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## Structural derivation and crystal chemistry of apatites

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The crystal structures of the  $[A(1)_2][A(2)_3](BO_4)_3X$  apatites and the related compounds  $[A(1)_2][A(2)_3](BO_5)_3X$  and  $[A(1)_2][A(2)_3](BO_3)_3X$  are collated and reviewed. The structural aristotype for this family is Mn<sub>5</sub>Si<sub>3</sub> (D8<sub>8</sub> type,  $P6_3/mcm$  symmetry), whose cation array approximates that of all derivatives and from which related structures arise through the systematic insertion of anions into tetrahedral, triangular or linear interstices. The construction of a hierarchy of spacegroups leads to three apatite families whose high-symmetry members are  $P6_3/m$ , Cmcm and  $P6_3$ cm. Alternatively, systematic crystallographic changes in apatite solid-solution series may be practically described as deviations from regular anion nets, with particular focus on the O(1)-A(1)-O(2)twist angle  $\varphi$  projected on (001) of the  $A(1)O_6$  metaprism. For apatites that contain the same A cation, it is shown that  $\varphi$ decreases linearly as a function of increasing average ionic radius of the formula unit. Large deviations from this simple relationship may indicate departures from  $P6_3/m$  symmetry or cation ordering. The inclusion of  $A(1)O_6$  metaprisms in structure drawings is useful for comparing apatites and condensed-apatites such as Sr<sub>5</sub>(BO<sub>3</sub>)<sub>3</sub>Br. The most common symmetry for the 74 chemically distinct  $[A(1)_2][A(2)_3]$ - $(BO_4)_3X$  apatites that were surveyed was  $P6_3/m$  (57%), with progressively more complex chemistries adopting  $P6_3$  (21%),  $P\bar{3}$  (9%),  $P\bar{6}$  (4.3%),  $P2_1/m$  (4.3%) and  $P2_1$  (4.3%). In chemically complex apatites, charge balance is usually maintained through charge-coupled cation substitutions, or through appropriate mixing of monovalent and divalent X anions or X-site vacancies. More rarely, charge compensation is achieved through insertion/removal of oxygen to produce BO<sub>5</sub> square pyramidal units (as in ReO<sub>5</sub>) or BO<sub>3</sub> triangular coordination (as in AsO<sub>3</sub>). Polysomatism arises through the ordered filling of [001] BO<sub>4</sub> tetrahedral strings to generate the apatite-nasonite family of structures.

#### 1. Introduction

Although the 'apatites' are an industrially important group of materials with applications in catalysis, environmental remediation, bone replacement and ceramic membranes amongst others, to our knowledge there has been no recently published collation of crystallographic data that systematizes the entire range of chemistries and compositions that have been studied to date. In general, the apatite varieties have not been placed in the context of the entire family, with a view to understanding their crystallographic derivation from simpler structures.

The apatite prototype  $Ca_5(PO_4)_3F$  structure was first determined by Naray-Szabo (1930) and was confirmed to adopt  $P6_3/m$  symmetry. An extensive apatite compilation was

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© 2003 International Union of Crystallography Printed in Great Britain – all rights reserved prepared by Wyckoff (1965), who noted that the lack of thorough structure determinations left many unanswered questions about the distribution of metals and oxyanions. Subsequently, numerous studies of biological, mineralogical and synthetic apatites and apatiterelated substances greatly expanded the range of known chemistries. McConnell (1973) summarized the work up to the 1970s and highlighted the limitations associated with failing to consider the underlying chemical principles that govern apatite crystallography and the need for precise chemical analyses to properly account for apatite properties. Elliot (1994) updated and expanded the work of McConnell by covering a wider range of apatites, reviewing iso- and altervalent substitutions, detailing the identification compounds of lower symmetry. Since then there has been a substantial increase in the number of apatites for which excellent crystallographic data exist. However, the extent of these chemistries, which are conventionally regarded as conforming to

 $A_5(BO_4)_3X$ , and the relative complexity of the structures have at times obscured either the underlying crystallochemical principles that govern the formation of apatite or the important systematic changes in crystallography that appear as a function of compositional variation.

In general, structure drawings highlight only the relatively regular BO<sub>4</sub> tetrahedra, while other features are usually grouped as 'irregular' polyhedra. In this paper, we have collated the data of more than 85 chemically distinct apatites for which crystallographic determinations have been completed. These structures are described as anion-stuffed cation arrays of the D8<sub>8</sub> alloy type (Wondratschek et al., 1964; O'Keeffe & Hyde, 1985; Vegas & Jansen, 2002) and as derivatives of hexagonal anion networks (Povarennykh, 1972). The former approach formalizes the structural hierarchy of apatites with the Mn<sub>5</sub>Si<sub>3</sub> structure type at the apex, while the latter proves useful for systematizing topological distortions as a function of average ionic radius. These descriptions also readily accommodate  $A_5(BO_3)_3X$  and  $A_5(BO_5)_3X$  derivatives and apatite polysomatism. This paper summarizes the current knowledge of apatite structures, particularly with respect to the literature that has appeared since the publication of Elliot's work, and complements the work of Nriagu & Moore (1984) and Brown & Constantz (1994), who deal with phosphate compounds.

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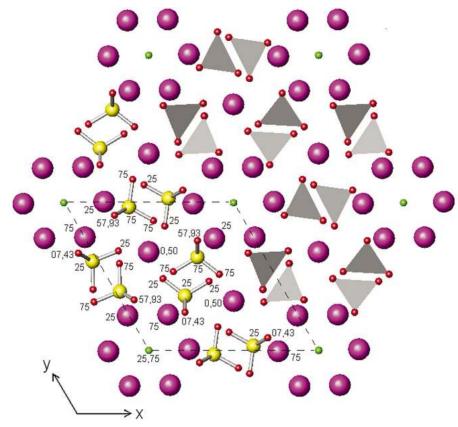


Figure 1 The [001] projection of fluorapatite  $Ca_5(PO_4)_3F$  represented in the conventional way with  $PO_4$  units highlighted, as ball and stick or tetrahedral representations, and with the oxygen coordination to the A cations de-emphasized.

#### 2. Structural derivation

## 2.1. The conventional description

The two most common representations of  $P6_3/m$  apatite – in this case  $Ca_5(PO_4)_3F$  (Sudarsanan  $et\ al.$ , 1972)<sup>1</sup> – are shown in Fig. 1. In general, drawings are projected along [001] and give prominence to the  $BO_4$  tetrahedra, either in ball-and-stick or tetrahedral display. This projection highlights aspects of the symmetry and the regular polyhedra, but usually the coordination of the A cations is not emphasized. Rather, the A(1) ions at the 4f position are described as having coordination to nine O atoms (six near and three more distant), while the A(2) ions on the 6h position are eight coordinated (to seven O atoms and one F atom). For fluorapatite, the F ion lies at position 2a with  $z=\frac{1}{4}$ , while for larger ions, such as Cl, the 2b position with z=0 is occupied. In the latter case, the symmetry can be reduced to monoclinic  $P2_1/b$  with the c axis unique and the X ion offset along [001] to statistically occupy the 4e

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 $<sup>^{1}</sup>$  The refined stoichiometry is  $\mathrm{Ca_{4.938}(PO_4)_3F_{0.906}}.$  It seems likely that many, or perhaps most, of the apatites that are reported as having integral stoichiometry are in fact somewhat non-stoichiometric, particularly with respect to X anions. It is, however, often difficult to determine such compositional variations directly. For this paper, which is concerned primarily with topological arrangements, small deviations from stoichiometry are not directly relevant.

 Table 1

 Bond lengths (Å) of anion-centred coordination polyhedra.

Ideal hexago	nal net	Chlorapatite		Fluorapatite		
O(1)-A(1) O(1)-B	2.300 × 2 1.675	O(1)-Ca(1) O(1)-P O(1)-Ca(2)	2.335 × 2 1.855 2.860	O(1)-Ca(1) O(1)-P O(1)-Ca(2)	2.391 × 2 1.534 2.818	
O(2)-A(1) O(2)-B O(2)-B O(2)-A(2)	2.300 × 2 2.901 1.592 3.138	O(2)-Ca(1) O(2)-P O(2)-P O(2)- Ca(2)	2.400 × 2 2.886 1.515 2.335	O(2)-Ca(1) O(2)-P O(2)-P O(2)-Ca(2)	2.451 × 2 3.147 1.540 2.384	
O(3)-A(1) O(3)-B O(3)-A(2) O(3)-A(2)	3.042 2.170 2.303 2.365	O(3)-Ca(1) O(3)-P O(3)-Ca(2) O(3)-Ca(2)	2.894 1.592 2.292 2.455	O(3)-Ca(1) O(3)-P O(3)-Ca(2) O(3)-Ca(2)	2.800 1.533 2.384 2.395	
X-A(2)	2.239 × 6	Cl-Ca(2)	2.932 × 6	F-Ca(2)	2.229 × 3	

position to yield an  $A(2)O_7Cl$  polyhedron [e.g. Bauer & Klee (1993), but see also Kim et al. (2000)]. These descriptions, while certainly correct, may not be entirely satisfactory when we seek to relate systematic crystallographic changes as a function of composition or to study structural relationships between apatites. These limitations have been discussed previously by O'Keeffe & Hyde (1985).

## 2.2. An anion-stuffed D8<sub>8</sub> alloy

Alternatively, apatite can be regarded as an anion-stuffed derivative of the  $Mn_5Si_3$  ( $D8_8$ ) alloy type (Wondratschek *et al.*, 1964; O'Keeffe & Hyde, 1985; Nyman & Andersson, 1979; Vegas *et al.*, 1991). Here the  $A_5B_3$  arrangement is emphasized, as in Fig. 2, where the  $Ca_5P_3$  topology of  $Ca_5(PO_4)_3F$  is shown.

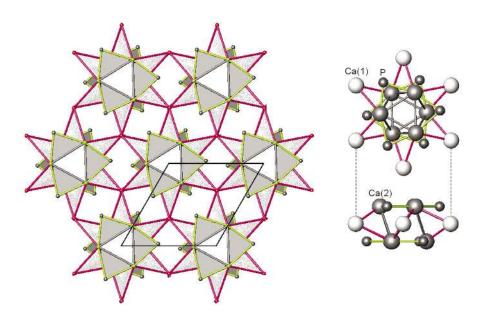


Figure 2 The structure of fluorapatite projected down [001]. We show only the  $D8_8$  cation array represented as corner-connected Ca(2) octahedra that are capped on all 12 edges by Ca(1) and P. On the right a single structural  $Ca_{12}P_6$  unit is shown projected on [001] in the upper portion and [110] in the lower.

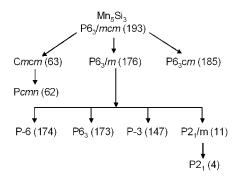


Figure 3
Symmetry relationships for apatite hettotypes that are derived from the Mn<sub>5</sub>Si<sub>3</sub> aristotype. [Adapted from Schriewer & Jeitschko (1993).]

This description highlights the regular arrangement of cations and uses a Ca(2)<sub>6</sub> octahedron with all edges capped by Ca(1) and P as the fundamental unit. These units are cornerconnected in (001) and face sharing along [001]. Anions are then introduced into cavities in the alloy. In fluorapatite, oxygen ions fill tetrahedral [O(1) and O(3)] or trigonal bipyramids [O(2)] while fluorine occupies the triangular interstices between the metals. In chlorapatite the halide moves from triangular to octahedral coordination. A comparison of the bond lengths in these anion-centred polyhedra is summarized in Table 1. Predictably, the oxygen polyhedra are highly distorted since they include A(1), A(2)and B cations, whereas the X-centred triangles and octahedra are regular as they involve X-A(2) bonding only. A similar description has been presented by Schriewer & Jeitschko (1993), who considered the arrangement of the  $A_5B_3X$  portion of apatite to be similar to that of the carbide Mo<sub>5</sub>Si<sub>3</sub>C. The

underlying physical basis for the stuffed-alloy model for apatites has recently been considered by Vegas & Jansen (2002), who list the alloy equivalents of 14 varieties.

If the  $D8_8$  alloy structure with space group P6<sub>3</sub>/mcm is taken as the aristotype – where this terminology is analogous to that adopted to describe perovskites (Lefkowitz et al., 1966; Megaw, 1973) - it follows that anionstuffed derivatives can adopt maximal non-isomorphic subgroups as shown by the three branches of the hierarchical tree in Fig. 3. Prototypical  $A_5(BO_4)_3X$  apatite with  $P6_3/m$ symmetry is at the head of the middle branch, and, as discussed below, apatites of lower symmetry ( $P6_3$ ,  $P\overline{3}$ , P6,  $P2_1/m$ ,  $P2_1$ ) are numerous. Indeed, it has been suggested by several workers (e.g. Huang & Sleight, 1993) that some apatites that are reported to conform to  $P6_3/m$  may have lower symmetry, a proposition

that appears to be gaining acceptance as increasing numbers of detailed structure solutions are presented. With respect to the other branches, Aneas *et al.* (1983) and Plaisier *et al.* (1995) have synthesized osmium and rhenium apatites (in which the oxygen coordination of B cations is raised to 5) that have Pcmn and  $P6_3cm$  symmetries.

### 2.3. Hexagonal anion nets

While many crystal structures are regarded as derivatives of near-regular anion packing, this description has not generally been applied to the apatites, although when taken in isolation the anion layers subscribe recognizable, though partially occupied, hexagonal networks. The similarity of the regular anion layers  $\dots b(ab)a \dots$  for idealized apatite and  $\dots b(ab'a'b)a \dots$  for (hexagonal) chlorapatite is clear (Fig. 4). The **b** and **b**' layers are differentiated through the twisting of alternate  $O_3$  triangles about [001].

2.3.1. Model I – idealized chlorapatite type. Idealized chlorapatite is constructed by filling trigonal prismatic and tetrahedral interstices in an ordered fashion, such that the A(1) cations are located in the **b** layers and the A(2) and Bcations occupy the a layers. The atomic coordinates of this derived apatite are given in Table 2, and its structure is illustrated in Fig. 5(a). The dominant feature is formed by the 6hA(2) cations that occupy  $A(2)O_4X_2$  trigonal prism trimers that share common edges along [001]. Filled and empty trigonal prisms alternate along  $\mathbf{c}$ . The A(1) cations form continuous isolated  $A(1)O_6$  trigonal prism columns parallel to **c** (Fig. 4a) – for clarity the three more distant O atoms that cap each of the square faces are not indicated. These columns are linked through the  $BO_4$  tetrahedra, which share edges with the A(2)trigonal prisms and corners with the A(1) trigonal prisms. Apatite is not generally represented in this way; however, a notable exception is the short description and associated figure given in Povarennykh (1972), where the A coordination polyhedra are emphasized and described as distorted trigonal prisms.

The similarity of the atomic coordinates and topology of the idealized structure and chlorapatite ( $P6_3/m$  symmetry) is obvious, as is the deviation from the perfect hexagonal net (Fig. 5b). In chlorapatite, the largest departure from the ideal position is for the A(2) cation, which is displaced from the centre of the trigonal prism such that it moves closer to O(1)/O(2) and increases its anion coordination from six to eight. This modification allows for the incorporation of larger cations into the structure.

The regular trigonal prisms of the A(1) cations are converted into metaprisms through the twisting of oxygen triangles, as described above, to generate a polyhedron that is intermediate between a trigonal prism and an octahedron (O'Keeffe & Hyde, 1996). This twist angle  $\varphi$ , which is the (001) projection of O(1)-A(1)-O(2), is equal to 19.1° (Fig. 6). The reduced volume of the metaprism permits the accommodation of relatively smaller cations.

**2.3.2.** Model II – idealized fluorapatite type. The ideal fluorapatite model retains  $A(1)O_6$  and  $BO_4$  polyhedra iden-

 Table 2

 Crystal data for idealized apatite compared with crystal data for chlorapatite and fluorapatite.

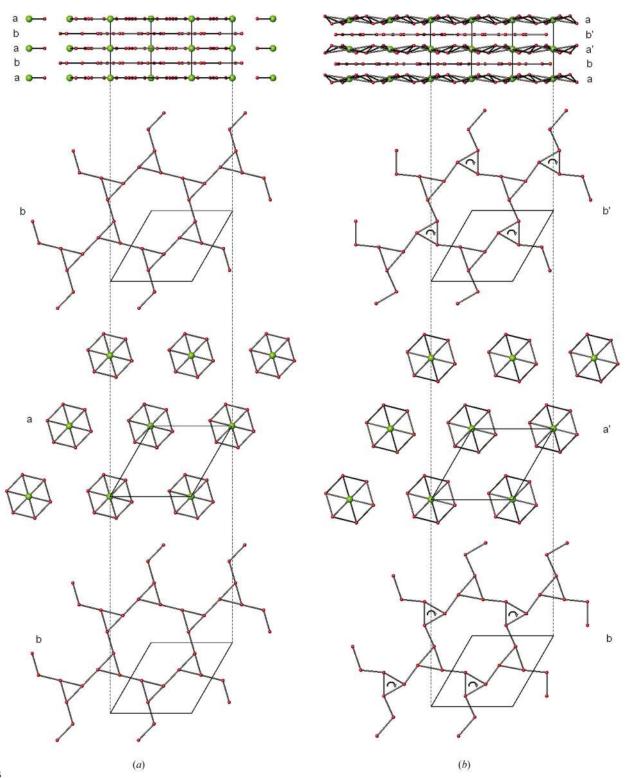
chiora	apauı	te and nu	югараш	e.					
with 2	4(2) t	exagonal rigonal p $c \sim 6.9 \text{ Å}$		lel I			c = 6.850	icks <i>et al</i> . ) Å	, 1932),
A(1) $A(2)$	4 <i>f</i> 6 <i>h</i>	1/3 0.1667	2/3 0.042	0 1/4	Ca(1) Ca(2)	4 <i>f</i> 6 <i>h</i>	1/3 1/4	2/3 0	0.000 1/4
В	6 <i>h</i>	(1/6) 0.4231	(1/24) 0.3846	1/4	P	6 <i>h</i>	0.417	0.361	1/4
O1	6 <i>h</i>	(11/26) 0.3847	(5/13) 0.5385	1/4	O1	6 <i>h</i>	1/3	1/2	1/4
O2	6 <i>h</i>	(5/13) 0.6153 (8/13)	(7/13) 0.4615 (6/13)	1/4	O2	6 <i>h</i>	0.600	0.467	1/4
О3	12 <i>i</i>	0.3077 (4/13)	0.2308 (3/13)	0	О3	12 <i>i</i>	1/3	1/4	0.064
X	2 <i>b</i>	Ò	Ò	0	Cl	2 <i>b</i>	0	0	0
wit		exagonal ) octahed						anan <i>et a</i> = 6.878 (2	
$\overline{A(1)}$	4 <i>f</i>	1/3	2/3	0	Ca(1)	4 <i>f</i>	1/3	2/3	0.0012
A(2)	6 <i>h</i>	0.1923 (5/26)	0.0769 (1/13)	1/4	Ca(2)	6 <i>h</i>	0.2413	0.0071	1/4
В	6 <i>h</i>	0.4231 (11/26)	0.3846 (5/13)	1/4	P	6 <i>h</i>	0.3982	0.3689	1/4
O1	6 <i>h</i>	0.3847 (5/13)	0.5385	1/4	O1	6 <i>h</i>	0.3268	0.4850	1/4
O2	6 <i>h</i>	0.6153 (8/13)	0.4615 (6/13)	1/4	O2	6 <i>h</i>	0.5881	0.4668	1/4
О3	12 <i>i</i>	0.3077 (4/13)	0.2308 (3/13)	0	О3	12 <i>i</i>	0.3415	0.2569	0.0704
X	2 <i>b</i>	0	0	1/4	F	2 <i>a</i>	0	0	1/4
III	with .	exagonal $A(1)$ octa $A(1) \sim 5.4$	hedra,	lel					
A(1)	4f	1/3	2/3	0					
A(2)	6h	0.1667 (1/6)	0.3333 (1/3)	1/4					
В	6h	0.3333 (1/3)	0.3333 (1/3)	1/4					
O1	6 <i>h</i>	0.3333 (1/3)	0.5000 (1/2)	1/4					
O2	6 <i>h</i>	0.6667 (2/3)	0.5000 (1/2)	1/4					
О3	12 <i>i</i>	0.1667 (1/6)	0.1667 (1/6)	0					
X	2b	0	0	0					

tical to those of the chlorapatite model; however, because the X anions are displaced by z/4, the coordination sphere of A(2) changes to octahedral  $A(2){\rm O}_5X$  (Fig. 7a, Table 2). These octahedra, which are corner-connected to each other and to the  $A(1){\rm O}_6$  trigonal prisms, are cation-centred on the  ${\bf b}$  anion nets, and form groupings of three, which are joined at a common X atom at one of their apices. The A(2) octahedra parallel to (001) edge-share with the  $B{\rm O}_4$  tetrahedra. This description of apatite was put forward by Alberius-Henning, Landa-Canovas *et al.* (1999), who regarded the  $A(2){\rm O}_5X$  octahedral arrangement as NaCl-like. For the real fluorapatite structure the A(1) metaprismatic twist angle  $\varphi = 23.3^{\circ}$  (Figs. 6 and 7b).

**2.3.3.** Model III – idealized glaserite type. In models I and II the  $A(1)O_6$  trigonal prismatic coordination was retained,

albeit somewhat elongated along [001]. However, in real chlorand fluorapatite, the twisting converted these polyhedra to metaprisms. The maximum possible twist angle  $\varphi$  is 60°, in which case  $A(1)O_6$  coordination is octahedral (Fig. 6), if collapse is invoked along [001] to maintain edges of equal

length, or elongated octahedral otherwise. The fractional coordinates of the atoms in this arrangement, which can be described as a double h.c.p. structure, are given in Table 2. While no real structure that adopts this arrangement could be identified, layers of this type are found in the glaserite-type



The anion arrangement in (a) idealized chlorapatite and (b) the real chlorapatite structure. The upper portion shows the [110] projection and emphasizes the stacking sequence using the conventional ... $\mathbf{b}(\mathbf{ab})\mathbf{a}$ ... notation. In real chlorapatite, anion displacements lead to a doubling of the [001] crystallographic repeat and hence to the sequence ... $\mathbf{b}(\mathbf{ab}'\mathbf{a}'\mathbf{b})\mathbf{a}$ .... In the lower portion three (001) anion slices are drawn. The anion layers are evidently not fully occupied although their derivation from a triangular network is clear.

structure. Indeed the formal relationship between glaserite [typified by  $P\bar{3}$  K<sub>3</sub>Sc(PO<sub>4</sub>)<sub>2</sub>] and apatite has been discussed by Wondratschek (1963).

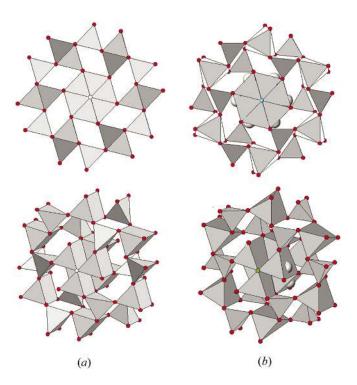


Figure 5 Arrangement of cation-centred polyhedra in (a) model I, which approximates the arrangement in chlorapatite, and (b) the refined structure of chlorapatite. In the upper part the projection along [001] is shown, while  $A(1)O_6$ ,  $A(2)O_4X_2$  and  $BO_4$  connectivity is emphasized in the lower clinographic projections. Note that the statistical occupancy of chlorine is not considered in this representation.

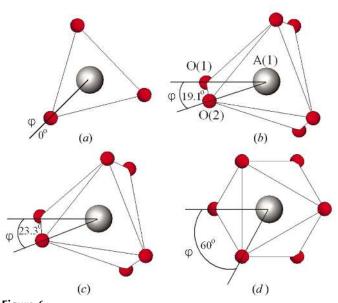


Figure 6 Twist angles of  $A(1)O_6$  polyhedra in (a) models I and II, (b) chlorapatite, (c) fluorapatite, and (d) model III (as found in glaserite).

## 3. Apatites with BO<sub>4</sub> tetrahedra

A compilation of 77 distinct  $A_5(BO_4)_3X$  compounds for which complete crystallographic data exist is presented in Table 3. It is recognized that the occurrence of chemical analogues is more widespread (e.g. Cockbain, 1968). However, for this analysis single-crystal determinations or powder refinements are essential. Similarly, intermediate members of solid-solution series were not included, except where information for the chemical endmembers was unavailable or complex chemistries led to lower symmetries.

While  $P6_3/m$  is dominant (57% of the total), apatites lower in the hierarchical tree are not uncommon, especially P63 (21%) and P3 (9%). Lower symmetries result from an increase in chemical complexity, as a greater number of acceptor sites (seven in  $P6_3/m$  and 18 in  $P2_1/m$ ) are required to accommodate the different bonding requirements of the A and B cations (Table 4). Note that apatites that adopt  $P2_1/b$  do so because statistical occupation of the 4e site by the X anion (Cl or OH) is required - the rest of the structure maintains  $P6_3/m$ . Therefore, monoclinic apatites such as Ca<sub>4.95</sub>(PO<sub>4</sub>)<sub>2.99</sub>Cl<sub>0.92</sub> (Ikoma et al., 1999) are not strictly accommodated through direct derivation from the D88 aristotype. Rather they can be considered as interpenetrating lattices of commensurate structure (that based on  $D8_8$ ) and (in)commensurate structure of resulting X-anion–X-vacancy ordering.

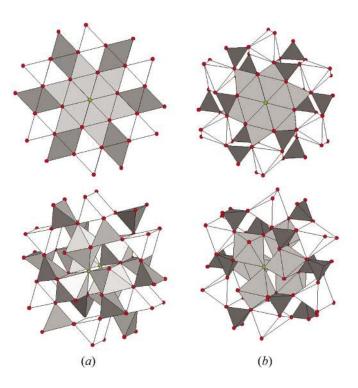


Figure 7 Arrangement of cation-centred polyhedra in (a) model II, which approximates the arrangement in fluorapatite, and (b) the refined structure of fluorapatite. In the upper part the projection along [001] is shown, while  $A(1)O_6$ ,  $A(2)O_5X$  and  $BO_4$  connectivity is emphasized in the lower clinographic projections.

**Table 3**  $[A(1)_2][A(2)_3](BO_4)_3X$  apatites.

Space group/ mineral name	Composition	Unit-cell parameters (Å)	Reference
P6 <sub>3</sub> /m (176)			
103111 (170)	$Na_{6.35}Ca_{3.65}(SO_4)_6F_{1.65}$	a = 9.4364 (21), $c = 6.9186$ (16)	Piotrowski, Kahlenberg & Fischer (2002)
	$Na_{6,39}Ca_{3,61}(SO_4)_6Cl_{1.61}$	a = 9.5423 (1), c = 6.8429 (1)	Piotrowski, Kahlenberg & Fischer (2002)
Carocolite (high)	$Na_3Pb_2(SO_4)_3Cl$	a = 9.810 (20), c = 7.140 (20)	Schneider (1967)
	$Na_3Pb_2(BeF_4)_3F$	a = 9.531 (3), c = 7.028 (2)	Engel & Fischer (1990)
	$Na_2Ca_6Sm_2(PO_4)_6F_2$	a = 9.3895 (3), c = 6.8950 (4)	Toumi et al. (2000)
Fluroapatite	$Ca_5(PO_4)_3F$	a = 9.363 (2), c = 6.878 (2)	Sudarsanan et al. (1972)
Chlorapatite	$Ca_5(PO_4)_3Cl$	a = 9.5902 (6), c = 6.7666 (2)	Kim <i>et al.</i> (2000); Hendricks <i>al.</i> (1932)
	$Ca_5(PO_4)_3Br$	a = 9.761 (1), c = 6.739 (1)	Elliot et al. (1981); Kim et al. (2000)
	Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	a = 9.4302 (5), c = 6.8911 (2)	Kim <i>et al.</i> (2000; Hughes <i>et a</i> (1989)
	$Ca_{15}(PO_4)_9IO\dagger$	a = 9.567, c = 20.758	Alberius-Henning, Lidin & Petrícek (1999)
Tournoourite	$Ca_5(CrO_4)_3OH$	a = 9.683, c = 7.010	Wilhelmi & Jonsson (1965)
Tourneaurite	Ca <sub>5</sub> (AsO <sub>4</sub> ) <sub>3</sub> Cl	a = 10.076 (1), c = 6.807 (1)	Wardojo & Hwu (1996)
Hedyphane	Ca <sub>2</sub> Pb <sub>3</sub> (AsO <sub>4</sub> ) <sub>3</sub> Cl Sr <sub>2</sub> (PO <sub>2</sub> ) <sub>2</sub> Cl	a = 10.140 (3), c = 7.185 (4) a = 9.877 (0), c = 7.189 (0)	Rouse <i>et al.</i> (1984)
	$Sr_5(PO_4)_3Cl$ $Sr_5(PO_4)_3Br$	a = 9.877 (0), c = 7.189 (0) a = 9.964 (0), c = 7.207 (0)	Nötzold <i>et al.</i> (1994) Nötzold & Wulff (1998)
	Sr <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	a = 9.745 (1), c = 7.267 (0) a = 9.745 (1), c = 7.265 (1)	Sudarsanan & Young (1972)
	$(Sr_{4.909}Nd_{0.061})(V_{0.972}O_4)_3F_{0.98}$	a = 10.0081 (1), c = 7.434 (1)	Corker <i>et al.</i> (1995)
	$(S1_4,9091 \times 0_{1,061})( V_{0.972} \cup 4/31_{0.98} $ $Sr_5(VO_4)_3(Cu_{0.896}O_{0.95})$	a = 10.0061 (1), c = 7.434 (1) a = 10.126 (1), c = 7.415 (1)	Carrillo-Cabrera & von Schnering (1999)
	$Cd_{4,92}(PO_4)_3Cl_{0.907}$	a = 9.633 (4), $c = 6.484$ (4)	Sudarsanan <i>et al.</i> (1977); Wilson <i>et al.</i> (1977)
	$Cd_5(PO_4)_3Cl$	a = 9.625 (4), $c = 6.504$ (2)	Ivanov <i>et al.</i> (1976); Sudarasanan <i>et al.</i> (1973)
	$Cd_{4.82}(PO_4)_3Br_{1.52}$	a = 9.733 (1), c = 6.468 (1)	Sudarsanan <i>et al.</i> (1977); Wilson <i>et al.</i> (1977)
	$Cd_5(PO_4)_3OH$	a = 9.335(2), c = 6.664(3)	Hata et al. (1978)
	$Cd_{4.92}(AsO_4)_3Br_{1.52}$	a = 10.100 (1), c = 6.519 (1)	Sudarsanan <i>et al.</i> (1977); Wilson <i>et al.</i> (1977)
	$Cd_{4.64}(VO_4)_3I_{1.39}$	a = 10.307 (1), c = 6.496 (1)	Sudarsanan <i>et al.</i> (1977); Wilson <i>et al.</i> (1977)
	$Cd_{4.86}(VO_4)_3Br_{1.41}$	a = 10.173 (2), $c = 6.532$ (1)	Sudarsanan <i>et al.</i> (1977); Wilson <i>et al.</i> (1977)
	$Ba_5(PO_4)_3F$	a = 10.153 (2), c = 7.733 (1)	Mathew et al. (1979)
	$Ba_5(PO_4)_3Cl$	a = 10.284 (2), c = 7.651 (3)	Hata et al. (1979)
	$Ba_5(AsO_4)_2SO_4S$	a = 10.526 (5), c = 7.737 (1)	Schiff-Francois et al. (1979)
	$Ba_5(MnO_4)_3F$	a = 10.3437, c = 7.8639	Dardenne et al. (1999)
	$Ba_5(MnO_4)_3Cl$	a = 10.469 (1), c = 7.760 (1)	Reinen et al. (1986)
	Pb <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> F	a = 9.760 (8), c = 7.300 (8)	Belokoneva <i>et al.</i> (1982); Kir <i>et al.</i> (2000)
Pyromorphite	Pb <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> Cl	a = 9.998 (1), c = 7.344 (1),	Dai & Hughes (1989); Kim <i>e al.</i> (2000)
	$Pb_5(PO_4)_3Br$ $Pb_5(PO_4)_3OH$	a = 10.0618 (3), $c = 7.3592$ (1) a = 9.866 (3), $c = 7.426$ (2)	Kim <i>et al.</i> (2000) Bruecker <i>et al.</i> (1995); Kim <i>e</i>
	ru <sub>5</sub> (rO <sub>4</sub> ) <sub>3</sub> On	u = 9.800 (3), c = 7.420 (2)	al. (1998) al. (1998)
Vanadinite	$Pb_5(VO_4)_3Cl$	a = 10.317 (3), $c = 7.338$ (3)	Dai & Hughes (1989)
	Pb <sub>9.85</sub> (VO <sub>4</sub> ) <sub>6</sub> I <sub>1.7</sub>	a = 10.442 (5), c = 7.467 (3)	Audubert <i>et al.</i> (1999)
Mimetite	Pb <sub>5</sub> (AsO <sub>4</sub> ) <sub>3</sub> Cl	a = 10.211 (1), c = 7.418 (4)	Calos & Kennard (1990)
	$Pb_5(GeO_4)(VO_4)_2$	a = 10.097 (3), c = 7.396 (2)	Ivanov & Zavodnik (1989); Ivanov (1990)
	$Pb_5(SO_4)(GeO_4)_2$	a = 10.058 (4), $c = 7.416$ (1)	Engel & Deppisch (1988)
	$Pb_5(CrO_4)(GeO_4)_2$	a = 10.105 (3), c = 7.428 (2)	Engel & Deppisch (1988)
	$Pb_3(PO_4)_2$	a = 9.826 (4), $c = 7.357$ (3)	Hata et al. (1980)
	$Pb_4Na(VO_4)_3$	a = 10.059 (1), c = 7.330 (1)	Sirotinkin et al. (1989)
	$Pb_8K_2(PO_4)_6$	a = 9.827 (1), c = 7.304 (1)	Mathew et al. (1980)
	$Mn_5(PO_4)_3Cl_{0.9}(OH)_{0.1}$	a = 9.532(1), c = 6.199(1)	Engel et al. (1975)
	$Nd_4Mn(SiO_4)_3O$	a = 9.499 (1), c = 6.944 (2)	Kluever & Mueller-Busch- baum (1995)
	$NaY_9(SiO_4)_6O_2$	a = 9.332(2), c = 6.759(1)	Gunawardane et al. (1982)
	$La_9Na(GeO_4)_6O_2$	a = 9.883 (2), $c = 7.267$ (3)	Takahashi et al. (1998)
	$La_{10}(Si_{3.96}B_{1.98}O_4)_6O_2$	a = 9.5587 (2), c = 7.2171 (2)	Mazza et al. (2000)
	$La_3Nd_{11}(SiO_4)_9O_3\dagger$	a = 9.638 (2), $c = 21.350$ (8)	Malinovskii et al. (1990)
Britholite	$(Ce_{0.4}Ca_{0.35}Sr_{0.25})_2(Ce_{0.86}Ca_{0.14})_3(SiO_4)_3(O_{0.5}F_{0.38})$	a = 9.638 (1), c = 7.081 (1)	Genkina et al. (1991)

Table 3 (continued)

Space group/ mineral name	Composition	Unit-cell parameters (Å)	Reference
P6 <sub>3</sub> (173)			
103 (173)	$K_3Sn_2(SO_4)_3Cl$	a = 10.230 (20), c = 7.560 (20)	Howie <i>et al.</i> (1973); Donaldson & Grimes (1984)
	$K_3Sn_2(SO_4)_3Br$	a = 10.280 (20), c = 7.570 (20)	Howie <i>et al.</i> (1973); Donaldson & Grimes (1984)
	$KNd_9(SiO_4)_6O_2$	a = 9.576 (2), $c = 7.009$ (2)	Pushcharovskii et al. (1978)
	$Ca_{10}(PO_4)_6S$	a = 9.455 (0), c = 8.840 (0)	Sutich et al. (1986)
	$Ca_4Bi(VO_4)_3O$	a = 9.819(2), c = 7.033(2)	Huang & Sleight (1993)
	$Sr_{7.3}Ca_{2.7}(PO_4)_6F_2$	a = 9.865 (8), $c = 7.115$ (3)	Pushcharovskii <i>et al.</i> (1987); Klevtsova (1964)
	Sr <sub>5</sub> (CrO <sub>4</sub> ) <sub>3</sub> Cl	a = 10.125, c = 7.328	Mueller-Buschbaum & Sander (1978)
	$Cd_5(PO_4)_3OH^{\dagger}$	a = 16.199(0), c = 6.648(0)	Hata & Marumo (1983)
	Ba <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	a = 10.190 (1), c = 7.721 (2)	Bondareva & Malinovskii (1986)
	Ba <sub>5</sub> (CrO <sub>4</sub> ) <sub>3</sub> OH	a = 10.428, c = 7.890	Mattausch & Mueller-Busch- baum (1973)
	$Ba_{5}[(Ge,C)(O,OH)_{4})]_{3}[(CO_{3}(OH)]_{1.5}(OH)$	a = 10.207 (3), c = 7.734 (2)	Malinovskii <i>et al.</i> (1975)
	$La_6Ca_{3.5}(SiO_4)_6(H_2O)F$	a = 9.664(3), c = 7.090(1)	Kalsbeek <i>et al.</i> (1990)
	$Na_{0.97}Ca_{1.40}La_{2.20}Ce_{3.69}Pr_{0.32}Nd_{0.80}[(Si_{5.69}P_{0.31})_4]_6(OH,F)$	a = 9.664 (3), c = 7.090 (1)	Kalsbeek <i>et al.</i> (1990)
	$Sm_{10}(SiO_4)_6N_2$	a = 9.517 (6), c = 6.981 (4)	Gaude <i>et al.</i> (1975)
	$(Sm_8Cr_2)(SiO_4)_6N_2$	a = 9.469 (5), c = 6.890 (4)	Maunaye <i>et al.</i> (1976)
$P\bar{3}$ (147)		0.456 (4) 6.005 (4)	T I. (1000)
	$Ca_{9.93}(P_{5.84}B_{0.16}O_4)_6(B_{0.67}O_{1.79})$	a = 9.456 (1), c = 6.905 (1)	Ito et al. (1988)
	$Sr_{9.402}Na_{0.209}(PO_4)_6B_{0.996}$	a = 9.734 (4), c = 7.279 (2)	Calvo <i>et al.</i> (1982)
D. L'.	$Ba_4Nd_3Na_3(PO_4)_6F_2$	a = 9.786 (2), c = 7.281 (1)	Mathew <i>et al.</i> (1979)
Belovite	$Na_{0.5}Ca_{0.3}Ce_{1.00}Sr_{2.95}(PO_4)_3OH$	a = 9.692 (3), c = 7.201 (1)	Nadezhina et al. (1987)
	$Na_{0.981}La_{0.999}Sr_{2.754}Ba_{0.12}Ca_{0.06}(PO_4)_3OH$	a = 9.664(0), c = 7.182(0)	Kabalov et al. (1997)
	$Na_2Ca_2Sr_6(PO_4)_6(OH)_2$	a = 9.620, c = 7.120	Klevtsova & Borisov (1964)
P6 (174)	N G (00 ) 000		T (2002)
Cesanite	$Na_{6.9}Ca_{3.1}(SO_4)_6OH_{1.1}$	a = 9.4434 (13), c = 6.8855 (14)	Piotrowski et al. (2002)
	$\operatorname{Ca}_{10}(\operatorname{PO}_4)_6\operatorname{O}$	a = 9.432, c = 6.881	Alberius-Henning, Landa-
	$Ba_3LaNa(PO_4)_3F$	a = 9.939(0), c = 7.442(1)	Canovas <i>et al.</i> (1999) Mathew <i>et al.</i> (1979)
	$Da_3LaNa(FO_4)_3F$	u = 9.939(0), c = 7.442(1)	Mathew <i>et al.</i> (1979)
P2 <sub>1</sub> /m (11)		0.476 (2) 1 0.500 (2)	C 1 (1000) II 1 0
Hydroxyellestad-	$Ca_{10}(SiO_4)_3(SO_4)_3[F_{0.16}Cl_{0.48}(OH)_{1.36}]$	a = 9.476 (2), b = 9.508 (2),	Sudarsanan (1980); Hughes &
tite Fermorite	$(Ca_{8.40}Sr_{1.61})(AsO_4)_{2.58}(PO_4)_{3.42}F_{0.69}(OH)_{1.31}$	$c = 6.919 (1), \gamma = 119.5^{\circ}$ a = 9.594 (2), b = 9.597 (2), $c = 6.975 (2), \gamma = 120.0 (0)^{\circ}$	Drexler (1991) Hughes & Drexler (1991)
Carocolite (low)	$Na_3Pb_2(SO_4)_3CI^{\dagger}$	a = 19.620, b = 9.810, c = 7.140,	Schneider (1969)
caroconte (1011)	1.43. 02(0.04/3.01)	$\gamma = 120^{\circ}$	Semielder (1767)
$P2_1/b$ (14)			
	Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	a = 9.421 (1), b = 18.843 (2), $c = 6.881 (1), \gamma = 120.0 (1)^{\circ}$	Elliot et al. (1973)
	Ca <sub>4.95</sub> (PO <sub>4</sub> ) <sub>2.99</sub> Cl <sub>0.92</sub>	a = 9.426 (3), $b = 18.856$ (5), $c = 6.887$ (1), $\gamma = 119.97$ (1)°	Ikoma et al. (1999)
	$Ca_{9.97}(PO_4)_3Cl_{1.94}\dagger$	a = 9.632 (7), $b = 19.226$ (20), $c = 6.776$ (5), $\gamma = 119.9$ (1)°	Bauer & Klee (1993); Mackie <i>et al.</i> (1972)
Clinomimetite	$Pb_5(AsO_4)_3Cl\dagger$	a = 10.189 (3), $b = 20.372$ (6), $c = 7.456$ (6), $\gamma = 119.0$ (0)°	Dai <i>et al.</i> (1991)
P2 <sub>1</sub> (4)		0.700(0) 1(0)	
Ellestadite (low)	$Ca_{10}(Si_{3.14}S_{2.94}C_{0.08}P_{0.02})O_{24}[(O\ H)_{1.12}Cl_{0.316}F_{0.05}]$	a = 9.526 (2), b = 9.506 (4), $c = 6.922 (1), \gamma = 120.0 (0)^{\circ}$	Organova et al. (1994)
Britholite	$(Na_{1.46}La_{8.55})(SiO_4)_6(F_{0.9}O_{0.11})$	a = 9.678 (1), b = 9.682 (3), $c = 7.1363 (1), y = 120.0 (0)^{\circ}$	Hughes et al. (1992)
Britholite (Ce)	$(Ca_{2.15}Ce_{2.85})(SiO_4)_3[F_{0.5}(OH)_{0.5}]$	a = 9.580 (5), b = 9.590 (4), $c = 6.980 (3), \gamma = 120.1 (0)^{\circ}$	Noe et al. (1993)

<sup>†</sup> Superstructure.

## 3.1. A(1) metaprism twist angle

The metaprism twist angle  $\varphi$  is, not surprisingly, sensitive to apatite composition – the smallest  $(\varphi = 5.2^{\circ})$  is found for

 $\text{Ca}_2\text{Pb}_3(\text{AsO}_4)_3\text{Cl}$ , while the largest  $(\varphi = 26.7^\circ)$  has been observed for  $\text{Pb}_5(\text{PO}_4)_3\text{OH}$  (Fig. 8). For  $P6_3/m$  apatites  $\varphi$  can be calculated for any individual metaprism from the fractional coordinates of A(1), O(1) and O(2) via the general expression

 Table 4

 Atom acceptor sites in apatites that adopt different space groups.

	Space group	Space group								
	$P6_3/m (176)$		P6 <sub>3</sub> (173)		P3 (147)		P6 (174)		P2 <sub>1</sub> /m (11)	
Site types	Wyckoff number	Number of site types	Wyckoff number	Number of site types	Wyckoff number	Number of site types	Wyckoff number	Number of site types	Wyckoff number	Number of site types
Large A cations	6h	1	6 <i>c</i>	1	6g	1	3k, 3j	2	$2a, 2e \times 2$	3
Small A cation	4f	1	$2b \times 2$	2	$2d \times 2$	2	2i, 2h	2	4f	1
B cations	6h	1	6 <i>c</i>	1	6g	1	3k, 3j	2	$2e \times 3$	3
Total number of cation acceptor sites		3		4		4		6		7
Oxygen anions	$6h \times 2, 12i$	3	$6c \times 4$	4	6 <i>g</i> × 4	4	$3k \times 2,$ $3j \times 2,$ $6l \times 2$	6	$ 2e \times 6, \\ 4f \times 3 $	9
X anions	2a or 2b or 4e	1	2 <i>a</i>	1	1a, 1b	2	1a, 1b or 2g	1/2	2a or 2e	2
Total number of anion acceptor sites		4		5		6		7/8		11
Examples of cation occupancy	[Ca <sub>6</sub> ][Ca <sub>4</sub> ][P	6]	$[K_6][K_2Sr$	$[S_6]$	[Ba <sub>4</sub> Nd <sub>1</sub> Na	a <sub>1</sub> ][Nd <sub>2</sub> Na <sub>2</sub> ][P <sub>6</sub> ]	[Ba <sub>5</sub> La <sub>1</sub> ][La <sub>1</sub> ]	Na <sub>1.3</sub> Ba <sub>1.7</sub> ][P <sub>6</sub> ]	[Ca <sub>6</sub> ][Ca <sub>4</sub> ][	Si <sub>3</sub> S <sub>3</sub> ]
[ $A_6$ ][ $A_4$ ][ $B_6$ ]- [ $O_{24}$ ][ $X_2$ ]	Ca <sub>10</sub> (PO <sub>4</sub> ) <sub>6</sub> F	2	K <sub>6</sub> Sn <sub>4</sub> (SC	0 <sub>4</sub> ) <sub>6</sub> Cl <sub>2</sub>	Ba <sub>4</sub> Nd <sub>3</sub> Na	<sub>3</sub> (PO <sub>4</sub> ) <sub>6</sub> F <sub>2</sub>	Ba <sub>6.7</sub> La <sub>2</sub> Na <sub>1.3</sub>	$(PO_4)_6F_2$	Ca <sub>10</sub> (SiO <sub>4</sub> ) <sub>3</sub> [F <sub>0.16</sub> Cl <sub>0.4</sub>	(SO <sub>4</sub> ) <sub>3</sub> <sub>18</sub> (OH) <sub>1.36</sub> ]

$$\cos \varphi = \left[ (G_3 - G_1)^2 + (H_3 - H_1)^2 + (G_1 - G_2)^2 + (H_1 - H_2)^2 - (G_2 - G_3)^2 - (H_2 - H_3)^2 \right]$$

$$/ \left( 2 \left\{ \left[ (G_3 - G_1)^2 + (H_3 - H_1)^2 \right] \right.$$

$$\times \left. \left[ (G_1 - G_2)^2 + (H_1 - H_2)^2 \right] \right\}^{1/2} \right),$$

where

$$G_1 = 0.866x_{A1}, H_1 = y_{A1} - 0.5x_{A1},$$

$$G_2 = 0.866x_{O1}, H_2 = y_{O1} - 0.5x_{O1},$$

$$G_3 = 0.866x_{O2}, H_3 = y_{O2} - 0.5x_{O2}.$$

When  $\varphi$  is calculated for a range of apatites, it is observed to vary inversely with average crystal radii and unit-cell volume (Table 5). In fact, for those  $P6_3/m$  apatites with the same A cation (e.g. calcium or cadmium), the relationship is very nearly linear (Fig. 9), regardless of whether substitution is made on the B or X sites. In other words, for apatites with smaller average crystal radii, the structure must collapse further to satisfy shorter bond-length requirements, and this fact is reflected in larger  $\varphi$  values, which lead to shorter A(1)— O distances. If  $A(1)O_6$  twisting is considered to be the pivotal structural feature, then the collapse towards octahedral bonding will also accommodate the shortening of the cell edges and a general reduction of A(2)—O—X bond lengths. A detailed consideration of the geometry of metaprism twisting is given in Dong & White (2003).] In contrast the BO<sub>4</sub> tetrahedra will essentially remain invariant. Therefore, the metaprism twist angle  $\varphi$  is a useful qualitative – and in some cases quantitative – predictor of apatite distortion from ideal h.c.p. packing. Indeed, for all the apatites in Table 5 only two compositions were exceptional. Ba<sub>5</sub>(MnO<sub>4</sub>)<sub>3</sub>Cl (Reinen *et al.*, 1986) and Pb<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>OH (Bruecker *et al.*, 1995) have  $\varphi$  values of 22.3° and 26.7°, respectively. These appear anomalously high compared with the average crystal radii and bond angle (Table 5). Possible explanations may be the adoption of lower symmetry, the distortion of the MnO<sub>4</sub> tetrahedra or *A*-site non-stoichiometry for the Pb compound.

## 3.2. X-site non-stoichiometry

It has long been understood that larger X anions (Cl, Br, I) may be displaced away from their 2a and 2b special positions, especially if the A(2) sites are occupied by smaller cations, and that a portion of the sites may be vacant. Substantial quantitative treatments of the phenomena have been undertaken by several workers (e.g. Kim et al., 2000; Hashimoto & Matsumoto, 1998; Hata et al., 1979; Mackie et al., 1972). While the location of the anions over available sites is generally assumed to be statistical, it has now been demonstrated that cadmium phosphate bromate and cadmium vanadate iodate exhibit incommensurate ordering of X anions and vacancies (Christy et al., 2001). For lead-rich apatites the X site may be completely vacant, as in Pb3(PO4)2, Pb4Na(VO4)3 and Pb<sub>8</sub>K<sub>2</sub>(PO<sub>4</sub>)<sub>6</sub>. It is reasonable to assume that the stereochemically active lone pair of Pb2+, which occupies a volume close to that of oxygen, allows such compounds to be stable (Mathew et al., 1980). Stoichiometrically similar non-leadbearing compounds [e.g. Cd<sub>5</sub>(PO<sub>4</sub>)<sub>2</sub>GeO<sub>4</sub>], which do not have the benefit of lone-pair stabilization, adopt the carnotite

**Table 5** Twist angle  $(\varphi)$ , average crystal radius per unit-cell content and unit-cell volume.

-	Metaprism angle	Average crystal	
Composition	(°)	radius (Å)†	Volume (Å <sup>3</sup> )
$Ca_{10}(PO_4)_6F_2$	23.3	1.143	522.6
$Ca_{10}(PO_4)_6OH_2$	23.2	1.146	530.3
$Ca_{10}(PO_4)_6Cl_2$	19.1	1.166	537.6
$Ca_{10}(CrO_4)_6OH_2$	17.8	1.171	569.2
$Ca_{10}(PO_4)_6Br_2$	16.3	1.173	556.1
$Ca_{10}(AsO_4)_6Cl_2$	13.0	1.189	598.5
$Ca_4Pb_6(AsO_4)_6Cl_2$	5.2	1.214	639.8
$Cd_{10}(PO_4)_6OH_2$	25.8	1.139	502.9
$Cd_{10}(PO_4)_6Cl_2$	19.5	1.158	521.1
$Cd_{10}(PO_4)_6Br_2$	16.0	1.165	530.6
$Cd_{10}(AsO_4)_6Br_2$	11.6	1.189	575.9
$Cd_{10}(VO_4)_6Br_2$	8.8	1.192	585.4
$Cd_{10}(VO_4)_6I_2$	8.4	1.203	597.6
$Sr_{10}(PO_4)_6OH_2$	23.0	1.183	597.5
$Sr_{10}(PO_4)_6Cl_2$	21.1	1.203	606.6
$Sr_{10}(PO_4)_6Br_2$	19.6	1.210	619.7
$Pb_{10}(PO_4)_6F_2$	23.5	1.185	602.2
$Pb_{10}(PO_4)_6OH_2$	26.7	1.189	603.2
$Pb_{10}(PO_4)_6Cl_2$	17.6	1.208	635.8
$Pb_{10}(VO_4)_6Cl_2$	17.5	1.235	676.5
$Pb_{10}(VO_4)_6I_2$	16.7	1.253	702.4
$Ba_{10}(PO_4)_6F_2$	22.5	1.219	690.3
$Ba_{10}(PO_4)_6Cl_2$	21.0	1.242	700.8
$Ba_{10}(MnO_4)_6Cl_2$	22.3	1.254	736.6
$Ba_{10}(AsO_4)_4(SO_4)_2S$	16.2	1.259	742.4

<sup>†</sup> Calculated using effective ionic radii of Shannon (1976).

structure type (Engel & Fischer, 1985). It is possible that trivalent bismuth-rich apatites may also exploit the space-filling properties of the lone pairs (Huang & Sleight, 1993; Buvaneswari & Varadaraju, 2000); however, this is yet to be demonstrated. The calculations of Le Bellac *et al.* (1995) have proved useful in locating electron lone pairs in complex oxides, such as Pb<sub>3</sub>P<sub>2</sub>O<sub>8</sub> and Bi<sub>2</sub>Sr<sub>2</sub>CaCu<sub>2</sub>O<sub>8</sub>, and, if applied to plumbous apatites, would underpin a quantitative discussion on this point.

It has been demonstrated that metals may also be incorporated into the 2a/2b sites. Carillo-Cabrera & von Schnering

(1999) successfully synthesized  $Sr_5(VO_4)_3(Cu_{0.896}O_{0.95})$ , in which (Cu—O) strings filled the X positions. While presently the only example of this type, it is not inconceivable that other linearly bonded cation–oxygen strings (e.g.  $Pd^{1+}$ —O,  $Ag^{1+}$ —O,  $Hg^{2+}$ —O) might also be prepared, provided larger A (e.g.  $Ba^{2+}$ ) and B (e.g.  $Mn^{5+}$ ) ions are available to sufficiently dilate the unit cell.

## 3.3. Polysomatism

All  $[A(1)_2][A(2)_3](BO_4)_3X$  apatites of whatever symmetry regularly fill every other BO<sub>4</sub> tetrahedral interstice in cornerconnected strings along [001] (Fig. 10). Alternative filling schemes are of course possible, such that the same overall occupancy is maintained while producing crystallographically distinct polysomates. Alternate filling has been observed in two minerals - nasonite and ganomalite (Engel, 1972). Nasonite, which has the ideal formula Pb<sub>6</sub>Ca<sub>4</sub>(Si<sub>2</sub>O<sub>7</sub>)<sub>3</sub>Cl<sub>2</sub>, was determined by Giuseppetti et al. (1971) to have P6<sub>3</sub>/m symmetry with consecutive sites filled as Si<sub>2</sub>O<sub>7</sub> units. Brès et al. (1987) used high-resolution electron microscopy to demonstrate that the situation is more complex, as on the unit-cell scale  $P6_3/m$  symmetry is broken. For the intermediate mineral ganomalite (Carlson et al., 1997) Si<sub>2</sub>O<sub>7</sub> units alternate with SiO<sub>4</sub> tetrahedra to give an ideal formula Pb<sub>6</sub>Ca<sub>3,33</sub>Mn<sub>0,67</sub>- $(SiO_4)_2(Si_2O_7)_2$  and a structure that conforms to  $P\bar{6}$  symmetry. In this compound Pb cations (IR 1.29 Å) are distributed into the larger 6l, 3k sites as expected; the Ca (1.12 Å) occupy the 2i, 2h, 1c metaprisms in an ordered way with a height along (001) of 3.5 Å, while Ca/Mn (average IR 1.04 Å) on 1e fill an untwisted prism of height 3.07 Å (Fig. 11). Intergrowth between apatite layers that have different heights and twist angles is yet another mechanism for accommodating cations of substantially different ionic radii. A more rigorous determination of natural nasonite will probably reveal similar detail. In addition, the synthetic analogue of ganomalite Pb<sub>5</sub>(GeO<sub>4</sub>)(Ge<sub>2</sub>O<sub>7</sub>) has been studied extensively because of its ferroelectric properties (Newnham et al., 1973; Iwata, 1977). Stemmermann (1992) synthesized the longer-period  $Pb_{40}(Si_2O_7)_6(Si_4O_{13})_3$  with c = 28.7 Å. All the members so far reported for the apatite-nasonite family are lead-rich with the role of lone-pair stabilization yet to be fully investigated. Note

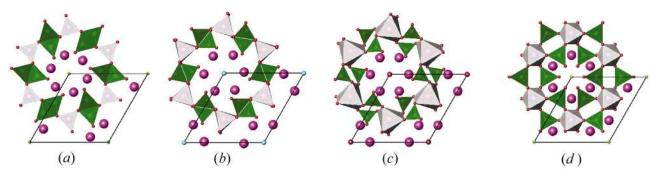


Figure 8
The preferred topological representation of four apatites; both the  $A(1)O_6$  trigonal prisms/metaprisms and the  $BO_4$  tetrahedra are emphasized. The idealized structural endmembers are emphasized in (a) model I with  $\varphi = 0^\circ$  and (d) model III with  $\varphi = 60^\circ$ . The intermediate structures shown are for (b) hedyphane  $Ca_4Pb_6(AsO_4)_6Cl_2$  with  $\varphi = 5.2^\circ$  and (c)  $Pb_{10}(PO_4)_6(OH)_2$  with  $\varphi = 26.7^\circ$ .

that stoichiometric adjustments are required in order to generate these tetrahedrally ordered apatite superstructures, and therefore these compounds are described as part of a polysomatic, rather than polytypic, series (Table 6).

## 4. Apatites with BO<sub>5</sub> and BO<sub>3</sub> polyhedra

The description of apatite as an anion-stuffed alloy proves especially compelling when the scope of the apatite family is broadened to include the less common  $[A(1)_2][A(2)_3](BO_5)_3X$  and  $[A(1)_2][A(2)_3](BO_3)_3X$  compounds (Table 7). In these cases a greater or lesser number of  $M_4$ ,  $M_3$  and  $M_2$  metal interstices are filled with oxygen while the distinctive metal arrangement of the  $D8_8$  aristotype is maintained.

Finnemanite Pb<sub>5</sub>(AsO<sub>3</sub>)<sub>4</sub>Cl (Effenberger & Pertlik, 1977) is a reduced form of mimetite Pb<sub>5</sub>(AsO<sub>4</sub>)<sub>4</sub>Cl (Dai et al., 1991). Both compounds adopt  $P6_3/m$  symmetry with the former missing one 6h oxygen to compensate for the substitution of  $As^{5+} \leftrightarrow As^{3+}$ . In finnemanite the  $BO_4$  tetrahedron is completely replaced by BO<sub>3</sub> with the arsenic lying above the triangular oxygen plane (Fig. 12a). To date, this is the only reported example of a completely  $[A(1)_2][A(2)_3](BO_3)_3X$  apatite. However, partial substitutions of BO<sub>3</sub> for BO<sub>4</sub> are known. In biologically important carbonated hydroxyapatite, recent studies confirm that partial replacement of  $PO_4^{3-} \leftrightarrow CO_3^{3-}$ , which is similar to  $AsO_4^{3-} \leftrightarrow AsO_3^{3-}$ , occur. However, ordering is incomplete, as either of the two possible O3 triangular faces that lie parallel to [001] of the BO<sub>4</sub> tetrahedron can be used to accommodate the carbonate unit (Ivanova et al., 2001). It must also be feasible to introduce boron in a similar way. et al. (1988) have reported the existence

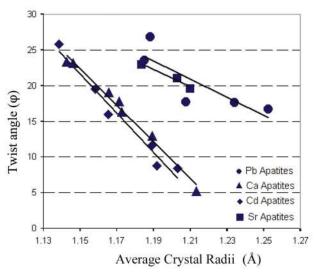


Figure 9 Relationship between twist angle  $(\varphi)$  and average crystal radius for several  $[A(1)_2][A(2)_3](BO_4)_3X$ ,  $P6_3/m$ , apatites in which the A cation is fixed and only B and X vary. For the Ca- and Cd-apatites there is very nearly a linear relationship. This also holds true for Sr-apatites although only three examples were included. While the general trend is also true for the Pb-apatites, the relationship is not so direct.

 $[Ca_{9.93}(P_{5.84}B_{0.16})_4]_6(B_{0.067}O_{1.79})$ ; however, whether a preference is shown by BO<sub>3</sub> to occupy particular face(s) of the O<sub>4</sub> tetrahedron has not been resolved.

A number of ruthenates and osmates have been described (Schriewer & Jeitschko, 1993) whose structures are also derived from the  $D8_8$  alloy (O'Keeffe & Hyde, 1985). In this case, four of the five O atoms of the  $BO_5$  square pyramid are in  $OM_4$  tetrahedra, while the last is in linear  $OM_2$  coordination (Fig. 12c). In  $[A(1)_2][A(2)_3](BO_5)_3X$  compounds the  $BO_5$  pyramids have a 'sense of direction' in that their apical O atoms can lie up or down along [001], and indeed examples of both types have been discovered (Schriewer & Jeitschko, 1993; Plaisier *et al.*, 1995). In  $Ba_5(ReO_5)_4Cl$ , with  $P6_3cm$  symmetry, all of the  $ReO_5$  pyramids' apices are unidirectional, whereas in Pnma  $Sr_5(ReO_5)_4Cl$  one-third of the  $ReO_5$  strings adopt antiparallel orientations (Fig. 13).

Finally, an interesting example of an apatite-related structure is  $Sr_5(BO_3)_3Br$  (Alekel & Keszler, 1992). It can be regarded as a condensed apatite phase, even though the stoichiometry is outwardly similar to that of finnemanite. In all other structures that we discuss, the  $AO_6$  metaprisms are connected exclusively through  $BO_3/BO_4/BO_5$  polyhedra, whereas in  $Sr_5(BO_3)_3Br$  some  $SrO_6$  prism columns share an edge with their neighbour – a comparison of Figs. 7(a) and 14 makes the relationship clear. Following the proposition of Moore & Araki (1977) it is possible to isolate  $Sr_4(BO_3)_{12}$  as the stable unit of this structure. This unit is analogous to  $Ca_4(PO_4)_{12}$  columns, which through rotational operations can generate significant segments of the structures of hydroxyapatite, octacalcium phosphate and samuelsonite. However,

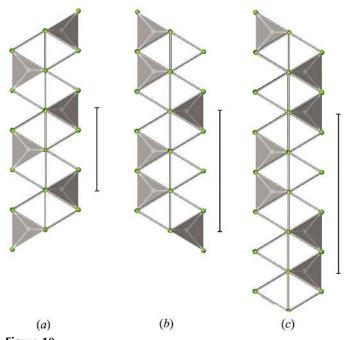


Figure 10 Tetrahedral strings and c-axis repetition in (a) apatite,  $c_{ap}$ , (b) ganomalite,  $\sim 1.33c_{ap}$ , and (c) nasonite,  $\sim 2c_{ap}$ .

<sup>&</sup>lt;sup>2</sup> Compositionally similar Mg<sub>5</sub>(BO<sub>3</sub>)<sub>3</sub>F is unrelated to finnemanite and is a leucophoenicite-type mineral (White & Hyde, 1983).

Table 6 Polysomatic apatites.

Mineral name Space group		Composition	Unit-cell parameters (Å)	Reference	
	$P\bar{6}$	$Pb_5(GeO_4)(Ge_2O_7)$	a = 10.260, c = 10.696	Iwata (1977)	
Ganomalite	$P\bar{6}$	$Pb_9Ca_5Mn(Si_2O_7)_3(SiO_4)_3$	a = 9.850 (50), c = 10.130 (50)	Carlson et al. (1997)	
	$P\overline{6}$	$Pb_9Ca_6(Si_2O_7)_3(SiO_4)_3$	a = 9.875, c = 10.176	Stemmermann (1992)	
	$P\bar{6}$	$Pb_{12}Ca_3(Si_2O_7)_3(SiO_4)_3$	a = 9.880, c = 10.210	Stemmermann (1992)	
	$P\bar{6}$ (?)	$Pb_3Ca_2(Si_2O_7)(SiO_4)$	a = 9.879(1), c = 10.178(1)	Engel (1972)	
	$P\bar{6}$ (?)	Pb <sub>3</sub> BiNa(Si <sub>2</sub> O <sub>7</sub> )(SiO <sub>4</sub> )	a = 9.876 (1), $c = 10.175$ (1)	Engel (1972)	
	$P\bar{6}$ (?)	$Pb_3Cd_2(Si_2O_7)(SiO_4)$	a = 9.810 (4), $c = 10.124$ (4)	Engel (1972)	
	P-6 (?)	$Pb_3Ca_2(Ge_2O_7)(GeO_4)$	a = 10.104 (1), c = 10.379 (1)	Engel (1972)	
	$P\overline{\underline{6}}$ (?)	$Pb_3BiNa(Ge_2O_7)(GeO_4)$	a = 10.084(1), c = 10.398(1)	Engel (1972)	
	$P\bar{6}$	Pb <sub>6</sub> Ca <sub>4</sub> (Si <sub>2</sub> O <sub>7</sub> ) <sub>3</sub> Cl <sub>2</sub>	a = 10.074, c = 13.234	Stemmermann (1992)	
Nasonite	$P\bar{6}$ (?)	$Pb_9Ca_4(Si_2O_7)_3$	a = 10.080, c = 13.270	Giuseppetti et al. (1971)	
	Monoclinic (?)	$Pb_{40}(Si_2O_7)_6(Si_4O_{13})_3$	$a=17.075,\ b=9.844,\ c=26.678,\ \beta=90.09^\circ$	Stemmermann (1992)	

as yet the range of available compounds is insufficient to test this hypothesis.

#### 5. Discussion and conclusions

summarizing the crystallographic data for the materials  $[A(1)_2][A(2)_3](BO_4)_3X$ ,  $[A(1)_2][A(2)_3](BO_5)_3X$  and  $[A(1)_2][A(2)_3](BO_3)_3X$ , their description as 'apatite' or 'apatite-related' can be replaced by a formal description as anion-stuffed hettotypes of the D8<sub>8</sub> aristotype. From this perspective, all are members of the apatite structural family, in the same way that even rather complex perovskites (such as giant magnetoresistance CaCu<sub>3</sub>Mn<sub>4</sub>O<sub>12</sub> and superconducting YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub>) possess similar genealogy. This proposition has been most fully articulated by O'Keeffe & Hyde (1985) but has also been recognized by Wondratschek et al. (1964) and more recently Schriewer & Jeitschko (1993) and Vegas & Jansen (2002). When formalized, this approach leads to a hierarchical tree of possible symmetries, which is increasingly pertinent as the structures of more apatites become available. While  $P6_3/m$  is the most common symmetry of  $[A(1)_2][A(2)_3](BO_4)_3X$  apatites, it is by no means dominant. With increasing chemical complexity, the lower symmetries  $P6_3$ ,  $P\overline{5}$ ,  $P\overline{6}$ ,  $P2_1/m$  and  $P2_1$  are required in order to accommodate cation ordering, and it may be expected that additional lower-symmetry varieties will be recognized. Huang & Sleight (1993) have noted that many  $P6_3/m$  apatites may actually have lower symmetry, and this prescience is borne out by high-precision redeterminations. The most recent of these is the reinvestigation of cesanite Na<sub>6.9</sub>Ca<sub>3.1</sub>(SO<sub>4</sub>)<sub>6</sub>(OH)<sub>1.1</sub> by Piotrowski, Kahlenberg, Fischer et al. (2002), who through the careful analysis of systematic absences were able to allocate P6 as the correct symmetry.

The middle branch of the structural tree, the branch that leads from  $P6_3/m$ , is evidentially the most heavily populated, with only a few representatives reported for the Cmcm and

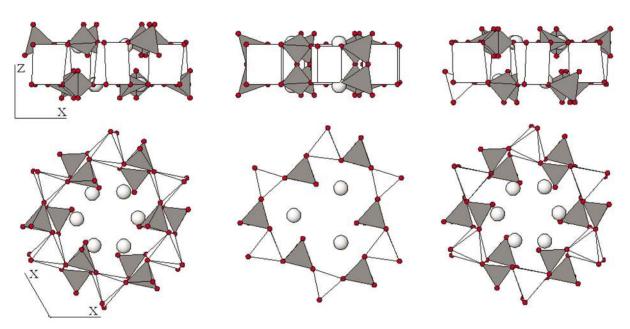


Figure 11 The three apatite layers in ganomalite  $Pb_6Ca_{3.33}Mn_{0.67}(SiO_4)_2(Si_2O_7)_2$ . The figure demonstrates the intergrowth of layers with different twist angles  $(\varphi)$ . Layers 1 and 3 contain trigonal prisms whose triangular faces are of different size and  $\varphi = 17.2^{\circ}$ . Layer 2, which contains the Mn atom, has  $\varphi = 0^{\circ}$ . In the upper portion of the drawing the connectivity of the SiO<sub>4</sub> tetrahedra are emphasized; the Si<sub>2</sub>O<sub>7</sub> unit occurs in the middle layer.

**Table 7**  $[A(1)_2][A(2)_3](BO_5)_3X$  and  $[A(1)_2][A(2)_3](BO_3)_3X$  apatites.

Mineral name	Space group	Composition	Unit-cell parameters (Å)	Reference
Finnemanite	$P6_{3}/m$	Pb <sub>5</sub> (AsO <sub>3</sub> ) <sub>3</sub> Cl	a = 10.322 (7), $c = 7.054$ (6)	Effenberger & Pertlik (1977)
	$C222_{1}$	$Sr_5(BO_3)_3Br$	<i>a</i> = 10.002 (2), <i>b</i> = 14.197 (2), <i>c</i> = 7.458 (1)	Alekel & Keszler (1992)
	P6 <sub>3</sub> cm P6 <sub>3</sub> cm P6 <sub>3</sub> cm	$\begin{array}{l} Ba_5(ReO_5)_3NO_4 \\ Ba_5(ReO_5)_3Cl \\ Ba_5(OsO_5)_3Cl \end{array}$	a = 11.054 (5), c = 7.718 (4) a = 10.926 (1), c = 7.782 (1) a = 10.928 (2), c = 7.824 (5)	Aneas et al. (1983) Schriewer & Jeitschko (1993); Besse et al. (1979) Plaisier et al. (1995)
	Pnma	$Sr_5(ReO_5)_3Cl$	a = 7.438 (1), $b = 18.434$ (2), $c = 10.563$ (2)	Schriewer & Jeitschko (1993)

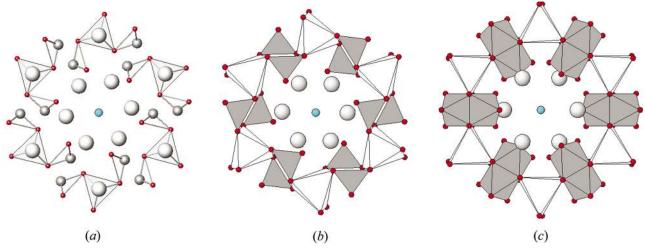
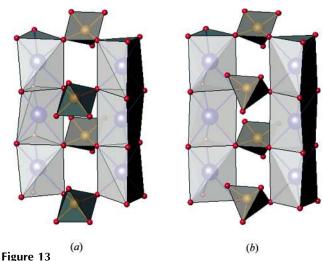


Figure 12 The structures of (a) finnemanite  $Pb_5(AsO_3)_4Cl$ , (b) mimetite  $Pb_5(AsO_4)_4Cl$  and (c)  $Ba_5(ReO_5)_4Cl$ . The figures show the progressive insertion of oxygen, the conversion of  $AsO_3$  to  $AsO_4$  and the  $ReO_5$  coordination. In finnemanite the Pb(1) ions are drawn as they occupy half-trigonal prisms (with three capping O atoms at a greater distance). Stereochemically active lone-pairs probably play a key role in stabilizing this structure.

 $P6_3cm$  branches. However, for the latter, the chemistries studied remain sparse, and it is possible that new representatives with stoichiometry  $[A(1)_2][A(2)_3](BO_5)_3X$  will be discovered.

The derivation of apatites from a regular oxygen sublattice also proves valuable, as it includes the BO<sub>4</sub> tetrahedra that are the common recognizable units shown in most structure drawings. These ideas, first used by Povarennykh (1972) and developed by Alberius-Henning, Landa-Canovas et al. (1999), have been extended in this paper to include the  $A(1)O_6$ trigonal metaprisms as a key structural unit, along with the BO<sub>4</sub> tetrahedra, in order to portray apatite chemical series. This depiction provides a highly visual and quantitative measure of apatite distortion from a perfect hexagonal anion net. For apatites that contain one species of A cation, the metaprismatic twist angle  $\varphi$  varies linearly over a wide range of compositions and changes inversely with atomic radius. The analysis of  $\varphi$  can be used to rapidly detect structures that fall outside expected bounds or deviate from compositional trends. Naturally,  $\varphi$  is not in itself sufficient to dismiss a structure solution as deficient; however,  $\varphi$  can indicate possible misinterpretation, particularly with respect to symmetry. For apatites that contain mixed and ordered A(1)cations, the interpretation becomes more complex, as the height of the metaprism (along [001]) is also influenced by the requirement to satisfy A-O bond lengths. In work to be published for  $(Ca_{10-x}Pb_x)(VO_4)_6(O,F)_{2-\delta}$ , 0 < x < 9, apatites (Dong & White, 2003),  $\varphi$  has been used as a sensitive probe to detect disequilibrium. Furthermore, microdomains observed directly by HRTEM were only removed after  $\varphi$  stabilized.



The arrangement of  $ReO_5$  square pyramids in (a)  $Sr_5(ReO_5)_4Cl$ , where the directionality is antiparallel between some  $SrO_6$  metaprisms, and (b)  $Ba_5(ReO_5)_4Cl$ , where the arrangement is always parallel, *i.e.* with the apical oxygen always pointing the same way.

The structural relationship between apatite and nasonite is well established (Engel, 1972), and members of this polysomatic series are possibly more prevalent than is currently  $Pb_6Ca_{3,33}Mn_{0.67}(SiO_4)_2(Si_2O_7)_2$ recognized. Ganomalite (Carlson et al., 1997) and Pb<sub>40</sub>(Si<sub>2</sub>O<sub>7</sub>)<sub>6</sub>(Si<sub>4</sub>O<sub>13</sub>)<sub>3</sub> (Stemmermann, 1992) have been identified as intermediate members of this homologous series. However, the structural principle of ordered filling of tetrahedral sites can give rise to an indefinite number of structures. It is less clear whether lead is a critical component of these structures but all current examples are substantially plumbous. At the atomic level, these structures may be quite complex. In the one instance where single-crystal X-ray diffraction and HRTEM have been undertaken, Brès et al. (1987) observed on the same sample a clear discrepancy in crystallographic features, the diffraction results being consistent with well ordered  $P6_3/m$ , while direct imaging showed microdomains of lower symmetry.

The rich chemistry and structural diversity of apatites provides fertile ground for the synthesis of technology-relevant compounds. Novel materials will continue to be discovered and possibly exploited in such diverse applications as zinc-rich bioactive apatites that accelerate tissue regrowth, mercury-bearing compounds analogous to  $\rm Sr_5(VO_4)_3(Cu_{0.896}O_{0.95})$  for polishing industrial wastewaters and catalytic apatites for photovoltaic devices.

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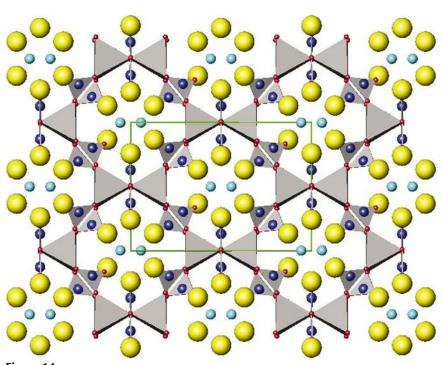


Figure 14  $Sr_5(BO_3)_3Br$  projected on (001). The apatite-like elements are obvious but are condensed such that some  $SrO_6$  trigonal prisms share edges rather than being connected through  $BO_3$  triangles.

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