2014).
375	11.	Deschler, F. et al. High Photoluminescence Efficiency and Optically Pumped Lasing in
376		Solution-Processed Mixed Halide Perovskite Semiconductors. J. Phys. Chem. Lett. 5.
377		1421-1426 (2014).
378	12.	Zhu, H. et al. Lead halide perovskite nanowire lasers with low lasing thresholds and high
379		quality factors. <i>Nature Mater.</i> <b>14</b> , 636-642 (2015).
380	13.	Wang, Y. et al. All-Inorganic Colloidal Perovskite Quantum Dots: A New Class of
381		Lasing Materials with Favorable Characteristics. Adv. Mater. 27, 7101–7108 (2015).
382	14.	Yakunin, S. et al. Low-threshold amplified spontaneous emission and lasing from
383		colloidal nanocrystals of caesium lead halide perovskites. <i>Nature Commun.</i> 6 (2015).
384	15.	Li, X. et al. CsPbX3 Quantum Dots for Lighting and Displays: Room-Temperature
385		Synthesis, Photoluminescence Superiorities, Underlying Origins and White Light-
386		Emitting Diodes. Adv. Funct. Mater. 26, 2435–2445 (2016).
387	16.	Song, J. et al. Ouantum Dot Light-Emitting Diodes Based on Inorganic Perovskite
388		Cesium Lead Halides (CsPbX3). Adv. Mater. 27, 7162–7167 (2015).
389	17.	Protesescu, L. et al. Nanocrystals of Cesium Lead Halide Perovskites (CsPbX3, X = Cl.
390		Br. and I): Novel Optoelectronic Materials Showing Bright Emission with Wide Color
391		Gamut. Nano Lett. (2015).
392	18.	Akkerman, O.A. et al. Tuning the Optical Properties of Cesium Lead Halide Perovskite
393		Nanocrystals by Anion Exchange Reactions. J. Am. Chem. Soc. 137, 10276-10281 (2015).
394	19.	Nedelcu, G. et al. Fast Anion-Exchange in Highly Luminescent Nanocrystals of Cesium
395		Lead Halide Perovskites (CsPbX3, X = Cl, Br, I). <i>Nano Lett.</i> <b>15</b> , 5635-5640 (2015).
396	20.	Swarnkar, A. et al. Colloidal CsPbBr3 Perovskite Nanocrystals: Luminescence beyond
397		Traditional Ouantum Dots. Angew. Chem., n/a-n/a (2015).
398	21.	Eaton, S.W. et al. Lasing in robust cesium lead halide perovskite nanowires. <i>Proc. Nat.</i>
399		Acad. Sci. U.S.A. 113, 1993-1998 (2016).
400	22.	Eaton, S.W., Fu, A., Wong, A.B., Ning, CZ. & Yang, P. Semiconductor nanowire lasers.
401		Nature Reviews Materials 1, 16028 (2016).

402	23.	Feng, L., Wong, Z.J., Ma, RM., Wang, Y. & Zhang, X. Single-mode laser by parity-
403		time symmetry breaking. Science 346, 972-975 (2014).
404	24.	Xiao, Y. et al. Single-Nanowire Single-Mode Laser. <i>Nano Lett.</i> <b>11</b> , 1122-1126 (2011).
405	25.	Sale, T.E. & Sale, T.E. Vertical cavity surface emitting lasers. (Research Studies Press,
406		1995).
407	26.	Yoshikawa, T. et al. Complete polarization control of 8×8 vertical-cavity surface-
408		emitting laser matrix arrays. Appl. Phys. Lett. 66, 908-910 (1995).
409	27.	Chen, R., Sun, H.D., Wang, T., Hui, K.N. & Choi, H.W. Optically pumped ultraviolet
410		lasing from nitride nanopillars at room temperature. Appl. Phys. Lett. 96, 241101 (2010).
411	28.	Sun, H.D. et al. Low-loss 1.3-µm GaInNAs saturable Bragg reflector for high-power
412		picosecond neodymium lasers. Opt. Lett. 27, 2124-2126 (2002).
413	29.	Calvez, S. et al. 1.3 μm GaInNAs optically-pumped vertical cavity semiconductor
414		optical amplifier. Electron. Lett 39, 100-102 (2003).
415	30.	Chen, S. et al. Gain-switching dynamics in optically pumped single-mode InGaN
416		vertical-cavity surface-emitting lasers. Opt. Express 22, 4196-4201 (2014).
417	31.	Marra, D.C., Aydil, E.S., Joo, SJ., Yoon, E. & Srdanov, V.I. Angle-dependent
418		photoluminescence spectra of hydrogenated amorphous silicon thin films. Appl. Phys.
419		Lett. 77, 3346-3348 (2000).
420	32.	Qiao, H. et al. Optical properties of II-VI colloidal quantum dot doped porous silicon
421		microcavities. Appl. Phys. Lett. 96, 161106 (2010).
422	33.	Wang, Y. et al. Nonlinear Absorption and Low-Threshold Multiphoton Pumped
423		Stimulated Emission from All-Inorganic Perovskite Nanocrystals. Nano Lett. 16, 448-453
424		(2016).
425	34.	Samuel, I.D.W., Namdas, E.B. & Turnbull, G.A. How to recognize lasing. Nature Photon.
426		<b>3</b> , 546-549 (2009).
427	35.	Chen, Q., Kiraz, A. & Fan, X. Optofluidic FRET lasers using aqueous quantum dots as
428		donors. Lab on a Chip 16, 353-359 (2016).
429	36.	Adachi, M.M. et al. Microsecond-sustained lasing from colloidal quantum dot solids.
430		<i>Nature Commun.</i> <b>6</b> (2015).
431	37.	Makarov, N.S. et al. Spectral and Dynamical Properties of Single Excitons, Biexcitons,
432		and Trions in Cesium-Lead-Halide Perovskite Quantum Dots. Nano Lett. 16, 2349-2362
433		(2016).
434	38.	Liao, Q. et al. An Organic Microlaser Array Based on a Lateral Microcavity of a Single
435		J-aggregation Microbelt. Angew. Chem. Int. Ed. 54, 7037-7041 (2015).
436	39.	Mhibik, O. et al. An ultra-narrow linewidth solution-processed organic laser. Light Sci
437		<i>Appl</i> <b>5</b> , e16026 (2016).
438	40.	Xu, D. et al. Polariton lasing in a ZnO microwire above 450 K. Appl. Phys. Lett. 104,
439		082101 (2014).
440	41.	Oron, D., Kazes, M. & Banin, U. Multiexcitons in type-II colloidal semiconductor
441		quantum dots. Phys. Rev. B 75, 035330 (2007).
442	42.	Xu, Y. et al. Two-Photon-Pumped Perovskite Semiconductor Nanocrystal Lasers. J. Am.
443		<i>Chem. Soc.</i> <b>138</b> , 3761-3768 (2016).
444	43.	Kemp, A.J. et al. Thermal management in vertical-external-cavity surface-emitting lasers:
445		finite-element analysis of a heatspreader approach. IEEE J. Quantum Electron. 41, 148-
446		155 (2005).
447	44.	Fan, F., Turkdogan, S., Liu, Z., Shelhammer, D. & Ning, C.Z. A monolithic white laser.
448		Nature Nanotech. 10, 796-803 (2015).
449	45.	Wang, Y. et al. Unraveling the ultralow threshold stimulated emission from CdZnS/ZnS
450		quantum dot and enabling high-Q microlasers. Laser Photonics Rev. 9, 507-516 (2015).

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#### 458 Author contribution

All authors contributed extensively to this work. Y.W. and V.N. conducted spectroscopic
experiment and analysis. Y.W. and H.S. conceived and fabricated the VCSEL structure. X.L. and
H.Z. synthesized the IPNC samples. H.S. and H.Z. supervised the project. Y.W., X.L., V.N., H.Z.,
and H.S. analyzed data and interpreted the results, and wrote the manuscript with input from all
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