# NANO EXPRESS

# Nanoscale Research Letters a SpringerOpen Journal

**Open Access** 

# Synthesis and characterizations of Ni-NiO nanoparticles on PDDA-modified graphene for oxygen reduction reaction

Tung-Yuan Yung<sup>1,2,3</sup>, Li-Ying Huang<sup>4</sup>, Tzu-Yi Chan<sup>5</sup>, Kuan-Syun Wang<sup>5</sup>, Ting-Yu Liu<sup>5\*</sup>, Po-Tuan Chen<sup>6</sup>, Chi-Yang Chao<sup>7</sup> and Ling-Kang Liu<sup>2\*</sup>

# Abstract

We are presenting our recent research results about the Ni-NiO nanoparticles on poly-(diallyldimethylammonium chloride)-modified graphene sheet (Ni-NiO/PDDA-G) nanocomposites prepared by the hydrothermal method at 90°C for 24 h. The Ni-NiO nanoparticles on PDDA-modified graphene sheets are measured by transmission electron microscopy (TEM), energy-dispersive X-ray spectroscopy (EDS), and selected area electron diffraction (SAED) pattern for exploring the structural evidence to apply in the electrochemical catalysts. The size of Ni-NiO nanoparticles is around 5 nm based on TEM observations. The X-ray diffraction (XRD) results show the Ni in the (012), (110), (110), (200), and (220) crystalline orientations, respectively. Moreover, the crystalline peaks of NiO are found in (111) and (220). The thermal gravimetric analysis (TGA) result represents the loading content of the Ni metal which is about 34.82 wt%. The electron spectroscopy for chemical analysis/X-ray photoelectron spectroscopy (ESCA/XPS) reveals the Ni<sup>0</sup> to Ni<sup>11</sup> ratio in metal phase. The electrochemical studies with Ni-NiO/PDDA-G in 0.5 M aqueous H<sub>2</sub>SO<sub>4</sub> were studied for oxygen reduction reaction (ORR).

Keywords: Graphene; PDDA; Ni; NiO; ORR; Fuel cells

# Background

Although platinum-based nanoparticles are thought as the best catalysts for oxygen reduction reaction (ORR), the increasing cost of Pt and the low abundance triggered scientists to develop non-noble metal catalysts for fuel cell applications [1-4]. The transition metal-based catalysts (based on Co, Ni, and Fe) are considered as a promising alternative due to their cheap cost and availability and have thus been studied for decades [5,6].

Catalysts for ORR of fuel cells (PEMFC and DMFC) have been the focus in recent years from the combination of Pt with varying metals to non-Pt-based metals [7-9]. Furthermore, carbon-supported nanocatalysts are also of great interest for scientists and engineers [7,10-14]. The ORR cathode is 6 or more orders of magnitude slower

Full list of author information is available at the end of the article

than the anode hydrogen oxidation reaction and thus limits performance, so almost all research and development focus on improving the cathode catalysts and electrodes [5]. The ORR catalysts are considered for mass production with the following factors: lower production of H<sub>2</sub>O<sub>2</sub> during the ORR and higher tolerance of the impurities (Cl- for instance). They must have the satisfied durability, and must be cost-effective. The three phenomena which lower the performance of fuel cells are kinetic losses, mass transport losses, and iR losses [5,7,15,16]. The ORR dominates the kinetic loss of fuel cells because the enhancement of the ORR activity would gain only 60 to 70 mV and kinetic losses are challenging. Moreover, the progress in catalyst development so far has achieved only modest cell voltage gains of tens of millivolts [5,17-19]. How to improve and enhance the catalyst electrochemical performances is the focus of scientists and engineers. Carbon-supported materials were introduced for fuel cell application. The supported materials would provide the surfaces for anchoring the catalysts and increasing the surface areas of the catalysts. Also,



© 2014 Yung et al.; licensee Springer. This is an Open Access article distributed under the terms of the Creative Commons Attribution License (http://creativecommons.org/licenses/by/4.0), which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly credited.

<sup>\*</sup> Correspondence: tyliu0322@gmail.com; liuu@chem.sinica.edu.tw
<sup>5</sup>Department of Materials Engineering, Ming Chi University of Technology,

New Taipei City 24301, Taiwan <sup>2</sup>Molecular Science and Technology, Taiwan International Graduate Program/

the supported material provides higher volume-to-mass ratio to make a good dispersive paste for electrode assembly. The size of Pt nanoparticles for the commercial Pt on carbon (Pt/C) is about 2 to 5 nm [5,20]. In addition to that, the Pt-based bimetallic system is interesting for ORR application, and the Pt<sub>3</sub>Ni bimetallic electrocatalyst on carbon support has also been known to serve as a catalyst for ORR [21]. Herein, we introduced additionally poly-(diallyldimethylammonium chloride) (PDDA) which further assists in the formation of a layer-to-layer structure for graphene surface modification (PDDA-G) on carbonsupported materials [22-25]. The synthesis of Ni-NiO nanoparticles on PDDA-G is done using the hydrothermal method. The results on hydrothermal synthesis of the Ni-NiO nanoparticles on PDDA-modified graphene for ORR application would be presented in this study.

# Methods

Graphene was prepared from graphite using the microwave synthesis method. Graphite (0.1 g; Sigma-Aldrich Co., St. Louis, MO, USA) was put into a 25-mL roundbottomed flask, and the flask was treated using a CEM Discover Du7046 microwave synthesis system (CEM, Matthews, NC, USA) with a power output of 20 W at 80°C for 10 s. The graphene was produced in 2 to 5 s with a sound of a bomb. Fifty milliliters of 3.5 wt% aqueous PDDA (Sigma-Aldrich) and 100 mg of graphene prepared by the method as mentioned were put into a 100-mL flask and then heated at 90°C for 4 h with a flux apparatus.

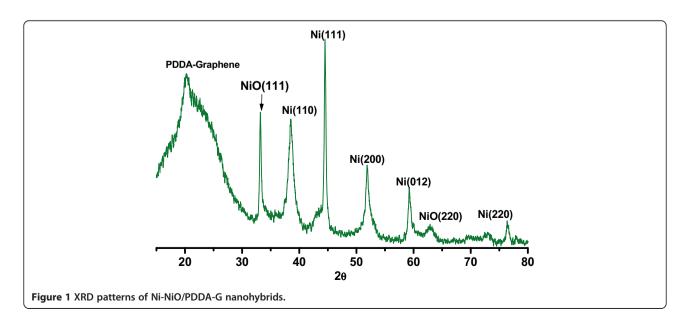
About 0.45 mmol Ni(NO<sub>3</sub>)<sub>2</sub>  $\cdot$  2.5H<sub>2</sub>O was added into the above mentioned PDDA-G solution, followed by the addition of hydrazine hydrate of about 20 mmol. Then, the mixed solution was transferred into a Teflon-lined autoclave and heated at 90°C for 24 h. The mixture was centrifuged and washed for three times prior to drying at 90°C to produce the Ni-NiO nanoparticles on the PDDA-modified graphene (Ni-NiO/PDDA-G).

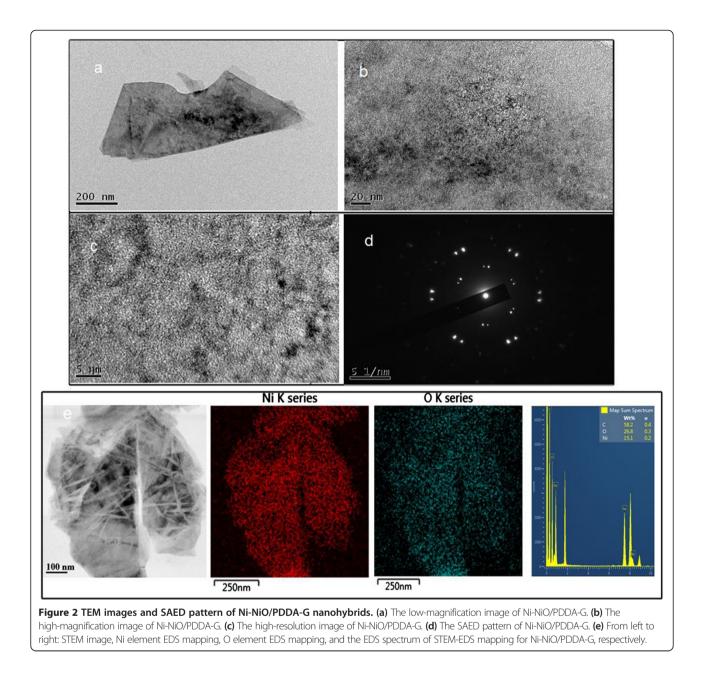
The crystalline structure of Ni-NiO/PDDA-G was examined by X-ray diffraction (XRD) using a Bruker D8 diffractometer (Bruker AXS, Karlsruhe, Germany) equipped with CuK $\alpha$  X-ray source. The chemical environments of Ni-NiO/PDDA-G were analyzed by electron spectroscopy for chemical analysis/X-ray photoelectron spectroscopy (ESCA/XPS) using a Thermo VG ESCAlab 250 (Thermo Fisher Scientific, Waltham, MA, USA) equipped with a dual-anode (MgK $\alpha$ /AlK $\alpha$ ) X-ray source.

The microstructures of Ni-NiO/PDDA-G were investigated with the high-resolution microstructural images produced using the JOEL FEM 2100F (JEOL Ltd., Akishima, Tokyo, Japan) equipped with an Oxford energy-dispersive X-ray spectroscope (EDS) for element analysis.

Thermal gravimetric analysis (TGA) for nanoparticle loading was carried out using a PerkinElmer Pyris 1 instrument (PerkinElmer, Waltham, MA, USA) and by applying a heating rate of 10°C/min from room temperature to 800°C in an oxygen-purged environment.

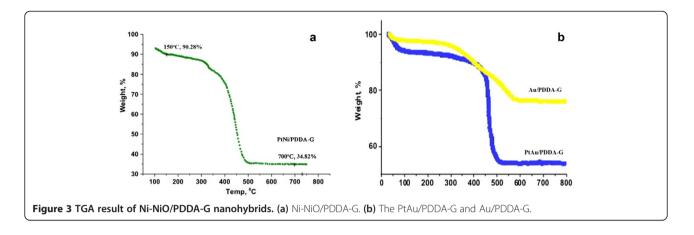
The ORR study was examined using an Autolab potentiostat/galvanostat PGSTAT30 (Eco Chemie BV, Utrecht, The Netherlands). The reference electrode is Ag/AgCl (ALS Co. Ltd., Tokyo, Japan), and the counter electrode is a 0.5 mm × 10 cm platinum wire. The working electrode is the glassy carbon whose surface is deposited 5.24  $\mu$ g/cm<sup>2</sup> of Ni-NiO/PDDA-G. Cyclic voltammetry was used to investigate the 0.5 M aqueous H<sub>2</sub>SO<sub>4</sub> and O<sub>2</sub>-saturated 0.5 M aqueous H<sub>2</sub>SO<sub>4</sub> with a scanning rate of 50 mV/s. The electrochemical impedance spectroscopy (EIS) is also used as a test with an amplitude of 10 mV from 1 to 100 mHz.





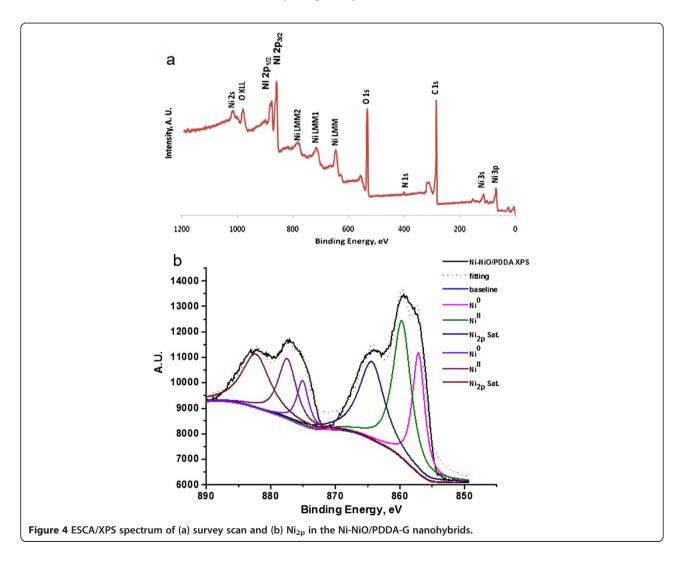
# **Results and discussion**

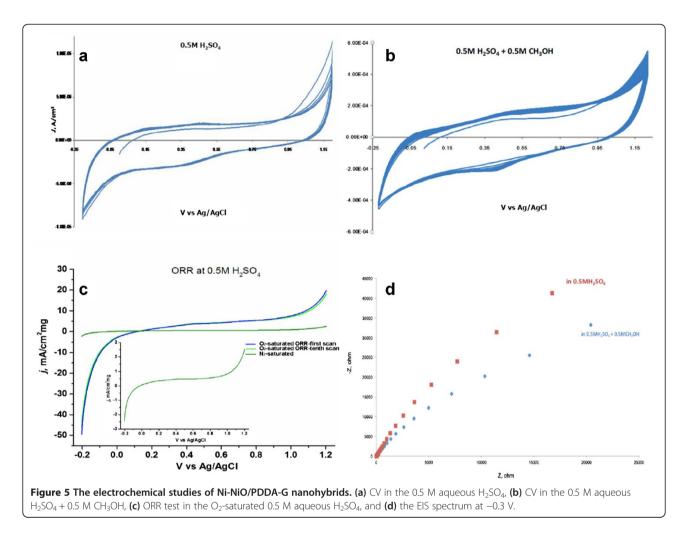
The crystallization of Ni-NiO/PDDA-G was examined by XRD as shown in Figure 1. The peaks of the (002) plane in the PDDA-modified graphene was shifted from 20.5° to 22°, which revealed the change in the layer-tolayer distance of graphene due to incorporation of PDDA [21]. The hydrothermal method for synthesis of the Ni-NiO alloy nanoparticles was one-pot synthesis with a mixture of PDDA-G, Ni precursors, and hydrazine hydrates at 90°C for 24 h. The XRD result of Ni-NiO/ PDDA-G indicated peaks assigned as Ni (111), Ni (200), Ni (012), Ni (222), NiO (111), NiO (012) and NiO (220), respectively [26,27]. The microstructures of the Ni-NiO/PDDA-G nanohybrids are shown in Figure 2. The PDDA-modified graphene is a layer-by-layer structure, shown in Figure 2a. The Ni-NiO nanoparticles are anchoring between the layers and the surfaces of PDDA-G. Figure 2b,c shows the high-resolution TEM images for Ni-NiO/PDDA-G. The different contrasts are shown: Ni (dark) and NiO (bright) nanoparticles. Both particle sizes are around 2 to 5 nm. Selected area electron diffraction (SAED) patterns for the Ni and NiO are shown in Figure 2d. The brighter and bigger spots are for the Ni nanoparticle electron diffraction patterns. The results of EDS mapping from the STEM method are shown in Figure 2e. The Ni and O



elements are colored red and blue to show the contribution for Ni-NiO nanoparticles on PDDA-G. The more condensed Ni element mapping is showing that the Ni-NiO nanoparticles exist. By EDS, the semi-quantified element ratios are Ni 15.1% and O 26.8% by weight (Ni 3.83% and O 24.7% by mole). The one-step synthesis with hydrothermal method is perfect for the synthesis process for the narrow size distribution of nanoparticles.

TGA shows that the loading content of the Ni-NiO nanoparticles is about 34.84 wt% on the PDDA-G surfaces. The





TGA result is shown in the Figure 3a. For comparison with the other metal loading contents by hydrothermal method, the Au/PDDA-G and PtAu/PDDA-G are observed in the Figure 3b. The same precursor loading (approximately 0.456 mmol) with the same batch PDDA-G was applied in the one-pot synthesis method. The nickel reduction rate is obviously lower than the reduction rate of gold and platinum by the metal loading amounts, which is in the order of 34.82, 58.2, and 74.1 wt%.

PDDA was used to modify the surface of graphene, and then the Ni-NiO nanoparticles could be embedded on the PDDA-G surface. The change of functional groups in the Ni-NiO/PDDA-G would be evaluated by ESCA/XPS in Figure 4a. The C<sub>1s</sub> binding energy of the C-C sp<sup>2</sup> (284.6 eV, 72.4%) and that of epoxy group (286.7 eV, 27.6%) are shown, respectively. The binding energy of O<sub>1s</sub> was fitted as 531.2 eV (C-O-Ni, 18.9%), 532.1 eV (C = O/O-Ni, 26.4%), 533.5 eV (C-OH/C-O-C, 30.0%), and 535.0 eV (COOH, 24.7), respectively. The N<sub>1s</sub> spectrum was fitted as 399.4 eV (binding PDDA, 54.4%) and 400.6 eV (free PDDA, 45.5%). Furthermore, the curve fitting of Ni<sub>2p</sub> spectrum results in Figure 4b shows Ni<sup>0</sup> (the binding energies for 857.1 and 875.0 eV) and Ni<sup>II</sup> (the binding energies for 859.7 and 877.5 eV) with the fitting ratio of 41.7% and 52.3%, respectively.

The electrochemical investigation of Ni-NiO/PDDA-G was applied in the 0.5 M aqueous H<sub>2</sub>SO<sub>4</sub> (shown in Figure 5a), 0.5 M aqueous  $H_2SO_4 + 0.5$  M CH<sub>3</sub>OH (shown in Figure 5b), and the  $O_2$ -saturated 0.5 M aqueous  $H_2SO_4$ (shown in Figure 5c). Figure 5c shows no significant difference, as evidenced by the blue line denoting the O2-saturated ORR first scan and the green line denoting the tenth scan. The inset in Figure 5c is the ORR test in the N2-saturated 0.5 M aqueous H<sub>2</sub>SO<sub>4</sub>. The O<sub>2</sub>-saturated ORR test current density at the -0.2 to 0.2 V vs. Ag/AgCl is about 25 times than that of the N<sub>2</sub>-saturated ORR test of Ni-NiO/PDDA-G. Furthermore, the O<sub>2</sub>-saturated ORR test current density at the 1.0 to 1.2 V vs. Ag/AgCl is about 5 times than that of the N2-saturated ORR test of Ni-NiO/PDDA-G. The electrochemical impedance spectroscopy result for testing the 0.5 M aqueous  $H_2SO_4$  and 0.5 M aqueous  $H_2SO_4 + 0.5$  M  $CH_3OH$  is shown in Figure 5d. The semicircle curve of Ni-NiO/ PDDA-G in the 0.5 M aqueous  $H_2SO_4$  is higher than that in the 0.5 M aqueous  $H_2SO_4 + 0.5$  M  $CH_3OH$ , showing the higher chemical reaction ability. Thus, the Ni-NiO/PDDA-G is more suitable for ORR than for the methanol oxygen reaction.

## Conclusions

We have successfully synthesized the Ni-NiO/PDDA-G nanohybrids, and the size of Ni-NiO nanoparticles was about 2 to 5 nm. The morphologies and chemical composition of Ni-NiO/PDDA-G were evaluated by TGA, XRD, TEM, and ESCA/XPS. The results show the phase of the Ni-NiO/PDDA-G, and the loading content of Ni-NiO is about 35 wt%. The CV and EIS results of Ni-NiO/PDDA-G in 0.5 M aqueous  $H_2SO_4$  are better than those in 0.5 M aqueous  $H_2SO_4 + 0.5$  M CH<sub>3</sub>OH. Therefore, Ni-NiO/PDDA-G in 0.5 M aqueous  $H_2SO_4$  is more suitable as ORR electrocatalyst and could be a candidate of for cathode electrocatalyst of fuel cells.

### **Competing interests**

The authors declare that they have no competing interests.

### Authors' contributions

TYY, LYH, and TYL conceived and designed the experiments. PTC, LYH, TYC, and KSW performed the experiments. TYY, LYH, TYC, CYC, and KSW contributed ideas and material analyses. TYY, TYL, and LKL wrote the manuscript. This work was performed under the supervision of LKL. All authors read and approved the final manuscript.

### Authors' information

TYY is an assistant engineer at the Institute of Nuclear Energy Research. LYH is a postdoctoral fellow at National Taiwan University of Science and Technology. PTC is a postdoctoral fellow at National Taiwan University. CYC is an associate professor at National Taiwan University. TYC and KSW are undergraduate students at Ming Chi University of Technology. TYL holds an assistant professor position at Ming Chi University of Technology. LKL is a research fellow at Academia Sinica and an adjunct professor at National Taiwan University.

### Acknowledgements

This work was financially supported by the National Science Council of Taiwan (NSC 102-2321-B-131-001) and partially supported by Academia Sinica.

### Author details

<sup>1</sup>Department of Physics, National Central University, Jhongli, Taoyuan 320, Taiwan. <sup>2</sup>Molecular Science and Technology, Taiwan International Graduate Program/Institute of Chemistry, Academia Sinica, Taipei 115, Taiwan. <sup>3</sup>Nuclear Fuels and Materials Division, Institute of Nuclear Energy Research, Lontan, Taoyuan 325, Taiwan. <sup>4</sup>Department of Materials Science and Engineering, National Taiwan University of Science and Technology, Taipei 106, Taiwan. <sup>5</sup>Department of Materials Engineering, Ming Chi University of Technology, New Taipei City 24301, Taiwan. <sup>6</sup>Center for Condensed Mater Sciences, National Taiwan University, Taipei 106, Taiwan.

### Received: 18 July 2014 Accepted: 18 August 2014 Published: 28 August 2014

### References

- 1. Li C, Yamauchi Y: Facile solution synthesis of Ag@Pt core-shell nanoparticles with dendritic Pt shells. *Phsy Chem Chem Phys* 2013, **5**:3490–3496.
- Ataee-Esfahani H, Imura M, Yamauchi Y: All-metal mesoporous nanocolloids: solution-phase synthesis of core-shell Pd@Pt nanoparticles with a designed concave surface. Angew Chem Int Ed 2013, 52:13611–13615.
- 3. Li C, Sato T, Yamauchi Y: Electrochemical synthesis of one-dimensional mesoporous Pt nanorods using the assembly of surfactant micelles in confined space. *Angew Chem Int Ed* 2013, **52**:8050–8053.

- Yamauchi Y: Field-induced alignment controls of one-dimensional mesochannels in mesoporous materials. J Ceram Soc Jpn 2013, 121:831–840.
- Debe MK: Electrocatalyst approaches and challenges for automotive fuel cells. Nature 2012, 486:43–51.
- Gasteiger HA, Kocha SS, Sompalli S, Wagner FT: Activity benchmarks and requirements for Pt, Pt-alloys, and non-Pt oxygen reduction catalysts for PEMFCs. *Appl Catal B* 2005, 56:9–35.
- Li W, Haldar P: Highly active carbon supported core-shell PtNi@Pt nanoparticles for oxygen reduction reaction. *Electrochem Solid State Lett* 2010, 13:B47–B49.
- Xin L, Zhang Z, Wang Z, Qi J, Li W: Carbon supported Ag nanoparticles as high performance cathode catalyst for H<sub>2</sub>/O<sub>2</sub> anion exchange membrane fuel cell. Front Chem 2013, 1:1–6.
- Toda T, Igarashi H, Uchida H, Watanabe M: Enhancement of the electroreduction of oxygen on Pt alloys with Fe, Ni, and Co. J Electrochem Soc 1999, 146:3750–3756.
- Xu C, Pietrasz P, Yang J, Soltis R, Sun K, Sulek M, Novak R: Pt-based ORR catalyst on carbon-supported amorphous niobium oxide support. ECS Trans 2013, 58:1779–1788.
- Neergat M, Gunasekar V, Rahul R: Carbon-supported Pd-Fe electrocatalysts for oxygen reduction reaction (ORR) and their methanol tolerance. *J Electranal Chem* 2011, 658:25–32.
- Liu CW, Chen HS, Lai CM, Tsai LD, Wang KW: Promotion of oxygen reduction reaction durability of carbon-supported PtAu catalysts by surface segregation and TiO<sub>2</sub> addition. ACS Appl Mater Interfaces 2014, 6:1589–1594.
- Hwang SC, Yoo SJ, Shin J, Cho YH, Jang JH, Cho E, Sung YE, Nam SW, Lim TH, Lee SC, Kim SK: Supported core@shell electrocatalysts for fuel cells: close encounter with reality. *Sci Rep* 2013, 3:1309.
- Ahmed J, Yuan Y, Zhou L, Kim S: Carbon supported cobalt oxide nanoparticles-iron phthalocyanine as alternative cathode catalyst for oxygen reduction in microbial fuel cells. J Power Sources 2012, 208:170–175.
- Wang C, Li D, Chi M, Pearson J, Rankin RB, Greeley J, Duan Z, Wang G, van der Vliet D, More KL, Markovic NM, Stamenkovic VR: Rational development of ternary alloy electrocatalysts. J Phys Chem Lett 2012, 3:1668–1673.
- Greeley J, Stephenes IE, Bondarenko AS, Johansson TP, Hansen HA, Jaramillo TF, Rossmeisl J, Chorkendorff I, Nørskov JK: Alloy of platinum and early transition metals as oxygen reduction electrocatalysts. *Nat Chem* 2009, 1:552–556.
- Sepa DB, Vojnovic MV, Damjanovic A: Reaction intermediates as a controlling factor in the kinetics and mechanism of oxygen reduction at platinum electrodes. *Electrochim Acta* 1981, 26:781–793.
- Garsany Y, Barurina OA, Swider-Lyons KE, Kocha SS: Experimental methods for quantifying the activity of platinum electrocatalysts for the oxygen reduction reaction. *Anal Chem* 2010, 82:6321–6328.
- Guo S, Sun S: FePt nanoparticles assembled on graphene as enhanced catalyst for oxygen reduction reaction. J Am Chem Soc 2012, 134:2492–2495.
- 20. Yung TY, Lee JY, Liu LK: Nanocomposite for methanol: synthesis and characterization of cubic Pt nanoparticles on graphene sheets. *Sci Technol Adv Mater* 2013, **14**:035001.
- Wu J, Zhang J, Peng Z, Yang S, Wangner FT, Yang H: Truncated octahedral Pt<sub>3</sub>Ni oxygen reduction reaction electrocatalysts. J Am Chem Soc 2010, 132:4984–4985.
- Wang Y, Wang S, Xiao M, Han D, Hickner M, Meng Y: Layer-by-layer self-assembly of PDDA/PSS-SPFEK composite membrane with low vanadium permeability for vanadium redox flow battery. *RSC Adv* 2013, 35:15467–15474.
- Wang S, Wang X, Jiang SP: Self-assembly of mixed Pt and Au nanoparticles on PDDA-functionalized graphene as effective electrocatalysts for formic acid oxidation fuel cells. *Phys Chem Chem Phys* 2011, 13:6883–6891.
- 24. Wang S, Yu D, Dai L, Chang JB: Polyelectrolyte-functionalized graphene as metal-free electrocatalysts for oxygen reduction. ACS Nano 2011, 5:6202–6209.
- Yuan L, He Y: Effect of surface charge of PDDA-protected gold nanoparticles on the specificity and efficiency of DNA polymerase chain reaction. *Analyst* 2012, 138:539–545.
- Zhu LP, Liao GH, Xiao HM, Wang JF, Fu SY: Self-assembled 3D flower-like hierarchical β-Ni(OH)<sub>2</sub> hollow architectures and their in situ thermal conversion to NiO. Nanoscale Res Lett 2009, 4:550–557.
- 27. Wang H, Kou X, Zhang J, Li J: Large scale synthesis and characterization of Ni nanoparticles by solution reaction method. *Bull Mater Sci* 2008, **31**:97–100.

### doi:10.1186/1556-276X-9-444

Cite this article as: Yung *et al.*: Synthesis and characterizations of Ni-NiO nanoparticles on PDDA-modified graphene for oxygen reduction reaction. *Nanoscale Research Letters* 2014 9:444.