

Research Article

Structural and Optical Properties of Aluminum Nitride Thin Films Deposited by Pulsed DC Magnetron Sputtering

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Aluminum nitride thin films were deposited on Si (100) substrate by pulsed DC (asymmetric bipolar) reactive magnetron sputtering under variable nitrogen flow in a gas mixture of argon and nitrogen. The deposited film was characterized by grazing incidence X-ray diffraction (GIXRD), atomic force microscope (AFM), spectroscopic ellipsometry, and secondary ion mass spectroscopy (SIMS). GIXRD results have shown (100) reflection of wurtzite AlN, whereas AFM micrographs have revealed very fine grained microstructure with average roughness in the range 6–8 nm. Spectroscopic ellipsometry measurements have indicated the band gap and refractive index of the film in the range 5.0–5.48 eV and 1.58–1.84, respectively. SIMS measurement has indicated the presence of oxygen in the film.

1. Introduction

Aluminum nitride, a III-V family compound, has excellent combination of physical, chemical, and mechanical properties. High-quality films of aluminum nitride have been used in various devices and sensors including the optical and optoelectronic devices. As far as the optical and optoelectronic applications are concerned, wide band gap (~6.2 eV) along with high-refractive index (~2.0) and low-absorption coefficient ($<10^{-3}$) makes AlN a very attractive material for these applications [1]. In addition to this, thermal and chemical stability of AlN films make it suitable for applications in difficult environment. Today, AlN films/coatings have been grown by several methods which include pulsed laser deposition [2], reactive molecular beam epitaxy [3], vacuum arc/cathodic arc deposition [4], DC/RF reactive sputtering [5–7], ion beam sputtering [8], metal-organic chemical vapor deposition (MOCVD) [9], and miscellaneous [10] other techniques. Due to simplicity, reproducibility, ease of scaling up, and lower cost, magnetron sputtering is one of the common methods for growing AlN films for various applications. As already known, it is difficult to obtain good

quality insulating films by DC magnetron sputtering, and RF magnetron sputtering has the disadvantage of lower deposition rate and higher cost of the RF power, whereas pulsed DC magnetron sputtering method has the advantage of higher deposition rate and it is suitable for producing good quality cost-effective dielectric films [11, 12].

Properties of AlN films depend upon the crystal structure, crystal orientation, microstructure, and chemical composition, which in turn depend upon the deposition conditions such as sputtering power, pulse frequency, duty cycle, growth temperature, nitrogen/argon flow ratio, and sputtering gas pressure. Effect of nitrogen/argon flow ratio on the optical properties of AlN films have been widely reported for DC and RF magnetron sputtering [5, 6, 13], but the same has not been studied in detail for pulsed DC magnetron sputtering. Therefore, in this study, we have deposited aluminum nitride thin films by pulsed DC reactive magnetron sputtering for a wide range of nitrogen/argon flow ratio and their effect on optical properties has been studied using the spectroscopic ellipsometry technique. Also, in the same work, study on structure and morphology of the film has been conducted using GIXRD and AFM.

2. Experimental Methods

A reactive pulsed DC balanced magnetron sputtering system coupled with asymmetric bipolar DC generator (reverse bias voltage: +36 V) was employed to deposit the AlN film with 99.99% pure aluminum used as a target material. Properly cleaned single crystal silicon wafer having (100) orientation was used as the substrate. Final cleaning of the substrate was done in situ by Ar⁺ ion bombardment for 30 min at a biasing voltage of -800 V and a pressure of 10⁻¹ mbar. Deposition was carried out at a working pressure of 1.5 × 10⁻³ mbar in argon and nitrogen atmosphere. Numerical values of key deposition parameters are listed in Table 1.

Crystal structure of the deposited film was revealed by GIXRD measurement at monochromatized CuK_α (λ = 0.154 nm) wavelength for an incidence angle of 1°. The experimental GIXRD patterns were matched with JCPDS file and the peak of aluminum nitride was identified. Surface morphology and roughness of the film were analyzed by AFM in tapping mode. Optical properties were measured using a phase modulated spectroscopic ellipsometer in the wavelength range 300–1200 nm. Chemical composition of the film was qualitatively analyzed by SIMS.

3. Results and Discussion

The rate of deposition of AlN (calculated as thickness measured in ellipsometry divided by the deposition time) for various flow ratio of nitrogen/argon is shown in Figure 1. It is clear that deposition rate decreases continuously with increase in flow ratio of nitrogen/argon mainly because of the increased proportion of N⁺ or N₂⁺ ions which transfer lower momentum to the target compared to the massive Ar⁺ ion; consequently, energy and fluence of the sputtered atoms decrease and this leads to a gradual fall in the growth rate of the film [14].

The AlN films deposited for different flow ratio of nitrogen/argon had different colours due to variation in stoichiometry [15]. Films deposited at a higher flow ratio were violet, whereas for intermediate flow ratios, colour was blue and for still lower flows ratio, it was yellow. Exact variation in colour is given in Table 2.

3.1. GIXRD and AFM Results. The GIXRD pattern is shown in Figure 2. It has been found that, for all flow ratio of nitrogen/argon, only the low intensity (100) reflection of wurtzite hexagonal phase of AlN was found with an appreciable amount of shift. This may be due to poor crystallinity of films with large amount of stress present, probably due to the low deposition temperature and sputtering power. In the present study, we could not obtain (001) reflection at any nitrogen/argon flow ratio. The reason may be that the (001) plane is the lowest energy plane with the highest atom density which requires high adatom mobility for its growth and the same can be achieved at a higher growth temperature and target power.

Figures 3(a)–3(c) shows the AFM micrographs. The micrographs show that at 80% flow ratio of nitrogen/argon,

TABLE 1: Deposition parameters for growth of AlN thin film.

Parameter	Numerical value
Base pressure	1.1 × 10 ⁻⁵ mbar
Power density of cathode	3.3 W/cm ²
Pulse frequency	125 kHz
Duty cycle	75%
Target-to-substrate distance	7.5 cm
Total gas (Ar + N ₂) flow rate	10 sccm
Substrate temperature	No external heating (~100°C)

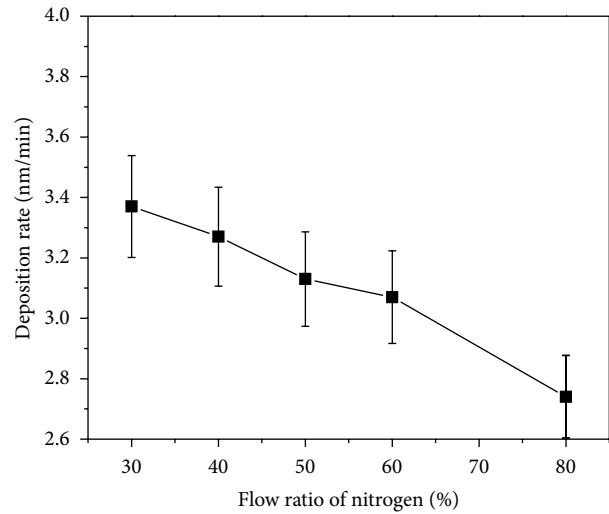


FIGURE 1: Deposition rate of AlN for different nitrogen/argon flow ratio.

TABLE 2: Variation in colour of AlN films with nitrogen/argon flow ratio.

N ₂ /Ar flow ratio (%)	Deposition time (min)	Colour of the film
30	60	Yellow
40	60	Pink
50	90	Blue
60	90	Blue
80	90	Violet

surface morphology was uniform with random presence of elongated grains at few locations, whereas at 50 and 30% flow ratio, grains were of spherical and conical shapes, respectively. The measured average roughness (R_a) values showed no marked effect of varying the nitrogen/argon flow ratio; for all the films, R_a values were in the range 6–8 nm. Some authors have reported decreased roughness of AlN film with increasing flow ratio of nitrogen/argon, attributing it to the lower growth rate of the film [14].

3.2. Spectroscopic Ellipsometry Results. In ellipsometry, the optical property is measured by measuring the complex

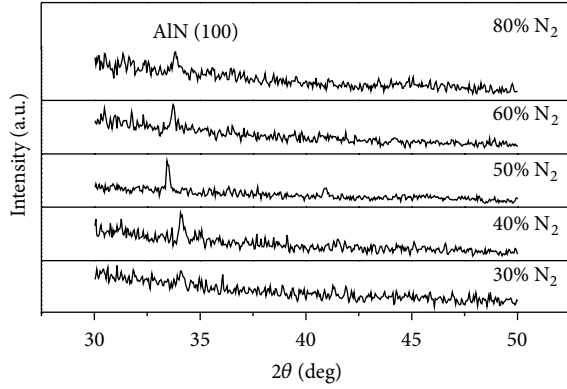


FIGURE 2: GIXRD pattern of AlN film deposited for various flow ratio of nitrogen/argon.

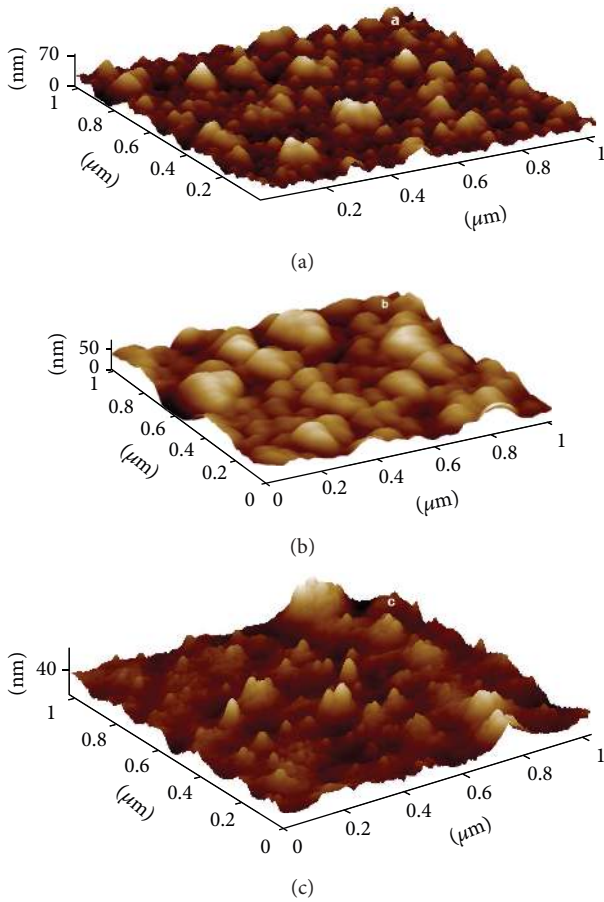


FIGURE 3: AFM micrographs of AlN film deposited for (a) 30, (b) 50, and (c) 80% flow ratio of nitrogen/argon.

reflectance ratio of incident light on a sample [16, 17]. The reflectance ratio is defined as

$$\rho = \frac{r_p}{r_s} = \tan(\psi) e^{i\Delta}. \quad (1)$$

Here, $\tan(\psi)$ is known as the amplitude ratio and Δ is called the phase shift and r_p and r_s represent the complex Fresnel

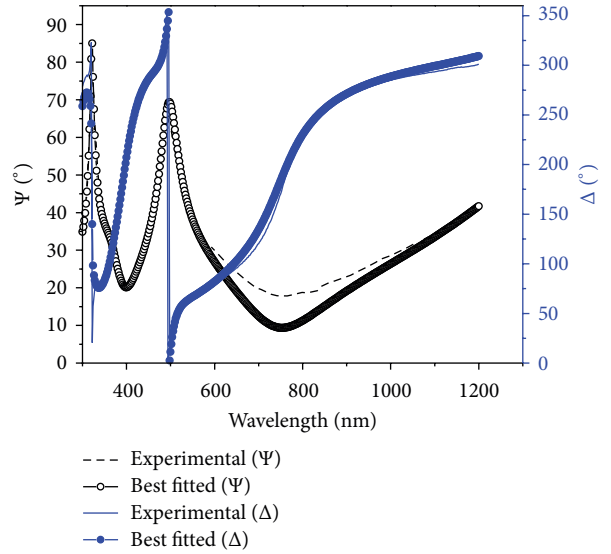


FIGURE 4: Experimental Ψ and Δ versus wavelength along with best-fit theoretical curve for AlN thin film deposited on Si (100) at 80% flow ratio of nitrogen/argon. Fitting was done with Tauc-Lorentz model.

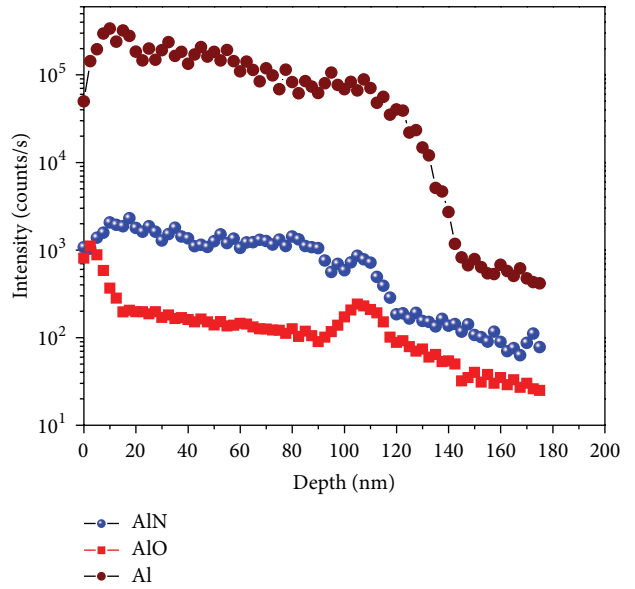


FIGURE 5: SIMS composition depth profile of AlN thin film deposited on Si (100).

coefficients for light parallel and perpendicular to the plane of incidence, respectively.

Figure 4 shows the experimental and the best fitted spectra of the ellipsometric parameters Ψ and Δ . During fitting, a two-layer sample structure was assumed on the silicon substrate. The top layer was assumed to be 50% void and 50% AlN, whereas, the lower layer was assumed as bulk (100% AlN). This is a well established technique in ellipsometry. The optical property of AlN film was generated

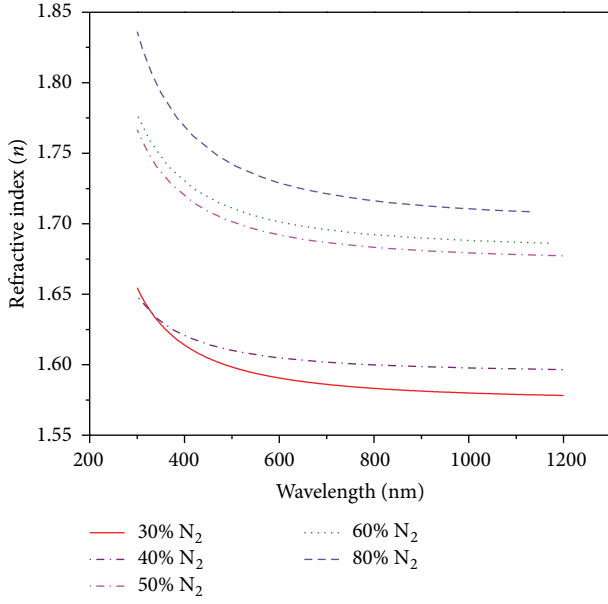


FIGURE 6: Nitrogen/argon flow ratio-dependent refractive indices of AlN thin film.

using the Tauc-Lorentz model. This model was used due to the poor crystalline nature of the deposited film.

The dielectric response of a material which is ultimately related to the ellipsometric parameters Ψ and Δ is described by a complex function as follows:

$$\varepsilon = \varepsilon_1 - i\varepsilon_2. \quad (2)$$

The expression for real part is described by Kramers-Kronig equation [18], which is written as

$$\varepsilon_1 = \varepsilon_\infty + \frac{2}{\pi} P \int_{E_g}^{\infty} \frac{\xi \varepsilon_2(\xi)}{\xi^2 - E^2} d\xi. \quad (3)$$

Here, ε_∞ represents the contribution of the optical transition at a higher energy.

The Tauc-Lorentz model gives an expression for the imaginary part (ε_2) [18] as follows

$$\varepsilon_2 = \begin{cases} \frac{AE_0\Gamma(E - E_g)^2}{(E^2 - E_0^2)^2 + \Gamma^2 E^2} \frac{1}{E} & (E > E_g) \\ 0 & (E \leq E_g), \end{cases} \quad (4)$$

where E_g is the band gap, E_0 is the peak transition energy, Γ is the broadening parameter, and A is the optical transition matrix element.

Table 3 shows the measured ellipsometric data of aluminum nitride film. From this table, it can be observed that there is a decrease in band gap from 5.48 to 5.00 eV and a substantial decrease in ε_∞ (optical contribution to the dielectric function at a higher photon energy) when the nitrogen/argon flow ratio was increased from 30 to 80%. Similar results on band gap variation of AlN thin films have been reported by García-Méndez et al. [19]. However, our results are contrary

TABLE 3: Measured ellipsometric data of wurtzite AlN.

N ₂ /Ar flow ratio (%)	Thickness (nm)	Band gap E_g (eV)	ε_∞
30	202	5.48	1.47
40	196	5.48	1.79
50	282	5.40	1.47
60	276	5.47	1.39
80	247	5.00	0.94

to those of Cho [13] and Wang et al. [15], who have used RF magnetron sputtering technique. Cho has attributed the increase in band gap to consistently improved crystallinity and stoichiometry of the films. However, Dumitru et al. [20] and Drüsedau and Koppenhagen [21] have observed different optical properties of AlN films deposited by DC and RF magnetron sputtering under identical operating conditions. A probable explanation for the decrease in band gap as observed in our work could be as follows. In our experiments, although high-purity argon and nitrogen gases (<2 vpm oxygen) were used, nitrogen was further purified to a vpb level of oxygen impurity before introducing it into the deposition chamber. This resulted in lower oxygen content in the chamber/deposit at higher nitrogen/argon flow ratios. This, in turn, reduced the proportion of the more ionic Al-O bonds over the Al-N bonds in the deposit which ultimately resulted in decreased band gap. Our explanation for decrease in band gap of AlN is based on the similar explanation given for TiO_xN_y films by Mohamed et al. [22]. To detect AlO in the film, one sample was analyzed in SIMS. The SIMS measurement (Figure 5) confirmed presence of AlO together with AlN in the deposited film.

Figure 6 shows the plot of refractive index versus wavelength of light for AlN films; it is clear that refractive index of sputtered AlN films depends on flow ratio of nitrogen/argon greatly. For a flow ratio greater than 50%, refractive index of the films has been in the range 1.67–1.84, which is lower than the reported values mostly because of the porous nature of the film. For lower flow ratios, refractive index of AlN has been observed to decrease because of the formation of nonstoichiometric films [23].

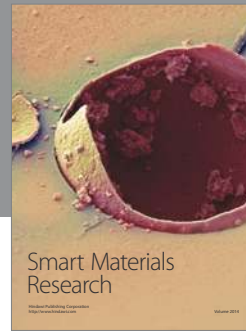
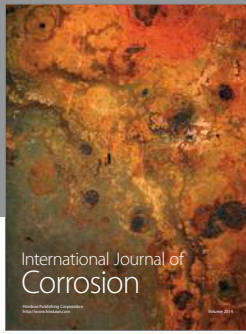
4. Conclusions

AlN thin films were deposited on silicon substrates by pulsed DC reactive magnetron sputtering. The deposited film has shown wurtzite crystal structure with (100) reflection, and flow ratio of nitrogen/argon has been found to have a marked effect on its optical properties. The deposited films exhibited a decreased band gap from 5.48 to 5.0 eV with increasing flow ratio of nitrogen/argon. Similarly, refractive index of the films was in the range 1.58–1.84.

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